



HAL
open science

4He glass phase: A model for liquid elements

Robert Tournier, Jacques Bossy

► **To cite this version:**

Robert Tournier, Jacques Bossy. 4He glass phase: A model for liquid elements. Chemical Physics Letters, 2016, 658, pp.282-286. 10.1016/J.cplett.2016.06.070 . hal-02936790

HAL Id: hal-02936790

<https://hal.science/hal-02936790>

Submitted on 2 Jun 2021

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



Research paper

⁴He glass phase: A model for liquid elements

Robert F. Tournier*, Jacques Bossy

Univ. Grenoble Alpes, Inst. NEEL, F-38042 Grenoble Cedex 9, France
 CNRS, Inst. NEEL, F-38042 Grenoble Cedex 9, France

ARTICLE INFO

Article history:

Received 9 May 2016

In final form 24 June 2016

Available online 25 June 2016

Keywords:

Liquid elements

Confined helium

He-4 glass phase

First-order liquid-glass transition

ABSTRACT

The specific heat of liquid helium confined under pressure in nanoporous material and the formation, in these conditions, of a glass phase accompanied by latent heat are known. These properties are in good agreement with a recent model predicting, in liquid elements, the formation of ultrastable glass having universal thermodynamic properties. The third law of thermodynamics involves that the specific heat decreases at low temperatures and consequently the effective transition temperature of the glass increases up to the temperature where the frozen enthalpy becomes equal to the predicted value. The glass residual entropy is about 23.6% of the melting entropy.

© 2016 Elsevier B.V. All rights reserved.

1. Introduction

The solid-liquid transformation of bulk helium depends on the pressure p and temperature T [1–6]. The melting entropy is determined from the Clapeyron relation knowing the volume and melting temperature changes associated with the solid-to-liquid transformation under a pressure p [2]. The phase diagram is deeply modified when liquid helium is confined in 25 Å mean diameter nano-pore media under pressures p where $3.58 \leq p \leq 5.27$ MPa [7]. Early studies of these involved specific heat anomaly measurements and they were viewed as a consequence of the formation of localized Bose-Einstein condensates on nanometer length scales analogous to a solid [7]. The glass phase has since been discovered using measurements of the static structure factor, $S(Q)$, of helium confined in the porous medium MCM-41 with pore diameter 47 ± 1.5 Å. A similar amorphous $S(Q)$ was also observed in 34 Å Gelsil [8]. The presence of an amorphous phase has been confirmed at higher pressures using porous Vycor glass [9]. In this work we consider that the supercooled liquid far below the melting temperature T_m is condensed in a glass phase accompanied by an exothermic latent heat associated with a first-order transition.

Recent work reviews earlier findings of glass formation in pure metals of small size and thickness [10–21]. It is clear that there is a need for a fundamental understanding of the resistance to crystallization of these glasses [21]. A recent model predicting the thermodynamic properties of the ultra-stable vitreous phase of liquid elements is used for that purpose [22]. Liquid helium at high

pressures is a normal liquid having properties analogous to those of liquid elements with much higher melting temperatures. Nevertheless, its zero point enthalpy H_0 is far from negligible at low temperatures [1,23]. The influence of the enthalpy change ΔH_0 has to be evaluated as a function of pressure to confirm that vitreous helium can be compared to other glasses. In addition the proximity to absolute zero and the third law of thermodynamics reduces the values of the specific heat of helium below and around the glass transition temperature T_g . Such influence does not occur in other liquids. A recent conference has been devoted to glass and entropy [24] pointing out a theoretical need to clarify whether residual entropy exists in the glass state at low temperatures [25]. In this work the model previously used for liquid elements having higher melting temperatures [22] is applied to these problems.

2. The model and its application to ⁴He under pressure

A complementary negative contribution $-v \times \Delta p = -v \times \varepsilon_{ls} \times \Delta H_m / V_m$ depending linearly on $\theta^2 = (T - T_m)^2 / T_m^2$, has been added to the classical Gibbs free energy change ΔG_{1s} for liquid-to-solid transformation. ΔH_m is the melting enthalpy per mole, ε_{ls} a fraction of ΔH_m , V_m the molar volume, Δp the complementary Laplace pressure and $v = 4\pi/3 \times R^3$ the solid nucleus volume of radius R . The new ΔG_{1s} per mole is given by (1):

$$\Delta G_{1s} = 4\pi R^3 3^{-1} \Delta H_m \times (\theta - \varepsilon_{ls}) + 4\pi R^2 (1 + \varepsilon_{ls}) \sigma_{1s} \Delta H_m. \quad (1)$$

ΔG_{1s} is associated with solid nucleus formation in a melt; ε_{ls} is the critical enthalpy saving coefficient, $(1 + \varepsilon_{ls}) \times \sigma_{1s}$ the new surface energy and σ_{1s} the classical surface energy for $\varepsilon_{ls} = 0$ [26]. This complementary enthalpy explains the presence of

* Corresponding author.

E-mail address: robert.tournier@creta.cnrs.fr (R.F. Tournier).

intrinsic long-lived metastable nuclei surviving above T_m and disappearing as the applied superheating rate increases [27,28]. Crystallization and melting are initiated by the formation of these solid or liquid growth nuclei accompanied by a volume change that is expected to obey Lindemann's rule in pure liquid elements [29]. Lindemann's description shows that the ratio of the mean square root ΔR of thermal vibrations and the interatomic distance R is a universal constant δ_{ls} at the melting temperature T_m .

The complement $\varepsilon_{ls0} \times \Delta H_m$ associated with the growth nucleus formation at the melting temperature T_m ($\theta = 0$) has been determined for many pure liquid elements and glass-forming melts. The coefficient $\varepsilon_{ls0} = 0.217$ is the same for many pure liquid elements [30] but is much larger than 1 and smaller than 2 in many glass-forming melts [26]. Lindemann's constant $\delta_{ls} = 0.103$ is directly deduced from $\varepsilon_{ls0} = 0.217$ when the classical Gibbs free energy change is extended to include this new contribution [22].

The glass state has been described as a thermodynamic equilibrium between crystal and liquid states [26]. The glass transition is induced by an enthalpy change. The critical enthalpy saving coefficients $\varepsilon_{ls}(\theta)$ and $\varepsilon_{gs}(\theta)$ for the formation of a solid growth nucleus at the reduced temperature θ in the liquid state and in the glass state are given by (2) and (3) where θ_{0m} and θ_{0g} are the reduced temperatures at which the enthalpy saving becomes zero and T_K is the Kauzmann temperature:

$$\varepsilon_{ls}(\theta) = \varepsilon_{ls0}(1 - \theta^2 \times \theta_{0m}^{-2}). \quad (2)$$

$$\varepsilon_{gs}(\theta) = \varepsilon_{gs0}(1 - \theta^2 \times \theta_{0g}^{-2}) \quad \text{for } T_K \leq T \leq T_g. \quad (3)$$

The enthalpy change per mole below T_g from the glass to the liquid is $[\varepsilon_{ls}(\theta) - \varepsilon_{gs}(\theta)] \times \Delta H_m$. The frozen enthalpy below T_g has a maximum value equal to $(\varepsilon_{ls0} - \varepsilon_{gs0}) \times \Delta H_m$ in strong glasses. Following cooling below T_g it takes centuries for the glass to become fully relaxed at the Kauzmann temperature, (point at which the frozen enthalpy is greatest), despite an easy recovery at T_g during heating. Therefore it is time-consuming to attain the thermodynamic equilibrium from the glass phase by relaxation. Various microscopic models prove the existence of a phase transition at T_g [31–40]. Here, we only use thermodynamic relations without considering the microscopic aspects of the liquid-glass transformation.

The universal value $\delta_{ls} = 0.103$ of the Lindemann constant obtained for pure metallic elements at their melting temperatures T_m has been used to build a model for their vitrification [22]. Its extension to liquid helium is proposed. The Gibbs free energy change below T_g cannot include any variation in structural relaxation enthalpy because δ_{ls} cannot be lowered below its minimum value. The enthalpy difference per mole is characterized by $\theta_{0m} = -2/3$ ($T_{0m} = T_m/3$), $\theta_{0g} = -1$ ($T_{0g} = 0$), $\varepsilon_{ls0} = \varepsilon_{gs0} = 0.217$, and is proportional to $\Delta\varepsilon_{ig}$ in (4) above $T_m/3$ and in (5) below $T_m/3$:

$$\Delta\varepsilon_{ig} \times \Delta H_m = -[0.217 \times 1.25 \times \theta^2] \Delta H_m. \quad (4)$$

$$\Delta\varepsilon_{ig} \times \Delta H_m = -[0.36165 \times (1 + \theta)] \Delta H_m. \quad (5)$$

The specific heat jump is maximum at $T = T_m/3$ and constant below $T_m/3$ because the contribution given by (2) becomes equal to zero below this temperature.

The glass transition temperature in liquid elements is $0.3777 \times T_m$ when the frozen enthalpy and the latent heat can be accommodated by the glass and liquid enthalpy [22]. In liquid helium, because of the specific heat reduction near the absolute zero, this is not possible. It has been shown that an enthalpy excess $\Delta\varepsilon \times \Delta H_m$ obtained after rapid quenching or vapor deposition increases $\varepsilon_{gs}(\theta)$ in the glass state [41] as shown in (6):

$$\varepsilon_{gs}(\theta) = \varepsilon_{gs0}(1 - \theta^2 \times \theta_{0g}^{-2}) + \Delta\varepsilon. \quad (6)$$

The reduced temperature $\theta_2 = [\varepsilon_{gs}(\theta_2) - 2]/3$ of homogeneous nucleation [29] which is equal to the glass transition θ_{g2} [26] is now a solution of the new quadratic Eq. (7):

$$\theta_{g2}^2 \times \varepsilon_{gs0} \times \theta_{0g}^{-2} + 3\theta_{g2} + 2 - \varepsilon_{gs0} - \Delta\varepsilon = 0. \quad (7)$$

The value of $\theta_2 = \theta_{g2}$ is given by (8) [40]:

$$\theta_{g2} = \left(-3 + \left[9 - 4(2 - \varepsilon_{gs0} - \Delta\varepsilon)\varepsilon_{gs0}/\theta_{0g}^2 \right]^{1/2} \right) \theta_{0g}^2 (2\varepsilon_{gs0})^{-1}. \quad (8)$$

The enthalpy excess coefficient $\Delta\varepsilon$ also exists in liquid helium-4 because of the specific heat reduction due to the third law of thermodynamics. In liquid elements described by (8) and $\Delta\varepsilon = 0$, we have $\theta_{g2} = \theta_g = -0.6224$ or $T_g = 0.3777 \times T_m$.

In spite of the specific heat reduction, the frozen enthalpy in the glass phase has to be equal to $0.105 \times \Delta H_m$ even if T_g is larger than $0.3777 \times T_m$. If the available glass enthalpy cannot attain $0.105 \times \Delta H_m$ at T_g , an excess enthalpy corresponding to $\Delta\varepsilon = 0.105$ would exist above this temperature during cooling. A new glass transition temperature given by (8) has to take place at $\theta_{g2} = -0.58398$. The new latent heat would be equal to $0.0925 \times \Delta H_m$ at this temperature instead of $0.105 \times \Delta H_m$ as given by (4). In fact, the effective glass transition still occurs above θ_{g2} up to the temperature T_{geff} where the frozen enthalpy becomes equal to $0.105 \times \Delta H_m$. The endothermic latent heat $0.0925 \times \Delta H_m$ begins to be recovered at the same temperature.

The specific heats of confined liquid helium at pressures of 3.58, 4.45, 4.89 and 5.27 MPa have been measured by [7] and are shown in Fig. 1.

The liquid specific heat in the cell does not have a strong dependence on pressure and is approximately proportional to the temperature T , so we use the approximation γT with $\gamma = 0.00593$. This introduces some error in the difference $\Delta C_p = C_{pl} - C_{pg}$ of liquid and glass specific heats as shown in Fig. 2 around 2 K.

The volume of confined helium in nano-pores is calculated using the linear specific heat measured at 2.28 MPa and is represented in Fig. 1a and c of [7] knowing that the volume open to bulk helium-4 is 38.5 mm^3 . This gives $V = 60.3 \text{ mm}^3$. The molar volume V_m under pressure is interpolated from known values [3] using (9):

$$V_m (\text{cm}^3 \text{ mole}^{-1}) = -2.246 \times \ln p + 23.176 \quad (9)$$

where p is the pressure in MPa.

The molar specific heat of liquid is calculated as a function of the mole fraction contained in the volume V and is equal to $\gamma T * V_m/V$. The values of γ and V_m/V are given in Table 1.

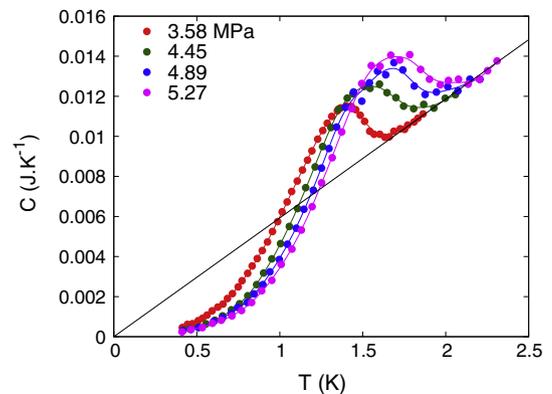


Fig. 1. Fig. 2 of Yamamoto et al. [7]. Copyright 2016 by the American Physical Society. Specific heat of confined helium-4 versus temperature at various pressures and added line $\gamma T = 0.00593 \times T$.

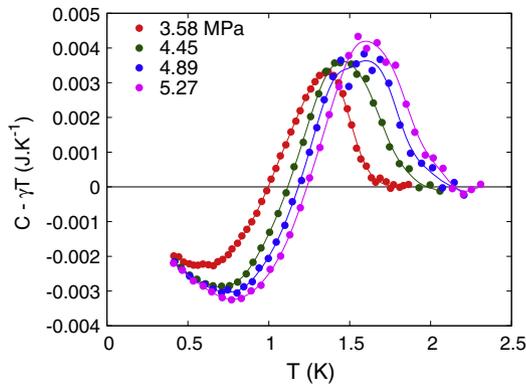


Fig. 2. Specific heat difference ($C - \gamma T$) of confined helium-4 versus temperature T (K) with $\gamma = 0.00593 \times T$. $H_2 \times V/V_m$ the area below zero and $H_1 \times V/V_m$ the area above zero.

Table 1
Application of the model to ^4He under pressure.

	p (MPa)	3.58	4.45	4.89	5.27
1	p (MPa)	3.58	4.45	4.89	5.27
2	V_m/V	359.3	350.3	346.6	343.5
3	T_m (K)	1.937	2.157	2.268	2.364
4	H_2 (J mole $^{-1}$)	0.52	0.696	0.766	0.843
5	H_1 (J mole $^{-1}$)	0.449	0.574	0.654	0.717
6	ΔV_m (cm 3)	1.353	1.3	1.29	1.27
7	V_m (cm 3) (liq)	21.66	21.12	20.9	20.71
8	T_K (K)	0.497	0.599	0.624	0.658
9	γ (J K $^{-2}$ mole $^{-1}$)	2.13	2.077	2.055	2.037
10	ΔS_m (J K $^{-1}$ mole $^{-1}$)	2.75	2.87	2.99	3.06
11	S_m (J K $^{-1}$ mole $^{-1}$)	4.72	4.99	5.13	5.25
12	s_R (J K $^{-1}$ mole $^{-1}$)	0.908	0.878	0.859	0.845
13	s_{RL} (J K $^{-1}$ mole $^{-1}$)	1.968	2.121	2.141	2.185
14	s_{Rg} (J K $^{-1}$ mole $^{-1}$)	1.06	1.23	1.21	1.24
15	θ_D (K)	24.67	26.1	26.83	27.46
16	T_{g2} (K) calc	0.806	0.897	0.944	0.983
17	T_{m0} (K) exp	0.86	0.93	0.92	0.988
18	ΔH_m (J mole $^{-1}$)	5.33	6.19	6.77	7.23
19	$H_2/\Delta H_m$	0.098	0.112	0.113	0.116
20	$H_1/\Delta H_m$	0.084	0.093	0.097	0.099
21	T_{geff} (K)	1.164	1.378	1.436	1.511
22	ΔH_0 (J mole $^{-1}$)	5.60	6.53	7.01	7.44
23	ΔU_0 (J mole $^{-1}$)	0.759	0.743	0.704	0.751

1-The pressure p in MPa, 2- V_m/V the molar volume divided by the confined helium volume, 3- T_m the melting temperature of helium crystals, 4- H_2 the frozen enthalpy in the glass phase 5- H_1 the enthalpy of fusion of the glass phase, 6- ΔV_m the volume change at T_m , 7- V_m the molar volume of the liquid, see (9), 8- S_m the total entropy change at T_m , see (15), 9- ΔS_m the melting entropy, see (12), 10- S_{RL} the residual entropy at zero K, see (17), 11- s_R , see (16), 12- s_{Rg} the residual entropy of the glass phase, 13- γT the liquid specific heat as observed in Fig. 1, 14- θ_D the Debye temperature of solid helium, see (11), 15- T_{g2} the calculated glass transition temperature, see (8) with $\Delta\epsilon = 0.105$, 16- T_{m0} the onset temperature of glass melting [7], 17- ΔH_m the melting enthalpy equal to $\Delta S_m \times T_m$, 18- $H_2/\Delta H_m$, 19- $H_1/\Delta H_m$, 20- T_{geff} the theoretical temperature where the glass enthalpy becomes equal to that of the liquid, 21- ΔH_0 the enthalpy at zero K, 22- ΔU_0 the internal energy at zero K.

The solid specific heat C_{vS} at constant volume depends on the Debye temperature according to (10) and the difference between C_{pS} and C_{vS} is considered to be negligible:

$$C_{vS} = 1952 \times T^3 \times \theta_D^{-3}. \quad (10)$$

The Debye temperature θ_D is interpolated from known values [3] using (11):

$$\theta_D(K) = 1.649 \times p(\text{MPa}) + 18.76. \quad (11)$$

The difference $C_p - C_v$ of liquid specific heats becomes smaller and smaller as the temperature decreases and at $T = 2.5$ K is only 3% of C_v [6]. In addition, the compressibility of the liquid and solid

are nearly equal when the solid is in equilibrium with the liquid [1]. At this point the specific heat values C_v are considered as being equal to C_p at low temperatures.

The specific heat minimum in Fig. 2 occurs at $T = T_m/3 = 0.646, 0.719, 0.756, \text{ and } 0.788$ K as predicted. The experimental minimum values: 0.810, 1.008, 1.051 and 1.107 are equal to or smaller than the theoretical ones: 0.995, 1.038, 1.081 and 1.107 J/K/mole. A consequence of the third law is that, instead of being constant and equal to $0.36165 \times \Delta S_m$ [22], the specific heat $\Delta C_p(T)$ falls sharply below $T_m/3$. The temperatures T_{g2} deduced from (7), using the enthalpy excess coefficient $\Delta\epsilon = 0.105$, are given in Table 1 and are close to the observed temperatures T_{m0} for the onset of glass phase melting [7] as shown in Table 1 and Fig. 3. The calculated temperatures T_{geff} are also closed to the freezing onset temperatures T_{FO} [42,43].

The frozen enthalpy H_2 is determined from the area below the zero line and above the curves in Fig. 2 using the multiplicative coefficients V_m/V given in Table 1 together with the melting temperature T_m , and the melting heat ΔH_m . The ratios $H_2/\Delta H_m$ are given in Table 1. Their mean value 0.11 is in good agreement with the predicted value of 0.105. The endothermic latent heat is still determined by the area below the zero line and below the specific heat curves in Fig. 2. The ratios $H_1/\Delta H_m$ for the melting enthalpy are given in Table 1. Their mean value of 0.0932 is in good agreement with the theoretical value of 0.0925. Other measurements indicate hysteresis between the melting temperature window observed by heating and that of freezing observed by cooling [42,43], confirming the first-order character of the glass transition. The radius reduction of nano-pores decreases the freezing onset temperature [43]. This phenomenon is due to an increase of the Laplace pressure reducing the applied pressure p in the nanopores, the melting temperature T_m and the glass transition temperature T_{g2} in quantitative agreement with the surface tension of ^4He .

The entropy ΔS deduced from the specific heat in Fig. 1 is represented as a function of temperature in Fig. 4 for the four pressures used.

The Kauzmann temperatures T_K are deduced by extrapolation of the straight lines in Fig. 4 down to zero entropy assuming that the solid entropy is negligible even if the experimental glass entropy is not fully negligible below these temperatures. The values of T_K are given in Table 1. The melting enthalpy ΔH_m is obtained using (12) and given in Table 1:

$$\Delta S_m = \gamma(T_m - T_K) - S_S \quad (12)$$

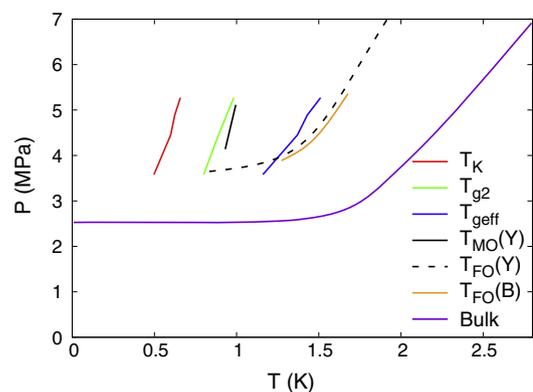


Fig. 3. The P-T phase diagram of ^4He in 2.5-nm and 2.4-nm nanoporous glasses determined in our present study and in [42,43]. The Kauzmann temperature is represented by the line T_K . The melting onset temperature T_{MO} is compared to the glass transition temperature T_{g2} and the freezing onset temperature T_{FO} to the effective glass transition temperature T_{geff} . The temperatures $T_{FO}(Y)$, $T_{MO}(Y)$ and $T_{FO}(B)$ have been measured by Yamamoto et al. and Bittner et al. respectively.

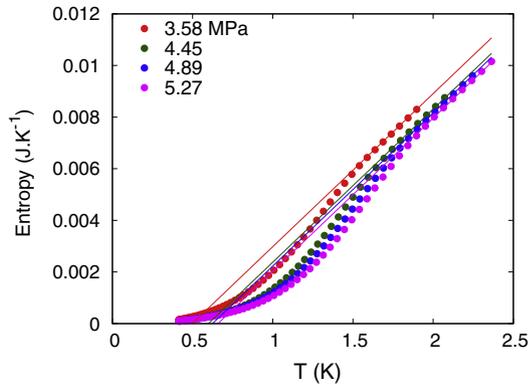


Fig. 4. The entropy ΔS of confined helium-4 versus the temperature T (K) for four pressures. The Kauzmann temperature T_K is extrapolated with the slope 0.00593 from the upper points.

where S_S is the solid entropy at T_m .

So our first conclusion is that the enthalpy excess of $0.105 \times \Delta H_m$ (equal to the frozen enthalpy) and the endothermic latent heat of $0.0925 \times \Delta H_m$, characterize the thermodynamic parameters at the glass transition temperature T_{g2} . These correspond to the experimental values T_{m0} given in Table 1 of the glass phase melting onset [7]. The endothermic latent heat is predicted to be fully recovered at the temperature T_{geff} given in Table 1 where the frozen enthalpy attains its theoretical value of $0.105 \times \Delta H_m$ as shown in Fig. 5 for $p = 3.58$ (line 2) and 5.27 MPa (line 3).

All the enthalpy values represented in Fig. 5 include the frozen enthalpy and the endothermic latent heat associated with the ultra-stable glass-to-liquid transformation. The increase of ΔH is determined using (13) from the Kauzmann temperature θ_K to $\theta = -2/3$ and (14) for $\theta \geq -2/3$:

$$\Delta H/\Delta H_m = 0.36165 \times (\theta - \theta_K) - 0.1975, \quad (13)$$

$$\Delta H/\Delta H_m = 0.217 \times 1.25 \times (\theta^2 - 4/9) + A, \quad (14)$$

where A is a constant equal to $\Delta H/\Delta H_m$ given by (13) for $\theta = -2/3$. Eqs. (13) and (14) are deduced from (4) and (5) including the frozen enthalpy and the latent heat. The enthalpy of ultra-stable glasses in other liquid elements are also represented in Fig. 5 line 1 [22] with demonstrating an exception of the 3rd law of thermodynamics.

3. Residual entropy and enthalpy at 0 K

Up to now we have used only the entropy ΔS and the enthalpy associated with the specific heat. The residual entropy and

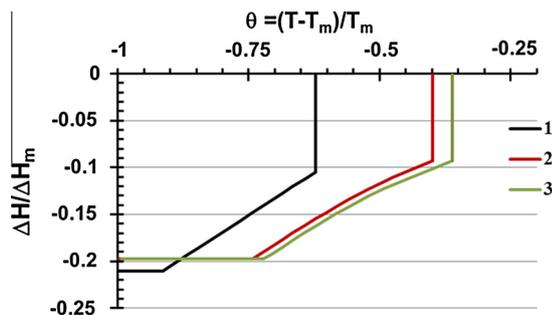


Fig. 5. Outline of the theoretical enthalpy ratio $\Delta H/\Delta H_m$ versus $\theta = (T - T_m)/T_m$. Line 2 for $p = 3.58$ MPa, $T_{geff} = 1.164$ K, $T_K = 0.497$ K and Line 3 for $p = 5.27$ MPa, $T_{geff} = 1.51$ K, $T_K = 0.658$ K. Line 1 for other liquid elements.

enthalpy at 0 K do not modify these as their derivatives are equal to zero. The amplitude of atomic vibrations at 0 K is large and cannot be neglected in liquid helium. Lindemann's rule introduces supplementary mean square amplitude of thermal vibrations to allow the melting of crystals. This phenomenon is associated with specific heat. The residual entropy and enthalpy at 0 K have a strong influence on the melting entropy S_m and on the melting enthalpy H_m . The question of the existence of residual entropy is an important theoretical problem in considering the nature of the glass phase [24]. The properties of vitrified helium help clarify it.

Knowing ΔV_m and dP/dT , the solid-liquid melting entropy differences $S_m = (S_L - S_S)$ in $\text{J K}^{-1} \text{mole}^{-1}$ have been measured or calculated from the Clapeyron relation [2,4,5]. They obey the relationship given in (15) in the range 1.772–2.5 K, where the melting entropy S_m and the melting temperature T_m are as given in Table 1:

$$S_m = S_L - S_S = 1.235 \times T_m + 2.33. \quad (15)$$

where S_S is the crystal entropy at T_m .

The liquid entropy is approximated by (16)

$$S_L = \gamma T + s_R, \quad (16)$$

As shown in Fig. 6 for $p = 3.58$ and 5.27 MPa, s_R is as given in Table 1.

The liquid entropy S_L is given by (17) as a function of the Kauzmann temperature T_K and of its residual entropy S_{RL} :

$$S_L = \gamma(T - T_K) + S_{RL}. \quad (17)$$

S_{RL} is equal to $\gamma T_K + s_R$ and to about 42% of S_m . The reduction of S_{RL} by the glass formation entropy $(0.0925 + 0.105) \times \Delta H_m/T_{geff}$ at 0 K leads to the residual entropy S_{Rg} of the glass given in Table 1. This experimental result demonstrates the existence at 0 K of residual entropy associated with structural disorder in glass equal to 23.6% of S_m and 56% of the residual entropy of the liquid at zero K [24,25]. The third law of thermodynamics predicts that the entropy change associated with a reversible transformation towards a thermodynamic equilibrium approaches zero as the temperature tends to zero. Residual entropy still remains in spite of the thermodynamic transition existence because the glass appears as an intermediate phase between liquid and crystal phases.

The total enthalpy change H_m during the transformation from solid to liquid at T_m is given by (18)

$$H_m = \gamma T_m^2 \times 2^{-1} + \Delta H_0 - H_S = T_m(S_L - S_S), \quad (18)$$

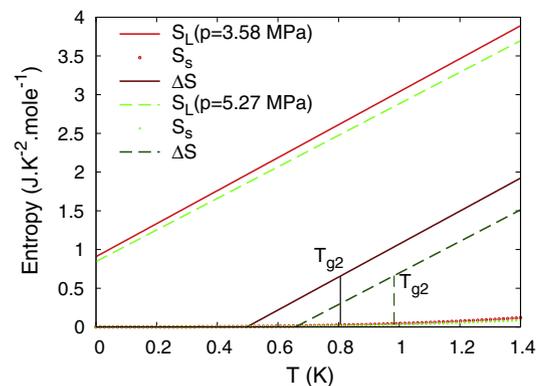


Fig. 6. Outline of liquid helium-4 entropy S_L and ΔS versus temperature T . The upper straight lines represent the total entropy $S_L(T)$ of liquids, including the residual entropy. The lower entropy lines $\Delta S(T)$ are deduced from the specific heat measurements. The broken lines correspond to $p = 5.27$ MPa and the continuous lines to $p = 3.58$ MPa. The points represent the solid entropies $S_S(T)$. The entropy changes ΔS at T_{g2} are also indicated.

where ΔH_0 is the difference of enthalpy at 0 K between the liquid and solid under the same pressure. ΔH_0 is due to the volume change ΔV_m . Values of ΔH_0 are given in Table 1 for the four pressures used. The mean ratio 0.56 J cm^{-3} of $\Delta U_0/\Delta V_m$ is obtained subtracting the contribution $p \times \Delta V_m$ in agreement with $\Delta U_0/\Delta V_m = 0.5 \text{ J cm}^{-3}$ of solid helium under pressure in the same range of molar volumes [1]. There is no enthalpy contribution associated with $v_{is} \times \Delta H_m$ tending to zero when the atom number n of superclusters increases [26].

The entropy change at T_{g2} in Fig. 6 is still too small to accommodate the endothermic latent heat and the frozen enthalpy. However it could be achieved at T_{geff} after a long time of isothermal relaxation. For example, the transformation at T_g of an ultra-stable glass of indomethacin in supercooled liquid has been observed after several hours of isothermal relaxation [44].

4. Conclusions

We have shown that the vitreous transition in liquid helium has a thermodynamic origin and is accompanied by a latent heat. There is no enthalpy relaxing below the glass transition down to the Kauzmann temperature because this enthalpy is delivered at the glass transition temperature and leads to a first-order transition. This helium glass is then ultra-stable. Our model of thermodynamic transition predicts the values of the latent heat, the frozen enthalpy, the specific heat minimum of $\Delta C_p(T)$ at $T = T_m/3$, and the glass transition temperature of liquid elements.

This strongly suggests that the growth nucleus formation is accompanied by a complementary enthalpy saving and that the classical nucleation equation has to be completed to be valid. In addition, we have confirmed that the glass entropy contains residual entropy associated with residual structural disorder at very low temperatures.

Acknowledgments

Our publication is based on specific heat measurements of K. Yamamoto, Y. Shibayama, and K. Shirahama [7]. The authors thank Prof. K. Shirahama from Keio University in Yokohama authorizing the reproduction of Fig. 2 and two anonymous reviewers for their help.

References

- [1] D.O. Edwards, R.C. Pandorf, *Phys. Rev.* 140 (1965) A816.
- [2] E.R. Grilly, *J. Low Temp. Phys.* 11 (1973) 33.
- [3] A. Driessen, E. van der Poll, I.F. Silvera, *Phys. Rev. B* 33 (1986) 3269.
- [4] W.H. Keesom, P.H. Keesom, *Physica* 2 (1935) 557.
- [5] E.R. Grilly, R.L. Mills, *Ann. Phys.* 18 (1962) 250.
- [6] R.D. Mc, Carty, *J. Phys. Chem.* 2 (1973) 923.
- [7] K. Yamamoto, Y. Shibayama, K. Shirahama, *Phys. Rev. Lett.* 100 (2008) 195301.
- [8] J. Bossy, T. Hansen, H.R. Glyde, *Phys. Rev. B* 81 (2010) 184507.
- [9] S. Bera, J. Maloney, N. Mulders, Z.G. Cheng, M.H.W. Chan, C.A. Burns, Z. Zhang, *Phys. Rev. B* 88 (2013) 054512.
- [10] L. Zhong, H. Wang, H. Sheng, Z. Zhang, S.X. Mao, *Nature* 512 (2014) 177.
- [11] W. Klement Jr., R.H. Willens, P. Duwez, *Nature* 187 (1960) 869.
- [12] H.A. Davies, J.B. Hull, *J. Mater. Sci.* 11 (1976) 215.
- [13] W. Buckel, R. Hirsch, *Z. Phys.* 138 (1954) 109.
- [14] K.H. Berhrndt, *J. Vac. Sci. Technol.* 7 (1970) 385.
- [15] D.M. Galenko, D.M. Herlach, *Mater. Sci. Eng. A* 34 (2007) 449.
- [16] S.R. Corriel, D. Turnbull, *Acta Metall.* 30 (1982) 2135.
- [17] M.J. Aziz, W.J. Boettinger, *Acta Metall. Mater.* 42 (1994) 527.
- [18] J.Q. Broughton, G.H. Gilmer, K.A. Jackson, *Phys. Rev. Lett.* 49 (1982) 1496.
- [19] Y.W. Kim, H.M. Lin, T.F. Kelly, *Acta Metall. Mater.* 37 (1989) 247.
- [20] Y.W. Kim, T.F. Kelly, *Acta Metall. Mater.* 39 (1991) 3237.
- [21] A.L. Greer, *Nat. Mater.* 14 (2015) 542.
- [22] R.F. Tournier, *Chem. Phys. Lett.* 651 (2016) 198.
- [23] A. Landesman II, *J. Phys. France, Colloq. C3* (10) (1970) 55.
- [24] L. Wondraczek, R. Conradt (Eds), *Proceedings of the international workshop on glass and entropy*, *J. Non-Cryst. Solids* 355 (2009) 581–758.
- [25] A. Takada, R. Conradt, P. Richet, *J. Non-Cryst. Solids* 429 (2015) 33.
- [26] R.F. Tournier, *Physica B* 454 (2014) 253.
- [27] R.F. Tournier, E. Beaunon, *Sci. Technol. Adv. Mater.* 10 (2009) 014501.
- [28] R.F. Tournier, *Metals* 4 (2014) 359.
- [29] F.A. Lindemann, *Phys. Z.* 11 (1911) 609.
- [30] R.F. Tournier, *Physica B* 392 (2007) 79.
- [31] T.R. Kirkpatrick, D. Thirumalai, P.G. Wolynes, *Phys. Rev. A* 40 (1989) 1045.
- [32] J. Souletie, *J. Phys. France* 51 (1990) 883.
- [33] L. Berthier, G. Biroli, J. Bouchaud, L. Cipelletti, D.E. Masri, D. L'Hôte, F. Ladieu, M. Pierno, *Science* 310 (2005) 1797.
- [34] C.A. Angell, in: P.G. Wolynes, V. Lubchenko (Eds.), *Structural Glasses and Supercooled Liquids*, Wiley and Sons, Hoboken, New Jersey, 2012, p. 237.
- [35] M.I. Ojovan, W.E. Lee, *J. Non-Cryst. Solids* 356 (2010) 2534.
- [36] M.I. Ojovan, K.P. Travis, R.J. Hand, *J. Phys. Condens. Matter.* 19 (2007) 415107.
- [37] J.F. Stanzione III, K.E. Strawhecker, R.P. Wool, *J. Non-Cryst. Solids* 357 (2011) 311.
- [38] R.P. Wool, *J. Polym. Sci. B* 46 (2008) 2765.
- [39] R.P. Wool, A. Campanella, *J. Polym. Phys. Part B: Polym. Phys.* 47 (2009) 2578.
- [40] D.S. Sanditov, *J. Non-Cryst. Solids* 385 (2014) 148.
- [41] R.F. Tournier, *Chem. Phys. Lett.* 641 (2015) 9.
- [42] K. Yamamoto, Y. Shibayama, K. Shirahama, *J. Phys. Soc. Jpn.* 77 (2008) 013601.
- [43] D.N. Bittner, E.D. Adams, *J. Low Temp. Phys.* 97 (1994) 519.
- [44] K.L. Kearns, K.R. Whitaker, M.D. Ediger, H. Huth, C. Schick, *J. Chem. Phys.* 133 (2010) 014702.