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¹ Kinetics and coupled dynamics of dewetting and chemical reaction

- $_{2}$ in Si/SiO $_{2}$ /Si system
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- 6 (Dated: June 30, 2020)

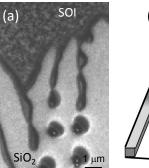
Abstract

We report on the observation of a coupling between the dewetting of a Si layer on SiO₂ induced by surface/interface energy minimization and the etching between both materials due to the Si+SiO₂ \longrightarrow 2 SiO_g[↑] reaction. In the limit of a thin SiO₂ layer (\le 10 nm) sandwiched between a Si layer and a Si handle wafer, the front of Si dewetting and the front of SiO₂ etching coexist in a narrow region. The interplay between both phenomena gives rise to specific morphologies. We show that extended Si fingers formed by dewetting are stabilized with respect to Rayleigh Plateau type instability over tenth of microns thanks to a localized etching of the SiO₂ layer. The break up of this structure occurs abruptly by an unzipping process combining dewetting and etching phenomena. We also put in evidence that Si rings are created with a thin SiO₂ layer in the center. These processes are thermally activated with an activation energy of 2.4±0.5 eV and 4.0±0.5 eV respectively for dewetting and the etching reaction. All these results highlight the respective roles of wetting and etching in Si/SiO₂/Si system dynamics and could be a stepping stone for the development of advanced processes based on Silicon-On-Insulator technology.

7 I. INTRODUCTION

Wetting of liquids on solids in reactive systems is of great importance with many industrial 9 applications such as soldering or adhesion [1]. The kinetics of reactive wetting is affected 10 by a number of factors like interfacial reactions, diffusion of constituents or dissolution of 11 the substrate [2 and 3]. Under reaction the contact line dynamics can lead to complex 12 behaviours such as oscillatory phenomenon of spreading and contraction [4–7], formation 13 of reaction rings [8] or even droplet motion [9–11]. For example, the reaction-mediated 14 spontaneous drift of millimeter-scale solid-state camphor particles on the surface of water 15 has been known for several centuries [12]. In solid systems only scarce studies have explored 16 the coupling between chemical reactions and wetting dynamics. For instance in metallic 17 systems the alloying driving force can induce the self-propelled motion of solid Sn islands 18 onto a Cu surface [13]. In the context of metal/oxide or semiconductor/oxide systems, solid 19 state wetting properties may also be affected by interfacial redox reactions. For instance in ₂₀ the Si/SiO₂ couple, that plays a key role in the perspective of future devices based on Silicon-21 On-Insulator (SOI) technology [14–17], wetting and reactivity properties of the interface have $_{22}$ been largely studied. Under moderate annealing ($\sim 1100\text{-}1300~\mathrm{K}$) SOI dewetting occurs 23 by edge retraction (see Fig.1 and [18–24]), the expelled silicon accumulates in a receding 24 rim that exhibits a capillary instability giving birth to elongated Si structures (called Si 25 fingers). These fingers are in turn destabilized and periodically form 3D Si islands by 26 a beading mechanism similar to a Rayleigh-Plateau instability. The driving force for this 27 dewetting process arises from the surface/interface free energy gain. In solids mass transport is mediated by surface diffusion [25–27] whereas hydrodynamics dominates in liquid films [28–31]. This makes that the contact line with the substrate is particularly sensitive to the 30 effect of the disjoining pressure in solid thin films whereas surface diffusion is much less 31 affected [32]. Considering high temperature regimes ($\sim 1300~\mathrm{K}$) Si and SiO₂ solids are also $_{32}$ known to react and form volatile SiO_q^{\uparrow} molecules. This is perfectly illustrated when a SOI 33 film is completely dewetted at 1000 K, then annealed at 1330 K: the reaction leads jointly 34 to the consumption of the 3D Si islands and to the etching of nanopits in the SiO₂ layer [33] 35 and 34 .

In this article we analyze the combined effects of etching reaction and dewetting on SOI films in the high temperature regime. Their interplay gives complex front dynamics: (i)



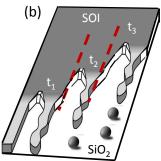


FIG. 1. (a) Low energy electron microscopy image of the dewetting of a SOI film (\sim 20 nm thick) at 1160 K. (b) sketch of the dewetting mechanism: a Si finger destabilizes by a lateral undulation (time t_1) that leads to beading and formation of a 3D Si island at its extremity (time $t_2 \sim t_1+120$ s). This process repeats periodically (time $t_3 \sim t_1+240$ s).

³⁸ formation of extended Si fingers stabilized by lateral etching, (ii) Si fingers break up by an ³⁹ unzipping mechanism and (iii) formation of Si rings with a thin SiO₂ layer in the center. ⁴⁰ The etching and dewetting fronts dynamics are studied experimentally *in situ* and in real ⁴¹ time by low energy electron microscopy (LEEM) and *ex situ* AFM measurements.

42 II. METHODS

The SOI wafers are originally fabricated at SOITEC (Bernin, France). They are cleaned and prepared in Ultra High Vacuum (UHV) conditions as described in Ref. [35]. The studied thickness is about 30 nm for the Si layer and 8 nm for the buried oxide layer (BOX) of SiO₂. Such thin structures are highly sensitive to Si layer dewetting and etching reaction between Si/SiO₂ under thermal annealing. In the temperature range ~ 1300 – 1400 K, a front of dewetting of the Si layer forms spontaneously onto a defect at the surface/interface and is followed by an etching reaction front. Their motion and morphology are recorded *in situ* and in real time (1 s per image) by low energy electron microscopy (LEEM) in the tilted bright field mode (electron energy: 4 eV). The characterization of the coupling between the etching and dewetting fronts dynamics is completed by *ex situ* analysis of the surface morphology by atomic force microscopy (AFM, PSIA-XE 100).

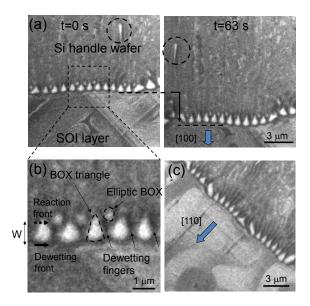


FIG. 2. (a) Sequence of two LEEM images of a thin SOI layer (30 nm) onto a BOX layer (8 nm) at 1370 K. The SOI dewetting front and the etching front Si/SiO₂ advance in the [100] direction (See Supplemental Material S1 at [URL will be inserted by publisher] for the complete movie). The Si handle wafer is on top of the image and the SOI layer is at the bottom. Dotted circles: SiO₂ layers moving opposite to the dewetting/etching front. The linear defects in the SOI layer (bottom right part of the image) occur due to remaining lateral temperature gradients during annealing. They have no influence on the fronts dynamics. (b) Close view of the area of coexistence of the dewetting front (black arrow) and reaction front (dashed arrow): The exposed SiO₂ layer has a bright contrast with a triangular or elliptical shape. Si fingers are formed in between triangular SiO₂ layers. (c) Same as (a) for an advancing front in the \(\lambda 110 \rangle \) direction.

54 III. RESULTS: INTERPLAY BETWEEN ETCHING REACTION AND DEWET55 TING FRONTS

56 A. Kinetics of fronts propagation

Figure 2(a) shows two LEEM images of the SOI (30 nm)/BOX (8 nm)/Si handle wafer during solid phase etching and dewetting at 1370 K. The SOI layer exhibits a black/white contrast coming from the $(1\times2)/(2\times1)$ surface reconstructions. The SOI dewetting front appears as an undulating grey line (fig. 2(b)) that separates the SOI layer and the underneath SiO₂ BOX layer (bright contrast). The fronts recede in a steady state at a constant velocity

of 53 ± 5 nm.s⁻¹. If the front is aligned along the [100] direction very regular structures (1.5 \pm 0.1 μ m period) occur whereas slightly more disordered ones can be observed when the front recedes in the [110] direction (Fig. 2(c)). A similar observation is obtained for Si(001) thin layer dewetting onto SiO₂ at lower temperature (1200 K) when etching reaction is neg-66 ligible. The [100] front forms regular fingers by front edge instability as described by [36]. In the case of a [110] front, as there are two equivalent directions for the fingering instability (100] and [010]), the competition between both degrades the long range order [23 and 24].

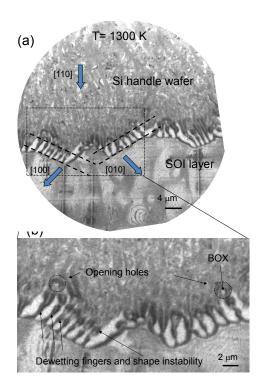


FIG. 3. (a) LEEM image of a SOI layer (30 nm) onto a BOX layer (8 nm) at 1300 K (See Supplemental Material S2 at [URL will be inserted by publisher] for the complete movie). The SOI dewetting front and the etching front Si/SiO₂ (middle) advance in the [110]. (b) Close view of the fronts: Si fingers onto SiO₂ layer are visible (black arrows) as well as circular holes in the Si layer (black dotted circles) exposing SiO₂.

As the SOI dewetting fronts recede (*via* fingers formation), the underlying SiO₂ BOX ⁷⁰ layer is exposed to vacuum (Fig. 2(b)). This free BOX layer shows triangular structures, ⁷¹ the base pointing towards the SOI layer and the tip pointing in the direction of the Si handle ⁷² wafer. We can infer that the edges of the triangles correspond to the reaction front between ⁷³ the remaining BOX layer and the Si handle wafer. As the front recedes in a steady state, the

 $_{74}$ decay of the BOX occurs by a constant etching reaction rate (Si+SiO $_2$ $\longrightarrow 2$ SiO $_g^{\uparrow})$ localized $_{75}$ at the $\mathrm{Si/SiO_2/vacuum}$ triple-line. To evaluate the activation energies of SOI dewetting ₇₆ E_d and etching reaction E_r we have studied the effect of the sample temperature on both 77 kinetics (Fig. 3(a)-(b)). As the temperature decreases from 1370 K to 1300 K the fronts 78 velocity v_f decreases from 53±5 nm.s⁻¹ to 18±3 nm.s⁻¹, the extension of the free BOX ₇₉ layer W increases from $0.9\pm0.1~\mu\mathrm{m}$ to $2.3\pm0.5~\mu\mathrm{m}$ and the disappearance time t_d of the free 80 BOX layer increases from 20 ± 1 s to 125 ± 20 s. The SOI front velocity v_f is only driven by 81 SOI dewetting kinetics and can be written as $v_f \sim e^{-\frac{E_d}{kT}}$ assuming it is thermally activated. 82 From the ratio of the velocities v_f at 1370 K and 1300 K, the dewetting activation energy 83 is estimated as $E_d=2.4\pm0.5$ eV that is in perfect adequation with previous SOI dewetting 84 studies [23] that show that Si surface diffusion is the dominant mechanism of mass transport ₈₅ mediating the dewetting process. The etching reaction activation energy E_r is not related 86 to the reaction front velocity. Indeed the reaction occurs at the $Si/SiO_2/vacuum$ triple-87 line and thus starts when the BOX is exposed to vacuum. Therefore the etching reaction 88 activation energy has to be evaluated from the disappearance time $t_d \sim e^{\frac{E_T}{kT}}$ of the free BOX 89 layer. Measurements of t_d at 1370 K and 1300 K provide an estimate of $E_r = 4.0 \pm 0.5$ 90 eV. This is also in adequation with the activation energy reported by [33] from temperature 91 dependent measurements of Si island decay kinetics onto SiO₂. At last we have also measured $_{92}$ the width W of the exposed BOX layer with respect temperature. It is related both to 93 dewetting and etching reaction activation energies as $W = v_f \times t_d \sim e^{\frac{E_r - E_d}{kT}}$. We measure 94 that $E_r-E_d=2.1\pm0.6$ eV. The large difference between the activation energies of Si 95 dewetting and etching reaction makes that the width W of the free BOX layer decreases 96 strongly with temperature as observed experimentally. At last let us note that despite the 97 difference between both activation energies, the dewetting and etching fronts move at the 98 same velocity (i.e. at the Si layer dewetting velocity, see supplementary material S1). This 99 is due to the fact that the etching reaction occurs only when the BOX layer is exposed to vacuum, i.e. after the dewetting front passage. Then it takes a time t_d for the BOX layer to be completely etched which can be seen as a delay time for the etching front to progress behind the dewetting front.

This evaluation of the activation energies can also be completed with the quantitative analysis of the rate of etching reaction and dewetting front velocity. From the linear decrease to find the BOX layer with time (triangular shape) the reaction rate $\nu_{reac} = \frac{dN}{dt}$ at the Si/SiO₂

we calculate therefore $\nu_{reac} \approx 3.2 \times 10^3 \text{ s}^{-1}$ per SiO₂ entity at the contact line at 1370 K and $\nu_{reac} \approx 5 \times 10^2 \text{ s}^{-1}$ per SiO₂ entity at 1300 K (oxide layer thickness: 8 nm). This result compares to previous report [33] on the reaction rate of Si islands onto SiO₂ (3 × 10² s⁻¹ per atom) in a slightly lower temperature regime. Concerning the dewetting velocity of SOI, Cheynis et al. [24] have shown that in the temperature range [1000 K - 1200 K] $v_f \simeq \frac{5.3 \times 10^{12}}{h^3 kT} e^{-2.0/kT}$ nm.s⁻¹ where h is the SOI film height expressed in nm and kT is given in eV. Assuming this expression for a 30 nm thick film at 1370 K and 1300 K we obtain respectively v_f =74 nm.s⁻¹ and v_f =31 nm.s⁻¹ which gives the correct order of magnitude of the kinetics, the overestimation arising probably from a slightly too low activation energy in the modeling.

Beyond the description of the energetics of SiO₂ etching, figures 2(b) and 3(b) show also very specific structures that highlight a complex interplay between the dynamics of dewetting and etching. This interplay occurs because the timescale for the formation of fingers and 3D islands by dewetting is comparable to the timescale for the consumption of the BOX layer. We will focus on three main structures: (i) SiO₂ BOX layer of elliptic shape located in between two free BOX triangles and separated by Si (dotted ellipse in fig. 2(b), at 1370 K). (ii) Elongated free BOX structures crossing the surface from time to time in the direction opposite to the dewetting front (dotted circles in fig. 2(a)) with a large velocity (> 100 nm.s⁻¹). (iii) SOI islands with a ring shape and exposing a circular BOX layer in the middle (see fig. 3(b) at 1300 K). To understand all these structures we will describe in the following part the main stages of their formation and propose different scenarios with coupled etching and dewetting phenomena.

B. Coupled dynamics of dewetting and etching

129

a. Steady state elliptic shape BOX structure Figures 2(a)-(b) show LEEM im131 ages of a dewetting/reaction front moving along the [100] direction. The regularly spaced
132 triangular BOX structures separate Si fingers that are short and perpendicular to the dewet133 ting front. Some Si fingers are prolonged by BOX elliptic structures (dotted ellipse in fig.
134 2(b)) that are stable during the steady motion of the fronts (See complete movie S1 at [URL
135 will be inserted by publisher]).

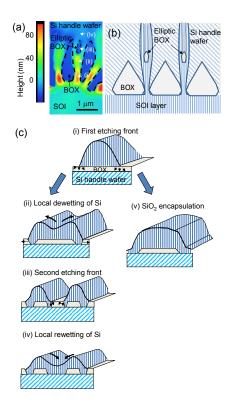


FIG. 4. (a) AFM image of the surface morphology of the front. Si fingers from the SOI layer separated by triangular shape BOX layers are visible. (i) to (iv) address the different steps of Si finger evolution (see scenario in (c)). (b) Scheme of the surface morphology considering elliptic holes (top view). (c) Scenario in four steps of the Si finger evolution: (i) Si finger formed by SOI layer dewetting and simultaneous SiO₂ BOX etching at the contact line. (ii) Complete lateral etching of the SiO₂ layer and contact between the Si finger (from SOI) and the Si handle wafer. Simultaneous lateral progression of the reaction front and Si mass shedding from the center of the finger. (iii) Splitting of the finger into two fingers and occurrence of new etching fronts with the free BOX. (iv) Complete etching of the inner SiO₂ BOX layer and rewetting of the SOI layer onto the Si handle wafer. (i)-(iv) are also sketched on AFM image in (a). (v) Scenario of direct BOX layer encapsulation (without finger splitting).

To analyze the morphology of these fronts combining dewetting and etching, figure 4(a) shows a typical AFM image. We can clearly distinguish the SOI layer (turquoise) and the Si fingers (yellow and red). The AFM image shows also the SiO₂ BOX layer (blue); it appears as extremely flat areas located in between Si fingers close to the SOI dewetting front. They have the same shape as those observed by LEEM (triangular structures in 2(c)). We can

141 also see from AFM that some Si fingers are unstable and split laterally into two branches 142 exposing a small SiO₂ BOX layer (\sim 500 nm) in between (elliptic shape) as the one measured 143 by LEEM. From the combination of both methods, a scenario of the process is proposed (fig. 4(b)-(c)) based on the combined effects of SiO₂ BOX etching and wetting/dewetting of the SOI layer. The scenario involves four steps labelled (i) to (iv) that can be observed also by AFM (fig. 4(a)) as time lag is related to the spatial position with respect to the front. 147 (i) First, a Si finger is formed by SOI dewetting on the SiO₂ BOX layer. This process is well known and arises from surface/interface energy minimization that kinetically destabilizes the receding front [37 and 38]. (ii) The Si finger formation results in the occurrence of a 150 triple line between the freshly exposed SiO₂ BOX layer, the Si finger and vacuum. This 151 contact line is very active for SiO₂ etching compared to SiO₂ dissolution through the SOI 152 layer. Indeed, as shown by [39], the limiting mechanism for SiO₂ etching is the diffusion 153 of dissolved O atoms created at the Si/SiO₂ interface. When O atoms reach the Si surface they combine to Si atoms and create volatile SiO_q^{\uparrow} molecules. The diffusion time is therefore 155 minimized when O atoms are directly formed at the triple line [33, 34, and 39]. The local 156 etching thins locally the SiO₂ BOX layer (~8 nm) until the Si fingers meet the Si handle 157 wafer. (iii) This has two consequences: (1) thermodynamically the Si finger tends to spread 158 laterally on the Si handle wafer thanks to perfect wetting. (2) Kinetically the newly formed 159 lateral etching front forces the finger to progress laterally. This enlargement of the fingers is 160 visible in LEEM (fig. 3(b)) and AFM images (fig. 4(a)). This forced lateral spreading gives 161 rise to a destabilization of the finger: it splits laterally in two parts with the occurrence of 162 a newly exposed SiO₂ BOX layer in the middle. (iv) This SiO₂ layer is in turn the place 163 for preferential etching at the triple line. When this SiO₂ layer in the middle is completely 164 etched, the two Si fingers meet again and form a flattened structure that encapsulates 165 the remaining SiO₂ BOX layers that are protected against further fast etching. When the splitting in two parts of the Si fingers does not occur, no elliptic SiO₂ layer is formed and another scenario with direct encapsulation of one SiO₂ BOX stripe occurs (see (v) in fig. 4(c)). This last scenario is clearly put in evidence in the unzipping process of Si fingers as detailed in the following.

b. Unzipping of Si fingers Figure 5(a) shows a sequence of LEEM images at 1370 K. We can observe a local anomaly at the reaction front edge: the triangular SiO₂ BOX layer is locally destabilized and this event suddenly generates a free SiO₂ layer that seems to

detach and propagate at high velocity $(210\pm10 \text{ nm.s}^{-1})$ in the direction opposite to the front. 174 To understand this process we have performed an ex situ analysis of the surface morphology by AFM. Fig. 5(c) shows an elongated silicon finger away from the front edge ($\sim 10 \ \mu \text{m}$) which is hollowed out in the middle. A SiO₂ BOX layer is clearly observed at the bottom of 177 this hollow as a very flat area surrounded by a Si rim and delimited by lateral trenches in the oxide layer (see height profile on fig. 5(c)). These very characteristic features of the SiO₂ 179 BOX layer, etched locally by Si, have already been observed in the context of conventional SOI dewetting [23 and 39] and are found here everywhere at the front edge demonstrating unambiguously that the SiO₂ BOX can be present far away from the front. The proposed 182 scenario is the following: The Si fingers formed by SOI dewetting locally etch the BOX 183 layer on the lateral sides. When a Si finger reaches the Si handle wafer, the SiO₂ BOX 184 layer may become encapsulated (see scenario (v) in fig. 4(c)) preventing further etching. This process is highly stabilized when the front recedes in the $\langle 100 \rangle$ direction resulting in the formation of extremely long SiO₂ BOX stripes (> 30 μ m) encapsulated into straight Si 187 fingers. Contrary to conventional dewetting of Si/SiO₂, the encapsulation of a SiO₂ layer inside a Si structure prevents the dewetting of the Si fingers (fig. 5(b)-(1)). However if the 189 dewetting front is destabilized and a finger detaches from the SOI layer edge (2), then a 190 new triple line Si/BOX/vacuum appears. Lateral shedding of Si by dewetting is promoted (3). This process unzips the Si finger from the middle (hollowed Si finger) and creates two 192 lateral rims of Si (4). Simultaneously the freshly exposed SiO₂ BOX layer is continuously 193 etched on the lateral sides giving the impression of the propagation of a SiO₂ layer at the 194 surface (5).

195 c. Si ring formation Figures 3(b) and 6(a) address a scenario occurring at slightly 196 lower temperature (1300 K). At this temperature the width W at the front of the freshly 197 exposed SiO₂ BOX layer is about 2 μ m (dashed lines in fig. 3(a)) whereas it is about 0.9 198 μ m at 1370 K. The width change arises because the activation energy for dewetting is lower 199 (2.4 eV±0.5 eV) than for etching (4.0±0.5 eV as described previously). Therefore the Si 200 fingers formed by dewetting are much longer and have now enough time to destabilize by 201 a Rayleigh-Plateau type instability. However together with the usual formation of 3D Si 202 islands, we also observe the formation of Si rings exposing in the middle a remaining SiO₂ 203 BOX layer (see circles in fig. 3(b) and ref. [40]). This is also clearly seen by AFM (fig. 204 6(a)). Then this BOX layer in the middle of a Si ring is etched progressively and the ring

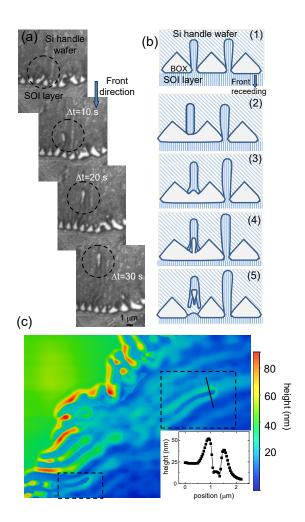


FIG. 5. (a) LEEM image sequence of Si/SiO_2 BOX front at 1370 K. Black dashed circles show the occurrence of a defect at the front that generates the linear propagation of a new front of SiO_2 BOX layer decomposition in the direction opposite to the dewetting front. (b) Scheme of the unzipping scenario of SOI finger with encapsulated SiO_2 . The finger is destabilized at the front side. This event allows for the lateral dewetting of Si and splits the finger into two parts. This process progresses all along the finger over tenth of microns (unzipping process). Simultaneous etching of the newly exposed SiO_2 BOX layer.(c) AFM image $(16\times10\ mum^2)$ showing the morphology of this unzipping process (black dashed areas). Inset: profile of a silicon finger hollowed out in the middle (black line): Si lateral rim, trenches in the the SiO_2 layer at the contact line with Si and flat area of BOX in the middle.

shape smoothens giving a final rough Si surface morphology. We propose a scenario similar to the one described for Si fingers (fig. 4(c)) but adapted to 3D islands: (i) The 3D Si islands

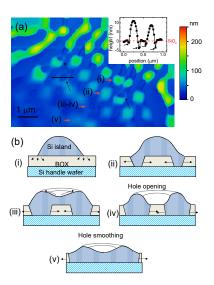


FIG. 6. (a) AFM image of the dewetting/reaction front (1300 K) obtained after quenching to RT. Elongated Si fingers are observed and agglomerate into 3D islands. The islands are unstable against Si rings formation (black arrow). Inset: profile of a circular hole (black line in the image) exposing a remaining SiO₂ BOX layer inside (BOX surface: red dashed line). Black arrows show the trenches in the BOX layer formed by local etching at the Si/SiO₂ contact lines. (b) Scenario of ring formation: (i) Si islands formed by dewetting and Si fingers. (ii) Local breaking through the SiO₂ layer and contact with the Si handle wafer. (iii) Radial progression of the reaction front and Si mass shedding from the center of the island. (iv) Opening of a hole into the center exposing the remaining SiO₂ BOX layer. (v) Etching of the SiO₂ BOX layer and spreading/smoothing of the Si layer.

formed by SOI finger dewetting instability [23 and 37] expose the BOX layer to vacuum. The circular contact line between the Si islands and the BOX layer is the place for preferential etching of the BOX layer. (ii) When the BOX layer is locally completely etched, the SOI layer and Si handle wafer are in contact and then the etching front progresses radially as for Si fingers. (iii) This process favors a mass transfer of Si from the center to the edge. (iv) This Si shedding destabilizes the compact island shape into a ring and a circular BOX layer

²¹³ appears in the center as it was protected by the 3D island. This observation is similar to the ²¹⁴ one obtained in the context of liquid Si droplets onto SiO₂ (quartz) forming spontaneously a ²¹⁵ liquid Si torus [41]. (v) Then the inner SiO₂ layer progressively disappears by lateral etching ²¹⁶ and when it is completely etched the Si morphology smoothens.

217 IV. CONCLUSION

We have shown by in situ LEEM and ex situ AFM that the Si/SiO₂ interface in ul-218 219 tra thin SOI system can lead to different structures/morphologies depending on annealing 220 conditions. The competition between (de)wetting and $Si+SiO_2 \longrightarrow 2 SiO_g$ reaction kinetics ₂₂₁ makes possible to achieve steady state regimes where the dewetting front and the etching 222 front coexist in a narrow space region. These regimes may create peculiar structures such 223 as 1D thin SiO₂ BOX stripes encapsulated into crystalline Si fingers or Si rings with a SiO₂ 224 BOX layer in the center. The formation of these structures follow different scenarios where the localization of the etching at the contact line between Si and SiO₂ and mass transfers ²²⁶ induced by Si/SiO₂ wetting properties play key roles. Small modifications at the front edge ²²⁷ may also lead to instabilities such as the fast unzipping of Si fingers with encapsulated SiO₂ ²²⁸ BOX stripes. This analysis of the front dynamics is completed with a quantitative study of the energetics of the process. We show that dewetting and etching reaction are thermally 230 activated with an activation energy respectively of 2.4 ± 0.5 eV and 4.0 ± 0.5 eV. To conclude, 231 these results propose a general picture of the dynamics of Si/SiO₂/Si interfaces at high tem-232 perature in the limit of a thin SiO_2 layer (≤ 8 nm). We believe these results may initiate 233 new approaches to fabricate specific structures based on SOI technology [42 and 43] and to ²³⁴ prevent SOI dewetting processes by encapsulation approaches.

235 V. SUPPLEMENTARY MATERIALS

See supplementary materials for the complete movies of the dewetting and reacting pro-

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