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White light emission from a zero-dimensional lead-chloride hybrid material

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Abstract:

Organic-inorganic hybrid (OIH) materials have emerged as high-performance class of light emitters, which can be used as phosphors for optically pumped white light-emitting diodes (WLEDs). Here, we report a new example of broadband white light (WL) emission from zero-dimensional lead chloride-based OIH material namely $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ (TAE = tris-(2-aminoethyl) ammonium ($\text{C}_6\text{N}_4\text{H}_{22}$)). Single crystals of this compound and its corresponding ammonium salt TAE.HCl have been synthesized by slow solvent evaporation method at room temperature and were characterized by X-ray diffraction, optical absorption (OA) and photoluminescence (PL) measurements. The crystal structure shows an isolated lead chloride edge sharing bi-octahedra $[\text{Pb}_2\text{Cl}_{10}]^{6-}$ interacting with two $[\text{TAE}]^{4+}$ cations and two Cl⁻ chlorine ions through multiple hydrogen bonds in a triclinic unit cell, which confers to this system its 0D character. Upon UV irradiation at 3.81 eV, $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ shows a very broadband WL emission resulting from the overlapping of two peaks located at 2.07 eV and 2.83 eV, respectively attributed to photoinduced free excitons confined in $\text{Pb}_2\text{Cl}_{10}$ inorganic bi-octahedra and the organic cations, respectively. When excited with a 4.66 eV laser, the full

width at half maximum of WL band luminescence reaches the unprecedented value of 1.25 eV (247nm), and exhibits the highest CRI value with 96 and a chromaticity coordinates closer to the reference white light. This was mainly attributed to light emission originating from some self-trapped excitonic states caused by structurally highly distorted inorganic subnetwork. The present results enrich the restricted family of hybrid perovskite WL emitters and highlight the importance of the 0D OIH materials for solid-state lighting applications.

Keywords: 0D hybrid material, Lead chloride, White light emission, FWHM, CRI.

Solid-state lighting, which presents an alternative solution to the inefficiencies of conventional lighting sources, has attracted great attentions in recent years.¹ Realizing WL emission from light-emitting diodes (LEDs) is of particular interest for illumination applications.² Different strategies for generating white light are well-known.^{3,4} One can use (i) a combination of red, green, and blue LEDs in a single device; (ii) a blue LED connected to a single or two color conversion phosphors excited by the blue light, and (iii) a UV LED connected to three different phosphors, which emit in the blue, green, and red regions of the visible spectrum, respectively. While these techniques have been widely commercially accepted, the resulting devices suffer from a variety of inevitable drawbacks, such as the spectral overlap of the different constituting phosphors which can result in self-absorption and hence losses of emission efficiency and poor color stability. Especially, white light stability is often described by three main characteristics: (i) the Commission Internationale d'Eclairage "CIE": this is the most widely used method to describe the composition of any color in terms of three primaries (red–green–blue) coordinates; (ii) the Correlated Color Temperature "CCT": which indicates the temperature of a blackbody radiator that is most similar in chromaticity to the light being described, and (iii) the Color Rendering Index "CRI": this is a numerical measure of how true a color appears to the human eye when illuminated with a light source. It can be rated on a scale from 0 to 100, where 100 is a true

color representation.⁵ Therefore, a major aim in solid-state lighting research has been concentrated recently in the development of a single-component broadband WL emitting, highly luminescent, and photostable material which achieve the criteria imposed by the above mentioned characteristics. In this context, organic-inorganic hybrid (OIH) materials represent an emerging class of crystalline photoactive materials, which have excellent solution processability with promising astounding physical properties for optoelectronics.⁶⁻⁹

While early investigations of OIH materials have been mainly focused on the structural and vibrational dynamics, they have been rediscovered over the past two decades and applied in promising optoelectronic fields,¹⁰⁻¹¹ such as light emitting diodes (LED),¹²⁻¹⁶ Field Effect Transistors or thin film transistors (TFT),¹⁶⁻²¹ Solar Cells²²⁻³¹ and Photovoltaics.³²⁻³⁸ Indeed, the inclusion of such materials into optoelectronics started with the pioneering works of Mitzi et al.¹⁶ who successfully assembled LED and TFT devices based on (H₂AEQT)PbCl₄ and [(PEA)₂SnI₄] hybrid bidimensional (2D) systems, respectively. Shortly after that, we have witnessed an unpredicted breakthrough and quick evolution in the field of Solar Cells and Photovoltaics, with the realization of highly efficient solid-state hybrid solar cells based on hybrid material absorbers. In fact, recent implementations of OIH absorber with an organic hole conductor enabled Power Conversion Efficiencies (PCEs) greater than 20%,²⁹⁻³¹ and has been recognized as the “next big thing in photovoltaics”.³²

More recently, the OIH perovskites have interestingly emerged as a highly promising class of light emitters, which can be used as phosphors for optically pumped White Light-Emitting Diodes (WLEDs).³⁹ Broadband emissions with excellent color tunability and high photoluminescence quantum efficiency (PLQE) have been observed in many low-dimensional OIH materials. The first examples of broadband WL emission based on 2D lead bromide and lead chloride based OIH were reported by Dohner et al.^{40,41} on (EDBE)[PbBr₄] and (N-MEDA)[PbBr₄] and then by Yangui et al.⁴² on the OIH perovskites (C₆H₁₁NH₃)₂[PbBr₄].

Since these pioneering works, several research groups have reported new examples,⁴³⁻⁶⁸ achieving great values of the PLQY of 45% and 36% in 1D lead bromide⁶⁸ and 0D lead chloride⁶⁶ hybrid materials. Moreover, broadband emission FWHM's (Full Width at Half Maximum) reached 0.9 and 0.95 eV in more than one material,^{40,41,46} and the CRI exceeded in some case the commercial requirements for indoor illumination,^{41,47,61,64,65,67} where both "warm" and "cold" white light have been generated.

From all these results, some concluding remarks can be drawn. Except for some recently published works,^{48,51,58,59,67} most of the WL emission reported are based on 1D and 2D lead bromide or chloride hybrid materials. The origin of the broadband emission in these OIH was deeply discussed and associated mainly with the presence of a short-range electron-distortion coupling in the strongly deformable lattice which generates self-trapped carriers in the inorganic substructures.^{43,45,49,52,60} It was also evidenced that lowering the dimensionality of OIH materials from 2D to 1D to 0D enables a strong quantum confinement of the self-trapped excited states that produce highly efficient below-gap broadband luminescence and give efficient bluish white-light emissions.^{43,46} Also, these single-component white-light emitters have an easily controlled emission, better color-rendering capability, and reproducibility compared to the properties of the multicomponent phosphors.⁶⁹ More importantly, it has been proven that the white-light emission intensity remains undiminished after continuous UV irradiation for 30 days under atmospheric conditions (~60% relative humidity).^{49,60}

In this work, we report the synthesis, crystal structure and photophysical properties of a new zero-dimensional organic lead chloride material, namely (TAE)₂[Pb₂Cl₁₀](Cl)₂ (TAE = tris-(2-aminoethyl) ammonium (C₆N₄H₂₂)), showing white light emission properties and exhibiting unprecedented values of 1.25 eV (247 nm) for the FWHM and CRI at 96 with chromaticity coordinates of (0.30,0.33). The temperature-dependence of this emission was also undertaken,

on single crystals of $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ and the corresponding ammonium salt TAE.HCl to investigate the origin and the thermal dependence of the WL performances of the material.

Experimental section

Single crystals of the OIH material $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ were synthesized by slow solvent evaporation at room temperature. Stoichiometric amounts of PbCl_2 and Tris (2-aminoethyl) amine, $(\text{NH}_2\text{CH}_2\text{CH}_2)_3\text{N}$, was stirred in aqueous solution of HCl and kept in the dark at room temperature. After two weeks, colorless platelets single crystals were formed. The crystal structure was determined using a supernova 4-circle micro-source diffractometer (Oxford Diffraction) equipped with a two-dimensional ATLAS detector, and using graphite monochromatized $\text{MoK}\alpha$ radiation ($\lambda = 0.71073 \text{ \AA}$).

All optical measurements were performed on $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ HOI and its corresponding ammonium salt TAE.HCl. The optical absorption (OA) spectra were measured at room temperature on spin-coated films using a conventional UV–Vis spectrophotometer (HITACHI, U-3300). Photoluminescence (PL) measurements were performed using a JOBIN YVON HR 320 spectrophotometer under 266 and 375 nm excitation wavelengths, and with 50 mW excitation powers. For temperature dependence measurements, we used a Janis closed cycle cryostat. Photoluminescence excitation (PLE) was recorded at room temperature on a Fluoromax-4 spectrofluorimeter equipped with a xenon lamp as an excitation source.

Crystal Structure

X-rays diffraction experiments performed at room temperature (300K) show that $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ crystallizes as an OIH material in the triclinic space group P-1 and the relevant crystallographic data are summarized in Table S1 of the supplementary material (SM). As shown in Figure 1, the crystal structure shows discrete $[\text{Pb}_2\text{Cl}_{10}]^{6-}$ entities, two $[\text{TAE}]^{4+}$ cations and two Cl^- ions packed in a triclinic unit cell. The $[\text{TAE}]^{4+}$ cations are located in general positions with their three ammonium groups involved in N-H...Cl

hydrogen bonds with the Cl atoms of $[\text{Pb}_2\text{Cl}_{10}]^{6-}$ groups and Cl^- counter ions. In this structure, there exist two different $[\text{Pb}_2\text{Cl}_{10}]^{6-}$ groups, which are both located on an inversion center forming Pb1 and Pb2 edge-sharing bi-octahedra stacked along the crystallographic a direction. It presents structural strains considering the distortions of the two $[\text{Pb}_2\text{Cl}_{10}]^{6-}$ bi-octahedra building blocks (see Figure S1-S5). These deformations involve modifications in Pb–Cl bond lengths (d) and Cl–Pb–Cl angles (α), and can be quantitatively evaluated using the followings formulas which introduce the octahedral elongation λ_{oct} and octahedral angle variance σ_{oct}^2 :⁷⁰

$$\lambda_{\text{oct}} = \frac{1}{6} \sum_{i=1}^6 (d_i/d_0)^2 \quad , \quad \sigma_{\text{oct}}^2 = \frac{1}{11} \sum_{i=1}^{12} (\alpha_i - 90)^2$$

Here, d_i are six independent Pb–Cl bond lengths, d_0 is the mean Pb–Cl bond length and α_i are the Cl–Pb–Cl angles. From the structural parameter in Table S2, the average values of Pb1–Cl and Pb2–Cl bond lengths are 2.9012 and 2.9035 Å, leading to an octahedral elongation λ_{oct} of 1.0011 and 1.0008, respectively. Furthermore, the corresponding octahedral angle variance, σ_{oct}^2 , are evaluated as 28.4 and 48.2 °², respectively. Such values are very large when compared with reported ones (5.98 °² for (CyBMA)[PbBr₄]⁵⁴, 8.28 °² for (C₄N₂H₁₄)[PbCl₄]⁶³ and 14.2 °² for (EDBE)[PbI₄]⁵⁰) and reveal the existence high degrees of distortion for both inorganic entities. In this context, it is well known that such distortion plays a crucial role in the formation of self-trapped excitons, which are at the origin broadband emission.^{45,50,52,53}

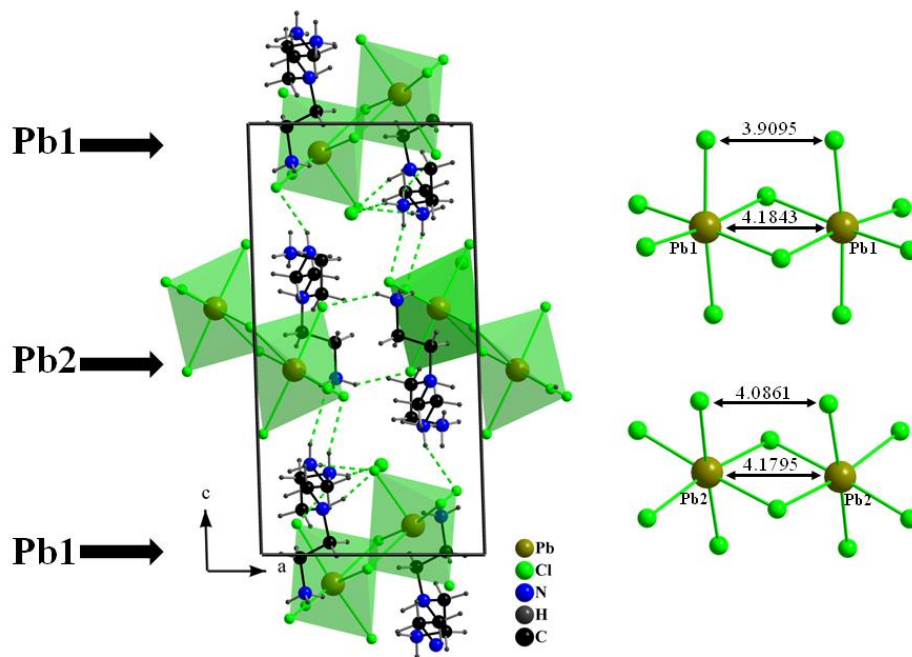


Figure 1. (Left) Crystal structure of the $(TAE)_2[Pb_2Cl_{10}](Cl)_2$ HOI material projected along the crystallographic b-axis. (Right) Two $[Pb_2Cl_{10}]^{6-}$ individual bi-octahedra showing different distortions values (T=300 K).

Photophysical Properties

Figure 2 presents the room temperature photophysical properties of the OIH material $(TAE)_2[Pb_2Cl_{10}](Cl)_2$ and its corresponding ammonium salt TAE.HCl. For the OIH material, the optical absorption spectrum (OA) (Figure 2(a)) shows a sharp absorption band at 3.77 eV (328 nm) with a characteristic shape of exciton absorption, similar to that reported in lead chloride-based OIH materials,^{16,40,47,61-66} and attributed to the lowest excitons localized in the inorganic bi-octahedra Pb_2Cl_{10} . Moreover, the OA spectrum of the organic salt (TAE.HCl) shows an HOMO-LUMO gap of the organic cation at almost the same energy (3.99 eV) and characterized by wide absorption band covering the entire of Pb_2Cl_{10} excitonic peak.

In addition, under UV irradiation $(TAE)_2[Pb_2Cl_{10}](Cl)_2$ single crystals show an impressive intense white light emission observable even with naked eyes at room temperature. Upon 3.81 eV laser (Figure 2(b)), the photoluminescence (PL) spectrum shows a broad band covering

the entire of the visible range and resulting from the overlap of two bands located at 2.07 eV and 2.83 eV, noted S1 and S2, respectively.

For a deep analysis of the PL spectrum and in order to investigate the contribution of the organic TAE cation and $\text{Pb}_2\text{Cl}_{10}$ inorganic anion in the emission process, we have studied under the same conditions the PL spectrum of the organic salt, which exhibits an intense and wide band at 3.07 eV. By comparison of both spectra, one can attribute the two components of the white light emission as follows: The S2 peak at 2.83 is unambiguously associated to the HOMO-LUMO electronic transitions within the organic cation. The 240 meV red shift could be assigned to the interaction between organic and inorganic components in the $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ leading to tune the band structure and to modify the luminescence properties.⁷¹

Whereas, the S1 peak at 2.07 eV originates from excitonic levels of the $\text{Pb}_2\text{Cl}_{10}$ bioctahedra, and to boost this assumption, we have recorded the photoluminescence excitation (PLE) spectrum at 2.07 eV (Figure 2(e)). The measurements reveal the presence of a sharp peak at 3.95 eV (313 nm), typically characteristic of the well-known Wannier-type excitons observed in similar compounds based on PbCl_6 octahedra.^{57,66} On the other hand, the very large Stokes shift with respect to excitons absorption (1.7 eV) may be interpreted as a result of the strong interaction between the organic and inorganic ions. From an optoelectronic point of view, this material is then assimilated to the self-assembly of two luminescent organic and inorganic ionic entities, placed in very close positions (2 to 3 Å) and having almost equal HOMO-LUMO gaps around (3.95 eV). Such characteristics have been observed in many metal halide hybrids and have been considered to favor strong interactions leading to the emission of broad bands of white light.^{38,53,61,72} It is well known that this type of interaction is characterized by a photoluminescence spectrum strongly dependent on the excitation wavelength.^{38,61,72}

In this context, we carried out photoluminescence measurements under different excitations presented in the right side of Figure 2. For an excitation of 4.66 eV, energy above the organic and inorganic gap, the two entities are excited and their resonant interaction strongly enhances the intensities of both S1 and S2 peaks, giving rise to a single very wide band and a white light of impressive intensity. For an excitation of 3.81 eV, energy included in the organic and inorganic absorption bands tails, the two species are excited and we observe both S1 and S2 resolved peaks, with a lower intensity than the previous one. When exciting by energy of 3.3 eV, only the organic cation is excited and the PL spectrum is reduced to the single band, as if the bioctahedra do not exist.

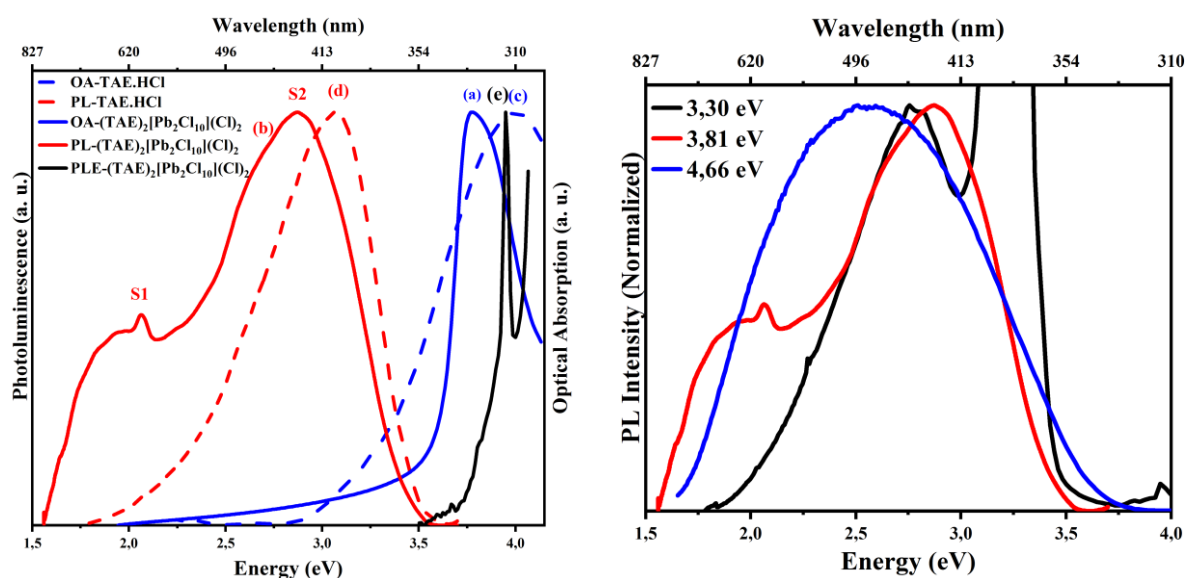


Figure 2. (Left) (a,b) Optical absorption spectrum (solid blue line) and the PL emission spectrum (solid red line) of the OIH material $(TAE)_2[Pb_2Cl_{10}](Cl)_2$. (c,d) Optical absorption spectrum (dashed blue line) and the PL emission spectrum (dashed red line) of the ammonium salt TAE.HCl. (e) PLE spectrum (solid black line) of the $(TAE)_2[Pb_2Cl_{10}](Cl)_2$ material. (Right) PL emission spectra of $(TAE)_2[Pb_2Cl_{10}](Cl)_2$ excited with (black) 3.30 eV, (red) 3.81 eV and (blue) 4.66 eV, ($T=300$ K).

White-Light Emission

As mentioned above, the studies conducted so far on hybrids based on lead halides have shown that the white light emission mechanisms as well as the profiles of the PL bands depend on many parameters, mainly, the optical activity of the organic cation, the distortion of the inorganic subnetwork and dimensionality. In order to better understand the white light

emission mechanism in our material, we have carried out a comparative study of the photophysical characteristics of the materials studied so far. In table 1, we have gathered their main optical characteristics. An overview of these properties shows without ambiguity that for chlorinated and brominated compounds, the lowering in dimensionality is accompanied by a blue shift of the emission peak. Indeed, it was recently demonstrated that lowering the system dimensionality helps in the formation of exciton self-trapping at any exciton-phonon interaction strength, and leads to the observation of broad band luminescence with large Stokes shifts.⁴³ Furthermore, these studies have shown that the common characteristic to all these materials lies in the fact that the emission of broad bands involves the recombinations of self-trapped excitons, due to the high distortion of the PbX_6 octahedra.

As we mentioned in the description of the structure, our material belonging to the low-dimensional 0D HOIM family, does not escape the rule and exhibits a remarkable distortion, stronger than those reported in the literature. In order to probe the contribution of such self-trapped excitonic states in the emission mechanism, we carried out a study of the PL intensity as a function of the excitation power, which presents an excellent tool to confirm the self-trapping excitons origin.^{40,44,46} As shown in Figure 4, the intensity of the white light emission excited at 4.66 eV showed a linear dependence on the excitation power density up to 25 MW/cm^2 , suggesting that the current WL emission originated from self-trapped excited states. Such a behavior is in good agreement with previous works, especially the results of Dohner et al. on $(\text{N-MEDA})\text{PbBr}_4$ ⁴¹ and $(\text{EDBE})\text{PbBr}_4$.⁴⁰ We can then conclude, that in addition to the organic and inorganic free excitons involved in the mechanism described above, is added the contribution of the self-trapped excitons. This can explain the impressive intensity of the white light emission, as well as the remarkable width of the PL band. Indeed, in the comparative table, the most remarkable property we can notice in terms of properties - structure correlation is the unprecedented width of the emission band of our compound (1.25

eV). To our knowledge, this value stand for the record of the broadening emission observed between overall the WL emitting OIH materials that did not exceed the value of 1 eV.

In order to better illustrate the interesting findings of our 0D OIH material $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$, the white light characteristics, namely the CIE coordinates, the (CCT) and the (CRI), are also presented for comparaison with those of other WL emitting HOI based materials in Table 1. In this study, the $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ excited with the 4.66 eV laser exhibits CIE coordinates at (0.30, 0.33) very close to the reference coordinates (0.33, 0.33) defined in the CIE chart as the white point or equal energy point (Figure 3). Also, the broad band emission from this material exhibits a CCT of 6741 that affords “cold” white light. Most importantly, the value of the CRI is evaluated at 96, which is the highest value revealed among the other WL emissions from other OIH materials⁶⁷ (see Table 1) and exceeds even the high CRI indexes (>90) exhibited by some mixed-phosphor light sources.^{3,7} Generally, the desired indoor lighting applications typically require values of CRI above 80 and closest to 100 which is, by definition, the black body radiator CRI.^{73,74} Moreover, from the CRI values depicted in Table 1, we concluded that the highest values are those obtained from chlorided materials and this agree well with the hypothesis conceived by Dohner et al. that chloride substitution improves the material’s properties (CRI).⁴¹ All of these characteristics clearly show the important originality of this new material, and can enable applications sought for many indoor lighting requiring inexpensive and large-area coatings.

Table1. Reported photophysical properties and white light characteristics in some OIH materials.

Compound	Photophysical properties (eV)				White light characteristics		
	PL	Abs	Stokes Shifts	FWHM	CIE (x,y)	CCT (K)	CRI
EDBE PbBr_4 (2D) ⁴⁰	2.16	3.34	1.18	0.8	0.39 , 0.42	3990	84
$(\text{C}_6\text{H}_{14}\text{NH}_3)_2 \text{PbBr}_4$ (2D) ⁴²	2	3.19	1.2	0.66	-	-	-
N MEDA PbBr_4 (2D) ⁴¹	2.22	3.13	0.91	0.9	0.36 , 0.41	4669	82
AEA PbBr_4 (2D) ⁴⁵	2	3.3	1.3	-	0.29 , 0.34	-	87
$\text{C}_4\text{N}_2\text{H}_{14} \text{PbBr}_4$ (1D) ⁴⁶	2.61	3.3	0.69	0.9	0.21 , 0.28	-	63
$\text{C}_4\text{N}_2\text{H}_{14} \text{PbBr}_4$ (1D) ⁴⁶	2.61	3.3	0.69	0.9	0.21 , 0.27	-	66

EDBE PbCl ₄ (2D) ⁴⁰	2.3	3.53	1.23	0.95	0.33 , 0.39	5509	81
PEPC PbCl ₄ (2D) ⁴⁷	2.1	3.64	1.54	-	0.37 , 0.42	4426	84
(C ₄ H ₉ NH ₃) ₂ PbCl ₄ (2D) ⁶⁵	2.28	-	-	-	0.37 , 0.40	4423	86
(2meptH ₂) PbCl ₄ (2D) ⁶²	2.15	3.51	1.36	0.73	0.39 , 0.44	3958	84
C ₆ H ₁₂ N ₃ PbCl ₄ (2D) ⁶⁴	2.16	3.26	1.1	0.86	0.36 , 0.37	4203	93
AQ PbCl ₄ (1D) ⁵⁷	2.3	3.22	0.92	-	0.29 , 0.43	-	-
C ₄ N ₂ H ₁₄ PbCl ₄ (1D) ⁶³	2.47	3.87	1.4	0.854	0.29 , 0.39	3880	-
C ₄ N ₂ H ₁₄ PbCl ₄ (1D) ⁶³	2.19	3.87	1.68	0.664	0.40 , 0.43	3880	-
C ₅ H ₁₄ N ₂ PbCl ₄ ·H ₂ O (1D) ⁶¹	2	-	-	-	0.39 , 0.37	3445	94
(TAE) ₂ [Pb ₂ Cl ₁₀](Cl) ₂ (0D)	2.51	3.77	1.26	1.25	0.30 , 0.33	6741	96

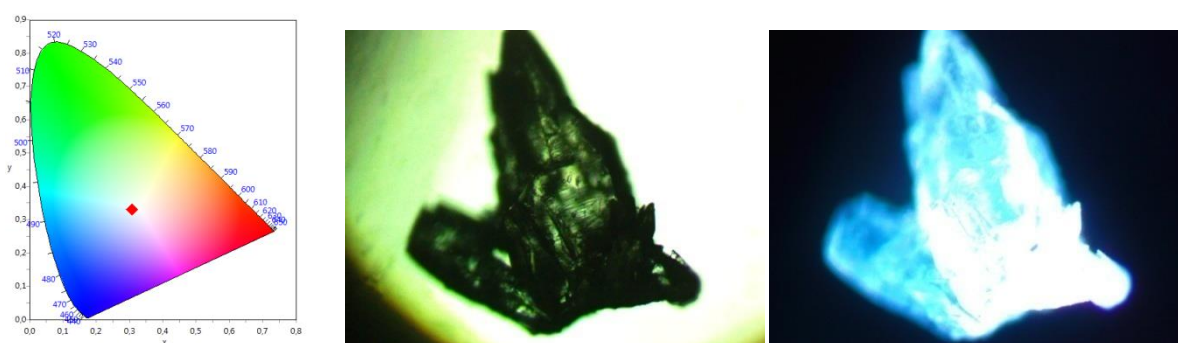


Figure 3. (Left) CIE Chromaticity coordinates of (TAE)₂[Pb₂Cl₁₀](Cl)₂ emission excited by 4.66 eV. (Right) Photographs showing the same microcrystal before and during photoluminescence under 3.30 eV irradiation (T=300 K).

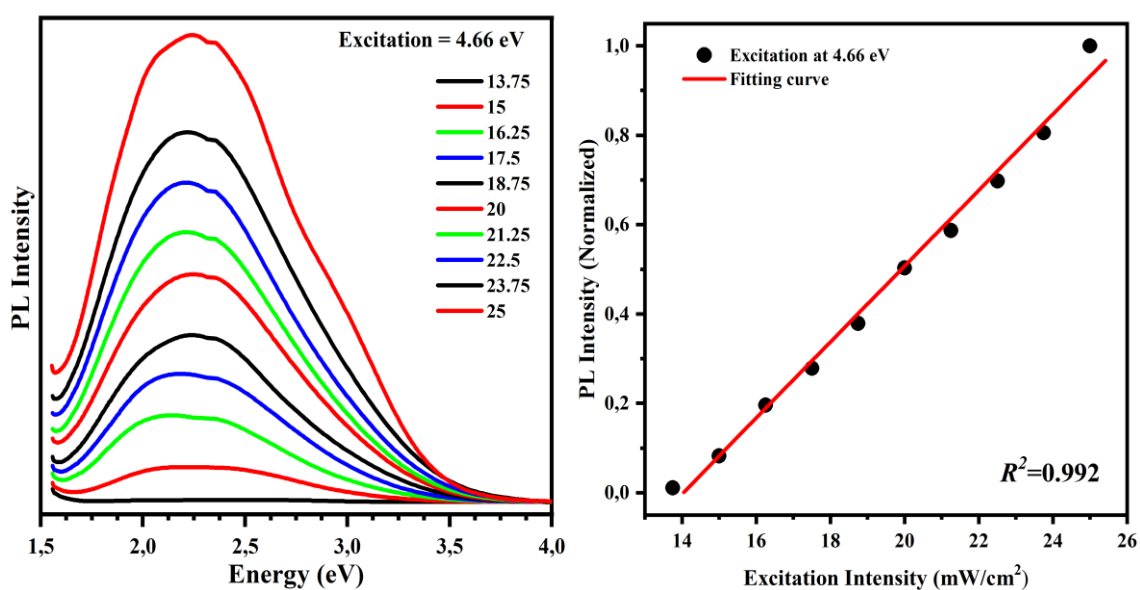


Figure 4. (Left) Variation of the emission spectra at 4.66 eV with different excitation intensities (13.75-25 mW/cm²). (Right) Linear dependence of emission intensity on excitation density and a linear fit, (T=300 K).

Temperature Dependence of WL Emission

To get further information about the stability of excitons, we measured the thermal dependence of the emission spectra of the OIH compound $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ in the temperature range [10-300 K]. As shown in Figure 5a, significant changes were observed on the behaviors of S1 and S2 PL lines. The integrated intensity of peak S1 continuously decreases as a function of temperature and shows an anomaly around 50 K. We described the quenching of S1 excitonic PL emission as a function of temperature, as a consequence of the competition with a non-radiative channel which described by the Arrhenius-type law,^{75,76}

$$I_{PL} = \frac{I_0}{1 + a \exp\left(\frac{-E_a}{k_B T}\right)} \quad (1)$$

where I_0 is the low-temperature PL intensity, k_B is the Boltzmann constant, a is the ratio between the radiative and the nonradiative decay rates, and E_a is the activation energy. The best fit gives $I_0 = 2.7 \times 10^4 \pm 480$ (a.u), $a = 25 \pm 7$, and $E_a = 5 \pm 0.5$ meV. This activation energy presents a good agreement with those measured in similar 2D OIHP such as $(\text{C}_6\text{H}_{11}\text{NH}_3)_2\text{PbBr}_4$ ⁴² and $(\text{C}_6\text{H}_5\text{C}_2\text{H}_4\text{NH}_3)_2\text{PbI}_4$.⁷⁷ On the other hand, S2 PL peak intensity shows three different behaviors as a function of temperature (Figure 5b): first, it decreases until reaching 50 K (region I), then increases between 50 and 180 K (region II) and finally quenches at higher temperature (region III). More importantly, at room temperature (curve b in Figure 5), the emission spectrum of $(\text{TAE})_2[\text{Pb}_2\text{Cl}_{10}](\text{Cl})_2$ was dominated by S2 PL band and on cooling, three different behaviors were observed. Indeed, as displayed in the inset of Figure 5b, in the temperature range (I), S1 peak is dominant and the ratio between S1 and S2 increases, then it suddenly decreases in region II, and below 180 K the S1 PL band becomes dominant, and the ratio between S1 and S2 PL intensities stabilizes in region III above 180 K. Such behavior can be explained by two ways. First, it is well known that excitons stability is

much higher at low temperature.⁷⁸ Therefore; the intensity of S1 excitonic peak enhances by lowering the temperature and becomes much sharper. Second, the presence of a singularity around 50 K on the thermal behavior of S1 could be the result of structural distortions often observed at low temperature in OIH materials.⁴⁰⁻⁴² On the other hand, the evidence of three different temperature ranges on the thermal behavior of the integrated intensity of S2 PL band further strengthens the hypothesis of the presence of critical structural changes especially around 50 and 180 K. Moreover, the abnormal increase of the intensity of S2 PL peak between 50 and 180 K is accompanied by the decreases of S1 PL intensity, which suggests the presence of a strong competition in the recombination processes of S1 and S2 PL bands, which can be activated by the presence of structural phase transitions in this temperature range.⁷⁹ Similar behavior was also found in the temperature dependence of the full width at half maximum (FWHM) of S1 and S2 PL peaks. Indeed, the FWHM of S1 excitonic peak slightly decreases in region I, and then continuously increases as a function of temperature (Figure 5c). Assuming linear electron-phonon coupling and harmonic lattice vibrations, the temperature dependence of the FWHM of S1 PL peak allows estimating the vibrational modes effective frequency coupled to the electronic transition using the following model:⁸⁰

$$\Gamma(T) = \Gamma_0 + \Gamma_{LO} \left(e^{\frac{E_{LO}}{k_B T}} - 1 \right)^{-1} + \Gamma_{inh} e^{\frac{-E_b}{k_B T}} \quad (2)$$

where, Γ_0 is the emission FWHM at 0 K, E_{LO} is the energy of the longitudinal optical phonon mode coupled to the electronic transition, E_b represents the average binding energy of emissive defect states, and the constants Γ_{LO} and Γ_{inh} give the relative contributions of electron-phonon coupling and inhomogeneous broadening to the emission line width. The best fit of the data (Figure 5c) gives $E_{LO} = 6 \pm 1$ meV, $E_b = 12 \pm 3$ meV, $\Gamma_0 = 193 \pm 8$ meV, $\Gamma_{LO} = 170 \pm 6$ meV and $\Gamma_{inh} = 93 \pm 7$ meV.

Here, some concluding remarks can be drawn. (i) First, these fitting values are in agreement with those obtained from previous broad emitting 2D perovskites.^{40,49,60} Although the value of Γ_{LO} (170 ± 6 meV) is slightly larger than the ones available in the literature of broad emitting lead bromide perovskites: (130 ± 7 meV)⁴⁰ and (131 ± 7 meV);⁶⁰ however all these values remain considerably larger (up to more than threefold) than those of WL emitting lead chloride based perovskites (49 ± 3 meV).⁴⁹ The present tendency clearly reveals the existence of a strong exciton-phonon coupling in (TAE)₂[Pb₂Cl₁₀](Cl)₂ material. (ii) The LO phonon energy ($E_{LO} = 6 \pm 1$ meV) obtained from this fitting of Eq. (2), agrees well with the activation energy ($E_a = 6 \pm 1$ meV) deduced from the Arrhenius fitting of the integrated intensity of S1 peak. In addition, this LO phonon energy along with the E_b energy can be coincided with the frequencies of 67 cm^{-1} (8.3 meV) and 90 cm^{-1} (11.2 meV), evidenced by Raman spectroscopy (Figure S5) and so well align well within the range of PbCl stretching frequencies. In the same context, we observed an intense Raman scattering of our material at room temperature, further confirming the existence of an electron-phonon coupling in a strongly deformable lattice. (iii) We can then attribute the exciton self-trapping process in the current material to originate primarily from the inorganic lattice like concluded by Hu et al. for (N-MEDA)[PbBr₄] material.⁴³

On the other hand, the temperature dependence of the FWHM of S2 peak shows three different regimes for each temperature range which suggests, once again, the possible presence of structural changes as a function of temperature. Concerning the thermal behavior of S1 and S2 PL position, they keep almost the same position as a function of temperature, with the presence of singularity around 180 K, for the position of S2 peak (Figure 5d).

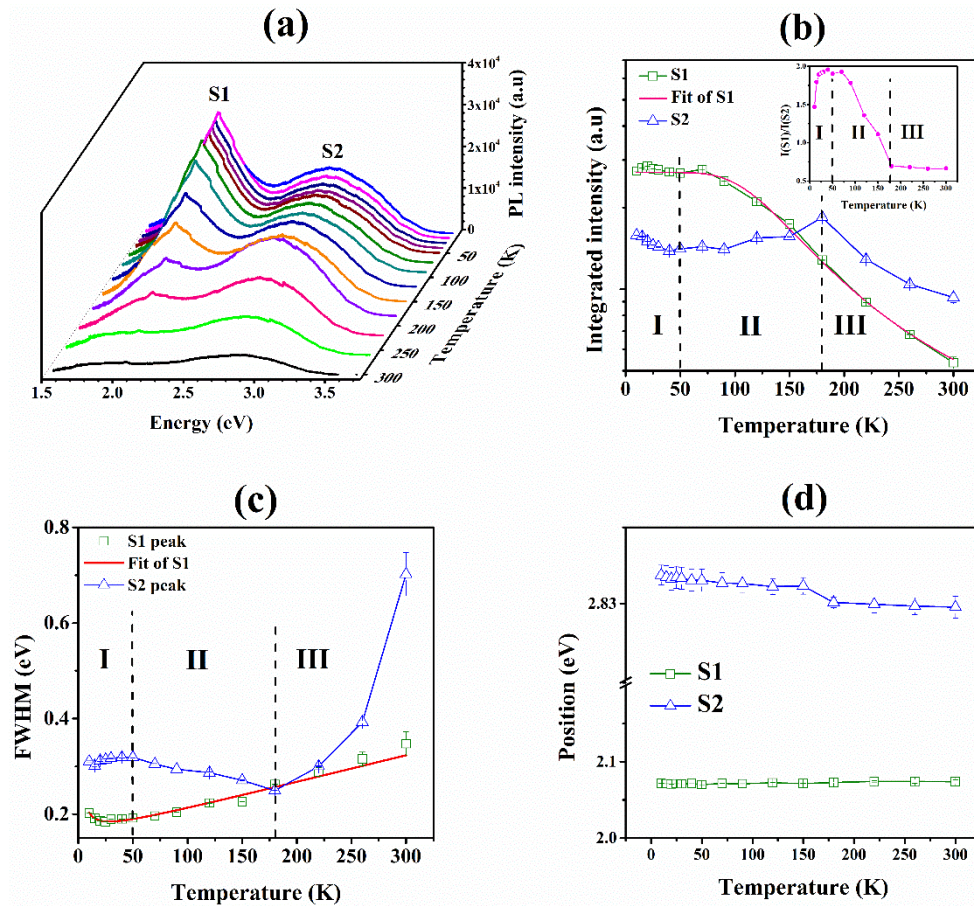


Figure 5. (a) 3D plot of the temperature dependence of the PL spectra of $(TAE)_2[Pb_2Cl_{10}](Cl)_2$ single crystals, in 10-300 K temperature range, under 3.816 eV excitation. Plot of temperature dependence of (b) the integrated intensity, (c) FWHM, and (d) the peak position of both S1 and S2 PL peaks. Inset of Figure 5b shows the ratio between intensities of S1 and S2 as a function of temperature.

Figure 6a, shows the temperature dependence of the PL spectra of the ammonium salt TAE.HCl. Here as well, the study of the thermal dependence of the integrated intensity, FWHM and the position of the PL peak (Figures 6b, 6c and 6d, respectively) shows the existence of three different temperature regimes (noted I, II, and III in Figure 6) which appear almost in the same temperature ranges as those observed in the OIH material $(TAE)_2[Pb_2Cl_{10}](Cl)_2$. Indeed, the intensity of the maximum emission first decreases until 60K (region I), then increases between 60 and 200 K (region II), and finally quenches at higher temperature (Figure 6b). Moreover, the FWHM of the emission band increases in region I, then decreases in region II and finally increases at higher temperature in region III.

More importantly, the maximum emission of TAE.HCl ammonium salt strongly blue-shifts between 100 and 130 K.

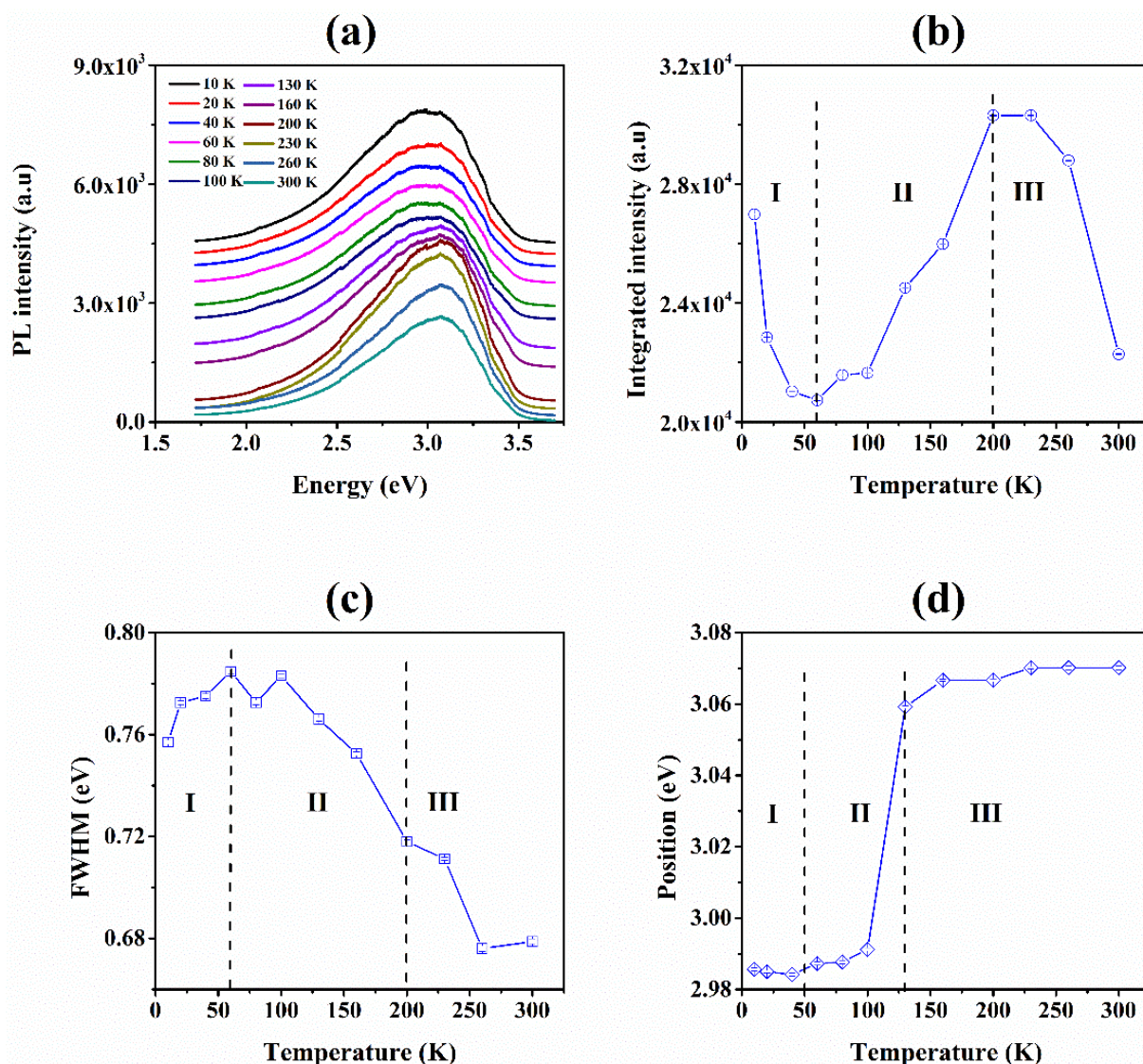


Figure 6. (a) Temperature dependence of the PL spectra of TAE.HCl ammonium salt. Plot of the temperature dependence of (b) the integrated intensity (c) the FWHM and (d) the position of the PL peak.

Conclusion

In summary, we have synthesized a novel low-dimensional lead chloride-based OIH material by slow solvent evaporation at room temperature. The crystal structure done using single crystal X-ray diffraction shows a 0D structure formed by discrete $[\text{Pb}_2\text{Cl}_{10}]^{6-}$ entities and two $[\text{TAE}]^{4+}$ organic cations located in general positions and two Cl^- chlorine ions. The

investigations of the photo-physical properties revealed a sharp excitonic absorption and a very broadband WL emission, even at room temperature, consisting of two PL bands (S1 and S2) located respectively at 2.07 and 2.83 eV, attributed respectively, to free excitons confined into the inorganic bi-octahedra $\text{Pb}_2\text{Cl}_{10}$ and in the organic salts, respectively. This material excited by 4.66 eV irradiation, exhibited a white light emission with record of the emission broadening with 1.25 eV (247 nm) as FWHM and offered the highest CRI value of 96. This is the effect of the supplementary light emissions from self trapped excitons generated within the highly distorted $\text{Pb}_2\text{Cl}_{10}$ bi-octahedras. We can state now that our material exhibits a very particular case of HOIM by showing a competition of three emission mechanisms, organic and inorganic free excitons and self trapped excitons. From the study of the temperature dependence of PL spectra of this OIH material, we deduced an exciton activation energy $E_a = 6$ meV and detected the presence of singularities around 50 and 180K on the behaviors of the thermal dependence integrated intensity, the FWHM, and the position of the PL peaks. These singularities were also tracked from the study of the thermal dependences of the PL spectra of the associated ammonium salt, and are attributed to structural phase transitions, often observed in OIH perovskites materials.

Considering the low cost character of the present new WL material, its room-temperature facile synthesis and excellent optical properties, this new perovskite phosphor which enriches the restricted list of WL OIH materials, represents a highly promising alternative to replace conventional inorganic rare-earth-based phosphors and quantum dot-based phosphors that currently dominate the field of optically pumped WLEDs.

Supporting Information: Crystallographic data, relevant structural parameters, crystal and molecular structures descriptions and Raman spectrum.

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