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Article

Facile Fabrication of Multifunctional ZnO Urchins on Surfaces

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Abstract: Functional ZnO nanostructured surfaces are important in a wide range of applications. Here we report the simple fabrication of ZnO surface structures at near room temperature with morphology resembling that of sea urchins, with densely packed, μm -long, tapered nanoneedles radiating from the urchin center. The ZnO urchin structures were successfully formed on several different substrates with high surface density and coverage, including silicon (Si), glass, polydimethylsiloxane (PDMS), and copper (Cu) sheets, as well as Si seeded with ZnO nanocrystals. Time-resolved SEM revealed growth kinetics of the ZnO nanostructures on Si, capturing the emergence of “infant” urchins at the early growth stage and subsequent progressive increases in the urchin nanoneedle length and density, whilst the spiky nanoneedle morphology was retained throughout the growth. $\varepsilon\text{-Zn(OH)}_2$ orthorhombic crystals were also observed alongside the urchins. The crystal structures of the nanostructures at different growth times were confirmed by synchrotron X-ray diffraction measurements. On seeded Si substrates, a two-stage growth mechanism was identified, with a primary growth step of vertically aligned ZnO nanoneedle arrays preceding the secondary growth of the urchins atop the nanoneedle array. The antibacterial, anti-reflective, and wetting functionality of the ZnO urchins—with spiky nanoneedles and at high surface density—on Si substrates was demonstrated. First, bacteria colonization was found to be suppressed on the surface after 24 h incubation in gram-negative *Escherichia coli* (*E. coli*) culture, in contrast to control substrates (bare Si and Si sputtered with a 20 nm ZnO thin film). Secondly, the ZnO urchin surface, exhibiting superhydrophilic property with a water contact angle $\sim 0^\circ$, could be rendered superhydrophobic with a simple silanization step, characterized by an apparent water contact angle θ of $159^\circ \pm 1.4^\circ$ and contact angle hysteresis $\Delta\theta < 7^\circ$. The dynamic superhydrophobicity of the surface was demonstrated by the bouncing-off of a falling 10 μL water droplet, with a contact time of 15.3 milliseconds (ms), captured using a high-speed camera. Thirdly, it was shown that the presence of dense spiky ZnO nanoneedles and urchins on the seeded Si substrate exhibited a reflectance $R < 1\%$ over the wavelength range $\lambda = 200\text{--}800\text{ nm}$. The ZnO urchins with a unique morphology fabricated via a simple route at room temperature, and readily implementable on different substrates, may be further exploited for multifunctional surfaces and product formulations.

Keywords: ZnO urchins; nanostructured surfaces; *E. coli*; superhydrophilic; superhydrophobic; anti-reflective surfaces

1. Introduction

Nanostructured surfaces are widespread in nature and many are ingeniously multifunctional. This has inspired considerable effort in the fabrication of hierarchical surface nanotextures. For instance, lotus leaf inspired nanostructures on silicon, TiO₂, and polymer surfaces exhibit superhydrophobic, self-cleaning and antibacterial characteristics [1–5]. Nanostructured surfaces on silicon and glass substrates mimicking cicada/dragonfly wings or moth eyes possess superhydrophobic and antimicrobial characteristics, as well as gradient refractive indices for anti-reflective applications [6–11].

Surfaces bearing ZnO nanostructures are among the most widely studied, with a wide range of applications in gas and bio-sensing, field emission devices, ultraviolet detectors, and photovoltaics [12] due to their thermostability [13], low-cost production [14], antimicrobial properties [15–19], and biocompatibility [20]. Several fabrication methods of ZnO nanostructured surfaces have been reported [21–28], such as hydrothermal synthesis, laser ablation, sputtering, thermal decomposition, evaporation induced self-assembly [29,30], and the sol-gel technique. Various ZnO nanostructures that have been reported include one-dimensional (1-D) morphologies such as nanowires [31–33], nanofibers [29,30], nanorods [34–37], micro-dendrites [38,39], and nanotubular structures [40–44], as well as 3-D architectures such as flowers/urchins [45–50], tetrapods/jack-like [51,52], and hedgehogs [53]. The 3-D nanostructures with enhanced surface area may be used as substitutes for 1-D nanostructure arrays with enhanced functionalities, but their fabrication requires either sophisticated instrumentation or elevated temperatures (and thus high energy input).

Here, we report the simple fabrication of multifunctional 3-D ZnO urchins on silicon (Si) substrates in a solution-based synthesis at near room temperature (RT) using a one-step procedure, which could also be applied on Si substrates seeded with ZnO crystals and various other substrates. The morphology of these ZnO surface structures resembles that of sea urchins, with tapered nanoneedles μm in length radiating from a central core. The growth kinetics of the ZnO urchins were studied by examining their intermediate morphologies at different growth time intervals (0.5–12 h). The ZnO urchin-coated surface exhibited high anti-reflectance, and superhydrophobicity after silanization, as investigated by contact angle, contact angle hysteresis and drop impact analyses. They were also bacteriophobic against *Escherichia coli* (*E. coli*). This simple fabrication method for ZnO urchins could also be adapted for a variety of materials such as polydimethylsiloxane (PDMS), copper sheets and glass substrates, demonstrating its versatility.

2. Materials and Methods

2.1. Substrate Preparation and Cleaning

Silicon substrates were prepared by cutting a silicon wafer (100 mm diameter, P type, B dopant, $\langle 100 \rangle$, 1–100 $\Omega\text{-cm}$, 500 μm thick, single-sided polish, test grade, University Wafer, Inc., South Boston, MA, USA) into 1 cm \times 1 cm squares. All substrates, including copper sheet, glass slide (Polar Industrial Corporation, Mumbai, India) and polydimethylsiloxane (Dow Corning, Sylgard 184, Cardiff, UK) were cleaned by a subsequent sonication in acetone, ethanol, and deionized water (10 min in each) followed by drying with N₂.

2.2. Seeding

A seeding procedure was used to produce nucleation sites for the growth of ZnO nanostructures [31]. Cleaned silicon substrates were dipped in a solution of zinc acetate dihydrate (Zn(CH₃COO)₂·2H₂O, 99%, Sigma Aldrich, Dorset, UK) in ethanol (CH₃CH₂OH, Absolute,

Sigma Aldrich) for a few seconds, rinsed with clean ethanol and then dried with N₂. This coating step was repeated 5 times for each seeded silicon substrate (see Figure S1). Subsequently, the substrates were heated to 300 °C on a hot plate, and annealed for 30 min in air to thermally decompose zinc acetate crystallites to ZnO islands with (0001) planes parallel to the silicon substrate surface [31].

2.3. ZnO Urchin Growth

To grow the ZnO nanostructures, 50 mL aqueous solution of zinc nitrate hexahydrate (Zn(NO₃)₂·6H₂O, 98%, Sigma Aldrich) solution was added dropwise to 50 mL of potassium hydroxide (KOH, 85%, Fisher Scientific) aqueous solution under constant stirring for 30 min. Next, the alkali solution of zincate ions was carefully poured into the glass Petri dish containing substrates (seeded Si, non-seeded Si, Cu sheet, glass and polydimethylsiloxane, separately for each type) sealed with a glass lid and a paraffin film, and kept at 20 °C ± 2 °C for 12 h. After the synthesis, samples were rinsed with Deionized (DI) water and dried with N₂.

2.4. Bacterial Growth Conditions and Sample Preparation

Isolated single colonies of *E. coli* K-12 (MG 1655) were used to prepare the pre-inoculum [54]. The culture was grown in Luria Bertani (LB) medium for 8 h at 37 °C with constant shaking at 160 rpm. A 0.2% pre-inoculum was added into 10 mL of LB medium and was allowed to grow to 0.3 optical density (O.D.) at 600 nm [55]. The cells were subsequently pelleted and washed with phosphate-buffered saline (PBS). A cell suspension of 0.01 O.D. at 600 nm was used for studying the bacterial interaction with the bare, unmodified silicon, silicon with a 20 nm thin film of ZnO, and ZnO nanourchin surfaces.

2.5. Scanning Electron Microscopy (SEM)

ZnO nanostructured surfaces were imaged using field emission scanning electron microscopy (JSM-IT300 SEM (JEOL)). Dimensions of the ZnO nanourchin-like structures were obtained from the SEM images using ImageJ software. For bacterial sample imaging, substrates were first washed with PBS to remove the loosely adhered cells. Substrates were then dipped in 2.5% of glutaraldehyde to fix the cells. After that all the substrates were allowed to dry in a vacuum. A thin layer of gold (15 nm) was sputtered on the samples using a Quorum sputter coater (Q150T) to avoid the charging effect while doing the SEM. Samples were scanned thoroughly using SEM (sample size 1.3 cm × 1.3 cm) and the imaging was done in triplicate for all the samples. Selecting three specimens from each type of sample (two controls and one ZnO Urchin surface) ensured the repeatability.

2.6. Urchin Dimension Analysis

ImageJ [56] was used to obtain the morphological information of the nanoneedles comprising the ZnO urchin and the vertical nanoneedles, using the length and angle measurement tools.

2.7. X-ray Diffraction (XRD)

The presence of crystalline ZnO after the growth procedure was confirmed using D8 Advance Bruker-AXS diffractometer (Cu K α radiation ($\lambda = 0.154178$ nm), $\theta - 2\theta$ scan, $2\theta = 10^\circ - 90^\circ$, step size -0.02 (2θ), step count -1.5 s).

2.8. Grazing Incidence X-ray Diffraction (GIXRD)

GIXRD analysis of the seeded and non-seeded silicon substrates used to study the time dependence growth of ZnO nanostructures, was performed at Beamline BM28 at the European Synchrotron Radiation Facility (France). Experimental parameters: radiation wavelength: 0.8856 Å, sample to detector distance: 0.24 m, detector: MAR165, calibrant: silver behenate and ZnO powder.

The diffraction patterns were reduced to one-dimensional line profiles using *pyFAI*, a pythonic library for 1D azimuthal/2D radial integrations of diffraction images [57].

2.9. Contact Angle Measurement and High-Speed Imaging

Apparent contact angle [58–60] measurements were carried out using a Krüss Drop Shape Analyser (DSA 100). A 10 μ L water droplet was placed gently on all the substrates to measure the contact angles. A custom-made goniometer setup was used to measure the contact angle hysteresis. The stage of the goniometer was tilted, and the droplet image was captured when the droplet first started to slide to measure the contact angle hysteresis. The advancing and receding contact angles were measured from the droplet image using ImageJ software. The contact angle hysteresis was obtained by calculating the difference between the advancing and receding contact angles. All the measurements were repeated three times. Droplet impact dynamics of water droplet on the superhydrophobic surface (silanized seeded silicon substrate with ZnO nanostructures) were captured using a high-speed camera (Photron FastCam SA4) at 10,000 fps (time resolution of 0.1 ms). The droplets were created by a micro-pipette and released from a height of 7 cm (Weber number)

$$(We) = \frac{\text{inertial force}}{\text{surface tension force}} = \frac{\rho V^2 D_w}{\gamma} = \frac{1000 \times 1.17^2 \times 0.0026}{0.072} = 49,$$

Density of water (ρ) = 1000 $\frac{\text{kg}}{\text{m}^3}$, Velocity of falling droplet (V) = $\sqrt{2gh} = 1.17 \frac{\text{m}}{\text{s}}$,
 Surface tension of water (γ) = 0.072 $\frac{\text{N}}{\text{m}}$, and diameter of droplet (D_w) = 0.0026 m).

2.10. Reflectance Measurements

To calculate the reflectance of all the surfaces, a Shimadzu MPC3600 UV-VIS-NIR Spectrometer with an absolute specular reflectance mode was used. Wavelengths ranging from 200 nm to 800 nm were used for the reflectance measurement. A D₂ light source was used for the range 200–310 nm and a Tungsten source was used for the range 310–800 nm. The angle of incidence and angle of reflection were set to 5° throughout the experiment. The equipment used a photomultiplier tube (PMT) detector. The reflectance from a surface is evaluated by its refractive index profile [6]. In general, a flat surface has a high reflectance due to the discontinuous refractive index profile, whereas a structured surface suppresses reflection with its graded refractive index profile [6].

3. Results and Discussion

Figure 1 shows the SEM images of densely packed urchins after growth in 5 mM zincate solution on a seeded silicon substrate (Figure S2) at ~20 °C for 12 h, and the simple fabrication method is described in Figure S1. XRD (Figure S11) confirmed that these urchins were ZnO. The morphology of an individual urchin (Figure 1c) revealed sharply tapered needles radiating from the center. The length of the needles was $L \sim 1.65 \pm 0.11 \mu\text{m}$, its width at the central base was $D_c \sim 156 \pm 24 \text{ nm}$ and at the tip $D_t \sim 13 \pm 7 \text{ nm}$, with a tapering angle $\theta \sim 10^\circ$ and an average tip-to-tip spacing $s \sim 504 \pm 119 \text{ nm}$ (see Figures S3–S5 for the size distribution analysis).

Previously, ZnO nano- and micro-structures resembling the urchin morphology have been reported. For instance, Elias et al. [50] prepared μm -sized hollow spheres with a ZnO nano-columns coating using atom layer deposition and electrodeposition, whilst the method by Shen et al. [49] used thermos-evaporation of metallic Zn powder at high temperature (~750 °C). Wahab et al. [45] fabricated ZnO nano-flowers with blunt tapering via pH-controlled reactions in a solution of zinc acetate dihydrate and sodium hydroxide at 90 °C. Using a similar approach, Gokarna et al. [46] synthesised ZnO urchin-like structures with columnar nanoneedles. Hieu et al. [47] sputtered zinc onto a polystyrene-sphere array and subsequent oxidation at 500 °C led to columnar ZnO urchins; while the method used by Taheri et al. [48] also involved depositing zinc acetate dihydrate precursor followed by calcination at 500 °C. The solution synthesis method we report here is relatively simpler (at RT and applicable to different surfaces as we show below) compared to these previous studies, producing

spiky tapered morphology of the urchin needles with a very high density previously unreported. The presence of high-density spiky ZnO urchins endows the surface with multi-functionalities as discussed below.

SEM images of the region not covered by the urchins reveal vertical ZnO nanoneedles (from the primary growth step as referred to in Figure 1d,g) with an average base diameter $\sim 90 \pm 20$ nm, smaller than that of the urchin needles (Figure 1d–g), and the cross-section SEM view confirms that the ZnO urchins were formed atop a layer of nanoneedle arrays (Figure 1a,b). Orthorhombic ϵ -Zn(OH)₂ crystals with facet edges ~ 10 – 30 μm were also observed (Figure S7), with ZnO urchins (the secondary growth in Figure 1d) decorating the facets. The tapered or spiky nanoneedle geometry might be attributed to the concentration gradient of the zinc ions in the vicinity of the substrate where the ZnO nanocrystal seeds provided the nucleation sites for ZnO nanoneedle growth (Figure 1f,g) [61].

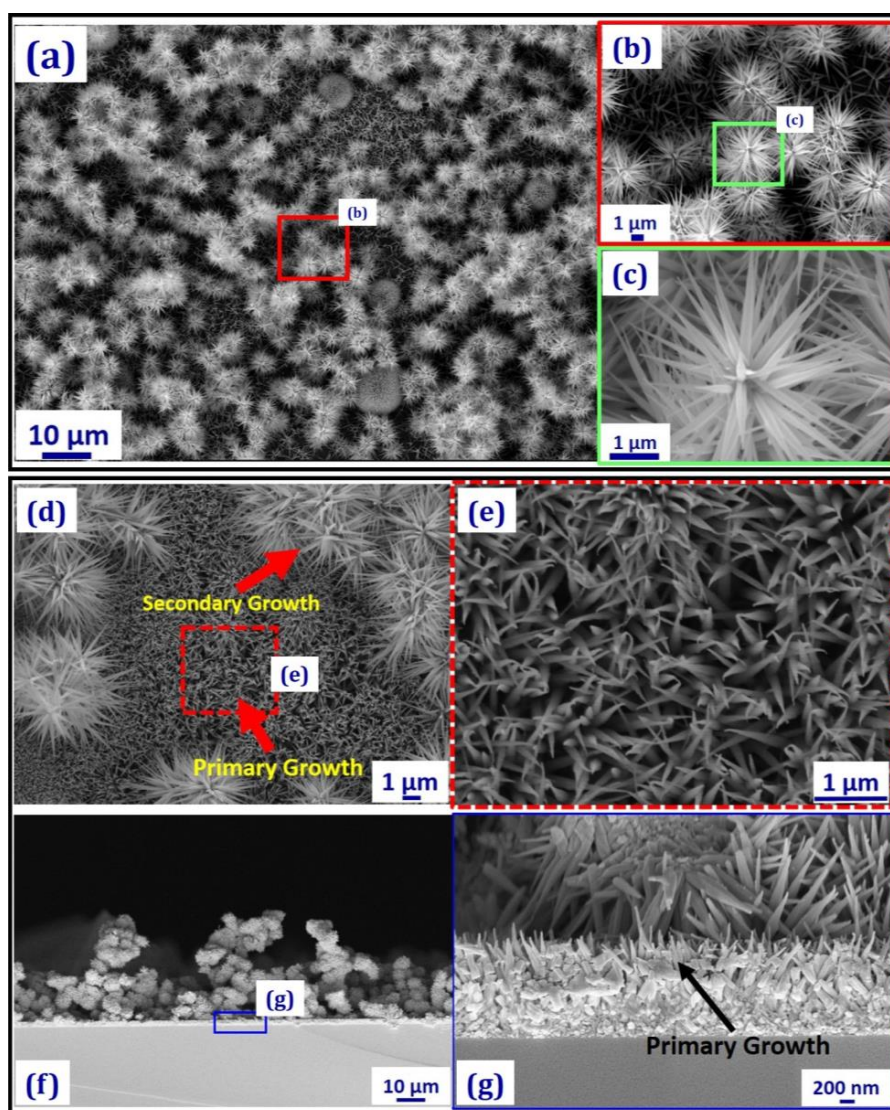


Figure 1. Scanning electron microscopy (SEM) images of ZnO urchins on a seeded silicon substrate (a), and SEM images of the region not covered by the urchins revealing vertical ZnO nanoneedles from the primary growth (d,g), with enlarged views shown in (b,c,e,f) respectively. (Seeding concentration of zinc acetate dihydrate—5 mM, synthesis time for ZnO urchin growth—12 h, and temperature—20 °C).

To understand the growth kinetics of the ZnO nanostructures, SEM images (Figure 2, Figures S8 and S9) of the seeded silicon substrates were taken at different growth time intervals, t . At $t = 0.5$ – 1

h (Figure 2a,b), only the primary growth of vertical nanoneedle structures was observed. At $t = 3$ h, the secondary growth of urchin-like structures atop the nanoneedle array with a diameter $\phi \sim 1 \mu\text{m}$ was observed. The density and size of the urchins and length of the urchin needle, L , then increased with time (see Figure S5 for urchin needle length, L vs. growth time, t), whilst the tapering angle remained largely constant at $\theta \sim 10^\circ$. Figure 2g presents the GIXRD data (see Section 2.8 in Materials and Methods) on the seeded silicon substrates at different growth time intervals. The intense (002) ZnO peak at $t < 3$ h is consistent with the primary growth of vertical ZnO nanoneedles. The $\epsilon\text{-Zn(OH)}_2$ phase (orthorhombic crystals) is visible for the samples with a reaction time $t > 3$ h (Figure 2g). This shows that the ZnO nanoneedles started forming after 30 min and Zn(OH)_2 crystals after 3 h, likely together with the secondary growth of ZnO urchins (Figure 2c–f).

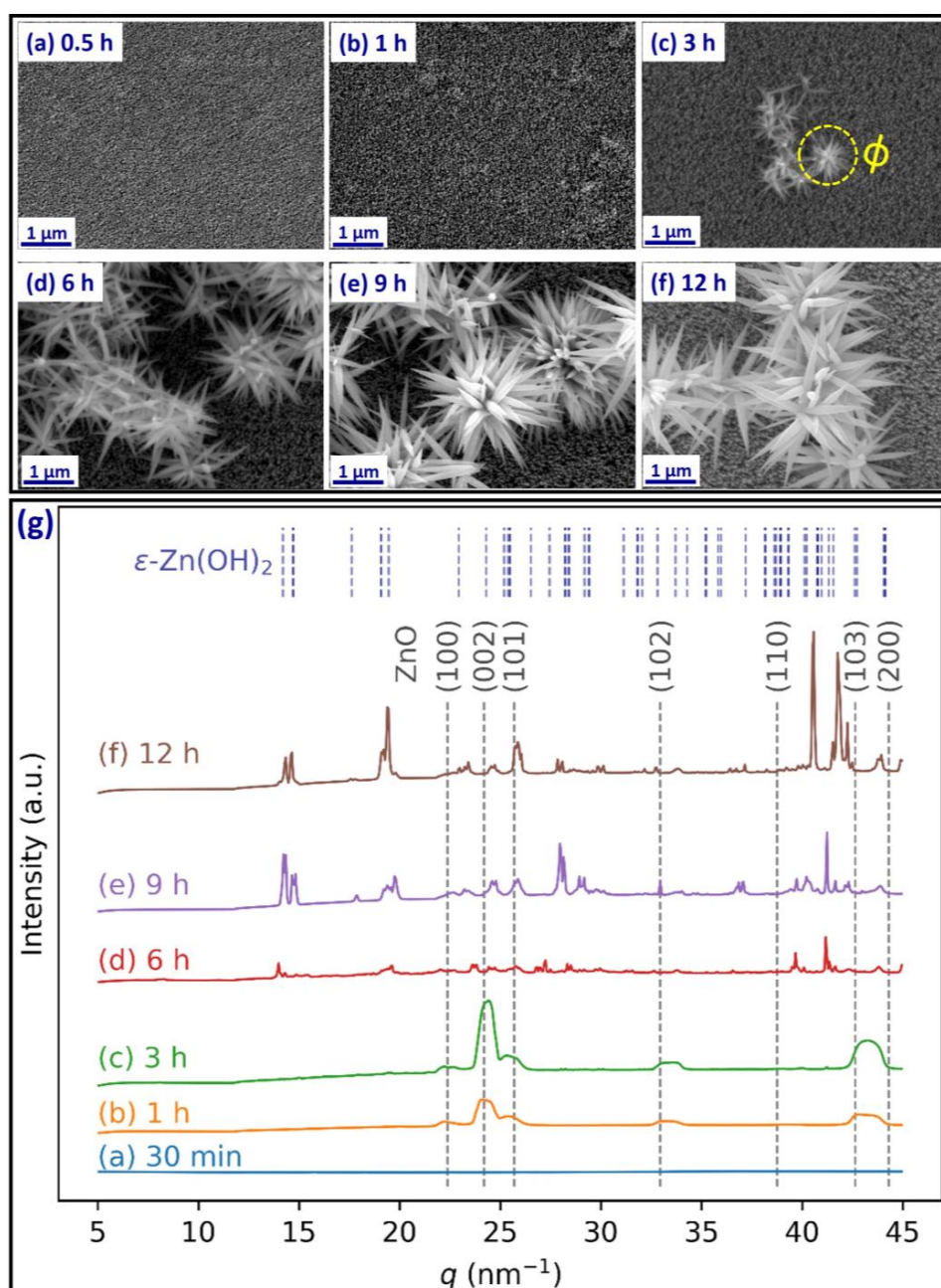


Figure 2. (a–f) SEM images of seeded silicon substrates in 5 mM zincate growth solution at different growth time intervals, $t = 0.5$ –12 h; and (g) corresponding GIXRD data for the substrates.

To evaluate the effect of substrate seeding on the ZnO urchin morphology, the one-step fabrication procedure was performed using unseeded silicon substrates (i.e., in 5 mM zincate solution at 20 °C, and 12 h growth time). The SEM image in Figure 3a shows that urchin-like ZnO nanostructures were also obtained with a high surface coverage (Figure 3a,c), with urchins found on orthorhombic ϵ -Zn(OH)₂ crystal facets (Figure 3b). There was no significant difference in the urchin morphology (urchin size, φ ; needle length, L ; diameter, D ; and tip angle, θ) compared to the seeded substrate. However, some urchins in the bare silicon area had a smaller number of nanoneedles (Figure 3d). Comparison of the XRD patterns in Figure S11 for 12 h growth shows that on the unseeded Si sample, the most intense peaks correspond to the ϵ -Zn(OH)₂ crystal structure, consistent with the SEM images shown in Figure 3.

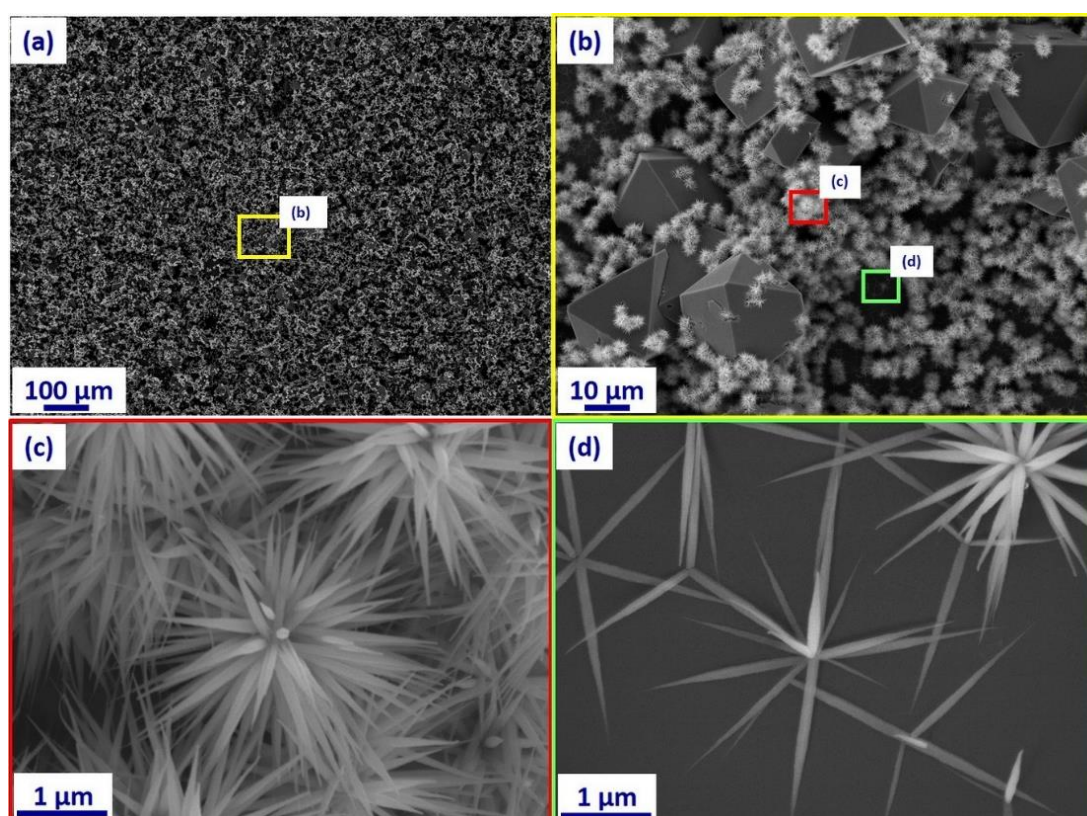


Figure 3. SEM images of ZnO nanostructures grown on the unseeded silicon substrate in the one-step synthesis (12 h, 20 °C), with a high coverage of ZnO urchins (a) and with morphology similar to that on seeded Si substrates (b,c). Orthorhombic ϵ -Zn(OH)₂ crystals were also observed (b) decorated with urchins. Urchins with a smaller number of needles were also observed in the bare silicon area (d).

Time dependent growth of ZnO urchins on non-seeded silicon substrate was also studied, and SEM images at different growth time intervals are shown in Figure S10. Similar to the seeded Si substrates, ZnO urchins emerged between 1–3 h of growth time (Figure S10a,e). Their dimensions and surface coverage increased with time (Figure S10), with orthorhombic ϵ -Zn(OH)₂ crystals starting to appear on the surface between 3–6 h of the synthesis (Figure S10b,h).

Figure S11 further compares the XRD on both seeded and non-seeded silicon substrates after growth in the zincate solution for 12 h. Peaks corresponding to both ZnO (PDF 36-1451) and ϵ -Zn(OH)₂ (PDF 38-385) were identified, confirming their formation on the surfaces. The most intense peak in the seeded silicon sample corresponds to the ZnO (002) plane, which is the preferential growth direction of ZnO nanoneedles [62]. Along with the SEM images, this confirms that on seeded Si the primary growth led to ZnO nanoneedles, with the ZnO urchins grown atop these in the secondary growth

stage. In the case of the unseeded Si sample, the most intense peaks correspond to the ϵ -Zn(OH)₂ crystal structure. This is consistent with the SEM image (Figure 3b) which shows that the substrate was covered by ZnO urchins grown on orthorhombic Zn(OH)₂ crystals and bare Si surface, with a significantly smaller number of arrayed ZnO nanoneedles (Figure 3d).

Growth of the ZnO nanoneedles and urchin structures was also trialed on copper sheets, soft PDMS and transparent glass substrates, all seeded by dipping the substrates in a solution of zinc acetate dihydrate in ethanol (Figure S1). The ZnO nanostructures grown on these surfaces (Figure 4g–i) exhibited similar urchin morphologies to the those on silicon substrates (Figure 1). The growth on glass was also evident, as it became translucent after the growth (Figure 4d and Figure S12). This demonstrated that the simple, room temperature synthesis method for ZnO nanoneedles and urchins is adaptable to various substrates.

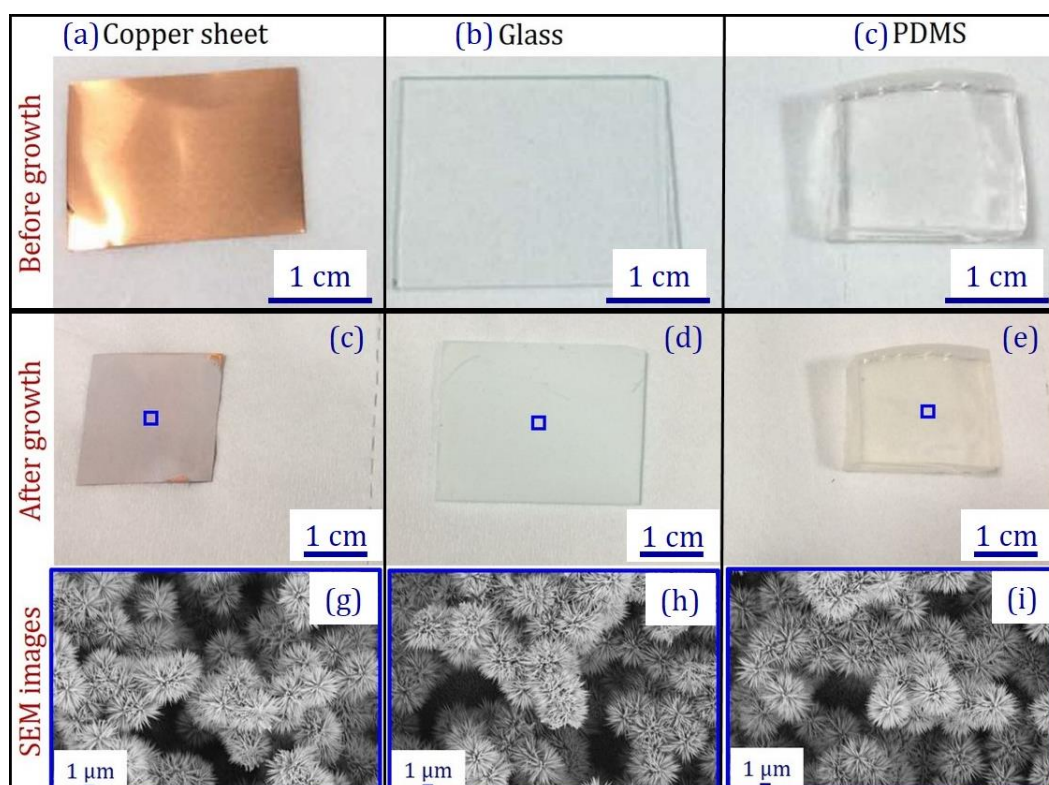


Figure 4. Photos of seeded substrates before and after growth of ZnO urchins, and corresponding SEM images of the ZnO urchins formed on the substrates: (a,c,g) copper, (b,d,h) glass slide, and (c,e,i) polydimethylsiloxane (PDMS) surfaces.

Si surfaces with ZnO urchins (both seeded and unseeded, with 12 h growth time) were superhydrophilic, with a water droplet placed on these surfaces spreading completely and showing a water contact angle $\theta \sim 0^\circ$ (Table S1). They became superhydrophobic after overnight silanization (Table S1) with an apparent water contact angle $\theta \sim 159^\circ \pm 1.4^\circ$ (Figure 5a) and a contact angle hysteresis $\Delta\theta < 7^\circ$ was observed. The water droplet exhibited a Cassie-Baxter state on the silanized ZnO urchin surface, which is evident from the high contact angle and low contact angle hysteresis. For comparison, Table S1 shows that bare Si and Si coated with a 20 nm ZnO thin film exhibited hydrophilic behavior with apparent water contact angles $\theta \sim 36^\circ$ and $\sim 61^\circ$, respectively.

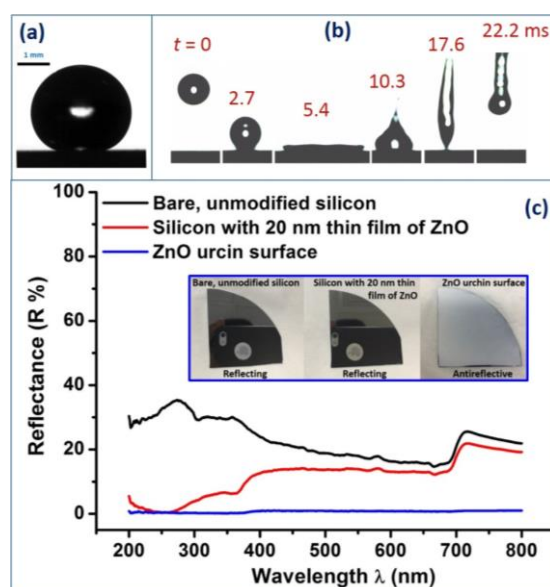


Figure 5. (a) Photograph of a 10 μL water droplet on ZnO urchin-coated surface after silanization, with a contact angle $\theta \sim 159^\circ \pm 1.4^\circ$ (the scale bar is 1 mm). Prior to silanization, the surface was highly hydrophilic, displaying complete wetting by water with $\theta \sim 0^\circ$ (Table S1). (b) High-speed camera images of a 10 μL droplet bouncing off the superhydrophobic ZnO urchin surface, with a contact time of 15.3 ms. (c) Reflectance from the ZnO urchin surface compared to the two control surfaces in the wavelength range $\lambda = 200\text{--}800$ nm. The ZnO urchin surface was found to be the most anti-reflective with reflectance $R < 1\%$ over the whole wavelength range. The inset shows the highly reflective bare Si and Si with 20 nm ZnO film (with the reflection of the iPhone used to take the photo clearly visible), in contrast to the anti-reflective characteristic of the ZnO urchin surface.

To further demonstrate the superhydrophobicity in dynamic conditions, a 10 μL water droplet was allowed to fall on the silanized ZnO urchin surface from a height of 7 cm with a *Weber number* $We = \frac{\rho V^2 D_w}{\gamma} = 49$, where $\rho = 1 \text{ g cm}^{-3}$ is the density of water, $V = \sqrt{2gh} = 1.17 \text{ m s}^{-1}$ the impact velocity of the falling droplet, $\gamma = 73.8 \text{ mN m}^{-1}$ the surface tension of water, and $D_w = 2.67 \text{ mm}$ the diameter of droplet (Table S1). The dynamic process was captured using a high-speed camera with a 10,000 frames per second (fps) capture rate. The water droplet bounced off the superhydrophic surface completely without leaving any visible residues, with a droplet-substrate contact time of 15.3 ms (Figure 5b and Video S1) which is close to the theoretical contact time $(\frac{\rho D_w^3}{\gamma})^{1/2} = 16.25 \text{ ms}$ for a water droplet on a superhydrophobic surface [63].

In addition, the ZnO urchin coated surface also demonstrated anti-reflective behavior. Figure 5c showing $<1\%$ reflectance (R) over the wavelength range $\lambda = 200\text{--}800$ nm on the urchin surface, compared to two control samples (bare Si and Si with 20 nm ZnO thin film). There have been previous reports on fabrication of nanostructured anti-reflective surfaces [6,11,64–70]. Different techniques such as dry etching, wet etching, metal assisted etching, nanoimprint lithography, etching using O_2 plasma and inductively coupled plasma, optical lithography followed by etching etc. [71] have been used to fabricate such surfaces on silicon, polymers, and glass to achieve low reflectance. In comparison, the fabrication method we used to achieve low reflectance ($R < 1\%$) was simple, with the synthesis undertaken at room temperature without the need for sophisticated instrumentation.

To evaluate the bacterial interaction with the fabricated ZnO urchins on the seeded Si surface, it was submerged in 2 mL *E. coli* culture in phosphate-buffered saline (PBS) for 24 h and then imaged with SEM. This was compared with two control samples: unmodified bare Si and Si with a 20 nm sputtered ZnO film (Figure S13). SEM images in Figure 6a (see also Table S2) showed bacteria growth on the control surfaces; in contrast, no bacterium was observed on the ZnO nanostructured

surface (Figure 6c,d). In addition, the SEM analysis revealed a change in the morphology of the ZnO urchins (inset in Figure 6c): after 24 h immersion in the *E. coli* culture, the spiky urchin structure was transformed into a hexagonal pyramid structure (see Figure S14a). This can be attributed to the formation of sodium zinc phosphate hydrate ($\text{NaZn-PO}_4 \cdot \text{H}_2\text{O}$) [72] due to the reaction between ZnO urchins and the PBS (cf. XRD data in Figure S14b). ZnO is known to exhibit antimicrobial efficacy [15–19], killing bacteria and prohibiting bacterial growth on its surface via release of Zn^{2+} ions which solicit generation of reactive oxygen species (ROS). A number of naturally occurring surfaces bearing spiky nanotextures (e.g., cicada wings, dragonfly wings, and gecko skin) have been reported to exhibit bactericidal efficacy, attributed to puncturing or stretching of the bacterial membrane, although the detailed mechanisms remain to be fully understood [3,11,73–76]. Hence, we suggest that the ZnO urchin surfaces could prevent bacterial growth by combining synergistically the inherent chemical activity of ZnO and the spiky morphology of the urchins that inhibit the bacteria from colonizing on the surface of the ZnO urchins [18,77–79]. The sharp topography of the nanoneedles is particularly effective in disrupting the bacterial cell wall by imparting localized stress on the bacterial membrane [80]. There are previous reports on reduced bacterial adhesion on superhydrophobic surfaces [81–84] where the antiwetting property of the surface plays a crucial role in the interaction of the bacteria with the surface; whereas our fabricated ZnO urchin surfaces exhibited bacteriophobic behavior in a hydrophilic state (with a water contact angle $\theta \sim 0^\circ$).

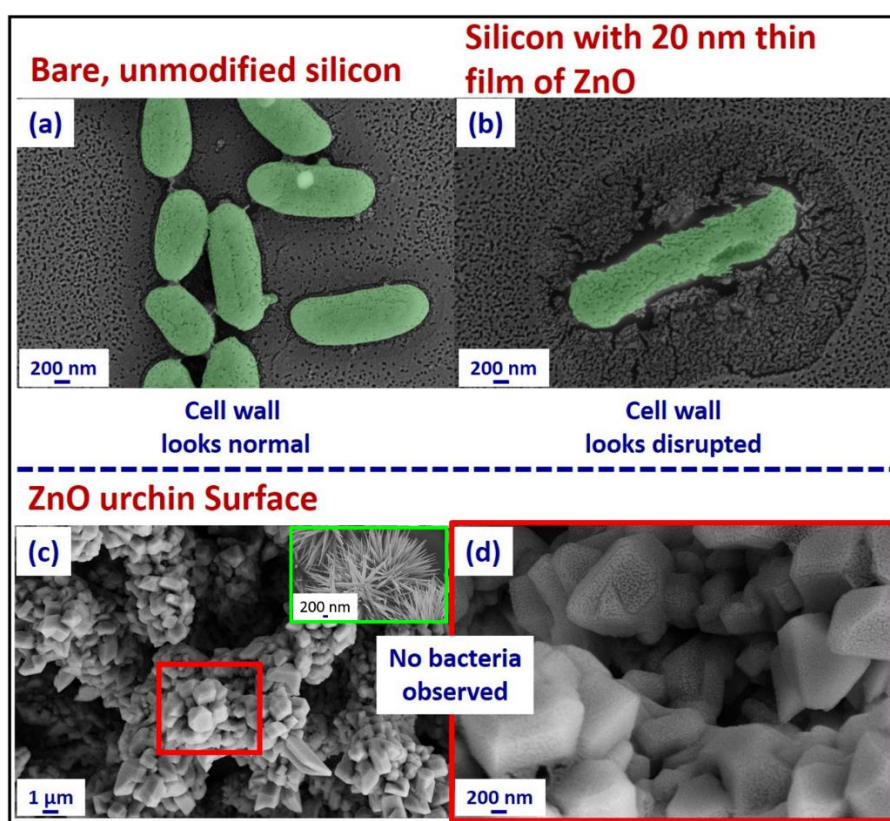


Figure 6. Representative SEM images of *E. coli* (false-colored green) on different surfaces after 24 h of bacterial culture: (a) bare, unmodified silicon, (b) silicon with a 20 nm thin film of ZnO, and (c,d) ZnO urchin surface. Inset in (c) shows the morphology of the ZnO urchin before bacterial culture on the urchin surface.

4. Conclusions

In summary, ZnO urchin structures were fabricated via a simple method involving submergence of a substrate (Si, PDMS, glass, or Cu) in an alkaline aqueous zincate ion ($\text{Zn}(\text{OH})_4^{2-}$) solution at

room temperature ($\sim 20^\circ\text{C}$). On the Si substrate seeded with ZnO nanocrystals, we observed ZnO urchin structures atop vertically aligned ZnO nanoneedles. The urchins consisted of densely packed, μm -long spiky ZnO nanoneedles radiating from the urchin center, with a tapering angle of $\sim 10^\circ$. The growth kinetics of the ZnO surface nanostructures were studied by time resolved SEM, revealing that the urchin morphology emerged at ~ 3 h reaction time with a small number of tapered nanoneedles. The nanoneedle density and length then progressively increased with the reaction time. On the seeded Si substrate, a primary growth step of the vertical ZnO nanoneedles was identified preceding the secondary urchin growth. The ZnO urchin coated surface exhibited anti-reflective properties, reducing the reflectance to less than 1%. It was highly hydrophilic, with a water contact angle of $\sim 0^\circ$; after silanization, it exhibited superhydrophobicity with a water contact angle of 159° and hysteresis smaller than 7° . In addition, its dynamic hydrophobicity was demonstrated by the bouncing-off of a water drop as captured by a high-speed camera, with a resident time of 15.9 ms. Furthermore, the ZnO urchin surface showed bacteriophobic behavior, as compared to the control Si and ZnO-coated surfaces, with no bacterium colonization observed on the surface after 24 h incubation in *E. coli*. This simple method for preparing ZnO urchins with the unique morphology of tapered nanoneedles, and its ready adaptability to different surfaces (including polymers), may open new routes for the fabrication of multifunctional ZnO nanostructured surfaces.

Supplementary Materials: The following are available online at <http://www.mdpi.com/2504-5377/2/4/74/s1>: Figure S1, process flow of the fabrication of ZnO nanostructures on seeded silicon substrates at near room temperature; Figure S2, SEM images of a silicon substrate seeded with ZnO nanoislands in 5 mM zinc acetate dihydrate solution (dipping $\times 5$ times and annealed at 300°C for 30 min); Figure S3, nanoneedle dimension measurement using ImageJ software. The values in the main text were averaged from 50 different needles, with the measurements of different parameters at each single needle repeated 5 times; Figure S4, size distribution from ImageJ analysis of the nanoneedle tip diameter at the top D_t urchins ranging from 7–30 nm with an average diameter of $D_t = 13 \pm 7$ nm; Figure S5, length of the nanoneedles, L , in the urchin structure vs. growth time. The length of the nanoneedle was found to increase with respect to synthesis time. Stars in the figure show that for time points 0, 0.5 and 1 h there was no formation of urchin structures; Figure S6, (a) the angled view after 9 h growth, showing ZnO urchins on the top of highly, (b) oriented ZnO nanoneedles; Figure S7, representative FESEM images of the seeded silicon substrate post 12 h synthesis in the zincate solution; Figure S8, SEM images of seeded silicon substrates after growth in zincate solution at different time intervals. The density of ZnO nanourchins was found to increase with the increase synthesis time; Figure S9, SEM images of seeded silicon substrates after growth in zincate solution at different time intervals. Zn(OH)_2 crystals were observed on the substrates for synthesis time $t > 3$ h; Figure S10, SEM images of ZnO urchins on a non-seeded silicon substrate, taken at different time intervals, (a) 3 h, (b) 6 h, (c) 9 h, and (d) 12 h. (e–h) show enlarged views of the square regions as labelled in (a–d), respectively; Figure S11, XRD of (a,b) seeded and (c,d) unseeded silicon substrates after growth in the zincate solution at 20°C for 12 h, with the ZnO and $\epsilon\text{-Zn(OH)}_2$ peaks indicated by * and \blacktriangledown , respectively, and the enlarged views shown on the right-hand side; Figure S12, optical image showing the loss in transparency after the formation of ZnO urchin/nanoneedles on the glass substrate; Figure S13, 12-well culture plate for pouring the bacterial suspension on different samples; Figure S14, (a) SEM and (b) XRD of the ZnO nanourchin surface 24 h after pouring the bacterial culture. Formation of sodium zinc phosphate hydrate ($\text{NaZn-PO}_4 \cdot \text{H}_2\text{O}$) was observed on the surface due to the reaction of ZnO with PBS leading to the change in morphology of the spiky ZnO nanowires; Table S1, apparent contact angles of a DI water droplet on different substrates; Table S2, SEM of *E. coli* on different substrates; Video S1, superhydrophobic ZnO urchin.

Author Contributions: A.T. and W.H.B. conceived and designed experiments; A.T. and P.W. synthesized and characterized the ZnO urchins; A.T., S.S., D.N. and P.S. performed experiments on anti-reflectivity, *E. coli* colonization, and dynamic superhydrophobicity of the urchin surfaces; P.W., O.B. and W.H.B. performed synchrotron XRD measurements. All authors contributed to data analysis and manuscript writing.

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