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Evolution of cluster free-energy during precipitation in concentrated binary alloys

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Abstract

Considering ordered and coherent Al3Li clusters embedded in concentrated AlLi alloys as a model case, precipitation is investigated using Atomistic Kinematic Monte Carlo (AKMC) simulations. The free energy of clusters is extracted from simulations as a function of their size and monitored from the initial disordered state until coarsening. It is found that all components of cluster free energy vary linearly with the chemical potential of monomers, a property easy to implement into classical precipitation models.

Keywords: precipitation, simulation, clusters, Monte Carlo, free energy

1. Introduction

Whatever the complexity of precipitation in modern multi-components alloys, the fundamental laws describing the behaviour of spherical and coherent precipitates in a model binary alloy remain an essential reference. The description of this ideal case slightly improved over a century, mostly through theoretical advances. Forty years ago, the development of numerical tools based on Statistical Physics brought new tools to explore this classical subject. Then, the computing capabilities became sufficient to confirm the validity of classical theories, mostly thanks to Atomistic Kinematic Monte Carlo simulations (AKMC); at least for alloys of low concentration and low supersaturation [1].

In the mean time, aluminium industry, for instance, raised questionings about the validity of classical approaches for alloys of non negligible concentrations and/or supersaturations, typical situations which were out of reach of numerical simulations until twenty years ago.

To investigate this question, using AKMC simulations as a reference, Al3Li clusters embedded in an aluminium matrix were chosen as a model case, for both industrial and academical reasons (e.g. the high solubility of this phase).

The properties of clusters (including fluctuations and precipitates) were first explored with an oversimplified atomic model limited to second near-neighbour bonds [2]. To move beyond this necessary simplification, a classical atomic potential accounting for both first- and second- near neighbour bonds (noted NN1 and NN2, respectively) has been proposed recently [3]. Furthermore, a method to directly extract the free energy of clusters from AKMC simulations has been generalised and applied to AlLi alloys at the solubility limit, in the range 50°C-300°C. In the present paper, this method is applied to clusters built during precipitation kinetics, in various conditions, in order to:

(i) check the validity of previous findings [2] using a more realistic potential,

(ii) try to express results in a form simple enough to be implemented into classical precipitation models like Cluster Dynamics (CD) [4] or KWN method [5].

It is reminded that in the Classical Nucleation Theory the free energy F_n of clusters of size *n* is based on the capillary approximation $F_n = G_\nu(n) + A(n)y(n)$ where $G_\nu(n)$ is the nucleation energy, γ the free surface energy and $A(n)$ the surface [6]. Note that this

description fails to describe small clusters while the form validated by Perrini et al. [7] is not limited by cluster size but requires 4 parameters (cf. next section). By analogy with the capillary approximation, the first term is usually identified with a volume term while the other terms can be grouped in a generalised surface term depending on cluster size [6].

In classical mean field approximations (e.g. ideal or regular solid solution), by construction, G_v varies during precipitation; if matrix frustration is taken into account, γ is no longer constant [6]. In both cases these changes directly arise from the evolution of cluster distributions. By consequent, although to our knowledge it is never explicitly discussed in the literature, the evolution of cluster free energy during precipitation is not a new concept. The present work is a contribution to build a more general theory, not limited by solute concentration, supersaturation or cluster size.

2. Methodology

2.1 A classical atomic model for AKMC simulations

AKMC simulations (see [1] for details) of the precipitation of δ' Al₃Li clusters (L1₂) structure) in aluminium have been performed with an atomic model model based on NN1 and NN2 [3]. It is mostly derived from the parametrization proposed by Garland and Sanchez [8] for the two effective pair interaction parameters, i.e. ω_2 =[1871J/mol] for NN2 and $\omega_1 = 2\omega_2$ for NN1. Although it is built to reproduce various thermodynamic and kinetic properties of this alloy, this atomic model should evolve, following the availability of DFT calculations. Nevertheless, it is very convenient for our present goal.

AKMC simulations were performed using a box of $200³$ or $500³$ atoms, depending on conditions and purposes. Figure 1 shows a typical sequence.

 $1 2 3 4 5 6$

Figure 1: a typical sequence of precipitation. T=85°C, C0=9%, 500³ sites. Only cluster of size larger than 100 (a) or 1000 (b) are shown. The colour indicates the number of solute neighbours. (a) early nucleation (b) coarsening

2.2 Definitions

The key quantity for precipitation is F_n , the free energy of clusters as a function of their size n (the number of solute atoms). However, for modelling purposes, the required quantity is the difference of free energy $F_{n+1}-F_n$ which can be fit with a 4 coefficients function [7]:

$$
F_{n+1} - F_n = a + b \left((n+1)^{2/3} - n^{2/3} \right) + c \left((n+1)^{1/3} - n^{1/3} \right) + d \ln \left((n+1)/n \right) \tag{1}
$$

To extract *a*, *b*, *c* and *d* from AKMC simulations, clusters are analysed following the 'CapRel' method based on the calculation of the so-called 'Capture' and 'Release' coefficients, $P_{n\rightarrow n+1}^*$ and $P_{n+1\rightarrow n}^*$ respectively (see [3] for details). The ratio of these two coefficients is related to the difference of free energy in Equation (1):

$$
\frac{P_{n\to n+1}^*}{P_{n+1\to n}^*} = \exp\left(\frac{h_1}{k_B T}\right) \exp\left(-\frac{F_{n+1} - F_n}{k_B T}\right)
$$
\n(2)

with h_l the monomer enthalpy, *T* the temperature and k_B the Boltzmann constant. The symbol '*' indicates that these coefficients account for the risk of coagulation between a cluster and its neighbours, when it captures a new solute atom. It has been shown that this ratio is the right quantity to be used in Cluster Dynamics (CD) to reproduce cluster distributions at equilibrium, which is an elementary test of validity [3].

Another fundamental quantity required to analyse results is the chemical potential of monomers u_1 [2]:

$$
\left(\frac{\mu_1}{kT}\right) = \left(\frac{h_1}{k_B T}\right) + \ln\left(\frac{C_1}{M_1}\right) \tag{3}
$$

with $C₁$ the concentration of monomers and $M₁$ the concentration of Al sites whose NN1 and NN2 are all occupied by Al atoms. In the following, \bar{x} denotes the quantity *x* normalised by $k_B T$; to simplify we note $\overline{a}^* = \overline{a} - \overline{h}_1$ and $\overline{\mu}_1^* = \overline{\mu}_1 - \overline{h}_1$. At the solubility limit $a=\mu$, thus $\overline{a}^* = \ln(C_1/M_1)$ [2-3]. Finally, with these notations, the quantity to fit writes:

$$
\frac{P_{n\to n+1}^{*}}{P_{n+1\to n}^{*}} = \exp\left(-\left(\overline{a}^{*} + \overline{b}\left((n+1)^{2/3} - n^{2/3}\right) + \overline{c}\left((n+1)^{1/3} - n^{1/3}\right) + \overline{d}\ln\left((n+1)/n\right)\right)\right)
$$
\n(4)

3. Results

This section examines the evolution of coefficients \vec{a}^* , \vec{b} , \vec{c} and \vec{d} during precipitation kinetics, analysing clusters with the 'CapRel' method [3].

3.1. General behaviour

Starting from a random distribution, a supersaturated system exhibits first a transient regime before reaching quasi-equilibrium (QE), as previously observed with a simplified atomic potential [2]. QE could be defined as a stage such that average properties of clusters are consistent with their distribution, like at true equilibrium [2]. Note that the onset of QE can be only assessed [2-3] but we checked that its choice is

not critical for our purpose.

To compare different situations, it is convenient to report $P_{n\rightarrow n+1}^* / P_{n+1\rightarrow n}^*$ versus the excess of chemical potential $\Delta \mu = \vec{u}_1(t) - \vec{u}_1(\infty)$, where $\vec{u}_1(\infty)$ is simply the value of $\vec{\mu}_1$ \mathbf{r}_1^* at the solubility limit (SL), supposed to be known [3].

In Figure 2, we have reported the measured values of $P_{n\rightarrow n+1}^* / P_{n+1\rightarrow n}^*$ for several intermediate states between the beginning of QE ($\Delta \mu$ =0.587) and the asymptotic state (SL) for T=200 $^{\circ}$ C, C₀=10%. As can be seen in Figure 2, in these conditions, the change of cluster properties during precipitation is rather large, which is likely to be important for the nucleation and growth stages.

Figure 2: Evolution of $P_{n\rightarrow n+1}^* / P_{n+1\rightarrow n}^*$ during precipitation for T=200°C, C_0 =10% (simulation box: 200³ sites). The upper bold curve "SL" refers to the solubility limit $(C_0=6.3\%)$ and the dashed line represents its asymptote. Symbols are related to measures performed on a given distribution of clusters obtained by AKMC simulations. The plain curves are the fits corresponding to the symbols of same colour. The lowest curve "<Eq." is taken in the transient stage preceding QE.

In Figure 3 we have reported the values of \vec{a} , \vec{b} , \vec{c} and \vec{d} versus $\Delta \mu$ corresponding to the fits shown in Figure 2. Of course, there is no 'exact' solution for any of these fits but many approximative solutions. Thus, any justified simplification is welcome to secure the fitting procedure. For instance, at 200 °C $\vec{a} \sim \Delta \mu$, like at the solubility limit, thus to simplify the fit we imposed this value. Generally speaking, the challenge is not to find the best fit for each cluster distribution but to find the best compromise between accuracy and simplicity, convenient for the whole range of precipitation kinetics. It is worth noting that whatever the simplifications the consequences for the coarsening stage are negligible, because the $P_{n \to n+1}^* / P_{n+1 \to n}^*$ curve becomes necessarily very close to the asymptote (see Figure 2) which can be known with a good accuracy [3].

The lowest curve (open red symbols) in Figure 2 is an example of early state, taken before reaching QE, i.e. when Equation (2) is not yet fulfilled. As can be seen, this curve exhibits a very different shape, thus its fit would require a very different combination of parameters. Using such a state as starting point in Cluster Dynamics, for instance, to predict the evolution of cluster distributions would require a too complicated model. On the contrary, this task is fairly easy if the initial state is taken around the onset of QE, as shown in Figure 3: all components of F_n vary linearly with $\Delta \mu$.

Figure 3: Evolution of the different coefficients used to fit $P_{n\rightarrow n+1}^{*} / P_{n+1\rightarrow n}^{*}$ vs. $\Delta \mu$ during precipitation, in two very different cases of high concentration**.** The beginning of QE is at the right end of curves while the left one corresponds to the solubility limit.

(a) (b)

This property of $P_{n\to n+1}^* \not P_{n+1\to n}^*$ vs. $\Delta \mu$ means that F_n^* can be written as a combination of both initial (QE) and final (SL) solutions:

$$
F_n^*(\lambda) = \lambda F_n^*(QE) + (1 - \lambda) F_n^*(SL)
$$
\n⁽⁵⁾

where λ is simply the ratio $\Delta \mu / (\mu(QE) - \mu(SL))$, varying from 1 to 0 (at *t* infinite).

3.2. Influence of temperature

Figure 2 is compared with two other situations of similar super-saturations: T=250°C and C_0 =10.5% (Figure 4a), T=85°C and C_0 =4.75% (Figure 4b). Like in Figure 2, the lower and upper curves correspond to the onset of QE and the solubility limit (SL), respectively. In addition, we have reported a few intermediate curves from nucleation to coarsening. For both cases we carefully fit the first curve, around the onset of QE, then

we assumed that coefficients \vec{a} , \vec{b} , \vec{c} and \vec{d} vary linearly with $\Delta \mu$, as found in the previous case (Figure 2). All other fits shown in Figures 4a-b follow this principle. As can be seen, these fits are in good agreement with measured data. The largest difference (a few %) with the best solution is observed in Figure 4a in the range *n*=5-30. The analysis of cluster distributions suggests that this small discrepancy is associated with a percolating heterogeneous phase containing up to 300000 connected solute atoms (100000 for $\Delta \mu$ =0.264 in Figure 4b) which coexists with a classical cluster distribution. This percolating phase appears during the nucleation stage and disappears before the coarsening stage, a behaviour which might be related to spinodal decomposition [8-9]. Although this behaviour might be interesting to explore, it is beyond the scope of the present work. However, it is interesting to note that even in such complex situations, the proposed procedure still provides good results for true clusters.

As can be seen in Figure 4a-b, although the temperature difference between these two cases is less than 200°C, the curves exhibit very different shapes and take very different values. This behaviour makes extrapolation between two known temperatures rather difficult.

Figure 4: Evolution of $P_{n\rightarrow n+1}^* / P_{n+1\rightarrow n}^*$ during precipitation in two cases of comparable supersaturation (C_0-C_{SL}) (i.e. 2.64% in (a) and 2.05% in (b)). In both cases the lower curve is the best fit of AKMC data (symbols) for the beginning of QE, while the upper curve (SL) refers to the solubility limit. Thin lines in (a) are the best fits of AKMC data.

3.3. Influence of concentration

In Figure 5, for two values of $\Delta \mu$, taken at early nucleation and early coarsening, respectively, we reported $P_{n\rightarrow n+1}^* / P_{n+1\rightarrow n}^*$ vs. $\Delta \mu$ at T=200°C for two close concentrations. As can be seen in Figure 5 the difference between two curves of same $\Delta \mu$ is small and the shape of curves is preserved. This suggests some relation between curves related to different concentrations, at constant temperature. Figure 5 shows also that this effect is obviously more important for nucleation and growth than for coarsening.

Figure 5: Influence of a small difference of solute concentration on $P_{n\rightarrow n+1}^* / P_{n+1\rightarrow n}^*$. Symbols are related to AKMC simulations while plain curves of same colour are the corresponding fits. "SL" corresponds to the solubility limit.

Figure 6 compares two situations of very different concentrations at T=85°C: C_0 =9%, reported for three values of $\Delta \mu$ (0.682, 0.516 and 0.34) and C_0 =4.75% for $\Delta \mu$ =0.34. First we fit the three first cases with the solution obtained for C₀=9% and reported in Figure 2b. Then to fit the fourth case with the same solution, we had to set Δμ~0.34/1.5. In other words, for a given temperature, the lower the initial excess of chemical potential $\Delta \mu$, the faster F_n converges towards its asymptotic value. To confirm the generality of this property, all curves in figure 4b for $C_0=4.75\%$ were fit with the data reported in Figure 3b for $C_0=9\%$, using $\Delta \mu/1.5$ instead of $\Delta \mu$ (the values at $\Delta \mu=0$ are unchanged by construction).

The quality of these fits proves that the relations between the different components of $P_{n\to n+1}^*$ / $P_{n+1\to n}^*$ are controlled only by temperature. Thus, to guess $P_{n\to n+1}^*$ / $P_{n+1\to n}^*$ for a concentration C_i , knowing the result at a concentration $C_i>C_i$, it is sufficient to determine the corresponding value of $\lambda(C_i)$ <1 at the beginning of QE for the new case C_i .

For this reason, it is preferable to spend computation time on a case of large supersaturation, e.g. $C_0=9\%$ here, which provides valuable information even with a $200³$ simulation box. Then, to explore a smaller concentration, e.g. $C_0=4.75\%$ here, it is worth using a large simulation box to get the required information about clusters at the onset of QE, the earlier possible starting point for classical models.

Figure 6: Influence of solute concentration on $P_{n\to n+1}^* / P_{n+1\to n}^*$. Symbols are related to AKMC simulations while plain curves of same colour are the corresponding fits. "SL" corresponds to the solubility limit. Note the magenta curve related to a concentration half of others.

3.4. General trends

No general solutions for the variations of \vec{a}^* , \vec{b} , \vec{c} and \vec{d} with $\Delta \mu$ as functions of temperature can be provided because of the influence of concentration. Nevertheless some general trends emerge.

As already mentioned, for T $\geq 200^{\circ}$ C it is easy to impose $\bar{a}^* = Ln(C_1/M_1)$ which means δ/\bar{a}^* $\delta(\Delta \mu)=1$. On the contrary, at lower temperatures, at the onset of QE, \bar{a}^* is already close to its asymptotic value, then δ/\bar{a}^* $\delta(\Delta \mu)$ tends towards 0. In other words, while \bar{a}^* instantaneously adjust with $\Delta \mu$ at high temperature, \bar{a}^* become less and less sensible to $\Delta \mu$ and concentration when T decreases. The evolution of δ^{b} / $\delta(\Delta \mu)$ is more dependent on concentration, but its range of variation at a given temperature remains fairly small. The evolution of $\delta^{\bar{c}}$ /δ(Δμ) is roughly similar to that of - $\delta^{\bar{b}}$ /δ(Δμ) with a

much higher dependence on the concentration, which leads to large variations. Finally, δ ^{\bar{d}} / δ ($\Delta \mu$) is weakly sensible to the concentration and its values remain of the order of a few units.

4. Conclusion

In this paper, some previous findings about the evolution of cluster free energy during precipitation [2] have been confirmed with AKMC simulations, using a more classical potential. The 'CapRel' method [3] has been used to extract the free energy of clusters during precipitation kinetics.

For the first time it is shown that the evolution of cluster free energy F_n and its components are linear functions of $\Delta \mu$, the excess of chemical potential of monomers. The simplicity of this result makes it easy to implement into classical precipitation models. Ideally, Δμ should be managed simultaneously in a consistent way; it depends on M_1 (Equation (3)), a variable closely related to the notion of cluster volume (discussed in a next paper). It is also shown how the information required to upgrade classical models can be obtained while minimizing the amount of atomistic calculations.

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