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DEPLETION-DRIVEN MORPHOLOGICAL TRANSITIONS

IN HEXAGONAL CRYSTALLITES OF VIRUS RODS

B. Sung,¹ H. H. Wensink,² E. Grelet,^{1,*}

¹ *Centre de Recherche Paul Pascal, UMR 5031,*

CNRS & Université de Bordeaux, 33600 Pessac, France

² *Laboratoire de Physique des Solides, UMR 8502,*

CNRS & Université Paris-Sud, Université Paris-Saclay, 91405 Orsay, France

Abstract

The assembly of nanometer-sized building blocks into complex morphologies is not only of fundamental interest but also plays a key role in material science and nanotechnology. We show that the shape of self-assembled superstructures formed by rod-shaped viruses can be controlled by tuning the attraction via the depletion interaction between the rods. Using non-adsorbing polymers as a depleting agent, we demonstrate that a hierarchical unidimensional self-organization into crystalline clusters emerges progressively upon increasing depletion attraction and enhanced growth kinetics. We observe a polymorphic change proceeding from two-dimensional (2D) crystalline monolayers at weak depletion to one-dimensional (1D) columnar fibers at strong depletion, via the formation of smectic fibrils at intermediate depletion strength. A simple theory for reversible polymerization enables us to determine the typical bond energy between monomeric units making up the smectic fibrils. We also demonstrate that gentle flow-assistance can be used to template filament-like structures into highly aligned supported films. Our results showcase a generic bottom-up approach for tuning the morphology of crystalline superstructures through modification of the interaction between non-spherical building blocks. This provides a convenient pathway for controlling self-organization, dimensionality and structure-formation of anisotropic nanoparticles for use in nanotechnology and functional materials.

31 **Introduction**

32 The ability to control the shape and symmetry of structures formed through self-assembly
33 processes is of fundamental importance in fabricating highly-ordered nano- and micro-structures
34 (1) (2). The large diversity of morphologies emerging from such superstructures finds applications
35 in photonics (3) (4) (5), electronics (6) (7), sustainable energy (8) (9) (10), drug delivery (11) (12)
36 and bioengineering (13) (14). Finding ways to guide nanoparticle self-assembly into mono- and
37 multi-layer structures, bundles, tubes, spherical or polyhedral shells, and more complex
38 geometries has been one of the most pursued research themes in nanoscience. One of the main
39 driving forces for studying controlled nanoparticle self-assembly is the many applications in
40 nanotechnology ranging from tunable optical properties (15) (16), anisotropic electron transport
41 (17) (18), to functional surface topography (19) (20). Among such complex architectures,
42 unidimensional (1D) fibers and bidimensional (2D) superlattices composed of rod-shaped
43 building blocks are of particular interest in view of their intrinsic anisotropic properties. For
44 example, fibers of 1D aligned carbon nanotubes, obtained from various spinning techniques,
45 exhibit high mechanical stiffness and outstanding electrical performance (21) (22) (23).
46 Moreover, 2D long-ranged crystalline monolayers of inorganic nanorods can be used for
47 generating surface plasmonics or piezoelectricity, or for creating photonic crystals (24) (25) (26)
48 (27).

49 Contrary to spherical nanoparticles, reports on controlled self-assembly of rod-shaped
50 particles are scarce, and no clear indications of polymorphisms in their self-assembled structures
51 have been adequately demonstrated so far. In view of this, we address the following two
52 questions. First, what variety of ordered anisotropic superstructures can be expected if we were
53 able to systematically tune the interparticle forces in a colloidal suspension of rods? And more
54 specifically, is it possible to define a pathway enabling us to control the shape of these self-
55 assembled objects from predominantly unidimensional (fiber-like) to bidimensional (disk-shaped)
56 without compromising their long-range crystalline order? For this purpose, the depletion
57 interaction (31) appears to be a suitable and versatile mechanism for controlling particle
58 interaction, as the range and strength of depletion interaction are determined by the polymer size
59 and concentration, respectively (32). We will show that introducing tunable depletion interactions
60 with a suitable range indeed leads to a high degree of controlled self-assembly generating a wide
61 variety of crystallite morphologies without the need to use microscale confinements (28) or
62 interfacial effects (29) (30).

63 It is well known that adding non-adsorbing polymers to colloidal suspension provides an
64 efficient method to induce effective particle-particle attraction that can direct hierarchical

65 organization processes (33) (34) (35) (36) (37) (38) (39). Experimental and theoretical studies on
66 model systems of rod-like particles such as fd viruses have established that the experimental
67 conditions are such that the formation of partially-ordered structures, such as nematic tactoids,
68 membranes and their stacks, as well as twisted ribbons (37) (40) are strongly favored. These
69 depletion-driven self-assembled morphologies have been obtained using high molecular-weight
70 depletants such as Dextran (molecular weight, $M_w = 500,000$ g/mol; radius of gyration, $R_g \sim 21$
71 nm), whose size is larger than the particle diameter. This results in the virus rods self-assembling
72 into super-structures with predominant liquid-like or short-ranged positional order in at least two
73 spatial directions (37) (40) (41) (42). However, in order to fully benefit from the morphology
74 effect in terms of enhanced anisotropic properties of the resulting materials, highly ordered
75 structures with crystalline, i.e. long-ranged positional order are required (4) (5) (10).

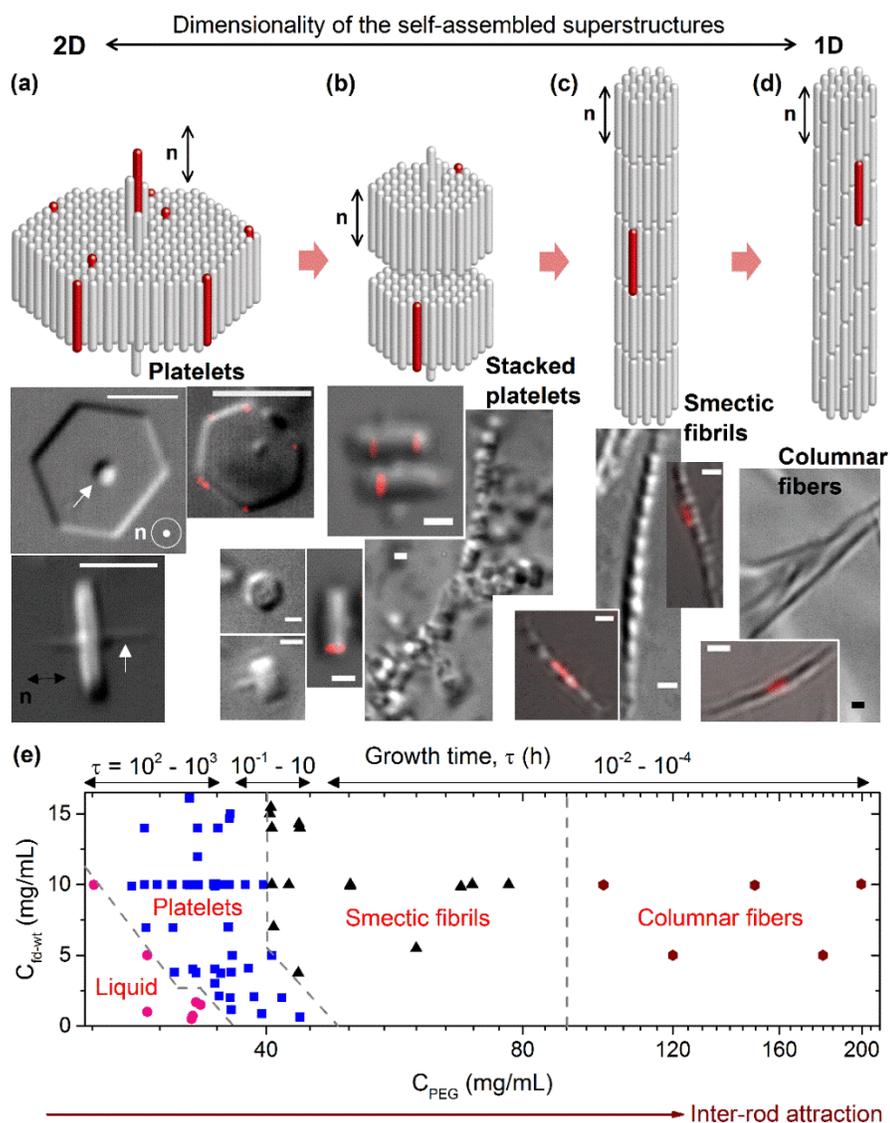
76 In order to promote dense self-assembled structures with crystalline order short-ranged
77 depletion attractions are required favoring small inter-rod distances, s . This effect can be achieved
78 using a non-adsorbing polymer of small molecular weight (PEG, $M_w = 8,000$ g/mol; $R_g \sim 4$ nm)
79 added to a suspension of fd virus. The key-point here is that corresponding polymer coil has about
80 the same size as the virus diameter, $2.R_g \sim d = 7$ nm. Taking advantage of the micrometer length of
81 fd virus we are able to analyze the structures formed by these filamentous rods at the single
82 particle scale using fluorescence optical microscopy (39) (43). We then investigate the detailed
83 morphology of the virus-based superstructures upon varying both the depletion strength (polymer
84 concentration) and rod volume fraction, while keeping the polymer size fixed thus ensuring the
85 rod-rod interactions to be short-ranged.

86 **Results and discussion**

87
88 The experimental results are presented in Figure 1. Here, we demonstrate that subtle
89 morphological changes occur in a hierarchical way upon increasing the depletion strength. At low
90 polymer concentration, when the depletion attraction is weak, the rods tend to aggregate side-by-
91 side (44) thus minimizing their excluded volume. This results in the formation of 2D plate-shaped
92 crystallites. These platelets have well-developed hexagonal facets (Figure 1a) and are composed
93 of a single layer of hexagonally packed aligned viral rods with a protruding needle-like defect
94 located at the core (43). The core-defect acts as a steric deterrent preventing the platelets to stack
95 on top of each other.

96 Enhancing the depletion attraction causes an increase in the density number ρ_0 of plate-
97 shaped nuclei and a simultaneous decrease of the platelet diameter ($D=1-2$ μm) as well as a length
98 reduction of the central core defect (Figures 1b and S2). This results in a decrease of the average

99 platelet diameter D with increasing the depletion strength (Figure S3), since $\rho_0 \propto 1/D^2$ as dictated
 100 by particle mass conservation.



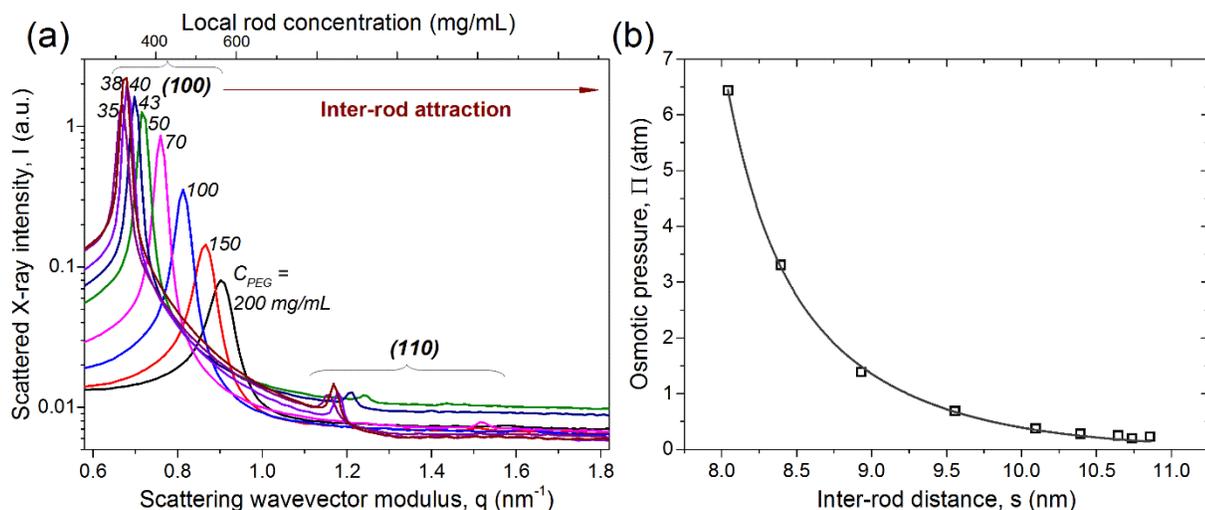
101 **Figure 1. Overview of morphological and dimensional changes in self-assembled structures formed**
 102 **by rod-like fd viruses driven by increasing depletion attraction.** The kinetic processes determining the
 103 typical growth time for the various structures speeds up considerably with increasing inter-rod attraction.
 104 The self-assembled morphology changes from (a) hexagonal platelets composed of a one-rod thick
 105 monolayer of aligned fd rods with a central core defect (indicated by a white arrow), (b) small hexagonal
 106 platelets and their stacks. When the attraction increases further, the rods self-assemble into (c) smectic
 107 fibrils composed of small polymerized platelet monomers, and then to (d) columnar fibers. The
 108 fluorescently labeled viruses in the DIC/fluorescence overlaid images show that the rods are aligned
 109 parallel to the main axis \mathbf{n} of the cluster (black double arrow). Scale bars, 5 μm for (a) and 1 μm for (b-d).
 110 (e) Phase diagram as a function of the rod (C_{fd-wt}) and polymer (C_{PEG}) concentrations, where the typical
 111 growth time τ is also indicated. At each phase boundary, a gradual change of morphology occurs rather
 112 than a sharp transition.
 113

114 Upon increasing the depletion strength, the self-assembly process speeds up considerably
115 resulting in much shorter typical growth times. Pushing the depletion strength even further we
116 observe a high population of small platelets (of about $D \sim 1 \mu\text{m}$). Without a detectable central
117 defect preventing their clustering, the small platelets begin to stack up and finally self-assemble
118 into smectic fibrils in which each layer represents a small hexagonal platelet with a width of
119 about one rod length (Figure 1c). This quasi-1D morphology referring to the overall resulting
120 shape of the self-assembly can be interpreted as the result of fast-growing hexagonal platelets
121 reversible polymerizing into smectic filaments. Clearly, the effective depletion attraction acting
122 between platelets promotes face-to-face stacking in order to maximize the free volume available
123 to the polymer (32). In order to estimate the typical bond energy between monomeric platelets
124 making up a smectic fibril, we invoke a simple reversible polymerization model described in the
125 Supporting Information. This theory assumes the smectic fibrils to operate in thermodynamic
126 equilibrium in which case the fibril height distribution can be determined analytically and
127 compared to experimental results depicted in Fig S4a. From the fits (see Supporting Information)
128 we estimate a typical bonding energy between monomeric platelets of about $10 k_B T$ and a typical
129 ‘bonding volume’ of effective range of attraction of about 8-10 times the polymer volume. The
130 ‘bonding volume’ which can be interpreted as the bonding range times the platelet area is
131 considerably less than the volume $\sim R^2 R_g$ with R the radius of a flat plate one would naively
132 obtain from free-volume theory (32) if the face-to-face bonding was entirely due to ideal
133 depletion attractions. Complicating factors such as thermal corrugation of the plate surface (due to
134 fluctuations of the rods around their centers of mass), electrostatic interactions and semi-
135 flexibility of the filamentous virus rods may account for the strongly reduced bond energy but
136 these effects are notoriously difficult to capture within a simple, tractable model.

137 At the highest depletion attraction the growth kinetics speeds up dramatically resulting in
138 the rods self-assembling into columnar bundles (Figure 1d). In these structures, there is no layered
139 organization along the main axis of the fiber but the hexagonal crystalline structure transverse to
140 the bundle director \mathbf{n} is preserved. The time required for the self-assembly processes, or growth
141 time, is reduced dramatically when the growth morphology changes from the predominantly
142 bidimensional platelets to the unidimensional fibers and bundles.

143 Throughout the range of cluster morphologies observed, the typical growth time varies by
144 more than six orders of magnitude, taking up days for large platelets to form, a few minutes for
145 the smectic fibrils, down to milliseconds for the fast formation of columnar fibers. Apart from the
146 intrinsic rod-rod attraction, kinetics play an important role in templating the self-assembled
147 structures obtained. In view of the ultrafast growth process, we expect the formation of columnar

148 bundles to be strongly marked by non-equilibrium effects, as columnar structures composed of
 149 long rods usually do not emerge from thermodynamic considerations (32) (46). In all the
 150 observed superstructures, the orientation of the rods (\mathbf{n}) remains parallel to the main axis of the
 151 crystallite, as evidenced from the presence of a small fraction of virus rods labelled with a red
 152 fluorescent dye (Figure 1) and using a full-wavelength retardation plate (Figure S1). The absence
 153 of any detectable particle self-diffusion within the clusters (Movies S1 to S5) is consistent with
 154 the long-range positional order along the direction normal to \mathbf{n} observed by small-angle X-ray
 155 scattering (SAXS) experiments (Figure 2).

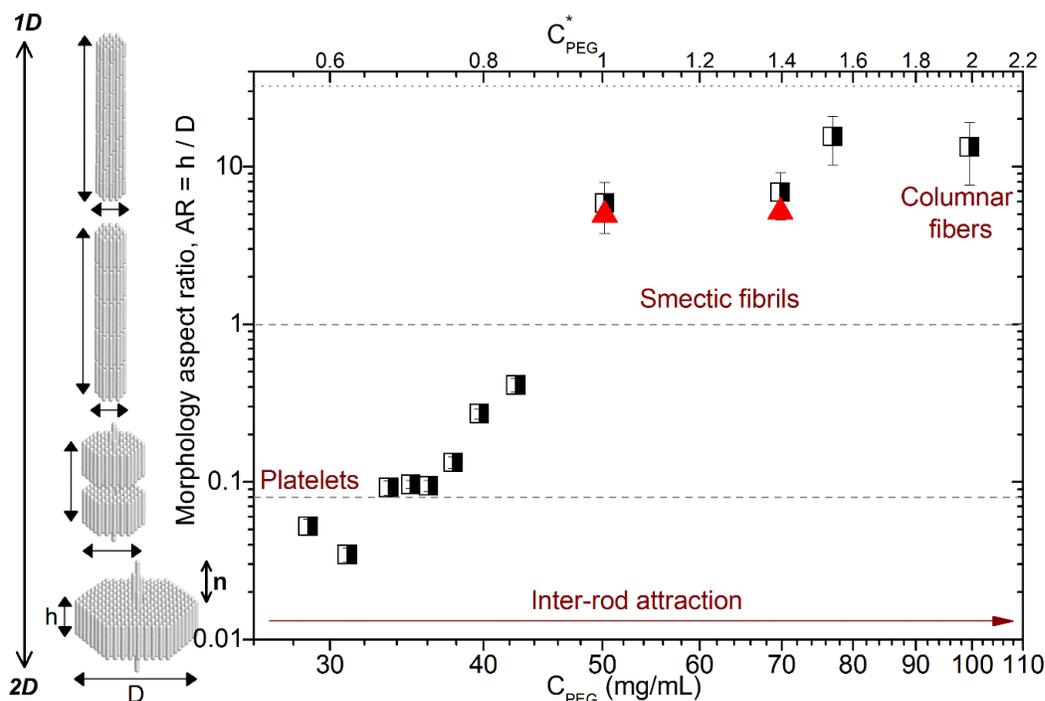


156
 157 **Figure 2. SAXS data of the different virus-based self-assembled superstructures obtained by varying**
 158 **the concentration of PEG depletant (C_{PEG}).** The initial virus concentration is set at 10 mg/mL. (a) SAXS
 159 spectra characteristic of hexagonal long-ranged positional order with the presence of sharp 100 and 110
 160 Bragg reflections. The virus concentration *within* the self-assembled clusters, referred to as the local
 161 concentration, is calculated from the 2D swelling law associated with a hexagonal ordering of long rods
 162 (46) for a given (100) peak position. (b) Osmotic pressure (Π) obtained from the PEG concentration (47)
 163 as a function of inter-rod distance, $s = (4\pi/\sqrt{3})q_{100}^{-1}$. The fit (black curve) is based on the model of Yasar
 164 et al. (48) described in the Supporting Information.

165
 166 The SAXS experiments have been conducted at a fixed rod concentration but at varying
 167 polymer concentration and depletion strength. All SAXS spectra (Figure 2a) display (100) and
 168 (110) diffraction peaks, characteristic of local hexagonal rod packing. The interaxial distance
 169 between the neighboring rods $s = (4\pi/\sqrt{3})q_{100}^{-1}$ can be directly obtained from the position of
 170 the main diffraction reflection. There is only little variation of the distance s (Figure 2b)
 171 accompanying the vast morphological changes observed throughout the same range of conditions.
 172 Since the interaxial rod distance s probed by SAXS is about two orders of magnitude smaller

173 compared to the rod length ℓ , a 2D swelling law ($q_{100} = \eta \cdot C_{\text{rod}}^{1/2}$) of the hexagonal lattice is
 174 expected, with $\eta = (8\pi^2 N_A \ell / (\sqrt{3} Mw))^{0.5}$ (46). From this, the local rod concentration
 175 associated with the various self-assembled morphologies can be deduced. This gives a value of
 176 around 300 mg/mL for the platelets, 350-400 mg/mL for the smectic fibrils, and 450-550 mg/mL
 177 for the columnar bundles (Figure 2a).

178 The depletion interaction between the virus rods can be accounted for by invoking a
 179 polymer-generated osmotic pressure Π acting on the virus arrays. Here, one simply assumes that
 180 most of the polymer is excluded from the virus-based superstructures (45). Such models,
 181 described in the Supporting Information, have been successfully applied to many charged rod-
 182 like particles (46) (47), and give an accurate description of the experimental data as shown in
 183 Figure 2b.



184
 185 **Figure 3. Morphological transition from 2D to 1D self-assembled structures observed by enhancing**
 186 **the inter-rod attraction, in turn controlled by the strength of the depletion interaction.** The cluster
 187 aspect ratio AR, defined as the dimension parallel to \mathbf{n} (h ; height) over the dimension perpendicular to \mathbf{n}
 188 (D ; diameter) where \mathbf{n} denotes the main axis of the cluster morphology, is plotted (black and white
 189 squares) as a function of the depletant concentration. Experimentally, the largest measurable length is
 190 around 20 μm (the corresponding aspect ratio is indicated by a dotted line), since longer fibrils no longer
 191 lie within the same focal plan. The red triangular symbols represent the theoretical predictions from a
 192 simple reversible polymerization theory describing the formation of smectic fibrils (see Supporting
 193 Information), and they show a good quantitative agreement with experimental data. The typical bond
 194 energy between monomeric platelets within the smectic fibrils is estimated to be about 10 $k_B T$.

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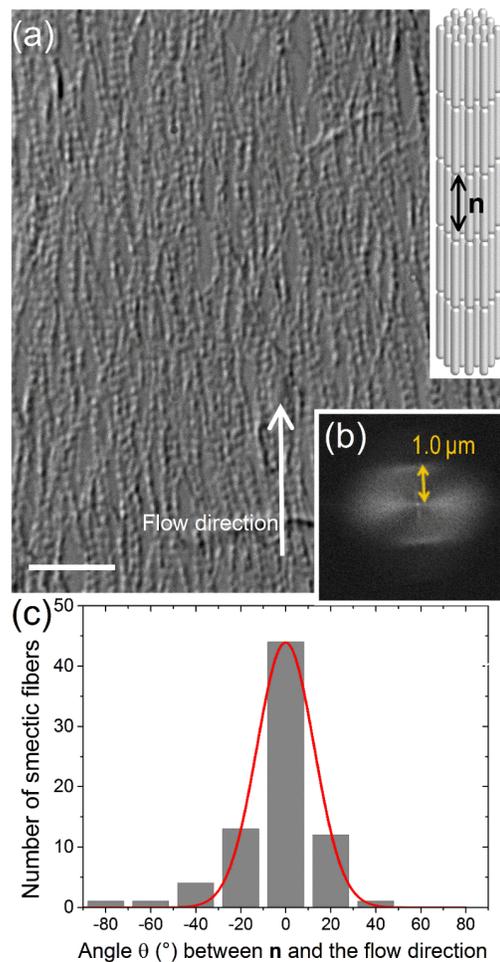
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Figure 3 displays the crossover from one self-assembled superstructure to another in terms of the aspect ratio $AR=h/D$ of the typical cluster morphology, with h and D indicating the height and diameter of the superstructure measured parallel and perpendicular to the rod director \mathbf{n} , respectively. Throughout the range of attractions, the diameter decreases from 20-30 μm for the platelets while reaching a minimum of around 0.9 μm for the smectic fibers and columnar bundles (Supporting Information, Figures S3 and S4). The height increases from one rod-length for single platelets to 10-20 μm for smectic fibers and columnar bundles (Supporting Information, Figures S3 and S4). Consequently, the aspect ratio AR of the cluster morphology varies by almost three orders of magnitude throughout the probed range of depletion strengths. This demonstrates a high degree of shape-tunability of the self-assembled clusters, while guaranteeing the long-ranged positional order within the structures as demonstrated by the SAXS experiments.



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Figure 4. Flow assisted alignment of smectic fibrils deposited on a glass substrate. (a) DIC image of the smectic fibrils, whose main axes (\mathbf{n}) are along the flow direction. Scale bar, 10 μm . (b) Fourier transform pattern of the DIC image in (a), with the one-rod-thick smectic layer spacing indicated. (c) Distribution of smectic fibril orientations fitted by a Gaussian function (red line), from which the 2D

orientational order parameter, $S = \langle 2 \cos^2(\theta) - 1 \rangle = 0.91$, can be obtained after numerical integration.

Films of highly aligned smectic fibers can be obtained upon applying weak shear flow, as shown in Figures 4a. The degree of alignment can be established from a 2D nematic order parameter $S = \langle 2 \cos^2(\theta) - 1 \rangle$ that we can extract from the orientational distribution of smectic fibrils (Figure 4c). Typical values we found are of about $S = 0.90$ (note that $S = 1$ indicates perfect alignment, and $S = 0$ complete orientational disorder). This shows that these self-assembled superstructures are strongly susceptible to external orientation stimuli and can be readily shaped into macroscopically aligned domains. The facile processability of these rod-based superstructures opens perspectives for applications requiring anisotropic stimuli-responsive materials with internal crystalline order (9) (10) (14) (29) (50).

Conclusion

In conclusion, we have described a tunable self-assembly process of rod-like particles forming superstructures with an unprecedented control of their cluster morphology, effective dimension and internal microstructure. In our system of filamentous fd virus rods mixed with PEG polymer, controlled self-assembly is achieved through the use of tunable short-ranged depletion interactions between the rods generated by the presence of non-adsorbing polymer with typical size comparable to the rod diameter. A systematic increase in the polymer concentration causes a number of morphological transitions affecting the principal dimensionality of the self-assembled structure. These range from isolated quasi-bidimensional platelets and stacked oligomeric platelets at weak depletion to unidimensional polymeric smectic fibrils, and columnar fibers at strong depletion. These polymorphological changes occur without compromising the long-ranged hexagonal crystal order within the superstructures. The average inter-rod distance is only weakly affected by the depletion strength. Simple thermodynamic considerations enable us to predict the typical fibril length and infer the typical bond energy between monomeric platelets within these structures. Last but not least, we have demonstrated the possibility of flow-assisted templating of such anisotropic superstructures into highly aligned supported films demonstrating a high degree of processability. By expanding the spectrum of bottom-up approaches towards anisotropic nanoparticles we argue that depletion-driven rod suspensions displaying hierarchically-tunable morphogenesis constitute a promising candidate for fabricating self-assembled structures for use in devices depending on anisotropic stimuli-responsive materials.

Materials and Methods

Virus stock suspensions

Wild-type fd bacteriophages were grown with ER2738 strain of *E. coli* as host bacteria, and were purified according to standard protocols (51). The fd-wt (wild type) viruses are rod-like particles, monodisperse in size and shape, with contour length $\ell = 880$ nm and diameter $d = 7$ nm. In suspension, the virus rods have been shown to behave as near-hard rods exhibiting a well-defined liquid-crystalline phase behavior (41) (39) (40) (46). Virus suspensions were extensively dialyzed against TRIS-NaCl-HCl buffer (pH 8.2) at 110 mM of ionic strength, to ensure the electrostatic repulsion between viral particles is strongly screened (Debye screening length, $\lambda_D = 0.9$ nm). The rods were subsequently concentrated using ultracentrifugation and redispersed at 30-35 mg/mL in the same buffer solution. The virus concentration was determined using spectrophotometry at the peak absorption wavelength of 269 nm with an optical density of $3.84 \text{ cm}^2/\text{mg}$ (52). Fluorescently-labeled virus batches were separately prepared by conjugating Alexa Fluor 488-TFP ester (Invitrogen) or Dylight550-NHS ester (ThermoFischer) to their coat proteins. These labeled viruses were added in tracer amounts (0.001% (w/w) to 0.1% (w/w)) to the non-labeled virus batches for tracking by fluorescence microscopy.

Sample preparation

Polyethylene glycol (PEG) of $M_w = 8,000$ g/mol (Sigma-Aldrich) was used as a non-absorbing polymer with a radius of gyration (R_g) of 4 nm in aqueous solution (53). All polymer-virus mixtures were prepared in TRIS-NaCl-HCl buffer (pH 8.2) adjusted at 110 mM of ionic strength, and injected into optical microscopy cells. The cells were made by a glass slide and a coverslip (initially cleaned with sulfochromic acid) separated by a Mylar or Parafilm spacer, to obtain a cell thickness of about 100 μm . The cells were then sealed with UV-cured glue (NOA81, Epotecn). For the film of aligned smectic fibers, the samples were either prepared by drop casting on a cover slip, or loaded and oriented in the cell through capillary forces. Samples for small-angle X-ray scattering (SAXS) experiments were prepared in cylindrical quartz capillaries (diameter ~ 1.5 mm; Mark-Röhrchen), filled with aqueous virus-polymer mixtures ($\sim 20 \mu\text{l}$ for each), and sealed by flame. The capillaries were positioned vertically (gravity along the main axis of a capillary) for about 4 weeks in order to induce macroscopic phase separation through sedimentation of the self-assembled structures at the capillary bottom.

Optical microscopy

Differential interference contrast (DIC) and epifluorescence images were obtained using an inverted microscope (IX71, Olympus) equipped with an oil-immersion objective (NA 1.4, $\times 100$

280 UPLSAPO), a mercury-halide excitation source (X-cite120Q, Excelitas), and a fluorescence
281 imaging camera (NEO sCMOS, Andor Technology). To enable simultaneous acquisition of DIC
282 and fluorescence images, an optical splitter setup (Optosplit II, Cairn Research) was used to
283 divide each image into two channels thanks to a dichroic beamsplitter and band pass filters. The
284 images obtained from both channels were then overlaid. The whole imaging system was operated
285 by computer-interface software (Meta-Morph, Molecular Devices).

286 *Small angle X-ray scattering (SAXS)*

287 SAXS measurements were performed at the SWING beamline at the synchrotron facility SOLEIL
288 (Orsay, France) operating at a source wavelength of $\lambda = 0.0995$ nm. The diffraction pattern was
289 recorded by an AVIEX CCD detector, which was located in a vacuum detection tunnel with a
290 sample-to-detector distance of 1.49 m. Angular integration was applied on the 2D SAXS pattern
291 to obtain the scattering intensity as a function of scattering vector modulus, q .

292 **References**

- 293 1. *Anisotropy of building blocks and their assembly into complex structures.* **S. C. Glotzer, M. J.**
294 **Solomon.** 2007, Nat. Mater., Vol. 6, pp. 557-562.
- 295 2. *Self-assembly: from crystals to cells.* **B. A. Grzybowski, C. E. Wilmer, J. Kim, K. P. Browne, K. J.**
296 **M. Bishop.** 2009, Soft Matter, Vol. 5, pp. 1110-1128.
- 297 3. *Morphological control in colloidal crystal templating of inverse opals, hierarchical structures, and*
298 *shaped particles.* **A. Stein, F. Li, N. R. Denny.** 2008, Chem. Mater., Vol. 20, pp. 649-666.
- 299 4. *Biomimetic self-templating supramolecular structures.* **W-J. Chung, J-W. Oh, K. Kwak, B. Y. Lee, J.**
300 **Meyer, E. Wang, A. Hexemer, S-W. Lee.** 2011, Nature, Vol. 478, pp. 364-368.
- 301 5. *Self-assembled colloidal superparticles from nanorods.* **T. Wang, J. Zhuang, J. Lynch, O. Chen, Z.**
302 **Wang, X. Wang, D. LaMontagne, H. Wu, Z. Wang, Y. C. Cao.** 2012, Science, Vol. 338, pp. 358-363.
- 303 6. *Chemical self-assembly for electronic applications.* **J. H. Fendler.** 2001, Chem. Mater., Vol. 13, pp.
304 3196-3210.
- 305 7. *Using self-assembly for the fabrication of nano-scale electronic and photonic devices.* **B. A. Parviz, D.**
306 **Ryan, G. M. Whitesides.** 2003, IEEE Trans. Adv. Packaging, Vol. 26, pp. 233-241.
- 307 8. *Recent progress in morphology control of supramolecular fullerene assemblies and its applications.* **S.**
308 **S. Babu, H. Möhwald, T. Nakanishi.** 2010, Chem. Soc. Rev., Vol. 39, pp. 4021-4035.
- 309 9. *Stamped microbattery electrodes based on self-assembled M13 viruses.* **K. T. Nam, R. Wartena, P. J.**
310 **Yoo, F. W. Liau, Y. J. Lee, Y-M. Chiang, P. T. Hammond, A. M. Belcher.** 2008, Proc. Natl. Acad. Sci.
311 U.S.A., Vol. 105, pp. 17227-17231.
- 312 10. *Virus-based piezoelectric energy generation.* **B. Y. Lee, J. Zhang, C. Zueger, W-J. Chung, S. Y.**
313 **Yoo, E. Wang, J. Meyer, R. Ramesh, S-W. Lee.** 2012, Nat. Nanotechnol., Vol. 7, pp. 351-356.

316 11. **D. D. Lasic, D. Papahadjopoulos, R. Podgornik.** Polymorphism of lipids, nucleic acids, and their
317 interactions. *Self-Assembling Complexes for Gene Delivery* (A. V. Kabanov, P. L. Felgner, L. W. Seymour).
318 Chichester : Wiley, 1998, pp. 3-26.

319 12. *Surfactant self-assembly objects as novel drug delivery vehicles.* **C. J. Drummond, C. Fong.** 2000,
320 *Curr. Opin. Colloid Interface Sci.*, Vol. 4, pp. 449-456.

321 13. *Biomimeticism and bioinspiration as tools for the design of innovative materials and systems.* **C.**
322 **Sanchez, H. Arribart, M. M. Giraud Guille.** 2005, *Nat. Mater.*, Vol. 4, pp. 277-288.

323 14. *Genetically engineered nanofiber-like viruses for tissue regenerating materials.* **A. Merzlyak, S.**
324 **Indrakanti, S-W. Lee.** 2009, *Nano Lett.*, Vol. 9, pp. 846-852.

325 15. *Self-assembly of photonic crystals from polymer colloids.* **J. Zhang, Z. Sun, B. Yang.** 2009, *Curr.*
326 *Opin. Colloid & Interface Sci.*, Vol. 14, pp. 103-114.

327 16. *Pyrite nanocrystals: shape-controlled synthesis and tunable optical properties via reversible self-*
328 *assembly.* **W. Li, M. Döblinger, A. Vaneski, A. L. Rogach, F. Jäckel, J. Feldmann.** 2011, *J. Mater.*
329 *Chem.*, Vol. 21, pp. 17946-17952.

330 17. *Columnar self-assembly of colloidal nanodisks.* **A. E. Saunders, A. Ghezelbash, D-M. Smilgies, M.**
331 **B. Sigman, Jr., B. A. Korgel.** 2006, *Nano Lett.*, Vol. 6, pp. 2959-2963.

332 18. *Hierarchical self-assembly of metal nanostructures on diblock copolymer scaffolds.* **W. A. Lopes, H.**
333 **M. Jaeger.** 2001, *Nature*, Vol. 414, pp. 735-738.

334 19. *Self-directed self-assembly of nanoparticle/copolymer mixtures.* **Y. Lin, A. Böker, J. He, K. Sill, H.**
335 **Xiang, C. Abetz, X. Li, J. Wang, T. Emrick, S. Long, Q. Wang, A. Balazs, T. P. Russel.** 2005, *Nature*,
336 Vol. 434, pp. 55-59.

337 20. *Gold mesostructures with tailored surface topography and their self-assembly arrays for surface-*
338 *enhanced raman spectroscopy.* **J. Fang, S. Du, S. Lebedkin, Z. Li, R. Kruk, M. Kappes, H. Hahn.**
339 2010, *Nano Lett.*, Vol. 10, pp. 5006-5013.

340 21. *High-performance carbon nanotube fiber.* **K. Koziol, J. Vilatela, A. Moisala, M. Motta, P. Cunniff,**
341 **M. Sennett, A. Windle.** 2007, *Science*, Vol. 318, pp. 1892-1895.

342 22. *Super-tough carbon-nanotube fibres.* **A. B. Dalton, S. Collins, E. Muñoz, J. M. Razal, V. H. Ebron,**
343 **J. P. Ferraris, J. N. Coleman, B. G. Kim, R. H. Baughman.** 2003, *Nature*, Vol. 423, p. 703.

344 23. *Carbon nanotube fiber microelectrodes.* **J. Wang, R. P. Deo, P. Poulin, M. Mangey.** 2003, *J. Am.*
345 *Chem. Soc.*, Vol. 125, pp. 14706-14707.

346 24. *Large-scale hexagonal-patterned growth of aligned ZnO nanorods for nano-optoelectronics and*
347 *nanosensor arrays.* **X. Wang, C. J. Summers, Z. L. Wang.** 2004, *Nano Lett.*, Vol. 4, pp. 423-426.

348 25. *Vertically aligned gold nanorod monolayer on arbitrary substrates: self-assembly and femtomolar*
349 *detection of food contaminants.* **B. Peng, G. Li, D. Li, S. Dobson, Q. Zhang, J. Zhang, Y. H. Lee, H. V.**
350 **Demir, X. Y. Ling, Q. Xiong.** 2013, *ACS Nano*, Vol. 7, pp. 5993-6000.

351 26. *Liquid crystal self-assembly of upconversion nanorods enriched by depletion forces for mesostructured*
352 *material preparation.* **Y. Xie, Y. Li, G. Wei, Q. Liu, H. Mundoor, Z. Chen, I. Smalyukh.** 2018,
353 *Nanoscale*, Vol. 10, pp. 4218–4227.

354 27. *Symmetry control of nanorod superlattice driven by a governing force.* **Y. Liang, Y. Xie, D. Chen, C.**
355 **Guo, S. Hou, T. Wen, F. Yang, K. Deng, X. Wu, I. I. Smalyukh, and Q. Liu.** 2017, *Nat. Commun.*,
356 Vol. 8, p. 1410.

357 28. *Controllable synthesis and self-assembly of PbCO₃ nanorods in shape-dependent nonionic w/o*
358 *microemulsions.* **J. Zhang, P. R. Lang, W. Pyckhout-Hintzen, J. K. G. Dhont.** 2013, *Soft Matter*, Vol.
359 9, pp. 7576-7582.

360 29. *Phage-based structural color sensors and their pattern recognition sensing system.* **J. H. Lee, B. Fan,**
361 **T. D. Samdin, D. A. Monteiro, M. S. Desai, O. Scheideler, H-E. Jin, S. Kim, S-W. Lee.** 2017, *ACS*
362 *Nano*, Vol. 11, pp. 3632-3641.

363 30. *Self-assembly of gold nanorods into symmetric superlattices directed by OH-terminated hexa(ethylene*
364 *glycol) alkanethiol.* **Y. Xie, S. Guo, Y. Ji, C. Guo, X. Liu, Z. Chen, X. Wu, Q. Liu.** 2011, *Langmuir*,
365 Vol. 27, pp. 11394-11400.

366 31. *Interaction between particles suspended in solutions of macromolecules.* **S. Asakura, F. Oosawa.**
367 1958, *J. Polym. Sci.*, Vol. 33, pp. 183-192.

368 32. **H. N. W. Lekkerkerker, R. Tuinier.** *Colloids and the Depletion Interaction.* Dordrecht : Springer,
369 2011.

370 33. *Macromolecular crowding for materials-directed controlled self-assembly.* **Y. Hata, T. Sawada, T.**
371 **Serizawa.** 2018, *J. Mater. Chem. B*, Vol. 6, pp. 6344-6359.

372 34. *Directing colloidal self-assembly through roughness-controlled depletion attractions.* **K. Zhao, T. G.**
373 **Mason.** 2007, *Phys. Rev. Lett.*, Vol. 99, p. 268301.

374 35. *Assembly of colloidal semiconductor nanorods in solution by depletion attraction.* **D. Baranov, A.**
375 **Fiore, M. van Huis, C. Giannini, A. Falqui, U. Lafont, H. Zandbergen, M. Zanella, R. Cingolani, L.**
376 **Manna.** 2010, *Nano Lett.*, Vol. 10, pp. 743-749.

377 36. *Assembly of reconfigurable one-dimensional colloidal superlattices due to a synergy of fundamental*
378 *nanoscale forces.* **K. L. Young, M. R. Jones, J. Zhang, R. J. Macfarlane, R. Esquivel-Sirvent, R. J.**
379 **Nap, J. Wu, G. C. Schatz, B. Lee, C. A. Mirkin.** 2012, *Proc. Natl. Acad. Sci. U.S.A.*, Vol. 109, pp. 2240-
380 2245.

381 37. *Reconfigurable self-assembly through chiral control of interfacial tension.* **T. Gibaud, E. Barry, M. J.**
382 **Zahkary, M. Henglin, A. Ward, Y. Yang, C. Berciu, R. Oldenbourg, M. F. Hagan, D. Nicastro, R. B.**
383 **Meyer, Z. Dogic.** 2012, *Nature*, Vol. 481, pp. 348-351.

384 38. *Phase Behavior of Mixtures of Rods (Tobacco Mosaic Virus) and Spheres (Polyethylene Oxide, Bovine*
385 *Serum Albumin).* **M. Adams, S. Fraden.** 1998, *Biophys. J.*, Vol. 74, pp. 669-677.

386 39. *Curvature instability of chiral colloidal membranes on crystallization.* **L. Saikia, T. Sarkar, M.**
387 **Thomas, V. A. Raghunathan, A. Sain, P Sharma.** 2017, *Nat. Commun.*, Vol. 8, pp. 1160.

- 388 40. *Self-assembly of 2D membranes from mixtures of hard rods and depleting polymers.* **Y. Yang, E.**
389 **Barry, Z. Dogic, M. F. Hagan.** 2012, *Soft Matter*, Vol. 8, pp. 707-714.
- 390 41. *Development of model colloidal liquid crystals and the kinetics of the isotropic–smectic transition.* **Z.**
391 **Dogic, S. Fraden.** 2001, *Phil. Trans. R. Soc. London A*, Vol. 359, pp. 997-1015.
- 392 42. *Filamentous phages as building blocks for reconfigurable and hierarchical self-assembly.* **Gibaud, T.**
393 2017, *J. Phys.: Condens. Matter*, Vol. 29, p. 493003.
- 394 43. *Chirality-controlled crystallization via screw dislocations.* **B. Sung, A. de la Cotte, E. Grelet.** 2018,
395 *Nat. Commun.*, Vol. 9, p. 1405.
- 396 44. *Tunable attractions directing nonequilibrium states in dispersions of hard rods.* **M. P. B. van**
397 **Bruggen, H. N. W. Lekkerkerker.** 2000, *Macromolecules*, Vol. 33, pp. 5532-5535.
- 398 45. *Direct Liquid to Crystal Transition in a Quasi-Two-Dimensional Colloidal Membrane.* **T. Gibaud, D.**
399 **Constantin.** 2018, *J. Phys. Chem. Lett.*, Vol. 9, pp. 4302-4307.
- 400 46. *From soft to hard rod behavior in liquid crystalline suspensions of sterically stabilized colloidal*
401 *filamentous particles.* **E. Grelet, R. Rana.** 2016, *Soft Matter*, Vol. 12, pp. 4621-4627.
- 402 47. *Measuring osmotic pressure of poly(ethylene glycol) solutions by sedimentation equilibrium*
403 *ultracentrifugation.* **C. B. Stanley, H. H. Strey.** 2003, *Macromolecules*, Vol. 36, pp. 6888-6893.
- 404 48. *Continuity of states between the cholesteric -> line hexatic transition and the condensation transition*
405 *in DNA solutions.* **S. Yasar, R. Podgornik, J. Valle-Orero, M. R. Johmson, V. A. Parsegian.** 2014, *Sci.*
406 *Rep.*, Vol. 4, p. 6877.
- 407 49. *Structural Effects of Single Mutations in a Filamentous Viral Capsid Across Multiple Length Scales.*
408 **G. Abramov, R. Shaharabani, O. Morag O, R. Avinery, A. Haimovich, I. Oz, R. Beck, A. Goldbourt.**
409 *Biomacromolecules.* 2017, Vol. 18, 2258-2266.
- 410 50. *Competition between crystal and fibril formation in molecular mutations of amyloidogenic peptides.* **N.**
411 **P. Reynolds, J. Adamcik, J. T. Berryman, S. Handschin, A. A. H. Zanjani, W. Li, K. Liu, A. Zhang,**
412 **R. Mezzenga.** 2017, *Nat. Commun.*, Vol. 8, pp. 1338.
- 413 51. **J. Sambrook, D. W. Russell.** *Molecular Cloning: A Laboratory Manual (3rd edition).* Cold Spring
414 Harbor, NY : Cold Spring Harbor Laboratory Press, 2001.
- 415 52. *A model liquid crystalline system based on rodlike viruses with variable chirality and persistence*
416 *length.* **E. Barry, D. Beller, Z. Dogic.** 2009, *Soft Matter*, Vol. 5, pp. 2563-2570.
- 417 53. *Asymptotic behavior and long-range interactions in aqueous solutions of poly(ethylene oxide).* **K.**
418 **Devanand, J. C. Selser.** 1991, *Macromolecules*, Vol. 24, pp. 5943-5947.
- 419 54. *Theory of Chain Association versus Liquid Condensation.* **R. van Roij.** 1996, *Phys. Rev. Lett.*, Vol.
420 76, pp. 3348.
- 421 55. *The equilibrium size distribution of rouleaux,* **A. S. Perelson, F. W. Wiegel.** 1982, *Biophys J.*, Vol.
422 37, pp. 515-522.
- 423 56. *Smectic filaments in colloidal suspensions of rods.* **D. Frenkel, T. Schilling.** 2002, *Phys. Rev. E*, Vol.
424 66, pp. 041606.

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437 **Corresponding author:** eric.grelet@crpp.cnrs.fr

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