



## Full Length Article

# Come to light: Detailed analysis of thermally treated Phenyl modified Carbon Nitride Polymorphs for bright phosphors in lighting applications

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## ABSTRACT

Carbon Nitride and its polymorphs have recently gained large interests for their huge properties and applications in different fields, from lighting to photocatalysis. Further, several attempts were recently devoted to tune and control its optical and electrical properties. In this report we analyze phenyl modified Carbon Nitride structures obtained by simple thermal polymerization at different temperatures (250–400 °C) of the starting precursor: 2,4-diamino-6-phenyl-1,3,5-triazine. A multi-technique experimental data (XRD patterns, Raman, TGA and DTG, steady- time and time resolved Luminescence, Photoluminescence Excitation spectra, Reflectivity spectra) was applied to analyze the relationship between structural and optical properties and to give more insight on the effect of synthesis procedure on the final polymer. The optical properties evidenced an interesting shift towards the visible region of the absorption spectrum of the phenyl modified g-C<sub>3</sub>N<sub>4</sub> polymer that, associated with the high optical quantum yield (about 60%) and to a broad emission in the green-red spectral region, makes the samples very suitable for lighting applications. Indeed, we report a first prototype of white LED emission by assembly of a commercial blue LED and the Phenyl modified g-C<sub>3</sub>N<sub>4</sub> powders as phosphor, verifying the structural and optical stability over about 10,000 working hours.

## 1. Introduction

Nowadays, research in new materials is instrumental to reach a sustainable, energy saving and eco-friendly development. New materials for lighting, catalysis and photocatalysis, photovoltaic and water splitting are of paramount importance for “next future” technologies; however, their application should be subjected to sustainable life-cycle assessment, too often not considered since the inception. Materials with very expensive synthesis procedures, both in terms of costs and/or energy consumption, as well as potentially eligible to produce dangerous or polluting waste need to be discarded. In this view, materials free of “critical” raw elements are very attractive and the research in this direction is boosted, looking for available alternatives, among which the most promising are new organic materials [1–3].

Within the organic chemistry panorama, polymers can guarantee very high versatility in terms of structural and optical response with high potential in several applications, from electronics to photovoltaics

[4–6], but their efficient exploitation in commercial devices is hampered by the low thermal and short time structural stability of the compounds.

Since 2014, an increasing attention has been devoted to graphitic Carbon Nitride (g-C<sub>3</sub>N<sub>4</sub>) defined as one of most reliable photocatalysts for artificial photosynthesis and pollution degradation [7]. It was indicated as a very promising material for photovoltaic applications [8], a very efficient phosphor for lighting applications, [9,10] and lastly as a suitable photocatalysts for hydrogen generation, [11] and water splitting [12].

Most of its high-ranking properties are related to its thermal and structural stability, taking advantage with its simply and low-cost synthesis procedure. g-C<sub>3</sub>N<sub>4</sub> is fabricated by direct condensation of nitrogen-containing organic precursors (for example, urea, thiourea, dicyandiamide, cyanamide, and guanidine hydrochloride) [13,14]. Recently, g-C<sub>3</sub>N<sub>4</sub> structure with thermal stability up to 550 °C was successfully obtained by direct condensation of melamine [15], leading

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to the formation of different polymorphs depending on the synthesis temperature [16].

Depending on the above-mentioned synthesis procedures, different optical properties were reported [17–19]. The possibility to tune the optical properties of carbon nitride based materials was largely explored, by doping with specific elements (among others Fe, Ag, Br, P, Br, I) [20–23] or by slightly changing the organic mesh [24]. In this scenario, recently doping with phenyl ring was applied [17,25], because of the promising results in tuning the band-gap of the organic semiconductor to lower energies. Indeed, the introduction of phenyl groups into the melon frame extends the  $\pi$ -conjugation in  $g\text{-C}_3\text{N}_4$ , leading to a decrease of the band gap and to an increase of the separation rate of electron–hole pairs [17]. To achieve the doped mesh, the strategy of election was a co-polymerization process of benzamide and urea [17,28]. Alternatively, a solvent free thermal treatment at 400 °C of a single precursor, 2,4-diamino-6-phenyl-1,3,5-triazine was successfully applied [26].

This last strategy seems very promising to produce sustainable materials: the solvent free reaction, the absence of metals and critical raw materials, the low cost of the starting precursor are fundamental parameters for the development of new green materials. However, a step by step study on the relationship between thermal synthesis, structural characterization and application/performance was never performed on this new class of Carbon Nitride materials.

In this paper we report structural and optical characterization of phenyl modified  $g\text{-C}_3\text{N}_4$  samples (hereafter PhCN) grown in controlled atmosphere at different temperature starting from 2,4-diamino-6-phenyl-1,3,5-triazine as precursor, without the use of any solvent. We applied a multi-techniques approach, which improves the predictive process, providing deep insights on the correlation among optical features, morphology and effective structure of the carbon nitride polymers. This opens the possibility to tune the properties of phenyl modified organic meshes for further technological exploitations. To test the bench strength, we provided a direct application of the phenyl modified structure for lighting. A first prototype of white LED is achieved by modifying a commercial Blue LED with one of the phosphors here described. The stability in time, the efficiency and the color rendering index are reported to provide a wide characterization and to suggest this class of organic compound as an efficient phosphor for lighting applications.

## 2. Experimental

### 2.1. Materials and methods

1 g of 6-phenyl-1,3,5-triazine-2,4-diamine powder (Ph-Triazine, Aldrich, 99%) was placed in a quartz tube accommodated in a tubular furnace and treated at different temperatures (250 °C, 300 °C, 350 °C and 400 °C) for 1 h (hereafter PhCN250, PhCN300, PhCN350 and PhCN400, respectively). The samples during the thermal treatment were covered with a quartz plate to prevent the vaporization at high temperature and to assure a re-condensation at high temperature. The heating rate was 30 °C/min, the syntheses were performed under constant nitrogen flux (30 ml/min). All samples were manually grounded to obtain powder-like samples and washed with methanol several times to remove the impurities.

### 2.2. Characterization techniques

X-Ray Diffraction measurements were made by using a Malver Panalytical Empyrean Diffractometer with 2 $\theta$  Bragg-Brentano geometry, Cu K $\alpha$  wavelength and X'Celerator silicon detector which uses a RTMS (Real Time Multiple Strip) 1D acquisition technology.

Raman spectra were acquired in back scattering geometry with excitation wavelength at 1064 nm generated by a Nd:YAG laser, to avoid luminescence contribution in the visible range. The system

operates in Stokes region up to 2500  $\text{cm}^{-1}$ . Measurements were performed in air at room temperature with a spectrometer BWTEK i-Raman ex with a spectral resolution of a 9  $\text{cm}^{-1}$ .

Elemental analysis was performed on the dry samples by a PerkinElmer 24,000 Series II CHN Analyzer.

High Resolution Mass Spectra (HRMS) of products were obtained using a High-Resolution Mass Spectrometer in fast atom bombardment (FAB+) ionization mode (ESI) acquired using a Bruker micrOTF-Q II or/and Agilent Q-TOF 6520.

Thermo-gravimetric analysis (TGA) was performed with Netzch 209 F1 Libra. The samples were heated from 25 °C to 800 °C (heating rate 10 °C/min) under constant nitrogen flux (20 ml/min). The same equipment was used for the on-line monitoring of the reaction, coupling the instrument to a BRUKER Optics FTIR. In this case, the synthesis procedure was simulated in the TGA (heating ramp from 20 °C to 400 °C with a rate of 20 °C/min followed by an isothermal step at 400 °C of 1 h, nitrogen flow 20 ml/min). Data exchange between NETZSCH PROTEUS software and Bruker OPUS software is done online during the measurement. The gases that evolved by thermal analysis are transferred into the IR spectrometer from Bruker Optics and analyzed with a Bruker Tensor2 FT-IR. The gas cell is heated to 200 °C and possesses a volume of 5.8 ml. IR -scanned range was from 400  $\text{cm}^{-1}$  to 5000  $\text{cm}^{-1}$ . Every spectrum is averaged from 16 scans.

Differential Scanning Calorimetry (DSC) measurements were acquired by using DSC1 METTLER STAR1 System equipped with a low temperature probe. The measurements were performed in nitrogen atmosphere. For all the samples, the thermal cycle was: isothermal step for 5 min at –30 °C, heating from –30 °C to 250 °C then cooling from 250 °C to –30 °C, registering a second isotherm at –30 °C for 5 min and heating again with the same heating ramp. The second thermogram was discussed in the text.

The absorption measurements were acquired by a UV-Vis-NIR Agilent Technologies Cary 5000. Measurements were performed by using a PbS solid state photodetector. The reflection configuration measures the diffuse reflection of a sample with respect to a reference sample which is considered to have a 100% reflectivity.

Steady-state photoluminescence measurements were performed with a 405 nm laser excitation coupled with an optical fiber to an Avantes Sensiline Avaspec-ULS-TEC Spectrometer. The measurements were acquired with 500 ms time window in a 300–800 nm spectral range.

Time resolved photoluminescence (TR-PL) measurements were recorded by exciting the samples with 200 fs long pulses delivered by an optical parametric amplifier (Light Conversion TOPAS-C) pumped by a regenerative Ti:sapphire amplifier (Coherent Libra-HE). The repetition frequency was 1 kHz and the PL signal was recovered by a streak camera (Hamamatsu C10910) equipped with a grating spectrometer (Princeton Instruments Acton SpectraPro SP-2300). All the measurements were collected in the front face configuration to reduce inner filter effects. Proper emission filters were applied to remove the reflected contribution of the excitation light.

The luminescence quantum yield was measured using a 10-inch diameter integrating sphere (Lab sphere, LMS-100) and a 385 nm LED (ThorLabs) for excitation and the signals were recorded using a CCD detector (Ocean optics, USB4000-VIS-NIR). The spectral intensity calibration was carried out using a standard Deuterium-Tungsten Halogen lamp (Ocean Optics, DH-3P-BAL-CAL). The sample was dispersed in a 1 mm thick PDMS pellet before placing inside the integrating sphere. The signals were recorded by placing the sample in-line with the excitation beam and for reference, the base support was placed in-line with the excitation beam. The internal quantum yield was measured by comparing the emitted photons with respect to absorbed photons. In order to validate the obtained values, the internal quantum yield of a YAG:Ce standard phosphor from OSRAM, Inc., was measured under 440 nm LED excitation using the same set-up and is found to be about 60% in place of 82%, which lead to apply a correction factor of. The

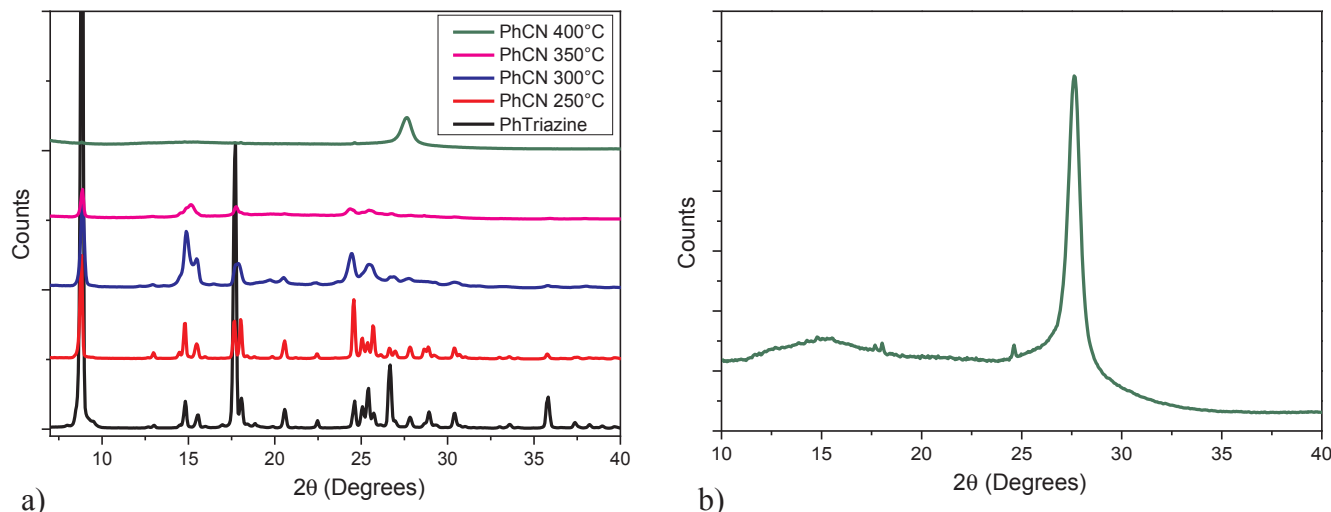


Fig. 1. (a) X-Ray diffraction patterns of the samples obtained at different temperature starting from the precursor (6-phenyl-1,3,5-triazine-2,4-diamine). (b) X-Ray diffraction pattern of the sample prepared at 400 °C.

total uncertainty in YAG:Ce quantum yield is up to 10%, which suggests an uncertainty of more than 10% in the quantum yield of a-YAB powders measured under near UV excitation.

### 3. Results and discussions

#### 3.1. Structural properties

##### 3.1.1. X-Ray diffraction measurements

To show how thermal treatment affects the final structure of the polymer, we report X-Ray diffraction patterns of the PhCN samples treated at different temperature obtained from Ph-Triazine precursor (Fig. 1a and b).

The plots display well defined diffraction peaks at 8.8°, 17.7°, 24.5°, 25.3°, 26.6° and 35.8° in  $2\theta$ , denoting a semi-crystalline structure. PhCN250 sample exhibits a reduction in the intensity of the main peaks at 8.8°, 17.7°, 26.6° and 35.8° and a variation in the intensity ratio in the 24.5–25.3° range without any further clear modification in the profile. Increasing the synthesis temperature up to 300 °C, it is possible to observe a broadening in all the diffraction peaks, with the overlapping of most of them. An overall reduction of the peaks counts with variation in the relative intensities among the diffraction peaks is observable for the pattern of the sample PhCN350. Finally, an abrupt variation is clearly detected in the pattern of the sample prepared at 400 °C, where, as already reported [19], only two broad bands, located at about 15° and 27.6°, can be distinguished. Considering the high similarity with the XRD diffraction pattern of undoped Carbon Nitride, the peak located at higher angles is assigned to (0 0 1) reflection and it is related to the distance among planes whilst the one at 15° and is associated to the reflection from (2 1 0) plane and it is related to the separation distance of heptazine chains [27].

In  $g\text{-C}_3\text{N}_4$  the higher angle diffraction peak was reported at 27.3° and 27.5° for the Carbon nitride samples treated at 500 and 600 °C, respectively [16] (slightly variations of these values were also reported, depending on minor changes in synthesis procedure) [28,29]. The observed peaks in the PhCN400 sample indicates that the general 2D structure of the  $g\text{-C}_3\text{N}_4$  is preserved but, with a smaller distance among the planes (from 3.26 Å to 3.24 Å in  $g\text{-C}_3\text{N}_4$ ). The broad diffraction peak at lower angles, with main features at about 13° and 15° in the PhCN400 diffraction pattern (13.2° in  $g\text{-C}_3\text{N}_4$ ) indicates a broad distribution of the separation distance among the heptazine chains in each plane (see in Fig. 1b the magnification of the XRD pattern of PhCN400 sample). In view of optical features, one has to bear in mind that a shortening in the inter-plane distance is the main cause of emission quenching in Carbon

nitride samples [16,30]. However, as we will better explain in the optical section, in the PhCN400 sample we observe a very high increase of the optical quantum yield with respect to  $g\text{-C}_3\text{N}_4$ , that underline the key role of the phenyl rings in the heptazine mesh.

##### 3.1.2. Raman analysis

The Raman spectra give information on the structure of the polymorphs (Fig. 2a). The spectrum of the phenyl-triazine presents several bands between 200 and 1600  $\text{cm}^{-1}$ , with prominent peaks at 1601  $\text{cm}^{-1}$ , related to the motion of the phenyl group, and at 1392  $\text{cm}^{-1}$ , assigned to conjugated stretching vibration (C–C) between the phenyl ring and the triazine ring [31]. Other prominent figures are at 1000  $\text{cm}^{-1}$  (breathing mode) and 674  $\text{cm}^{-1}$ , related to  $\text{NH}_2$  modes (Fig. S1-a). The spectra of the samples treated at different temperatures present slight variation in the peaks position. The main change is gathered in the relative contribution of the Raman peaks, enlightened by comparing the intensities at 1392 and 1601  $\text{cm}^{-1}$  (Fig. 2b). The ratio  $I_{1600/1390}$  value (about 1) keeps constant up to 250 °C then it increases to 1.34, at 300 °C, and to 1.57 at 350 °C. Finally, an abrupt increase up to 3.54 is observed in the sample treated at 400 °C.

This trend can be understood by analyzing deeply the vibration modes: since the band at 1392  $\text{cm}^{-1}$  is related to the conjugated motion of the triazine and the phenyl ring through the stretching mode of the C–C bonding, a decrease of the vibration intensity is expected when increasing the mass structure of at least one of the two units. Thus, the increase of the ratio with respect to the breathing mode of the phenyl ring suggests the formation of different polymorphs, until the formation of the heptazine units at 400 °C is achieved.

##### 3.1.3. Elemental analysis

The formation of different polymorphs is confirmed by elemental analysis of the compounds; the C/N molar ratio spans from 1.54 for the phenyl-triazine precursor down to 1.50 in the sample treated at 300 °C (PhCN300). Finally, a drastic decrease of the C/N molar ratio to 1.24 was measured for the sample prepared at 400 °C (S2-Table S1) an indication of the increase of the polymerization degree.

##### 3.1.4. HRMS analysis

To better understand the polymerization process and to identify the structure of the intermediate products, high resolution mass spectrometry (HRMS) analysis was carried out. As previously observed in the XRD and Raman analysis, no variations are observed for the sample heated at 250 °C with respect to the starting Phenyl-Triazine. At 300 °C the partial formation of the dimer by linear polymerization is indicated

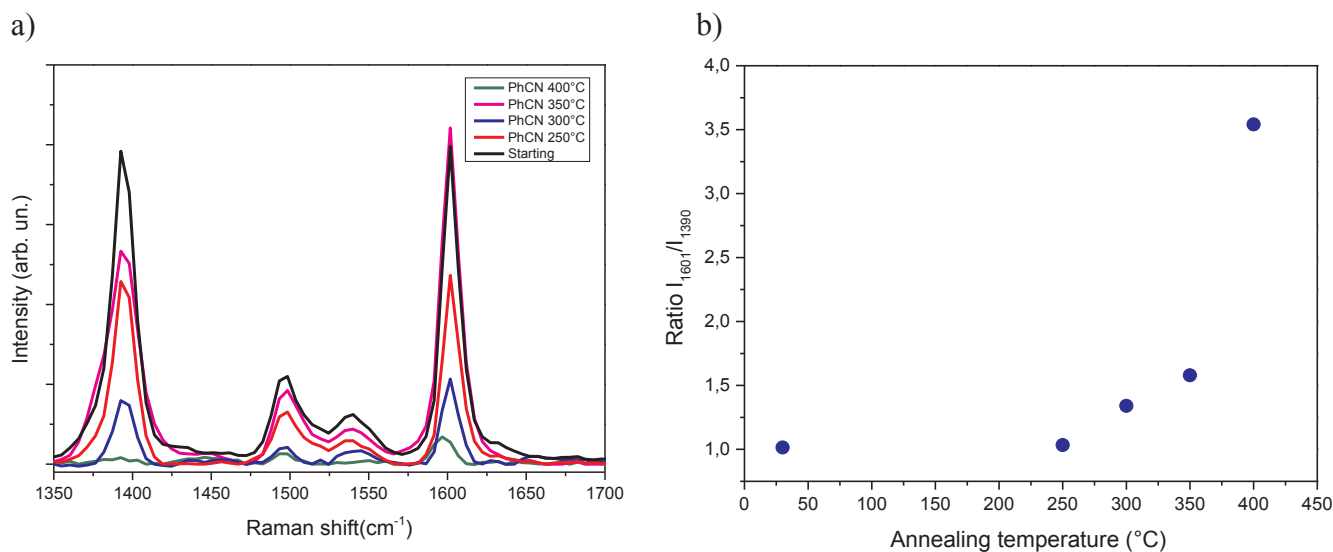


Fig. 2. (a) Raman spectra behavior of the temperature treated samples. (b) Ratio of intensities  $I_{1601}/I_{1390}$ .

by the peak at  $358 \text{ (mZ}^{-1}\text{)}$ , this process is further evidenced increasing the curing temperature to  $350 \text{ }^\circ\text{C}$ , where the formation of trimer units at  $528 \text{ (mZ}^{-1}\text{)}$  are also evidenced (Fig. S3). According to the previous analysis, the sample treated at  $400 \text{ }^\circ\text{C}$  was expected to give rise to the formation of the heptazinic system, but due to its very poor solubility in several solvents, it was not possible to analyze the product.

### 3.1.5. Thermogravimetric analysis

Thermogravimetric (TGA) and Differential Scanning Calorimetry (DSC) measurements confirm that different structures of phenyl modified carbon nitride were obtained. Fig. 3a reports TGA curves measured for the samples cured at different temperatures (PhCN300, PhCN350, PhCN400 and Ph-Triazine). The analysis suggests a rather different volatilization process for the different samples. The Ph-Triazine shows an immediate and almost complete volatilization at temperatures lower than  $300 \text{ }^\circ\text{C}$ . A similar behavior is present both in PhCN300 and in PhCN350 whilst PhCN400 shows a completely different degradation mechanism. More considerations could be done considering DTG curves (Fig. S4 in supporting Information files). As above mentioned, Ph-Triazine presents a single volatilization peak at  $280 \text{ }^\circ\text{C}$ . Similarly, PhCN300 and PhCN350 samples present a major degradation peak at the same temperature. Moreover, in those samples other degradation peaks at higher temperatures appear ( $320 \text{ }^\circ\text{C}$ ,  $460 \text{ }^\circ\text{C}$  and then  $580 \text{ }^\circ\text{C}$  only in PhCN350). This is consistent with an increase of molecular weight in the samples, in good agreement with all the other experiments performed that showed the formation of dimer, trimer, or larger aggregates (i.e. compounds with larger molecular weight). Interestingly, the higher the temperature of pre-treatment, the less important is the degradation peak at lower temperature. In fact, PhCN400 shows the lowest peak (in terms of integrated intensity) at temperatures lower than  $300 \text{ }^\circ\text{C}$ . Regarding the other degradation peaks, in PhCN400°C DTG curve is present a shoulder at  $380 \text{ }^\circ\text{C}$ , indicating the presence of low temperature polymorphs. Moreover, new peaks appear, with a maximum of degradation rate at  $660 \text{ }^\circ\text{C}$ . In good agreement with the diffraction analysis, the appearance of signals at elevated temperatures are related to graphitic structure, and in general to large molecular units. The formation of heptazine ring is further suggested by the comparison with the DTG analysis of pure graphitic Carbon nitride samples, where degradation signals were observed at  $380 \text{ }^\circ\text{C}$  and  $590 \text{ }^\circ\text{C}$  [32]. Finally, changing from inert to oxidative environment (Fig. S5), samples synthesized up to  $350 \text{ }^\circ\text{C}$  do not show any change in TGA trend, implying their complete degradation during TG analysis in inert atmosphere. On the contrary, PhCN400 sample exhibits a different

behavior, showing an abrupt loss mass. This is consistent with the formation of a certain amount of chair content, which commonly occurs for higher molecular weight compounds [33].

To have a more detailed understanding of the process, a TGA measurement coupled with in-line FT-IR was performed on Ph-Triazine. The thermal treatment performed was selected to simulate the synthesis procedure (Fig. 3b), collecting IR spectra of the volatilized compounds. The heat map reported in Fig. 3c demonstrate the appearance of IR peaks mainly at two wavenumbers: about  $3750 \text{ cm}^{-1}$  and about  $1500 \text{ cm}^{-1}$ . A clearer analysis could be performed analyzing representative peaks selected at different stages of the process (Fig. 3d).

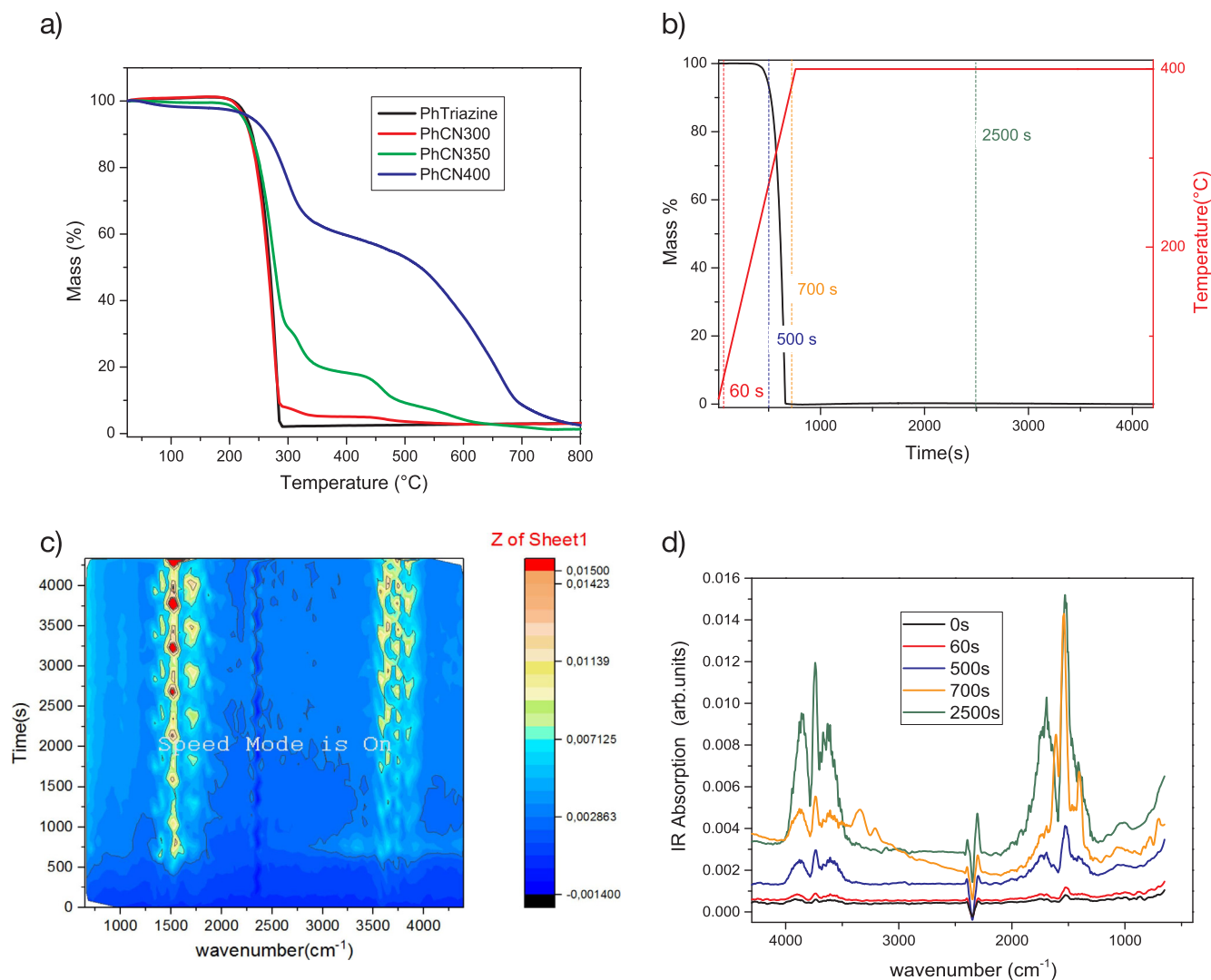
At low temperature (0 s i.e.  $20 \text{ }^\circ\text{C}$  and 60 s i.e.  $50 \text{ }^\circ\text{C}$ ) no peaks were observed, which is related to the fact that no volatile compounds were evolved. Increasing the temperature (spectra at 500 s, i.e.  $270 \text{ }^\circ\text{C}$ ), absorption at  $1540 \text{ cm}^{-1}$  (triazine ring) is observed, indicating the begin of the volatilization of the starting material. Further increasing the temperature (spectra at 700 s, i.e.  $350 \text{ }^\circ\text{C}$ ), the gas species present a spectrum close to phenyl triazine, indicating that all the species are volatilized and available for gas phase reaction [34]. This is clearer in the spectra collected during the isothermal step (spectra at 2500 s), in which a new large band centered at  $1690 \text{ cm}^{-1}$  is visible. This is compatible with the formation of aromatic adducts close to triazine ring, confirming the formation of polymerized structures [34].

### 3.1.6. Differential Scanning Calorimetry measurements

By DSC measurements, we retrieved melting temperatures ( $T_m$ ) and glass transition ( $T_g$ ) temperatures of all the samples (Table 1 and Fig. S6).

Ph-Triazine shows a clear melting temperature at  $233 \text{ }^\circ\text{C}$ ; this value is slightly higher than the reported  $T_m$  ( $220 \text{ }^\circ\text{C}$ ), however this could be related to the heating rate which may shift melting to higher temperatures. Similarly, PhCN300 sample shows a single melting peak at a slightly lower temperature than Ph-Triazine ( $230 \text{ }^\circ\text{C}$ ). This is in good agreement with XRD spectra, which show a similar crystalline structure, even if at lower extent; as a further confirmation, a decrease of enthalpy peak was observed. This trend is more evident in PhCN350, with a further decrease of both  $T_m$  and peak area. Interestingly, in this sample appears a  $T_g$  at  $105 \text{ }^\circ\text{C}$ . This amorphous area could be explained once again taking into consideration the increase of molecular weight in the structures by increasing the temperature of thermal treatment. In fact, the higher the molecular weight, the lower the mobility of the macromolecules, which could lead to have amorphous region. This could clearly be observed in PhCN400 samples, which do not present





**Fig. 3.** TGA curves for the temperature treated samples (a). Thermal curve and relative weight loss (b). The heat map in panel (c) report the evolution of the FTIR spectra registered during the thermal curve. Representative FTIR spectra acquired at different time during the thermal curve (d).

**Table 1**

DSC parameters of PhTriazine and samples prepared at 300, 350, 400 °C. Melting temperatures ( $T_m$ ), glass transition ( $T_g$ ) temperatures and enthalpy change of thermal transitions per unit weight of sample ( $\Delta H$ ).

Sample	$T_m$ (°C)	$T_g$ (°C)	$\Delta H$ (W/g)
PhTriazine	233	–	–22.4
PhCN 300 °C	230	–	–20.9
PhCN 350 °C	221	105	–9.7
PhCN 400 °C	–	115	–

any melting peak but a clear  $T_g$  at 115 °C, which means that the powder is mainly amorphous. This is in good agreement with the observed formation of heptazine structures, which present low mobility due to the large extent of conjugated structures, not allowing the formation of long-range crystal organization. The polymerization scheme in function of the temperature is reported in Fig. 4.

### 3.2. Optical characterization

#### 3.2.1. UV-Vis measurements

The presence of the phenyl group in the carbon nitride structure allows to decrease the absorption bandgap. Indeed, the mixing of  $\pi/\pi^*$  interactions of phenyl group for LUMO and LUMO' results in a

symmetry matching that allows the transitions from HOMO/HOMO' to LUMO/LUMO', forbidden in the pure  $g\text{-C}_3\text{N}_4$  structure [35]. On this view, the number and the position of the phenyl groups has obvious consequence on the main optical properties of the compound. Fig. 5a shows the optical absorption in the UV-visible range of the samples treated at different temperatures, evidencing the different behavior of the samples. A progressive redshift of the optical absorption maxima is clearly visible as the treating temperature is increased.

The absorption spectrum of the starting material exhibits its main maximum at 320 nm (Fig. 5b).

The increase of the annealing temperature at 250 °C results in a slight broadening of the absorption edge towards higher wavelengths. The sample prepared at 300 °C presents a steady increase of the absorption in the 350–550 nm region, whilst keeping relatively unchanged the main absorption band at the higher energy. The sample prepared at 350 °C exhibits a further increase of the absorption in the visible region of the spectrum, showing two additional maxima at 350 and 433 nm, respectively. Finally, the sample prepared at 400 °C exhibits a broad band with the absorption edge red shifted up to 450 nm.

The optical bandgaps of the CN products were calculated according to the Tauc plots  $((\alpha h\nu)^{1/2}$  versus  $h\nu$ , in which  $\alpha$  is the diffuse absorption coefficient,  $h$  is the Planck constant, and  $\nu$  is the light frequency). As reported in Fig. S7 of the supporting material, the bandgap of the starting material and PhCN250 is estimated at about 3.55 eV

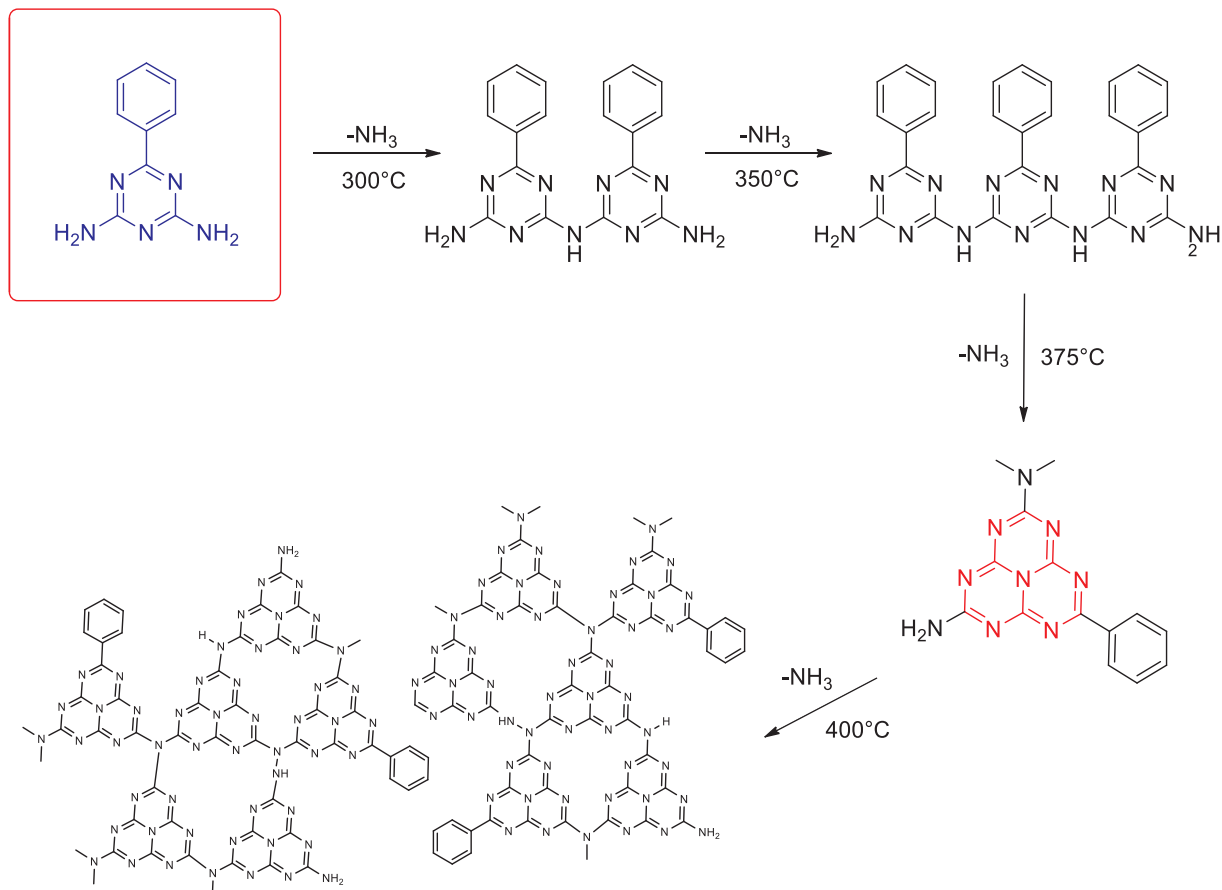


Fig. 4. Polymerization scheme of 6-phenyl-1,3,5-triazine-2,4-diamine.

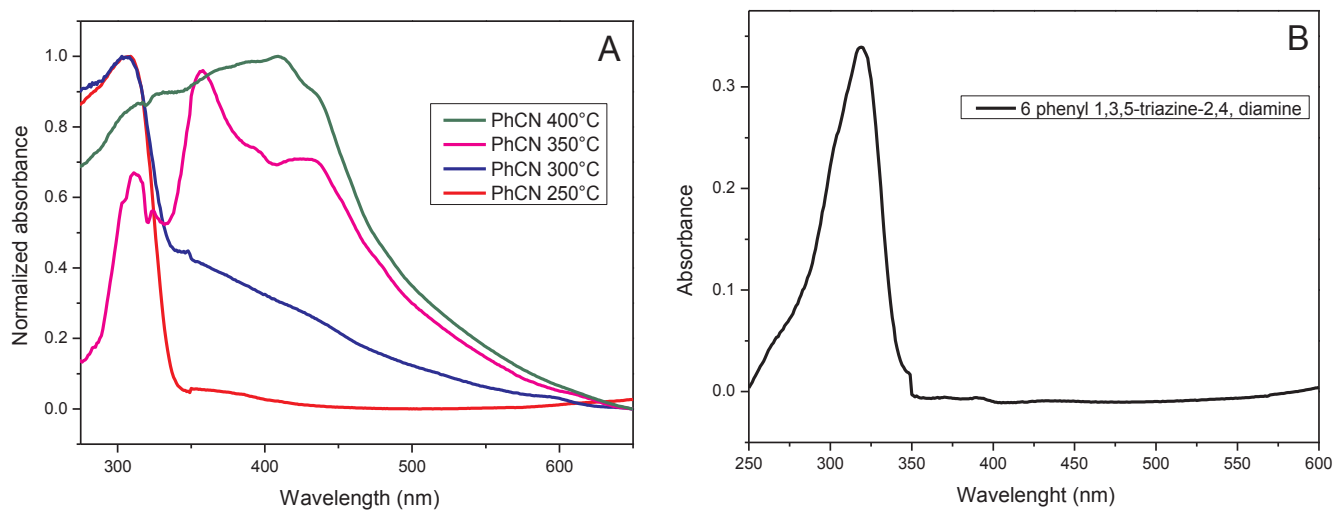


Fig. 5. (a) Absorption spectra of the samples prepared at different temperatures. (b) Absorption spectrum of 6-phenyl-1,3,5-triazine-2,4-diamine.

(350 nm). The samples PhCN300 and 350 show the evidence of multiple absorption edges, indicating that the product is constituted by different polymorphs, as already indicated in the HMRS analysis, with bandgap at about 3.2 and 2.1 eV (387 and 590 nm), respectively. Finally, the absorption onset of the sample PhCN400 presents a clear rise with a bandgap at about 2.0 eV (620 nm). It is worth to note that there is no general agreement in the literature on phenyl modified Carbon Nitride concerning the bandgap, values in the 2.8–2.4 eV being reported, suggesting that different structures were possibly obtained depending on the synthesis properties.

### 3.2.2. Steady State Luminescence measurements

Fig. S8 reports the steady state luminescence spectra of the compounds obtained at different temperatures, showing a progressive redshift of the emissions as the temperature increases. In the framework of different polymorphs synthesized as a function of the increasing temperature, the reported emission redshift should be interpreted in terms of the conjugation length of the  $\pi$  system. It was already pointed out that the aromatic molecules can be easily modelled as a particle, where the optical electron is confined in the aromatic ring system representing a linear box [36]. In this approximation of a quantum model, as the box

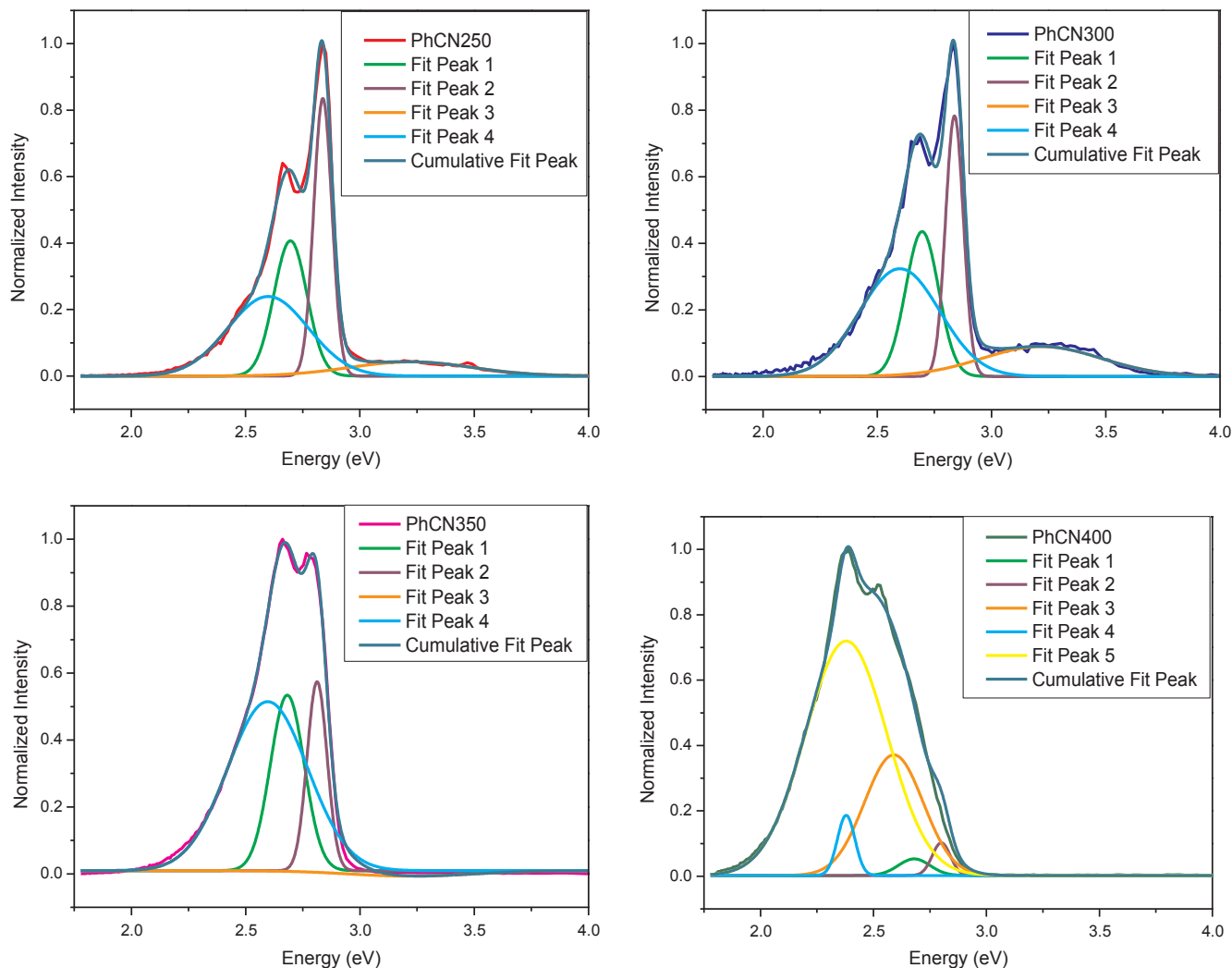


Fig. 6. Gaussian fitting of the PL spectra of the samples prepared at 250, 300, 350 and 400 °C respectively.

dimension increases the energy separation between highest occupied and lowest unoccupied orbitals decreases. Accordingly, an increase of the conjugation length of the  $\pi$  system generates a shift of the emission spectra towards the longer wavelength [36], in good agreement with the reported emission spectra. To outline the contribution of the possible different polymorphs, the emission spectra were analyzed by means of Gaussian bands in the energy space; each band represents a specific radiative channel of recombination, whose relative integrated intensity is related to the contribution of the different species in the samples (Fig. 6).

All the spectra were successfully deconvoluted with 3 or 4 Gaussian bands, whose spectral features are summed up in Table 2.

Taking into account the three levels scheme and the different electronic states in Carbon Nitride systems, ( $sp^3$  C-N  $\sigma$  band,  $sp^2$  C-N  $\pi$  band and the lone pair state of the bridging nitrogen atom), three main

recombination channels are recognized: from  $\sigma^*$  to LP, from  $\pi^*$  to LP and from  $\pi^*$  to  $\pi$  [17,35]. The emission bands extracted by the fitting procedure can be associated to these transitions (Fig. 7).

It is worth to note that the contribution of the higher energy band progressively decreases by increasing the temperature of synthesis, whilst the one of the large bands at lower energy increases, two more bands are required to fit the large emission at low energy for the sample cured at 400 °C. Indeed, as the conjugation length increases, the contribution from the  $\pi$  level increases with a progressive depletion of the  $\sigma$  level due to non-radiative energy transfer to the  $\pi$  levels. Thus, the high energy contribution is reduced and the one at lower energy is increased because of the increase of the reaction degree with increasing curing temperature [37].

Table 2

Gaussian fit results of samples: PhCN250, PhCN300, PhCN350 and PhCN400 as obtained in Fig. 5.

Sample	Band 1			Band 2			Band 3			Band 4			Band 5		
	Xc (eV)	FWHM	Area	Xc (eV)	FWHM	Area	Xc (eV)	FWHM	Area	Xc (eV)	FWHM	Area	Xc (eV)	FWHM	Area
PhCN250	3.22	0.62	0.28	2.83	0.09	0.083	2.69	0.17	0.073	2.56	0.41	0.010	-	-	-
PhCN300	3.22	0.61	0.06	2.83	0.09	0.076	2.69	0.17	0.078	2.58	0.412	0.142	-	-	-
PhCN350	-	-	-	2.81	0.10	0.06	2.68	0.17	0.09	2.59	0.412	0.22	-	-	-
PhCN400	-	-	-	2.81	0.09	0.01	2.68	0.17	0.01	2.59	0.30	0.12	2.38	0.42	0.324

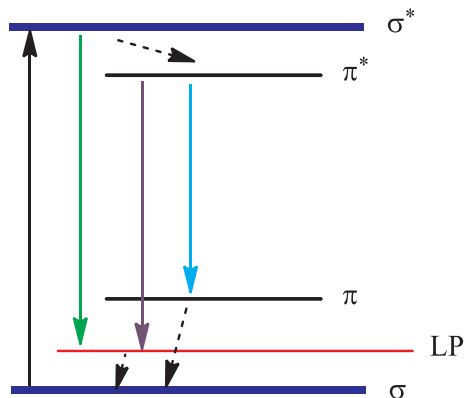


Fig. 7. Energy levels scheme in PhCN polymorphs.

3.2.3. Time resolved measurements

To further characterize the optical properties of PhCN samples we performed time resolved measurements exciting at 350 nm (well above the optical bandgap of all the samples) and analyzing the decay time profiles with a multiexponential decay function, [38]

$$y = y_0 + \sum_i A_i e^{-\frac{t}{\tau_i}}$$

Fit results are presented in Fig. 8, and the retrieved decay parameters are presented in Table 3.

Table 3  
Time parameters obtained from fits in Fig. 8.

Sample	A1	t1 (μs)	A2	t2 (μs)	A3	t3 (μs)
PhCN250	562.3	0.935	–	–	–	–
PhCN300	529.7	0.999	3810.4	0.425	–	–
PhCN350	501	0.901	4695	0.173	–	–
PhCN400	43.6	0.745	809.91	0.151	2676.53	0.03667

Different time behaviors are observed depending on the thermal treatment. In general, as the curing temperature increases the time profile appears faster and more complicated. Sample cured at lower temperature presents single, very slow recombination channel, with lifetime of almost 1 ms ( $\tau = 0.935$  ms). By increasing the curing temperature, the decay shortens, and two new faster components are required to successfully fit the time profile, with lifetime of 425 ns and 173 ns for the PhCN300 and PhCN350 samples, respectively. The time profile of the sample treated at 400 °C (PhCN400) is even faster, with a very small contribution in the microsecond region and two decays at 151 and 36 ns. This is a further indication that the synthesis conditions modify both the structure and the optical features of Phenyl modified g-C<sub>3</sub>N<sub>4</sub> compounds, time and spectral tunability being achieved by different thermal treatments. In addition, due to the large red shift of PhCN400 sample, we measured the decay time by exciting in the visible region (at 410 nm, but still above its bandgap). The time decay and the spectral profile seem to be not affected by the variation of the excitation

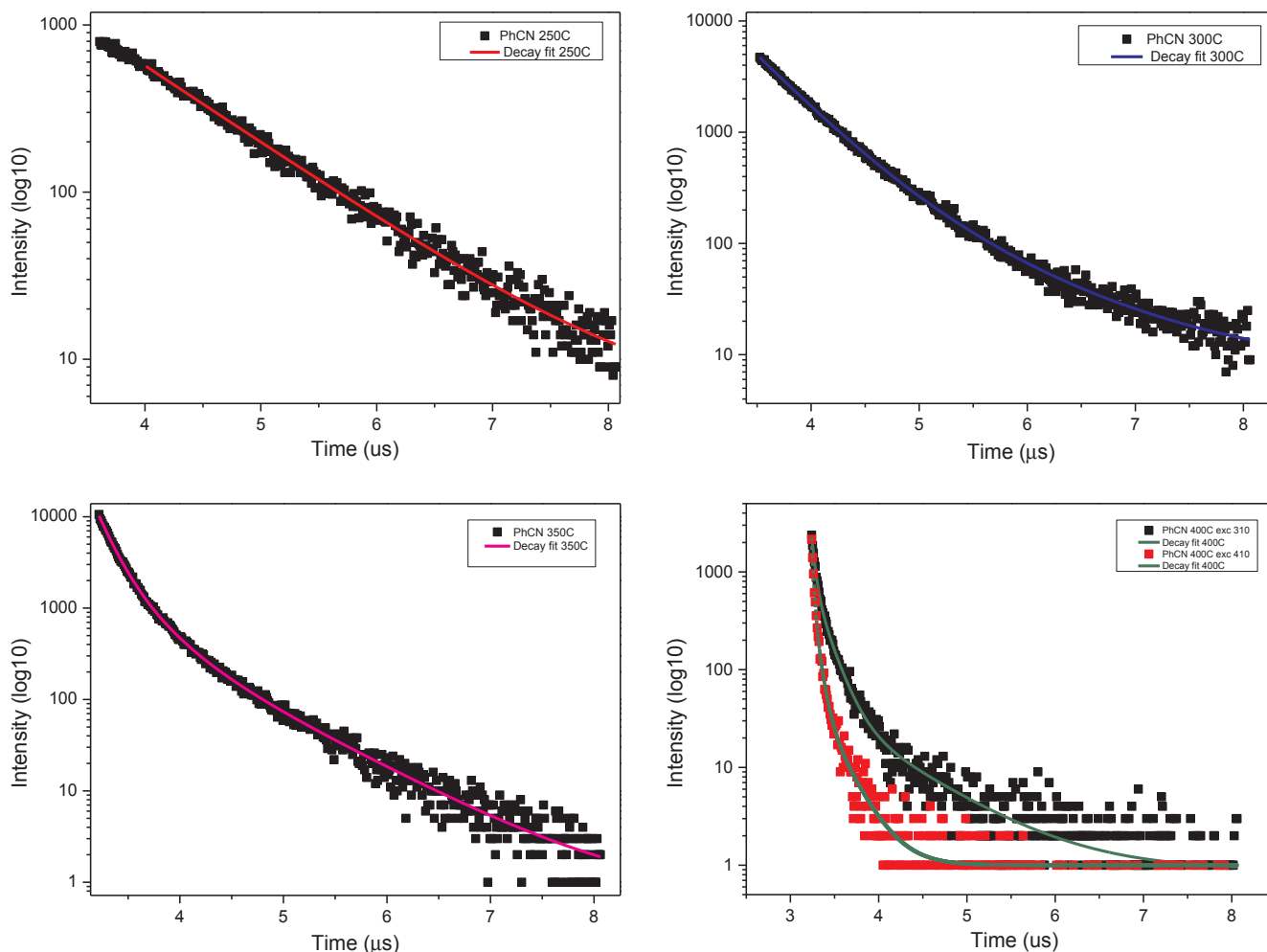
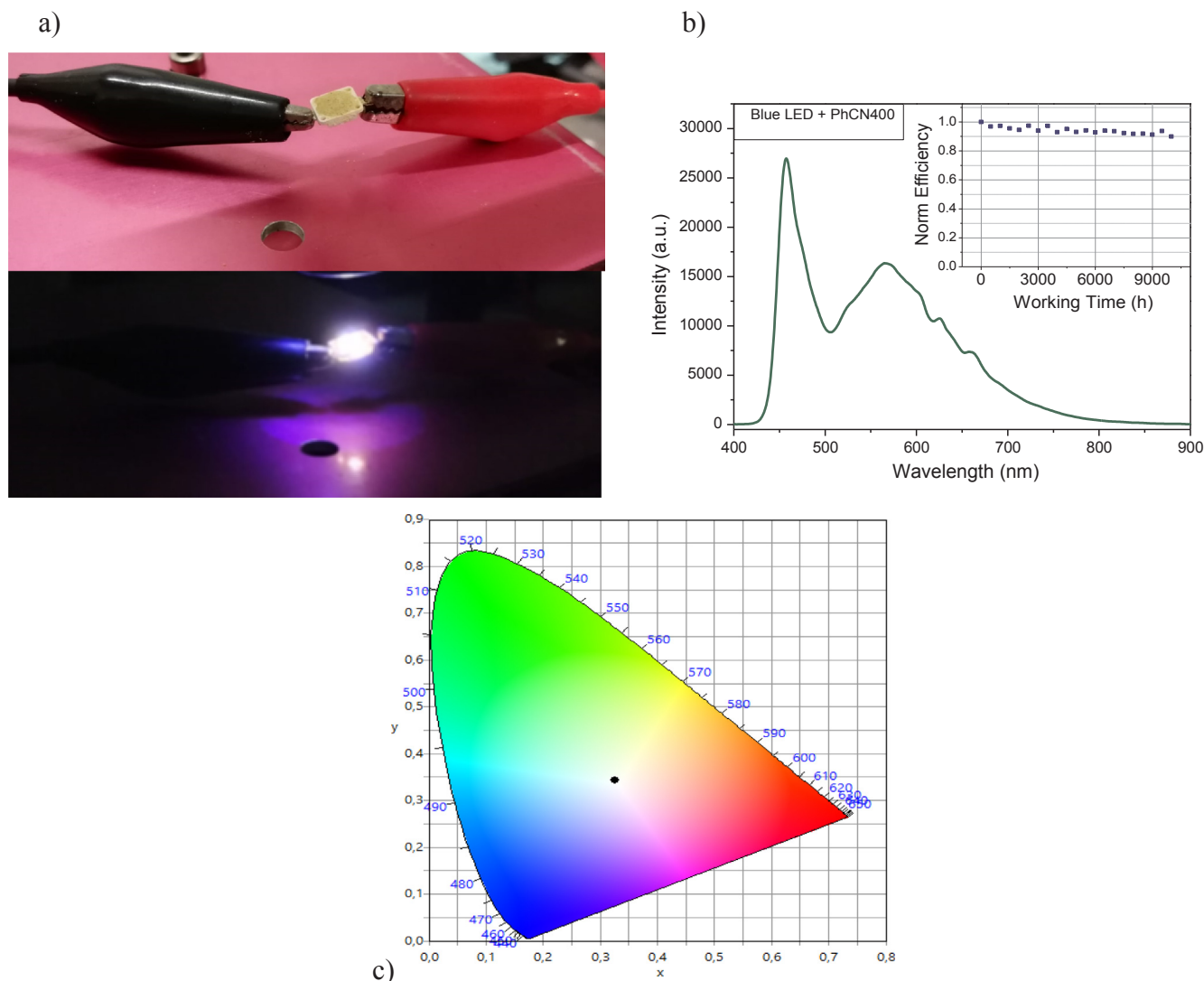


Fig. 8. Time decay photoluminescence fits of PhCN250, PhCN300, PhCN350 samples with a 310 nm excitation wavelength and PhCN 400 °C sample with a 310 and 410 nm excitation wavelength, acquired over a 5 μs time window.





**Fig. 9.** (a) Prototype of a White LED. (b) Steady state luminescence spectrum of prototype. (c) CIE. The inset in the panel B reports the overall emission registered for continuously working time regime.

wavelength. The time spectral profiles were acquired over two different time range, 100 ns and 2 ms (Fig. S9), to enlighten possible different contributions to the overall emission spectrum. As evidenced, the spectral profile shows the presence of at least two contributions, one at high energy, characterized by a faster decay, and one at lower energy with a slower decay time. This trend resembles the decay kinetics of pure  $g\text{-C}_3\text{N}_4$ , where two decay times were observed and related to the two excited levels ( $\sigma^*$  and  $\pi^*$ ) [16,35]. The recombination from the highest conduction level  $\sigma^*$  is affected to a larger extent by non-radiative process related to the non-radiative interaction among planes and, hence, presents a faster decay time. However, the time scale of the recombination in the  $g\text{-C}_3\text{N}_4$  (in the ns time domain) is very different with respect to the one of Phenyl modified CN compounds. In that case a very reduced emission efficiency with an estimated Quantum Yield (QY) of about 7% is observed, because of the presence of non-radiative recombination channels between planes. On the contrary, the estimated QY of PhCN400 sample was of 62%, and very similar values were measured for all the other samples (55% for PhCN250, 51% for PhCN300, and 57% for PhCN350).

#### 4. Application

The optical features above reported, and in particular the ones of

the PhCN400 sample, promote PhCNs as good phosphors for white LED (WLED) applications. White light in LED is generally achieved by combing the monochromatic emission of a pumping LED (generally chosen at about 450 nm, as a good compromise between short wavelength emission and eye sensitivity) with a broad emission in the green to red spectral region (500–650 nm) of a suitable phosphor. Research of new phosphors is very active and partially driven by the necessity to substitute Ce:YAG, because of its low Color Rendering Index, and the high supply disruption of Rare Earth Elements [1]. The optical properties of PhCN400 match very well the excitation and the emission requirements for WLED applications, coupled with high quantum efficiency and fast decays to avoid saturation effect. In addition, a high structural and thermal stability is required, since the phosphor is located close to the LED chip and only approximately 80% of the input power is converted to optical power; the remaining power produces instead heat to be dissipated by the device. Given the small chip area and the limited phosphor area, temperatures from 100 °C up to 200 °C can be easily reached close to the chip, thus the thermal resistance up to 400 °C of PhCN compounds is a very boosting feature for these organic compounds that can be prepared by a very facile and low-cost synthesis procedure. For this reason, a first prototype of a WLED was realized, with the combination of a blue LED (Golden Dragon Osram) with the main emission at 465 nm, with the PhCN400 compound deposited in

close contact with the junction (Fig. 9a). The amount of the organic compound was set to partially transmit the emission from the pumping LED and to use the absorbed light to generate the broad emission in the 500–650 nm spectral range (Fig. 9b). The prototype generates an overall emission (LED + phosphor) with CIE coordinates of  $x = 0.3256$  and  $y = 0.3437$  for an optimal white color, associated to a Color Rendering Index of 88. (Fig. 9c). The prototype was tested setting the current at 250 mA and monitoring the emission for about 10,000 h (inset panel 9b). The spectral shape remained almost stable in that period and the emission intensity remains stable above 90% with respect to the starting one.

## 5. Conclusions

The correlation between structural and optical properties in phenyl modified g-C<sub>3</sub>N<sub>4</sub> was analyzed. Different structures were observed as a function of the temperature in the thermal solid-state reaction of the starting material (6-phenyl-1,3,5-triazine-2,4-diamine powder). Firstly, the formation of dimer/trimer conjugated triazine units were observed with subsequently, the formation of the heptazine structure and, finally, the formation of the g-C<sub>3</sub>N<sub>4</sub>-like structures. A progressive red shift of the main optical features (absorption/emission) were observed as the conjugation increases.

The optimal sample for lighting application was identified in the phenyl modified g-C<sub>3</sub>N<sub>4</sub> sample, obtained with thermal treatment at 400 °C, that presents the optical absorption tail at 450 nm and a broad, emission in the green-red part of the visible range. These features, associated with high internal quantum yield (above 60%), and high thermal and flux stability, were proven suitable for efficient white LED emission.

Finally, the similarity in the structure of the sample with the g-C<sub>3</sub>N<sub>4</sub> but for the optical absorption shifted in the visible range, strongly suggests an easy achievable way to tune and control the properties of such polymorphs for other applications, like photocatalysis.

## Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

## Appendix A. Supplementary material

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.apsusc.2019.144330>.

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