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# Reactive surface of glass particles under aqueous corrosion 

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#### Abstract

Glass dissolution rates are normalized to the glass surface area in contact with solution, and experiments are very often carried out using crushed and sieved materials whose size is narrowed between an upper and a lower value. Surface area of such particles could be determined by gas adsorption or geometric considerations. Although crushed particles cannot be assimilated with simple geometric shapes, rates normalized with-spheres of the same size-geometric surface area are underestimated but are close to those found for polished monoliths. Overestimation of the reactive surface when using gas adsorption measurements is discussed. © 2017 The Authors. Published by Elsevier B.V. This is an open access article under the CC BY-NC-ND license (http://creativecommons.org/licenses/by-nc-nd/4.0/). Peer-review under responsibility of the organizing committee of WRI-15 Keywords: Glass; Surface; Initial rate; ISG; BET; AFM.


## 1. Introduction

With the choice of geological repository for nuclear waste glasses, many countries are driven to study aqueous dissolution of glass to assess long-term behavior of glass canisters. Many experimental protocols meant to study glass dissolution mechanisms and kinetics involve particles. Two main methods are currently used to measure their surface area: gas adsorption with application of the BET model ${ }^{1}$ or geometrical measurement-assuming glass particles as smooth and non-porous spheres.

Surface areas measured by gas adoption ( $S_{\mathrm{BET}}$ ) are systematically higher than geometric surface areas ( $S_{\text {geo }}$ ) by a factor of 1.6 to $4.5^{2,3}$. This gap raise a persisting debate in the community about the best way to evaluate the "reactive surface" of glass particles, defined here as the relevant glass/water interfacial area.

[^0]The quantification of this surface is important for the description and the modeling of interfacial processes such as glass dissolution. This surface area has also a direct influence on the normalized alteration rates measured during leaching tests (commonly expressed in $\mathrm{g} \cdot \mathrm{m}^{-2} \cdot \mathrm{~d}^{-1}$ ). Previous studies ${ }^{2-5}$ concluded that alteration rates normalized to $S_{\text {geo }}$ are closer-than those normalized to $S_{\mathrm{BET}}$ - to those measured on polished monoliths. However, crushed glass particles are not spherical and their surface irregularities are not taken into account in $S_{\text {geo }}$ calculation. Improving the methodology for reactive surface measurement and understanding the difference between $S_{\text {BET }}$ and $S_{\text {geo }}$ is the objective of this experimental study.

## 2. Materials and methods

### 2.1. Glass samples

Two glasses (Table 1) were used in the present study: the international nuclear reference glass (ISG glass) and Type S SiLibeads ${ }^{\circledR}$ glass beads (Sigmund Lindner) whose geometry is mastered and produced from soda lime glass by a thermal rounding process. ISG glass powders were obtained by successive crushing and sieving steps. Rectangular monoliths of these two glasses were cut and polished using SiC abrasive papers before a final polishing involving diamond suspensions of $6 \mu \mathrm{~m}, 3 \mu \mathrm{~m}$ and $1 \mu \mathrm{~m}$.

Table 1. International Simple Glass (ISG) and Type S glass compositions expressed in oxide weight percent.

| ISG glass | $\mathrm{SiO}_{2}$ | $\mathrm{~B}_{2} \mathrm{O}_{3}$ | $\mathrm{Na}_{2} \mathrm{O}$ | $\mathrm{Al}_{2} \mathrm{O}_{3}$ | CaO | $\mathrm{ZrO}_{2}$ |
| :--- | :--- | :--- | :--- | :--- | :--- | :--- |
| Oxide wt $\%$ | $56.2 \pm 1.5$ | $17.3 \pm 0.9$ | $12.2 \pm 0.7$ | $6.1 \pm 0.8$ | $5.0 \pm 0.6$ | $3.3 \pm 0.5$ |
| Type S glass | $\mathrm{SiO}_{2}$ | $\mathrm{Na}_{2} \mathrm{O}$ | CaO | MgO | $\mathrm{Al}_{2} \mathrm{O}_{3}$ |  |
| Oxide wt $\%$ | $72.5 \pm 0.9$ | $13.0 \pm 0.8$ | $9.1 \pm 0.3$ | $4.2 \pm 0.1$ | $0.6 \pm 0.2$ |  |

### 2.2. Solid analyses

Surface area measurements. $S_{\text {BET }}$ was measured by Kr adsorption on sample surface (Micromeritics ASAP 2020). The estimation of the amount of gas needed to form a monolayer on the solid surface was estimated by the BET model ${ }^{1}$. $S_{\text {geo }}$ was calculated assuming a spherical geometry for smooth non-porous glass particles with normally distributed sizes: $S_{\text {geo }}=3 /(\rho \cdot R)$ where $\rho$ is the glass density and $R$ the average radius of the particles. $S_{\text {geo }}$ of monoliths was determined using a digital caliper (their surface area is far too low to perform $S_{\mathrm{BET}}$ measurements).

Atomic Force Microscopy (AFM). A Multimode 8 and Nanoscope V controller (Veeco, Santa Barbara, CA) was used in Tapping Mode (free amplitude $\approx 1 \mathrm{~V}$ with RTESP antimony ( n ) doped silicium tips, $K=40 \mathrm{~N} \cdot \mathrm{~nm}^{-1}$, $f_{0}=300 \mathrm{kHz}$, Bruker) or PeakForce Mode (force $\approx 100 \mathrm{pN}$ with a SNL silicon tip on Nitride lever, $K=0.35 \mathrm{~N} \cdot \mathrm{~nm}^{-1}$, Bruker). Measured topography data were processed by NanoScope Analysis software v1.40 to calculate the difference $\Delta S$ between the analyzed region's three dimensional surface area and its two-dimensional, footprint area.

Leaching tests and solution analysis. Glass initial dissolution rates ( $r_{0}$ ) were determined by static leaching tests at $90^{\circ} \mathrm{C}$. The leaching solution consisted of a KOH solution giving a $\mathrm{pH}_{90^{\circ} \mathrm{C}}$ of $10 \pm 0.1$, continuously stirred and always sufficiently diluted ( $[\mathrm{Si}]<3 \mathrm{mg} \cdot \mathrm{L}^{-1}$ ). Dissolved silica concentrations-used for $r_{0}$ calculations-were determined photometrically (Merck Spectroquant ${ }^{\circledR}$ Silicate Test, Cary ${ }^{\circledR} 50$ Scan UV-Vis spectrophotometer) with a method analogous to ASTM D859-10.

## 3. Results and discussion

## 3.1. $S_{B E T}$ vs $S_{g e o}$

For ISG crushed glass particles, the $S_{\text {BET }} / S_{\text {geo }}$ ratio is $2.6 \pm 0.2$ (Table 2), showing the difference between a smooth, non-porous sphere and a glass particle with a complex shape and presenting surface features accessible to gas atoms. For Type $S$ beads-highly spherical sample (sphericity values of 0.92 to $0.97^{6}$ )—this ratio is reduced to $1.7 \pm 0.4$ (Table 2), but still higher than 1 . This indicates that glass surface present defects accessible to gas atoms, increasing $S_{\mathrm{BET}}$. The difference between $S_{\mathrm{BET}}$ and $S_{\text {geo }}$ can thus be explained by the existence of a "shape factor" $F_{s}$
(i.e. the difference between a sphere and the real shape of the particle) and a "roughness factor" $F_{r}$ due to the sample surface features.

Table 2. Values of $S_{\mathrm{BET}}$ and $S_{\text {geo }}$ for ISG and Type S glass particles as a function of particle size.

| Glass | Size fraction $(\mu \mathrm{m})$ | $S_{\text {BET }}\left(\mathrm{cm}^{2} \cdot \mathrm{~g}^{-1}\right)$ | $S_{\text {geo }}\left(\mathrm{cm}^{2} \cdot \mathrm{~g}^{-1}\right)$ | $S_{\text {BET }} / S_{\text {geo }}$ |
| :--- | :--- | :--- | :--- | :--- |
| ISG | $20-40$ | 2065 | 800 | 2.6 |
| ISG | $40-63$ | 1120 | 466 | 2.4 |
| ISG | $63-125$ | 630 | 255 | 2.5 |
| ISG | $125-250$ | 345 | 128 | 2.7 |
| Type S SiLibeads $^{\circledR( }$ | $90-150$ | 270 | 200 | 1.4 |
| Type S SiLibeads $^{\circledR}$ | $150-250$ | 155 | 120 | 1.3 |
| Type S SiLibeads $^{\circledR}$ | $250-500$ | 125 | 64 | 2.0 |
| Type S SiLibeads $^{\circledR}$ | $500-750$ | 75 | 38 | 2.0 |
| Type S SiLibeads $^{\circledR}$ | $750-1000$ | 51 | 27 | 1.9 |

### 3.2. Characterization of samples roughness measurable by AFM

For polished monoliths, $\Delta S$ (the surface increase due to roughness) is close to $1 \%$ (Fig. 1.a). For Type S glass beads (Fig. 1.b), roughness measurements are similar within the same particle size but vary from one granulometry to another. $\Delta S$ of Type S glass beads does not exceed $10 \%$ - note that the higher $\Delta S$ is, the higher is the $S_{\mathrm{BET}} / S_{\text {geo }}$ ratio. AFM analyses are much more difficult to be performed for ISG glass crushed particles (Fig. 1.c) because of the step heights (e.g. conchoidal fractures): the only areas on which a measurement is possible are those located near the upper surface and oriented in parallel to the scanning plane. In such areas, $\Delta S$ doesn't exceed $30 \%$. For both types of samples-and despite the non-representativeness of ISG glass particle analyses- $\Delta S$ values cannot explain the entire gap between $S_{\mathrm{BET}}$ and $S_{\text {geo. }}$. This difference could then be explained by the fact that the cross section of a Kr atom (20 $\AA^{2}$ ) is smaller than the AFM tip (10-20 nm). These two "probes" do not have the same size, so they do not address the same defects size scale: surface features taken into account for $S_{\text {BET }}$ are predominantly smaller than those achievable for AFM.


Fig. 1. $10 \mu \mathrm{~m} \times 10 \mu \mathrm{~m}$ AFM images of (a) ISG glass polished monolith ( $\Delta S=1.0 \%$ );
(b) 500-750 $\mu \mathrm{m}$ Type $S$ glass bead ( $\Delta S=4.6 \%$ ); (c) 63-125 $\mu \mathrm{m}$ ISG glass crushed particle ( $\Delta S=1.1 \%$ ).

### 3.3. Initial dissolution rate measurements

For ISG glass particles, $r_{0}$ measurements (Fig. 2.a) were conducted 5 times on the same 63-125 $\mu \mathrm{m}$ particle batch (relative standard deviation $\operatorname{RSD}=2.6 \cdot 10^{-2}$ ), on 5 monoliths ( $\mathrm{RSD}=4.5 \cdot 10^{-2}$ ), and twice on the three other particle sizes $\left(\operatorname{RSD}=8.5 \cdot 10^{-2}\right.$ ). RSD is lower for tests conducted on the same particle batch (good repeatability of measurements) and is multiplied by $\approx 3$ when various glass particle sizes are used. For Type $S$ glass (Fig. 2.b), the measurements were performed twice for each bead size $\left(\operatorname{RSD}=13.7 \cdot 10^{-2}\right)$ and on 3 monoliths ( $\mathrm{RSD}=7.4 \cdot 10^{-2}$ ).

Measurements performed on ISG glass and normalized to $S_{\text {geo }}$ are closer to those obtained with monoliths but do not overlap (Fig. 2.a). This overlap is clearly visible for Type $S$ glass beads because of their spherical shape (Fig. 2.b). Fig. 2 indicates that for ISG crushed particles: $F_{s}=1.3, F_{r}=1.9$ and for Type S glass beads: $F_{s} \approx 1, F_{r}=1.7$. These results show that surface features accessible to gas atoms-predominantly contributing to $S_{\mathrm{BET}}$-have little effect on the glass dissolution rate, presumably because they are too small to contribute for a significant time to the ingress of water.


Fig. 2. Comparison of $r_{0}$ measurements on (a) ISG glass and (b) Type S glass. $r_{0}$ measured on glass particles are normalized to $S_{\text {geo }}$ or $S_{\text {BET }}$.

## 4. Conclusions

The accurate measurement of the external surface of fragmented and non-porous materials-with complex morphologies and roughness-is difficult and the uncertainty of $r_{0}$ measurements can reach $\pm 25 \%$. Both reasons partly explain why the debate around glass reactive surface is lasting for so long. In order to be compared to the dissolution rate measured on polished monoliths, the rate measured on glass crushed particles ( $S_{\mathrm{BET}} / S_{\text {geo }} \approx 2.5$ ) and normalized to $S_{\text {geo }}$ should be divided by 1.3 while the rate normalized to $S_{\mathrm{BET}}$ should be multiplied by 1.9.

Both particles shape and their surface state contribute to the difference between $S_{\text {BET }}$ and $S_{\text {geo }}$. This difference is mainly explained by surface features accessible to gas atoms. Their nature and the way they interact with water are unknown. It can also be concluded that $S_{\text {BET }}$ should be used with caution for glass/water reactive interfacial area measurement.

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