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Physical and mechanical studies on binary blended Portland cements containing mordenite-rich tuff and limestone filler

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Abstract. In this research, we made physical and mechanical analyses thanks to which we deduced that the potential of replacing the Portland cement up to 20% by mordenite-rich tuff (TM) and limestone filler (LF). The experimental procedure was done in two main stages. The first stage was dedicated to the pastes, including water requirement, setting time, hydration heat and free $\text{Ca}(\text{OH})_2$ content. In the second stage, mechanical and physical properties of mortars, water porosity and pore size distribution were studied. The final results show that LF mixtures closest to the control mixtures unlike the MT mixtures which exhibited a high water demand, less free $\text{Ca}(\text{OH})_2$, rapid appearance of the maximum thermal effect, an increase in the total and water accessible porosity, there is also a delay in improving of strength.

Keywords: mordenite-rich tuff / cement-based materials / hydration / porosity / mechanical strengths

Résumé. Étude physico-mécanique des mortiers de ciment binaires contenant un tuf riche en mordenite et un filler calcaire. En s'appuyant sur des essais physiques et mécaniques, l'objectif de cette étude est de mettre en évidence l'intérêt du remplacement jusqu'à 20 % du ciment Portland par un tuf riche en mordenite et un filler calcaire. La campagne expérimentale a été réalisée sur deux parties, la première dédiée aux pâtes, évalue le besoin en eau, le temps de prise, la chaleur d'hydratation et la teneur en $\text{Ca}(\text{OH})_2$ libre. Dans la seconde partie, les propriétés mécaniques et physiques des mortiers, la porosité à l'eau et la distribution de la taille des pores ont été étudiées. Les principaux résultats ont montré que, à l'opposé des mélanges de tuf riche en mordenite, ceux de filler calcaire semblent être les plus proches au mélange témoin. Les mélanges de tuf se caractérisent par : une forte demande en eau, une baisse de la quantité de $\text{Ca}(\text{OH})_2$ libre, une apparition rapide du pic thermique maximal, une amélioration de la porosité accessible à l'eau et de la porosité totale et enfin, un retard dans l'amélioration de la résistance à la flexion et à la compression.

Mots clés: mordenite / matériaux cimentaires / hydratation / porosité / résistances mécaniques

1 Introduction

The cement industry has been always among the largest sources of carbon dioxide (CO_2) emission making up for 5–7% of the CO_2 emissions worldwide [1]. Traditional cements are made from clinker and researches have shown that the fact of reducing the use of clinker by 1 kg when producing one tone of cement, can reduce the CO_2 released in air by 1 kg [2].

Taking into consideration the environmental protection and the economical aspect, a large variety of supplementary cementitious materials (SCMs) are used

to substitute either Portland cement clinker in the production of blended cements or cement in the concrete production.

The SCMs are highly attractive options to reduce the problem of CO_2 emission which helps to decrease the negative impact on the environment and promote energy savings [1,3]. It is further reported that the demand for cement and concrete will increase from nearly 2 billion tons a year, nowadays to 5.8 billion tons by 2050 [4], much of which should be met through a combination of SCMs and clinker in order to meet environmental restrictions [5].

Since the production of by-products is not expected to increase, there is a great need to explore alternative SCMs to meet the increasing demand. Natural mineral deposits, virtually endless, are some of promising sources of alternative SCMs such as limestone filler and natural pozzolans [5,6].

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The increasing interest in the use of limestone filler in cement-based materials was motivated and justified by the several advantages linked to their use, including: lower costs, a better homogeneity, the worldwide abundance and the ease in controlling quality [7].

The use of pozzolanic materials is gaining a considerable attention because of their ability to react with calcium hydroxide to form compounds which possess cementitious properties. Among the pozzolans which can be used in cement-based materials, the natural zeolites that have emerged as an important binding material in recent years [8].

The pozzolanic activity of the natural zeolites and their incorporation in the blended cements have been studied and several studies revealed that zeolitic minerals enhances the formation of hydration products similar to those produced during cement hydration because of the presence of $\text{Ca}(\text{OH})_2$ [9–11].

The possibility of cement substitution by the natural zeolite, mordenite, has been investigated [12–15]. In spite of numerous efforts to investigate the strength properties and durability of concretes and mortars containing natural zeolite [9,11,12,15–20], in our knowledge, in the case of natural mordenite, physical and microstructural behavior like hydration heat, stiffness, water porosity and pore size distribution are not highlighted yet. For such a reason, studies and investigations remain limited in this field. Therefore, this study aims to highlight the benefits of using mordenite-rich tuff as a partial replacement of Portland cement from the physico-mechanical and microstructural point of view. Willing to contribute to the state of an exhaustive knowledge about the basic mechanisms related to mordenite effects, consists on the novelty of our study. We, especially, aim to use the mordenite-rich tuff in construction technology instead of using Portland cement in cement-based materials which was monitored at 7, 28, 90 and 180 days to evaluate its influence on the physico-mechanical properties and the hydration of the blended mixes.

2 Materials, mixtures and curing conditions

2.1 Cement, sand and limestone filler

The main binder used in this study is a cement CEM I52.5R, in accordance with European standard NF EN 197-1 (2012) from Saint-Pierre La Cour plant (France) [21]. The cement has a density of 3140 kg/m^3 and a Blaine fineness of $440 \text{ m}^2/\text{kg}$. The chemical composition of the cement (PC) is given in Table 1. The Bogue's formula gave the following mass percentages of the main clinker phases: 61% of C_3S , 15% of C_2S , 8% of C_3A and 10% of C_4AF .

Standardized siliceous sand of Leucate (France), with a particle size ranging from 0 to 2 mm and an absolute density of 2600 kg/m^3 was used to manufacture the mortars NF EN 196-1 (2016) [22].

The Limestone Filler (LF), containing 97.30% of CaCO_3 , used is a BETOCARB P2, produced at Erbray's plant (France). This limestone filler is characterized by a Blaine fineness of $397 \text{ m}^2/\text{kg}$, an absolute density of 2710 kg/m^3 and a water content of 0.1%.

Table 1. Chemical composition of cement (mass %).

Tableau 1. Composition chimique du ciment (% massique).

Element	Massic content (%)
SiO_2	20.07
Al_2O_3	4.86
Fe_2O_3	3.07
CaO	64.25
MgO	0.95
SO_3	3.55
K_2O	1.00
Na_2O	0.20
$\text{Na}_2\text{O}_{\text{eq}}$	0.85

Table 2. Chemical properties of Mordenite-rich tuff (MT) (mass %).

Tableau 2. Propriétés chimique du tuf riche en mordenite (% massique).

Element	Massic content (%)
SiO_2	64.81
Al_2O_3	15.58
Fe_2O_3	2.68
CaO	2.85
MgO	0.51
SO_3	0.18
K_2O	3.00
Na_2O	3.72

2.2 Mordenite-rich tuff

Mordenite-rich Tuff (MT) used in this research was extracted from natural deposits collected in the Tinebdar area, located in the province of Bejaia, Algeria. First, the bulk material was crushed to obtain a material with a particle size smaller than 2 mm. Then, the material was ground for 6 minutes using a vibratory disc mill to obtain a specific surface area similar to that of cement.

The density of mordenite-rich tuff was 2290 kg/m^3 and its specific area was $460 \text{ m}^2/\text{kg}$. The chemical composition and physical properties of the finely-ground zeolitic tuff (MT) are shown in Table 2. Powder X-ray diffraction pattern of MT showed the presence of the crystalline phases in a greater quantity of mordenite (58%) and a smaller quantity of heulandites (12%), albite (15%), sanidine (6%) and muscovite (9%) (Fig. 1). The morphology of the MT powder is examined under a scanning electron microscope as shown in Figure 2. A SEM-EDX analysis showed that the mordenite-rich tuff was mainly composed of oxygen, silicon, and aluminum. Whereas, potassium, iron and magnesium were found in minor quantities (Fig. 2).

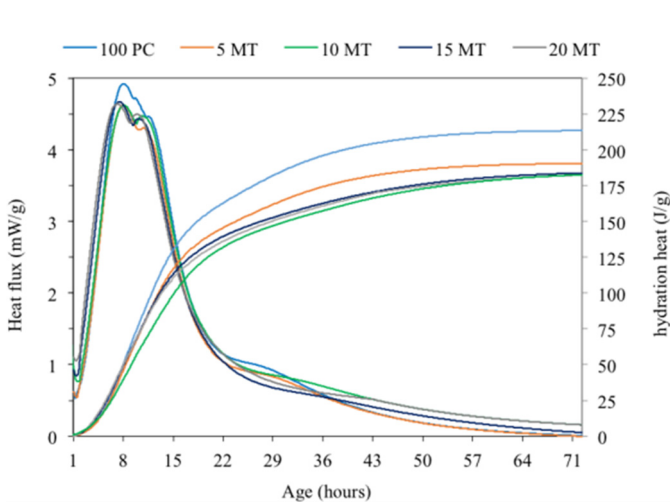
The particle size distributions of the cement, mordenite-rich tuff and limestone filler were measured by using the laser granulometry are shown in Figure 3. The fractions of the different size classes are summarized in Table 3.

Table 3. Particle size classes of Portland cement (PC), limestone filler (LF) and tuff (MT).**Tableau 3.** Classification des tailles des particules du ciment Portland (PC), filler calcaire (LF) et tuf (MT).

	PC (μm)	LF (μm)	MT (μm)
90% of particles with \emptyset inferior or equal to	28.92	41.53	123.6
50% of particles with \emptyset inferior or equal to	10.09	11.1	19.69
10% of particles with \emptyset inferior or equal to	1.36	1.12	1.45

Table 4. Composition parameters, density and flow time with a LCPC-maniabilimeter of the different mortars investigated (the indicated ratios are mass ratios, PC: cement; MT: mordenite-rich tuff; LF: limestone filler; B: binder).**Tableau 4.** Paramètres de composition, densité et temps d'écoulement mesuré par le maniabilimètre LCPC des différents mortiers (rapport massique; PC: ciment; MT: tuf riche en mordenite; LF: filler calcaire; B: liant).

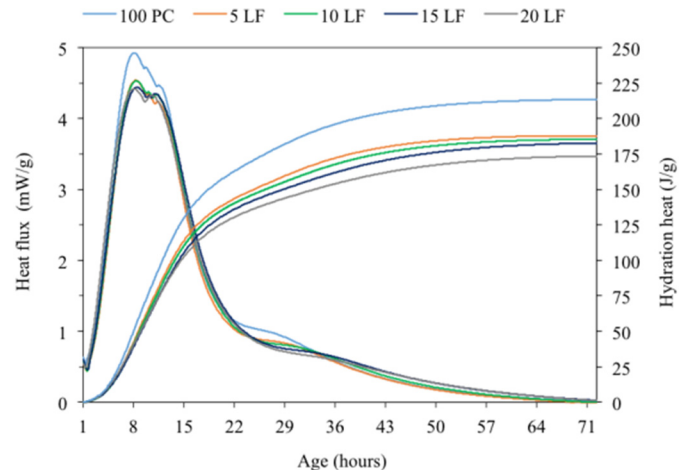
	100 PC	5 MT	10 MT	15 MT	20 MT	5 LF	10 LF	15 LF	20 LF
C/B	1.00	0.95	0.9	0.85	0.8	0.95	0.9	0.85	0.8
MT/B	–	0.05	0.1	0.15	0.2	–	–	–	–
LF/B	–	–	–	–	–	0.05	0.1	0.15	0.2
W/B	0.5	0.5	0.5	0.5	0.5	0.5	0.5	0.5	0.5
Flow time (s)	4	13	23	60	148	2	2	2	2
Bulk density	2.264	2.241	2.235	2.220	2.211	2.242	2.249	2.253	2.245
Dry bulk density	2.089	2.068	2.061	2.044	2.030	2.079	2.076	2.078	2.071

**Fig. 4.** Heat flux and hydration heat of PC paste and MT pastes.**Fig. 4.** Flux de chaleur et chaleur d'hydratation des pâtes de CP et de tuf.

(1995) [27]. Setting time analysis of the blended cements and the reference Portland cement were performed by using an automatic Vicat machine.

3.2.2 Hydration heat

The hydration heat evolution rate and cumulative hydration heat of blended cement pastes containing MT or LF were measured with a multi-channel isothermal microcalorimeter of TAM Air type. The tests were performed at a constant temperature nearing 20 °C. TAM Air has eight parallel twin-chamber measuring channels: one chamber contains the

**Fig. 5.** Heat flux and hydration heat of PC paste and LF pastes.**Fig. 5.** Flux de chaleur et chaleur d'hydratation des pâtes de CP et de filler calcaire.

sample, another contains the reference. After mixing homogeneously, the pastes were immediately placed in bottles, weighed and loaded into the microcalorimeter. The hydration heat evolution rate and cumulative hydration heat of blended cements was continuously monitored as a function of time (Figs. 4–6).

3.2.3 Free $\text{Ca}(\text{OH})_2$ content of hardened pastes

The free $\text{Ca}(\text{OH})_2$ content of hardened cement pastes was determined at 7, 28, and 90 days by thermo-gravimetric analysis (TGA) using a Mettler Toledo TGA/DSC 1

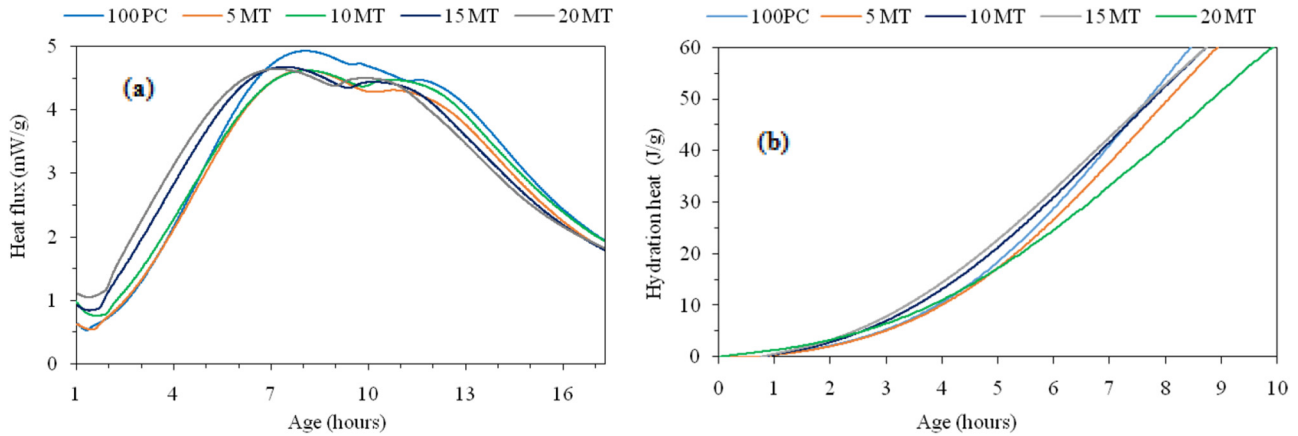


Fig. 6. Detail of heat flux and hydration heat of MT pastes.

Fig. 6. Détail du flux de chaleur et de chaleur d'hydratation des pâtes de tuf.

instrument. At specified test ages, the pastes were crushed and sieved to obtain particles smaller than $80\ \mu\text{m}$, then, they were immediately analyzed between room temperature and 1050°C at a heating rate of $10^\circ\text{C}/\text{min}$ in an atmosphere of carbon dioxide free nitrogen.

3.3 Mortars

3.3.1 Mass variations and total shrinkage

The total shrinkage and mass variation of mortar specimens were measured using an extensometer with $1\ \mu\text{m}$ of precision and a balance with $0.01\ \text{g}$ of precision, respectively. The measurements were conducted from 1 to up to 180 days after the casting of the mortars. The average of three parallel tests was taken as the representative value.

3.3.2 Dynamic Young's modulus and mechanical strengths

Dynamic Young's modulus of the mortar specimens was measured according to the European standard NFP 18-414 (2017) [26]. The elastic modulus was calculated by a non-destructive resonance frequency measurement. This frequency was determined via impulse excitation tests with a Grindosonic[®] device. The mortar prisms were lightly and elastically struck with an impuler and the flexural resonant frequency was recorded with a vibration sensor. The resonance frequency was manually recorded via data acquisition incorporated with the excitation system. For each mortar, the determination of Young's modulus was performed on three specimen-replicates manufactured from the same mixture.

The flexural and compressive strength were determined by means of a 250-KN and 25-KN double automatic compression flexion press-class 1 device in accordance with the European standard NF EN 196-1 (2016) procedure [22]. The reported data represent the average values obtained from three measurements for flexural strength and six for compressive strength.

3.3.3 Water porosity

The open porosity of mortars was measured using the water vacuum saturation method recommended by AFPC-AFREM 11-12 (1997) [25]. The samples were placed for four hours in a desiccator which was connected to a vacuum pump in order to remove the air from it. Then, these samples were kept in water for a period of 72 h. After the saturation sequence, the previous samples were weighed thanks to the hydrostatic weighing. Next, the samples were dried for about two weeks in an oven at 105°C until the obtention of a constant mass. The porosity P is then calculated using the equation (1).

$$P = \frac{M_{sat} - M_{dry}}{(M_{sat} - M_{water})}, \quad (1)$$

where M_{sat} is the mass of saturated mortar sample measured in air ambience (g); M_{dry} is the mass of dry sample (g); M_{water} is the mass of saturated mortar sample measured in water (g).

For each mortar composition, two specimens were tested in order to calculate an average value of open porosity accessible to water.

3.3.4 Pore size distribution

The porous structure was performed by mercury intrusion porosimetry (MIP) at 28 and 90 days using an AutoPore IV Micrometrics device. The range of applied pressure corresponded to a pore radius of $10\ \text{nm}$ to $100\ \mu\text{m}$. Since oven-drying at $100\text{--}105^\circ\text{C}$ may cause damage to microstructure and pore system of the hardened mortars, the samples of approximately $3\ \text{cm}^3$ of volume were dried in a hot chamber at 60°C until obtaining the constant weight to removal moisture. After being dried, the samples were tested by MIP in order to determine the total porosity and the pore size distribution (PSD).

Table 5. Normal consistency water content and setting time of the cement pastes.**Tableau 5.** Consistance normalisée, contenu en eau et temps de prise des pâtes de ciment.

	100 PC	5 MT	10 MT	15 MT	20 MT	5 LF	10 LF	15 LF	20 LF
Water demand (w/w %)	30.5	32	32.5	34	35	30	30	30	30
Initial setting time (min)	59	63	65	65	72	64	66	72	79
Final setting time (min)	414	359	339	305	279	355	339	334	334

4 Results and discussion

4.1 Normal consistency and setting time

The amount of water require for normal consistency of cement pastes and the setting time of the tested samples are shown in Table 5. The results indicate that the water demand of 5, 10, 15 and 20MT blended-cements was approximately 5, 7, 11 and 15% higher than the one of 100PC, respectively. Increased water requirement of blended cements when compared to 100PC is attributed to a higher capacity of MT particles to absorb water.

Initial setting times of the blended cements containing mordenite-rich tuff/limestone filler are slightly greater than that of 100PC. This result is in accordance with a previous report [28]. However, an opposite trend is observed in final setting times. Thus, the final setting times of MT pastes are shorter than that of 100CP by approximately 55, 75, 109 and 135 min for 5, 10, 15 and 20, respectively. These results are in a good agreement with those of Shon and Kim (2013). They noticed that the presence of zeolite material in the cement system causes a setting acceleration [29]. The latter is probably attributed to a decrease in consistency of the blended cement pastes which is caused by the incessant water absorption by MT particles. So, the setting time decreased when the water demand increased. This acceleration in final setting time can also be attributed to the controlled diffusion which occurs during the pozzolanic reaction of MT.

4.2 Hydration heat

The heat flux and hydration heat, shown by kinetic curves in Figures 4 and 5 for cement pastes with varying contents of MT and LF, indicate that the kinetic curves are very close to those of PC paste. The release of heat is primarily associated with the quick but short-lived hydration of C_3S which has a very high intrinsic solubility. When the hydration reactions accelerate and become controlled by nucleation, the acceleration period begins and the heat flow reaches its maximum. Thus, small nuclei of hydration products precipitate out of the pore solution and grow in size. Consequently, the increase in the rate of hydration during the maximum heat flux occurs because the number of nuclei is increasing.

As shown in Figures 4 and 5, the magnitude of heat flux at its maximum and its position on the temporal axis is strongly correlated with the amount of MT and LF added to the reaction system. When the MT and LF content in cement pastes increases, the magnitude of the main heat flux peak decreases. This is due to less cement mass

(lower C_3S content) after being substituted by the mineral additions.

A detail of the early time hydration of MT pastes compared to that of PC paste is presented in Figure 6. During the initial period up to 7 h, the heat flux increased as the amount of MT increased in the blended binder (Fig. 6a). The baseline of heat evolution during the induction period increased as the MT content increased. Moreover as illustrated in Figure 6b, the development of the hydration heat is faster in the initial hydration phase. Apparently, the onset of C_3S hydration was accelerated by the presence of MT in the mixture. The earlier occurrence of the peak heat rate is due to the dilution of the system by MT addition. The acceleration stage is related to the adsorption of Ca^{2+} on the pozzolan (MT), which would allow the C_3S to dissolve more rapidly [30,31].

The heat flux at its maximum decreased with increasing amount of LF in the binder and similarly the required time to reach the maximum increased (Fig. 5). An opposite trend is observed with increasing the MT content (Figs. 4 and 6). During the first 7 h, 10 and 15 MT increased the hydration heat and accelerated the maximum thermal effect compared to 5, 20 MT and the control cement paste (Fig. 6b). The increase in the MT content causes an increase of the final hydration heat due to the pozzolanic reaction between MT (glassy phases of SiO_2) and calcium hydroxide released during Portland cement hydration. Nonetheless after 7 h, the strongly exothermic reactions and the ascendant slope for the high curves illustrate that the ordinary cement released somewhat higher hydration heat relatively to the blended cements. These decreased the heat released as the MT content increased. Whereas, the basic shapes of the recorded curves are the same as the curves for the sample without addition. This indicates that using MT as an addition material to system reactions does not change the basic processes as well as the control mechanism of the hydration.

The heat flux peak corresponding approximately to 12 h associated to the transformation of the trisulfoalumite to the monosulfoalumite phase, was completely missing for the ordinary Portland cement (Fig. 6a). The recorded peak has an insignificant form in 5 MT and LF pastes but is well pronounced in 10, 15 and 20 MT pastes. The time of occurrence of this peak decreased as the amount of MT increased. This emphasized even more the accelerating effect of MT on the hydration process.

At 72 h, the heat flows of 10, 15 and 20 MT pastes were higher than that of 5 MT and 100 PC; this indicates that the MT was fully involved in the hydration reaction. The limit rate found in Vejmelková et al. (2015) is about 10% of zeolite [32].

Table 6. Ca(OH)₂ content (%) of the hardened cement pastes.**Tableau 6.** Contenu en Ca(OH)₂ (%) des pâtes de ciment durcies.

Cement pastes	100 PC	5 MT	10 MT	15 MT	20 MT	5 LF	10 LF	15 LF	20 LF
7-day	19.25	19.18	18.16	18.43	17.81	19.11	18.64	17.39	16.94
28-day	20.97	20.30	20.28	20.01	19.08	20.70	19.70	19.61	18.68
90-day	22.27	19.17	18.68	17.58	17.71	22.40	21.66	20.56	19.82

4.3 Free Ca(OH)₂ content

The Ca(OH)₂ content in hardened cement pastes was determined by TGA at 7, 28 and 90 days, and the results are shown in Table 6. We have to note that the content of Portlandite is calculated by Eq. (2) [33]:

$$\text{Ca(OH)}_2(\%) = 4,11(L_{dx}) + 1,68(L_{dc} - L_{dca}), \quad (2)$$

where

L_{dx} (%): loss of mass due to dehydroxylation of portlandite.

L_{dc} (%): loss of mass due to decarbonation of calcite.

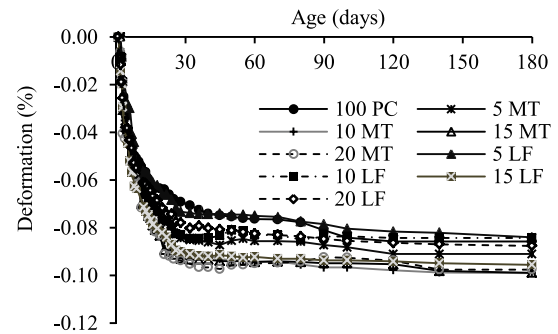
L_{dca} (%): loss of mass due to anhydrous materials carbonation.

Factors 4.11 and 1.68 correct for calcium hydroxide formed during hydroxylation and decarbonation, respectively.

According to Table 6, it can be seen that, free Ca(OH)₂ content of blended cement pastes (MT and LF) is lower than that of 100 PC paste at all test ages. This result is the consequence of lower Ca(OH)₂ production due to the dilution effect, i.e: the decrease of the clinker content in the LF cements as well as its consumption by the pozzolanic activity of the MT [34,35]. Therefore, it can be concluded that after comparing to the 100 PC paste sample, all the MT samples reduced the free Ca(OH)₂ content at the three curing periods of times as well as the consumption increased as the level of substitution increased. On the basis of the hydrated cements, the free Ca(OH)₂ levels in the cements which contains 5–20% of MT is 0.4–7.5, 3–9, and 14–21% are lower than in 100CP pastes at 7, 28 and 90 days, respectively. The free Ca(OH)₂ content in the cement paste is very important since portlandite is a reactive compound that may cause durability problems.

The small values of Ca(OH)₂ content in the MT blended cement pastes during the first days indicate that pozzolanic action of MT particles occurred slowly during the first stage of hydration. In the other hand, the pozzolanic activity of the mordenite crystals in cementitious systems is associated with the transformation of mordenite into amorphous material able to react in a high-pH medium resulting from presence of Ca(OH)₂. The slow pozzolanic reaction in the pastes which contain different amounts of the MT may be due to the lower pH of the pore solution. This medium is inadequate for the decomposition of the mordenite crystal structure [16,36].

As shown in the results presented in section 4.2 (Hydration heat), at early ages, the onset of alite (C₃S) hydration is accelerated by the presence of the MT in the mixture. So, the Ca(OH)₂ produced in the MT mixtures

**Fig. 7.** Total shrinkage versus age.**Fig. 7.** Le retrait total fonction du temps.

increases as the content of MT increases; this is due to the dilution effect. But at later ages, the Ca(OH)₂ content decreases because of the pozzolanic reaction.

4.5 Mass variations and total shrinkage

The Figure 7 presents the total shrinkage test results of the MT and the LF mixtures together with those obtained in the ordinary cement mortar mixture plotted versus age.

The total shrinkage evolves faster during the first four weeks of drying for all the mixes because of their small size favoring the desiccation. At early age, the shrinkage is almost independent of the mortar's composition; the effect of the mineral additions appears only after the first week. On a long term, the LF does not have a significant effect on the total shrinkage where its values meet with those of ordinary cement. The total shrinkage of MT blended cement mixtures is higher than that of the reference sample, and the shrinkage values, at a given age, increases as the MT content increases.

These higher shrinkage values originate from the absorption of part of the mix water by the mordenite particles having an open crystal structure and from the low dynamic modulus of the MT mortars. These results are in contrast with those of Shon and Kim (2013) [29] and Najimi's et al. (2012) [18]. These later found that the replacement of cement at a level of 15 and 30% by natural zeolite led to an important decrease in shrinkage, about 84 and 64% at 90 days, respectively.

However, our results are in accordance with those of Jana (2007) who reported that the shrinkage of natural zeolite mixtures with 10 and 20% Portland cement replacement levels were similar or slightly higher than that of the control mixture [37]. Itim et al. (2011) [17] showed that the substitution of cement by 10% of natural

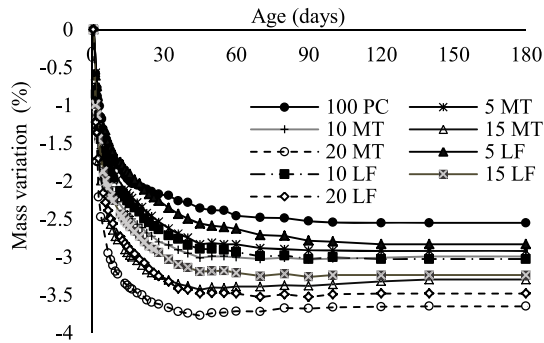


Fig. 8. Mass loss versus age.

Fig. 8. Perte de masse fonction du temps.

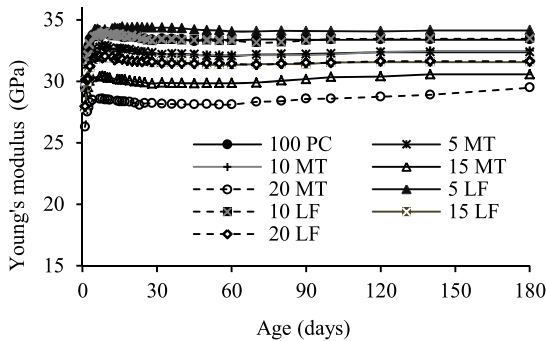


Fig. 9. Dynamic Young modulus versus age.

Fig. 9. Module dynamique de Young fonction de temps.

pozzolan caused a very important shrinkage exceeding that of the ordinary cement. In addition that, they reported that the shrinkage of the 20% natural pozzolan sample reaches that one without any mineral additions, whereas the shrinkage of those containing 30% natural pozzolan remains slightly lower.

The mass variation of the different studied mortars is shown in Figure 8. The mass variation curves corresponded approximately to those observed on the shrinkage test. The water evaporation rate is high up to four weeks, but beyond, the mass variation is very low, even almost null. Importantly, it can be noted that the mass variation of the MT and the LF mortars are greater than that of the reference mortar. This increased evaporation was likely related to the decrease in the Portland cement content in these mortars. Thus, the amount of chemically-bound water during the hydration decreased, which increases the free water fraction susceptible to be evaporated.

4.6 Dynamic Young's modulus

The data shown in Figure 9 indicate that, compared to the elastic modulus of 100CP mortar, the one of 5 LF is higher, those of 10 LF and 15 LF are equal, whereas that of 20 LF is slightly lower. All of the MT mortars presented an elastic modulus lower than that of 100 PC mortar; however, the difference in strength was not significant. This indicates

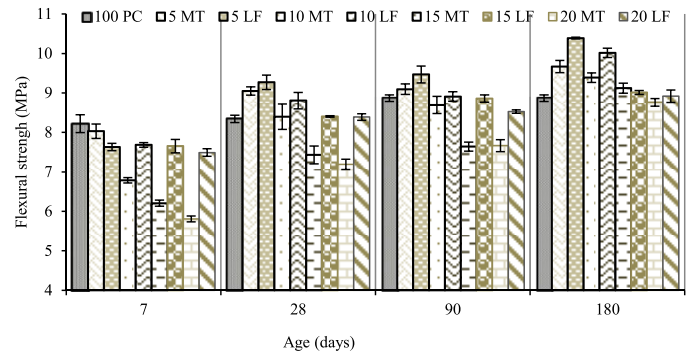


Fig. 10. Flexural strength development of different mixtures.

Fig. 10. Développement de la résistance à la flexion des différentes formulations.

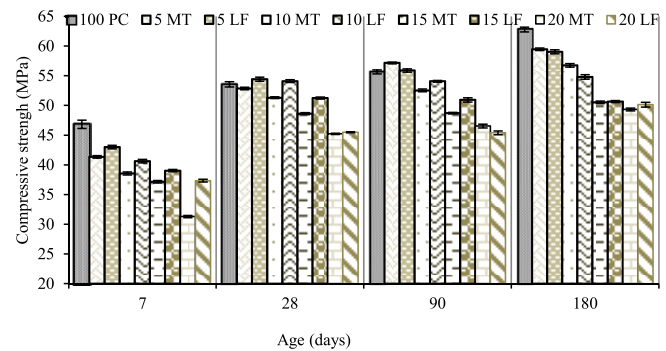


Fig. 11. Compressive strength development of different mixtures.

Fig. 11. Développement de la résistance à la compression des différentes formulations.

that the elastic modulus is not proportional to the compressive strength.

The elastic moduli of MT mortars are lower than that of ordinary mortar which is in agreement with the report of American Concrete Institute Committee (ACIC). The ACIC noted that mortar containing pozzolans typically has a lower modulus of elasticity than a similar mortar without pozzolans [28]. The elastic modulus of the mortar is a function of the modulus of each constituent and their relative mixture proportion. The elastic modulus of the reference mortar is the highest; this is because the modulus of the Portland cement particles is greater than that of lightweight materials particles (MT particles) [38].

4.7 Flexural and compressive strength of mortars

The flexural and compressive strengths of the mortars were determined at 7, 28, 90 and 180 days, the results are presented in Figures 10 and 11. Apart from 7 days, most of the flexural strength values of the LF mortars were better than that obtained from the reference mixture. For MT mortars, a delay of strength is observed at 7 days as the tuff content increases in the binder. From 28 to 180 days, the 5 and 10 MT samples developed a flexural strength greater

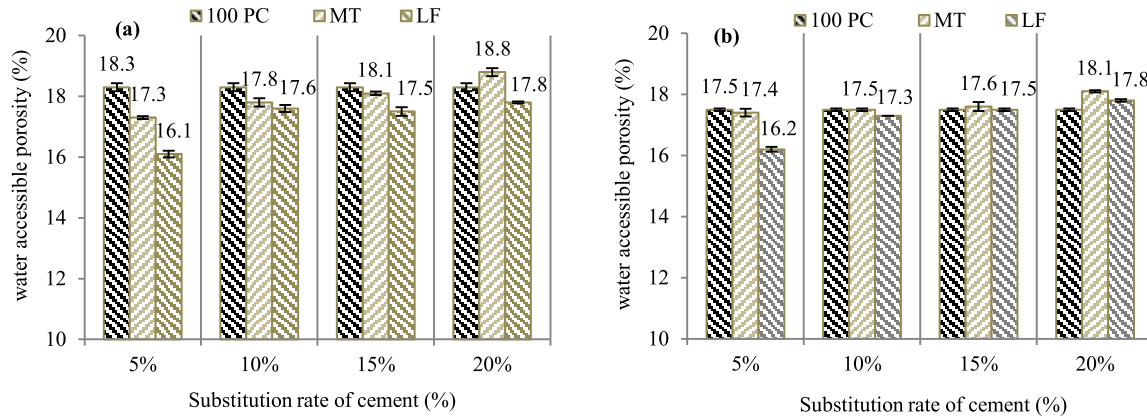


Fig. 12. Water accessible porosity: (a) 28 days; (b) 90 days.

Fig. 12. Porosité accessible à l'eau : (a) 28 jours ; (b) 90 jours.

than that of the ordinary cement mortar. The 15 and 20 MT mortars delayed the enhancement of the flexural strength and improved it only after 180 days. Similar results were found by Messaoudene et al. (2012) [39] when volcanic ash were used as mineral admixture. In another study by Kocak et al. (2013), the use of clinoptilolite (zeolite) decreased the flexural strength at all the test ages [40]. The reaching of acceptable levels of mortar strengths after 1 year of cure is attributed to the slow pozzolanic reaction of M.

As expected, the compressive strength increased over time. The rate of increase depended upon the level of mineral addition replacement and age. As seen in Figure 11, it appears that mordenite-rich tuff used in this study have a moderate strengthening effect. Thus, compared to the reference mortar (100 PC), MT mortars have a lower compressive strength which decreases as the MT content increased; apart from 5 MT mortar for which an increasing of compressive strength was observed at 90 days. This is most likely owing to the reduction in $\text{Ca}(\text{OH})_2$ levels in the medium despite the increase in MT content [41]. The second possible reason may be the particle size of MT compared to PC and LF particles size distribution. Thus, although PC and the mineral additions have close specific surface areas, particle size distribution appears to be different; MT is the coarser [42]. Although a constant W/B ratio in the mixtures, the presence of the MT (which consumes water) decreased the workability compared to the 100 PC sample (Table 4). This ultimately resulted in an insufficient amount of water available for the pozzolanic reaction. The loss in compressive strength was 5.3, 9.62, 19.5 and 21.44% for 5, 10, 15 and 20% MT replacement of PC, respectively, at 180 days. This is consistent with Senhaji et al. (2014) study which reported that the natural pozzolan decreased the compressive strengths of mortar at all levels of replacement [19]. The decrease in the strength can also be explained by the fact that more than 15% of MT content in mortars led to a surplus of the small-sized fractions which separate from Portland cement grains, causing an unpacking of the system and decreasing of the mortar strength [43]. This is in agreement with Benaissa et al. (2016) and Bibi et al. (2008) who found that partial

replacement of cement at a level of 10% was the optimum amount to enhance the compressive strength of concrete [44,45]. Similar results were reported by Perraki et al. (2010) who observed that zeolite cements based on CEMI 52.5 develop lower strength than pure cement at ages up to 28 days, while they develop higher strength than pure cement at 90, 180 and 360 days [46]. In the other hand, 5 and 10 MT blended cements have acceptable strengths and could match the limits of FDP 15-010 (1997) for CEMI 52.5 Portland cement [47], whereas the 15 and 20 MT mixes could reach the 180 days strength of the commercial Portland cement.

In the case of LF mortars, the results showed that the compressive strength decreased with increasing LF content at all ages. The substitution of Portland cement by LF at a level of 5 and 10% has significantly increased the 7 and 28-day mortars strengths. Such increase of strength may be explained by the very fine particles of FC, which may have acted as nuclei for precipitation of C-S-H and thus increased the degree of cement hydration [48].

4.8 Water-accessible porosity

Bulk density, dry bulk density and porosity of the mortar mixtures are given in Table 4. The results indicate that addition of LF did not have any substantial influence on the bulk density which was almost equal for all mixtures as the difference was less than 1%. The values of MT mortars were at the most 2.5% lower than the reference mortar. Similar results were seen in the dry bulk density; this most likely related to the porosity of the material and its low specific gravity. From the results of water-accessible porosity test showed in Figure 12, it can be noticed that the porosity of the LF samples are lower than that of 100 PC at 28 days but, nearly equal at 90 days for all the replacement levels. In the other hand, at 28 days, the water-accessible porosities of 5, 10 and 15 MT samples are lower than that of 100 PC, apart from 20 MT whose porosity greater by 3%. The porosity of MT samples decreased over the time so that at 90 days, almost the MT mixtures accrued a significant decrease on their porosities to equalize that of 100 PC. At this age, the porosity of 20 MT sample is less than that

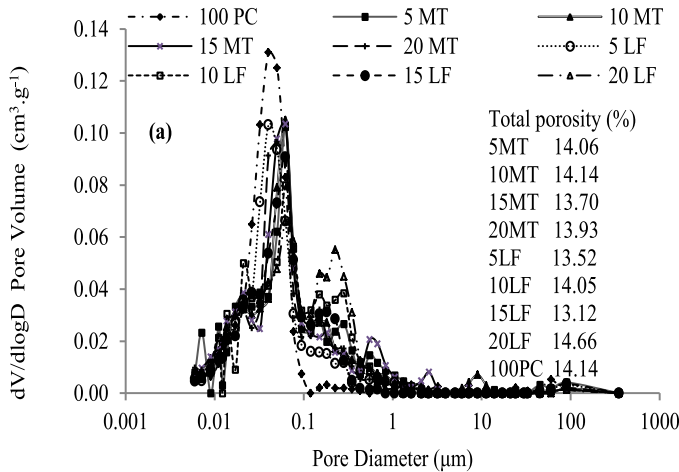


Fig. 13. Pore size distribution and total porosity of mortars: (a) at 28 days; (b) at 90 days.

Fig. 13. Distribution de la taille des pores et porosité totale des mortiers: (a) 28 jours; (b) 90 jours.

recorded at 28 days by 3% but higher than that of 100 PC sample by 3%. This low increase in porosity is not deleterious and may be attributed to the porous structure of mordenite-rich tuff. In our point of view, this porosity is non-deleterious because the increase of pore volume, measured by MIP, in 20MT mortar was mainly due to the small pores of diameter below $0.1 \mu\text{m}$ (Fig. 13). And, it is known that the pores smaller than $0.5 \mu\text{m}$ are more influential on drying shrinkage than on strength and permeability. Also, since the water absorption of the MT is higher than that of cement, increased water absorption is inevitable. The substitution of Portland cement by pozzolans modifies the relative content in hydrates as well as the microstructure. The most important changes concern the transition zone between the aggregate and the matrix paste, where the $\text{Ca}(\text{OH})_2$ content decreases and the porosity increases [49].

4.9 Pore size distribution

At 28 days, all MT and LF mixtures, except the 20LF, resulted in a lower porosity compared to the 100 PC mortars (Fig. 13a). This result is in consistent with the result of Poon et al. (1999) [50], who reported that at all ages; the porosity of the paste with 15% zeolite replacement is lower than the one of Portland cement paste having the same water/binder ratio. A drastic change is observed in pore size distribution and total porosity of all mixtures between 28 and 90 days. Clearly, the total porosity of all mixtures decreased as the curing age increased. Indeed, at 90 days, the pores having diameters between 0.1 and $1 \mu\text{m}$ are relatively sealed (Fig. 13b). This decrease trend of porosity in the MT mortars indicates an increasing degree of pozzolanic reaction in the blended mixes as the formation and growth of pozzolanic reaction products filled the pores [50]. The difference between the total porosity of 5 and 10 MT mortars and that of 100 PC

increased as the curing age increased. For example, the porosity of 5 MT mortar is 0.57% lower than that of 100 PC mortar at 28 days but is 5.35% lower than the latter at 90 days. Although the total porosity of 15 and 20 MT mortars decreased at 90 days, it remained slightly greater than the one of 100 PC (Fig. 13b). This indicates that the pore reducing ability of the MT is limited and only a small percentage of replacement reduced porosity. The increase in porosity is most likely due to the preservation of microcrystalline structure of zeolite in the hardened mixes.

The results of total pore volume (Fig. 13), obtained by MIP agreed well with the measurement of basic physical properties by water vacuum saturation (Fig. 12). The increase of pore volume in some samples is mainly due to the small pores of diameter below $0.1 \mu\text{m}$ (Fig. 13). It is known that the pore size distribution controls the properties of hardened cementitious systems. Pores larger than $0.5 \mu\text{m}$ are more influential on strength characteristics whereas those smaller than $0.5 \mu\text{m}$ are more influential on drying shrinkage. Therefore, increased total porosity would not be detrimental on strength (Fig. 11) [51].

5 Conclusion

Through this study, we show the potential advantages of replacing Portland cement up to 20% by mordenite-rich tuff and limestone filler in cementitious materials. Based on the physico-mechanical and microstructural test data, the following conclusions can be drawn:

- the LF mixtures seem to be the closest to the control mixture than the MT ones;
- incorporation the MT at 10, 15 and 20% levels increased the hydration heat up to 7 h and accelerated the appearance of the maximum thermal effect by allowing the C_3A to undergo renewed rapid hydration and convert the ettringite to monosulfoaluminate;
- the pore reducing ability of the MT is limited to a Portland cement replacement rate of 10%. Beyond this level, the microcrystalline structure of mordenite may be preserved in the hardened mixes to slightly increase the porosity;
- the improvement in water accessible porosity of the MT is also limited to up to 10% replacement of Portland cement. However, the use of the MT at a level up to 20% is advisable since the water accessible porosity was only slightly affected;
- the blended cements containing the MT exhibited less free $\text{Ca}(\text{OH})_2$ in hardened pastes because of the pozzolanic reactivity of the mordenite-rich tuff;
- the MT mortars delayed the enhancement of the flexural strength, so that 15 and 20% MT have shown their strengthening effect only after 180 days;
- the replacement of ordinary Portland cement by 5 and 10% of mordenite-rich tuff offers to mortars a suitable compressive strength and could match the limits of FDP 15-010 (1997) for CEMI 52.5 Portland cement. The mixes containing 15 and 20% of the MT could reach the 180 days strength of the commercial Portland cement; the resulting strengths are satisfactory for structural applications;

- mortars containing the MT presented a slightly greater shrinkage and a lower Young modulus;
- the water demand of the MT-blended cement pastes increased with the MT substitution ratio because of the micropores inherent to its structure. The addition of the MT delayed the initial setting time but accelerated the final setting time.

In the light of the above and the conclusions drawn, the use of mordenite-rich tuff as a partial substitute for Portland cement presents physico-mechanical and micro-structural advantages. The economic virtue is not studied in the present research, but the action of substituting Portland cement by an abundant pozzolanic product will imperatively reduce the cost of manufacturing Portland cement and the emission CO₂ into the environment. A more detailed measurement of mortars durability indicators will be made in a follow-up study. Additional studies are underway to assess the feasibility of Portland cement substitution by the MT on concrete.

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