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# A quantitative study of the morphology of montmorillonite filled thermosets based on a tailor made homogenisation model

Mélissa Poncet-Malige, Stéphane Corn\*, Aurélie Taguet, José-Marie Lopez-Cuesta

Centre des Matériaux des mines d'Alès, C2MA, École des mines d'Alès, 6 avenue de Clavières, F-30319 Alès Cedex, France

## A B S T R A C T

Thermoset/montmorillonite nanocomposites were fabricated and their elastic modulus was measured using experimental modal analysis. The morphology of the nanocomposite was considered as a distribution of several components: exfoliated clay platelets, intercalated clay layers, primary particles and clay agglomerates. A novel homogenisation model, which involves a five-phase sequence based on the Halpin–Tsai equations, was developed to calculate the elastic modulus of the nanocomposites. This model was then used to quantify the morphology of the nanocomposites by back calculating the exfoliation, intercalation and agglomeration fractions from the measured values of the elastic modulus. Additionally, this approach led to quantify the efficiency of the fabrication process, which proved to be optimal for 2.5% clay content.

### Keywords:

- A. Particle-reinforcement
- B. Elasticity
- C. Computational modelling
- E. Thermosetting resin

## 1. Introduction

In recent years, nanoclays have attracted great interest in both academic and industrial research as they potentially permit a significant enhancement at low filler content [1]. Furthermore, they can exhibit added multifunctionality such as fire reaction, thermal stability and gas permeability barrier properties. For the last twenty years, layered silicate polymer nanocomposites have been specifically studied [1]. The most widely used layered silicates are based on montmorillonite, which is composed of extremely thin clay platelets that exhibit large surface areas and high aspect ratios. The elastic modulus of these platelets is very high compared to that of the polymer matrix. In order to obtain nanocomposites with enhanced properties compared to conventional composites, the natural montmorillonite has to be organically modified [2], for example by a quaternary ammonium salt bearing long carbon chains. Hence, the dispersion of the clay platelets is improved and exfoliated structures can occur. The reinforcement effect of silicate layers strongly depends on the degree of dispersion of the platelets and the structure of the nanocomposite [3–5]. Polymer/clay composites morphology can be divided into four categories [6]: conventional miscible composite, intercalated nanocomposite, exfoliated nanocomposite and partially intercalated and partially exfoliated nanocomposite. Generally, the

structure of clay/polymer nanocomposites is partially intercalated and partially exfoliated, with intercalated clusters and individual platelets randomly distributed in the matrix.

The development of nanocomposites requires a better understanding of the structure–properties relationship and the ability to make predictions using adapted micromechanical models. Continuum models provide a rapid assessment of the key factors that control the reinforcement and the mechanical properties of conventional particle-polymer composites. Among the existing models, Halpin–Tsai and Mori–Tanaka homogenisation schemes [7,8] were used to evaluate the properties of nanocomposites [9]. The mechanical behaviour of polymer/clay ‘ideal’ nanocomposites (e.g. assuming the complete exfoliation of clay platelets, their uniform dispersion and a perfect bonding between the matrix and the particles) can be predicted rather well with these models. But, due to the inherent complexity of the ‘real’ nanocomposite structure, discrepancies occur. Therefore, it is necessary to adapt existing models in order to take into account a realistic morphology of the nanocomposite [10–13]. In this paper, the reinforcing effect of an organomodified montmorillonite dispersed in a vinyl ester matrix is studied and a new micromechanical model, based on the Halpin–Tsai equations, is proposed to calculate the Young’s modulus of polymer/layered silicate nanocomposites. Finally, by updating the morphological parameters of the model to match the experimental values of the elastic modulus, this new homogenisation scheme is used to predict the microstructure of the nanocomposites.

\* Corresponding author.

E-mail address: [Stephane.Corn@mines-ales.fr](mailto:Stephane.Corn@mines-ales.fr) (S. Corn).

## 2. Experimental characterisation

### 2.1. Materials

Ashland Derakane<sup>®</sup> Momentum<sup>™</sup> 411-350 vinyl ester resin was used as the matrix. The clay material used was Cloisite<sup>®</sup> 30B supplied from Southern Clay Products. It is a natural montmorillonite mineral modified with methyl tallow bis-2-hydroxyethyl quaternary ammonium salt. The montmorillonite particles are disk-shaped stacks of thin silicate layers with a diameter ranging from 100 nm to several micrometres. Cloisite<sup>®</sup> 30B typical physical properties are the following: a layer spacing of 1.85 nm and a specific gravity of 1.98 for the clay particles and of 2.6 for an individual layer.

### 2.2. Processing

Five compositions of nanocomposites, containing the montmorillonite weight fraction of 0, 1.25, 2.5, 3.75 and 5 wt%, were performed in this study. Suspensions of organomodified clay in resin were manually stirred for 5 min, then mechanically stirred with a VELP stirrer model LH, followed by ultrasonic mixing with a Branson digital sonifier<sup>®</sup> S-450D. The optimised process parameters used are given in Table 1. The catalyst system used was 1.5 phr<sup>1</sup> of methyl ethyl ketone peroxide and 0.30 phr of cobalt naphthenate (3% in styrene). Mixtures were cast into silicone moulds to form blocks of 240 mm × 40 mm × 6 mm. The blocks samples were then cured for 3 h at 80 °C.

### 2.3. Morphology

The structure of the nanocomposites was characterised by X-ray diffraction (Bruker AXS D8 ADVANCE) and scanning electron microscopy (Hitachi S-4300 ESEM). The d-spacing was obtained from X-ray diffraction. Cryo-fractured pieces of test specimens and cryo-microtomes of test specimens were used as SEM samples. Suspensions of organomodified clay in resin, filled in a hollowed specimen holder, were used for the X-ray diffraction analysis.

### 2.4. Elastic parameters measurement set-up

An experimental vibration technique called ‘modal analysis’ was used to determine the viscoelastic parameters (elastic modulus, and loss tangent not presented here) of the nanocomposite materials. The set-up simulates free boundary conditions by supporting the sample with soft suspensions [14]. The sample, a solid block, is excited in free vibration by an impulse hammer and its response is monitored by an accelerometer. The excitation and response are fed into the Fast Fourier Transform (FFT) analyser which computes and displays the frequency response curve. The natural frequencies of the sample are given by the series of peaks in the frequency response curve. The software MODAN,<sup>2</sup> developed by the FEMTO-ST laboratory, from University of Franche-Comté, was used to extract natural frequencies and damping ratios from the frequency response function. Finite element model of each sample was correlated with the modal analysis results to identify the Young’s modulus and Poisson’s ratio of the materials. The samples used for modal analysis were parallelepipedic specimens with dimensions being (240 ± 0.5) mm × (40 ± 0.5) mm × (6 ± 0.02) mm. The accurate dimensions of each sample were updated in the finite element model used to calculate the elastic modulus. The elastic measurement were conducted at room temperature (23 °C).

**Table 1**

Parameters of the mechanical stirring and of the ultrasonic mixing.

Weight fraction (%)	Mechanical stirring		Ultrasonic mixing	
	Speed (rpm)	Time (min)	Power (W)	Time (min)
1.25	1200	60	72	30
2.5	1500	60	72	45
3.75	1500	90	80	45
5	1500	120	80	60

## 3. Results and modelling

### 3.1. Morphology

As shown in Figs. 1 and 2, the nanocomposite materials contain montmorillonite primary particles with a layer-spacing of 1.82 nm (which is very close to the supplied data), and montmorillonite intercalated particles with a layer-spacing of 3.81 nm. Moreover, agglomerates of montmorillonite were found in all the studied formulations (Figs. 3 and 4).

### 3.2. Modelling procedure

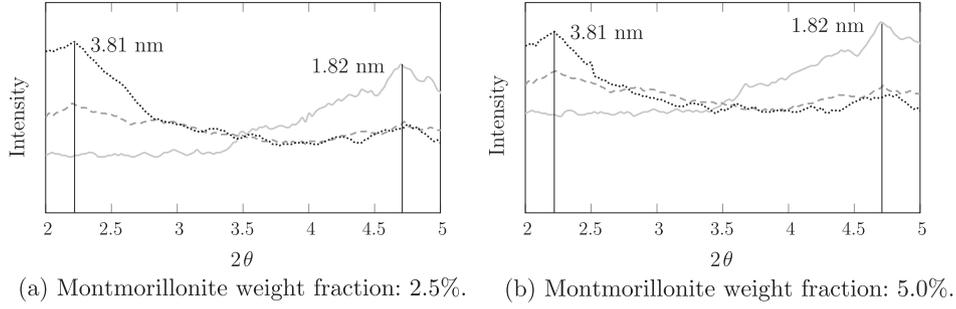
The whole structure of a polymer/clay nanocomposite is generally described as partially intercalated and partially exfoliated, but as seen with the morphological characterisation, the structure of the nanocomposite materials in this study also contains primary particles, with a layer-spacing of 1.82 nm, and clay agglomerates. The structure of the nanocomposites is represented by a five-phase model, including the thermosetting matrix, exfoliated clay platelets, intercalated clay layers, primary particles and clay agglomerates. The proposed model is adapted from the work of Luo and Daniel [10].

#### 3.2.1. Assumptions of the model

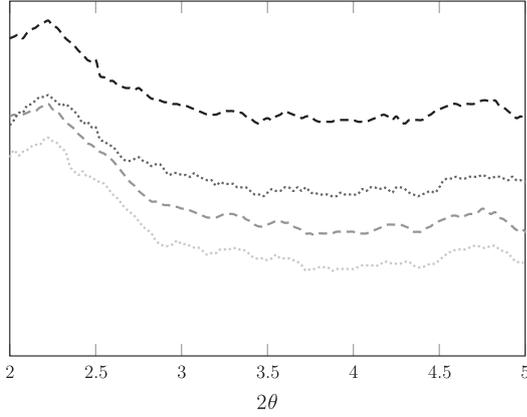
The matrix and the filler are assumed to exhibit linearly elastic and isotropic behaviour and to be firmly bonded. Platelets are assumed to be disks of diameter  $L$  and thickness  $d_s$  (Fig. 5a). It is assumed that a primary particle (Fig. 5b) is a stack of  $N_0$  non intercalated silicate layers, with a layer spacing  $d_0 = 1.82$  nm. It is assumed that an intercalated particle (Fig. 5c) is a stack of  $N$  intercalated silicate layers, with an interlayer spacing  $d_{001} = 3.81$  nm. And finally it is assumed that a montmorillonite agglomerate (Figs. 5d and 4) is a spherical cluster of primary particles. The exfoliated platelets, primary particles, intercalated particles and clusters of montmorillonite are assumed to be homogeneously distributed and randomly oriented in the matrix. For the exfoliated platelets, primary and intercalated particles, the axes parallel to the platelets are named the parallel  $\parallel$  directions and the axis perpendicular to the platelets is named the transverse  $\perp$  direction (cf. Fig. 5a). The model built in this study is based on the distribution of montmorillonite platelets given in Fig. 6: the clay volume fraction  $\phi_r$  is divided into a volume fraction  $\phi_e$  of exfoliated platelets, characterised by the exfoliation fraction  $f_e$ , and a volume fraction  $\phi_{ne}$  of non exfoliated particles. The volume fraction of non exfoliated particles is divided into a volume fraction  $\phi_{pi}$  of intercalated particles and a volume fraction  $\phi_{pp}$  of primary particles and the intercalation fraction of non exfoliated clay particles is noted  $f_i$ . The volume fraction of primary particles is divided into a volume fraction  $\phi_{ppna}$  of non agglomerated primary particles and a volume fraction  $\phi_{ppa}$  of agglomerated primary particles and the agglomeration fraction of the primary particles is noted  $f_a$ .

<sup>1</sup> Parts per hundred resin moulding compound.

<sup>2</sup> Pr. Emmanuel Foltête – ENSMM, Département de Mécanique Appliquée – 24 chemin de l’Épitaphe – 25000 Besançon, France.



**Fig. 1.** X-ray diffractograms: — manual mixing - - - mechanical stirring; ··· ultrasonic mixing.



**Fig. 2.** X-ray diffractograms after ultrasonic mixing for different montmorillonite weight fractions: ····· 1.25% - - - 2.5% ··· 3.75% - - - 5%.

### 3.2.2. Building of the homogenisation model

The elastic modulus  $E$  of a nanocomposite consisting of a matrix and platelet-like fillers with a completely three-dimensional random orientation and homogenous distribution can be, using the laminated model [15], calculated as:

$$E = 0.49E_{\parallel} + 0.51E_{\perp} \quad (1)$$

where  $E_{\perp}$  and  $E_{\parallel}$  are the composite moduli perpendicular (transverse) and parallel to the major axis of the filler, respectively. The Eq. (1) [9] is an approximation proposed by van Es et al. using laminates theory [16,17]. The composite elastic modulus perpendicular

to the filler major axis,  $E_{\perp}$ , and the composite elastic modulus parallel to the filler major axis,  $E_{\parallel}$ , can be calculated using Halpin-Tsai equations:

$$E_{\perp} = \frac{1 + \zeta_{\perp} \eta_{\perp} \phi_f}{1 - \eta_{\perp} \phi_f} E_m \quad E_{\parallel} = \frac{1 + \zeta_{\parallel} \eta_{\parallel} \phi_f}{1 - \eta_{\parallel} \phi_f} E_m \quad (2)$$

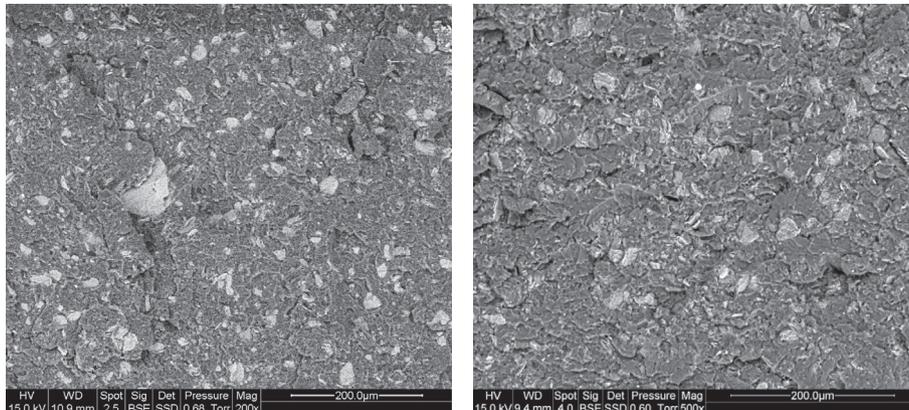
where  $E_m$  is the elastic modulus of the matrix,  $\phi_f$  is the volume fraction of filler and where the parameters  $\eta_{\perp}$  and  $\eta_{\parallel}$ , in the directions perpendicular and parallel to the filler are given by the following equations:

$$\eta_{\perp} = \frac{\frac{E_f}{E_m} - 1}{\frac{E_f}{E_m} + \zeta_{\perp}} \quad \eta_{\parallel} = \frac{\frac{E_f}{E_m} - 1}{\frac{E_f}{E_m} + \zeta_{\parallel}} \quad (3)$$

where  $E_f$  is the elastic modulus of the filler and where  $\zeta_{\perp}$  and  $\zeta_{\parallel}$  are the shape parameter of the filler, in the directions transverse and parallel to the filler, respectively.

For a nanocomposite material containing more than one variety of filler, the modelling procedure becomes a multi-step approach. As the nanocomposite materials in this study contain exfoliated platelets, intercalated and primary particles as well as agglomerates (Fig. 5), the modelling procedure is divided in a 4 steps approach:

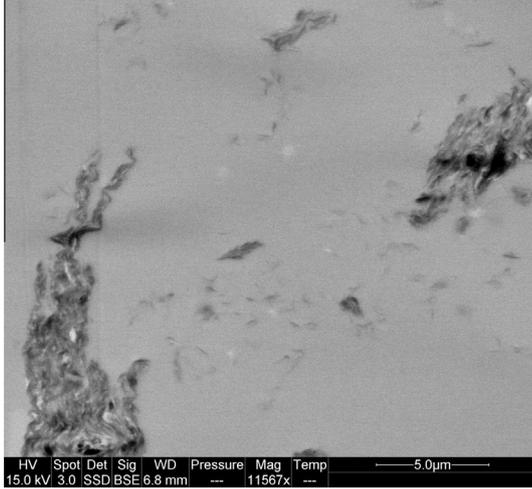
1. Halpin-Tsai model is applied to the matrix and to the exfoliated platelets to determine a parallel ( $E_{\parallel}$ ) and a transverse ( $E_{\perp}$ ) elastic moduli, then the equation proposed by van Es (cf. Eq. (1)) is used to determine the elastic modulus  $E_1$  of the first effective matrix.



(a) Montmorillonite weight fraction: 2.5%.

(b) Montmorillonite weight fraction: 5.0%.

**Fig. 3.** Momentum™ 411-350/Cloisite® 30B composites micrographs.



**Fig. 4.** Momentum™ 411-350/Cloisite® 30B composites STEM micrograph – Montmorillonite weight fraction: 2.5%.

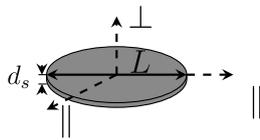
2. The same principle is applied to the first effective matrix and to the effective intercalated particles, to determine the elastic modulus  $E_2$  of the second effective matrix.
3. Then the same is done with the second effective matrix and the non agglomerated effective primary particles, to determine the elastic modulus  $E_3$  of the third effective matrix.
4. Finally, one last repetition, with the third effective matrix and the agglomerated primary particles allows determining the elastic modulus  $E_c$  of the nanocomposite.

For each step  $i$  of the procedure,  $i$  ranging from 1 to 4, the elastic modulus  $E_i$  is calculated using the following equations:

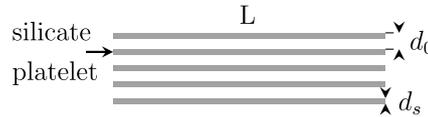
$$E_i = 0.49E_{i\parallel} + 0.51E_{i\perp} \quad (4)$$

$$E_{i\parallel} = \frac{1 + \eta_{i\parallel} \zeta_{i\parallel} \phi_i}{1 - \eta_{i\parallel} \phi_i} E_{i-1} \quad E_{i\perp} = \frac{1 + \eta_{i\perp} \zeta_{i\perp} \phi_i}{1 - \eta_{i\perp} \phi_i} E_{i-1} \quad (5)$$

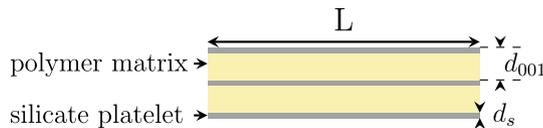
$$\eta_{i\parallel} = \frac{\frac{E_{fi}}{E_{i-1}} - 1}{\frac{E_{fi}}{E_{i-1}} + \zeta_{i\parallel}} \quad \eta_{i\perp} = \frac{\frac{E_{fi}}{E_{i-1}} - 1}{\frac{E_{fi}}{E_{i-1}} + \zeta_{i\perp}} \quad (6)$$



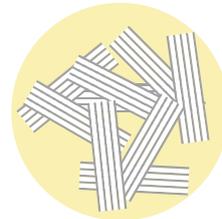
(a) Disk-like platelet.



(b) Primary particle.

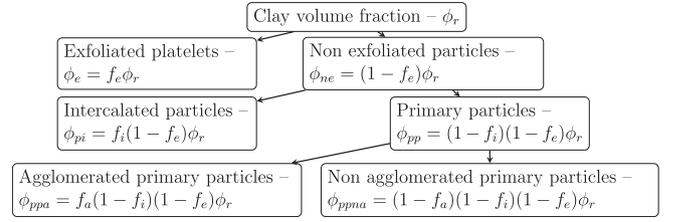


(c) Intercalated particle.



(d) Agglomerate.

**Fig. 5.** Schematic representation of a montmorillonite exfoliated platelet (a), primary particle (b), intercalated particle (c) and agglomerate (d). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)



**Fig. 6.** Distribution of the montmorillonite platelets in the nanocomposites.

**Table 2**

Elastic moduli, shape parameters and volume fraction of the various fillers –  $L$  is the diameter of a silicate platelet, and  $d_s$  is its thickness.

	$E_{fi}$	$\zeta_{i\parallel}$	$\zeta_{i\perp}$	$\phi_i$
$i = 1$	$E_r$	$2\frac{L}{d_s}$	2	$\phi_e$
$i = 2$	$E_{pi}$	$2\frac{L}{(N-1)d_{001} + d_s}$	2	$\phi_2 = \frac{\phi_{pi}}{\phi_{rpi}}$
$i = 3$	$E_{pp}$	$2\frac{L}{(N_0-1)d_0 + d_s}$	2	$\phi_3 = \frac{\phi_{ppa}}{\phi_{pp}}$
$i = 4$	$E_a$	2	2	$\phi_{ppa}$

with  $E_0 = E_m$ , the elastic modulus of the matrix and  $E_4 = E_c$ , the elastic modulus of the nanocomposite containing the four varieties of montmorillonite fillers. The volume fraction, shape parameters and elastic moduli of the different types of filler are given in Table 2.

The elastic modulus of an intercalated particle  $E_{pi}$ , with an interlayer spacing of  $d_{001}$  and a number  $N$  of silicate layers for each intercalated particle, is calculated by using the rule of mixtures, as proposed by Brune and Bicerano [18]:

$$E_{pi} = \phi_{rpi} E_r + (1 - \phi_{rpi}) E_m \quad (7)$$

where  $\phi_{rpi}$ , the volume fraction of silicate in an intercalated particle (cf. Fig. 5b), is the ratio of the volume of silicate in an intercalated particle to the total volume of an intercalated particle, and is given by the following equation:

$$\phi_{rpi} = \frac{N d_s}{(N-1) d_{001} + d_s} \quad (8)$$

and the elastic modulus of an effective primary particle  $E_{pp}$ , with an interlayer spacing of  $d_0$  and a number  $N_0$  of silicate layers for each primary particle, is calculated using:

$$E_{pp} = \phi_{rpp} E_r \quad (9)$$

where  $\phi_{rpp}$ , the volume fraction of silicate in a primary particle (cf. Fig. 5c), is the ratio of the volume of silicate in a primary particle to the total volume of a primary particle, and is given by the following equation:

$$\phi_{rpp} = \frac{N_0 d_s}{(N_0 - 1)d_0 + d_s} \quad (10)$$

$\phi_2$ , the effective volume fraction of intercalated particles, is given by:

$$\phi_2 = \frac{\phi_{pi}}{\phi_{rpi}} = \frac{f_i(1 - f_e)\phi_r}{\phi_{rpi}} \quad (11)$$

and  $\phi_3$ , the volume fraction of primary particles, is given by:

$$\phi_3 = \frac{\phi_{ppna}}{\phi_{rpp}} = \frac{(1 - f_a)(1 - f_i)(1 - f_e)\phi_r}{\phi_{rpp}} \quad (12)$$

The explanation for the expression and values of  $\zeta_{i||}$  and  $\zeta_{i\perp}$  are given elsewhere [19,9].

### 3.3. Values of the input parameters of the model

The homogenisation model described above uses several input parameters, whose values are discussed below.

**Numbers  $N_0$  and  $N$  of platelets** – Considering that the thickness of a montmorillonite particle is substantially constant and of the order of 8 to 10 nm [20] (when the interlayer distance increases, the number of silicate platelets per stack decreases), the average number  $N_0$  of platelets per stack in a primary particle can be estimated by the following equation<sup>3</sup>:

$$\left\lfloor \frac{8 - d_s}{d_0} + 1 \right\rfloor \leq N_0 \leq \left\lceil \frac{10 - d_s}{d_0} + 1 \right\rceil \quad (13)$$

where  $d_s$  is the thickness (in nm) of a silicate platelet and where  $d_0$  is the interlayer distance (in nm) in a montmorillonite primary particle. Thus, here, a primary particle is formed of an average stack of 5.4 platelets.

In the same way, the average number  $N$  of platelets per stack in an intercalated particle, with an interlayer distance of  $d_{001}$  can be estimated by the following equation:

$$\left\lfloor \frac{8 - d_s}{d_{001}} + 1 \right\rfloor \leq N \leq \left\lceil \frac{10 - d_s}{d_{001}} + 1 \right\rceil \quad (14)$$

Thus, here, an intercalated particle is formed of an average stack  $N$  of 3 platelets.

**Thickness  $d_s$  of a silicate platelet** – From the data of Fornes and Paul [9], the thickness  $d_s$  of a single platelet of montmorillonite is considered to be 0.94 nm.

**Elastic modulus  $E_r$  of a single silicate platelet** – Chen and Evans [21] have established that the elastic modulus of smectite clay platelets is between 178 and 265 GPa. For this study, an average value of 220 GPa was used for  $E_r$ , the elastic modulus of a single montmorillonite platelet.

**Elastic modulus  $E_a$  of a clay agglomerate** – The elastic modulus of a clay agglomerate  $E_a$  can be estimated at 50 GPa [21].

**Elastic modulus  $E_m$  of the thermoset matrix** – The elastic modulus of the vinyl ester matrix is 3.22 GPa  $\pm$  0.05 GPa (determined by modal analysis, cf. Table 3).

**Width  $L$  of a silicate platelet** – The values of  $L$  given in the literature often extend between 100 and 200 nm [9,22]. In the following, the value of  $L$  used is 150 nm.

**Table 3**

Experimental results of the elastic modulus and standard deviation (GPa).

Weight fraction (%)	Elastic modulus (GPa)
0	3.22 $\pm$ 0.05
1.25	3.52 $\pm$ 0.05
2.5	3.85 $\pm$ 0.05
3.75	3.90 $\pm$ 0.05
5	3.97 $\pm$ 0.05

**Exfoliation fraction  $f_e$**  – Following the example of Luo and Daniel [10], if uniform dispersion and random orientation of the filler are assumed, an upper limit to the exfoliated volume fraction  $\phi_e$  can be estimated by the ratio of the volume of an exfoliated platelet to the cube of its largest dimension (Eq. (15)).

$$\phi_e = f_e \phi_r \leq \frac{\pi d_s}{4L} \quad (15)$$

Thus the maximum exfoliation fraction  $f_{e \max \text{ 3D}}$ , defined as the upper limit of the exfoliation fraction is given by Eq. (16).

$$f_{e \max \text{ 3D}} = \frac{\pi d_s}{4L \phi_r} \quad (16)$$

Thereafter, the exfoliation fraction of the nanocomposite is restricted by this upper limit.

**Intercalation and agglomeration fractions** – The intercalation fraction  $f_i$  of non exfoliated montmorillonite particles and the agglomeration fraction  $f_a$  of the primary particles, can theoretically vary between 0 and 1.

### 3.4. Experimental results

The elastic moduli of the montmorillonite reinforced nanocomposites, measured by experimental modal analysis, are listed in the Table 3.

The addition of 2.5% weight fraction of montmorillonite increases the elastic modulus of the matrix of about 20% whereas the addition of 5% weight fraction of montmorillonite increases the elastic modulus of the matrix of about 23%. The increase of the elastic modulus of a thermosetting matrix by the addition of montmorillonite is an expected result already observed in the literature, with the intensity of the increase varying according to the nature of the materials used and the manufacturing process. For example, using montmorillonite with a silane treatment as a filler in an unsaturated polyester matrix, Kornmann et al. [23] observed an increase of 28% in the elastic modulus, for a montmorillonite concentration of 5 wt%. With the same matrix and Cloisite® 30B montmorillonite, Inceoglu and Yilmazer [24] obtained a 17% increase of the elastic modulus, for a montmorillonite concentration of 5 wt%. With the same organomodified montmorillonite and an epoxy matrix, Yasmin et al. [25] obtained an increase of 18% of the elastic modulus, for a clay concentration of 5 wt%, whereas with another organomodified montmorillonite (Nanomer® I.28E), the increase in the elastic modulus reached 25%, for the same clay concentration.

### 3.5. Comparison between model predictions and experimental results

The five-phase model developed allows determining the elastic modulus of the composite ( $E_c$ ) by varying the morphology and especially the distribution of the montmorillonite platelets in the composite (cf. Figs. 5 and 6). Two extreme cases can be described: the most unfavourable one where the platelets are poorly exfoliated, and the most favourable one where the platelets are highly exfoliated.

<sup>3</sup>  $\lfloor \cdot \rfloor$  and  $\lceil \cdot \rceil$  representing the floor and ceiling functions.

- In the most unfavourable composite the montmorillonite filler is uniformly dispersed in the matrix as clay agglomerates. The theoretical minimal elastic modulus  $E_{\text{mod}}^{\text{min}}$  of such a composite is defined as the elastic modulus of a composite with a zero exfoliation fraction, a zero intercalation fraction of non exfoliated particles, and with an agglomeration fraction of primary particles of 1 ( $f_e = 0, f_i = 0, f_a = 1$ ). Hence the increase of the elastic modulus of the matrix is only due to the addition of uniformly dispersed montmorillonite agglomerates, with neither exfoliation nor intercalation.
- In the most favourable composite, consisting of a partially exfoliated and partially intercalated nanocomposite, the exfoliation fraction is equal to the upper limit of the exfoliation fraction as defined above and in which all the non exfoliated particles are intercalated, with an interlayer distance equal to  $d_{001}$ . The theoretical maximal elastic modulus  $E_{\text{mod}}^{\text{max}}$  of such a nanocomposite is defined as the elastic modulus of a nanocomposite with an exfoliation fraction  $f_e$  equal to  $f_{e \text{ max } 3D}$ , an intercalation fraction of non exfoliated particle  $f_i$  of 1, and with an agglomeration fraction of primary particles  $f_a$  of 0. Hence the increase in the elastic modulus of the matrix is due to the addition of uniformly dispersed montmorillonite, which shows the maximal exfoliation fraction that can be obtained, and to a complete intercalation of the remaining montmorillonite particle, without any agglomerates.

The theoretical minimal and theoretical maximal elastic moduli, estimated using the input parameters listed in Section 3.3, are shown in Fig. 7, together with the experimental data.

As expected for the nanocomposites of this study which present a morphology intermediary between these two extreme cases, the experimental values of the elastic modulus of the composites are between the theoretical minimum and maximum elastic moduli. Calculating the ratio of the experimental elastic modulus on the theoretical maximal elastic modulus (cf. Table 4) reveals that, for the lower levels of filler, the nanocomposite materials have an elastic modulus less than 10% lower than the theoretical maximal elastic modulus, which is a good result. The higher the filler weight fraction, the greater the difference between experimental and maximal values.

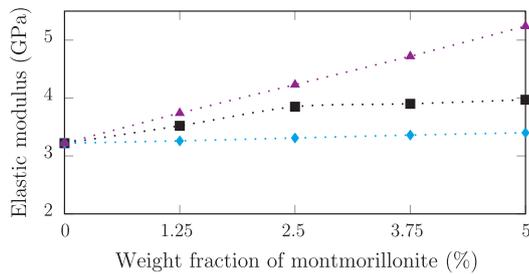


Fig. 7.  $\diamond$  Theoretical minimal elastic modulus  $E_{\text{mod}}^{\text{min}}$  and  $\blacktriangle$  theoretical maximal elastic modulus  $E_{\text{mod}}^{\text{max}}$  and  $\blacksquare$  experimental data for several weight fractions of montmorillonite. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

Table 4  
Ratio of the experimental elastic modulus on the theoretical maximal elastic modulus.

Weight fraction (%)	$\frac{E_{\text{mod}}}{E_{\text{mod}}^{\text{max}}}$ (%)
1.25	93.9
2.5	91.0
3.75	82.6
5	75.8

The parameters involved in the increase of the elastic modulus of the nanocomposites that can be influenced by the fabrication process are the exfoliation, intercalation and agglomeration fractions.

The effective increase of the elastic modulus of the nanocomposite  $R_e$  and the theoretical maximal increase in the elastic modulus of the nanocomposite  $R_{\text{max}}$  can be defined as:

$$R_e = \frac{E_c - E_{\text{mod}}^{\text{min}}}{E_{\text{mod}}^{\text{min}}} \quad R_{\text{max}} = \frac{E_{\text{mod}}^{\text{max}} - E_{\text{mod}}^{\text{min}}}{E_{\text{mod}}^{\text{min}}} \quad (17)$$

In addition, as a mean to evaluate the efficiency  $\Xi$  of the fabrication process in terms of exfoliation, intercalation and rupture of agglomerates,  $\Xi$  is defined as the ratio of the effective increase in the elastic modulus of the nanocomposite  $R_e$  on the theoretical maximal increase in the elastic modulus of the nanocomposite  $R_{\text{max}}$ :

$$\Xi = \frac{R_e}{R_{\text{max}}} = \frac{E_c - E_{\text{mod}}^{\text{min}}}{E_{\text{mod}}^{\text{max}} - E_{\text{mod}}^{\text{min}}} \quad (18)$$

The effective increase  $R_e$  and the theoretical maximal increase in the elastic modulus of the nanocomposite  $R_{\text{max}}$ , together with the efficiency  $\Xi$  of the fabrication process in terms of exfoliation, intercalation and rupture of agglomerates are shown Fig. 8.

The efficiency of the fabrication process in terms of exfoliation, intercalation and rupture of agglomerates reaches its maximum for a weight fraction of montmorillonite of 2.5%, the effective increase in the elastic modulus is then approximately 59% of the theoretical maximum increase of the elastic modulus. For higher weight fractions of montmorillonite, it is very noticeable that the delamination efficiency decreases with the filler weight fraction hence confirming that the nanocomposite material processing was optimal for 2.5 weight fraction of montmorillonite and should be improved for higher contents. Indeed, the fabrication process described above was chosen because it can be easily transposable for industrial purposes. But, it is most efficient for low amounts of montmorillonite. When the amount of montmorillonite is increased the process efficiency is not sufficient to improve the dispersion.

### 3.6. Morphology optimisation by back calculation of the exfoliation, intercalation and agglomeration fractions

To better identify the weaknesses in the manufacturing process, exfoliation, intercalation and agglomeration fractions, for which

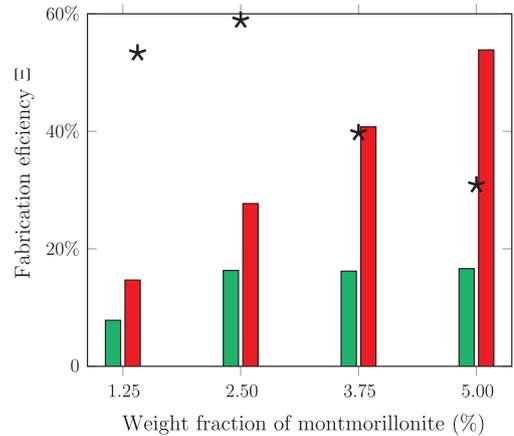


Fig. 8.  $\blacksquare$  Effective increase  $R_e$  and  $\blacksquare$  theoretical maximal increase  $R_{\text{max}}$  in the elastic modulus of the nanocomposite.  $\star$  Efficiency  $\Xi$  of the fabrication process in terms of exfoliation, intercalation and rupture of agglomerates. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

the modelled elastic modulus is equal to the experimental elastic modulus, are back calculated using the modelling procedure described previously and the results are displayed in Fig. 9.

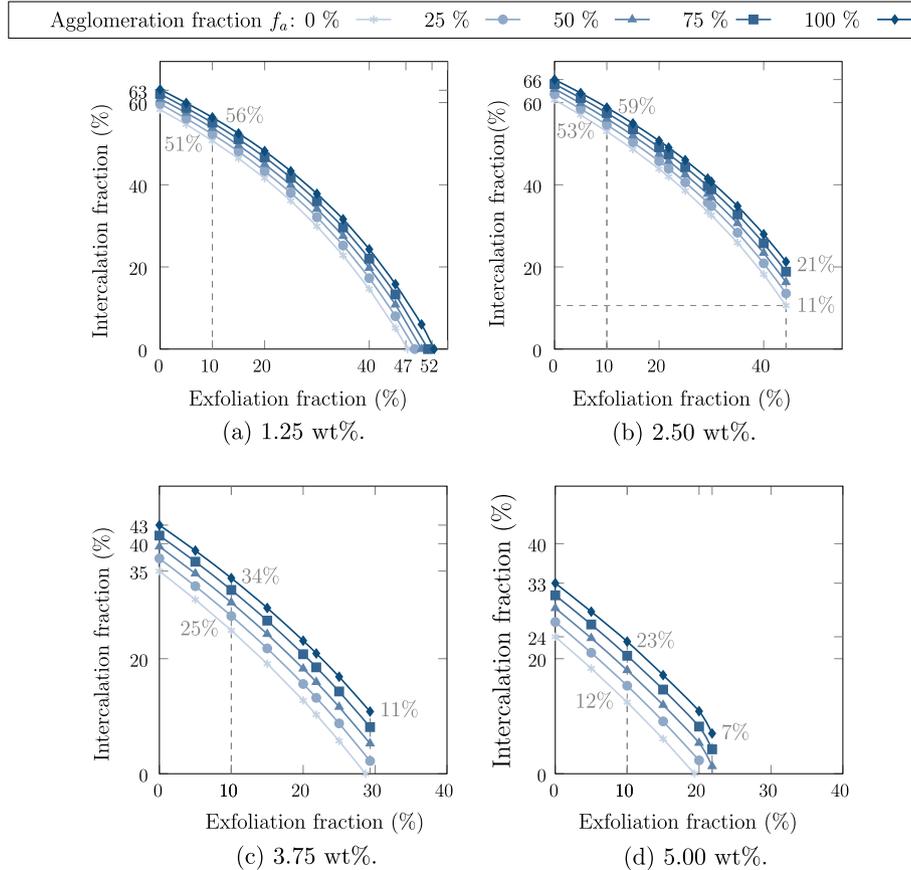
For a montmorillonite weight fraction of 1.25%, the theoretical maximum exfoliation fraction  $f_{e \text{ max } 3D}$  is about 89%. However, the maximum exfoliation fraction back calculated is only between 47% and 52%, depending on the agglomeration fraction (cf. Fig. 9a), meaning that the maximum exfoliation fraction was not achieved during the processing of this nanocomposite whereas for the other compositions, the theoretical maximal exfoliation fraction  $f_{e \text{ max } 3D}$  may have been reached. In addition, the intercalation fraction of non exfoliated particles reaches a maximum value of 63% for a montmorillonite weight fraction of 1.25% (cf. Fig. 9a), 66% for a montmorillonite weight fraction of 2.5% (cf. Fig. 9b), 43% for a montmorillonite weight fraction of 3.75% (cf. Fig. 9c) and 33% for a montmorillonite weight fraction of 5% (cf. Fig. 9d). Hence, a significant fraction of primary clay particles are not even intercalated. These results confirm that the higher the filler weight fraction, the lower the process efficiency in terms of intercalation. If the extend of the intercalation depends mainly on the type of organic treatment of the montmorillonite [26], intercalation fractions as high as 60–66% should also be achievable for the highest fractions of montmorillonite. Thus, in order to increase the elastic modulus of the nanocomposites, for the highest fractions of filler, their intercalation fraction should be increased to values ranging from 60% to 66%, as it is already the case for the montmorillonite weight fraction of 2.5%. If one considers that the exfoliation fraction is around 10%, which is a reasonable assumption according to the literature [10], the possible ranges of the intercalation fraction are given in Table 5.

**Table 5**

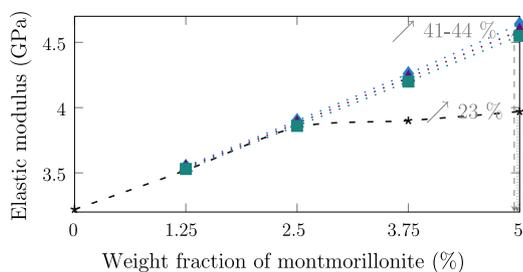
Possible ranges for the intercalation fraction ( $f_i$ ), for an exfoliation fraction of 10%.

Weight fraction (%)	$f_i$ (%)
1.25	51–56
2.5	53–59
3.75	25–34
5	12–23

Using an intercalation fraction of 60% and an exfoliation fraction of 10%, the elastic moduli of the different composition have been estimated and are represented in Fig. 10. First, it is shown that, as expected, varying the agglomeration fraction do not change much the values of the elastic modulus. Moreover, for 1.25% and 2.5% weight fraction of montmorillonite, this distribution of the nanoclay ( $f_e = 10\%$  and  $f_i = 60\%$ ) gives a value of the elastic modulus very close (cf. Fig. 10) to the one measured by modal analysis, demonstrating that the morphology of those two nanocomposites is probably well-described by the modified Halpin–Tsai model built. If one uses this distribution of the nanoclay for the highest fractions of montmorillonite (3.75% and 5%) however, huge discrepancies are found between estimated and measured values of the elastic modulus (cf. Fig. 10, Table 5). This result is to be expected as the highest intercalation fraction actually achieved is two to four times lower than 60%, that is the intercalation fraction of 1.25% and 2.5% montmorillonite weight fraction nanocomposites (cf. Table 5). For the highest fractions of filler (5%), an increase of the efficiency of the process in term of intercalation, from 30% (cf. Fig. 8) to 60% (cf. Fig. 10) would lead to an increase in the elastic modulus of the matrix of 41% and 44%, for an agglomeration



**Fig. 9.** Influence of the agglomeration fraction ( $f_a$ ) on the back calculation (assuming  $E_{mod} = E_c$  as given in Table 3) of the exfoliation and intercalation fractions ( $f_e$  and  $f_i$ , respectively), for several montmorillonite weight fractions. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)



**Fig. 10.** \* Experimental elastic modulus and modelled elastic modulus for exfoliation fraction ( $f_e$ ) of 10% and intercalation fraction ( $f_i$ ) of 60% and for agglomeration fraction ( $f_a$ ) of 0% (♦), 50% (▲) and 100% (■), for several weight fraction of montmorillonite. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

fraction of 100% and 0%, respectively. By comparing these values of 41–44% to the value of 23% experimentally obtained for a montmorillonite weight fraction of 5%, the increase in the elastic modulus that one would obtain with an intercalation fraction as high as 60% (the one obtained for a weight fraction of 2.5%) can thus be estimated.

#### 4. Summary and conclusions

In this work, thermoset/montmorillonite nanocomposites were studied. Besides the expected increase of the elastic modulus of the nanocomposites with the clay concentration, it was shown that the morphology of the nanocomposites filler could be effectively modelled as partially exfoliated and partially intercalated platelets, with the presence of non intercalated primary particles and clay agglomerates. A five-phase homogenisation model based on the Halpin–Tsai equations was developed to relate the elastic modulus of the materials to their morphology and to estimate the efficiency of the fabrication process in terms of exfoliation, intercalation and rupture of agglomerates. Moreover, this model was used to back calculate the exfoliation, intercalation and agglomeration fractions of the nanocomposite materials from the elastic modulus measured by an original modal analysis technique. Thereby, it was shown that with a limited exfoliation fraction (10%), the intercalation fraction ranges from 12% (for 5% weight fraction of montmorillonite) to 59% (for 2.5% weight fraction of montmorillonite). The hypothesis of a distribution of montmorillonite platelets consisting of 10% of exfoliated platelets and 60% of intercalated particles was shown to be reliable for the 1.25% and 2.5% weight fractions of montmorillonite. Finally, this work was useful to describe a realistic distribution of montmorillonite particles assuming a limited exfoliation fraction.

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