



HAL
open science

Versatile lysine dendrigrafts and polyethylene glycol hydrogels with inherent biological properties: in vitro cell behavior modulation and in vivo biocompatibility

Mariana Carrancá, Louise Griveau, Noëlle Remoué, Chloé Lorion, Dominique Sigau-do-roussel, Pierre Weiss, Clément Faye, Daniel Ferri-angulo, Romain Debret, Jérôme Sohier

► To cite this version:

Mariana Carrancá, Louise Griveau, Noëlle Remoué, Chloé Lorion, Dominique Sigau-do-roussel, et al.. Versatile lysine dendrigrafts and polyethylene glycol hydrogels with inherent biological properties: in vitro cell behavior modulation and in vivo biocompatibility. *Journal of Biomedical Materials Research Part A*, 2020, 10.1002/jbm.a.37083 . hal-03016904

HAL Id: hal-03016904

<https://hal.science/hal-03016904>

Submitted on 1 Dec 2020

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



Versatile lysine dendrigrafts and polyethylene glycol hydrogels with inherent biological properties: in vitro cell behavior modulation and in vivo biocompatibility

Journal:	<i>Journal of Biomedical Materials Research: Part A</i>
Manuscript ID	JBMR-A-20-0077
Wiley - Manuscript type:	Original Article
Date Submitted by the Author:	05-Feb-2020
Complete List of Authors:	Carrancá, Mariana; CNRS, LBTI; MATEIS, I2B Griveau, Louise; CNRS, LBTI; MATEIS, I2B Remoué, Noëlle ; CNRS, LBTI Lorion, Chloé; CNRS, LBTI Weiss, Pierre; INSERM U791, Faculty of Dental Surgery Orea, Valérie; CNRS, LBTI Sigaudou-Roussel, Dominique ; CNRS, LBTI Faye, Clément ; COLCOM, GL BIOCONTROL Ferri-Angulo, Daniel ; MATEIS, I2B Debret, Romain; CNRS, LBTI Sohier, Jérôme; MATEIS, I2B
Keywords:	PEG based Hydrogels, Mechanical Properties, Poly(L-lysine) Dendrimers, Cell interaction, Biocompatibility

SCHOLARONE™
 Manuscripts

Versatile lysine dendrigrafts and polyethylene glycol hydrogels with inherent biological properties: *in vitro* cell behavior modulation and *in vivo* biocompatibility.

Mariana Carrancá^{1,2}, Louise Griveau^{1,2}, Noëlle Remoué¹, Chloé Lorion¹, Pierre Weiss³, Valérie Orea¹, Dominique Sigaudou-Roussel¹, Clément Faye^{4#}, Daniel Ferri-Angulo², Romain Debret¹, Jérôme Sohier^{1,2}

¹CNRS Université Lyon 1, UMR 5305, Laboratory of Tissue Biology and Therapeutic Engineering, IBCP, 7 Passage du Vercors, 69367 Lyon Cedex 07, France

²CNRS INSA, UMR 5510, Laboratory for Materials Engineering and Science, Bat. B. Pascal, 7 Avenue Jean Capelle 69621 Villeurbanne Cedex, France

³INSERM, UMR_S 791, Laboratory of Osteo-Articular and Dental Engineering, 1 Place Alexis Ricordeau, 44042 Nantes, France

⁴COLCOM, Bat CAP ALPHA, 9 avenue de l'Europe 34830 Clapiers, France

[#]Current address : GLPBiocontrol, Bat CAP ALPHA, 9 avenue de l'Europe 34830, Clapiers, France

Author information:

Corresponding author: Jérôme Sohier

E-mail: jerome.sohier@insa-lyon.fr

Abstract:

Poly(ethylene glycol) (PEG) hydrogels have been extensively used as scaffolds for tissue engineering applications, owing to their biocompatibility, chemical versatility and tunable mechanical properties. However, their bio-inert properties require them to be associated with additional functional moieties to interact with cells. We propose here to reticulate PEG molecules with poly(L-lysine) dendrigrafts (DGL) to provide inherent cell functionalities to PEG-based hydrogels. The physico-chemical characteristics of the resulting hydrogels were studied in regard of the concentration of each component. With increasing amounts of DGL, the cross-linking time and swelling ratio could be decreased, conversely to mechanical properties, which could be tailored from 7.7 ± 0.7 to 90 ± 28.8 kPa. Furthermore, fibroblasts adhesion, viability and morphology on hydrogels were then assessed. While cell adhesion significantly increased with the concentration of DGL, cell viability was dependant of the ratio of DGL and PEG. Cell morphology and proliferation, however appeared mainly related to the overall hydrogel rigidity. To allow cell infiltration and cell growth in 3D, the hydrogels were rendered porous. The biocompatibility of resulting hydrogels of different compositions and porosities was evaluated by 3-week subcutaneous implantations in mice. Hydrogels allowed an extensive cellular infiltration with a mild foreign body reaction, histological evidence of hydrogel degradation and neovascularization.

Keywords: PEG based Hydrogels; Mechanical Properties; Poly(L-lysine) Dendrimers; Cell interaction; Biocompatibility

1. Introduction

The interest in hydrogels has grown rapidly in the past decade due to their vast potential in tissue engineering, tissue regeneration and drug delivery.⁽¹⁾ These hydrophilic polymer networks are of interest through their biocompatibility, viscoelasticity, permeability to oxygen and nutrients, and high water content.^{(2)–(4)} While natural (e.g., collagen, gelatin, fibrin) confer to hydrogels their inherent extracellular matrix (ECM) structure and qualities, they often show weak mechanical properties, batch-to-batch variability⁽⁵⁾ and risk of pathogen transfer.⁽⁶⁾ Conversely, synthetic hydrogels have an exact composition and multi-tunable properties, but may lack bioactivity to promote cellular activities.⁽³⁾ For instance, PEG hydrogels, which have been extensively used for controlled drug delivery, as cell vehicles or tissue engineering scaffolds⁽⁷⁾, are bio-inert and require an association with other functional moieties such as Arginine-Glycine-Aspartic (RGD) peptides to promote survival or function of adherent cells.⁽⁸⁾

PEG hydrogels can be obtained via covalent cross-links between PEG molecules through different paths, in which the PEG molecules are either functionalized with reactive end groups (methacrylate or acrylate for instance)^{(9),(10)} and activated by initiators or associated with multifunctional monomers to avoid the use of initiators. In the latter approach, dendrimers have shown potential as cross-linking monomers, thanks to their versatile chemical composition, highly organized 3D arborescent structure and ease of surface functionalization.^{(11)–(14)} Poly(amidoamine) (PAMAM) dendrimers, for instance, have been successfully associated with PEG to form

1
2
3
4
5 hydrogels for drug delivery applications that do not require interactions with cells.^{(15)–(17)}
6
7
8 Conversely, other dendritic structures with amine end groups, such as DGL, could be of interest
9
10 to provide cell interactions.⁽¹⁸⁾ Indeed, DGL used as coatings on surfaces have been shown to
11
12 increase cellular adhesion and proliferation of human skin fibroblasts, therefore providing
13
14 bioactivity.⁽¹⁹⁾ Interestingly, DGL has never been evaluated as cross-linking monomers to form
15
16 PEG hydrogels, while such hydrogels could benefit from their inherent bioactivity, without the
17
18 need for further association with functional moieties.
19
20
21

22
23
24 We therefore proposed to assess this possibility through the development of a novel hydrogel
25
26 composed of DGL and PEG, with extensive and straightforward tuneable mechanical properties,
27
28 to serve as matrix or cell support for soft tissue engineering applications.⁽²⁰⁾ The characteristics of
29
30 these hydrogels as cross-linking time, swelling and mechanical properties were determined. Cell
31
32 adhesion, viability, proliferation and morphology of dermal fibroblasts on dense hydrogels were
33
34 determined, in relation to hydrogel composition. Additionally, to allow cell colonisation and
35
36 infiltration, hydrogels were rendered porous by particulate/leaching technique and their
37
38 biocompatibility was determined by *in vivo* subcutaneous implantations in mice.
39
40
41
42
43
44
45

46 **2. Materials and Methods**

47 48 49 **2.1. Preparation of dense DGL/PEG hydrogels**

50
51
52 Dense hydrogels of different ratios of DGL/PEG were prepared by adding PEG-bis(N-
53
54 succinimidyl succinate) (PEG-NHS, Sigma-Aldrich) in anhydrous dimethylformamide (DMF,
55
56
57
58
59
60

1
2
3
4 Sigma Aldrich) solution, to DGL (COLCOM) in phosphate-buffered saline (PBS) in 2ml conic
5
6 tubes (Maxymum Recovery, Axygen) at 4°C. After cross-linking, the conical bottom of the tubes
7
8 was cut off, and the tubes were immersed in ethanol for 5 min to subsequently retrieve the
9
10 hydrogels. The resulting cylindrical hydrogels were then sectioned (2 mm thickness) using a
11
12 vibratome (7550 Integraslice, Campden Instruments Ltd.) and finally rehydrated and stored in PBS
13
14 at 4°C. For 2D cellular *in vitro* studies, hydrogels were prepared on top of coverslips for convenient
15
16 handling (12 mm in diameter). The desired concentrations of DGL and PEG were mixed and
17
18 swiftly deposited between a hydrophobic glass slide and a round coverslip (for a hydrogel
19
20 thickness of 0.6 mm). Hydrophobic glass slides were obtained by dipping in
21
22 dichlorodimethylsilane (Sigma Aldrich). After cross-linking, the hydrogels were sterilized
23
24 overnight in EtOH:PBS (70:30, v/v) solution, washed 3x30 min with sterile PBS and kept at 4°C
25
26 prior use. Concentrated hydrogels, which cross-linking time was faster than 10 s, were prepared in
27
28 a cold room to slow down cross-linking.
29
30
31
32
33

34 35 **2.2. Cross-linking speed**

36
37
38 To monitor cross-linking speed, hydrogel components were added to a glass vial (8x25x35 mm)
39
40 under agitation (300rpm) with a magnetic rod (5 mm) placed at exactly 4 cm from the magnetic
41
42 stirrer. Cross-linking time was defined, in our study, as the time needed to halt the magnetic rod
43
44 after adding all the hydrogel components.
45
46
47
48

49 50 **2.3. Swelling Ratio (Qs)**

51
52
53 The swelling ratio (Qs) of 2 mm-thick dense hydrogels discs was determined in PBS at 37 °C.
54
55 Prior incubation in PBS, samples were frozen in liquid nitrogen and freeze-dried to measure their
56
57
58
59
60

dry weight. Samples were blotted before each weight measurement, performed at 1, 2, 6 and 24 hours. The swelling ratio was defined as, $((W_s - W_i)/W_i)$, where W_s is the weight of swollen hydrogel and W_i is its initial dry weight after freeze-drying.

2.4. Mechanical testing

The mechanical properties of DGL/PEG hydrogels (2x9.1 mm) of different compositions were analyzed by cyclic compression with a dynamic mechanical analyzer (DMA 242 E Artemis, NEZSTCH). The hydrogel's domain of linearity was first determined with a strain sweep and compression test. Samples immersed in PBS were then subjected to compression at 10% strain and 60 μ m amplitude, with increasing frequencies (1-20 Hz) at room temperature.

2.5. *In vitro* cell culture studies

Human dermal fibroblasts (Promocell, Heidelberg) were seeded at a density of 10,000 cell/cm² on the surface of hydrogel-covered coverslips and cultured in DMEM-F12 medium (Gibco) supplemented with 10% FBS (Life Technologies) and 1% penicillin/streptomycin (PAA Laboratories) at 37°C and 5% CO₂. Culture medium was refreshed every second day.

Hydrogels cytocompatibility was evaluated with a live/dead assay. Cells were washed once with sterile PBS and incubated 30 min with a 6 μ M propidium iodide (Sigma Aldrich) and 1 μ M Calcein (Sigma Aldrich) solution and subsequently observed with a fluorescence microscope (Nikon TiE, Nikon Instruments). After observation, samples were washed with PBS, culture medium refreshed and cultured at 37°C and 5% CO₂ until next measurement. The number of alive and dead cells was

1
2
3
4
5 determined from image analysis (Imagej⁽²¹⁾), using 5 different fields of view randomly acquired
6
7 per replicate. Viability was determined as the percentage of alive cells from the total number of
8
9 cells. Cell adhesion was expressed as the total number of cells (alive or dead) present on the surface
10
11 of the hydrogels 24 h post-seeding. The effect of hydrogel composition on cell morphology was
12
13 evaluated by phase contrast microscopy (Nikon TiE) and by actin staining. Phase contrast
14
15 microscopy pictures were acquired after 24 and 72 h of culture. Cell spreading area, circularity
16
17 and feret diameter were determined to compare cell morphologies. Control were only analyzed
18
19 after 24 h, since cell confluence impeded images analysis after 72 h. Actin cytoskeleton was
20
21 observed by phalloidin (Sigma Aldrich) staining after 1, 3 and 8 days of culture. Cells on the
22
23 hydrogels were fixed with 4% paraformaldehyde (PFA, Thermo Fischer Scientific) for 10 min
24
25 followed by 20 min permeabilization with a 0.1% triton solution in PBS. After washing twice with
26
27 PBS, the samples were incubated for 10 min with a 2 μ g/ml 4',6-Diamidino-2'-phenylindole
28
29 dihydrochloride (DAPI, Sigma Aldrich) and 2 μ g/ml phalloidin (Thermo Fischer Scientific)
30
31 solution in PBS to stain the cell nucleus and actin cytoskeleton, respectively. Finally, hydrogels
32
33 were observed with a fluorescence microscope (Nikon TiE). Cell proliferation was determined by
34
35 counting DAPI stained cell nucleus after 1, 3 and 8 days of culture.
36
37
38
39
40
41
42
43
44
45
46

47 **2.6. Preparation of porous hydrogels**

48
49
50
51
52 Porous hydrogels were prepared by particulate/leaching technique using paraffin microspheres as
53
54 porogens.⁽²²⁾ Briefly, 10 g paraffin (Histolab AB, Västra Frölunda) and 250 ml 0.5% poly(vinyl
55
56
57
58
59
60

1
2
3
4
5 alcohol) (PVA, Sigma Aldrich) were heated to 80°C under stirring. After 20 min, the suspension
6
7 was poured into ice water. The paraffin microspheres formed were sieved at 50, 100 and 180 µm
8
9 and the resulting fractions were washed with distilled water, freeze-dried and conserved at 4°C.

10
11
12 To prepare porous hydrogels, 400 mg of paraffin microsphere, previously prepared, were
13
14 compacted by centrifugation in 2ml microtubes (2000g, 10s) to obtain a flat surface.
15
16 Consecutively, DGL and PEG-NHS solutions were mixed by vortex and rapidly transferred to the
17
18 microtubes containing the paraffin at 0°C. After cross-linking, the conical bottom of the
19
20 microtubes was sectioned, paraffin was extracted with boiling EtOH for 40 min and the hydrogels
21
22 were removed from the microtubes. Finally, hydrogels were cut into 2 mm thick discs with a
23
24 vibratome. After that, the remaining paraffin was removed by several cycles (40 min) in reflux
25
26 EtOH. The obtained scaffold discs were sterilized overnight in EtOH 70%, washed 3x1h with
27
28 sterile PBS, and kept at 4 °C prior use.
29
30
31
32
33
34
35
36

37 **2.7. Biocompatibility**

38
39
40 Biocompatibility of synthesized hydrogels was assessed by subcutaneous implantation of acellular
41
42 hydrogels. After approval by local ethics committee, five hydrogels discs per condition (2x6 mm
43
44 diameter) were implanted under the back skin of 8-weeks old SKH1 mice (Charles River, Ecully)
45
46 under sedation by intraperitoneal xylazin-ketamin injection (four hydrogels per mouse). A small
47
48 incision was performed at the low back of the mice and 4 subcutaneous pockets created with a
49
50 sterile spatula. The hydrogels were inserted in the pockets and the incision sutured. Mice, fed ad
51
52 libitum, were monitored every day for recovery and signs of distress. After three weeks, the mice
53
54
55
56
57
58
59
60

1
2
3
4 were euthanized by anesthetic overdoses (intracardiac injection of sodium pentobarbital), the
5
6 hydrogel samples recovered with surrounding tissue, fixed in 4% paraformaldehyde (PFA)
7
8 solution in PBS, embedded in paraffin, sectioned and stained with Masson's trichrome. To
9
10 highlight the penetration of blood vessels in the implanted hydrogels, sections were stained for α -
11
12 SMA (Abcam #ab5694 1:250) by immunohistochemistry, cell nuclei were counter-stained with 2
13
14 $\mu\text{g/ml}$ DAPI solution and observed by confocal microscopy (ZEISS, LSM 800). Round blood
15
16 vessels with a diameter superior to 10 and 25 μm per hydrogel cross section were counted by image
17
18 analysis.
19
20
21
22

23 **2.8. Statistical analysis**

24
25
26
27 Statistical analyses were performed with Graphpad prism or Kaleidagraph. Test were performed
28
29 using variance analysis (ANOVA). Data values are presented as mean \pm standard error (SE) and
30
31 p -values of 0.05 and below were considered significant.
32
33
34

35 **3. Results**

36 **3.1 Effect of reactants concentration on the formation of self-standing hydrogels**

37
38
39 DGL/PEG hydrogel formation was straight forward. Concretely, hydrogels were obtained by
40
41 covalent reaction between amine groups in the DGL and the NHS ester ends groups in PEG by
42
43 simply mixing the solutions at room temperature (Figure 1.). Cross-linking velocity could be
44
45 varied between 5 to 145 s by modifying reagent concentrations (between 1 to 4 mM for DGL and
46
47 19 to 30 mM for PEG-NHS), predominantly through the DGL concentration, as presented in Table
48
49
50
51
52
53
54
55
56 1. Interestingly, by considering the amount of amine groups and NHS functions theoretically
57
58
59
60

available for the covalent reaction to occur, similar ratios of available amine to NHS functions resulted in different cross-linking velocities.

3.2 Swelling ratio

To mimic the physiological conditions, swelling ratio was determined in PBS at 37°C. As observed in Figure 2, the swelling was decreased with an increased concentration of DGL or PEG-NHS. After 2 hours, a great quantity of liquid has been absorbed and reached equilibrium after 6 hours. The composition with lowest concentration of both components showed the greatest swelling ratio of 22.89 ± 1.95 after 24 h.

3.3 Mechanical properties of dense DGL/PEG hydrogels

The mechanical properties of hydrogels of different compositions were measured with mechanical dynamical analysis. The loss (E''), storage (E') and complex modulus (E^*) were determined at increasing frequencies between 1 and 20 Hz. Figure 3 shows that for all hydrogels, E' was greater than E'' over the entire frequency range studied. This was further expressed by tan delta, which is the ratio of moduli (E''/E') that describes the viscous energy dissipation relative to the stored elastic energy. While E' was constant for all applied frequencies, E'' increased with frequency, which is a typical behavior of rubbery elastics⁽²³⁾. Overall, the complex modulus of the different hydrogels was increased with an increased concentration of PEG-NHS and DGL, varying from 7.7 ± 0.7 to 90.4 ± 28.8 kPa.

3.4 Cytocompatibility of the DGL/PEG hydrogels and effect of hydrogel composition on cell morphology and proliferation

To assess the potential of the DGL/PEG hydrogels as a cell substrate, the adhesion and viability of human fibroblasts seeded on the surface of hydrogels of different compositions were evaluated with a life/dead assay (Figure 4.A). Seeded cells readily adhered on the hydrogels surface after 24 hours with a very low cytotoxicity, except for the hydrogel 3/19 mM (DGL/PEG), which showed a mortality of 79.4%. As shown in Figure 4.B, the hydrogel composition had an apparent effect on cell adhesion, concretely, with DGL concentration. With an increase of DGL, the total number of adhered cells was increased while adhesion was minimal when the DGL concentration was 1 mM. After 3 days of culture, compositions 1/19, 1/28 and 2/19 mM DGL/PEG presented around 84% of viability while more concentrated compositions (2/28 and 3/28 mM DGL/PEG) showed a viability of 97.9 and 96%, respectively (Figure 4.C), with no difference compared to controls (tissue culture plastic). Of note, an increase in PEG-NHS concentration resulted in an increase of the viability of the adhered cells.

Cell morphology was also modified by hydrogel composition (Figure 5.A). Cells on hydrogels with lower concentrations of DGL (1mM) and therefore lower percentage of adhesion presented a round morphology similar to non-adherent cells, with a mean circularity of 0.8 and a small ferret diameter (26.9–28.9 μm). An increase in hydrogel concentration had an impact on cell morphology. Except for hydrogel composition 2/28 and 3/28 (DGL/PEG) which showed a similar

1
2
3
4
5 cell morphology. After 72 hours, cell spreading area was increased in all compositions (Figure
6
7
8 5.B). The effect of hydrogel composition on cell morphology and spreading was further confirmed
9
10 by phalloidin staining of f-actin fibres after 1, 3 and 8 days of culture (Figure 6.A.).
11
12
13 Regarding proliferation, seeded cells were able to proliferate over time (Figure 6.B), except for
14
15 hydrogels with the lower concentration of DGL, where cells stayed round and showed proliferation
16
17 rates close to zero. While hydrogels 2/28 and 3/28 mM showed similar proliferation rates of 69.9
18
19 ± 8.1 and 66.7 ± 14.6 , respectively, which were lower than tissue culture plastic controls (rate of
20
21 135.2 ± 26.19).
22
23
24
25

26 27 **3.5 Biocompatibility**

28
29
30 The behavior of DGL/PEG hydrogels of different compositions and porosities was evaluated *in*
31
32 *vivo* by subcutaneous implantation in mice. As shown in Figure 7, regardless of pore size or
33
34 composition, all hydrogels exhibited a mild foreign body reaction with the formation of a fibrous
35
36 capsule after 3 weeks of implantation. All porous hydrogels were deeply infiltrated by cells,
37
38 contrarily to dense hydrogels (Supplementary Figure 1). An important population of macrophages
39
40 were visible, highly concentrated at the rim and within the porous implants and on the external
41
42 edge of dense hydrogels. The macrophages were able to degrade the hydrogel through
43
44 phagocytosis, further opening the pores, as exemplified in the blue close-ups on Figure 7.
45
46
47
48
49 Interestingly, no granulocyte or lymphocytes could be observed, suggesting a mild inflammatory
50
51
52
53
54
55
56
57
58
59
60 reaction. While the different porosities did not result in differences of cellular infiltration, the

1
2
3
4
5 hydrogels composition clearly induced structural and degradation-related differences in the
6
7 retrieved implants after 3 weeks. Softer hydrogels (7.7 ± 0.4 kPa) appeared condensed while harder
8
9 ones (41.5 ± 5.0 and 76.7 ± 15.5 kPa) conserved their initial porous structure. Similar observations
10
11 were made in regards of phagocytosis, with harder hydrogels being less prone to degradation by
12
13 macrophages. The presence of neo-tissue within the hydrogels, revealing deposits of collagen,
14
15 followed a similar pattern with an increased occurrence in harder hydrogels than in soft ones.
16
17
18 Curiously, an important amount of blood vessels were present within and throughout the porous
19
20 hydrogels, as could be observed by IHC staining of α -SMA (Figure 8). Fewer blood vessels were
21
22 present in soft hydrogels compared to more rigid compositions, with a preference to porous
23
24 hydrogels with a broader range of pore size (50-180 μ m). When considering only the blood vessels
25
26 with a diameter over 25 μ m, a preference to bigger pores was observed for more rigid compositions
27
28 (Figure 8.B).
29
30
31
32
33
34
35
36
37

38 **4. Discussion**

39
40
41 PEG-based hydrogels have been the focus of much attention for various tissue engineering
42
43 applications over the past years.⁽⁷⁾ With the aim of further broadening their potential by providing
44
45 them with inherent cellular adhesion and interactions, we defined and evaluated a novel hydrogel
46
47 where homobifunctionalized PEG-NHS is cross-linked by poly(L-lysine) dendrigrafts (DGL).
48
49
50 Indeed, the mixing of PEG-NHS with DGL in aqueous solutions results in the swift formation of
51
52
53
54
55
56
57
58
59
60

1
2
3
4
5 self-standing hydrogels. Among the various dendrimer structures that have been associated with
6
7 PEG to form hydrogels,⁽¹³⁾ DGL was so far never evaluated in this respect.

10 By varying the concentrations of hydrogel components, the cross-linking velocity, swelling and
11
12 mechanical properties can be tailored. DGL of third generation used in this study presents an
13
14 important density of amine groups on their surface (123), and it has been determined that only
15
16 97.7% of these amines residues are available as binding sites per molecules (114).⁽²⁴⁾ PEG-NHS,
17
18 on the contrary, possess only two able NHS ester group per molecules. Hence, the ratio of available
19
20 amines groups *versus* NHS is always higher than 1, but we did not observe a relationship between
21
22 this ratio and the cross-linking speed, mechanical properties or swelling ratio. Indeed, solutions of
23
24 increasing DGL and PEG-NHS concentrations but of similar amines/NHS ratios resulted in
25
26 increasing cross-link velocities and mechanical properties. Similarly to PEG-PAMAM gels, the
27
28 effect of the polymer content could be logically ascribed to an increase of cross-linking density
29
30 due to a closer presence of the amine (DGL) and NHS (PEG) reactive groups. Once a PEG
31
32 molecule is attached to the surface of the dendrimer, the second NHS function on the other end of
33
34 the PEG chain will have lower mobility, which will increase its probability to react with an amine
35
36 function on the same dendrimer and form intramolecular loops.⁽²⁵⁾ Therefore, the dilution of the
37
38 reaction mixture increases the space between the dendrimers, forming many intramolecular loops
39
40 at the expense of effective networks chains. Conversely, concentrating the mixture results in the
41
42 opposite and increases the cross-link velocity.⁽²⁶⁾ This hypothesis is further reflected in the
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1
2
3
4
5 concomitant increase of mechanical properties and decrease of swelling observed in our hydrogels
6
7 when increasing the polymer concentration. Since the arborescent molecules function as junction
8
9 points in the cross-linking system, an increase in cross-links density will limit the expansion of the
10
11 network, meaning a more compact mesh that leads to a decrease in swelling and an increase in
12
13 rigidity. Similarly, Gitsov *et al.* observed that the swelling properties of hydrogels prepared from
14
15 PEG and various dendritic fragments were not related to the PEG concentration or the dendrimer
16
17 generation but mainly to the total increase in components concentration.⁽²⁷⁾
18
19
20
21
22
23

24 In addition to global polymer concentration, the DGL appears to have a predominant control of
25
26 the hydrogel cross-linking. While its concentration doubling from 1 to 2 mM for a constant PEG-
27
28 NHS concentration induces a 10-fold increase of the cross-linking velocity, a 1.6-fold increase of
29
30 PEG-NHS for a constant DGL concentration only marginally decreases cross-linking speed.
31
32
33 Although a DGL presents 114 amine groups to act as a cross-linker, it is unlikely that all the end
34
35 groups will be able to react with the PEG molecules due to steric crowding, which will limit the
36
37 maximum number of PEGs that can be grafted to a single dendrimer.⁽²⁶⁾ The fact that an increase
38
39 of PEG-NHS concentration for a constant DGL does not result in a significant increase of the
40
41 cross-linking speed supports this hypothesis by indicating that a maximal number of PEG per DGL
42
43 has been reached. Contrariwise, an increase of DGL at constant PEG-NHS allows to reduce the
44
45 steric crowding and increases further the cross-linking velocity.
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1
2
3
4
5 To provide to PEG hydrogels the ability to inherently interact with cells, our hypothesis that DGL
6
7 bioactivity could be conveyed to the hydrogel bulk was confirmed, as human fibroblasts were able
8
9
10 to adhere and proliferate on the surface of DGL/PEG hydrogels prepared without supplementary
11
12 coating before seeding. Similarly to DGLs coated on surfaces,⁽¹⁹⁾ the ability of cells to attach and
13
14 spread to DGL-containing hydrogels can be attributed to early electrostatic interactions between
15
16 the polyanionic cell surfaces and the polycationic charges brought by the DGL's amino groups.
17
18 This was further supported by the increase in cell adhesion with an increase in DGL concentration.
19
20
21 While the concentration of DGL plays a role in cell adhesion to the hydrogels, their viability seems
22
23 dependent of the ratio between DGL and PEG-NHS. Mortality was indeed observed when the ratio
24
25 of amines to NHS groups was greater than 9. The cationic nature of DGL that provides cell
26
27 adhesion properties is possibly also the cause of their cytotoxic effect, since excessive cationic
28
29 charges can affect the integrity of cell membranes.⁽²⁸⁾ When coupled with PEG molecules, PEG
30
31 might shield the dendritic cationic charges and improve the DGL viability, similarly to Tang *et al.*
32
33 who showed that PEG-gelation of DGL vectors for gene therapy allowed to decrease toxicity.⁽²⁹⁾
34
35
36 Unlike cell adhesion and viability, cell morphology and proliferation appear related to the overall
37
38 hydrogel rigidity. For instance, for a constant DGL concentration (2mM), an increase of PEG-
39
40 NHS concentration from 19 to 28 mM results in an increase of substrate rigidity (from 41.5±5.0
41
42 to 90.4±14.4 kPa) that is correlated with an increase of cell proliferation and a fusiform spread-
43
44 like morphology. Oppositely, an increase of DGL concentration from 2 to 3 mM for a constant
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1
2
3
4
5 PEG-NHS concentration (28mM) does not result in a significant increase of substrate rigidity and
6
7 does not influence cell proliferation or morphology. These observations are in good agreement
8
9
10 with several groups that have showed an increase in cell spreading area, stress fibers and
11
12 proliferation with an increase of substrate rigidity.^{(20),(30)–(34)}

13
14
15 As could be expected from PEG-based materials,⁽³⁵⁾ no intense immune reaction was observed
16
17 upon implantation of the DGL/PEG hydrogels under the back skin of mice for three weeks.
18
19
20
21 Consequently, the presence of DGL does not seem to induce a specific inflammatory reaction. The
22
23 controlled porosity in the hydrogels showed sufficiently interconnected to allow cell ingrowth,
24
25 vascularization and nutrient diffusion, which are important prerequisite for tissue engineering
26
27 applications.^{(36)–(38)} Interestingly, broader pores size distribution in the hydrogels resulted in a
28
29 higher vascularization, especially of vessels greater than 25 μm in diameter. Logically, bigger
30
31 pores allow bigger blood vessels to penetrate the hydrogels; however, this could only be observed
32
33 in rigid hydrogels since they are able to maintain their structure during implantation. In addition
34
35 to a good biocompatibility and a good cellular infiltration, the important presence of blood vessels
36
37 in the porous hydrogels is a further suggestion of the potential use of these novel hydrogels to
38
39 support tissue formation.
40
41
42
43
44
45
46
47

48 **5. Conclusions**

49
50
51 Through straight forward approaches, we have developed a novel PEG-hydrogel system cross-
52
53 linked with DGL which inherently allows human dermal fibroblasts attachment and proliferation.
54
55 Macroscopically, the hydrogels cross-link velocity, swelling and rigidity can be tailored by the
56
57
58
59
60

concentration of both components. The ability of cells to adhere on the hydrogels surface appears solely dependent of the concentration of DGL while a minimal concentration of PEG is needed to sustain the viability of the adhered cells. While cell behavior and morphology appear linked to substrate rigidity rather than hydrogel composition. Of note, the hydrogels are biocompatible and allow cell infiltration and blood vessels invasion when porous. All these elements emphasize the potential of these novel hydrogels for a vast range soft tissue applications.

Acknowledgments

This work was supported by la Région Auvergne-Rhône-Alpes (grant 17 002601 ARC 2016), CONACyT, i²t² and the French national research grant DHERMIC (ANR TECSAN 016-01). The authors would like to thank the platform PriMaTiss for the histological sample preparation and the IGFL (UMR 5242) for the access to the vibratome.

Conflicts of Interest

The authors declare no conflict of interest.

References

1. Tibbitt MW, Anseth KS. Hydrogels as Extracellular Matrix Mimics for 3D Cell Culture. *Biotechnol Bioeng.* 2009;103:655–63.
2. Jia X, Kiick KL. Hybrid multicomponent hydrogels for tissue engineering. *Macromol Biosci.* 2009;9:140–56.
3. Park S, Park K. Engineered Polymeric Hydrogels for 3D Tissue Models. *Polymers (Basel).* 2016;8:23.
4. Figueiredo L, Pace R, D'Arros C, Réthoré G, Guicheux J, Le Visage C, Weiss P. Assessing glucose and oxygen diffusion in hydrogels for the rational design of 3D stem cell scaffolds in regenerative medicine. *J Tissue Eng Regen Med.* 2018;12:1238–46.
5. Van Vlierberghe S, Dubruel P, Schacht E. Biopolymer-based hydrogels as scaffolds for tissue engineering applications: a review. *Biomacromolecules* [Internet]. 2011;12:1387–408. Available from: <http://dx.doi.org/10.1021/bm200083n>
6. Franz S, Rammelt S, Scharnweber D, Simon JC. Biomaterials Immune responses to implants - A review of

- the implications for the design of immunomodulatory biomaterials. *Biomaterials* [Internet]. Elsevier Ltd; 2011;32:6692–709. Available from: <http://dx.doi.org/10.1016/j.biomaterials.2011.05.078>
7. Lin C, Anseth KS. PEG Hydrogels for the Controlled Release of Biomolecules in Regenerative Medicine. *Pharm Res.* 2009;26:631–43.
 8. Burdick JA, Anseth KS. Photoencapsulation of osteoblasts in injectable RGD-modified PEG hydrogels for bone tissue engineering. 2002;23:4315–23.
 9. Rizzi SC, Hubbell JA. Recombinant Protein-co-PEG Networks as Cell-Adhesive and Proteolytically Degradable Hydrogel Matrixes. Part I: Development and Physicochemical Characteristics. *Biomacromolecules.* 2005;6:1226–38.
 10. Zhu J. Bioactive modification of poly(ethylene glycol) hydrogels for tissue engineering. *Biomaterials.* 2010;31:4639–56.
 11. Wathier M, Jung PJ, Carnahan MA, Kim T, Grinstaff MW. Dendritic Macromers as in Situ Polymerizing Biomaterials for Securing Cataract Incisions. *J Am Chem Soc.* 2004;126:12744–5.
 12. Wathier M, Johnson CS, Kim T, Grinstaff MW. Hydrogels Formed by Multiple Peptide Ligation Reactions To Fasten Corneal Transplants. *Bioconjugate Chem.* 2006;17:873–6.
 13. Kaga S, Arslan M, Sanyal R, Sanyal A. Dendrimers and Dendrons as Versatile Building Blocks for the Fabrication of Functional Hydrogels. *Molecules.* 2016;21.
 14. Oliveira JM, Salgado AJ, Sousa N, Mano JF, Reis RL. Dendrimers and derivatives as a potential therapeutic tool in regenerative medicine strategies—A review. *Prog Polym Sci* [Internet]. Elsevier Ltd; 2010;35:1163–94. Available from: <http://linkinghub.elsevier.com/retrieve/pii/S0079670010000456>
 15. N. Desai P, Yuan Q, Yang H. Synthesis and Characterization of Photocurable Polyamidoamine Dendrimer Hydrogels as a Versatile Platform for Tissue Engineering and Drug Delivery. *Biomacromolecules.* 2011;11:666–73.
 16. Navath RS, Menjoge AR, Dai H, Romero R, Kannan S, Kannan RM. Injectable PAMAM Dendrimer À PEG Hydrogels for the Treatment of Genital Infections : Formulation and in Vitro and in Vivo Evaluation. *Mol Pharm.* 2011;8:1209–23.
 17. Unal B, Hedden RC. Gelation and swelling behavior of end-linked hydrogels prepared from linear poly(ethylene glycol) and poly(amidoamine) dendrimers. *Polymer (Guildf).* 2006;47:8173–82.
 18. Huang R, Liu S, Shao K, Han L, Ke W, Liu Y, Li J, Huang S, Jiang C. Evaluation and mechanism studies of PEGylated dendrigraft poly-L-lysines as novel gene delivery vectors. *Nanotechnology* [Internet]. 2010;21:265101. Available from: <http://www.ncbi.nlm.nih.gov/pubmed/20522929>
 19. Lorion C, Faye C, Maret B, Trimaille T, Régnier T, Sommer P, Debret R. Biosynthetic support based on dendritic poly(L-lysine) improves human skin fibroblasts attachment. *J Biomater Sci Polym Ed* [Internet]. 2014 [cited 2016 Feb 3];25:136–49. Available from: <http://www.ncbi.nlm.nih.gov/pubmed/24116875>
 20. Nemir S, L. West J. Synthetic Materials in the Study of Cell Response to Substrate Rigidity. *Ann Biomed Eng.* 2010;38:2–20.
 21. Schneider CA, Rasband WS, Eliceiri KW. NIH Image to ImageJ: 25 years of image analysis. *Nat Methods* [Internet]. Nature Publishing Group; 2012 [cited 2019 Oct 25];9:671–5. Available from: <http://www.nature.com/articles/nmeth.2089>

- 1
2
3
4 22. Ma PX, Choi PDJ. Biodegradable Polymer Scaffolds with Well-Defined Interconnected Spherical Pore
5 Network. *Tissue e Eng.* 2001;7:23–33.
6
7 23. Anseth KS, Bowman CN, Brannon-Peppas L. Mechanical properties of hydrogels and their experimental
8 determination. *Biomaterials.* 1996;17:1647–57.
9
10 24. Coussot G, Nicol E, Commeryras A, Desvignes I, Pascal R, Vandenabeele-trambouze O. Colorimetric
11 quantification of amino groups in linear and dendritic structures. *Polym Int.* 2009;58:511–8.
12
13 25. Wang Y, Zhao Q, Zhang H, Yang S, Jia X. A novel poly(amido amine)-dendrimer-based hydrogel as a mimic
14 for the extracellular matrix. *Adv Mater.* 2014;26:4163–7.
15
16 26. Unal B, Hedden RC. Gelation and swelling behavior of end-linked hydrogels prepared from linear poly (
17 ethylene glycol) and poly (amidoamine) dendrimers. *Polym J.* 2006;47:8173–82.
18
19 27. Gitsov I, Zhu C. Amphiphilic hydrogels constructed by poly(ethylene glycol) and shape-persistent dendritic
20 fragments. *Macromolecules.* 2002;35:8418–27.
21
22 28. Quinton MP, Philpott CW. A Role for Anionic Sites in Epithelial Architecture: Effects of Cationic Polymers
23 on Cell Membrane Structure. *J Cell Biol.* 1978;56:787–96.
24
25 29. Tang M, Dong H, Li Y, Ren T. Harnessing the PEG-cleavable strategy to balance cytotoxicity, intracellular
26 release and the therapeutic effect of dendrigraft poly-L-lysine for cancer gene therapy. 1284 | *J Mater Chem*
27 *B* [Internet]. 2016 [cited 2019 Oct 21];4:1284. Available from: www.rsc.org/MaterialsB
28
29 30. Jiang G, Huang AH, Cai Y, Tanase M, Sheetz MP. Rigidity Sensing at the Leading Edge through avb3
30 Integrins and RPTPa. *Biophys J.* 2006;90:1804–9.
31
32 31. Yeung T, Georges PC, Flanagan LA, Marg B, Ortiz M, Funaki M, Zahir N, Ming W, Weaver V, Janmey PA.
33 Effects of Substrate Stiffness on Cell Morphology , Cytoskeletal Structure , and Adhesion. *Cell Motil*
34 *Cytoskeleton.* 2005;60:24–34.
35
36 32. Ghosh K, Pan Z, Guan E, Ge S, Liu Y, Nakamura T, Ren XD, Rafailovich M, Clark RAF. Cell adaptation to
37 a physiologically relevant ECM mimic with different viscoelastic properties. *Biomaterials.* 2007.
38
39 33. Lo CM, Wang HB, Dembo M, Wang YL. Cell movement is guided by the rigidity of the substrate. *Biophys J*
40 [Internet]. Elsevier; 2000;79:144–52. Available from: [http://dx.doi.org/10.1016/S0006-3495\(00\)76279-5](http://dx.doi.org/10.1016/S0006-3495(00)76279-5)
41
42 34. Solon J, Levental I, Sengupta K, Georges PC, Janmey PA. Fibroblast Adaptation and Stiffness Matching to
43 Soft Elastic Substrates. *Biophys J.* 2007;93:4453–61.
44
45 35. Lynn AD, Kyriakides TR, Bryant SJ. Characterization of the in vitro macrophage response and in vivo host
46 response to poly(ethylene glycol)-based hydrogels. *J Biomed Mater Res - Part A.* 2010;93:941–53.
47
48 36. J. Griffon D, Sedighi MR, Schaeffer D V., Eurell JA, Johnson AL. Chitosan scaffolds : Interconnective pore
49 size and cartilage engineering. *Acta Biomater.* 2006;2:313–20.
50
51 37. Rouwkema J, Rivron NC, Blitterswijk CA Van. Vascularization in tissue engineering. *Trends Biotechnol.*
52 2008;26:434–41.
53
54 38. Annabi N, Nichol JW, Zhong X, Ji C, Koshy S, Khademhosseini A, Dehghani F. Controlling the porosity and
55 microarchitecture of hydrogels for tissue engineering. *Tissue Eng Part B Rev* [Internet]. 2010 [cited 2015 Dec
56 5];16:371–83. Available from:
57 <http://www.pubmedcentral.nih.gov/articlerender.fcgi?artid=2946907&tool=pmcentrez&rendertype=abstract>
58
59
60

Figures

Figure 1. Photographs of the hydrogel at room temperature A) Dense 2/19 mM DGL/PEG hydrogel disc of 2mm thickness and 9.1mm diameter. B) Hydrogel in an inverted conic tube.

Figure 2. Swelling ratio of hydrogels of different compositions (hydrogel concentration is expressed as the ratio DGL/PEG in mM) swelling measured in PBS at 37°C. (Two-way ANOVA, $p < 0.05$ * at 24 hours).

Figure 3. Mechanical properties of dense hydrogel discs (2x9.1mm) of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM) measured at room temperature under PBS immersion by dynamical mechanical analysis in compression at a 10% strain and 60 μ m of amplitude. A) complex modulus, B) tan delta, C) storage modulus and D) loss modulus. (Two-way ANOVA, $p < 0.05$, *compared to 1/19, τ compared to 1/28 and γ compared to 2/19).

Figure 4. Dermal human fibroblast viability. A) Cytotoxicity by live/dead assay after 1 and 3 days of culture on the surface of hydrogels of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM). Live cells are observed in green and dead cells in red B) Total cell adhesion after 24 hours of cell seeding on top of hydrogels of different compositions. (One-way ANOVA, $p < 0.05$, *compared to control, τ compared to 1/19 and γ compared to 1/28) C) Viability percentage obtained from images analysis ($p < 0.05$ * compared to control).

Figure 5. Dermal fibroblast morphology when cultivated on the surface of hydrogels of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM) A) contrast phase photos after 24 hours and 72 hours of cell seeding B) Morphology parameters (spreading area, circularity and feret diameter) obtained from image analysis of contrast phase photos. Control was only analyzed after 24 hours since after 72 hours images analysis was not possible due to cell confluence. ($p < 0.05$, * compared to control at 24 hours).

Figure 6. A) Cytoskeleton structure of dermal fibroblasts seeded on the surface of hydrogel of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in

mM) observed by DAPI-Phalloidin staining after 1, 4 and 8 days of culture (in blue cell nuclei and in red actin fibers). B) Cell proliferation rate of fibroblasts seeded on the surface of hydrogel of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM) obtained from cell nuclei counting after 1, 4 and 8 days of culture. (One-way ANOVA, $p < 0.05$, *compared to control, τ compared to 1/19 and γ compared to 1/28).

Figure 7. Subcutaneous implantation in mice for three weeks of DGL/PEG porous hydrogels of different compositions (soft, medium and rigid, 7.7 ± 0.4 , 41.5 ± 5.0 and 76.7 ± 15.5 kPa, respectively) and with different pore sizes distribution. Masson's trichrome staining of the full explants and close-ups highlighting the hydrogel (#), the fibrous capsule (*), macrophages (+), synthesised collagen (α) and blood vessels (Φ).

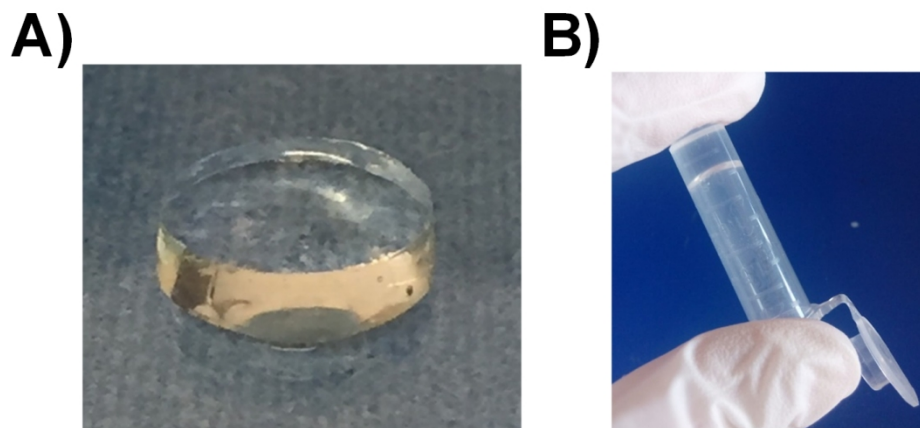
Figure 8. Blood vessels present after three weeks of subcutaneous implantation in mice in porous hydrogels of different compositions (soft, medium and rigid, 7.7 ± 0.4 , 41.5 ± 5.0 and 76.7 ± 15.5 kPa, respectively) A) Immunofluorescence staining of α -SMA (red) indicating blood vessels present within the porous hydrogels while cell nuclei are stained with DAPI in blue; white lines delimit the border of the hydrogel and arrows point to sectioned blood vessels. B) Number of blood vessels present in the hydrogel of a diameter greater than $10 \mu\text{m}$ or $25 \mu\text{m}$ obtained from image analysis of mosaic images of complete slides of porous hydrogels. (Two-way ANOVA, * $p < 0.05$).

Tables

Table 1. Cross-linking time to form self-standing hydrogels in relation to the final concentration of DGL and PEG-NHS and ratio of available amine/NHS functions (Two-way ANOVA between PEG and DGL concentration, * indicates that there is a statistically significant difference when DGL concentration is modified, for any given PEG concentration, $p < 0.05$)

Supplementary data

Supplementary figure 1. Subcutaneous implantation of dense DGL/PEG hydrogel (2/19mM DGL/PEG). Masson's trichrome staining of the full explants and close-ups highlighting the hydrogel (#), the fibrous capsule (*), macrophages (+).

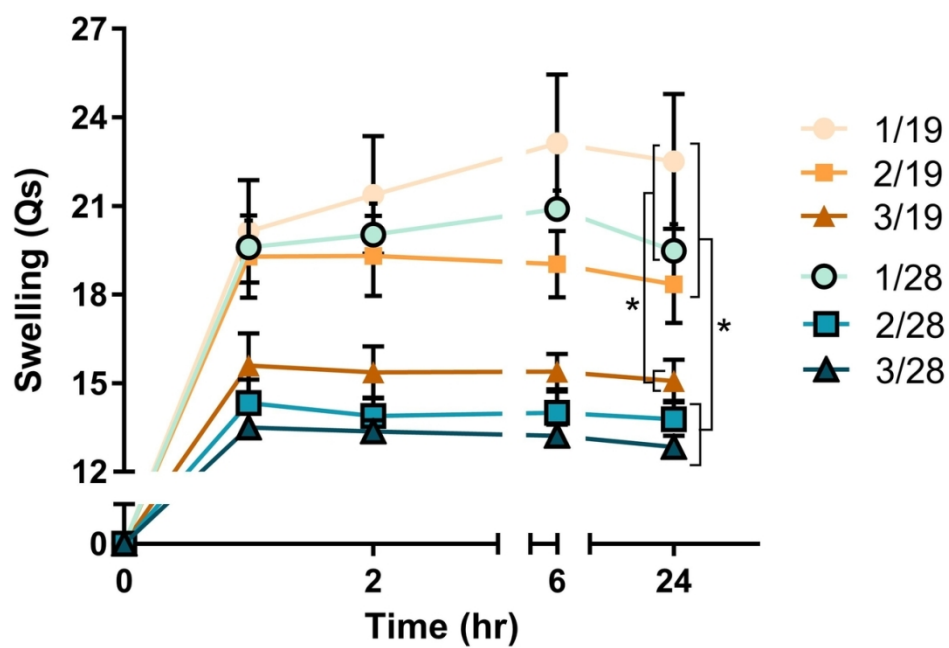


Photographs of the hydrogel at room temperature A) Dense 2/19 mM DGL/PEG hydrogel disc of 2mm thickness and 9.1mm diameter. B) Hydrogel in an inverted conic tube.

129x63mm (300 x 300 DPI)

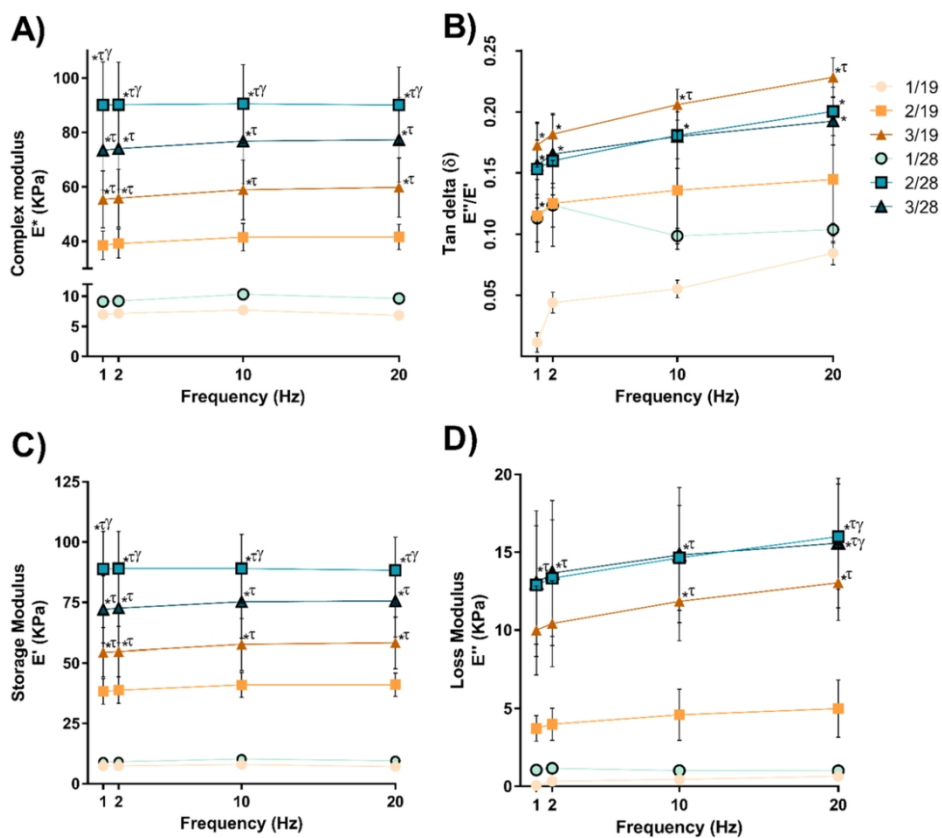
Table 1. Crosslinking time to form self-standing gels in relation to the final concentration of DGL and PEG-NHS and ratio of available amine/NHS functions (Two-way ANOVA between PEG and DGL concentration, * indicates that there is a statistically significant difference when DGL concentration is modified, for any given PEG concentration, $p < 0.05$)

Time (s) (Ratio amines/NHS)		PEG-NHS (mM) (Available NHS, mM)		
		19 (38)	28 (56)	30 (60)
DGL (mM)* (Available amines, mM)	1 (114)	144.6 ± 23.3 (3)	122.0 ± 24.0 (2.03)	140.2 ± 15.8 (1.9)
	2 (228)	14.94 ± 1.1 (6)	13.17 ± 1.9 (4.07)	12.28 ± 3.5 (3.8)
	3 (342)	10.47 ± 4.5 (9)	8.4 ± 1.2 (6.1)	7.66 ± 1.1 (5.7)
	4 (456)	6.77 ± 0.9 (12)	4.76 ± 0.6 (8.14)	5.47 ± 0.6 (7.6)



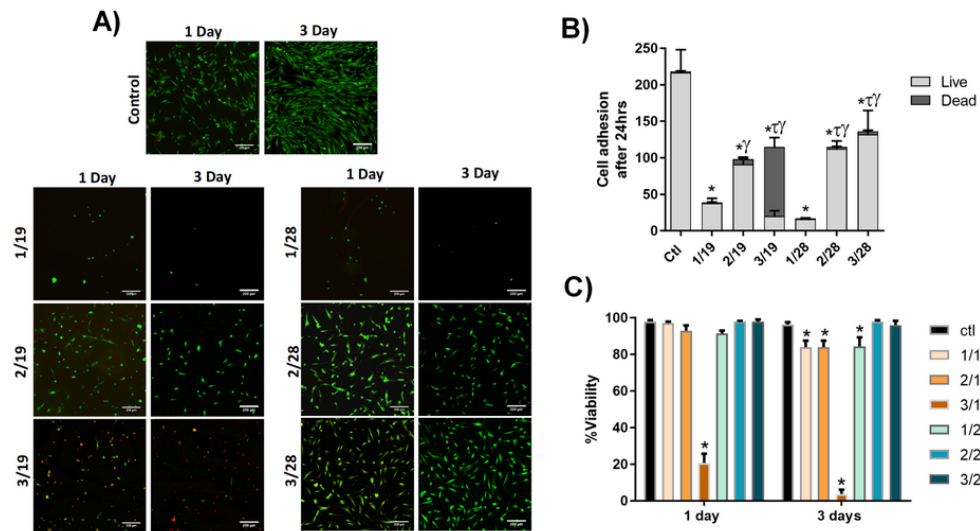
Swelling ratio of hydrogels of different compositions (hydrogel concentration is expressed as the ratio DGL/PEG in mM) swelling measured in PBS at 37°C. (Two-way ANOVA, $p < 0.05$ * at 24 hours).

117x83mm (300 x 300 DPI)



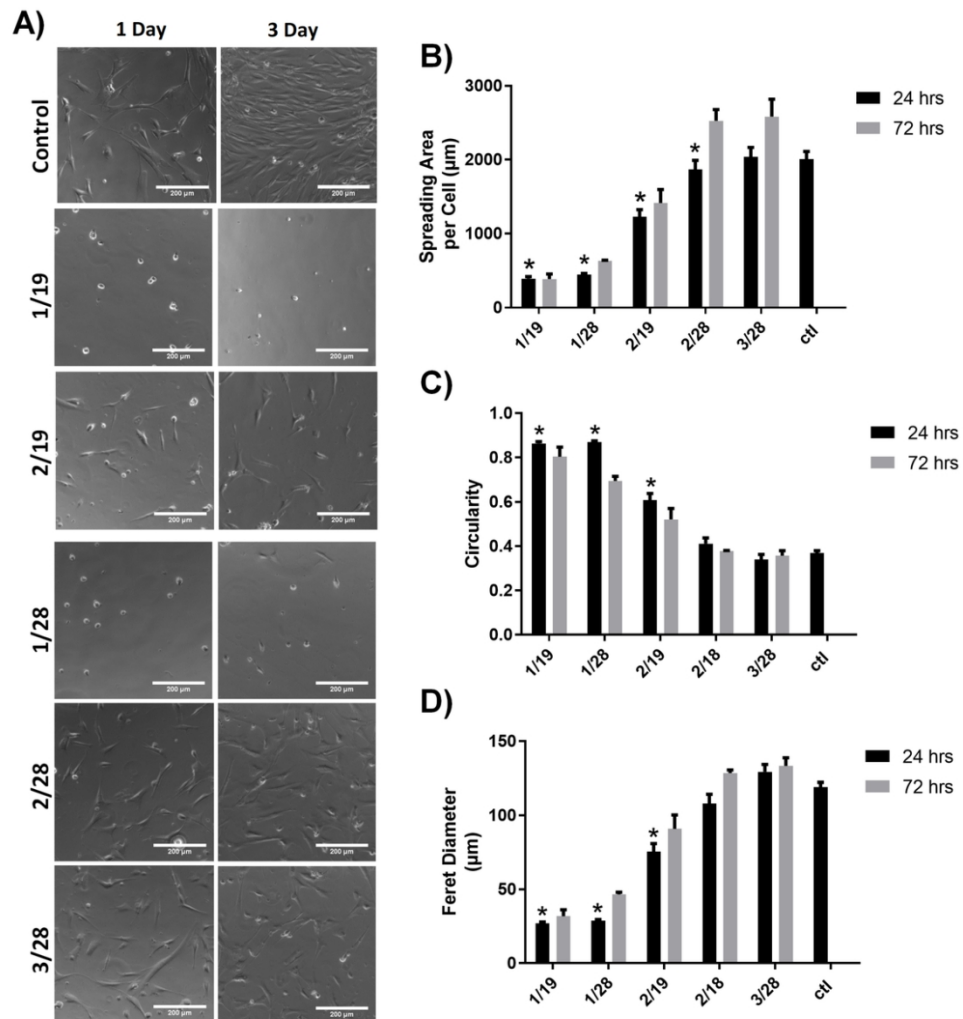
Mechanical properties of dense hydrogel discs (2x9.1mm) of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM) measured at room temperature under PBS immersion by dynamical mechanical analysis in compression at a 10% strain and 60 μ m of amplitude. A) complex modulus, B) tan delta, C) storage modulus and D) loss modulus. (Two-way ANOVA, $p < 0.05$, *compared to 1/19, τ compared to 1/28 and γ compared to 2/19).

99x88mm (300 x 300 DPI)



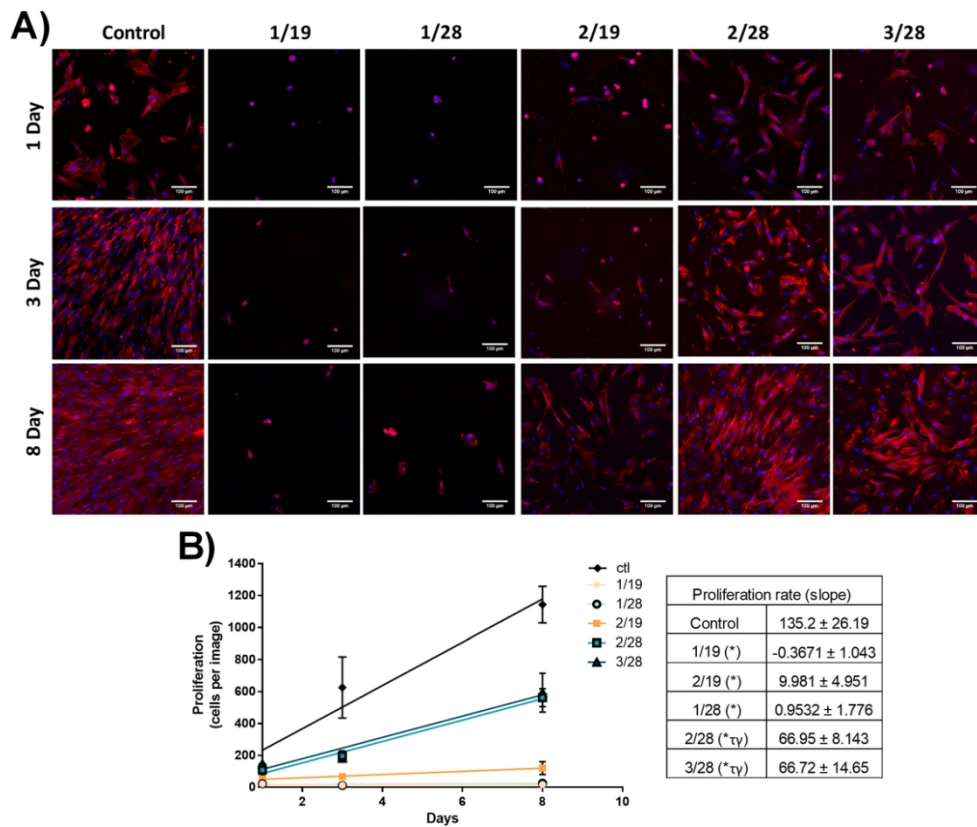
Dermal human fibroblast viability. A) Cytotoxicity by live/dead assay after 1 and 3 days of culture on the surface of hydrogels of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM). Live cells are observed in green and dead cells in red B) Total cell adhesion after 24 hours of cell seeding on top of hydrogels of different compositions. (One-way ANOVA, $p < 0.05$, *compared to control, τ compared to 1/19 and γ compared to 1/28) C) Viability percentage obtained from images analysis ($p < 0.05$ * compared to control).

79x43mm (300 x 300 DPI)



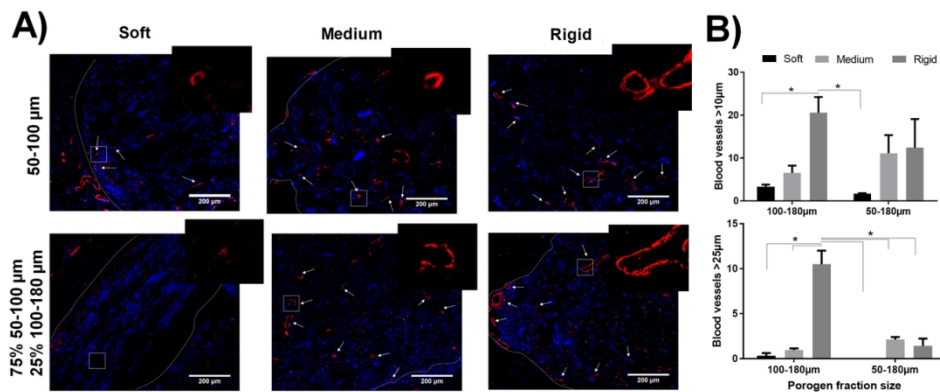
Dermal fibroblast morphology when cultivated on the surface of hydrogels of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM) A) contrast phase photos after 24 hours and 72 hours of cell seeding B) Morphology parameters (spreading area, circularity and feret diameter) obtained from image analysis of contrast phase photos. Control was only analyzed after 24 hours since after 72 hours images analysis was not possible due to cell confluence. ($p < 0.05$, * compared to control at 24 hours).

99x104mm (300 x 300 DPI)



A) Cytoskeleton structure of dermal fibroblasts seeded on the surface of hydrogel of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM) observed by DAPI-Phalloidin staining after 1, 4 and 8 days of culture (in blue cell nuclei and in red actin fibers). B) Cell proliferation rate of fibroblasts seeded on the surface of hydrogel of different compositions (hydrogel concentration is expressed as de ratio DGL/PEG in mM) obtained from cell nuclei counting after 1, 4 and 8 days of culture. (One-way ANOVA, $p < 0.05$, *compared to control, τ compared to 1/19 and γ compared to 1/28).

99x86mm (300 x 300 DPI)



Blood vessels present after three weeks of subcutaneous implantation in mice in porous hydrogels of different compositions (soft, medium and rigid, 7.7 ± 0.4 , 41.5 ± 5.0 and 76.7 ± 15.5 kPa, respectively) A) Immunofluorescence staining of α -SMA (red) indicating blood vessels present within the porous hydrogels while cell nuclei are stained with DAPI in blue; white lines delimit the border of the hydrogel and arrows point to sectioned blood vessels. B) Number of blood vessels present in the hydrogel of a diameter greater than 10 μm or 25 μm obtained from image analysis of mosaic images of complete slides of porous hydrogels. (Two-way ANOVA, * $p < 0.05$).

119x51mm (300 x 300 DPI)