

Causal inference machine learning leads original experimental discovery in CdSe/CdS core/shell nanoparticles

Rulin Liu, Junjie Hao, Jiagen Li, Shujie Wang, Haochen Liu, Ziming Zhou, Marie-Hélène Delville, Jiaji Cheng, Kai Wang, Xi Zhu

▶ To cite this version:

Rulin Liu, Junjie Hao, Jiagen Li, Shujie Wang, Haochen Liu, et al.. Causal inference machine learning leads original experimental discovery in CdSe/CdS core/shell nanoparticles. Journal of Physical Chemistry Letters, 2020, 11, pp.7232-7238. 10.1021/acs.jpclett.0c02115. hal-02922356

HAL Id: hal-02922356 https://hal.science/hal-02922356v1

Submitted on 26 Aug 2020

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

The Causal Inference Machine Learning Leads Original Experimental Discovery in CdSe/CdS Core/Shell Nanoparticles

Rulin Liu^{1,†}, Junjie Hao^{2,3†}, Jiagen Li¹, Shujie Wang¹, Haochen Liu², Ziming Zhou²,

Marie-Hélène Delville³, Jiaji Cheng⁴, Kai Wang^{2,*}, Xi Zhu^{1,*}

1 Shenzhen Institute of Artificial Intelligence and Robotics for Society (AIRS),

The Chinese University of Hong Kong, Shenzhen, Guangdong 518172, China

2 Department of Electrical and Electronic Engineering, Southern University of Science and Technology, Shenzhen, 518055, China

3 CNRS, Univ. Bordeaux, Bordeaux INP, ICMCB, UMR 5026, Pessac, F-33608,

France

4 School of Materials Science and Engineering, Hubei University, Wuhan 430062, China

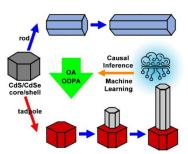
- † These authors contributed equally to this study.
- * Address correspondence to

wangk@sustc.edu.cn, zhuxi@cuhk.edu.cn

Keywords: Machine Learning, Causality Analysis, Entropy, Nucleation Theory, Quantum Dots, Heterostructure

ABSTRACT: The synthesis of CdSe/CdS core/shell nanoparticles was revisited with the help of a causal inference machine learning framework. The tadpole morphology with 1-2 tails was experimentally discovered. The causal inference model revealed the causality between the oleic acid (OA), octadecyl-phosphonic acid (ODPA) ligands and the detailed tail shape of the tadpole morphology. Further, with the identified causality, a neural network was provided to predict and directly lead to the original experimental discovery of new tadpole-shaped structures. An entropy-driven nucleation theory was developed to understand both the ligand & temperature dependent experimental data and the causal inference from the machine learning framework. This work provided a vivid example of how the artificial intelligence technology, including machine learning, could benefit the materials science research for the discovery.

TOC Graphic



The application of artificial intelligent technology, including machine learning (ML) in materials science, has recently generated intense research attention. ¹⁻⁶ The utilization of the ML technology could significantly reduce the workload and accelerate discoveries in both computational and experimental explorations. ⁶⁻⁸ As a numerical methodology, the ML technology is advantageous in extremely complicated systems, such as nanoscale materials.

In the present work, we revisited the CdSe/CdS core/shell quantum dots (QDs), a popular type of optoelectronic materials.⁹⁻¹¹ The morphologies of the CdSe/CdS core/shell structure are as various as hexagonal shape¹², triangular shape¹³⁻¹⁴, rod-like shape^{10, 15}, etc. The causality of the morphology formation is ingenious yet intricate, where the general understanding is *via* ligand passivation¹⁶⁻¹⁷ for the preference of ligands to bind with specific facets.¹⁸⁻¹⁹ One of the routine methods to investigate the ligand-surface binding phenomena and ligand-morphology causality is the density functional theory (DFT)¹⁶. However, except for the intrinsic drawbacks,²⁰ temperature also degrades the performance of DFT. The surface energy provided by DFT is a

thermodynamic quantity at zero K, yet most of the ligand-surface binding events occur above 500 K. Thus, at high temperature where QDs nucleate and grow, it is difficult to trust the temperature-morphology causality derived from the zero-temperature-level DFT methodology.

Since most of the properties of QDs are strictly dependent on the morphology, ²¹⁻²² an alternative and rational methodology is essential to understand the morphology causality. A causal inference in ML framework is hereby provided to explore the long-standing morphology causality problems in CdSe/CdS core-shell QDs. An ML framework was trained, which automatically found that oleic acid (OA) is the principal factor for tadpoles. A separate neural network was trained to predict the ODPA amount dependence of the number of tails. The predictions for the three-tail and five-tail tadpole structures agree well with the experimental data. Besides the causal inference, the growth mechanism of the tadpoles was also analyzed *via* entropy endorsed nucleation theory.

The synthesis was done in AIR-Chem²³ and takes two steps as in **Figure 1a-b**. Detailed descriptions can be found in **S1**. The process yielded tadpoles at 320 °C, and triangles at 260 °C.

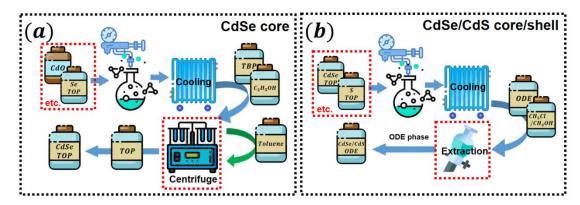


Figure 1, Experimental setup. Experimental setup of the synthesis process of the wurtzite CdSe cores, and the seeded growth to the CdSe/CdS core-shell structures. Red dotted frames in Figures 1a and 1b emphasize the difference between the two procedures. The full method is detailed in **S1**.

As an illustration, the tadpole morphology was obtained as mentioned in S1, and its characterizations by TEM versus time are shown in Figure 2. Figures 2a-g show the TEM images of the CdSe/CdS core-shell samples at different growth reaction times (0 s, 30 s, 1 min, 2 min, 4 min, and 8 min, respectively) in the presence of OA (1.02 ml) and ODPA (285 mg). The QDs first grew triangular, and a tail-like shape formed only after 2 minutes and kept on growing until 8 minutes, as shown in Figures 2f and 2g. The tail grows along the [0002] axis of the wurtzite CdS phase²⁴, the CdS phase is coherent with the head part and the CdSe core in the center of this head. The tadpole shape indicates a complete anisotropic growth pattern due to the routine selective facet coating of ligands.¹⁶

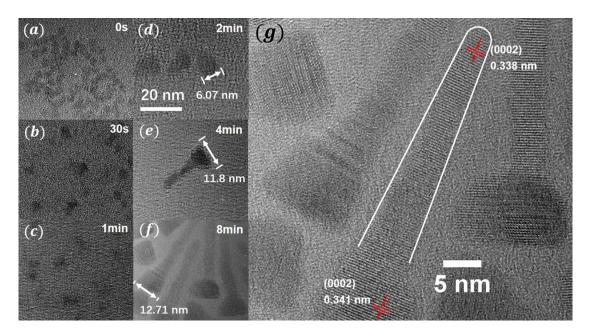


Figure 2, Time-dependent TEM images. TEM images of the tadpole morphology with respect to time: (a) core at 0 s, (b) 30 s, (c) 1 min, (d) 2 min, (e) 4 min, (f) 8 min. The nanoparticles grew first into a triangular shape, and the tail started to grow between 2 min and 4 min. The images before 2 min are too blurred to recognize any pattern. (g) A closer look at the 8 min products. The fringes show clearly that the tail grew along the (0002) facet.

DFT is one of the popular methodologies used to study the surface science problems^{16, 25-28}, which has successfully explain numerically some ligands-facets related anisotropic growth problem.^{16, 25-30} Yet, the "DFT success" is still an issue because of its treatment of temperature.

Here, we present the causal inference (CI) in ML framework³¹⁻³³ to identify the causal relationship between the experiments parameters and the morphology of the resulting nanomaterials. The CI takes into account the treatment $a \in \{0, 1\}$, the covariates X,

and the response y, to search for the most efficient covariates, and then uses methods such as inverse probability weighting³⁴ to eliminate the difference resulting from the distribution of these covariates. The causal relationship was evaluated via the standardized mean difference (SMD) of the covariates by dividing the data into two classes based on the treatment a, as the horizontal axis in **Figure 3a**. The SMD of two groups are

$$SMD = \frac{\mu_1 - \mu_2}{\sqrt{\sigma_1^2 + \sigma_2^2 - 2\sigma_{12}}},$$

where μ is the mean of the covariates in different classes, σ is the standard deviation, and the subscripts denote the classes of data. The two considered classes are the two denoted in the treatment a. This parameter is effective in expressing the size of the intervention effect of different covariates in X. The covariates with the highest standardized mean difference are believed to be the most relevant to the variability of the response y. In our case, X refers to the space of experimental parameters, y is the feature space of the QDs. Since the Bayesian probability $P(tadpole \mid ligand)$ of OA is the largest, the addition of OA was taken as the treatment a. Since the temperature and time are trivial as causalities for the QDs growth, we focus more on the ligands, such as the effects of ODPA on the number of tails, and study $P(n_{tail} \mid ODPA, T = 320^{\circ}C, t = 8min)$. As shown in Figure 3a, it was found that OA and ODPA were most relevant to the feature "tail or not". The different synthesis routes have been sketched in Figure 3a, where the coloring scheme is the same as in Figure

4a, denoteing the different growth mechanisms that are going to be analyzed in the following sections.

CI is capable of explaining the role of ligands in the formation of tadpole in a qualitative way. With more data acquired,³⁵⁻⁴⁴ the CI could extend its capability to the synthesis of other morphologies of other materials. Data collected from other publications are listed as Supplementary file **S4** "Experimental data" and some related features analyzed in **S5**.

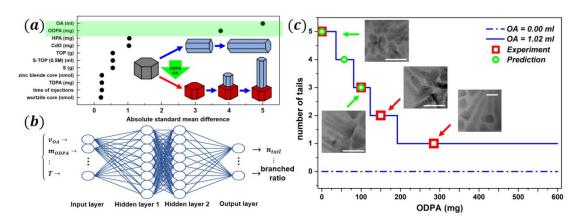


Figure 3, Causal Inference Analysis. (a), Causal relationship for the transition from quantum rod to tadpole. The SMD for OA and ODPA are the highest. The figure inserted shows that the growth route can be altered by the addition of OA and ODPA. The coloring scheme of the structures is the same as the ones shown in Figure 4a. (b) The detailed structure of the neural network. The *Hidden layer 1* and the *Output layer* both take a tanh activation function and a linear operation afterward, while the others are purely linear. The dimension of the neural network is $n \times 8 \times 8 \times m$, where n is the number of controls, and m is the number of QD features. In the experiment present, n = 4 refers to the amount of OA, ODPA, HPA, temperature, and r = 1

refers to the number of tails, respectively. **(c)**, Neural network predicted causal relationship between the amount of ODPA and the number of tails at 320 °C, trained with the experimental data. The solid red rectangles refer to the experimental data with tail numbers 1 and 2 and their corresponding TEM images; the green circles refer to the predicted structures made by the neural network with 3, 4 and 5 tails. The scale bars in the figures denote represent 20 nm. Full-sized TEM image is provided in **S6**.

The causal inference provided a qualitative understanding, while the neural network is quantitative. The neural network is shown in **Figure 3b**, whose structure provided was chosen because of its monotonicity and proper numerical range. The dimension of the neural network is $n \times 8 \times 8 \times m$, where n is the number of controls, and m is the number of QD features. In the present experiment, n = 4, m = 1, refer to OA, ODPA, HPA, temperature (the highest causality with the tadpole morphology in **Figure 3a**) as four inputs for n, and the number of tails as the output m. The loss function was MSE, and an adaptive backpropagation was conducted for training.

Since ODPA was found to be relevant to the number of tails, the prediction between the number of tails and the amount of ODPA was first examined using the trained network, in which OA was fixed at 0, 1.02 mL, and HPA at 0 g (HPA was found to be relevant to the formation of rods, and it did not appear in the formal experiments where tadpole was first discovered, so it was set to 0 g), and ODPA ranging from 0 to 600 mg. The result is plotted in **Figure 3c**, where the prediction agrees well with the

experimental data for 0, 1, 2 tails, and the prediction of new morphologies with 3, 4, 5 tails is also given.

According to the prediction in **Figure 3c**, 100 mg, 50 mg, 5 mg (picked from around the middle of the temperature range shown by the prediction curve) of ODPA was picked for validation. By applying these parameters, we successfully synthesized tadpole with 3 and 5 tails with 100mg and 5mg ODPA, the corresponding TEM images of the samples were shown beside the prediction data points in **Figure 3c**. The 4-tail samples were not obtained maybe because of the narrow tolerance window.

As shown in **Figure 3a**, OA was the main reason accounting for the formation of tadpoles. OA restricted the growth of the elongated part along the radial directions ([01 $\overline{10}$] and [10 $\overline{10}$]) of CdSe/CdS because of the OA/facets binding.⁴⁵ Thus, the vertical elongation carried on, while the horizontal does not, which resulted in the thinner tail. Another proof is that the more the number of tails of a tadpole QD, the larger its head (as shown in the TEM images provided in **Figure 3c**), since more tails provides larger surface area, which dissipates the OA/facet binding. ODPA is relevant to the growth along the vertical direction (0001), and thus the number of tails in a discrete manner, as shown by the blue lines of CI and neural network results in **Figure 3c**. Upon reaching one threshold by increasing the amount of ODPA, the number of the tails decreases by one. Quantitative calculations of the ligands can be found in **S7**.

Besides the ligands, the reaction temperature and time also matter. For the same experimental conditions as in **Figure 2**, but with temperature changed from 320 °C to 260 °C, the morphology would remain small and triangular even after 8 min (the time

dependent size is as depicted in the 260 °C curve in **Figure 4b**, and the morphology is similar to the one in **Figure 2d**). Since the tadpole occurs only at a higher temperature (320 °C), and the morphology is different from the Platonic solids described with enthalpy in classical theories, $^{46-48}$ and since the tadpole morphology is an hybrid of sphere and rod, the mechanism could be an entropy-driven two-step nucleation. The entropy considered in the present work is assumed to be proportional to the surface area of the QDs by facet dependent coefficients defined as $\chi(Jm^{-2}K^{-1})$. The expression of the free energy from the surfaces is thus:

$$\Delta G = \Delta H - T\Delta S = \sum_{i} (\gamma_i - T\chi_i) \Delta A_i \# (1)$$

where the subscript i refers to different facets, γ is the surface energy, ΔA is the change of the surface area. Thus, the morphology favoured under different temperature ranges can be specified by taking the minimum of the free energy ΔG .

Assuming the volume of the hexagonal prism QDs is V, and the additional volume from growth is $\Delta V (\Delta V \ll V)$. If ΔV grows to keep the hexagonal prism,

$$\Delta A_{hex} = \left(\frac{9\sqrt{3}}{16} + 3\right) \frac{16}{9\sqrt{3}} V^{\frac{2}{3}} \frac{\Delta V}{V} \# (2)$$

if ΔV induces a triangular morphology,

$$\Delta A_{tri} = \left(\frac{8}{3\sqrt{3}}\right)^{\frac{2}{3}} \left(\frac{3}{2} + \frac{3\sqrt{3}}{4}\right) V^{\frac{2}{3}} \frac{\Delta V}{V} \#(3)$$

if ΔV gives a rod-like morphology along the (0001) facet,

$$\Delta A_{rod} = \frac{16}{\sqrt{3}} \left(\frac{3\sqrt{3}}{8} \right)^{\frac{1}{3}} V^{\frac{2}{3}} \Delta V \# (4)$$

Eq (2) - (4) was plugged into eq (1) and ΔG minimized to determine the temperaturedependent favored facet. Note that ΔG depends on V and ΔV , so it is only upon reaching a critical volume that the growth turns to favor another morphology. One can thus define the most favored morphology as a function of the temperature, together with the size where the transition starts. **Figure 4a** shows the ΔG plot with V=158.47n m^3 , $\Delta V=1nm^3$. The gray, red and blue colors refer to different growth mechanisms analyzed above, and the directions and lengths of the green arrows denote the growth directions and rates. An explicit temperature dependence for the morphologies can be observed: below 140°C, hexagonal is favored; while a triangle phase emerges in the range 140 °C ~320 °C $^{50-52}$; the rod-like morphology emerges at 320 °C. This corresponds to the transition from triangular to rod-like morphology as shown by the TEM images in **Figure 2**.

Next, the kinetic mechanism of the tadpole morphology was investigated. The growth rates of the head and tail parts of the tadpole morphology are:

$$dr_{head} = dr,$$

$$dl_{tail} = \frac{\Theta(\Delta H_{diff} - T\Delta S_{diff})}{\sum_{n_{tail}=1}^{5} n_{tail} (\Theta(c_{ODPA} - c_{tail}^{n_{tail}}) - \Theta(c_{ODPA} - c_{tail}^{n_{-1}}))} dh$$

$$= \begin{cases} 0, & \text{if } \Delta H_{diff} < T\Delta S_{diff} \\ \frac{dh}{n_{tail}}, & \text{if } \Delta H_{diff} > T\Delta S_{diff}' \end{cases}$$
(6)

where

$$\frac{dr}{dt} = \frac{v_m c_\infty D_{head} k_{head}}{D_{head} + k_{head} r} \left(\frac{2\gamma_{head} v_m}{RTr} \right) \left(r \left(\frac{2\gamma_{head} v_m - c_\infty}{RT - c - c_\infty} \right)^{-1} - 1 \right)$$

$$\approx \frac{2\gamma_{head} v_m^2 c_\infty D_{head}}{RTr^2} \left(r \left(\frac{2\gamma_{head} v_m - c_\infty}{RT - c - c_\infty} \right)^{-1} - 1 \right), \tag{7}$$

$$\frac{dh}{dt} = \frac{v_m c_\infty D_{tail} k_{tail}}{D_{tail} + k_{tail} (\lambda - h)} \left(\frac{2\gamma_{tail} v_m}{RTr_{tail}} \right) \left(r_{tail} \left(\frac{2\gamma_{tail} v_m - c_\infty}{RT - c - c_\infty} \right)^{-1} - 1 \right). \tag{8}$$

 r_{head} in eq 5 is the radius of the tadpole's head, l_{tail} in eq 6 is the length of the tail; r and h in eqs 5 and 6 refer to the radius and height of the sphere and cylinder (detailed in S8) that form a simplified spherical morphology with a cylindrical tail, into which the tadpole morphology has been simplified. v_m is the molar volume of the material; c_{∞} is the bulk concentration of the monomers, c is the concentration of the monomers in solution; $D_{i}k$ are the diffusion and reaction coefficients, where the subscripts denote the part they refer to; γ is the surface energy, with subscripts referring to the different parts; λ is a considerable enough distance from a QD, where the concentration of the monomers c does not deviate from c_{∞} (detailed in S8).

The subscript diff of ΔH and ΔS in the numerator of eq 6 refers to the difference in them between the triangular and tadpole-like morphologies. Note that ΔH_{diff} refers to the chemical bonding energy which is almost temperature independent, thus, $T\Delta$ S_{diff} becomes more decisive at high temperatures. As can be observed from this model, an increasing monomer concentration would result in a longer and pointier tail, since that the growth route according to **Figure 4a** depends on only the temperature and ligands, and that the parameters D_ik_ic will increase with the monomer concentration.

The growth rate of the triangle morphology is

$$\frac{dx}{dt} = \frac{kc_{\infty}\gamma v_m^2}{RT} \left(\left(\frac{\gamma v_m \quad c_{\infty}}{RT \ c - c_{\infty}} \right)^{-1} - \frac{1}{h} + \frac{1}{x - 2l} \right), \tag{9}$$

$$\frac{dl}{dt} = \frac{Dkv_m \quad c_{\infty}\gamma v_m}{Dln\left(\delta/l\right) + kl \quad RT} \left(\left(\frac{\gamma v_m \quad c_{\infty}}{RT \ c - c_{\infty}} \right)^{-1} - \frac{1}{h} + \frac{1}{l - 2x} \right). \tag{10}$$

where x is the distance from the center of the triangular plane to the edge of the facet that grows faster, and l is the distance to the slower one, δ is defined the same as λ above, h here is different from above and denotes the thickness of the triangle. The models are detailed in **S9**.

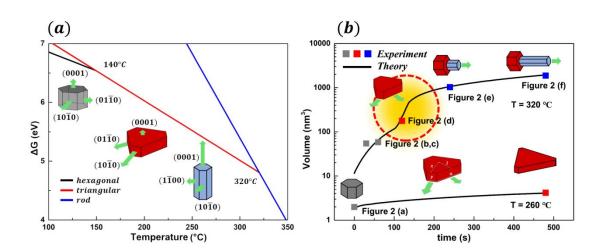


Figure 4, The entropy-driven nucleation theory. (a), ΔG of growth into different morphologies, plotted with respect to the temperature. Different colors in the figure refer to different growth mechanisms; the direction of the green arrows refer to the growth directions, and the magnitude of them refer to the respective growth rate. Since this plot was calculated with $V = 158.47nm^3$, as measured from the TEM in Figure 2d, it is only when the nanoparticle grows to this size that it begins to grow a tail, while below which the growth will be triangular, accounting for the triangular shape that occurred in the TEM image at 2 min. The surface entropy⁴⁹ $\chi_{uniform} = 10 \times 10^{-4}Jm^{-2}$, $\chi_{triangle} = 15 \times 10^{-4}Jm^{-2}$, $\chi_{tail} = 20 \times 10^{-4}Jm^{-2}$. The surface energies γ were taken as the same in the growth models. (b), The total volume of the tadpoles at $T = 320 \, ^{\circ}C$, together with the triangle at $T = 260 \, ^{\circ}C$. Both the theoretical and

experimental results are included. The data point labeled as **Figure 2a** shows up in both theoretical curves since it refers to the seed. The golden-colored red-dotted-line-circled emphasized area in the 320 °C curve denotes the emergence of the tail, as the second stage of the growth.

Figure 4b shows the numerical plot of eq 5 and 6, the time-dependent growth at 320 °C, and eq 9 and 10 at 260 °C. The colors in **Figure 4b** refer to the same growth mechanisms in Figure 4a. The experimental data shown in **Figure 2a** to **2f** are also labelled in Figure 4b, which agrees well with our theoretical model. By eq 6, the entropic competition between ΔH and $T\Delta S$ items in the Θ function is the critical factor for the system to choose the triangle or tadpole channel.

The emergence of the tadpole morphology requires a more aggressive volume increase, indicating a three-stage growth mechanism induced by the entropy. The three stages are: (1) the growth of the head to reach a critical volume at a given temperature, which will be slow and diffusion-controlled in later time; (2) the fast reaction controlled emergence of the tail; and (3) the slowing down of the tail's growth, where both the head and the tail end up in diffusion-controlled growth. This mechanism cannot be established without the help of entropy, for enthalpy analysis considers the competition amongst existing facets, rather than the appearance of a new one. The enthalpy analysis based on surface energy and chemical potential⁴⁷ can describe the morphology transition of nanoparticles, yet it is more like a ripening phenomenon of existing surfaces. The endorsement of entropy enables us to find another alternative growth

route for the nanoparticles besides the morphology transition from enthalpy competition.

To conclude, in this work, the causal inference has successfully learned from the existing experimental data of ours of 0, 1, 2-tail tadpoles, and managed to provide predictions for 3, 4, 5-tail tadpoles. Experiments were then conducted, and 3, 5-tail tadpoles were synthesized as expected. An entropy-driven three-stages growth mechanism was developed under the guidance of causal inference. This work provides a vivid example of how ML can help researchers in recognizing the pattern of the experimental data, and even in developing new knowledge.

ASSOCIATED CONTENT

Supporting Information:

Additional data and figures are included in the Supplementary Information. S1: Experimental methodology; S2: Characterization of tadpoles, S3: Drawbacks of DFT; S4: Experimental data (The experimental data from literatures, which were used for the training of CI, were formatted in an Excel-formatted PDF file and attached in an "S4. Experimental Data.pdf"), S5: CI for other morphologies, S6: Full TEM images of tadpoles, S7: Attachment of ligand molecules, S8: Derivation of growth in reduced dimensions, S9: Derivation and numerical calculation of triangular and tadpole

morphology. This material is available free of charge *via* the Internet at http://pubs.acs.org.

AUTHOR INFORMATION

Corresponding Authors:

Kai Wang – Department of Electrical and Electronic Engineering, Southern

University of Science and Technology, Shenzhen, 518055, China, orcid.org/00000003-0443-6955 Email: wangk@sustc.edu.cn

Xi Zhu – Shenzhen Institute of Artificial Intelligence and Robotics for Society (AIRS),

The Chinese University of Hong Kong, Shenzhen (CUHK-Shenzhen), Shenzhen,

Guangdong 518172, China, Email: zhuxi@cuhk.edu.cn

Authors

Rulin Liu — Shenzhen Institute of Artificial Intelligence and Robotics for Society

(AIRS), The Chinese University of Hong Kong, Shenzhen, Guangdong 518172, China

Junjie Hao — Department of Electrical and Electronic Engineering, Southern

University of Science and Technology, Shenzhen, 518055, China and CNRS, Univ.

Bordeaux, Bordeaux INP, ICMCB, UMR 5026, Pessac, F-33608, France

Jiagen Li - Shenzhen Institute of Artificial Intelligence and Robotics for Society (AIRS), The Chinese University of Hong Kong, Shenzhen, Guangdong 518172, China

Shujie Wang — Shenzhen Institute of Artificial Intelligence and Robotics for Society (AIRS), The Chinese University of Hong Kong, Shenzhen, Guangdong 518172, China

Haocheng Liu – Department of Electrical and Electronic Engineering, Southern
University of Science and Technology, Shenzhen, 518055, China

Ziming Zhou – Department of Electrical and Electronic Engineering, Southern
University of Science and Technology, Shenzhen, 518055, China

Marie-Hélène Delville – CNRS, Univ. Bordeaux, Bordeaux INP, ICMCB, UMR 5026, Pessac, F-33608, France

Jiaji Cheng – School of Materials Science and Engineering, Hubei University, Wuhan 430062, China, orcid.org/0000-0002-2663-7881. Email: jiajicheng@hubu.edu.cn

Author Contributions

R. Liu and J. Hao contributed equally to this work.

ACKNOWLEDGMENTS

This work is supported by the Shenzhen Fundamental Research Foundation (JCYJ20170818103918295, JCYJ20180508162801893), and the National Natural Science Foundation of China (grant no. 21805234 and 61875082). Also supported by funding (2019-INT018,2020-IND002) from Shenzhen Institute of Artificial Intelligence and Robotics for Society (AIRS). TEM data were obtained using

equipment maintained by Southern University of Science and Technology Core Research Facilities, and the authors would like to acknowledge the technical support from Dongsheng He and Yang Qiu in SUSTech CRF.

REFERENCE

- 1. Meredig, B. Five High-Impact Research Areas in Machine Learning for Materials Science. *Chem. Mater.* **2019,** *31* (23), 9579-9581.
- 2. Sun, W.; Zheng, Y.; Yang, K.; Zhang, Q.; Shah, A. A.; Wu, Z.; Sun, Y.; Feng, L.; Chen, D.; Xiao, Z.; Lu, S.; Li, Y.; Sun, K. Machine learning—assisted molecular design and efficiency prediction for high-performance organic photovoltaic materials. *Sci.Adv.* **2019**, *5* (11), eaay4275.
- 3. Granda, J. M.; Donina, L.; Dragone, V.; Long, D.-L.; Cronin, L. Controlling an organic synthesis robot with machine learning to search for new reactivity. *Nature* **2018**, *559* (7714), 377-381.
- 4. Hase, F.; Roch, L. M.; Kreisbeck, C.; Aspuru-Guzik, A. Phoenics: a Bayesian optimizer for chemistry. *ACS Cent. Sci.* **2018,** *4* (9), 1134-1145.
- 5. Bai, Y.; Wilbraham, L.; Slater, B. J.; Zwijnenburg, M. A.; Sprick, R. S.; Cooper, A. I. Accelerated discovery of organic polymer photocatalysts for hydrogen evolution from water through the integration of experiment and theory. *J. Am. Chem. Soc.* **2019**, *141* (22), 9063-9071.
- 6. Butler, K. T.; Davies, D. W.; Cartwright, H.; Isayev, O.; Walsh, A. Machine learning for molecular and materials science. *Nature* **2018**, *559* (7715), 547-555.

- 7. Kitchin, J. R. Machine learning in catalysis. *Nat. Catal.* **2018**, *1* (4), 230-232.
- 8. Li, F.; Thevenon, A.; Rosas-Hernández, A.; Wang, Z.; Li, Y.; Gabardo, C. M.; Ozden, A.; Dinh, C. T.; Li, J.; Wang, Y.; Edwards, J. P.; Xu, Y.; McCallum, C.; Tao, L.; Liang, Z.-Q.; Luo, M.; Wang, X.; Li, H.; O'Brien, C. P.; Tan, C.-S.; Nam, D.-H.; Quintero-Bermudez, R.; Zhuang, T.-T.; Li, Y. C.; Han, Z.; Britt, R. D.; Sinton, D.; Agapie, T.; Peters, J. C.; Sargent, E. H. Molecular tuning of CO2-to-ethylene conversion. *Nature* **2020**, *577* (7791), 509-513.
- 9. Alivisatos, A. P. Semiconductor clusters, nanocrystals, and quantum dots. *Science* **1996,** *271* (5251), 933-937.
- 10. Zavelani-Rossi, M.; Lupo, M. G.; Krahne, R.; Manna, L.; Lanzani, G. Lasing in self-assembled microcavities of CdSe/CdS core/shell colloidal quantum rods. *Nanoscale* **2010**, *2* (6), 931-935.

11. Nan, W.; Niu, Y.; Qin, H.; Cui, F.; Yang, Y.; Lai, R.; Lin, W.; Peng, X. Crystal

- structure control of zinc-blende CdSe/CdS core/shell nanocrystals: synthesis and structure-dependent optical properties. *J. Am. Chem. Soc.* **2012**, *134* (48), 19685-19693.

 12. Cassette, E.; Mahler, B.; Guigner, J.-M.; Patriarche, G.; Dubertret, B.; Pons, T. Colloidal CdSe/CdS dot-in-plate nanocrystals with 2D-polarized emission. *ACS Nano* **2012**, *6* (8), 6741-6750.
- 13. Sugunan, A.; Zhao, Y.; Mitra, S.; Dong, L.; Li, S.; Popov, S.; Marcinkevicius, S.; Toprak, M. S.; Muhammed, M. Synthesis of tetrahedral quasi-type-II CdSe–CdS core–shell quantum dots. *Nanotechnology* **2011**, *22* (42), 425202.

- 14. Song, X.; Vladislav, K.; Liu, C.; Xu, W. The preparation of anisotropic hybrid nanostructures based on CdSe and CdS by the ligand combination method. *Nanoscale* **2017,** *9* (11), 3934-3940.
- 15. Qin, J.; Hao, J.; Chen, W.; Deng, J.; Wang, D.; Xu, B.; Wu, D.; Wang, K.; Li, S.; Wen, Z. In 34-2: A Rapid, Highly Emissive Procedure Synthesize of Giant Pure Red Coreshell Quantum Rods by Using Modified Tributylphosphine-assisted Method, SID Symposium Digest of Technical Papers, Wiley Online Library: 2016; pp 428-431.
- 16. Puzder, A.; Williamson, A. J.; Zaitseva, N.; Galli, G.; Manna, L.; Alivisatos, A. P. The effect of organic ligand binding on the growth of CdSe nanoparticles probed by ab initio calculations. *Nano Lett.* **2004,** *4* (12), 2361-2365.
- 17. Manna, L.; Wang, L. W.; Cingolani, R.; Alivisatos, A. P. First-principles modeling of unpassivated and surfactant-passivated bulk facets of wurtzite CdSe: A model system for studying the anisotropic growth of CdSe nanocrystals. *J. Phys. Chem. B* **2005**, *109* (13), 6183-6192.
- 18. Peng, Z. A.; Peng, X. Mechanisms of the shape evolution of CdSe nanocrystals. *J. Am. Chem. Soc.* **2001,** *123* (7), 1389-1395.
- 19. Yin, Y.; Erdonmez, C.; Aloni, S.; Alivisatos, A. P. Faceting of nanocrystals during chemical transformation: From solid silver spheres to hollow gold octahedra. *J. Am. Chem. Soc.* **2006**, *128* (39), 12671-12673.
- 20. Lejaeghere, K.; Van Speybroeck, V.; Cottenier, S.; Van Oost, G. The ground state elemental crystals as a benchmark set for solid state DFT: intrinsic accuracy and code comparison. *Crit. Rev. Solid State Mater. Sci.* **2012**, *39*, 1.

- 21. Brus, L. Electronic wave functions in semiconductor clusters: experiment and theory. *J. Phys. Chem.* **1986**, *90* (12), 2555-2560.
- 22. Markovich, G.; Collier, C. P.; Henrichs, S. E.; Remacle, F.; Levine, R. D.; Heath, J. R. Architectonic quantum dot solids. *Acc. Chem. Res.* **1999**, *32* (5), 415-423.
- 23. Li, J.; Lu, Y.; Xu, Y.; Liu, C.; Tu, Y.; Ye, S.; Liu, H.; Xie, Y.; Qian, H.; Zhu, X. AIR-Chem: Authentic Intelligent Robotics for Chemistry. *J. Phys. Chem. A* **2018**, *122* (46), 9142-9148.
- 24. Wang, X.; Chen, S.; Thota, S.; Wang, Y.; Tan, H.; Tang, M.; Quan, Z.; Zhao, J. Anisotropic Arm Growth in Unconventional Semiconductor CdSe/CdS Nanotetrapod Synthesis Using Core/Shell CdSe/CdS as Seeds. *J. Phys. Chem. C* **2019**, *123* (31), 19238-19245.
- 25. Yamamoto, M.; Chan, C. T.; Ho, K.; Kurahashi, M.; Naito, S. First-principles calculation of the longitudinal phonon in the surface-normal direction of a zirconium (0001) slab: Localization mode at the subsurface. *Phys. Rev. B* **1996**, *53* (20), 13772.

 26. Rempel, J. Y.; Trout, B. L.; Bawendi, M. G.; Jensen, K. F. Density functional theory study of ligand binding on CdSe (0001),(0001), and (1120) single crystal relaxed and reconstructed surfaces: implications for nanocrystalline growth. *J. Phys. Chem. B* **2006**, *110* (36), 18007-18016.
- 27. Aruguete, D. M.; Marcus, M. A.; Li, L.-s.; Williamson, A.; Fakra, S.; Gygi, F.; Galli, G. A.; Alivisatos, A. P. Surface structure of CdSe nanorods revealed by combined X-ray absorption fine structure measurements and ab initio calculations. *J. Phys. Chem. C* **2007**, *111* (1), 75-79.

- 28. Kilin, D. S.; Prezhdo, O. V.; Xia, Y. Shape-controlled synthesis of silver nanoparticles: Ab initio study of preferential surface coordination with citric acid. *Chem. Phys. Lett.* **2008**, *458* (1-3), 113-116.
- 29. Bealing, C. R.; Baumgardner, W. J.; Choi, J. J.; Hanrath, T.; Hennig, R. G. Predicting nanocrystal shape through consideration of surface-ligand interactions. *ACS Nano* **2012**, *6* (3), 2118-2127.
- 30. Hu, H.; Reven, L.; Rey, A. First-principles density functional theory (DFT) study of gold nanorod and its interaction with alkanethiol ligands. *J. Phys. Chem. B* **2013**, *117* (41), 12625-12631.
- 31. Verma, T.; Pearl, J. *Equivalence and synthesis of causal models*. UCLA, Computer Science Department: 1991.
- 32. Spirtes, P.; Glymour, C. N.; Scheines, R.; Heckerman, D. *Causation, prediction, and search*. MIT press: 2000.
- 33. Pearl, J. Causal inference in statistics: An overview. Stat. Surv. 2009, 3, 96-146.
- 34. Mansournia, M. A.; Altman, D. G. STATISTICS NOTES Inverse probability weighting. *BMJ-BRIT MED J* **2016**, *352*.
- 35. Carbone, L.; Nobile, C.; De Giorgi, M.; Sala, F. D.; Morello, G.; Pompa, P.; Hytch, M.; Snoeck, E.; Fiore, A.; Franchini, I. R. Synthesis and micrometer-scale assembly of colloidal CdSe/CdS nanorods prepared by a seeded growth approach. *Nano letters* **2007,** *7* (10), 2942-2950.
- 36. Fiore, A.; Mastria, R.; Lupo, M. G.; Lanzani, G.; Giannini, C.; Carlino, E.; Morello, G.; De Giorgi, M.; Li, Y.; Cingolani, R. Tetrapod-shaped colloidal nanocrystals of II—

VI semiconductors prepared by seeded growth. *J. Am. Chem. Soc.* **2009,** *131* (6), 2274-2282.

- 37. Gafni, T.; Cohen, K. Learning in restless multi-armed bandits via adaptive arm sequencing rules. *arXiv preprint arXiv:1906.08120* **2019**.
- 38. Kunneman, L. T.; Zanella, M.; Manna, L.; Siebbeles, L. D.; Schins, J. M. Mobility and spatial distribution of photoexcited electrons in cdse/cds nanorods. *The Journal of Physical Chemistry C* **2013**, *117* (6), 3146-3151.
- 39. Manna, L.; Scher, E. C.; Li, L.-S.; Alivisatos, A. P. Epitaxial growth and photochemical annealing of graded CdS/ZnS shells on colloidal CdSe nanorods. *J. Am. Chem. Soc.* **2002**, *124* (24), 7136-7145.
- 40. Peng, X. Mechanisms for the shape-control and shape-evolution of colloidal semiconductor nanocrystals. *Adv. Mater.* **2003**, *15* (5), 459-463.
- 41. Peng, Z. A.; Peng, X. Nearly monodisperse and shape-controlled CdSe nanocrystals via alternative routes: nucleation and growth. *J. Am. Chem. Soc.* **2002**, *124* (13), 3343-3353.
- 42. Talapin, D. V.; Nelson, J. H.; Shevchenko, E. V.; Aloni, S.; Sadtler, B.; Alivisatos, A. P. Seeded growth of highly luminescent CdSe/CdS nanoheterostructures with rod and tetrapod morphologies. *Nano Lett.* **2007**, *7* (10), 2951-2959.
- 43. Wang, W.; Banerjee, S.; Jia, S.; Steigerwald, M. L.; Herman, I. P. Ligand control of growth, morphology, and capping structure of colloidal CdSe nanorods. *Chem. Mater.* **2007**, *19* (10), 2573-2580.

- 44. Wang, X.; Feng, Z.; Fan, D.; Fan, F.; Li, C. Shape-controlled synthesis of CdS nanostructures via a solvothermal method. *Crystal growth & design* **2010**, *10* (12), 5312-5318.
- 45. Shanavas, K.; Sharma, S. M.; Dasgupta, I.; Nag, A.; Hazarika, A.; Sarma, D. First-principles study of the effect of organic ligands on the crystal structure of CdS nanoparticles. *The Journal of Physical Chemistry C* **2012**, *116* (11), 6507-6511.
- 46. Lifshitz, I. M.; Slyozov, V. V. The kinetics of precipitation from supersaturated solid solutions. *J. Phys. Chem. Solids* **1961**, *19* (1-2), 35-50.
- 47. Sugimoto, T. Preparation of monodispersed colloidal particles. *Adv. Colloid Interface Sci.* **1987**, *28*, 65-108.
- 48. Wagner, C. Theorie der alterung von niederschlägen durch umlösen (Ostwaldreifung). *Z. Elektrochem.* **1961,** *65* (7-8), 581-591.
- 49. Molleman, B.; Hiemstra, T. Size and shape dependency of the surface energy of metallic nanoparticles: unifying the atomic and thermodynamic approaches. *PCCP* **2018**, *20* (31), 20575-20587.
- 50. Chen, W.; Chen, K.; Peng, Q.; Li, Y. Triangular CdS nanocrystals: rational solvothermal synthesis and optical studies. *Small* **2009**, *5* (6), 681-684.
- 51. Poulose, A. C.; Veeranarayanan, S.; Yoshida, Y.; Maekawa, T.; Kumar, D. S. Rapid synthesis of triangular CdS nanocrystals without any trap emission. *J. Nanopart. Res.* **2012,** *14* (4), 789.
- 52. Warner, J. H.; Tilley, R. D. Synthesis and Self-Assembly of Triangular and Hexagonal CdS Nanocrystals. *Adv. Mater.* **2005,** *17* (24), 2997-3001.