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Chiral versus collinear magnetic order dynamics: faster chiral recovery after optical excitation revealed by femtosecond XUV scattering

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Abstract

While chiral spin structures stabilized by Dzyaloshinskii-Moriya interaction (DMI) are candidates as novel information carriers, their dynamics on the fs-ps timescale is little known. Since with the bulk Heisenberg exchange and the interfacial DMI two distinct exchange mechanisms are at play, the ultra-fast dynamics of the chiral order needs to be ascertained and compared to the dynamics of the conventional collinear order. Using an XUV free-electron laser we determine the fs-ps temporal evolution of the chiral order in domain walls in a magnetic thin film sample by an IR pump - X-ray magnetic scattering probe experiment. Upon demagnetisation we observe that the dichroic (CL-CR) signal from the chiral order in the domain walls recovers significantly faster than the (CL+CR) sum signal from the average collinear domain magnetisation. We explore possible explanations based on spin structure dynamics and reduced transversal magnetisation fluctuations inside the domain walls and find that the latter can explain the experimental data leading to distinctly different dynamics for collinear magnetic order and chiral magnetic order.

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Introduction

In the field of magnetism and spintronics chiral magnetic structures, such as spin spirals, domain walls and skyrmions [1-6], are intensively investigated due to their fascinating properties such as potentially enhanced stability and efficient spin-orbit torque driven dynamics [7-10]. It has been shown that these structures are stabilized by the Dzyaloshinskii-Moriya interaction (DMI) [11-12] that favours a chiral winding of the magnetisation. This antisymmetric indirect exchange interaction requires materials with large spin orbit coupling as well as a broken inversion symmetry, present in special bulk systems such as B20 compounds where skyrmions have been first discovered experimentally [1-2] or in interfacial systems such as heavy metal/ferromagnet multilayer stacks [3-6, 8, 10]. The domain wall type (Néel or Bloch) and chirality of the spin textures can be accessed in real space by imaging techniques [4, 13-15] or in reciprocal space by resonant magnetic X-ray scattering [16-18]. The chiral wall spin structure is of key importance as it governs the dynamical properties of domain walls and skyrmions [7-10, 19]. While the investigation of static structures and slow (ns) dynamics of chiral magnetic structures has been intensified recently, experimental studies addressing the ultimate fs-ps dynamics of the chirality have been elusive so far as ultrafast pump-probe experiments have concentrated on the collinear order in magnetic systems only [20-28]. In particular, as the ferromagnetic alignment is governed by the Heisenberg exchange, while the chiral order is resulting from the DMI, the ultrafast dynamics of both needs to be probed individually. And since the characteristic time scale for the onset of the chiral magnetic order and its ultrafast dynamics are unexplored up to now, we need to ascertain both, as they hold fundamental insights into the underlying physical mechanisms and allow us to gauge the ultimate speed for the manipulation of chiral magnetism, e.g. for ultrafast writing of chiral spin textures.

In this work we employ circularly polarized light pulses from an XUV free-electron laser and investigate time-resolved the evolution of the chirality of domain walls in magnetic thin film samples by an IR pump - X-ray magnetic scattering (XRMS) probe experiment. Using samples with interfacial DMI and perpendicular magnetic anisotropy exhibiting labyrinth-like domain patterns, we measure in the same experiment both the sum signal corresponding to the ferromagnetic order in the domains and the difference signal corresponding to the average chiral order in the domain walls. We find an ultrafast intensity decrease of both signals in the sub-ps regime with similar time constants. However, a significantly faster recovery of the chiral signal in the sub-ns timescale is observed. We subsequently investigate the origin of the faster recovery of the chiral signal by performing numerical simulations of the scattering signal, which reproduce the experimental findings.

Results

Magnetic small angle X-ray scattering

The experimental setup for the scattering experiment is shown schematically in Fig. 1 a. Circularly polarized extreme ultraviolet (XUV) radiation of 60 fs pulse duration was tuned to a wavelength of 23.0 nm corresponding to the Fe $M_{2,3}$ dichroic transition, which exhibits magnetic scattering contrast due to the X-ray magnetic circular dichroism (XMCD) effect [27-30]. The limited transmission in the XUV regime required us to perform the experiment in reflection geometry with an incident angle of Θ = 44° [28] yielding an effective XUV penetration depth of 8 nm. Due to an isotropic disordered domain structure we observe the typical ring-like diffraction feature well known from magnetic transmission SAXS experiments representing a broad distribution of Fourier components of the magnetic domain pattern [27-28]. The sample is pumped with a 60 fs IR laser pulse of 780 nm wavelength impinging on the sample with a small 2 degree offset with respect to the XUV beam (further setup information can be found in the Methods section).

The SAXS pattern was detected on a 2D CCD-detector and investigated as a function of the pump-probe delay over a time span of 100 ps. For each time delay and helicity, 7000 scattering patterns have been measured, normalized to the incoming flux and averaged. The area around the beamstop and charge scattering streaks have been masked. The background from the charge scattering due to the reflection geometry has been subtracted for each pattern leading to the corrected patterns used in the further analysis.

The investigated [Ta(5.3 nm)/Co₂₀Fe₆₀B₂₀(0.93 nm)/Ta(0.08 nm)/MgO(2.0 nm)] $_{x20}$ /Ta(1.6 nm) multilayer stack was produced by dc magnetron sputtering and exhibits perpendicular magnetic anisotropy (PMA). The material was imaged via magnetic force microscopy (MFM) as shown in Fig. 1b and exhibits, after out-of-plane demagnetisation, labyrinth-like magnetic domains at zero field. The domain pattern extended through all ferromagnetic layers due to stray field coupling exhibits a domain periodicity of (455±30) nm. In reciprocal space the MFM images yields a first order peak at q = (13.8±0.9) µm⁻¹. Due to partial alignment of the magnetic stripes, higher order peaks are visible as well [31]. Further material analysis with a superconducting quantum interference device (SQUID) yields a saturation magnetisation of M_s = (844±28) kA/m and an effective perpendicular anisotropy K_{eff} = (133±17) kJ/m³ of the sample at room temperature (for more sample details see Methods section). The Ta(5.3 nm)/Co₂₀Fe₆₀B₂₀ interface leads to a positive DMI, with reported values for such stacks ranging from 0.06 to 0.30 mJ/m² [32-34], which favours right-handed Néel-type domain walls (see Fig. 1c). The exact domain wall arrangement is finally determined by an interplay between interfacial DMI and dipolar interactions, which can lead to complex domain wall arrangements as shown recently [35]. We also note that considering the small penetration depth of the XUV radiation, we are sensitive to the magnetisation of the topmost Co₂₀Fe₆₀B₂₀ layer only.

X-ray scattering from such magnetic structures leads to the ring shaped SAXS patterns shown in Fig. 1(d-g). In Fig. 1f the sum image of the circular left (CL) and right (CR) polarized light (Fig. 1d & 1e) is shown, which is predominantly connected to the average domain magnetisation magnitude (I \propto M²) [27, 36]. This signal shows a ring-like diffraction pattern at a q-value of (14.3±0.1) μ m⁻¹ that we assign to the 1st order scattering peak of the magnetic stripe domains. The dichroic signal (CL-CR) defined as the difference of the diffracted left- and right-circularly polarized light [17] shown in Fig. 1g contains the information about the average chiral magnetisation magnitude and exhibits a pronounced angular asymmetry with an amplitude of 14 % of the sum signal. This provides us a tool to individually probe the time resolved dynamics of the chiral magnetic order in the domain walls. It has recently been shown that one can directly determine the domain wall chirality (left/right-handed) and character (Bloch (helical), Néel (cycloidal)) from the dichroic signal [17-18]. Fig. 2g shows the orthoradial profile of the dichroic signal for different pump-probe delay times. We used a fitting model of A^* cos(Ψ - ϕ), where A is the amplitude, Ψ the azimuthal angle and the phase ϕ determines the domain wall angle and with that the domain wall chirality and character [17]. Fitting the azimuthal profile leads to a value of ϕ = 90° which indeed confirms the presence of fully right-handed Néel-type domain walls [17-18].

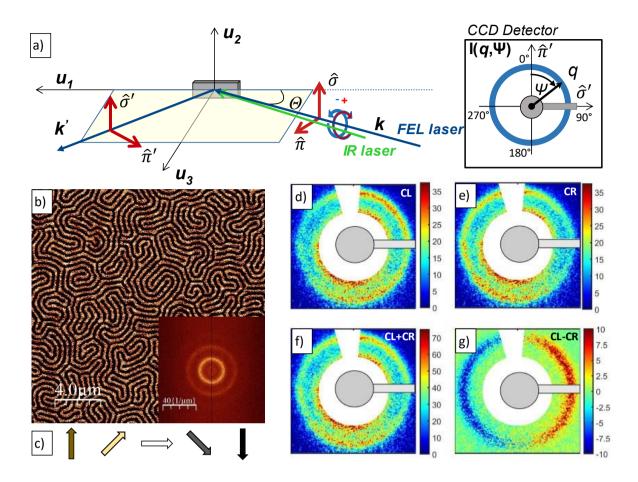


Figure 1: Experimental setup and diffraction images. (a) Measurement geometry: A magnetic thin film sample is pumped by an optical infrared laser pulse and probed by a circularly polarized X-ray FEL pulse with an incident angle of Θ = 44° (wavevector k, s-polarisation(σ) and p-polarisation(π)) that scatters on the sample. Afterwards an IR-protected charge-coupled device (CCD) detector records the magnetic SAXS pattern. (b) MFM image of a typical labyrinth domain pattern of the [Ta(5.3 nm)/Co₂₀Fe₆₀B₂₀(0.93 nm)/Ta(0.08 nm)/MgO(2.0 nm)]_{x20}/Ta(1.6 nm) sample. The inset shows the FFT with the first order peak corresponding to isotropic distributed labyrinthine stripes with a domain periodicity of (455+/-30) nm. These magnetic structures lead to the SAXS signals for (d) left-hand (CL) and (e) right-hand (CR) circular polarized incident x-rays. (f) The resulting sum = CL+CR of the diffraction pattern, confirms that the diffraction corresponds to the magnetic domains observed by MFM. (g) The azimuthal asymmetry in the dichroic signal (difference = CL-CR) confirms the presence of (c) right-handed chiral Néel (cycloidal) domain walls.

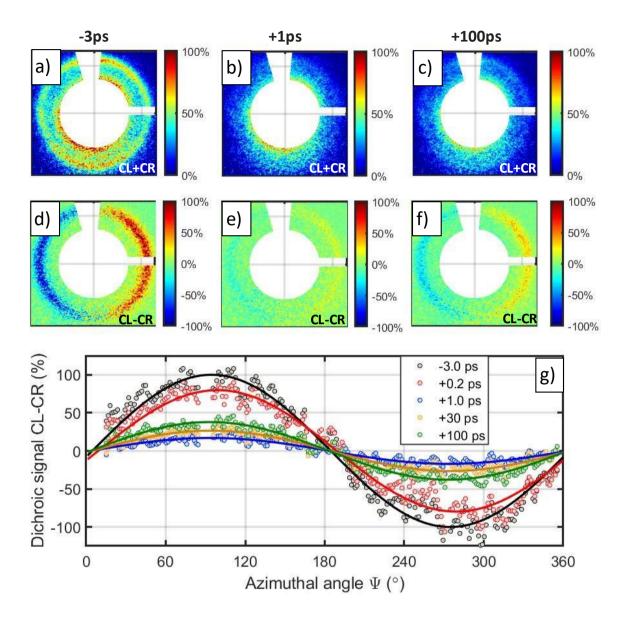


Figure 2: Time dependence of diffraction images. Diffraction patterns for the sum CL+CR (top) and the difference CL-CR (tot) images for the initial (tot), +1.0 ps (tot), e) and +100 ps (tot), states. (tot) The experimental (tot) and fitted (tot) and fitted (tot) orthoradial dichroic profiles for different pump-probe delay times showing a maximum at 90° and a minimum at 270° corresponding to a right-handed cycloidal winding of the domain walls. While the amplitude of the dichroic signal drops within a ps upon pumping and recovers afterwards, no significant angle shift in the fitted phase tot can be detected. The beamstop is located at an azimuthal angle of 90°.

Ultrafast time resolved pump-probe experiments

In a next step we investigated the time evolution of the scattering signals upon IR laser excitation. Fig. 2(a-f) shows the diffraction patterns of the sum and the dichroic signals at different pump-probe delays. At +1.0 ps (Fig. 2b & 2e)) the intensity in both signals has dropped down significantly in comparison to the initial states (Fig. 2a & 2d) with a visible partial recovery of both signals after +100 ps (Fig. 2c & 2f).

By azimuthal integration of the dichroic signal we can identify that within our experimental accuracy no significant shift in the phase ϕ of the orthoradial profile (Fig. 2g) has been observed upon pumping. Thereby we can exclude any significant ultrafast change of the domain wall angle. This means that the DMI is even at the ultrafast timescale contributing to the wall chirality and therewith the stability of chiral spin structures is retained.

The key step is the comparison of the ferromagnetic order dynamics and the chiral order dynamics. To ascertain both, we calculated the average intensity for the sum and the difference signal as a function of the wavevector Q_r for different time delays (Fig. 3a & 3b). The error bars from counting statistics are of the order of 0.01 % and the fluctuations visible are due to speckles from the coherent illumination. Numerical integration of the radial profiles for each time delay leads to the data shown in Fig. 4a & 4b which demonstrates the evolution of the total intensity of the sum and the difference signal as a function of delay time on linear (a) and logarithmic time scale (b) normalized to the unpumped total intensity.

We used the model reported in equation (1) of the Methods section to fit the experimental data [27]. The model includes the time constant τ_d of the demagnetisation process of the ferromagnetic collinear order respectively the dechiralisation process (vanishing of the chiral order in the domain walls), and the time constants τ_{r1} and τ_{r2} of the fast and slow recovery processes. The initial ultrafast demagnetisation time constant $\tau_d = 0.37 \pm 0.10$ ps is in agreement with the typical demagnetisation times of ferromagnetic 3d-transition metals [21-23, 27-28]. The main results of these experiments are firstly a dechiralisation process ($\tau_d = 0.33 \pm 0.10$ ps) that occurs on similar timescales as the demagnetisation of the collinear domains, and secondly we observe a clear difference in the slower recovery process of the two signals. The time constant $\tau_{r2} = 312 \pm 6$ ps of the difference signal is significantly smaller than the one of the sum signal ($\tau_{r2} = 905 \pm 28$ ps) demonstrating a faster recovery of the dichroic signal that reflects the chiral magnetic order in the domain walls.

To understand the origin of this difference in the recovery, we next investigate the time evolution of the 2nd moments of the radial distributions as displayed in Fig. 4 c & 4d. The 2nd moments (calculated using equation (2) in the Methods section) entail additional information about possible ultrafast changes of the radial distribution widths upon IR excitation. The 2nd moment of the sum signal increases significantly upon pumping within 1 ps, indicating a wider radial distribution and therefore we can identify stronger fluctuations leading to a reduced correlation length of the domain-domain correlation function. After 100 ps it has just recovered partially to its unpumped value. In contrast, the 2nd moment of the difference signal decreases slightly within 1 ps to about 90 % after optical excitation but recovers afterwards within a few ps to values around the unpumped value.

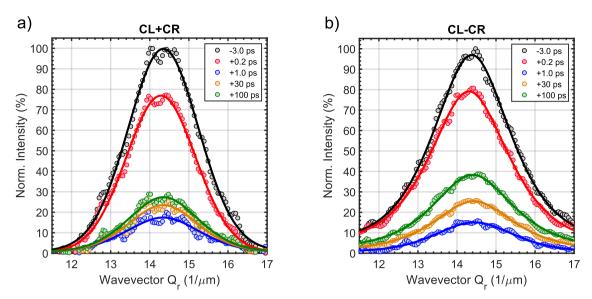


Figure 3: Time dependence of radial scattering intensities. The radial profiles of the sum CL+CR (a) and difference CL-CR (b) signals for selected delay times. Error bars from counting statistic are smaller than the symbol sizes. The visible fluctuations in the profiles are due to speckles originating from the coherent illumination.

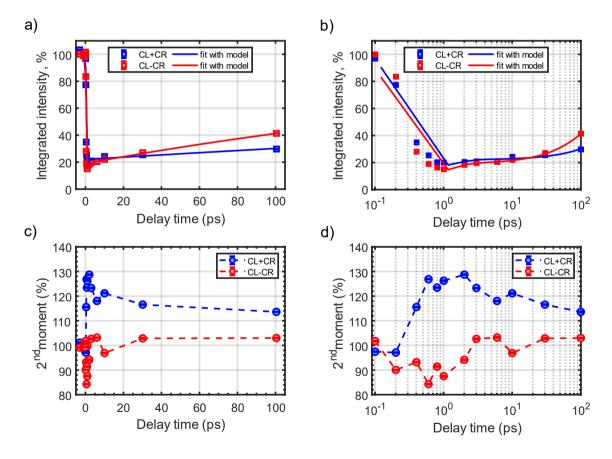


Figure 4: Time dependence of total scattering intensities and second moments. Time evolution of the (**a,b**) radially integrated intensity and (**c,d**) the 2nd moments of the sum signal (ferromagnetic order) and the difference signal (chiral magnetic order) in linear scale (**a,c**) and logarithmic scale (**b,d**). The solid lines in (**a,b**) depict the fitted dependences using equation (1) and highlight the faster recovery of the chiral signal. The dashed lines in (**c,d**) provide a guide to the eye. Error bars based on photon counting statistics and numerical integration of the peaks in Fig. 3 are smaller than the symbol sizes.

Discussion

The experimental results show an ultrafast decrease within the first ps of the total intensity of the sum signal that reflects the conventional collinear magnetic order inside the domains, as well as the dichroic signal, that corresponds to the chiral order of the domain walls. Both signals decay on similar timescales, indicating that for the ultrafast initial demagnetisation process the order of the magnetisation itself does not play a major role in our sample system.

In order to explain the key finding of the experimentally observed faster recovery of the chiral signal after laser excitation we can envisage two different mechanisms: (1) a change in the size ratio between domain walls and domains caused by an increase of the domain wall width during the whole investigated time frame or (2) a faster recovery dynamics of the chiral order within the domain walls compared to the ferromagnetic order in the domains leading to a faster build-up of the chiral magnetisation.

The increase of the domain-wall width as required for mechanism (1) can results from the temperature dependence of the micromagnetic parameters, more precisely the effective anisotropy and the exchange constant, as shown in the Supplementary S3. Since the average domain magnetisation is directly proportional to the square root of the total intensity of the sum signal one can estimate the temperature evolution of the spin system (assuming the 3-temperature model [21]) during the pumpprobe experiment using the fitted temperature dependence of the saturation magnetisation of the sample (see Supplementary S2). This leads to an increased spin system temperature of (493±22) K at a delay time of 1 ps, where the sample is significantly demagnetized, and an already decreased temperature of (469±13) K at 100 ps. Such a sudden change of the micromagnetic parameters is expected to excite a breathing mode of the domain walls. The breathing mode corresponds to a damped ps-scale oscillation of the width of the domain wall around the new equilibrium wall width as determined by the temperature-scaled micromagnetic parameters. We estimate the equilibrium wall width at the increased spin system temperature to be 37 nm (see Supplementary S3). Therefore, we estimate that for our system the wall width oscillates between values of 27 nm and 47 nm due to the sudden change of the micromagnetic parameters. Domain-wall breathing frequencies lie in the range of several GHz. Assuming a breathing frequency of 5 GHz, one-half period of the breathing mode amounts to 100 ps, in which the wall expands by 20 nm from 27 nm to 47 nm.

Mechanism (2) is based on a faster recovery of the chiral order parameter of the system in comparison to the collinear order parameter [37-39]. The chiral order parameter is defined here as the in-plane component of the vector $\langle S_i \times S_{i+1} \rangle / |\sin \theta_0|$ with θ_0 denoting the average angle between the neighbouring spins S_i and S_{i+1} in the wall [37-39]. The chiral order parameter defined in this way is a pseudo-scalar variable analogous to an Ising spin, which takes values of +1 for right-rotating walls and -1 for left-rotating walls, before the pump pulse arrives.

Between 1 ps and 10 ps the electronic part of the system recovers its ground-state properties to a large extent, while the spin system is still excited with a relatively high spin temperature of about 490 K, which slowly decays on the scale of 10 - 1000 ps towards room temperature. It is in this time window that the recovery rate of the chiral order parameter is significantly higher than that of the magnetization. This is based on different temperature dependences of the chiral and scalar order parameters as observed in various spin systems, with the chiral order parameter restoring to its ground state value faster as the temperature is lowered [39]. The faster recovery behaviour can be traced back to the properties of the chiral order parameter as an essentially Ising type variable [37-38, 40], which is also the basis of exotic spin states such as chiral spin liquids [37-38, 41].

In order to check the appropriateness of these two mechanisms for our case, we performed numerical simulations of the scattering signal and evaluated - similar to the experiment - the total intensity and the 2nd moments of the sum and difference signal under the influence of these two mechanisms. We start from an initial labyrinth stripe state shown in Fig. 1b with a domain periodicity of 450 nm and a domain

wall width of 27 nm as estimated from the magnetic parameters (S3). For mechanism (1) the domain wall width was altered, while keeping the domain periodicity fixed. Mechanism (2) was implemented by introducing smaller amplitudes of the (transversal) magnetisation fluctuations to the domain walls compared to the domains emulating the faster build-up of the chiral order parameter.

In Fig. 5a we show the simulated total scattering intensities of CL+CR and CL-CR as a function of the domain wall width. For an estimated maximum increase of the wall width towards 47 nm the signal strength of CL-CR increases to 125 %, while CL+CR decreases to 95 %. The ratio of both signals thus increases by a factor of 1.3 which fits very well the experimental values of 1.33 as measured at 100 ps (Fig. 4b).

In Fig. 5b we show the dependence of the simulated total scattering intensities of CL+CR and CL-CR on the fluctuation tilt angle, which characterizes the strength of the transverse spin fluctuations. The dashed and dotted lines in Fig. 5b correspond to a homogeneous distribution of fluctuations throughout the system, while the solid lines represent the case of reduced fluctuations in the walls, but not in the domains. First, we note that the experimentally observed drop of the intensities to 20 % during the first ps in Fig. 4b) correspond to a fluctuation tilt angle of 63° in Fig. 5b (dashed and dotted lines). We assume that during the following 100 ps the fluctuations in the walls reduce faster than in the domains caused by the above discussed temperature (and by that delay time) scaling of the order parameters such that finally the solid lines in Fig. 5b describe the intensities. At 100 ps the value of CL+CR observed in Fig. 4b) is 30 % and CL-CR is 41 % in good agreement with the numerical intensities at a tilt angle of 62° (solid blue and red line in Fig. 5b).

Fig. 5c displays the evolution of the 2nd moment of the scattering peak S(Q) from the simulation as a function of wall width. In contrast to the overall intensity (Fig. 5a) we do not observe significant changes in the second moments on increasing the relevant domain widths between 27 nm and 47 nm.

Fluctuations, however, can change the 2nd moments significantly (Fig. 5d). While the 2nd moments of the CL-CR signal are insensitive to the tilt angle (red lines), the CL+CR signal increases with increasing fluctuations - in good agreement with the experimental findings (Fig. 4d). Therefore, the experimentally observed decrease of CL+CR can be explained by a reduction of the fluctuation strength with increasing delay time and a faster reduction of fluctuations in the domain wall leading to a faster build-up of the chiral order.

Thus, the findings obtained from the second moments support the presence of mechanism (2): the fluctuation mechanism, while not necessarily ruling out the additional contribution of (1) the breathing mode mechanism to the faster recovery of the total intensity of the difference signal, since we showed that both mechanisms can explain the observed faster recovery of the total intensity of CL-CR.

However, this implicates the necessity of breathing frequencies lower than 5 GHz, since otherwise also a decrease of the DW width and accordingly CL-CR should be observed within the 100 ps timeframe, while frequencies below 5 GHz would lead to a reduced domain wall increase within the 100 ps timeframe. Analytical calculations of the domain wall oscillation frequency [42] using the measured parameters suggest that even the strongly reduced anisotropy values at the high spin system temperatures after the IR excitation lead to oscillation frequencies in the range of 13-17 GHz. This does not fit the experimental data that could only be explained by an oscillation of < 5 GHz.

Therefore, we conclude that the main driver behind our experimental findings is likely mechanism (2) leading to a faster recovery of the chiral order in the domain walls in comparison to the ferromagnetic order in the domains. The responsible mechanism(s) might in addition be also material system dependent. So for the future, one can explore further multilayer stacks and the time dependence of the total intensity on even longer timescales to probe if one can identify additionally the effects of a damped breathing mode of the domain wall excited by the IR laser.

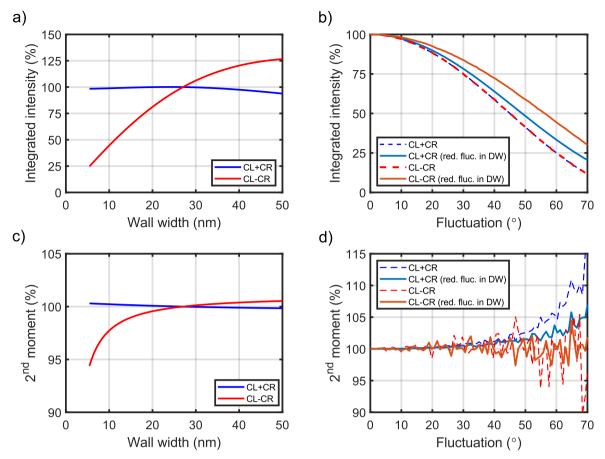


Figure 5: Numerical calculations of total scattering intensities and second moments. The numerically calculated (**a,b**) integrated intensity and (**c,d**) 2nd moments of the sum signal (ferromagnetic order) and the difference signal (chiral magnetic order) as a function of the (**a,c**) domain wall width and (**b,d**) the transversal magnetisation fluctuation strength (cone angle in degrees). The 100 % value in (**a,c**) corresponds to the room temperature domain wall width of 27 nm. The dashed lines in (**b,d**) correspond to the case of fluctuation in the whole system, while the solid lines correspond to the case of reduced fluctuations within the domain walls.

Conclusion

In conclusion, using resonant magnetic scattering at an XUV FEL we identify a distinctly faster recovery of the scattering intensity signal stemming from the chiral magnetic order in the domain walls compared to the signal from the collinear magnetic order in the domains. We also observe an increase of the width of the structure factor peak for the collinear signal while the chiral scattering peak does not increase in width upon IR pumping indicating strong fluctuations. We explain both observations with the onset of strong transversal magnetic fluctuations after pumping in the domain systems. Importantly, however, our experimental findings imply reduced fluctuations within the chiral domain walls leading to a chiral order that is restored faster. We can connect this to the realisation that the characteristic time scale for the DMI-driven onset of the chiral magnetic order is significantly lower than the onset of the ferromagnetic order driven by the Heisenberg exchange interaction. In the future further fundamental aspects can be studied in detail e.g. the dependence of the timescales of the chiral order build-up on the absolute strength of the DMI by varying the heavy metal layers. The better control of the DMI and the chirality of spin structures on the ultrafast timescale can finally allow the controlled ultrafast manipulation of chiral magnetism, e.g. ultrafast writing of chiral topological objects such as skyrmions and pave the path to applications in the field of ultrafast chiral spintronics.

Methods

Sample fabrication and characterisation

The magnetic multilayer sample has been grown at room temperature by dc magnetron sputtering and consist of (thicknesses in nm) Si/SiO₂/[Ta(5.3)/Co₂0Fe₆₀B₂₀(0.93)/Ta(0.08)/MgO(2.0)]_{x20}/Ta(1.6). The Ta(0.08 nm) interlayer was inserted to tune the anisotropy stemming from the Co₂₀Fe₆₀B₂₀/MgO interface, while the 1.6 nm Ta capping was reduced to a thickness that prevented the sample from oxidation, but simultaneously led to sufficient penetration of the XUV light into the topmost Co₂₀Fe₆₀B₂₀ layer to ensure sufficient statistics. The magnetic properties of the sample were characterized using superconducting quantum interference magnetometry (SQUID) in the temperature range 10 K to 390 K and a vibrating sample magnetometer (VSM) in the temperature range 300 K to 600 K. Both out-of-plane and in-plane hysteresis loops were measured (see Supplementary S1) for each temperature. From those the saturation magnetisation M_s and the effective perpendicular anisotropy constant Keff were determined. The latter was ascertained from the difference in the areas of the in-plane and out-of plane loops [43]. This leads to the temperature dependencies shown in Supplementary S1 yielding a Curie temperature of (553±1) K and parameters $M_s(300 \text{ K}) = (844\pm28) \text{ kA/m}$ and $K_{eff}(300 \text{ K}) = (133\pm17) \text{ kJ/m}^3$ at room temperature. Additionally, a 20x20 µm² MFM image of the multilayers was performed at room temperature as shown in Fig. 1b. The sample exhibits labyrinth-like stripe domains with a domain periodicity of (455+/-30) nm, as extracted from the Fourier spectrum which exhibits a first order peak at q = (13.8+/-0.9) μm⁻¹ (see Supplementary S4). The interfacial DMI arises from the Ta(5.3 nm)/ Co₂₀Fe₆₀B₂₀ interface, which favours right-handed chiral Néel domain walls, as observed in the dichroic SAXS signal in Fig. 1f. One must take into consideration that the Ta(0.08 nm) insertion layer is not even a complete Ta monolayer and therefore a net interfacial DMI is still prominent in the sample. Finally, the interplay between interfacial DMI and dipolar interactions determines the equilibrium domain and domain wall arrangement in the sample [35].

Pump-probe experiments

The measurements have been performed at the DiProl beamline [44-45] at the FERMI FEL facility in Trieste, Italy [46-47]. For the experiment, the FEL was tuned to the Fe M_{2,3}-edge at a wavelength of 23.0 nm, a pulse duration of 60 fs, a repetition rate of 50 Hz, and an attenuated pulse energy of 0.94 µJ. Using a Kirkpatrick-Baez (KB) optics, the beam was focused down to a size of (200) × (200) µm² leading to an energy density of 3,0 mJ/cm². The optical laser for pump-probe experiments is the same as the Ti:sapphire seed laser used for generating the FEL pulses in the HGHG scheme and therefore is intrinsically synchronized to the XUV-FEL pulses with a jitter of less than 10 fs. We used as a pump a 780 nm IR pulse of 100 fs duration, with an energy of 2.62 µJ and a size of (300) × (300) µm², leading to a pump energy density of 3,7 mJ/cm², well below the material damage threshold and the intra-pulse self- demagnetization process observed with FEL radiation on similar magnetic structures [48]. The IR and FEL beams are nearly parallel with a small angular offset of 2° which allows for a constant temporal resolution during the rotation of the sample through the beam. The resonant magnetic XUV reflectivity experiments were performed for left- and right-hand circular polarization with a charge-coupled device (CCD) area detector (2,048×2,048 pixels, 13.5×13.5 µm² pixel size) placed 145 mm behind the sample. The circularly polarized incident X-rays hit the sample at an angle of 44 degrees yielding an effective penetration depth of 8 nm. The intense reflected primary beam is blocked by a beam stop. The CCD is IR protected by an Al filter.

SAXS data treatment

For each measured pump probe time delay, 7000 SAXS patterns have been measured for each helicity. The patterns have been normalized to the incoming flux and averaged. The area around the beamstop and charge scattering streaks have been masked. To remove the contribution of the charge component, the background signal must be calculated for each pattern. Using the areas inside and outside the

diffraction ring, we can reconstruct the whole pattern without the magnetic part. As a result of the background removal, we obtain the corrected patterns for post-proceeding analysis.

Afterwards the average intensity of the sum (CL+CR) and difference (CL-CR) diffraction images was determined as a function of the wavevector Q_r for different time delays as depicted in Fig. 3. The radial profiles for each time delay were then integrated in order to obtain the evolution of the total intensity of the sum and the difference signal as a function of delay time. The time evolution of the total intensity normalized to the unpumped total intensity is displayed in Fig. 4 a & 4b.

The time dependence of the total intensities was fitted with the sum of three exponential functions convolved with a Gaussian distribution containing the time resolution σ_t of the experiment given by the pulse duration of the FEL:

$$\begin{split} &I(t)\Big/_{I_{unpumped}} = \Big\{1 - H(t-t_0) \cdot \Big[A_d \cdot exp\left(-\frac{t-t_0}{\tau_d}\right) + A_{r1} \cdot exp\left(-\frac{t-t_0}{\tau_{r1}}\right) + A_{r2} \cdot exp\left(-\frac{t-t_0}{\tau_{r2}}\right)\Big]\Big\} * \Big[\frac{1}{\sqrt{2\pi}\sigma_t} \cdot exp\left(-\frac{(t-t_0)^2}{2\sigma_t^2}\right)\Big], \end{split} \tag{1}$$

where τ_d denotes the time constant of the demagnetisation respectively the dechiralisation process and τ_{r1} and τ_{r2} the time constants of the fast and slow recovery processes, whereas A_d , A_{r1} and A_{r2} denote the strength of these processes (fit parameters displayed in Supplementary S5). H(t) is the Heaviside step function and time zero t_0 was obtained from the fit.

The 2nd moments of scattering peak S(Q) (radial scattering distribution) displayed in Fig. 4c & 4d are calculated by:

$$2^{nd} \ moment = \frac{\int (q - q_0)^2 \ S(q) \ dq}{\int S(q) \ dq}$$
 (2)

where the peak center qo is determined for each delay time individually.

Numerical calculation of the diffracted intensity

In the simulation setup we assume a system of homochiral walls, where the -+ and +- walls are modelled by:

$$m_x^{W^{-+}}(x) = -m_x^{W^{+-}}(x) = -\sqrt{1 - \left[\tanh\frac{x - x_i}{w}\right]^2}$$

$$m_y^{W^{-+}}(x) = m_y^{W^{+-}}(x) = 0$$

$$m_z^{W^{-+}}(x) = -m_z^{W^{+-}}(x) = \tanh\frac{x - x_i}{w}, \quad (3)$$

We model the diffracted intensity by:

$$I(Q) = |(\varepsilon \times \varepsilon') \cdot \int_{-L/2}^{L/2} \boldsymbol{m}(x) e^{iQx} dx|^2, \quad (4)$$

where L is the size of the simulation box. We used L=225 μ m, corresponding to 500 domain periodicities of 450 nm. The magnetization $\mathbf{m}(x)$ in the system without fluctuations is obtained by alternating -+ and +- walls spaced 225 nm apart. In order to model the effect of transverse fluctuations we first tilt the magnetization at every point by a polar tilt angle. After that we rotate the magnetization at every point by an azimuthal angle obtained from a random number generator. We characterize the fluctuation strength by the polar tilt angle.

In order to model the case where fluctuations are reduced in the walls (due to the proposed faster recovery of the chiral order parameter) we switch off the fluctuations in the region $-1.1w < x - x_i < 1.1w$ for every wall i.

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Author Contributions

C.G. and M.K. proposed the study. C.G., M.K., and Y.M. supervised the respective members of the study. N.K. and J.C. fabricated samples. N.K., C.G., H.Z., F.C., E.P. and I.L.-Q. performed the FEL experiment. D.K. and C.G. analysed the beamtime data. N.K., B.S., K.L. and D.L. performed the sample characterization. F.F., Y.M., N.K., C. G., M. K. and D.K. developed the two discussed mechanism. F.F. performed numerical calculations of the scattering signal. All authors participated in the discussion and interpreted results. N.K. drafted the manuscript with the help of D.K., F.F. and assistance from C.G., M.K. and Y.M. All authors commented on the manuscript.

Competing interests

The authors declare no competing interests.

Supplementary

S1: Determination of material parameters and their temperature dependence by SQUID/VSM

Hysteresis loops were recorded by SQUID and VSM in order to determine the saturation magnetisation M_{s} as well as the effective anisotropy K_{eff} of the sample. The curves displayed in Fig. S1 show the out-of-plane (OOP) and in-plane (IP) hysteresis loops at 300 K. The effective perpendicular anisotropy K_{eff} was determined by the difference in the areas of the in-plane and out-of-plane hysteresis loops and corresponds to $K_{u} - \mu_{0}/2$ M_{s}^{2} with the uniaxial anisotropy K_{u} [43].

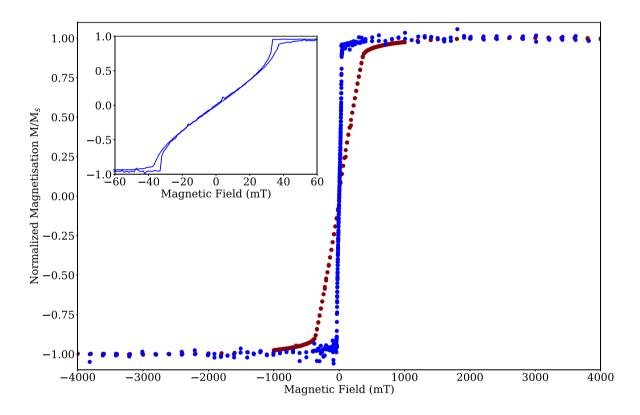


Figure S1: Hysteresis loops of multilayer stack. OOP (blue) and IP (red) hysteresis loops recorded at 300 K by SQUID magnetometry. Inset: The zoomed in OOP curve displays the typical behaviour of a multidomain stripe sample.

The hysteresis loops displayed in Fig. S1 measured at room temperature (300 K) via SQUID were also performed at temperatures of 10 K, 50 K, 100 K, 200 K, 300 K, 350 K and 390 K by SQUID and 300 K, 350 K, 390 K, 450 K, 500 K, 550 K, and 600 K by VSM. The saturation magnetisation was determined and afterwards the following model was used to fit the temperature dependence of the saturation magnetisation:

$$M_s(T) = M_s(0 K) (1 - (T/T_c)^{\alpha})^{\beta},$$
 (5)

with a saturation magnetisation $M_s(0 \text{ K}) = (976\pm10) \text{ kA/m}$, a Curie temperature $T_c = (553\pm1) \text{ K}$, $\alpha = (1.73\pm0.13)$ and $\beta = (0.63\pm0.04)$.

The temperature dependence of the effective anisotropy constant was fitted using:

$$K_{eff}(T) = K_u(T) - \frac{\mu_0}{2} (M_s(T))^2,$$
 (6)

where the magnetisation dependence of the anisotropy constant was fitted by a power law:

$$\frac{K_u(T)}{K_u(0\ K)} = \left(\frac{M_s(T)}{M_s(0\ K)}\right)^c,\tag{7}$$

leading to an anisotropy constant $K_u(0 \text{ K}) = (1031\pm11) \text{ kJ/m}^3$ and $c = (2.16\pm0.02)$.

Using the fit parameters, we obtain finally a room temperature saturation magnetisation of $M_s(300 \text{ K}) = (844\pm28) \text{ kA/m}$ and an effective anisotropy of $K_{eff}(300 \text{ K}) = (133\pm17) \text{ kJ/m}^3$.

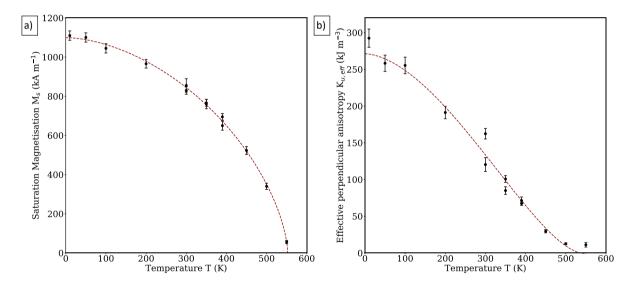


Figure S2: Temperature dependence of M_s and K_{eff} . Temperature dependence of the saturation magnetisation M_s (a) and effective anisotropy K_{eff} (b). The datapoints were fitted by the models displayed in equation (5) respectively equation (6).

S2: Pump-probe-delay time dependent spin system temperature

The total intensity of the sum signal displayed in Fig. 4a & 4b is connected to average domain magnetisation (I \propto M²). Using the temperature dependence of the saturation magnetisation according to Fig. S2 one can estimate the temperature evolution of the spin system during the pump-probe experiment displayed in Fig. S3. Upon optical excitation the spin system temperature increases within a ps to (493±22) K and relaxes afterwards slowly back to room temperature with an already decreased temperature of (469±13) K at 100 ps.

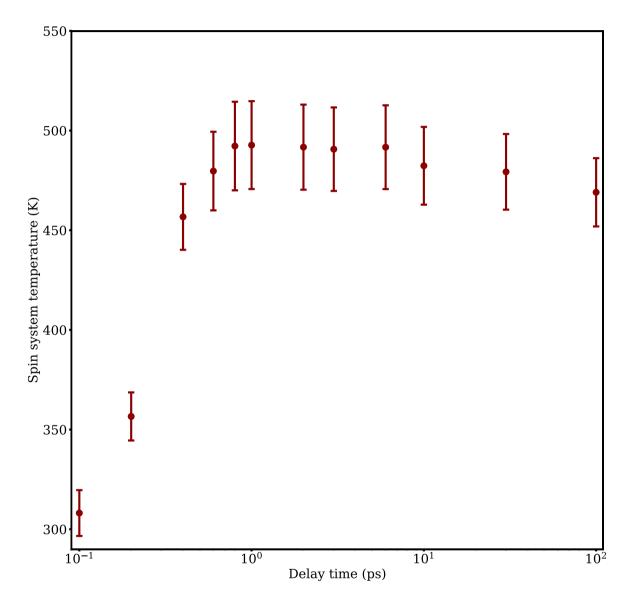


Figure S3: Time dependence of spin system temperature. The evolution of the spin system temperature as a function of pump-probe delay time in logarithmic scale.

S3: Temperature dependence of the domain wall width

Using the temperature dependence of the effective anisotropy constant obtained in Fig. S2 one can estimate the temperature dependence of the (equilibrated) domain wall width given by:

$$\Delta = \pi \sqrt{A/K_{u,eff}} . \qquad (8)$$

The temperature dependence of the exchange stiffness constant A is in mean-field approximation described by a power law A $\propto M_s^2$ leading to the temperature scaling of the domain wall width displayed in Fig. S4. One can observe that the domain wall width increases with increasing temperature due to a stronger scaling of the effective anisotropy constant in comparison to the exchange constant. It was also shown that fluctuation corrections from nonlinear spin-wave effects lead to a scaling A $\propto M_s^\kappa$ with κ <2 depending on the lattice structure [49], which would lead to an even stronger scaling of the domain wall width with the temperature.

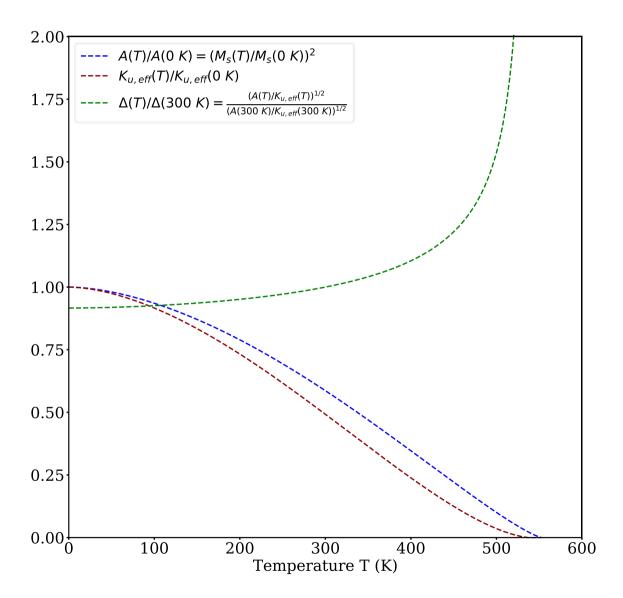


Figure S4: Temperature dependence of K_{eff} , A and Δ . Normalized temperature dependence of the effective anisotropy constant (red), the exchange stiffness constant (blue) and the domain wall width (green).

Using the assumed exchange stiffness constant of A=10 pJ/m at room temperature (300 K) leads to the absolute domain wall widths displayed in Fig. S5.

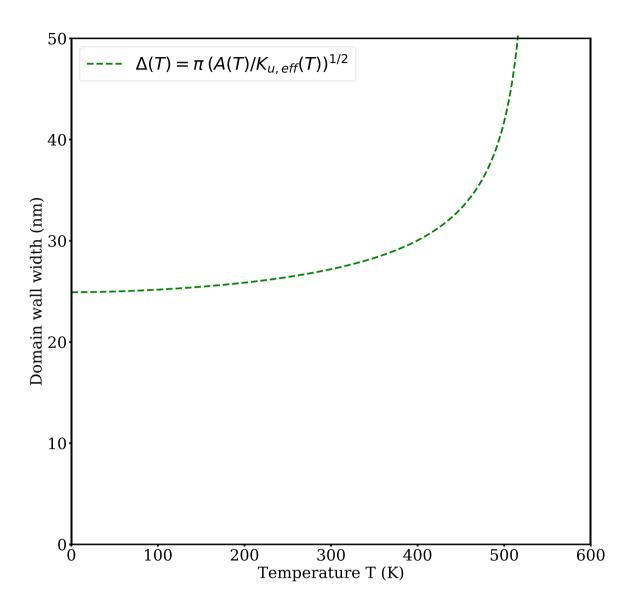


Figure S5: Temperature dependence of Δ . Temperature dependence of the domain wall width based on the temperature scaling of the exchange stiffness constant and effective anisotropy.

Combining the time delay dependence of the spin system temperature displayed in Fig. S3 with the here obtained temperature dependence of the domain wall width one can estimate the increase of the equilibrium domain wall width to approximately 10 nm for delay times between 1 ps to 100 ps.

S4: Fourier spectrum of the MFM image

The magnetic force microscopy image displayed in Fig. 1b was Fourier transformed leading to the spectrum displayed in Fig. S6. One can see a clear first order peak at a value of $q=(13.8+/-0.9) \mu m^{-1}$ as well as a second order peak at $q=(27.6+/-0.9) \mu m^{-1}$ and a third order peak at $q=(41.4+/-1.0) \mu m^{-1}$ due to partial short range stripe alignment of the OOP demagnetized magnetic labyrinth pattern. The first order peak indicates a stripe periodicity of (455+/-30) nm, which agrees well with the real space stripe data measured.

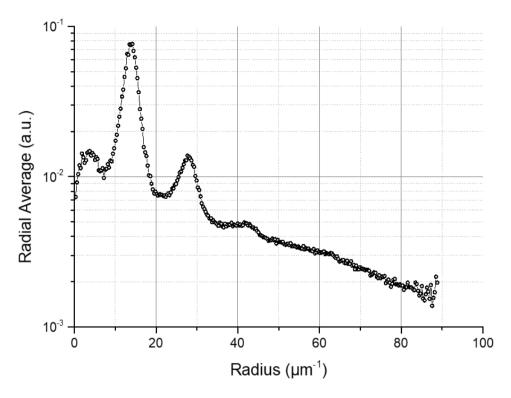


Figure S6: Fourier Spectrum of MFM image. Logarithmic Fourier spectrum of the MFM image of the demagnetized labyrinth stripe state showing a first, second and third order peak.

S5: Fit parameters of the pump-probe delay curves

	CL+CR	CL-CR
$\tau_{\sf d}$	0.37±0.10 ps	0.33±0.10 ps
τ _{r1}	0.48±0.08 ps	0.44±0.11 ps
τ _{r2}	905±28 ps	312±6 ps
A _d	-2,99±0,02	-3,00±0.01
A _{r1}	1,89±0,09	1,88±0,09
A _{r2}	0,78±0.01	0,81±0.01

Table S1: Pump-probe fit parameters. Fit parameters obtained by fitting the pump-probe data shown in Fig. 4a & 4b with the model described in equation (1).

Continued References:

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