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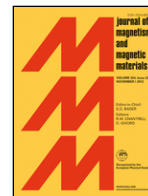
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Research articles

Magnetic properties, anisotropy parameters and magnetocaloric effect of flux grown MnFe_4Si_3 single crystal

H. Yibole^a, W. Hanggai^a, Z.Q. Ou^a, R. Hamane^b, V. Hardy^b, F. Guillou^a

^a Inner Mongolia Key Laboratory for Physics and Chemistry of Functional Materials, Inner Mongolia Normal University, 81 Zhaowuda Rd, Saihan District, Hohhot 010022, China

^b Laboratoire CRISMAT, UMR 6508, CNRS/ENSICAEN/UCBN, 6 B^d Marechal Juin, 14050 Caen Cedex, France

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ABSTRACT

The transition-metal based alloy MnFe_4Si_3 not only is a potential candidate for room temperature magnetocaloric applications, but also shows a large magnetic anisotropy forming an interesting case study in the search for rare-earth free permanent magnets. However, former polycrystalline and single crystal studies led to major disagreements about the order of the magnetic transition and the magnetocrystalline anisotropy scheme, which are two essential points for the understanding of this alloy. Here, magnetic, magnetocaloric properties and the magnetic anisotropy of MnFe_4Si_3 ($\text{Mn}_{0.86}\text{Fe}_{4.24}\text{Si}_{2.90}$) are investigated on a high quality single crystal grown by flux method, and compared to polycrystalline materials. Using the recently proposed criterion of field dependence of the magnetocaloric effect, we show that the ferromagnetic transition is more likely to be of second order, which is fully compatible with the absence of thermal hysteresis at the ferromagnetic transition in the present MnFe_4Si_3 crystal. The c axis is confirmed to be the hard magnetic axis, both in single crystal and polycrystalline MnFe_4Si_3 , and a large, dominant, K_1 anisotropy constant ($\sim -2.5 \text{ MJ m}^{-3}$) is found at low temperatures.

1. Introduction

Initially investigated in the 60s and 70s [1–4], transition metal-based compounds deriving from $\text{Mn}_{5-x}\text{Fe}_x\text{Si}_3$ are currently receiving a renewed interest for their rich magnetic properties and for their potential use in magnetocaloric applications [5–11]. $\text{Mn}_{5-x}\text{Fe}_x\text{Si}_3$ compounds were originally believed to crystallize in the hexagonal $P_{63/mcm}$ structure with Mn and Fe distributed on 4d and 6g sites, and Si metalloid atoms on 6g sites with different coordinates. Magnetic properties were found to be highly dependent on the Fe content [5,6]. Of particular interest is the $x \approx 4$ composition, MnFe_4Si_3 , which is ferromagnetic with a saturation magnetization of 1.5–1.7 $\mu_B/\text{T.M.}$ and presents a Curie temperature around 300 K making it a potential candidate for magnetocaloric applications near room temperature [5]. Some recent studies have deepened the understanding of this materials family using single crystals grown by Czochralski method [8]. Combined neutron and X-ray diffraction data have led to the emergence of a new structural model [8] and detailed inelastic neutron scattering experiments and Mössbauer spectroscopy have clarified the spin and lattice dynamics of MnFe_4Si_3 [9,10].

Despite these extensive investigations on MnFe_4Si_3 , some controversies persist. First, the order of the magnetic transition MnFe_4Si_3 is unclear. The ferromagnetic transition was originally classified as a contin-

uous second order transition, due to the absence of field-induced transition on the magnetization curves of bulk polycrystalline materials [5]. However, the recent observation of a large thermal hysteresis of $\sim 10 \text{ K}$ in single crystal is characteristic of a first order transition [8]. Further physical properties pointed out a complex behavior potentially closer to a second order behavior [9]. Usually, observing a large thermal hysteresis at a magnetic transition is typical of a first order transition with a significant latent heat and large discontinuities in the lattice parameters or a change in crystal symmetry. All these features should in turn lead to appreciable magnetocaloric entropy change [12]. On the contrary, MnFe_4Si_3 seems to combine a relatively large thermal hysteresis with only a modest entropy change of 2–2.5 $\text{J kg}^{-1} \text{K}^{-1}$ for 2 T, making it an unusual magnetocaloric compound [5,8,9]. The second debated issue in MnFe_4Si_3 concerns the magnetic anisotropy, originally believed to correspond to a c easy axis from neutron diffraction measurements and X-ray diffraction on in-field oriented powders [3], but later proposed to be a c hard axis with an easy (a,b) plane based on single crystal data [8].

The purpose of this work is to clarify the two aforementioned points, by a detailed study of the order of the transition from magnetization and magnetocaloric measurements, and by investigating the magnetic anisotropy of MnFe_4Si_3 both in single crystal and polycrystalline forms. A single crystal growth technique different from Ref. [8]

E-mail address: francoisguillou@imnu.edu.cn (F. Guillou)

was selected for this study in order to investigate whether sample preparation can explain part of the disagreements in the literature. In addition, a quantitative estimate of the magnetic anisotropy parameters over a large temperature range turns out to be needed for MnFe_4Si_3 , since other 5:3 transition-metal:metalloid alloys such as $\text{Fe}_5(\text{Si},\text{B})_3$ or $\text{Fe}_5(\text{P},\text{B})_3$ are currently explored for their potential use as rare-earth free permanent magnets [13–15].

2. Experimental details

MnFe_4Si_3 single crystals were grown by tin flux method. High-purity Mn (99.9%), Fe (99.99%) and Si (99.999%) starting materials were arc-melted with Sn (99.999%) in Ar atmosphere in a water-cooled copper crucible. Due to the smaller solubility of Si in molten Sn relative to Mn and Fe, we used extra amount of Si starting materials compared to the metal:metalloid stoichiometric ratio of 5:3. The resulting ingot was then sealed in quartz ampoule in 200 mbar Ar. The charge to flux ratio was 1:20 wt%. The sealed ampoule was then placed in a vertical furnace and heated up to 1423 K. The sample was maintained at this temperature for 100 h, and then slowly cooled at a rate of 2 K/h down to 700 K, at which temperature the excess of Sn was removed. The remaining flux was finally removed by etching with diluted hydrochloric acid. The as-grown crystals present the well-formed prismatic shape of an elongated needle. Microanalyses were performed using Hitachi SU-8010 scanning electron microscope (SEM) equipped with an energy dispersive spectrometer (EDS). The chemical compositions of the single crystals were determined with EDS by probing several locations on each crystal at an accelerating voltage of 15 keV. Reference materials were used to correct EDS data and make the chemical analysis more quantitative.

A polycrystalline MnFe_4Si_3 ingot was prepared by arc-melting elemental starting materials in purified Ar atmosphere. The button was flipped 6 times to ensure homogenization. The grain size distribution of this as-prepared button is quite large, with a few crystallites at the button surface having an elongated shape and reaching the mm in length, while the inner part of the button present a columnar microstructure with an average grain size of approximately $120 \mu\text{m} \times 40 \mu\text{m}$. The resulting sample is then cut, ground using an agate mortar and sieved. Particles smaller than $36 \mu\text{m}$ are used for powder x-ray diffraction, and part of the powder is oriented in a magnetic field of 11 kOe using epoxy glue as binder. Powder x-ray diffraction XRD experiments were carried out at room temperature using a Panalytical Empyrean diffractometer employing $\text{Cu K}\alpha$ radiation.

Magnetic measurements were performed in a VersaLab Quantum Design system employing a Vibrating Sample Magnetometer option. To limit possible thermalization artefacts, magnetization measurements as a function of the temperature were carried out while settling the temperature at each point and/or using slow temperature sweep ($< 1 \text{ K min}^{-1}$). The limited size of the sample ($m \sim 0.019 \text{ mg}$, estimated from SEM pictures and the theoretical density of MnFe_4Si_3) makes such measurements challenging. First, to avoid contamination, the needle is mounted under optical microscope on a fresh half-straw quartz holder using non-magnetic tools and only a tiny amount of superglue as fixer. A background signal with only the glue was measured to ensure it results only in a minor diamagnetic contribution. Then, large oscillations (4 mm) and long integration times (up to 10 s/point) were used to optimize the signal/noise ratio. Even in the most unfavorable situation, such as M - T curve recorded along the hard axis with an external field of 100 Oe, the signal in the paramagnetic phase (9.10^{-6} emu) remains significantly larger than the noise taken as the scattering between different points in the same conditions (noise level $\sim 2.10^{-7} \text{ emu}$). Neglecting potential error on the density, most of the uncertainty on the massic or volumic magnetization actually originates from the determination of the radius of the needle on SEM pictures, $\Delta r/r \approx 4\%$, which leads to sizable uncertainty on the sample volume $\Delta V/V \approx 8\%$ and thus

on the normalized magnetization, estimated as $\Delta M/M \approx \Delta m_{\text{raw}}/m_{\text{raw}} + \Delta V/V \approx 10\%$. The magnetization versus temperature curves are presented in terms of external magnetic field. For M vs H measurements with $H//(\text{a},\text{b})$ plane, i.e. for a magnetic field applied perpendicular to the long axis of the needle, a demagnetization factor of $N = 1/2$ was accounted for in order to present M - H data in terms of internal magnetic field H_{in} , according to the usual approximation $H_{\text{in}} \approx H - NM$, where H represents the applied magnetic field.

3. Results and discussion

Fig. 1(a) shows an SEM image of the as-grown crystal used in this study. The surfaces of the crystal are regular and homogeneous without noticeable flux contamination, and the needle shape presents a length:diameter ratio more than 30:1. The composition calculated from EDS spectrometry is approximately $\text{Mn}_{-0.86}\text{Fe}_{-4.24}\text{Si}_{-2.90}$, i.e. close to the targeted composition but with possibly a slight excess of Fe. For simplicity only the MnFe_4Si_3 nominal composition is mentioned hereafter. Fig. 1(b) and (c) show magnetization curves for the single crystal measured along different directions. In the direction perpendicular to the c axis, i.e. perpendicular to the long axis of the needle, the only distinguishable feature is the magnetization step corresponding to the ferromagnetic transition at 323 K taken from the dM/dT maximum in an external field of 1 kOe. This transition temperature is slightly higher than the range 300–310 K reported in former studies on MnFe_4Si_3 , a feature which can be ascribed to the small deviation in stoichiometry inherent to our flux crystal growth method. As a matter of fact, observing T_C slightly higher than 300–310 K is fully compatible with a minor Fe excess ($x \geq 4$) in the $\text{Mn}_{5-x}\text{Fe}_x\text{Si}_3$ phase diagram [6]. While no significant magnetic anisotropy could be distinguished when measuring magnetization along 3 random directions with $H \perp c$ (not shown), the shape of the $M(T)$ curve is strongly modified and the magnetization strongly reduced when performing measurements in $H//c$ direction. This indicates a combination of an easy (a,b) plane with an hard c axis uniaxial magnetic anisotropy, in line with recent single crystal investigation [8]. However in contrast to a former study reporting a 10 K thermal hysteresis [8], our data shows that the shift between heating and cooling $M(T)$ curves is smaller than 0.1 K (both for measurements parallel or perpendicular to the c axis), pointing to the absence of significant thermal hysteresis at the ferromagnetic transition in the present crystal. On the one hand, the presence of thermal hysteresis at a magnetic transition can be regarded as a signature of its first-order nature. On the other hand, absence of hysteresis does not demonstrate one is dealing with a second-order transition, since there are first-order magnetic transitions with very small thermal hysteresis [16].

To get further insight into the ferromagnetic transition order in MnFe_4Si_3 , we used the method recently proposed by V. Franco et al. which consists in analyzing the temperature and field dependence of the magnetocaloric effect [17,18]. It requires to first determine the magnetocaloric ΔS , as presented in Fig. 2 in the easy plane direction, which is the most relevant for caloric purposes. Given that no significant thermal hysteresis could be detected, the entropy change was calculated based on magnetization versus field curves measured at different temperatures and using the magnetocaloric Maxwell equation [19]. Finite entropy change values are found over a broad temperature range around the Curie temperature. The maximal ΔS are -1.8 , -2.9 and $-3.7 \text{ J kg}^{-1} \text{ K}^{-1}$ for field changes $\Delta B = 1, 2$ and 3 T , respectively. These entropy change values are –as expected– larger than the $\sim 2 \text{ J kg}^{-1} \text{ K}^{-1}$ for $\Delta B = 2 \text{ T}$ reported in polycrystalline materials [5], and in line with 2 – $2.9 \text{ J kg}^{-1} \text{ K}^{-1}$ reported for single crystals [8,9] or calculated by Monte Carlo simulations [11]. While sizable, these entropy values are nevertheless relatively limited, as the magnetocaloric effect spreads over a large temperature range, mimicking the broadening of $M(T)$ curves at high field. Such a behavior is most often typical of magnetocaloric materials with second order continuous transitions.

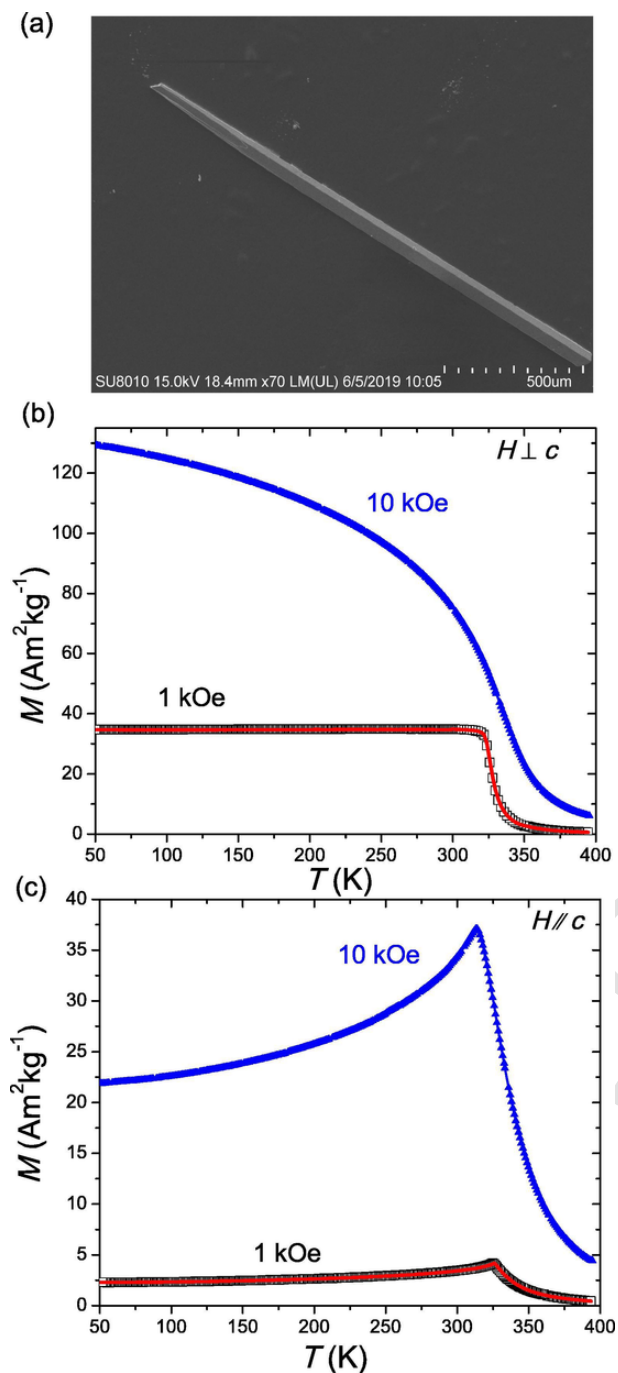


Fig. 1. SEM picture of MnFe_4Si_3 crystal used in this study (a) and magnetization measurements as a function of the temperature in an external field of 1 kOe upon cooling (open symbol) and upon heating (line) and 10 kOe perpendicular to the long axis (b) and parallel (c).

Then, assuming that the field-dependence of the isothermal entropy change obeys a power law ($\Delta S \propto H^n$), one calculates the exponent n from the expression: $n(T, H) = d \ln |\Delta S| / d \ln H$ [18]. Several studies have shown that the field dependence of the magnetocaloric effect is intrinsically different depending on the order of the magnetic transition [17,18,20]. In particular, it has recently been proposed that $n > 2$ in the transition region is characteristic of first order magnetic transition [18]. Bottom panel of Fig. 2 presents the local field dependence of ΔS in MnFe_4Si_3 single crystal. Clearly, n remains lower than 2 near T_C for $\Delta B = 2$ T, and similar results were obtained for other field change val-

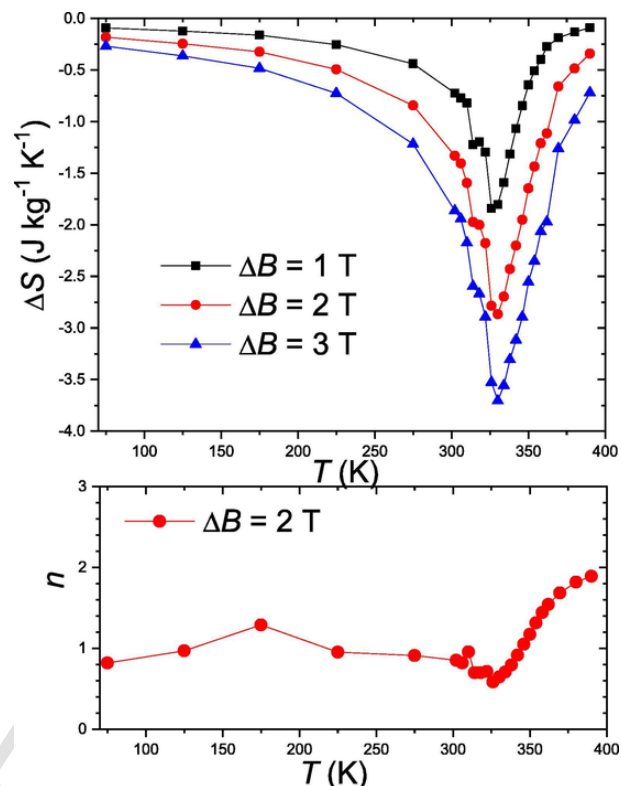


Fig. 2. Entropy change as a function of the temperature for MnFe_4Si_3 crystal in the easy plane direction. Bottom panel, local field dependence of the entropy change at 2 T.

ues (not shown). Not only the n values, but also their temperature evolution, $\sim 1 \rightarrow \text{minimum} \rightarrow \sim 2$, are typical of materials with second order transition [18]. When $T \gg T_C$, n tends toward 2, the values expected for a conventional paramagnet. The analysis of the field dependence of the ΔS thus demonstrates the second order character of the ferromagnetic transition, in line with the relative broadness of transition at intermediate magnetic fields and the absence of thermal hysteresis on the $M(T)$ curves on this crystal.

Another point of controversy that emerged from past studies concerns the magnetic configuration and magnetic anisotropy. The first powder neutron diffractions studies reported for MnFe_4Si_3 spin arrangement with moments tilted by $\sim 40^\circ$ of the c axis [6] or oriented along the c axis [7], in a way similar to the parent Fe_5Si_3 [2]. On the other hand, more recent studies based on single crystals revealed a spin arrangement with moment in the (a,b) plane from magnetization measurements and at 136° to $[1\ 0\ 0]$ -direction from neutron diffraction [8]. Magnetization versus field curves were systematically recorded on our single crystal in the (a,b) plane and along the c axis, a selection of the most representative temperatures are presented in Fig. 3. Only the first quadrant is shown, but the measurements were performed along 5 quadrants and pointed toward a negligibly small magnetic hysteresis. For instance, at 50 K, the coercive field H_C is less than 5 Oe and remnant magnetization M_{rem} less than $0.05 \text{ Am}^2 \text{ kg}^{-1}$. In agreement with temperature dependence of the magnetization in Fig. 1, $M(H)$ data indicate an (a,b) easy plane with a c hard axis anisotropy. At 50 K, the magnetic anisotropy appears particularly large, as the anisotropy field (H_{an}) is significantly higher than our applied field of 3 T. A graphical linear extrapolation suggests that H_{an} is of the order of ~ 6 T. This anisotropy slowly decreases while increasing the temperature until approaching the Curie temperature $T_C \approx 323$ K where it decreases more rapidly. Around 300 K, the magnetization parallel and perpendicular to the c axis does not coincide, which points toward a certain anisotropy of the magnetization and the presence of non-zero orbital moments in

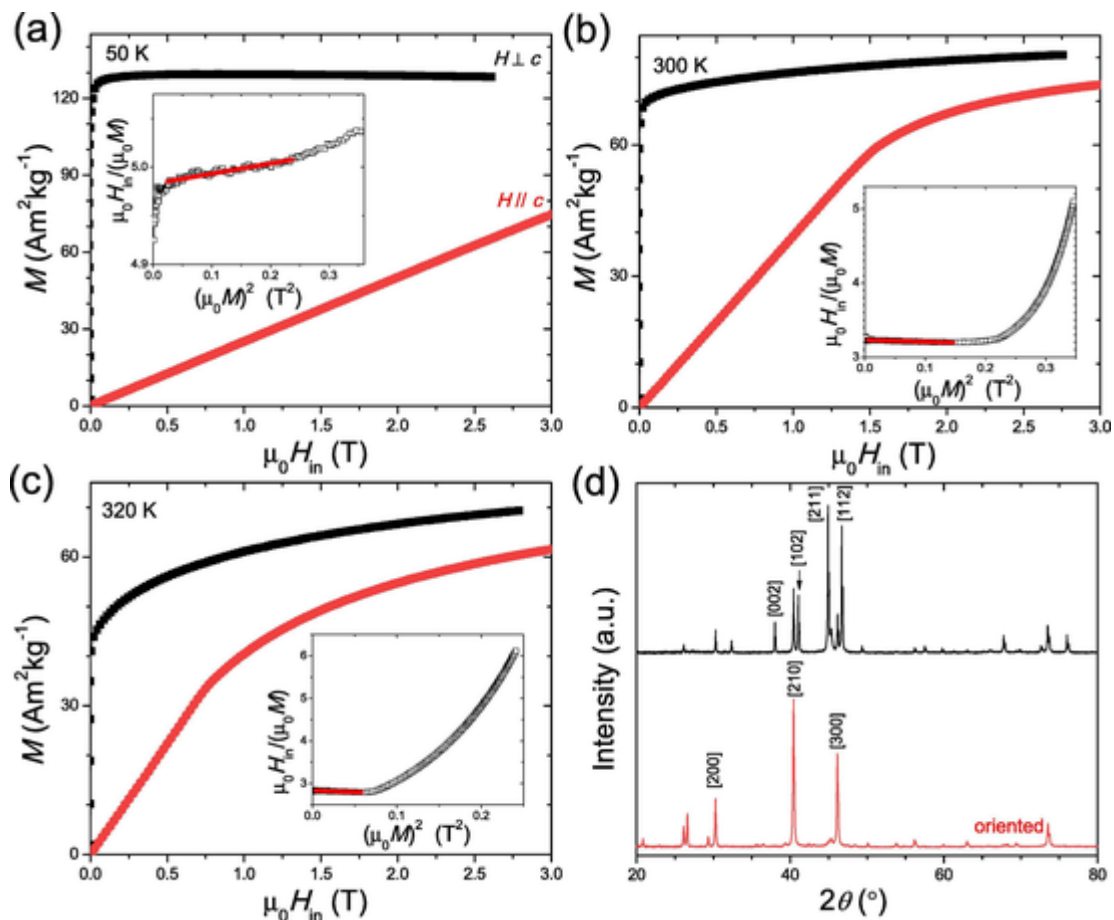


Fig. 3. Magnetization curves for MnFe_4Si_3 at selected temperatures along and perpendicular to the c axis (Panels a to c) with in inset the determination of the anisotropy constants by the Sucksmith and Thompson method. Panel d, comparison between random (top curve) and oriented (bottom curve) powder XRD with peak experiencing large changes being indexed.

MnFe_4Si_3 . These observations, in particular the c axis being the hard magnetic axis, are in line with formerly reported magnetization curves on single crystals [8] and at odd with polycrystalline studies [3,6,7]. As the cooling rate during single crystal synthesis is much slower than polycrystalline materials prepared by induction melting [7] or arc-melting with annealing followed by quenching [6], preferential site occupation maybe different in materials with different crystalline forms. To investigate this possibility we prepared a polycrystalline batch of MnFe_4Si_3 by arc-melting. Then, the sample is hand crushed into particles and part of this powder is mixed with an epoxy binder, shaped into a plate, and oriented in magnetic field with the orientation field perpendicular to the plate. XRD measurements were made on the plate surface. XRD measurements for random and oriented powders show a stark contrast. As a result of magnetic orientation, one observes an almost complete disappearance of the reflections having a c axis component like for instance the [0 0 2] direction; while Bragg peaks having an a or b axis component are strongly enhanced, for instance [2 0 0], [2 1 0] and [3 0 0] directions. These results appear opposite to that obtained for Fe_5Si_3 and MnFe_4Si_3 by a similar method, which were indicating an easy axis along the c axis for both alloys [3]. Our powder oriented polycrystalline sample indicates that in polycrystalline materials the (a,b) plane is an easy plane and c is the hard axis, in agreement with single crystal magnetization data. Different crystalline forms can thus not explain the discrepancy in magnetic anisotropy reported for MnFe_4Si_3 .

The large magnetic anisotropy of MnFe_4Si_3 turns out to deserve a more quantitative determination, in particular as transition metal-based alloys could be considered as potential candidates as rare-earth free

permanent magnets. Usually for an hexagonal system, the first (K_1) and second-order (K_2) anisotropy constants are considered for the magneto-crystalline anisotropy energy $E \approx K_1 \sin^2\theta + K_2 \sin^4\theta$. To determine K_1 and K_2 from magnetization measurements along the hard c axis direction, the Sucksmith and Thompson method has been used [21]. This method is more customarily employed to derive the two magneto-crystalline anisotropy constants of materials with c being the easy magnetic axis. It can nevertheless be applied to the case of an easy magnetization direction perpendicular to the hexagonal axis. In such a case, the slope and intercept of an H/M vs M^2 plot will correspond to $\text{slope} = 4K_2M_S^4$ and $\text{intercept} = (-2K_1 - 4K_2)/M_S^2$, respectively [21]. The applicability of the Sucksmith and Thompson method to various magneto-crystalline anisotropy scheme has been confirmed from modelling studies [22] and used in different materials with a combination of easy plane and hard c axis [23,24]. Beside systematic error, due to the significant relative error on magnetization, the differences between maxima and minima for the intercepts and slopes of H_{in}/M vs $(\mu_0M)^2$ linear fits are large too, resulting in sizable uncertainties on K_1 and K_2 . Fig. 4 presents the temperature evolution of K_1 and K_2 . K_2 appears negligible compared to the large K_1 . While K_1 exhibits negative values, K_2 might experience a change of sign around 150 K. The predominance of K_1 and its negative sign is in line with an easy plane magnetic direction perpendicular to the c axis [25]. Neglecting K_2 , our estimate of the anisotropy field ($H_{an} \sim 6$ T at $T = 50$ K) provides us a mean for an alternative estimation of K_1 considering $M = M_S$ at H_{an} , it results in $K_1 \approx -\frac{1}{2}(H_{an}M_S) \approx -3.0 \text{ MJm}^{-3}$. This estimate of K_1 is fully compatible with that obtained by H/M vs M^2 fittings ($K_1 \approx -2.8 \text{ MJm}^{-3}$).

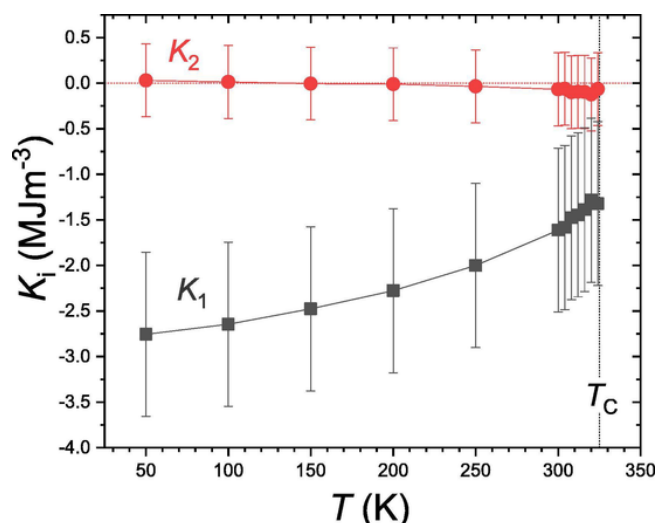


Fig. 4. Temperature dependence of the magneto-crystalline anisotropy parameters for MnFe_4Si_3 .

The binary parent Fe_5Si_3 is ferromagnetic with a $T_C \sim 380$ K, but its magnetic anisotropy could unfortunately not be compared to MnFe_4Si_3 , as being a metastable phase at room temperature stabilized by rapid solidification or nanostructuring, little single crystal information could be found. On the other hand, ternaries with metalloid substitutions on Si have recently received a great interest as they offer a possibility to reach Curie temperatures higher than those of the binary parents, making them potential candidates for rare-earth free permanent magnet. In tetragonal Fe_5SiB_2 , between $T_C \sim 784$ K and 140 K (spin re-orientation transition) the easy magnetic axis is along the c axis with an anisotropy $K_1 = 0.25$ MJ m⁻³, and an easy plane anisotropy below 140 K with only a modest anisotropy (MAE = -0.28 MJ m⁻³) [13,15]. In this latter tetragonal compounds, P for Si substitution favors an easy axis magnetization with magnetic anisotropy $K_1 \sim 0.38$ MJ m⁻³ at room temperature [13,14]. In hexagonal ternary alloys deriving from Fe_5Si_3 such as MnFe_4Si_3 , a different configuration is observed with a relatively strong hard c axis anisotropy and an easy plane. Beside the difference in crystal structure, the incorporation of Mn in Fe_5Si_3 may also play a role in the strengthening of the magnetic anisotropy. Mn atoms carry a significantly larger overall magnetic moment than Fe [6,7], and may also favor a stronger spin-orbit coupling. In addition, spin-polarized calculations pointed out the important role played by Mn d states in the filling of the electronic structure near the Fermi level [7]. Therefore it appears that minor Fe:Mn compositional changes in $\text{Mn}_{5-x}\text{Fe}_x\text{Si}_3$ alloys can yield major changes in magnetic configuration, which might in turn explain (at least partly) the observed scatter in the literature about the magnetic anisotropy scheme for MnFe_4Si_3 .

4. Conclusions

Both the magnetocaloric properties and magnetic anisotropy of MnFe_4Si_3 single crystal grown by flux method were investigated. By analyzing the field dependence of the magnetocaloric effect, the ferromagnetic transition at $T_C \sim 323$ K is found to be of second-order, which is fully compatible with the absence of thermal or magnetic hysteresis related to the transition. A relatively strong magnetic anisotropy for a transition-metals alloy is observed in MnFe_4Si_3 single crystal with the c axis being a hard magnetic direction and an easy plane perpendicular to c . The anisotropy constants are determined over a broad temperature range, leading to a relatively large negative K_1 . In contrast to former reports, X-ray diffraction experiments on field oriented polycrystalline materials confirm the present magnetic anisotropy scheme, indicating that the crystalline form or the single crystal preparation method is not

the primary origin for the differences existing in the literature on MnFe_4Si_3 .

CRediT authorship contribution statement

H. Yibole: Conceptualization, Investigation. W. Haggai: Investigation. Z.Q. Ou: Investigation. R. Hamane: Investigation. V. Hardy: Investigation. F. Guillou: Investigation.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.jmmm.2020.166597>.

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