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## The Strong Confinement Regime in HgTe Two-Dimensional Nanoplatelets

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**Abstract:** The confinement in colloidal HgTe nanocrystals enables this material to be promising for colloidal optoelectronics over a wide range of energies, from the THz spectral range up to the visible region. Herein, by using a combination of high energy absorption HgTe nanoplatelets and low energy absorption HgTe nanocrystals, we probe optical transmission of HgTe nanoparticles over the 0.26–1.8 eV range, from 0 K to 300 K temperatures and under simultaneous pressure, up to 4 GPa. While the pressure dependence of nanoplatelets follows the one observed for bulk and nanocrystals, the temperature dependence dramatically differs for nanoplatelets. The modeling of the electronic energy dispersion using up to 14-band **k.p** formalism suggests that the second conduction band and higher bands of HgTe play a vital role to describe and explain the HgTe nanoparticle spectroscopies.

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reduce the device dark current. Alternatively, the development of HgTe based colloidal heterostructures appears promising to expand the reachable range of wavelength of the 2D NPLs,<sup>30,31</sup> but also raises questions relative to the understanding of the optical spectrum to disentangle charge delocalization effect from pressure effect resulting from non-lattice matched shells.<sup>32</sup>

In this paper, we explore in a systematic way the effect of confinement, temperature and pressure on the energy dispersion of HgTe. In practice, we grow a series of HgTe NCs with various confinement energies from 0.26 to 1.8 eV. The temperature is tuned from 0 K to 300 K which corresponds to cryogenic operating condition for IR sensors and finally the pressure is tuned up to 4 GPa, corresponding to the full range of existence of the zinc blende low pressure phase of HgTe nanoparticles. While the pressure dependence of the spectroscopic properties appears to be weakly affected by the confinement, we show that the temperature dependence of the 2D NPLs is opposite to the one observed for bulk and weakly confined NCs. Our multiband **k.p** modelling points to the central role of the second conduction band in the origin of this observation. Finally, thanks to a 14 bands **k.p** model we are able to propose a set of parameters which correctly account for temperature and pressure dependence and describe the dispersion relation of HgTe over an extended range of the Brillouin zone.

## METHODS

**Chemicals:** Mercury chloride (HgCl<sub>2</sub>, Strem Chemicals, 99%), Hg(Acetate)<sub>2</sub> (Hg(Ac)<sub>2</sub> Aldrich, > 99.0 %), **Mercury compounds are highly toxic. Handle them with special care.** Tellurium powder (Te, Sigma-Aldrich, 99.99%), cadmium oxide (CdO) (Aldrich 99.99 %), trioctylphosphine (TOP) (Alfa, 90 %), oleic acid (OA, Sigma, 90%), propionic acid (Aldrich, 99 %), oleylamine (OLA, Acros, 80-90%), dodecanethiol (DDT, Sigma-Aldrich, 98%), Trioctylamine (TOA, Aldrich), Octadecene (ODE) (Aldrich, 90 %), chloroform (Carlo Erba), methanol (VWR, 98.5%) acetone (VWR rechapur), ethanol (absolute VWR), toluene (VWR, 99.8 %). All chemicals are used without further purification.

**Cadmium propanoate (Cd(Prop)<sub>2</sub>):** 1.036 g of CdO is mixed in 10 mL of propionic acid under Ar for 1 hour. Then the flask is open to air and the temperature risen to 140 °C up to the point the volume gets divided by a factor of two. The whitish solution is precipitated by addition of acetone. After centrifugation, the solid is dried under vacuum for 24 hours.

**1 M TOP:Te precursor:** 2.54 g of Te powder is mixed in 20 mL of TOP in a three neck flask. The flask is kept under vacuum at room temperature for 5 min and then the temperature is raised to 100 °C. Furthermore, degassing of flask is conducted for the next 20 min. The atmosphere is switched to Ar and the temperature is raised to 275 °C. The solution is stirred until a clear orange coloration is obtained. The flask is cooled down to room temperature and the color switches to yellow. Finally, this solution is transferred to an Ar filled glove box for storage.

**Synthesis of 3 MLs CdTe NPLs:** The procedure follows the procedure proposed by Pedetti *et al.*<sup>33</sup> In a three-neck flask, Cd(prop)<sub>2</sub> (390 mg, 1.5 mmol), oleic acid (240 μL, 0,75 mmol), and 30 mL of octadecene are degassed under vacuum at 90 °C for 1 hour. Under argon flow, TOP:Te at 1 M (300 μL) diluted in 1.5 mL of octadecene is swiftly injected at 210 °C. After 30 minutes of reaction, the mixture is cooled down and 1.7 mL of oleic acid is introduced. The resulting nanoplatelets are precipitated twice with an excess of ethanol and then are redispersed in 30 mL of hexane.

**Cation exchange for synthesis of 3 MLs HgTe NPLs with TOA:** In a glass tube, 2 mL of hexane and 200 μL of HgTe NPLs are introduced. 60 μL of a fresh 0.1 M HgAc<sub>2</sub> in trioctylamine solution is added at room temperature (The Cd:Hg ratio is 1:2). The cation exchange is followed by absorption spectroscopy. Once the first excitonic peak has reached 824 nm, 80 μL of oleic acid are added to stabilize the HgTe NPLs. The excess of precursor is removed by a two-step centrifugation (6000 rpm, 30 s and we keep the supernatant then 6000 rpm, 5 min). The resulting NPLs are resuspended in 1 mL of toluene. Procedures for other surface chemistry on the HgTe NPL surface are given in the SI

**HgTe NCs synthesis with band edge at 4000 cm<sup>-1</sup>:** 513 mg of HgCl<sub>2</sub> was added to 60 mL of oleylamine in a 100 mL round flask. The solution was placed under vacuum and heated to 110 °C for 1 h. Then, the temperature is decreased to 80 °C and solution is placed to Ar atmosphere. 1.9 mL of TOP:Te (1 M) with 10 mL of oleylamine is added to the mercury solution. The solution color gradually turns to dark brown and the reaction is stopped at 3 min. A solution made of 1 mL of dodecanethiol and 9mL of toluene is quickly added to quench the reaction. The nanocrystals are then precipitated with ethanol. After centrifugation, the nanocrystals are redispersed in chloroform. The washing step is repeated one more time. The solution is filtered with a 0.2 µm filter and redispersed in 6mL of chloroform. Procedures for other NC sizes are given in the SI.

**Electron microscopy:** For transmission electron microscopy (TEM) pictures, a drop of the NC solution is drop-casted onto a copper grid covered with an amorphous carbon film. The grid is degassed overnight to reduce future contamination. A JEOL 2010F is used for acquisition of pictures and operated at 200 kV.

**Infrared spectroscopy as a function of temperature:** The solution of nanocrystal is drop-casted onto a double side polished Si wafer for the sample in the mid-infrared and onto a CaF<sub>2</sub> substrate for the sample in the near-infrared. The sample is mounted on the cold finger of a closed cycle cryostat. The latter is used in a transmission configuration with two ZnSe windows. The cryostat is then introduced into the Fourier Transform Infrared spectrometer (FTIR, Thermo Fischer iS 50). Two sources have been used: a white light in the near infrared and a globar in the mid infrared. Two beam splitters have been used: a CaF<sub>2</sub> beam splitter in the near infrared and an extended KBr in the mid infrared. Two detectors have been used: an InGaAs sensor in the near infrared and a DTGS in the mid infrared. In all case, the background is made at room temperature using the same optical configuration but without sample onto the substrate. Spectra are acquired with a 4 cm<sup>-1</sup> resolution and typically averaged 100 times.

**X-ray diffraction under pressure:** Experiments were carried out using a membrane diamond anvil cell equipped with Böhler-type anvils with 500 micrometer culets. A stainless-steel gasket of 200 micrometer thickness was pre-indented to 70 µm, provided with a 200 µm hole, and loaded with the sample, a 4:1 methanol-ethanol pressure transmitting medium and a 10 µm diameter ruby sphere which served as pressure marker. The "Mao hydrostatic" pressure scale was used to determine pressures. X-ray powder diffraction measurements were performed at the XRD platform of the IMPMC on a Rigaku MM007HF diffractometer equipped with a Mo rotating anode ( $\lambda_{K\alpha 1} = 0.709319 \text{ \AA}$ ,  $\lambda_{K\alpha 2} = 0.713609 \text{ \AA}$ ), Varimax focusing optics, and a RAXIS4++ image plate detector. X-ray data were collected at 20 °C. A LaB<sub>6</sub> standard sample was measured in the same experimental conditions to calibrate the Fit2D program, the image processing software used to integrate the intensities around the Debye-Scherrer rings and to get the 1D patterns.

**Infrared spectroscopy under pressure:** Membrane diamond anvil cells (DACs), equipped with IIa-type diamonds having a 400-500 µm culets, are used. Stainless steel gaskets with a thickness of 200 µm are indented down to 50 µm. A 150 µm hole is then drilled by electro-erosion. The gasket is placed on the top of one of the DAC diamonds, and the gasket hole filled with NaCl. After introducing a ruby crystal, the DAC is closed under gentle pressure until the salt forms a clear window. A drop of a diluted suspension of NCs in toluene is then added and dried on top of the NaCl window to form a dry and uniform film of NCs. The DAC is then closed and introduced in the Cassegrain microscope. the pressure is generated by the means of a metallic membrane inflated with He gas. The *in-situ* pressure is monitored by the ruby luminescence technique. A 532 nm laser is shone through one of the Cassegrain objectives and focused on the ruby crystal embedded in the cell. Ruby photoluminescence light is collected and sent to an OceanOptics spectrometer, showing a characteristic doublet around 694 nm at ambient pressure. This photoluminescence is fitted and the pressure is computed using the main peak position. Once the pressure is stabilized, an infrared absorbance spectrum is acquired using a Thermo Fisher Nicolet iS50 FTIR with a globar source, a quartz beam splitter, and a silicon detector in transmission configuration. This operation is repeated for various pressure from 0 GPa to 4 GPa. When the maximum pressure is reached, the membrane is deflated and several spectra are acquired to check the reversibility of the pressure effect.

**k.p modeling and simulations:** 3x3 and 14x14 **k.p**  $H(k,T,P)$  hamiltonian matrices are processed symbolically and diagonalized numerically for each couple of temperature and pressure  $(T,P)$  using the parameters in Table 1. Bulk material energy dispersion diagrams  $E(k)$  are calculated using the wave vector  $k$  as a free-varying independence variable. On the contrary, spectroscopic energies of the form  $E(T,P)$  are extracted by modifying  $k$ , with the following procedure. Since thermal dilation or pressure contraction occurs, the nanoparticles are implicitly labelled by their discretized wave vector  $k_0 = 2\pi/L_0$  in (0 K,0 GPa) conditions. Based on the starting energy  $E_{start}$  in the starting conditions  $(T_{start}, P_{start})$ , the dispersion relation  $E(k)$  is first inverted giving out a reference wave vector  $k_{start}$ . This  $k_{start}$  wavevector is then dilated or contracted back to (0 K,0 GPa) conditions following equation S6b, thus giving the  $k_0$  wavevector labelling the nanoparticle. The temperature  $T$  and pressure  $P$  are then freely scanned. For each  $(T,P)$ , the norm of the discretized wave vector  $k_0$  is changed into  $k(k_0,T,P)$  according to the dilation or contraction of the lattice parameter. The **k.p** matrix is finally numerically evaluated at this altered wave vector  $k$  and the eigen energies are reported against T and/or P.

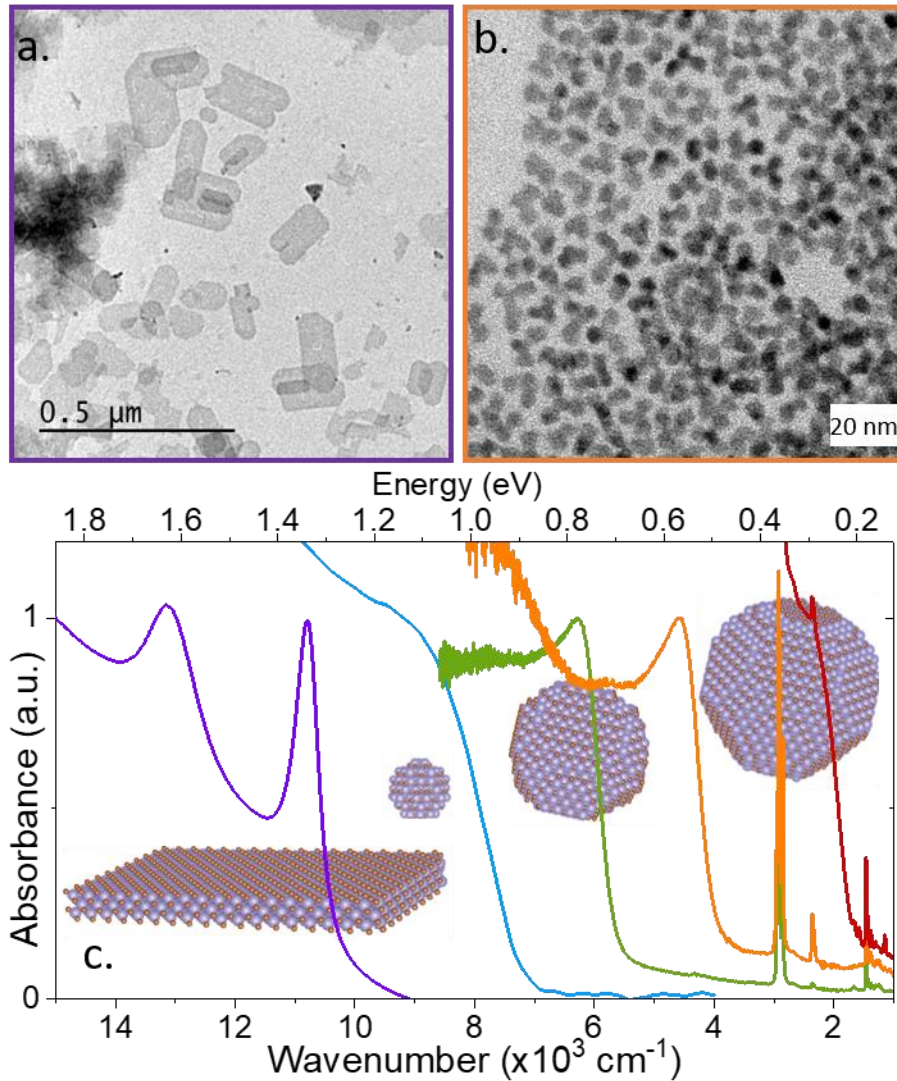
## RESULTS AND DISCUSSION

### From weak to strong confinement in HgTe Nanocrystals

We grow a series of HgTe NCs with various sizes to widely tune the confinement energy. As a strongly confined form of HgTe, 2D NPLs are grown.<sup>15</sup> The HgTe NPLs are obtained using a two-step approach where CdTe NPLs are first synthesized,<sup>34</sup> see Figure S1. The NPLs are 3.5 monolayers thick (3 planes of Te surrounded by 4 planes of Cd) corresponding to a thickness of 1.1 nm.<sup>34</sup> Those NPLs are then exposed to a bulky Hg precursor (Hg oleate), which transforms them into pure HgTe NPLs due to cation exchange.<sup>15</sup> Given that CdTe (0.648 nm) and HgTe (0.646 nm<sup>35</sup>) have similar lattice parameter (0.3 % difference) the final thickness of the HgTe NPL is 1.1 nm<sup>15</sup> which is confirmed by the fact that the two types of material present similar X-ray diffractogram, see Figure S2.

The lateral extension of the particles is large and can reach several hundred of nm according to electronic microscopy, see **Figure 2a** and S3-S5. Due to the thickness, the material is strongly confined along <001> crystal direction and the band edge absorption appears around 900 nm ( $\approx 11000 \text{ cm}^{-1}$  or  $\approx 1.4 \text{ eV}$  at room temperature), see **Figure 2c**. It has been demonstrated that the exact value of the band-edge energy can be further tuned with the capping ligands which also take part in the effective delocalization length.<sup>15</sup> This enables a fine tuning of the confinement energy while keeping the inorganic core unchanged.

In addition, we grow a series of HgTe NCs with room temperature band edge between  $9000 \text{ cm}^{-1}$  (1.11 eV) and  $2000 \text{ cm}^{-1}$  (248 meV), see **Figure 2b-c** and S6. The syntheses follow the procedure of Keuleyan *et al.*<sup>36</sup> (for large NCs) and Geiregat *et al.*<sup>37</sup> (for the smallest NCs). The combination of the HgTe NPLs and NCs allows us to tune the confinement energies over more than a factor of 5, from strongly to weakly confined materials as we switch from NPLs to NCs.

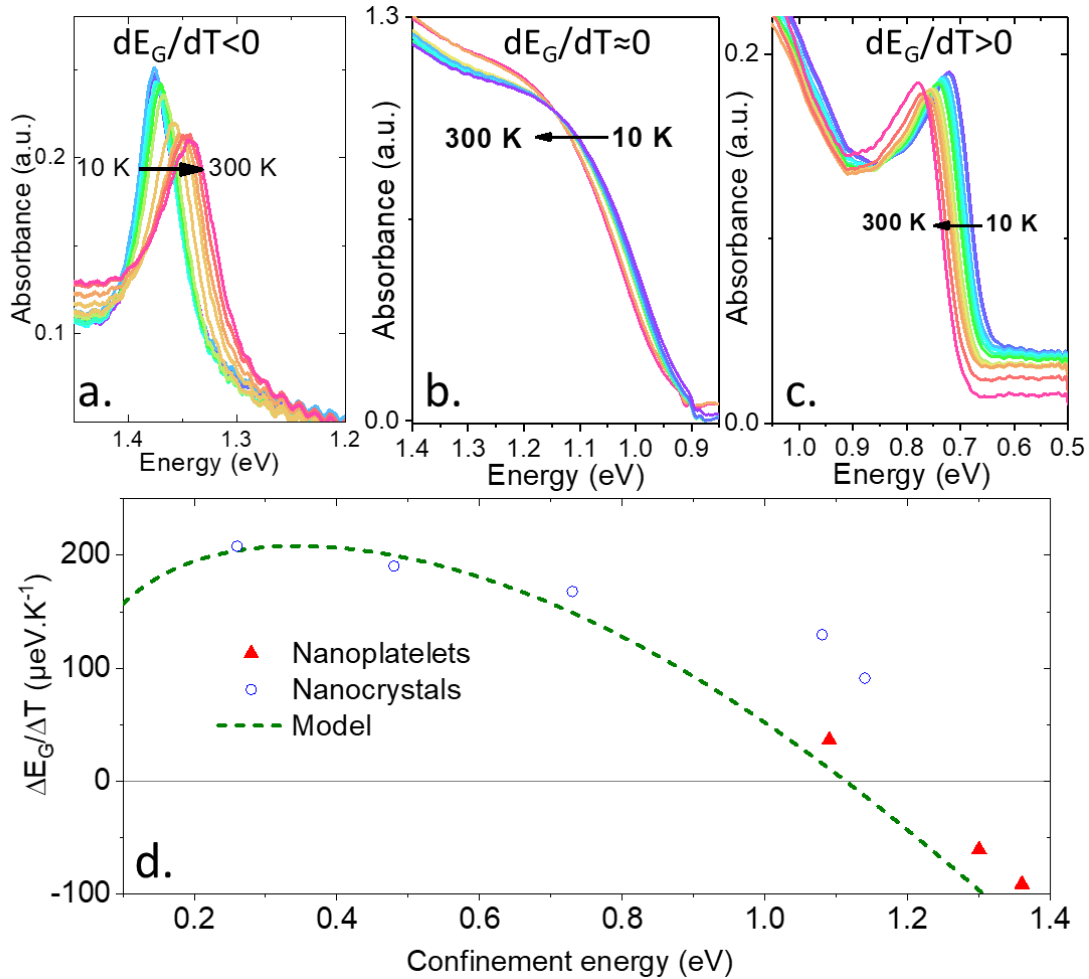


**Figure 2 Confined NCs of HgTe.** TEM images of a. HgTe NPLs and b. NCs. The associated absorption spectrum appears in purple and orange, respectively. c. Infrared absorption spectra for HgTe NPLs and NCs with various sizes.

### Effect of temperature of the infrared absorption

We then probe the effect of temperature on the IR spectrum of the material, see the spectra in **Figure 3a-c** and S8-S9. The temperature dependence on the material is dramatically influenced by the material confinement, as reported in **Figure 3a-c**. For large (*ie* weakly confined) HgTe NCs (**Figure 3c**), the  $E_G$  confinement energy redshifts as the temperature is lowered, corresponding to a positive value for  $dE_G/dT$ . This behavior is similar to the one observed in bulk HgTe,<sup>38,39</sup> but is opposite to the trend observed for wider band gap II-VI semiconductors such as CdTe<sup>40</sup> or CdSe<sup>41,42</sup> where whatever the particle size only negative value have been observed for  $dE_G/dT$ . In NCs, such positive value for  $dE_G/dT$  has been observed for lead halide perovskites<sup>43</sup> or large PbS NCs.<sup>44</sup> Nevertheless, as the confinement is increased, we observe a reduction of the value of  $dE_G/dT$  (**Figure 3b**) and surprisingly even an inversion of its sign in the case of NPLs (**Figure 3a**).





**Figure 3 Effect of the temperature on the infrared spectra of HgTe NCs.** *a.* Infrared absorption spectra for HgTe NPLs at various temperature between 10 K and 300 K. The absorption blueshifts as temperature is lowered. *b.* Infrared absorption spectra for small HgTe NC at various temperature between 10 K and 300 K. The absorption is barely affected by temperature. *c.* Infrared absorption spectra for large HgTe NC at various temperature between 10 K and 300 K. The absorption redshifts as temperature is lowered. *d.* Experimental temperature dependence of the band gap variation as a function of HgTe NC and NPL  $E_G$  confinement energy (at  $T=10$  K). The green dotted line is from the (3+1)  $k.p$  model.

This change of sign for  $dE_G/dT$  excludes that the observed shifts may result solely from a thermal dilation or contraction of the nanoparticles, the size of which will itself tune the confinement. In addition, this contribution of temperature has been found to be quantitatively weak (few  $\mu\text{eV} \cdot \text{K}^{-1}$ )<sup>23</sup> due to the very small thermal dilation, see Figure S11b. Note that the dimensionality does not seem to be a driving parameter for the magnitude of  $dE_G/dT$  since NCs and NPLs with the same  $E_G$  confinement energy present similar values for  $dE_G/dT$ , see circles and triangles in **Figure 3d**. A similar trend for  $dE_G/dT$  (reduction of the amplitude and change of sign) as a function of the inverse of the particle size was observed in the case of PbS NCs. By excluding other possibility, the band gap shift with temperature was finally attributed to a size dependent electron-phonon coupling.<sup>23,44</sup>

### Effect of pressure of the infrared absorption

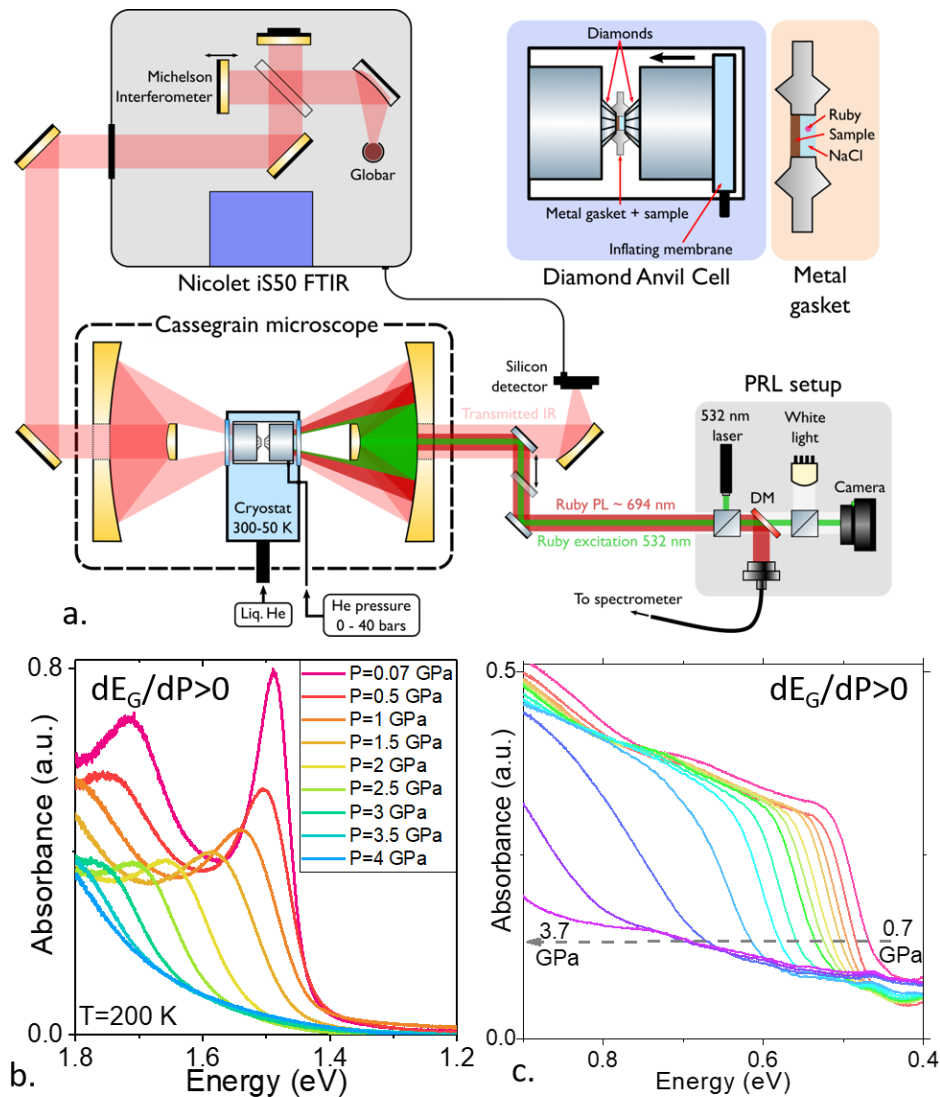
Similarly, we measured the pressure dependence of the band gap for the HgTe NPLs at various fixed temperatures. Note that for HgTe NCs the pressure dependence of the band gap was already reported in ref<sup>45</sup> at room temperature. This is why the effect of pressure is discussed in the case of NPL here. In practice, the sample is loaded in a diamond anvil cell together with NaCl which is used as an infrared transparent pressure transmission medium. A ruby is also loaded in the cell and used to determine the applied pressure,



see **Figure 4a** for a scheme of the experiment and the Supporting Information (SI) for details. All the reported observations appear to be fully reversible in the targeted range of pressure (*i.e.* zero pressure spectrum is recovered after pressure releasing). On the other hand, the process is strongly hysteretic: the full release of pressure is required to recover the initial spectrum.

To understand pressure induced change in the infrared spectrum, it is first necessary to determine if the changes might be the result of structural phase transition. Bulk HgTe crystallizes in the cubic zinc blende structure at ambient pressure and undergoes several phase changes as pressure is increased. For bulk material, the cinnabar phase appears around 1.4 GPa.<sup>46,47</sup> Under NC form, the zinc-blende phase remains stable up to higher pressures.<sup>45</sup> In the range between 3 and 5 GPa, we can notice the presence of both zinc blende and cinnabar phase (Figure S7). It is however unclear if the two phases exist within the same NPL or if the signal is associated with different particles. In the zinc-blende phase, we identify a decrease in the lattice parameter with initial pressure ( $da/dP \approx 7$  pm/GPa), which corresponds to a bulk modulus of 30 GPa for the HgTe NPLs. This value is similar to the one measured for the HgTe NCs<sup>47</sup> (33 GPa) which was relatively close to the one of the bulk (34 GPa).<sup>48</sup>

The  $E_G$  confinement energy appears to be dependent on the applied pressure, see **Figure 4b-c**, 6c and S10. The confinement energy gradually increases with pressure corresponding to a positive value for  $dE_G/dP$ . This is similar to what have been measured for HgTe NCs,<sup>45</sup> and wider band gap II-VI semiconductor quantum dots such as CdSe<sup>49</sup> and core shell made of CdSe/ZnS.<sup>50,51</sup> Applying pressure in the range of existence of the zinc blende phase can leads to a shift of the confinement energy up to 300 meV, which is far more than the shift induced by cooling the sample from room temperature to cryogenic condition ( $\approx 35$  meV). This order of magnitude is partially expected from the lattice parameter change under pressure around 6% in Figure S7c and estimated around only 0.08% over 300K temperature range in Figure S11b. Above 4 GPa, no excitonic feature is identified. This observation is correlated to the disappearance of the zinc-blende phase under similar pressure, see Figure S7.



**Figure 4 Effect of the pressure on the infrared spectra of HgTe NPLs.** a. Scheme of the experimental setup of the SMIS beamline of synchrotron SOLEIL to probe the infrared spectrum of a HgTe NPLs under pressure. b. Infrared spectrum of HgTe NPLs at 200 K and under various pressure in the range of 0 to 4 GPa. c. Infrared spectrum of HgTe NCs at 300 K and under various pressure in the range of 0 to 4 GPa

## Multiband $\mathbf{k}\cdot\mathbf{p}$ modelling

In order to address the striking temperature and pressure dependence of the spectra observed in **Figure 3** and **Figure 4**, we model the nanoparticles energies using multiband  $\mathbf{k}\cdot\mathbf{p}$  simulations. HgTe dispersion appears to be non-parabolic<sup>26,52</sup> and the effective mass approximation (1 band model) is only valid over 1% of the Brillouin zone. For wavevector far away for the  $\Gamma$  point, the dispersion becomes first linear and then further slightly bends down for wavevectors corresponding to the confinement of NPL (confinement energy above 1 eV). More bands need to be included in the  $\mathbf{k}\cdot\mathbf{p}$  model to obtain an accurate description of the dispersion relation.

When a detailed description of the valence band is required, 8 band  $\mathbf{k}\cdot\mathbf{p}$  model describing the  $\Gamma_6$ ,  $\Gamma_7$  and  $\Gamma_8$  bands and their degeneracy is used.<sup>53,54</sup> It has been estimated that the 8 band  $\mathbf{k}\cdot\mathbf{p}$  model correctly described the Brillouin zone over 8% of it, corresponding to energies significantly smaller than the ones of the NPLs which are around 1.1-1.8 eV.<sup>53</sup> While, the effective mass approximation only describes the very bottom parabolic part of the light hole like ( $\Gamma_8$ ) band, the inclusion of additional conduction band is critical to capture the linear dispersion occurring at larger wavevector and higher energies.

In HgTe contrary to PbS, the hole and electron effective masses strongly differ, the valence band mass being more than 10 times the conduction band mass. So the confinement is mostly the result of the conduction band and the latter is the one requiring the most accuracy. In the SI (Figure S14-S17), we provide a 14-band **k.p** model, which includes the minimum amount of bands to properly take into account additional conduction bands.<sup>55</sup>

This 14-band model aims at supporting a simpler (3+1) effective bands model later described in the main text. This simple model provides the driving concepts leading to the realistic calculated 14-band structure. The 14-band model includes notably the spin-orbit coupling, disregarded in the simplified model, and all relevant interactions. Note that coulombic renormalization of the band gap has been neglected since these effects have found to be small in the case of HgTe NCs.<sup>26</sup> Even though the conclusions are qualitatively the same using each version of the model, the complete version may somehow blurry the message with a large amount of involved parameters. This is why, in the following, we describe a simplified approach. We use a (3+1) bands one-dimensional **k.p**-like model, referred to as the simple model and described in the SI. Its illustrative role is to be tractable and to account for all the experimental features with a minimal set of necessary ingredients, defined in **Figure 1**. The associated parameters are given in Table 1.

The (3+1) band model accounts for the 2 bands with  $\Gamma_8$  symmetry at  $\vec{k} = 0$  (called HH and LH in **Figure 1**) which are the bands around the Fermi level and involved in the absorption. We also include the  $\Gamma_6$  valence band and a second conduction band. The introduction of this effective high energy band in the conduction band is central to accurately describe the bending of the LH band occurring at large wave vector.

All the experimental data are then accounted from the variation with temperature and pressure of only 2 bulk band gap energies:  $E_g(T,P)$  and  $E_{gC}(T,P)$ . As these gaps will be affected by pressure and temperature, the bands curvature will be modified (through the  $E_p$  and  $E_{pC}$  parameters coupling the bands), unveiling effect of pressure and temperature in presence of confinement (*i.e.* away from the  $\Gamma$  point). The strong confinement in the HgTe nanoparticles is modeled by the discretization of the  $\vec{k}$  wavevector in the reciprocal space, typically at  $k = \pm \frac{\pi}{L}$  along [001] where  $L$  is size of the nanoparticle, accounting for the contraction and thermal dilation of the nanoparticle with pressure and temperature respectively.

*Table 1 Values of the simple (3+1) and the 14-band model parameters used to describe the confinement, temperature and pressure dependences. n.a. means non applicable.*

Parameter	Simple band value	(3+1)-model	14-band model value	k.p	Reference
$E_g^0$	0.31 eV		0.375 eV		0.303 eV in Ref. 56
Simulated $E_g(k=0)$			0.305 eV		
$E_{gC}^0$	4 eV		2.5 eV		4.11 eV (QSGW) and 3.07 eV (LDA) in Ref. 52
Simulated $E_{gC}(k=0)$			2.78 eV		
$E_p$	25 eV		30 eV		Fitted, 24.8 eV in Ref. 56
$E_{pC}$	15 eV		20 eV		Fitted
$E_{pV}$	2.5 eV		10 eV		Fitted
$\partial E_g / \partial T$	from $E_g(T,P)$ , typically $\approx -0.61 \text{ meV.K}^{-1}$		from $E_g(T,P)$ plus a constant $0.1 \text{ meV.K}^{-1}$ , typically $\approx -0.51 \text{ meV.K}^{-1}$		See $E_g(T,P)$ formula in SI
$\partial E_{gC} / \partial T$	$-2.4 \text{ meV.K}^{-1}$		$-3 \text{ meV.K}^{-1}$		Fitted
$\partial E_g / \partial P$	$-85 \text{ meV/GPa}$		$-95 \text{ meV/GPa}$		Fitted
$\partial E_{gC} / \partial P$	$+220 \text{ meV/GPa}$		$+600 \text{ meV/GPa}$		Fitted

$a$ bulk HgTe lattice parameter	0.6465 nm		Ref 35 at 300K
$a_0^{NP}$ HgTe nanoparticle lattice parameter	0.6500 nm		From measurement Fig. S6c on NPLs at 300K
$B_0$ HgTe nanoparticle elasticity modulus	30 GPa		From measurement Fig. S6c on NPLs at 300K
$\partial B_0/\partial P$	-1.5		From measurement Fig. S6c on NPLs at 300K
$m_{LH}^*$	0.013 $m_0$	0.013 $m_0$	Output
$m_{HH}^*$	0.33 $m_0$	0.69 $m_0$	Output
Spin-orbit coupling $\Delta$	n.a.	0.5 eV	0.775 eV (hQSGW) in Ref. 52 1.08 eV in Ref. 52
Simulated spin-orbit coupling at $k=0$	n.a.	0.732 eV	
$\Delta_c$	n.a.	1 eV	1.06 eV (QSGW) 1.17 eV (LDA) in Ref. 52
Simulated spin-orbit coupling at $k=0$	n.a.	0.768 eV	
$\Delta'$	n.a.	-1.5 eV	Fitted

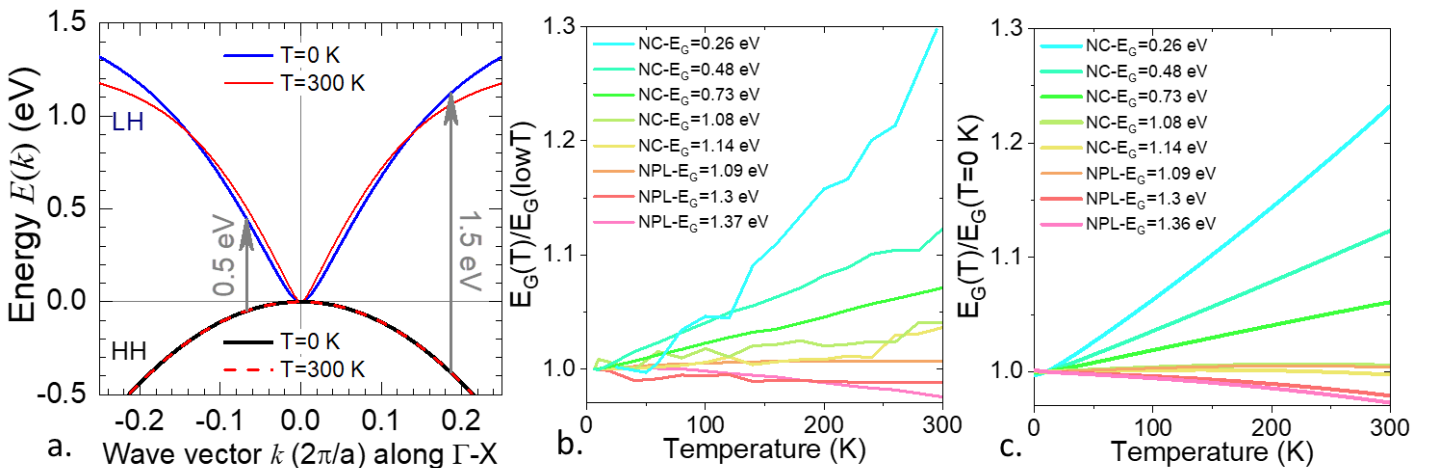
**Figure 5a** and **Figure 6a** show respectively the effects of temperature and pressure on the relation dispersion. When temperature or pressure increases, the band gaps  $E_g$  and  $E_{gC}$  are altered as depicted in Figure S11c. The effect on the LH band effective mass  $m_{LH}^*$ , defined at small  $k$ , is given by:

$$E_{LH}(k, T, P) \approx \left( 1 + \frac{E_p}{E_g(T, P)} - \frac{E_{pC}}{E_{gC}(T, P)} \right) \frac{\hbar^2 k^2}{2m_0} = \frac{\hbar^2 k^2}{2m_{LH}^*} \quad (1)$$

As seen in the equation (1), these variations lead to a competition between the two gaps  $E_g(T, P)$  and  $E_{gC}(T, P)$  in the alteration of  $E_{LH}(k)$  relation dispersion.

At low energy (*ie* small  $k$ , see the left arrow in the dispersion relation) the dispersion is ruled by the  $m_{LH}^*$  effective mass reduction. The energy blue-shifts as shown in **Figure 5a** for temperature, and **Figure 6a** for pressure as the result of the drastic reduction of  $E_g$  from 0.31 eV at 0 K down to 0.12 eV at room temperature.  $E_g$  being one order of magnitude smaller than  $E_{gC}$ , and  $\partial E_g/\partial T/E_g$  and  $\partial E_{gC}/\partial T/E_{gC}$  partial derivatives being comparable, the effective mass is essentially governed by  $E_g$  (*ie* the second conduction band plays no role in the vicinity of the  $\Gamma$  point). The same rule of thumb holds for pressure.

However, at high energies (*ie* high  $k$ , see the right arrow in the dispersion relation), the parabolic equation (1) ceases to be valid. When temperature increases, the second conduction band counteracts  $m_{LH}^*$  reduction. As the result of the shrinking of  $E_{gC}$ , the second conduction band becomes closer to the LH band and repeals it downwards as seen in **Figure 5a**. In other words, the second conduction band is responsible for the sublinear relation dispersion observed at large  $k$  and leads to an inversion of the confinement energy variation with a simulated red-shift of the  $E_G$  confinement energy.

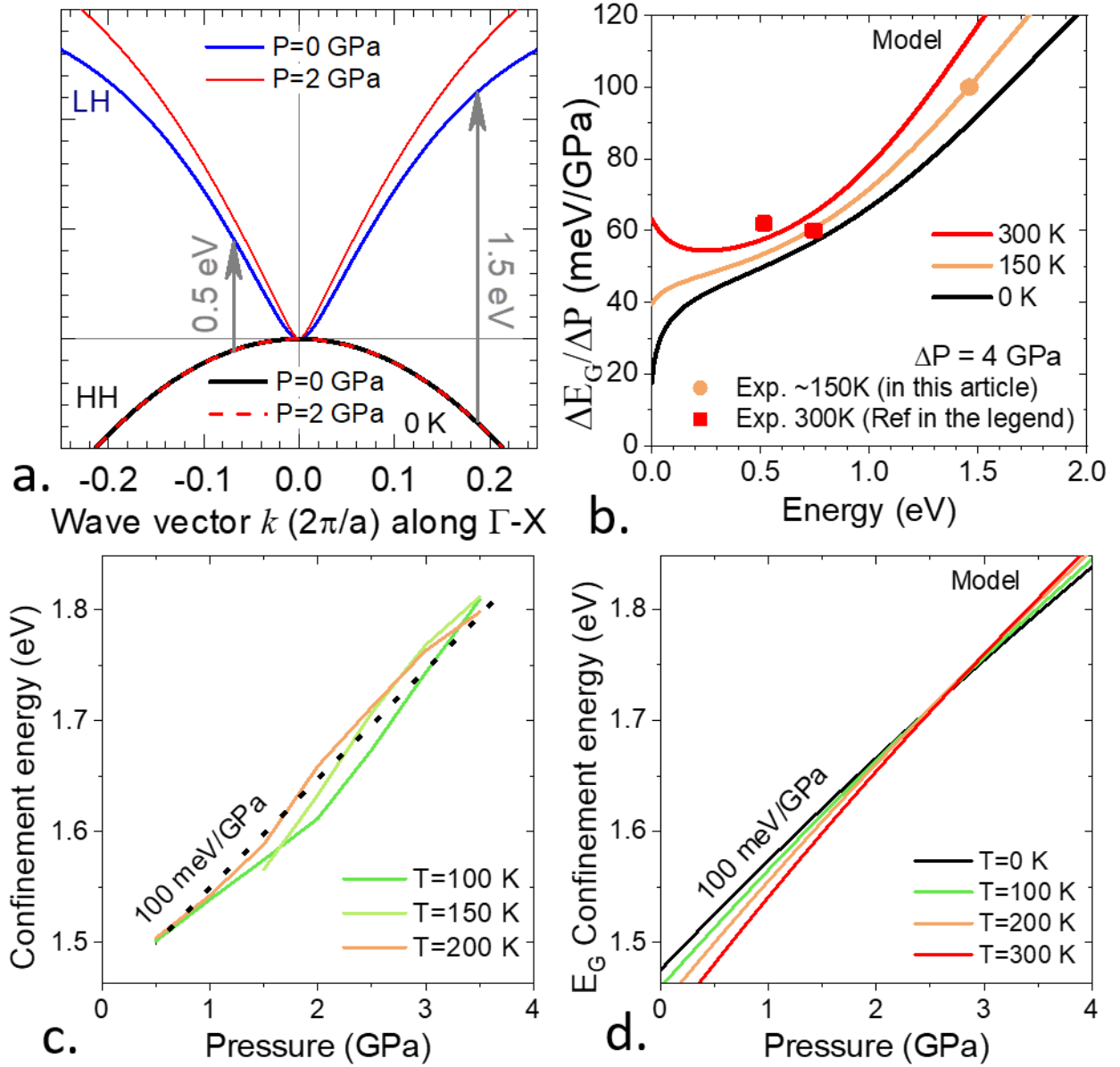


**Figure 5 Modelling the temperature effect of the dispersion relation.** a. Effect of temperature on  $E(k)$  when the two bands gaps  $E_g$  and  $E_{gC}$  varies. The gray arrows corresponds to typical confinement energies for the NPL (1.5 eV) and for NCs (0.5 eV). b. Experimental normalized shift of the band gap as a function of temperature for different 10 K sizes of HgTe NCs and 3 ML HgTe NPLs with different surface chemistry. c. Simulated normalized shift of the confinement energy  $E_G$  as a function of temperature for different sizes of HgTe NCs and 3 ML HgTe NPLs with different surface chemistry.

Using the (3+1) band model and  $-2.4 \text{ meV.K}^{-1}$  value for  $\frac{\partial E_{gC}}{\partial T}$ , the experimental temperature dependence of the band gap (**Figure 5b**) is reproduced, as shown in **Figure 5c**. With low  $T=10\text{K}$ , the blue shift ( $E_G(T)/E_G(\text{low}T)>1$ ) at small energies is due to the shrinking of the  $E_g$  band gap. However, at large energies,  $E_{gC}$  also shrinks and the closer second conduction band pushes the first one downward, leading to a red-shift ( $E_G(T)/E_G(\text{low}T)<1$ ). This interpretation is further analyzed quantitatively by plotting the simulated slopes  $\frac{\Delta E_G}{\Delta T}$  as a function of the confinement energy of the particles, see the dashed green line in **Figure 3d**. We see that the model appears in fair agreement with the observed slopes over the whole range of experimental energies supporting the interpretation of the shrinking of the two gaps energies.

Regarding the pressure dependence, the observed energy shifts with hydrostatic pressure present an average slope  $\frac{\Delta E_G}{\Delta P} \simeq 100 \text{ meV/GPa}$  in the case of NPL (**Figure 4b**, **Figure 6c**, and S10) which has to be compared with the value of  $\approx 60 \text{ meV/GPa}$  for the nanocrystals<sup>45</sup>(**Figure 4c** and S13). Contrary to the case of temperature, the confinement only quantitatively affects the pressure dependence of the confinement energy. This result contrasts to CdSe where the size dependence of the  $\frac{\Delta E_G}{\Delta P}$  parameter has been found to be larger with an increase of  $\frac{\Delta E_G}{\Delta P}$  from 40 to 80 meV/GPa as the size changes by only a factor two.<sup>49</sup>

Under increasing pressure, the second conduction band goes away from the first one. Pressure simply pushes  $E_G$  confinement energy even higher, see **Figure 6a**. In this case, due to pressure, the second conduction band adds an additional contribution to the one of  $E_g$  to the observed blue-shift of the confinement energy  $E_G$ .



**Figure 6 Modelling the pressure effect of the dispersion relation** a. Effect of pressure on the conduction and valence bands. The gray arrows corresponds to typical confinement energies for the NPL (1.5 eV) and for NCs (0.5 eV). b. Simulated confinement energy variation over  $\Delta P=4$  GPa pressure difference for 3 temperatures. The light orange dot is the experimental measurement around 150 K. The two red squares are the measurements from Ref <sup>45</sup> at 300 K. c. Experimental confinement energy  $E_G$  shift as a function of an applied hydrostatic pressure for HgTe NPLs for three different temperatures. d. Simulated confinement energy  $E_G$  shifts as a function of an applied pressure for HgTe NPLs from 0K to 300K. The indicated slope corresponds to 0 K

Using  $-85$  meV/GPa and  $+220$  meV/GPa respectively for  $\partial E_g/\partial P$  and  $\partial E_{gC}/\partial P$ , the (3+1) bands model can reproduce the pressure dependence of the confinement energy of the NPL, see **Figure 6c** and d. Also note that using the temperature dependence determined previously, it is possible to reproduce the weak temperature dependence of the pressure potential  $dE_G/dP$  of the confinement energy in the case of NPL, see the four simulated temperatures in **Figure 6d**, while capturing the moderate confinement dependence as depicted in **Figure 6b** and S13a.

In all, though both temperature and pressure alter the bulk band gaps  $E_g$  and  $E_{gC}$ , only blue shifts are observed with increasing pressure, while increasing temperature also shows red-shift. We thus conclude that the unusual change of the optical band gap with temperature and its change of sign reflects the bulk material electronic structure and that phonon does not have to be invoked. <sup>44</sup>

## CONCLUSION

In conclusion, we have shown that the strong confinement in NPL causes a non-linear dispersion of high energy state energies in the electronic structure of HgTe nanoparticles. Measurements of the temperature dependence of the optical band gap appears as a direct probe of the dispersion relation. We found that a single parameter set in the (3+1) effective band model can account for all the observed features in energy, temperature, and pressure, providing that the relative proximity of the second conduction band of HgTe be included. We show that the second conduction band appears as a paramount ingredient for absorption energies above  $\approx 0.5$ – $1$  eV to describe realistically the energies and electronic states of strongly confined HgTe nanoparticles.

## ASSOCIATED CONTENT

### Supporting information

The Supporting Information is available free of charge at

Supporting Information include details about material synthesis, material characterization, X-ray diffraction measurement under pressure, raw spectrum about material temperature and pressure dependence and additional information regarding the (3+1)-band **k.p**-like and 14 band **k.p** models.

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## TOC graphic

