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Particular H₂O dissolution mechanism in iron-rich melt: application to martian basaltic melt genesis.

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Abstract

Martian basalts are different from Earth by their iron-rich abundance with 18 wt % FeO_{tot} in average for Mars upper crust. The H₂O behavior in this atypical melt composition is not well understood. We have synthesized H₂O-bearing martian basaltic glasses (> 15.5 wt % FeOⁱⁿⁱ) under high pressures (0.5-1.5 GPa) and temperatures (> 1500 °C) conditions. We used Raman spectra to investigate the effect of H₂O as well as the high FeO_{tot} content on the molecular structure of Fe-rich glasses.

Increasing Fe content appears to inhibit the dissolution of H₂O in the melt. We observed the formation of Free OH groups at 3660 cm⁻¹ at relatively low H₂O content (1 wt %) in Fe-rich glasses whereas it only appears at high H₂O content (~6.5 wt %) for Fe-poor glasses. We suggest that the Free OH are bonded to Fe²⁺ cations in the melt forming isolating clusters of Fe(OH)₂. Such configurations is suspected to induce an increase in the melt polymerization;

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3 24 however, we did not clearly observe it and further investigation is requested. The major
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5 25 implication of these results is the possibility to form an immiscible hydrated Fe-rich phase that
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8 26 will favor formation of Fe-oxides at Mars surface.
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10 27 **1. Introduction**

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13 28 Many evidences of features related to liquid flows and aqueous phases at the surface of Mars
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16 29 led to the conclusion that water was present in its youth (e.g., [1-3]). To stabilize liquid water
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18 30 at the surface of Mars, an atmosphere thicker and denser than the present one is needed [4-7].
19
20 31 Such primitive atmosphere would either require: 1) volatiles from large extra-terrestrial impacts
21
22
23 32 or 2) volatiles degassing from the interior of the planet through volcanism. Indeed, Mars went
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25 33 through an intense volcanic activity during the Noachian (> 3.7 Ga), resulting in large provinces
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27 34 of igneous basaltic rocks covering the surface [8-12].

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29
30 35 The quantification of the volatile fluxes from a planetary mantle to its atmosphere depends
31
32 36 chiefly on the volatile content at the source and their respective solubility in the magmatic
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34 37 system. During volcanic eruptions, volatile species such as CO₂, H₂O and SO₂ (for the most
35
36 38 important) are degassed into the atmosphere. The behavior of volatile species has been widely
37
38 39 investigated for the different terrestrial melt compositions: 1) CO₂, H₂O and SO₄²⁻ for basaltic
39
40 40 melt compositions (e.g., [13-16]); 2) CO₂ and H₂O for silica-poor melts (e.g., [17-20]). While
41
42 41 studies have focused their work on the volatile solubility in Earth magmatic systems, only few
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44 42 of them have focused on the solubility of volatile elements in martian magmas to constrain the
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46 43 volatile geochemical cycle of Mars. From geochemical modeling and meteorite analyses,
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48 44 estimations of the H₂O in a martian melt have been proposed although values are very scattered
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50 45 from thousands of ppm to 2 wt % H₂O [21 - 23].
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56 46 Martian interior can be characterized by a wide range of redox conditions: from IW (i.e., Iron
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58 47 Wüstite) to above the QFM buffer (i.e., Quart-Fayalite-Magnetite) [24-26]. Experimental
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3 48 studies on martian melt were mainly focused on the C behavior under oxidizing and reducing
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5 49 conditions (e.g., [27-30]). Studies of the crystallization sequence for martian melt analogues in
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7 50 presence of H₂O have been proposed [31-33] and the water content determined. However, the
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9 51 H₂O quantification established in those studies cannot reflect the solubility of H₂O in an iron-
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11 52 rich melt. Up to now, there is no experimental study scrutinizing the H₂O dissolution
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13 53 mechanisms for iron-rich glass.
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16
17 54 The Raman spectroscopy is a dedicated tool used to investigate qualitatively and quantitatively
18
19 55 both crystallized and amorphous materials. Spectroscopic studies were achieved to establish
20
21 56 calibrations from Raman spectra in crystallized materials (e.g., [34-36]). Many studies on
22
23 57 amorphous materials based on the investigation of CO₂ and S behaviors in a silicate melt have
24
25 58 led to subsequent calibrations (e.g. [37-39]). Calibrations to quantify the H₂O in melts by
26
27 59 Raman spectroscopy have been proposed by [40-43]. Nonetheless, the behavior of volatile
28
29 60 species dissolved in an iron-rich melt prevailing on Mars has been merely addressed by Raman
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31 61 spectroscopy.
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36 62 In the present study we investigated the behavior of H₂O in an iron-rich melt. We synthesized
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38 63 Fe-rich basalt glasses (15.5 and 23.5 wt % FeOⁱⁿⁱ) under high-pressure conditions (0.5-1.5 GPa)
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40 64 in equilibrium with a H₂O fluid phase (up to 3 wt % starting H₂O content). The results obtained
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42 65 on Fe-rich glasses are compared to a Fe-poor terrestrial basalt analogue (~ 9 wt %). Experiments
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44 66 at high H₂O content (> 6 wt %) have also been performed on the terrestrial analogue
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46 67 composition. The silicate structure of the obtained glasses has been characterized by Raman
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48 68 spectroscopy. Fe²⁺ and FeO_{tot} (total iron) wet chemistry analyses have been performed in order
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50 69 to discuss the interplay of the Fe coordination state and the H₂O dissolution mechanism. The
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52 70 growth of Free OH molecular groups is observed at very low H₂O content in Fe-rich basaltic
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54 71 glasses and the possible impact of these molecular clusters on element transportation within
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56 72 Mars interiors is discussed.
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2. Methods

2.1. Starting Material

Investigated synthetic compositions were prepared in the $\text{SiO}_2\text{-Al}_2\text{O}_3\text{-FeO-MgO-CaO-Na}_2\text{O}$ system from a mixture of oxides and carbonate (Na_2CO_3). The Fe-rich synthetic basaltic compositions were made according to basaltic rocks identified at the surface of Mars by the Spirit and Curiosity rovers at Gusev and Gale craters, respectively [7, 12]. The two Fe-rich synthetic basaltic compositions were prepared with 15.5 and 23.5 wt % FeO and were referred as GC and CL respectively. A natural Popping Rock 2 π D43 glass (i.e., PR, ~9 wt % FeO_{tot}) has also been studied to investigate the H_2O behavior in a terrestrial analogue. The glass chemical compositions are reported in Table 1.

The natural Popping Rock and the synthetic compositions were crushed in an agate mortar for homogenization. Some powders were decarbonated at 850°C for at least 4 hours in a 1 atm furnace (i.e., GCD and PR samples). The wet chemistry results of the starting materials showed that iron is mostly under Fe^{3+} state, $\text{Fe}^{2+} / \sum\text{Fe}$ at 0.03 and 0.02 for GC and CL starting powders, respectively (see Table 1). Prior to the high pressure experiments, the Fe is almost fully oxidized.

The initial degree of polymerization representing the silicate network tetrahedral units interconnection, calculated according to Brooker et al. [18], is represented by the NBO/T parameter (i.e., Non-Bridging Oxygen per Tetrahedron) in the Table 1. For this calculation, Fe is considered as Fe^{3+} in the mixtures and therefore as a network former cation like Si^{4+} or Al^{3+} . The studied compositions have comparable NBO/T with 0.38, 0.56 and 0.33, for PR, GC and CL, respectively; which represents slightly depolymerized composition.

2.2. High pressure experiments

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3 96 Pure H₂O was added with a micro-syringe in several proportions (1.0 to 6.5 wt %) at the bottom
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5 97 of a Pt capsule before loading the starting powder. The capsules were welded shut and weighed
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7 98 to check for mass loss before and after the experiment. No evidence for weight loss was
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9 99 observed from the recovered capsules.
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13 100 The experiments were carried out using piston-cylinder apparatus in the pressure range of 0.5-
14
15 101 1.5 GPa (~80 – 120 km in depth for Mars) and 1500 – 1600 °C for temperature range. A ¾ inch
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17 102 talc-pyrex assembly was used. Previous work [44] showed that using talc-pyrex assemblies
18
19 103 apply intrinsic fO_2 conditions close to QFM+1 buffer. Temperature was controlled with a B-
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21 104 type thermocouple (PtRh₆—PtRh₃₀) with an accuracy of ±5°C. The accuracy on pressure was
22
23 105 ±10 % and a 10 % friction correction was applied during the experiment. The run duration
24
25 106 varied between 1 and 6 hours. Quenching was achieved by cutting off the power and the quench
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27 107 rate was measured and estimated to be above 80°C/s.
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32 108 2.3. Analytical methods

33 34 35 109 2.3.1. Spectroscopic methods

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38 110 The Fourier Transform Infrared spectroscopy (i.e., FTIR) was used for several samples.
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40 111 The acquisitions were acquired with a IR microscope attached to a Nicolet FTIR 5700, with a
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42 112 X15 Cassegrain objective. The analysis conditions were: a white light source, an InSb detector
43
44 113 and CaF₂ beamsplitter. Acquisitions were performed in the near IR area between 3000 – 6000
45
46 114 cm⁻¹ with a spectral resolution of 4 cm⁻¹. Each sample is an accumulation of 100 repetitive
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48 115 scans. The samples were not polished due to their small sizes. Measurements of the thickness
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50 116 of each glass chips were realized with a digitometer (with an accuracy of ± 1 µm) to link the
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52 117 absorption to the H₂O content using the Beer-Lambert approximation [13, 14].
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3 119 We used a Jobin-Yvon Labram HR800 Raman spectrometer equipped with a 532 nm
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5 120 solid-state laser at the Laboratory of Planetology and Geodynamic (LPG, Nantes University,
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7 121 France). The output power was set at 50 mW to prevent sample damage. We used a grating of
8
9 122 1200 grooves/mm corresponding to a spectral resolution of $\sim 0.8 \text{ cm}^{-1}$. The acquisition time was
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11 123 typically 30 sec with 6 repetitive scans. Spectra were acquired in a non-confocal mode with a
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13 124 slit aperture of $200 \mu\text{m}$ and a spot size estimated $\sim 1 \mu\text{m}$ in diameter. A X50 Olympus objective
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15 125 was used. At least 5 spectra were acquired on each sample to investigate the homogeneity of
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17 126 the recovered sample with respect to the dissolved H_2O . The studied spectral range was between
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19 127 200 and 1250 cm^{-1} and between 2600 and 4000 cm^{-1} , covering the area of the silicate network
20
21 128 and the H_2O vibrations, respectively.

22
23
24 129 H_2O dissolved in silicate glasses exhibits Raman vibrations corresponding to the vibrations
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26 130 stretch and bending (ν_1 and ν_2) of $\text{H}_2\text{O}_{\text{mol}}$ and OH^- represented by an envelope with a peak
27
28 131 maximum at $\sim 3550 \text{ cm}^{-1}$. A shoulder at 3660 cm^{-1} can be observed and is attributed to Free OH
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30 132 groups vibrations [45-47]. We tested several calibrations to quantify H_2O in glasses: [40, 41,
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32 133 43]; using the ratio between the HF area (i.e., high frequency) band in the $800 - 1200 \text{ cm}^{-1}$ and
33
34 134 the H_2O band at 3550 cm^{-1} [40, 43] and the intensity of the same band [41]. Although, these
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36 135 calibrations were not fully tested to quantify H_2O in Fe-rich basaltic glasses; at a given
37
38 136 composition, the H_2O content can be compared from one sample to another.

137 2.3.2. SEM and EPMA analysis

138 We performed Electron Probe Micro-Analysis (EPMA) and Secondary Electron Microscopy
139 (SEM) equipped with Energy Dispersive Spectroscopy to determine the major elements
140 compositions of our glasses. The SEM analyses were performed with a JEOL JSM 5800LV
141 with a beam current at 5 nA and a voltage at 15 kV, at Institut des Matériaux Jean Rouxel of
142 Nantes (IMN, France).

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3 143 The EPMA analyses were made using a Cameca SX 100 at IFREMER of Brest (France). A spot
4
5 144 size of 10 μm and a beam current at 6 nA were chosen for the glass components. The acquisition
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7 145 time was 10 s on the sample and 5 s on the background. The standards chosen for our analysis
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9
10 146 were: wollastonite (Si, Ca), corindon (Al), TiMn oxide, andradite (Fe), NiO, albite (Na),
11
12 147 forsterite (Mg), apatite (P), orthose (K), Cr_2O_3 oxide. The associated error is on the order of 2
13
14 148 % in relative to the measured value (see Table 2).

15
16
17 149 We can see in the Table 2, that the PR-3H-6h sample has experienced a strong Fe loss (2.7 wt
18
19 150 % FeO_{tot}) during the experiment considering that the FeO_{tot} in the PR composition is ~ 9 wt %.
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21 151 This Fe-loss has previously been observed [48] and it is known from experimental study that
22
23 152 the Fe can form an alloy with the Pt capsule. Even knowing the loss of Fe during experiments,
24
25 153 the Fe-enrichment of the capsules has not proven its efficiency either. Then the Fe-loss cannot
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27 154 be accurately controlled in high pressure and temperature experiments. However, all the
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29 155 experiments have experienced depletion with respect to Fe. The following results are then
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31 156 considering the FeO_{tot} after the experiments.
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158 2.3.3. Colorimetric determination of the $\text{Fe}^{2+} / \sum\text{Fe}$ ratio

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43 159 The $\text{Fe}^{2+} / \sum\text{Fe}$ ratios were measured using the colorimetric method of Wilson [49] recently
44
45 160 modified by Schuessler et al. [50]. About 3 mg of samples were weighted in 7 mL crystal
46
47 161 polypropylene beakers with a precision of 10^{-3} mg and dissolved for 3 days at room temperature
48
49 162 in 1 mL of concentrated HF and 1 mL of 0.139 mol. L^{-1} ammonium vanadate (V^{5+}). As the
50
51 163 digestion progresses in such low pH solutions, the Fe^{2+} released from the silicate matrix reacts
52
53 164 immediately with the V^{5+} to form Fe^{3+} and V^{4+} . This reaction, which prevents any accidental
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55 165 and irreversible oxidation of Fe^{2+} into Fe^{3+} [49], may be reversed if the pH of the solution is
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57 166 raised up to 5. Once the digestions were completed, 5 mL of beryllium sulfate solution (500
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3 167 g.L⁻¹) was added to the beakers in order to neutralize the excess of HF and breakdown all
4
5 168 insoluble fluorides. The final solutions for FeO colorimetric measurement were prepared in
6
7 169 volumetric flasks filled with 10 mL of ammonium acetate buffer solution at 500 g.mol⁻¹ (to
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9 170 raises pH up to 5), 5 mL of 2:2'-dipyridil solution as the ferrous colorimetric reagent and ultra-
10
11 171 pure water to filled up to 100 mL. The FeO_{tot.} analytical solutions were prepared by adding
12
13 172 between 5 to 10 mg of hydroxylamine hydrochloride to an aliquot of the ferrous analytical
14
15 173 solutions [50]. This strong reducing agent ensure the quantitative reduction of all remaining
16
17 174 Fe³⁺ into Fe²⁺. The colorimetric measurements were made with the UV/VIS spectrophotometer
18
19 175 CARY UV500 (Varian). The optical density measurements were done at 525 nm where the
20
21 176 ferrous 2.2' -dipyridil complex absorption is maximum [49]. The peaks heights were measured
22
23 177 relative to the baseline at 700 nm and the Fe²⁺ / \sum Fe ratios were determined by dividing the
24
25 178 absorbance ratios corrected for the procedural blanks. The results of the Fe²⁺ / \sum Fe
26
27 179 measurements on the international reference material JB-2 (Fe²⁺ / \sum Fe = 0.78 \pm 0.005, N=19)
28
29 180 are in good agreement with the recommended values (i.e., 0.78, [51]).
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3. Results

3.1. Raman spectra

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45 184 The Raman spectra obtained on H₂O-bearing glasses synthesized at 1 GPa are represented in
46
47 185 Fig. 1A. The Raman spectra were treated with Labspec 6© software and a 5-degree polynomial
48
49 186 fit which goes through entrenched points (200 – 250 cm⁻¹, ~600, ~800 and > 1150 cm⁻¹) to
50
51 187 subtract the baseline is used [41]. The spectra are arranged according to their FeO_{tot.} We
52
53 188 distinguish two parts: the HF region 800 – 1200 cm⁻¹ and the LF region 200 – 600 cm⁻¹. The
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55 189 800 – 1200 cm⁻¹ region represents the stretching of the silicate network (ν_1). The region between
56
57 190 200 and 600 cm⁻¹ in amorphous material is still not well constrained owing to the complexity
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59
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3 191 of the many vibrations occurring and overlapping together. Previous studies [52] have shown
4
5 192 that the LF region is a complex mixture between bending of tetrahedron from the silicate
6
7
8 193 network (e.g., Al, Si, Ti or Fe in tetrahedral coordination) and bending of tetrahedron in relation
9
10 194 to cation network modifiers (e.g., Mg, Ca, Na, K and Fe in octahedral coordination). Although
11
12 195 in the subsequent discussion, we will address in more detail the HF region, we observe that the
13
14 196 LF region is more intense than the HF region for the PR-6Hb and PR-3H-6h, with 4.5 and 2.7
15
16 197 wt % FeO_{tot} respectively. There is an opposite behavior for the Fe-rich synthetic glasses (e.g.,
17
18 198 GC-3H-6h, GC- H_2O -4h, GCD-3H-6h, CL-3H-6h). In the PR samples (PR-3H-6h and PR-6Hb),
19
20 199 the HF region is asymmetric and exhibits three main features: 850, ~1000 and 1050 cm^{-1} . The
21
22 200 most intense peak is localized around 1000 cm^{-1} . In the GC samples (GC- H_2O -4h and GCD-
23
24 201 3H-6h) peaks at 850 and 1050 cm^{-1} are smoothen but distinguishable. An intense peak around
25
26 202 950 cm^{-1} is present for GC synthetic glasses. Finally, for CL-3H-6h, the HF region shape is
27
28 203 almost symmetric with an intense peak centered at 950 cm^{-1} . A Lorentzian-shape peak is
29
30 204 observed between the LF and HF region a 670 cm^{-1} for the iron-rich sample CL-3H-6h and the
31
32 205 Popping Rock PR-6Hb.

3.2. H_2O quantification by Raman spectroscopy

37
38 206 Raman spectra of glasses synthesized 1 GPa are shown in Fig. 1B in the H_2O vibration region
39
40 207 between 3400 and 3700 cm^{-1} . Acquisitions were also performed between 4000 and 4300 cm^{-1}
41
42 208 to investigate the possible presence of H_2 at 4135 cm^{-1} [53] but no signal was found. A main
43
44 209 broad feature is observed at 3550 cm^{-1} . At 3660 cm^{-1} , we can observe a shoulder only presents
45
46 210 for the CL-3H-6h and PR-6H-b samples corresponding to Free OH species.

47
48 211 The quantification of H_2O in our samples was performed using the calibration of Mercier et al.
49
50 212 [41] and according to the following equation (1):

$$213 \quad (1) \text{H}_2\text{O} (\text{wt} \%) = \frac{I_{\text{H}_2\text{O}_{\text{tot}}}}{I_{\text{HF}}} \times \frac{1}{a}$$

215 With $I_{\text{H}_2\text{O}_{\text{tot}}}$ the maximum peak intensity at 3550 cm^{-1} , HF the intensity taken at 960 cm^{-1}
216 and a the correlation coefficient chosen for a basaltic composition (i.e., 0.26).

217 Quantifications have been done on at least 5 spectra. The estimated error on the quantification
218 method is $\pm 0.06 \text{ wt } \% \text{ H}_2\text{O}$ [41]. Therefore, the H_2O errors mentioned in Table 2 are a
219 combination of 1) the error relative to the H_2O homogeneity in the glass if above the error
220 calibration provided by Mercier et al. [41], 2) the calibration error (0.1 wt %) if H_2O
221 homogeneity is lower than this value.

222 In Fig. 1B, spectra are normalized according to the concomitant HF peak intensity in Fig. 1A.
223 The PR glass with 4.5 wt % FeO_{tot} dissolves 6.5 wt % H_2O and shows the highest peak at 3550
224 cm^{-1} (i.e., PR-6Hb). The peak intensity is decreasing towards higher FeO_{tot} contents: for the
225 CL-3H-6h (with 16.0 wt % FeO_{tot}) 0.9 wt % H_2O is recovered. The highest 3550 cm^{-1} peak
226 intensity corresponds to the highest H_2O content.

227 The derived H_2O contents are reported in Table 2. High water content experiments at 1 GPa
228 have been measured for the PR composition with 4.1 and then 6.5 wt % H_2O dissolved in the
229 melt with 5.7 and 6.5 wt % $\text{H}_2\text{O}^{\text{ini}}$, respectively (i.e., PR-6H-4h and PR-6Hb). Experiment with
230 0.9 wt % $\text{H}_2\text{O}^{\text{ini}}$, at 1 GPa, has been tested for the GC composition (i.e., GC-1H-4h sample).
231 Almost all the H_2O initially added is recovered in the melt: 0.8 wt % H_2O . Experiments with
232 more than 3 wt % H_2O for the synthetic martian basalt compositions were performed at 1 GPa
233 but no quenched glass could be obtained. At 1.5 GPa, 0.8 wt % H_2O is determined in GCD-3H-
234 6h-1.5GPa, with $\sim 3 \text{ wt } \% \text{ H}_2\text{O}^{\text{ini}}$. Finally, at low pressure 0.9 wt % H_2O is dissolved in the same
235 starting composition (i.e., GCD).

236 3.3. H_2O quantification by FTIR versus Raman

237 FTIR spectroscopy has been conducted on several glass samples in order to investigate the
238 applicability of the Mercier et al. [41] calibration on Fe-rich basalts. Not all samples could be

analyzed because of 1) sample size (small glass chips < 1mm); 2) quenched crystals at the rim of the samples. Only pure glass chips with a size > 2 mm have been analyzed: GC-3H-6h, GC-1H-4h, GCD-3H-6h, PR-3H-6h, PR-6H-4h and PR-6Hb. The determination of water content by FTIR follows the Beer-Lambert equation [13, 14]:

$$(2) C_{H_2O} = \frac{100 \times MM \times A}{d \times \rho \times \epsilon}$$

Where C is the H₂O^{mol} or OH⁻ content in wt %, A the height of the band for OH⁻ (4500 cm⁻¹) or H₂O_{mol} (5200 cm⁻¹), MM the molar mass in g/mol (H₂O or OH), d the thickness in cm, ρ the density in g/L and ε the linear molar absorption coefficient in L/mol/cm.

For PR samples, we used the extinction coefficients derived from Ohlhosrt et al. [54] 0.56 L/mol/cm for both the peak at 4500 and 5200 cm⁻¹. For our Fe-rich we used the extinction coefficient given by Stolper [55] considering that the studied basalts are closer in composition to ours. The molar absorptivity coefficients are 0.98 and 1.76 L/mol/cm for 4500 and 5200 cm⁻¹, respectively.

Glass density measurements were not possible considering the size of our glass chips. It was calculated using the calibration of Schiavi et al. [43]; which takes into account the chemical composition and is applicable to Fe-rich glass compositions.

In Fig. 2, we observe that the H₂O quantified with Mercier et al. [41] provided better correspondence to the FTIR results. At low H₂O content, for Fe-rich glasses, the H₂O quantification by Mercier et al. [41] is close to the 1:1 line. As an example, for the GC-3H-6h sample, 1.4 wt % and 1.5 wt % H₂O are determined by FTIR and Raman, respectively. At higher H₂O content (> 2 wt %), the results are more scattered, but still in reasonable agreement with the FTIR results. For the water-rich sample (PR-6Hb), 6.9 wt % H₂O is determined by FTIR, where 6.5 is calculated by the method of Mercier et al. [41]. There is a strong discrepancy for PR-6H-4h with 4.1 wt % H₂O determined by Raman whereas 6.1 wt % is determined by FTIR.

263 However, this sample has the highest error bar due to possible heterogeneity in the H₂O
264 distribution with the glass.

265 In Fig. 2, the calibration provided by Behrens et al. [40] slightly overestimates the water content
266 for Fe-rich glasses at low H₂O content. For GC-1H-4h, 1.1 wt % H₂O is quantified with Raman
267 and 1.0 wt % by FTIR; although 0.9 wt % H₂O was initially added before the experiment. The
268 discrepancy becomes more important at high H₂O content in PR-6Hb glass: 8.1 wt % H₂O by
269 Raman with 6.5 wt % H₂Oⁱⁿⁱ. However, it should be emphasized that the calibration is accurate
270 to 4.7 wt % H₂O; and at higher content, the error in determining the amount of H₂O will be
271 larger.

272 The calibration established by Schiavi et al. [43] shows good results for low H₂O content in
273 Fig. 2 with 0.7 wt % for GC-1H-4h against 1.0 wt % H₂O by FTIR. However, with increasing
274 H₂O content, there is an increasing deviation with the FTIR results. This difference can be
275 ascribed by the high Fe content of basalts in Schiavi et al. [43] (FeO_{tot}, > 7 wt %) compared to
276 our Fe-poor basalts (< 6 wt % FeO_{tot}).

277 The comparison provided in Fig. 2 suggests that we can reasonably assess that the Raman
278 calibration established by Mercier et al. [41] is more appropriate to quantify H₂O in our Fe-rich
279 and Fe-poor basaltic glasses. The results from all H₂O quantifications by IR and Raman
280 spectroscopies are reported in the Table S1 (Supporting Information).

281

282

4. Discussion

4.1. H₂O content in Fe-rich basalts

284 Differences between H₂O dissolved in the melt and H₂Oⁱⁿⁱ can be observed for PR samples
285 (Table 2). For 5.7 wt % H₂O added in PR-6H-4h experiment, less than 4 wt % H₂O is dissolved.

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3 286 For 6.5 wt %, ~6.5 wt % H₂O is recovered in the PR glass. Under these conditions (i.e. 1 GPa),
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5 287 H₂O solubility is close 7 wt % [56]. The difference between initial and final amount of H₂O for
6
7 288 PR glasses is explained by the thermodynamic equilibrium of the H₂O species in the liquid and
8
9 289 fluid phase [57]; by extension, there is a partitioning of H₂O between the fluid phase and the
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12 290 melt phase.

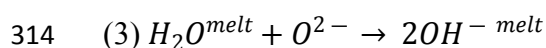
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15 291 Some mixtures were not decarbonated and CO₂ was present in the fluid phase during the
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17 292 experiments. However after the experiments, no CO₃²⁻ signal at 1080 cm⁻¹ was observed by
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19 293 Raman spectroscopy [37], implying that only hundreds ppm of CO₂ could be dissolved. It is
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21 294 consistent with previous studies that estimated the CO₂ solubility in these iron-rich melt at ~0.3
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23 295 wt % under oxidized conditions [27, 28].

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27 296 The presented results in Fig. 1B and Table 2, suggest that increasing Fe content will induce a
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29 297 substantial decrease in the H₂O dissolution. For instance, if we compare the PR-3H-6h and
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31 298 GCD-3H-6h samples with 2.9 and 2.7 wt % H₂Oⁱⁿⁱ, the total H₂O dissolved in the melt is 2.8
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33 299 and 0.7 wt % H₂O, respectively. For those two experiments made at identical intensive
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35 300 conditions and with no CO₂ present, there is a difference of ~2 wt % between H₂O dissolved
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37 301 and the H₂O initially loaded in the sample charge. The most likely hypothesis to explain this
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39 302 difference is the change in the Fe content: ~13 wt % for GC samples and 5 wt % FeO_{tot} for PR.

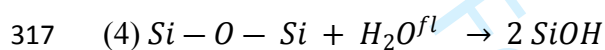
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44 303 We have presented the H₂O dissolved as a function of the FeO_{tot} for 1 GPa experiments in Fig.
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46 304 3. We observe a decrease of the H₂O content from 6.5 wt % H₂O at 5 wt % FeO_{tot} for PR-6Hb,
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48 305 to less than 1 wt % at 16 wt % FeO_{tot} for CL-3H-6h (see Table 2). Hence, increasing the FeO_{tot}
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50 306 in the melt composition seems to hamper the H₂O dissolution. This trend has also been shown
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52 307 in the recent work of Hou et al. [57] investigating the immiscibility of an Fe-rich melt in
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54 308 presence of H₂O.

309 4.2. Peculiar H₂O dissolution mechanisms in Fe-rich basalt

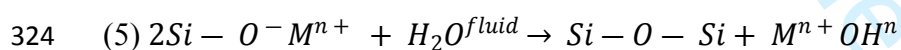
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3 310 Previous spectroscopic investigations [13, 45, 46, 54] have shown that H₂O can dissolve in
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5 311 silicate glasses as two main species: 1) hydroxyl groups (i.e., OH) or 2) molecular H₂O (i.e.,
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7 312 H₂O_{mol}). At low H₂O content, OH is the predominant species to be formed in silicate glasses
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9 313 owing to the water dissociation reaction:



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16 315 This dissolution mechanism induces a depolymerization of the silicate network through the
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18 316 formation of silanol groups (SiOH):



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24 318 At higher H₂O content, H₂O_{mol} becomes the dominant species whereas OH remains almost
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26 319 constant. An additional H₂O dissolution mechanism has been inferred in recent studies based
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28 320 on Raman and NMR spectroscopic results [46, 58] and corresponds to the formation of Free
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30 321 OH groups. It has been suggested that the formation of Free OH will induce a polymerization
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32 322 of the silicate melt structure as the network modifying cation charges are consumed by the OH-
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34 323 negative charges according to the following equation (5):



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42 325 With M for cations (i.e., Ca, Mg, Fe) and n representing the charge of the cation.

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45 326 A shoulder at 3660 cm⁻¹ corresponding to Free OH vibrations [45-47] is observed for the Fe-
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47 327 rich glass (CL-3H-6h) and the water-rich terrestrial analogue (PR-6Hb) in Fig. 1B. Coupled
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49 328 with the shoulder at 3660 cm⁻¹, a peculiar peak at 670 cm⁻¹ has been detected for those two
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51 329 spectra (Fig. 1A) with a Lorentzian shape. The occurrence of the 3660 and 670 cm⁻¹ peaks
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53 330 appear correlated, although the assignment of the 670 cm⁻¹ peak to a peculiar vibration remains
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55 331 to be determined. In the study of Di Genova et al. [42], Raman spectra acquired on Fe-rich
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57 332 basaltic glasses exhibit a specific peak at ~690 cm⁻¹ that is correlated with nanolite particles due
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333 to high Fe content, but also due to high H₂O content. Furthermore, we have noticed that some
334 other experiments showed the same peaks: GC-H₂O-chiller, GC-H₂O-4h and GCD-3H-6h-
335 1.5GPa.

336 To determine the proportion of the 3660 cm⁻¹ peak, we carried out spectrum simulation using 4
337 peaks: 3 Gaussians and one Voigt (combination of a Gaussian and a Lorentzian component).
338 Values of simulations are reported in the Supporting Information in Table S2. Typical
339 simulations for PR-6Hb and CL-3H-6h are shown in Fig. 4. In the simulations in Fig. 4, CL-
340 3H-6h sample exhibits ~17 % of Free OH and only 2 % is calculated for PR-6Hb.

341 In Fig. 5A, we have reported the proportions of Free OH as a function of the proportion of the
342 670 cm⁻¹ peak (i.e., 670* corresponding to the ratio between the 670 cm⁻¹ area and the total area
343 of H₂O) for the samples: CL-3H-6h, PR-6Hb, GC-H₂O-chiller, GC-H₂O-4h and GCD-3H-6h-
344 1.5GPa. Each data reported in this graph corresponds to the proportion determined from the
345 simulation of a given spectrum. In Fig. 5A, we observe a correlation between the percentage of
346 Free OH and the 670 cm⁻¹ peak. The increase of the 670 cm⁻¹ is correlated to an increase in Free
347 OH. For PR-6Hb, we measure the 670* = 6 % and Free OH = 2 %. For CL-3H-6h, we obtain
348 670* ~60 % and 20 % of Free OH.

349 In Fig. 5B, the Free OH percentages are represented against the FeO_{tot}. The Free OH fraction
350 appears also correlated to the FeO_{tot} in the sample. The Fe-rich sample, CL-3H-6h with 16.0 wt
351 % FeO_{tot}, presents ~17 % of Free OH. The GC and GCD samples exhibit values from 3 to 7 %
352 of Free OH, with ~12 and ~14 wt % FeO_{tot}, respectively. On the other hand, PR-6Hb with 4.5
353 wt % FeO_{tot}, shows only 2 % of Free OH.

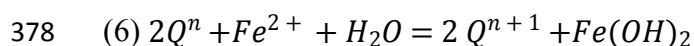
354 Previous works have shown that the formation of Free OH is observed 1) when H₂O is
355 dissolving in a strongly depolymerized composition [57], 2) with high H₂O concentrations [45],
356 and 3) in aqueous solutions [47]. The presence of Free OH is rather surprising in Fe-rich glasses

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3 357 investigated here considering that the total H₂O content dissolved does not exceed 1 wt %. One
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5 358 possible reason for the existence of Free OH groups in Fe-rich glasses could be ascribed to the
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7 359 structural role of iron within the glass. Recovered Fe-rich glasses (i.e., GC and CL) from high
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9 360 pressure experiments have a Fe ratio between $Fe^{2+} / \sum Fe = 0.4$ and 0.6, respectively (see Table
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11 361 2) suggesting that Fe plays a network modifying role [59] within the glass when under its Fe²⁺
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13 362 state. Although additional work is clearly needed to better characterize the correlation between
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15 363 the Free OH groups the Fe redox state; it is possible that Fe²⁺ has a high affinity to form isolated
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17 364 complexes with OH.

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22 365 The observation of Free OH in Fe-rich melt at low H₂O content (~1 wt %) has implications on
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24 366 the mineralogical composition at the Mars surface. The formation of Free OH molecular
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26 367 clusters could constitute a precursor for an immiscible Fe-rich phase; in a similar manner to the
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28 368 immiscibility observed between carbonatite and silicate melts owing to the formation of Free
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30 369 carbonates [60]. Previous spatial missions identified multiple forms of Fe-oxides, mostly
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32 370 hematite and magnetite at the surface of Mars [61-63]. The separation of Fe-(OH)₂ rich phase
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34 371 could favor the formation of Fe-oxides by weathering of the basalt at the surface.

35 36 37 38 39 372 4.3. Evolution of the silicate network signature as a function of the Fe redox state 40 41 373 estimation

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44 374 The presence of Free OH should induce a change on the overall silicate network considering
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46 375 that Fe²⁺ cations are scavenged by the OH species. Therefore, the deficit of positive charges
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48 376 from the network modifying cations should induce an increase in the degree of polymerization
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50 377 according to the reaction (6):



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57 379 This reaction equivalent to Eq. 5 shows that Qⁿ species proportions are changing due to the
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59 380 formation of Free OH clusters. For instance, Q³ species should merge to form Q⁴ species upon

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3 381 the dissolution of water as Free OH clusters. This change is likely to be observed by Raman
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5 382 spectroscopy.

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8 383 The HF region is a composition of several Gaussians peaks [60] referred to a Q^n species with,
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10 384 n the number of bridging O varying from 0 to 4. We conducted spectrum deconvolution of the
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13 385 HF region that are exhibited in Fig. 6. The results from the deconvolutions are reported in the
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15 386 Table S2 (Supporting Information). The deconvolutions of the Raman spectra for PR-6H-4h,
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17 387 GCD-3H-6h and CL-3H-6h (i.e. synthetic martian basalts) are reported in Fig. 6. The
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20 388 assignment of each Gaussian line is based on previous works [64]: Q^4 at 1080 cm^{-1} , Q^3 at ~ 1020
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22 389 cm^{-1} , Q^2 at $\sim 950\text{ cm}^{-1}$ and Q^1 at $\sim 880\text{ cm}^{-1}$.

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25 390 For the PR-6H-4h sample, the deconvolution shows that the Q^3 ($\sim 1010\text{ cm}^{-1}$) is the most intense
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27 391 peak whereas the GCD sample shows a more intense peak at lower Raman shift with the Q^2
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29 392 ($\sim 950\text{ cm}^{-1}$), leading to less polymerized network. Then, for the CL sample, we can see that the
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31 393 Q^2 peak is more intense but the Q^3 is almost at the same height.

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35 394 Regarding solely at the Fe^{2+} content reported in Fig. 6, we can see an increase of the Fe^{2+} content
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37 395 in the melt from the PR-6H-4h toward CL-3H-6h. Indeed, if the GCD sample is the most
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39 396 oxidized, the bulk Fe^{2+} is higher as compared to the PR sample: 5.4 and 4.2 wt % Fe^{2+} ,
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41 397 respectively. The presence of high Fe^{2+} content enhances the depolymerization of the structure,
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43 398 which is confirmed by the deconvolution of the GCD glass in Fig. 6. Then, the low content of
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45 399 Fe^{2+} coupled with the high SiO_2 content ($> 51\text{ wt } \%$) of the PR-6H-4h sample enable a
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47 400 polymerization of the silicate glass. We can see that the iron-rich composition (CL sample) is
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49 401 richer in Fe^{2+} species (9.8 wt %). Hence, we expected a more depolymerized structure correlated
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51 402 with the high Fe^{2+} in the melt. However, the opposite behavior is observed from subsequent
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53 403 simulation reported in Fig. 6.
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3 404 In order to discriminate the possible effect of Free OH related to Fe^{2+} on the degree of
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5 405 polymerization, we represent in the Fig. 7 the ratio of Q^4 over the sum of Q^n species as a function
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7 406 of Fe^{2+} normalized by the H_2O content. We choose to show the change in Q^4 species as this is
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9 407 the most polymerized unit (4 bridging oxygens) and the trend is the best obtained between the
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11 408 Q^n species; however, trends for the other species can be found in the Fig. S1 (Supporting
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13 409 information). The deconvolution of glasses obtained at all pressures associated with a relative
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15 410 error of 10 % are reported in Fig. 7. The Fig. 7 shows two major information. 1) Increasing Fe^{2+}
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17 411 in the glass induces a decrease in the fraction of Q^4 (from 0.23 to 0.16 between $\text{Fe}^{2+}/\text{H}_2\text{O} \sim 1$ to
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19 412 11) and therefore a depolymerization of the glass. These results appear consistent with the
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21 413 suggested role of Fe^{2+} as a network modifier. 2) Free OH species have only a limited effect on
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23 414 the degree of polymerization for PR-6H-4h (without Free OH) and PR-6Hb (with Free OH)
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25 415 $Q^4/\sum Q^n = 0.23$ and 0.26. The same applies for GCD-3H-6h without Free OH and GCD-3H-
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27 416 1.5G with Free OH: 0.18 and 0.19, respectively.

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33 417 We expected measurable effect of Free OH with the increase of the polymerization degree
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35 418 observed in Fig. 6 (CL-3H-6h). Several aspects could explain the absence of effect shown in
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37 419 Fig. 7: 1) although the Free OH are visible in Raman spectra, their abundance is small compared
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39 420 to the $\text{H}_2\text{O}_{\text{tot.}}$; Thus the effect on the degree of polymerization is not visible or within error the
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41 421 Q^n simulation, 2) it is also possible that the Free OH species are not exclusively linked to Fe^{2+}
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43 422 and other cations such as Mg^{2+} and Ca^{2+} form Free OH clusters. However, the trend observed
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45 423 in Fig. 5B strongly favors an affinity of OH^- with Fe^{2+} .

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51 52 53 425 **Summary**

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56 426 In the present work, we have shown that Fe concentration in glass has a non-negligible effect
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58 427 on the H_2O dissolution within the melt structure. Increasing the Fe content appears to induce a
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3 428 decrease in the H₂O dissolution. This result has major implications for Mars, notably for the
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5 429 transport of H₂O in Fe-rich melt from the mantle to the surface that will be less efficient than
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8 430 on Earth.

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10 431 We inferred the formation of Free OH in Fe-rich glasses (i.e., martian basalt analogues) at
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12 432 relatively low H₂O content (~1 wt %) whereas such molecular groups appear at very high H₂O
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14 433 content for terrestrial basaltic glasses (> 6 wt %). The presence of Free OH molecular clusters
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16 434 seems to be dependent on the Fe content by increasing with FeO^{tot}. In detail, Free OH species
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18 435 are likely to be correlated to the Fe²⁺ in Fe-rich composition. Free OH are supposed to induce
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20 436 a polymerization effect on the silicate network; although further work is currently needed to
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22 437 investigate the exact molecular configuration in the surrounding of cations forming Free OH
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24 438 groups. The possibility for Free OH in the vicinity of Fe²⁺ cations represents a precursor for
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26 439 producing Fe-rich mineral phases through weathering of the erupted Fe-rich basalts, which will
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28 440 be consistent with Fe-oxides found at Mars surface.

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44
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47 447 and analyses.

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Tables:

635 Table 1: Starting compositions for PR (natural glass of Popping Rock) an Earth basalt analogue, CL and
 636 GC synthetic martian basalts.

637	Oxides (wt %)	PR ^a	GC ^b	CL ^b
638	SiO ₂	50.3	48.1	46.5
639	TiO ₂	1.8		
640	Al ₂ O ₃	14.8	10.2	8.2
641	FeO	8.8	15.5	23.5
642	Fe ₂ O ₃	1.3		
643	MnO	0.2		
644	MgO	7.7	13.1	9.8
645	CaO	10.8	8.1	7.6
646	Na ₂ O	2.9	4.9	4.3
647	K ₂ O	0.7		
648	P ₂ O ₅	0.3		
649	Total	99.6	100.0	100.0
650	NBO/T ^c	0.38	0.56	0.33
651	$\frac{Fe^{2+}}{\sum Fe}$ ^d		0.03 (1)	0.02 (1)

Notes :

^a Natural glass of Popping Rock^b Synthetic analogues to Martian basalt (> 15 wt % FeO)^c Calculated on the basis of Fe in its tetrahedral coordination (i.e. Fe³⁺), acting as a network former like Si, Al and Ti tetrahedron. NBO/T parameter ranging from 0 to 4. Calculations made on an anhydrous basis^d $\frac{Fe^{2+}}{\sum Fe}$ Ratio calculated from wet chemistry analyses for decarbonated powders.Standards errors (1 σ) are reported in brackets at 10⁻²

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Samples	GC-H ₂ O-chiller	GC-H ₂ O-4h	GC-3H-6h	GC-1H-4h	GCD-3H-6h	GCD-3H-6h-1.5GPa	GCD-2H-0.5GPa ^a	CL-3H-6h	PR-3H-6h	PR-6H-4h	PR-6Hb ^a
1 H ₂ O ⁱⁿⁱ (wt %)	2.6	2.6	2.4	0.9	2.7	3.0	1.9	2.9	2.9	5.7	6.5
2 CO ₂ ⁱⁿⁱ (wt %)	3.5	3.5	3.5	3.5				3.1			
3 P (GPa)	1	1	1	1	1	1.5	0.5	1	1	1	1
4 T (°C)	1600	1600	1600	1600	1600	1600	1600	1600	1600	1600	1500
5 t (h)	1	4	6	4	6	6	6	6	6	4	1
6 SiO ₂	48.6 (8)	50.1 (3)	50.9 (3)	51.1 (3)	48.5 (2)	47.8 (4)	49.4 (2)	50.9 (2)	54.8 (3)	51.2 (3)	51.14 (20)
7 TiO ₂									1.9 (1)	1.7 (1)	1.51 (6)
8 Al ₂ O ₃	11.5 (5)	10.7 (4)	10.8 (2)	10.6 (2)	10.2 (2)	10.4 (20)	10.9 (1)	8.7 (2)	15.2 (3)	14.5 (3)	14.49 (11)
9 FeO _{tot}	13.4 (4)	12.1 (3)	10.6 (3)	11.6 (3)	14.2 (3)	14.1 (3)	13.4 (1)	16.0 (4)	2.7 (1)	5.8 (2)	4.54 (11)
10 MnO									0.1 (0)	0.1 (0)	0.11 (2)
11 MgO	12.7 (1.0)	12.7 (4)	12.8 (3)	12.9 (3)	12.3 (3)	12.0 (3)	11.3 (1)	10.0 (1)	7.9 (2)	7.4 (2)	8.45 (21)
12 CaO	5.9 (1)	5.6 (3)	5.9 (2)	6.0 (1)	6.9 (3)	6.6 (2)	7.6 (1)	5.6 (2)	10.7 (2)	10.1 (3)	10.71 (18)
13 Na ₂ O	3.6 (3)	4.0 (6)	4.3 (2)	4.5 (1)	4.2 (3)	3.7 (3)	5.1 (1)	4.0 (2)	3.0 (1)	2.5 (1)	3.13 (5)
14 K ₂ O									0.7 (0)	0.7 (0)	0.61 (3)
15 P ₂ O ₅									0.3 (1)	0.3 (1)	0.26 (3)
16 Total	95.6	95.2	95.3	96.8	96.4	94.7	97.8	95.3	95.2	92.8	93.61
17 H ₂ O _{final} (wt %)	1.0 (1)	1.0 (2)	1.5 (1)	0.8 (1)	0.7 (1)	0.8 (1)	0.9 (1)	0.9 (2)	2.7 (2)	4.1 (2)	6.5 (3)
18 Free OH (%)	6.0	3.0				6.6		17.1			2.0
19 $\frac{Fe^{2+}}{\sum Fe}$ ^b	0.47	0.49	0.53	0.57	0.37	0.46	0.44	0.58	0.83	0.70	0.82

Notes:

^a Final chemical compositions determined by SEM only^b $\frac{Fe^{2+}}{\sum Fe}$ Ratio calculated from wet chemistry analyses

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Table 2: Experimental settings for the different samples (P, T, t, H₂O content). Chemical compositions of the samples determined by EPMA and SEM analyses. Final content of H₂O and FeO_{tot} are also reported, associated with wet chemistry results. The Free OH percentages are added for samples presenting the features at 3660 cm⁻¹. The calculation of Free OH percentages is made on the deconvolution of the H₂O area. Detailed of deconvolutions are reported in the Table S2 (Supporting Information). Standards errors (1 σ) are reported in brackets at 10⁻¹. The associated standard errors of wet chemistry measurements are 0.01.

Supporting Information Tables:

	PR-3H-6h	PR-6H-4h	PR-6Hb	GC-3H-6h	GC-1H-4h	GCD-3H-6h
ϵ_{4500} *	0.56 ^a	0.56	0.56	0.98 ^b	0.98	0.98
ϵ_{5200}	0.56	0.56	0.56	1.76	1.76	1.76
Density ** (g/L)	2789	2765	2722	2815	2832	2833
H ₂ O wt % (FTIR)	3.6 (3)	6.1 (9)	6.9 (4)	1.4 (2)	1.0 (3)	0.8 (2)
	H ₂ O from Raman spectroscopy (% wt):					
Mercier et al. [41]	2.7 (2)	4.1 (2)	6.5 (3)	1.5 (1)	0.8 (1)	0.7 (1)
Behrens et al. [40]	2.2 (1)	5.0 (1)	8.1 (4)	2.3 (1)	1.1 (1)	1.1 (1)
Schiavi et al. [43]	1.4 (1)	3.0 (1)	4.9 (3)	1.4 (2)	0.7 (2)	0.7 (2)

Notes :

* : extinction coefficient L/mol/cm

** : calculated according to Schiavi et al. [43]

^a: From Ohlhorst et al. [54]

^b: From Stolper [55]

Table S1: FTIR and Raman H₂O quantifications. Description of the parameters chosen for FTIR calculations.

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		GC-H ₂ O-chiller	GC-H ₂ O-4h	GC-3H-6h	GC-1H-4h	GCD-3H-6h	GCD-3H-6h-1.5G	GCD-2H-0.5G	CL-3H-6h	PR-3H-6h	PR-6H-4h	PR-6Hb	
HF region	Q1	Position (cm ⁻¹)	884	887	874	886	877	882	875	882	879	885	905
		FWHM	80	81	82	78	78	82	77	84	84	79	70
		Area %	20	20	18	20	22	20	18	20	15	15	14
	Q2	Position (cm ⁻¹)	948	951	940	951	940	946	940	947	951	950	969
		FWHM	82	86	81	83	87	87	76	83	78	82	69
		Area %	33	32	32	32	33	32	34	34	26	28	31
	Q3	Position (cm ⁻¹)	1011	1014	1001	1012	1001	1009	999	1011	1007	1005	1020
		FWHM	82	88	90	82	94	87	81	83	74	77	66
		Area %	29	29	29	28	27	29	29	31	32	34	31
	Q4	Position (cm ⁻¹)	1079	1082	1073	1081	1072	1076	1065	1075	1077	1077	1078
		FWHM	86	91	90	83	90	90	79	84	72	83	63
		Area %	19	20	21	19	18	19	19	16	25	23	26
H ₂ O region	D1	Position (cm ⁻¹)	3357	3252	3278	3380	3323	3374	3398	3477	3410	3278	3279
		FWHM	208	200	222	219	126	284	126	220	201	356	228
		Area %	16	23	16	28	40	20	18	28	42	24	19
	D2	Position (cm ⁻¹)	3473	3478	3391	3480	3446	3514	3558	3590	3573	3447	3459
		FWHM	146	220	109	129	122	160	137	130	149	176	184
		Area %	30	40	23	31	50	32	82	55	58	32	29
	D3	Position (cm ⁻¹)	3581	3590	3556	3596	3571	3606				3576	3586
		FWHM	130	130	179	116	133	130				141	148
		Area %	48	34	61	41	10	42				44	50
	Free OH	Position (cm ⁻¹)	3666	3672				3673		3661			3665
		FWHM	52	50				48		54			43
		Area %	6	3				7		17			2

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675 Table S2: Deconvolution results for each sample in the HF (800 – 1250 cm⁻¹) region and the H₂O region (3000 – 4000 cm⁻¹). For each deconvolution, the
676 position peak is first described, then the FWHM (i.e., Full Width at Half Maximum) and the area percentage (area %). The addition of D1 to D3 coupled with
677 the Free OH peak represents the entire H₂O vibration.

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Figure captions

Fig. 1: A) Raman spectra of the 200 – 1250 cm^{-1} region for the Fe-poor glasses (PR) and the synthetic Fe-rich glasses obtained at 1 GPa. B) Raman spectra of the same samples in the high frequency region (2600 – 4000 cm^{-1}). Spectra are ranked according to the FeO_{tot} content in the samples. Spectra are normalized to the most intense peak (arbitrary units).

Fig. 2: H_2O quantifications by FTIR against calibrations established by Raman spectroscopy for Mercier et al. [41] in black, Behrens et al. [40] in blue and Schiavi et al. [43] in green. Error bars reported are the error from the homogenization of the sample when it is above the calibration error of each study.

Fig. 3: H_2O dissolved in the melt versus the FeO_{tot} measured by EPMA analysis in our samples at 1 GPa. The $\text{H}_2\text{O}^{\text{ini}}$ content is also reported.

Fig. 4: Deconvolution of the H_2O region (2600 – 4000 cm^{-1}) for CL-3H-6h and PR-6Hb samples. In blue is highlighted the Free OH area. The CL-3H-6h sample has been increased x5 for clarity.

Fig. 5: A) Percentage of Free OH (i.e., $A_{3660}/A_{\text{H}_2\text{O}_{\text{tot}}}$; with A the area from deconvolution of the 3660 cm^{-1} and the water band from 3000 to 4000 cm^{-1}) versus the percentage of the 670 cm^{-1} peak (i.e., 670*) normalized by the total H_2O area ($A_{\text{H}_2\text{O}_{\text{tot}}}$); B) Free OH percentages in function of the total iron content (FeO_{tot}).

Fig. 6: Deconvolution of the high frequency region (i.e., HF; silicate matrix signal) for CL-3H-6h, GCD-3H-6h and PR-6H-4h. Gaussian lines are in green, cumulative peaks of the Gaussian lines are the black and the residuals from simulations is the black dotted lines. H_2O and FeO_{tot} final content for each experiment are localized next to their respective spectrum. The Fe^{2+} content is given for each sample. Spectra are normalized to the most intense peak (arbitrary units).

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Fig. 7: Ratio of $Q^4/\sum Q^n$ species versus the Fe^{2+} normalized by the H_2O content for all samples. PR samples are in green, where synthetic Fe-rich experiments are represented in red. The black stars are samples presenting Free OH molecular groups. A relative error of 10 % has been applied.

Supporting Information Figure:

Fig. S1: Ratio of the Q^n species in function of the Fe^{2+} normalized by the H_2O content for PR glasses (i.e., Fe-poor) in green, GC/CL samples (Fe-rich synthetic glasses) in red, and the glasses presenting Free OH molecular groups in black stars: A) $Q^1/\sum Q^n$, B) $Q^2/\sum Q^n$ and C) $Q^3/\sum Q^n$. A relative error of 10 % on the deconvolution is applied.

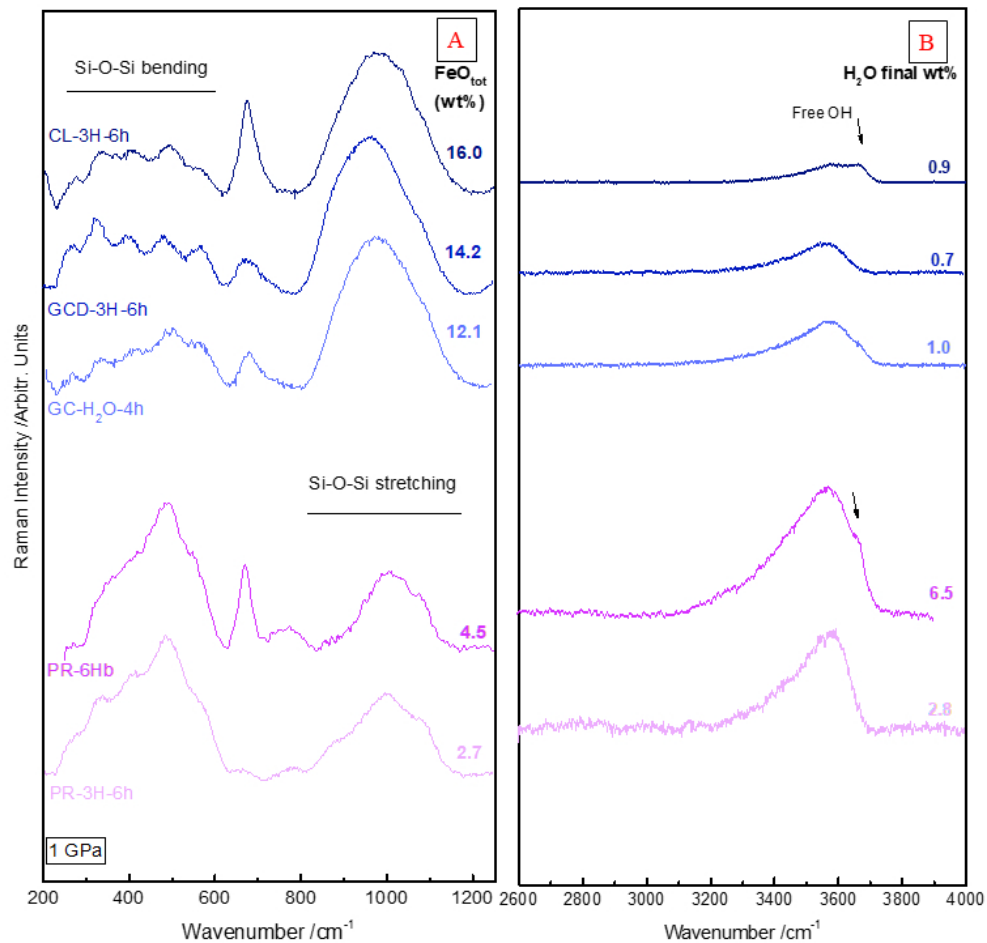


Fig. 1: A) Raman spectra of the 200 – 1250 cm⁻¹ region for the Fe-poor glasses (PR) and the synthetic Fe-rich glasses obtained at 1 GPa. B) Raman spectra of the same samples in the high frequency region (2600 – 4000 cm⁻¹). Spectra are ranked according to the FeO_{tot} content in the samples. Spectra are normalized to the most intense peak (arbitrary units).

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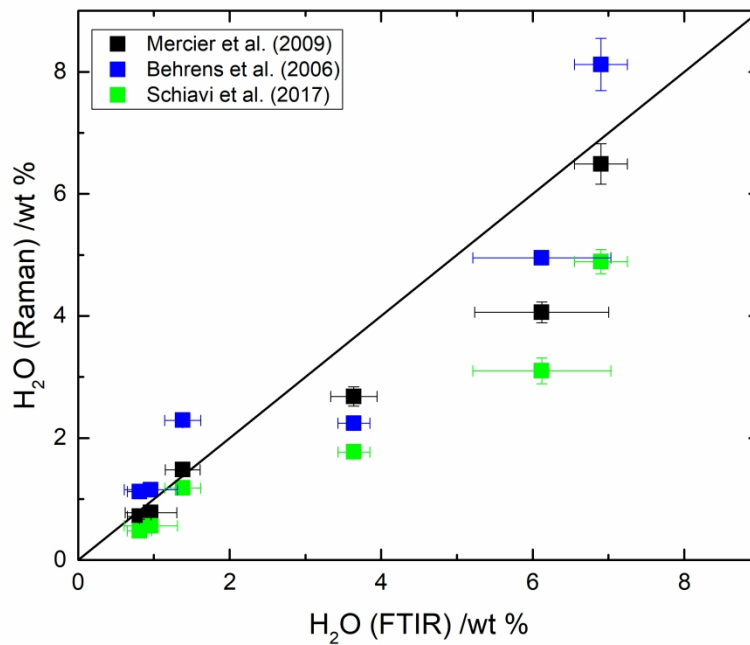


Fig. 2: H₂O quantifications by FTIR against calibrations established by Raman spectroscopy for Mercier et al. [41] in black, Behrens et al. [40] in blue and Schiavi et al. [43] in green. Error bars reported are the error from the homogenization of the sample when it is above the calibration error of each study.

272x208mm (300 x 300 DPI)

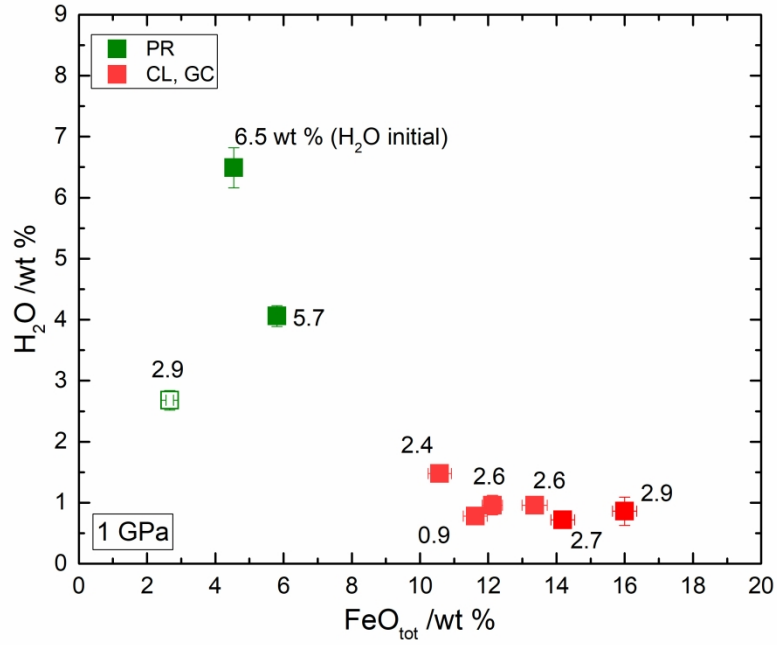


Fig. 3: H₂O dissolved in the melt versus the FeO_{tot} measured by EPMA analysis in our samples at 1 GPa. The H₂O_{ini} content is also reported.

272x208mm (300 x 300 DPI)

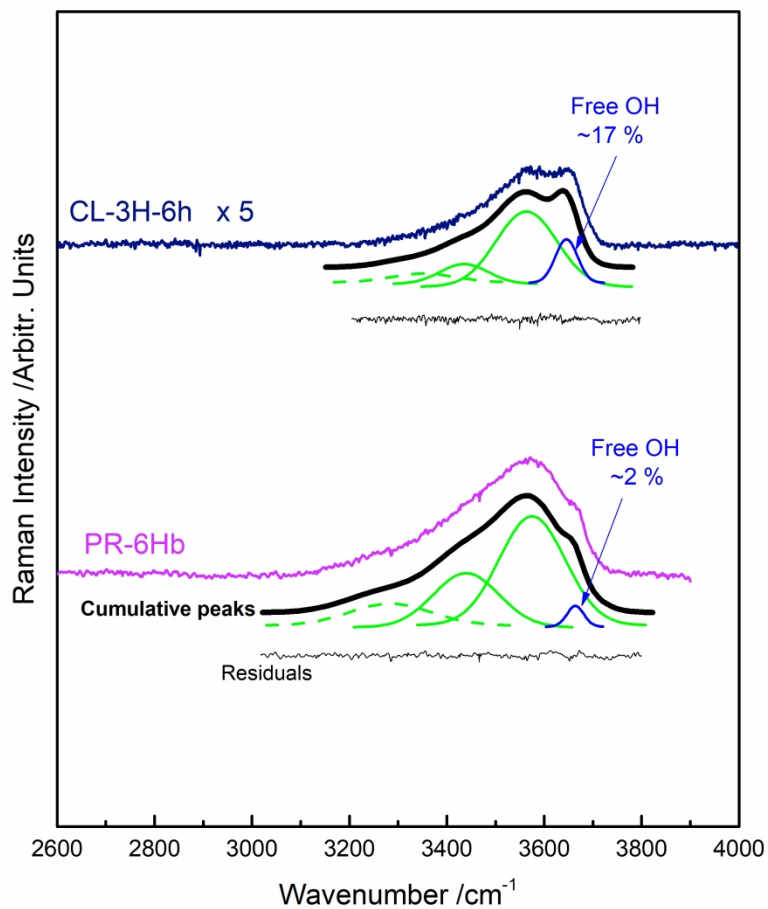


Fig. 4: Deconvolution of the H₂O region (2600 – 4000 cm^{-1}) for CL-3H-6h and PR-6Hb samples. In blue is highlighted the Free OH area. The CL-3H-6h sample has been increased x5 for clarity.

272x308mm (300 x 300 DPI)

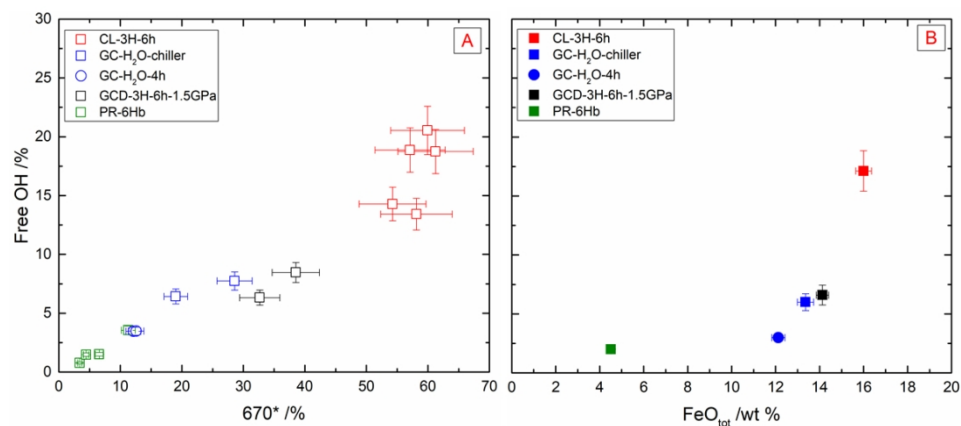


Fig. 5: A) Percentage of Free OH (i.e., $A_{3660}/A_{H_2O_{tot}}$; with A the area from deconvolution of the 3660 cm^{-1} and the water band from 3000 to 4000 cm^{-1}) versus the percentage of the 670 cm^{-1} peak (i.e., 670^*) normalized by the total H_2O area ($A_{H_2O_{tot}}$); B) Free OH percentages in function of the total iron content (FeO_{tot}).

338x190mm (96 x 96 DPI)

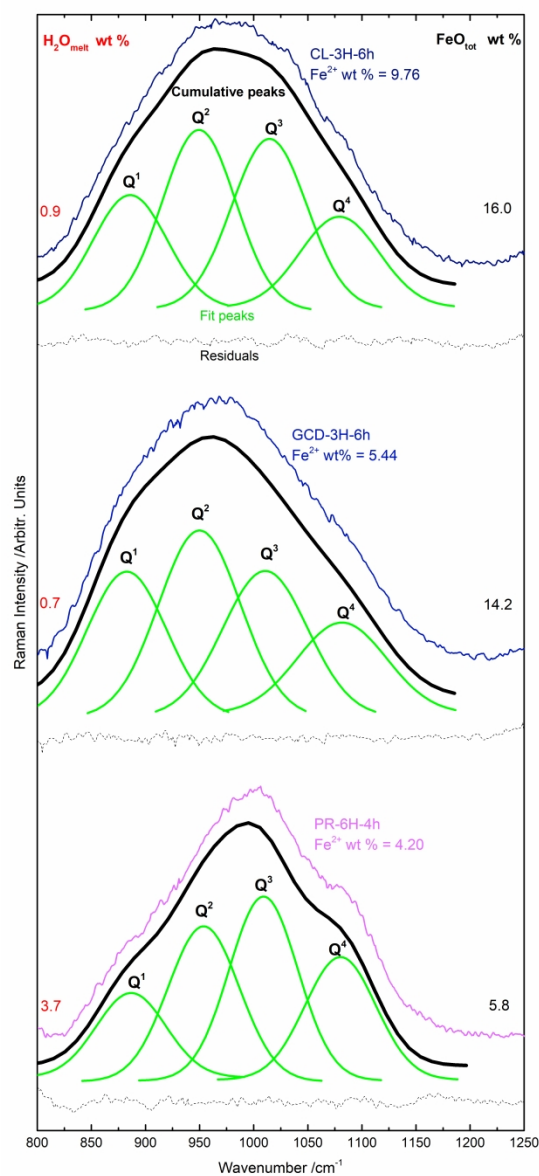


Fig. 6: Deconvolution of the high frequency region (i.e., HF; silicate matrix signal) for CL-3H-6h, GCD-3H-6h and PR-6H-4h. Gaussian lines are in green, cumulative peaks of the Gaussian lines are the black and the residuals from simulations is the black dotted lines. H₂O and FeO_{tot} final content for each experiment are localized next to their respective spectrum. The Fe²⁺ content is given for each sample. Spectra are normalized to the most intense peak (arbitrary units).

289x624mm (300 x 300 DPI)

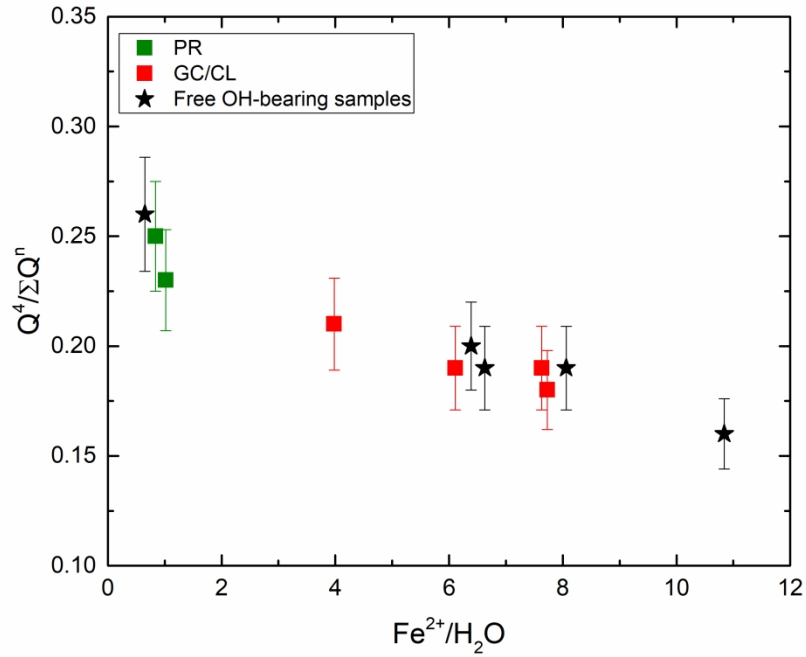


Fig. 7: Ratio of $Q^4/\Sigma Q^n$ species versus the Fe^{2+} normalized by the H_2O content for all samples. PR samples are in green, where synthetic Fe-rich experiments are represented in red. The black stars are samples presenting Free OH molecular groups. A relative error of 10 % has been applied.

272x208mm (300 x 300 DPI)

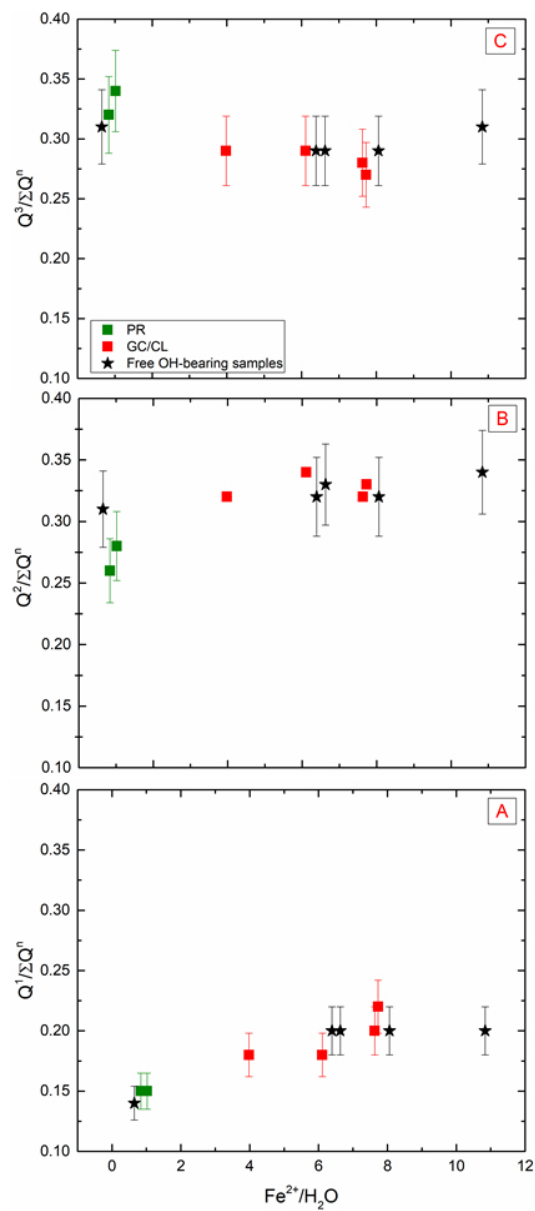


Fig. S1: Ratio of the Qn species in function of the Fe²⁺ normalized by the H₂O content for PR glasses (i.e., Fe-poor) in green, GC/CL samples (Fe-rich synthetic glasses) in red, and the glasses presenting Free OH molecular groups in black stars: A) Q₁/ΣQ_n, B) Q₂/ΣQ_n and C) Q₃/ΣQ_n. A relative error of 10 % on the deconvolution is applied.

117x254mm (96 x 96 DPI)