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# Modification of commercial UF membranes by electrospray deposition of polymers for tailoring physicochemical properties and enhancing filtration performances

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Elizaveta Korzhova, Sébastien Déon, Zakaryae Koubaa, Patrick Fievet, Dmitry Lopatin, et al.. Modification of commercial UF membranes by electrospray deposition of polymers for tailoring physicochemical properties and enhancing filtration performances. *Journal of Membrane Science*, 2020, 598, pp.117805. 10.1016/j.memsci.2019.117805 . hal-02469682

**HAL Id: hal-02469682**

**<https://hal.science/hal-02469682>**

Submitted on 21 Jul 2022

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1                    **Modification of commercial UF membranes by**  
2                    **electrospray deposition of polymers for tailoring**  
3                    **physicochemical properties and enhancing filtration**  
4                    **performances**

5  
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18                    Submitted to

19                    **Journal of membrane Science**

20                    2019

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## 30 **Abstract**

31 The main challenge for a widespread use of nanoporous membranes in the removal of ionic  
32 contaminants lies in the adjustment of their physicochemical properties to allow adequate ion  
33 rejection and mitigate fouling based on the targeted application. Most of the commercial  
34 membranes are negatively charged and their use is thus not necessarily relevant for divalent  
35 cation rejection. The main objective for researchers is therefore to provide novel tailored  
36 membranes by developing specific synthesis or modifying available membranes. It is proposed  
37 here to tailor physicochemical properties of a commercial low molecular weight cut-off  
38 ultrafiltration membrane by electrospray deposition of polyethylenimine (PEI) and polystyrene  
39 sulfonate (PSS). In this study, it is highlighted that, with adequate conditions, it is possible to  
40 adjust the charge of the membrane surface, which can reach values from -40 to +40 mV  
41 (compared with -20 mV for pristine membrane). Surface hydrophilicity has also been increased  
42 with a contact angle decreased from 60 to 30° with a PSS surface layer. In terms of filtration  
43 performances, it is shown that the permeation flux is not reduced by the electrospray deposition  
44 of polymer and can even be slightly enhanced in specific conditions. When polymer  
45 concentration is sufficient, the deposit is able to face the shear stress induced by cross-flow  
46 filtration, probably due to viscosity effect. The positive PEI surface layer leads to a strong  
47 enhancement in the rejection of divalent cations whereas that of divalent anions is notably  
48 decreased due to electrostatic interactions between the charge of divalent ions and that of the  
49 membrane. The weaker impact of electrostatic interactions on monovalent ions allows  
50 adjustment of the separation selectivity between cations and anions as well as between mono-  
51 and divalent ions. Finally, it is also demonstrated that rejection performances are mostly  
52 governed by the surface layer, even if the underlying deposit layers and membrane also have a  
53 non-negligible impact on ion rejection.

## 54 **Keywords**

55 Ultrafiltration membrane; electrospray deposition; layer-by-layer assembly; ion rejection;  
56 zeta potential; hydrophilicity.

## 57 **I. Introduction**

58 In the perspective of sustainable development and environmental conservation, membrane  
59 processes appear as growing technologies for water treatment and pollution removal [1]. Indeed,  
60 they exhibit several advantages compared to other technologies due to low energy requirements,  
61 weak detrimental impact on environment, easy implementation, and high separation selectivity  
62 between components. Therefore, they currently represent one of the most viable options to  
63 remove a broad range of contaminants from polluted waters due to the wide variety of processes  
64 and materials available. For instance, membrane processes are a competitive technique to remove  
65 ionic or ionizable contaminants from polluted wastewaters by combining both steric effects and  
66 electrostatic interactions [2, 3].

67 Unfortunately, commercial membranes are often inadequate for particular applications,  
68 especially for removing multivalent cations such as heavy metals [4]. Moreover, their fouling  
69 also turns out to be a major drawback, impeding large-scale implementation [5, 6]. To overcome  
70 these issues, the development of specific membranes with adapted structural or physicochemical  
71 properties has become a key challenge for the expansion of membrane technologies. The target  
72 objective is to impart antifouling and/or antibacterial properties, and to obtain the best trade-off  
73 between permeability and separation selectivity [7-9]. This goal can be achieved by either  
74 synthesizing more effective membranes (containing mixed polymers or additives) [10-15] or  
75 improving surface properties through chemical or physical modifications (coating, grafting,  
76 etc...) [16-19]. Modifying commercial membranes by coating them with polyelectrolytes  
77 (charged polymers) seems to be a relevant way but usual methods show some drawbacks and the  
78 development of new modification methods remains a challenge for the membrane community.  
79 Lots of routes are available in literature for surface coating (*e.g.* dip-coating [20, 21], spin-coating  
80 [22, 23], dynamic coating [24-26], electrodeposition [27, 28]) but dip-coating is often chosen due  
81 to its easy implementation. The latter, which consists in immersing the pristine membrane in a

82 polymer solution, is often used but its potential use for industrial application is impeded by the  
83 large volume of polymer solution required for membrane immersion [29]. Spray deposition is an  
84 alternative way to reduce polymer quantity and deposition time [30, 31]. However, its use is often  
85 restricted to loading nanoparticles onto membrane surfaces [32-34].

86 In this study, the original technique of electrospray (ES) is proposed to deposit a small quantity  
87 of polymer on the membrane surface. The electrospray deposition is a common technique which,  
88 up to now, has not often been used for membrane modification [35]. With this technique, fine  
89 droplets of polymer solution are sprayed on the membrane surface under a high voltage between  
90 the needle containing the solution and the metallic support on which the membrane is stuck. The  
91 advantages of this process lie in the small quantities of polymer that are coated on the membrane  
92 surface (compared with immersion technique) and the inhomogeneous dispersion of the polymer  
93 at the surface. In this study, two commercial membranes were modified by deposition of two  
94 polyelectrolytes, namely polyethyleneimine (PEI) and polystyrene sulfonate (PSS). The impact  
95 of various deposits obtained for various volumes and concentrations of sprayed polymer  
96 solutions, was investigated on filtration performances, such as permeation flux, ion rejection and  
97 separation selectivity, and discussed in terms of surface charge and hydrophilicity.

98

## 99 2. Material and methods

### 100 2.1. Membrane and solutions

101 Impact of polymer electrospray deposition was investigated on two ultrafiltration commercial  
102 flat-sheet membranes: a PLEIADE polyethersulfone (PES) membrane supplied by Orelis  
103 Environnement SAS (France) and a Desal GK thin film composite (TFC) membrane made up  
104 of a polyamide (PA) ultrathin layer on a polysulfone support (PS) supplied by GE Water &  
105 Process Technologies (Trevose, USA). Properties of the two membranes are summarized in  
106 Table 1.

107 **Table1:** Properties of the two investigated membranes

Membranes	MWCO* (Da)	Maximum pressure** (bar)	Wet thickness** ( $\mu\text{m}$ )	isoelectric point**	mean pore radius** (nm)	Intrinsic Permeability ** (m)
<b>Plejade</b>	3000	5	190	3.4	2.5	$8.4 \times 10^{-14}$
<b>Desal GK</b>	3000**	27	135	4.5	1.8	$6 \times 10^{-15}$

108 \* provided by supplier

109 \*\* assessed during this study

110 \*\*Molecular weight cut-off was provided by supplier, but the value seems notably overestimated

111

112 Solutions were prepared by dissolving the organic and mineral solutes in ultrapure water (18.2  
113 M $\Omega$ /cm). All experiments were carried out at the natural pH of water, namely  $5.7 \pm 0.2$ .

114 The two polymers used for deposition were polyethylenimine (PEI, branched) with a molecular  
115 weight of ~25,000 and polystyrene sulfonate (poly(sodium-p-styrene sulfonate), PSS) with a  
116 molecular weight of 70,000, which were purchased from Sigma-Aldrich (USA) and Acros  
117 organics (USA), respectively.

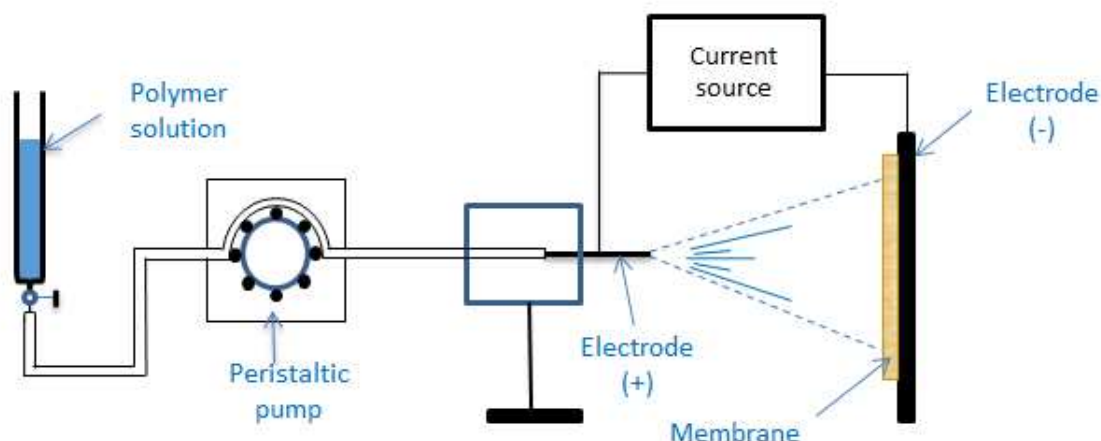
118 Deposition was carried out by spraying various volumes of polymer solutions. The polymer  
119 concentration was varied from  $1.5 \times 10^{-2}$  to 1.5 mol/L of functional groups (*i.e.* amino and  
120 carboxyl groups, respectively for PEI and PSS).

121 Concentrations of  $10^{-3}$  mol/L for single salt solutions (namely NaCl, MgCl<sub>2</sub> and Na<sub>2</sub>SO<sub>4</sub>) and  
122  $5 \times 10^{-4}$  mol/L for salt mixtures were chosen. Salts were supplied by Fisher Scientific, UK. Salt

123 and sucrose concentrations were estimated by ionic chromatography 883 Basic IC Plus  
124 (Metrohm) and refractometer RFM970-T (Bellingham & Stanley, supplied by Xylem  
125 Analytics), respectively.

## 126 **2.2. Electrospray deposition**

127 The modification of membrane properties was obtained by coating a small amount of polymer  
128 solution (PEI or PSS) on the surface by electrospray deposition. This technique, which was  
129 accurately described in literature [36], consists in spraying fine droplets of polymer solution by  
130 applying an electric current between a needle containing the polymer solution and a metallic  
131 support on which the membrane is stuck. The setup used in this study for deposition is depicted  
132 in Fig. 1.



133

134 **Fig. 1:** Scheme of the setup used for electrospray deposition

135 **Polymer is firstly dissolved in aqueous solution.** The polymer solution is pushed into a metallic  
136 needle at a constant flow rate by a peristaltic pump. An electrical field is generated by a high-  
137 voltage regulated DC current source (XP Power Q101-5) between the needle and an aluminum  
138 plate (100 cm<sup>2</sup> area) used as counter electrode. The distance between the needle tip and the  
139 grounded target is 7 cm, the applied voltage was varied from 7 to 12 kV, and the flow rate was  
140 1 mL/min. Deposition was carried out at ambient temperature. **The needle and thus the spray**

141 can be moved to cover all the membrane surface and the duration of spraying is calculated  
142 according to membrane area, *i.e.* 140 cm<sup>2</sup> for filtrations and 30 cm<sup>2</sup> for characterizations (IR,  
143 electrokinetic, contact angle measurements).

144 In order to determine the optimal conditions, the influence of sprayed volume and concentration  
145 of monomer unit (amine or sulfonate groups) was investigated in the ranges of 5-50 μL/cm<sup>2</sup>  
146 and 0.15-1.5 mol/L, respectively.

147 Membranes were firstly immersed in a solution of KCl 10<sup>-3</sup> mol/L before being coated by  
148 spraying alternatively positively charged polymer (PEI) and negatively charged polymer PSS  
149 (*i.e.* PEI, PSS/PEI and PEI/PSS/PEI) so that electrostatic attraction enables adhesion of the  
150 sprayed layer on the previous layer (*i.e.* previously coated polymer or negatively charged  
151 membrane surface). Membrane always remains wet and no drying step is implemented, so that  
152 polymer remains dissolved during all the procedure.

### 153 ***2.3. Filtration experiments***

154 A cross-flow lab-scale pilot was used in this study for ultrafiltration of solutions by flat-sheet  
155 organic membrane. Flow rate is induced by a volumetric pump whereas transmembrane  
156 pressure is generated by partially closing a regulation valve. Experiments were carried out in  
157 full recycling mode (both permeate and retentate streams) to maintain feed concentration  
158 constant and avoid variation of performances over time [37]. Circulation of cold water within  
159 the feed tank jacket was used to set temperature at 25 ± 1°C. Filtrations were carried out at three  
160 applied pressures (5, 15 and 25 bar) but results are provided here only for ΔP = 15 bar in order  
161 to minimize experimental data and simplify graphs. It was not possible to duplicate each  
162 measurement and error bars cannot therefore be displayed. However, few experiments were  
163 repeated to validate the reproducibility, and discrepancies were lower than 5%. Moreover, it  
164 should be mentioned that experiments before and after modification were always carried out  
165 with the same membrane sample to overcome discrepancy between membrane sheets.



166 Performances obtained with pristine and modified membranes were estimated by calculating  
167 permeation flux  $J_v$  and ion rejection  $R_i$  with Eqs. 1 and 2 from the mass of permeate  $m_p$  measured  
168 during a filtration time  $\Delta t$  and feed  $C_{i,f}$  and permeate  $C_{i,p}$  concentrations.

$$169 \quad J_v = \frac{m_p}{\Delta t \rho S_m} \quad (1)$$

$$170 \quad R_i = 1 - \frac{C_{i,p}}{C_{i,f}} \quad (2)$$

171 where  $S_m$  is the membrane area and  $\rho$  the solution density.

## 172 **2.4. Membrane characterization**

### 173 **2.4.1. Zeta potential measurements**

174  $\zeta$ -potential of the external surface was measured before and after polymer spraying by  
175 Tangential Streaming Current (TSC) measurements, which were carried out with a ZetaCAD  
176 zetameter supplied by CAD Inst. The principle and setup were already accurately described in  
177 previous papers [38]. Briefly, nitrogen gas is injected in the feed tank, which leads to flow of  
178 the electrolyte solution (KCl  $10^{-3}$  mol/L) across a rectangular channel made by two identical  
179 membranes facing each other. A pressure drop through the channel  $\Delta P$  is induced by this flow.  
180 Additionally, excess of counter-ions within the diffuse layer is carried by the solution flow  
181 generating a streaming current  $I_s$ , which is measured by two Ag/AgCl wire electrodes for each  
182 pressure drop. The slope of  $I_s = f(\Delta P)$  can then be used to estimate the  $\zeta$ -potential value by Eq.  
183 3, provided that the channel geometry is known [39]:

$$184 \quad \zeta = \left( \frac{l_c}{h_c L_c} \right) \frac{\eta}{\varepsilon_0 \varepsilon_r} \times \left( - \frac{I_s}{\Delta P} \right) \quad (3)$$

185 with  $l_c$ ,  $L_c$  and  $h_c$  the channel length, width and height, respectively, and  $\eta$ ,  $\varepsilon_0$  and  $\varepsilon_r$  the dynamic  
186 viscosity, vacuum permittivity and the dielectric constant of the solution.

187

188 **2.4.2. Surface hydrophilicity estimation**

189 Variation of the membrane surface hydrophilicity was estimated by contact angle  
190 measurements. On each sample, 20 drops of ultrapure water were placed at different positions  
191 on the dry membrane surface at ambient temperature. Images obtained with a digital video  
192 camera were processed using computer program and contact angles were measured 10 seconds  
193 after applying the drop. Then, average values and standard deviation were calculated for  
194 discussion.

195 **2.4.3. Surface chemical analysis**

196 The surface of the membrane before and after deposition was chemically investigated by  
197 attenuated total reflection - Fourier transform infrared (ATR-FTIR) spectroscopy. A FTIR  
198 spectrometer iS50 (Thermo Scientific) with ATR attachment VarigATR (Harrick) were used  
199 with an angle of 65° and spectra were acquired in the spectral region between 800 and  
200 4000 cm<sup>-1</sup> with 2 cm<sup>-1</sup> resolution.

201 **2.4.4. Morphological analysis**

202 The morphology (roughness) of membrane surface was investigated by AFM. Images of  
203 membrane surface (with and without deposition of PEI and PSS) were performed in contact  
204 mode with the commercial Nano Observer atomic force microscope from CSI (France). The  
205 treatment of AFM pictures and roughness assessment were performed using MountainsMap®  
206 software from Digital Surf.

207 Images of the membrane cross section were obtained by Scanning Electron Microscopy (SEM)  
208 after Focused Ion Beam (FIB) irradiation. To avoid charge accumulation on the surface, the  
209 membrane surface was covered with a chromium layer of 20 nm by cathodic sputtering. The  
210 membrane surface was irradiated by the focused Ga<sup>+</sup> ions (FIB Orsay-Physic with ionic column  
211 CANION 31). After this irradiation step, the cross-section was observed by SEM (Gemini  
212 Column, Zeiss) with a tilt of 50°.

213 **2.4.5. Structural characterization**

214 Membrane intrinsic permeability and mean pore radius were estimated from filtration of pure  
 215 water and neutral solutes, namely sucrose and PEG 2000 for Desal GK and PLEIADE  
 216 membranes, respectively.

217 Membrane intrinsic permeability  $L_p$  was assessed from pure water flux  $J_w = f(\Delta P)$  by Darcy's  
 218 law:

$$219 \quad J_w = \frac{L_p}{\eta} \Delta P \quad (4)$$

220  $L_p$  is therefore a structural property since viscosity is not included in this parameter. Hence,  
 221 considering a laminar flow in cylindrical pores, Hagen-Poiseuille equation can be coupled with  
 222 Darcy's law to obtain the expression of  $L_p$ :

$$223 \quad L_p = \frac{\varepsilon r_p^2}{8L} \quad (5)$$

224 A change in membrane intrinsic permeability can be imputed to variations in mean pore radius  
 225  $r_p$ , porosity  $\varepsilon$  or pore length  $L$  (*i.e.* membrane thickness if tortuosity is neglected).

226 Mean pore radius is estimated by fitting rejection curves  $R = f(J_v)$  with a steric hindrance model  
 227 [40, 41] (Eqs. 6, 7, 8) for which, mean pore radius is the only adjustable parameter, diffusivities  
 228 at infinite dilution  $D_{i,\infty}$  and Stokes radius  $r_{i,s}$  of neutral solutes being retrieved from literature.

$$229 \quad R = 1 - \frac{\phi_i K_{i,c}}{1 - \left[ (1 - \phi_i K_{i,c}) \left( \exp\left(-\frac{K_{i,c} \Delta P}{8K_{i,d} D_{i,\infty} \eta} r_p^2\right) \right) \right]} \quad (6)$$

230 where  $\phi_i$ ,  $K_{i,c}$ ,  $K_{i,d}$  are the steric partitioning coefficient, and convection and diffusion hindrance  
 231 factors, respectively, calculated by [42]:

$$232 \quad \phi_i = \left( 1 - \frac{r_{i,s}}{r_p} \right)^2 = (1 - \lambda_i)^2 \quad (7)$$

$$233 \quad K_{i,c} = (2 - \phi_i)(1 + 0.054\lambda_i - 0.988\lambda_i^2 - 0.44\lambda_i^3) \quad (8)$$

234

$$K_{i,d} = 1 - 2.30\lambda_i + 1.154\lambda_i^2 + 0.224\lambda_i^3 \quad (9)$$

235

## 236 **3. Results & Discussion**

### 237 **3.1. Influence of electrical field**

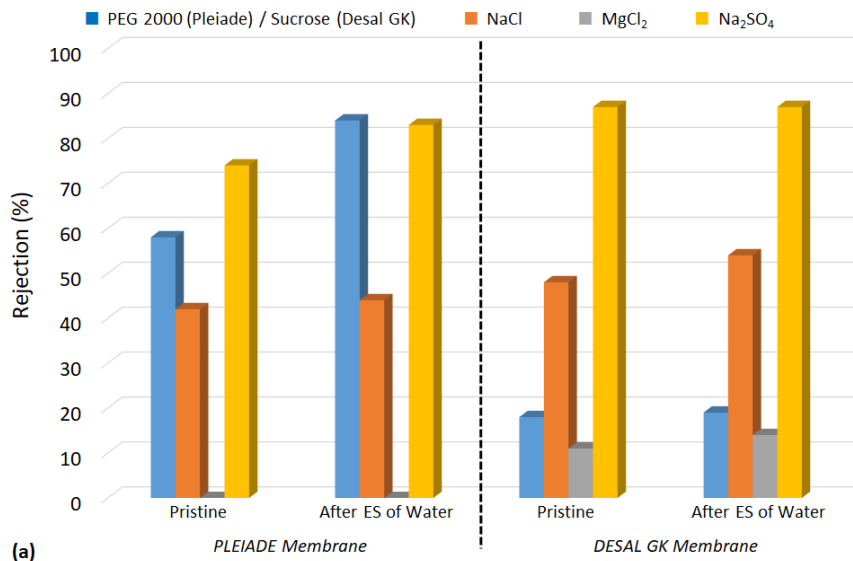
238 Electro spray deposition technique requires a high voltage, which can potentially have an  
239 influence on the membrane polymer. In a first step, the influence of the electrical field on  
240 membrane was investigated by spraying only water, so that modification of membrane  
241 performances cannot be attributed to deposition. The applied voltage was chosen at 12 kV and  
242 the impact was studied by estimating the rejection  $R$  and permeation flux  $J_v$  of salts and neutral  
243 solutes (*cf.* Fig. 2) as well as **intrinsic** permeability  $L_p$  and mean pore radius  $r_p$  (*cf.* Table 1),  
244 obtained with PLEIADE and GK membranes before and after water electro spray.

245 From Fig. 2, it can be seen that permeation fluxes obtained with PLEIADE membrane clearly  
246 decrease after spraying water, which cannot be attributed to any deposit. Rejection of salts was  
247 not significantly impacted except that of  $\text{Na}_2\text{SO}_4$ , which was slightly increased by electro spray.  
248 Oppositely, electrical field involved in electro spray of water seems to enhance flux of Desal  
249 GK membrane, whereas rejection is not noticeably affected, except a very slight increase in  
250  $\text{NaCl}$  rejection.

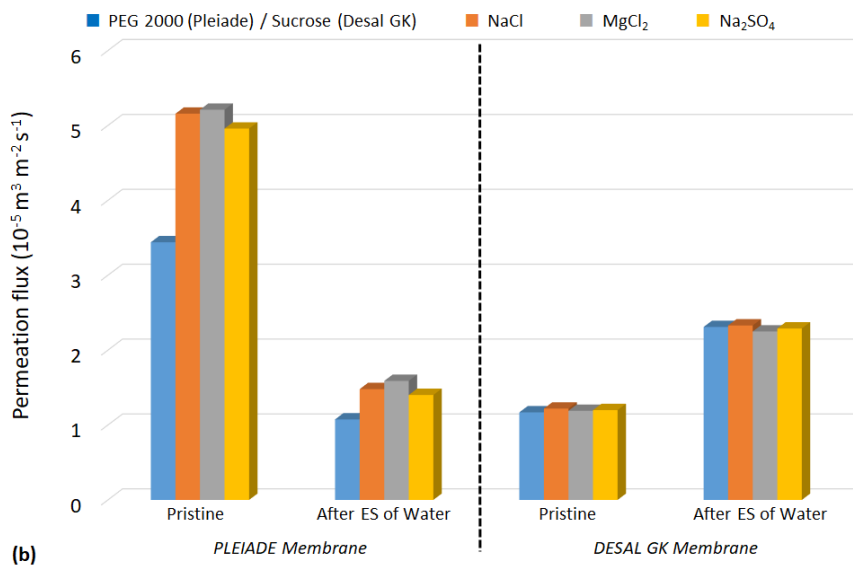
251 The rejection of salts is governed by both steric and electric mechanisms. The fact that salt  
252 rejection is not impacted tends to show that membrane charge is probably not strongly affected  
253 by electrical field. The large size of  $\text{SO}_4^{2-}$  ion compared with other ions can probably explain  
254 the slight increase of its rejection due to a stronger steric exclusion. Flux and rejection of neutral  
255 solutes are mainly governed by structural properties. The values of **intrinsic** permeability  $L_p$  and  
256 the mean pore radius  $r_p$ , **estimated by fitting pure water flux and rejection of neutral solutes**  
257 (PEG 2000 for PLEIADE membrane and Sucrose for Desal GK) **with Eqs. 4 and 6**, before and  
258 after water electro spray, are provided in **Table 2** for deeper discussion.

259

260



261



262

263 **Fig. 2:** Rejection (2a) and permeation flux (2b) of salts and neutral solutes obtained with PLEIADE  
 264 and GK membranes, before and after electro-spray of water with a voltage of 12 kV

265 **Table 2** demonstrates that structural properties of the PLEIADE membrane (mean pore radius  
 266 and **intrinsic** permeability) are clearly affected by the application of an electrical field through  
 267 the membrane, whereas only permeability seems to be impacted for the GK membrane. Indeed,  
 268 **intrinsic** permeability of PEIADE membrane was decreased by 60% and mean pore radius by  
 269 almost 30% because of the electrical field applied for spraying solutions. This means that the  
 270 PES constituting this membrane is probably damaged by the technique. **Indeed, the electric field**  
 271 **may cause local temperature increase, which could expand membrane polymer and therefore**

272 lead to a decrease of mean pore size (increase in PEG rejection) and permeation flux.

273 Oppositely, the polyamide of the Desal GK is positively affected by electric field since an

274 increase in intrinsic permeability is observed. However, the mean pore radius after water

275 electrospray is unchanged. Considering the expression of intrinsic permeability (Eq. 5), the

276 increase in permeability induced by electric field can be attributed to an increase in porosity or

277 a decrease in membrane thickness. It is perhaps possible that the electrical field passing through

278 the membrane leads to the creation of additional pores within the active layer (without changing

279 the mean pore size) and/or the shrinkage of this thin layer.

280 In any case, the aim of this study being to optimize filtration performances, the rest of this study

281 has been implemented only with Desal GK membrane in order to exploit the positive impact of

282 electric field on its performances. Similar trends (but to a lesser extent) were obtained at 7 kV

283 for both Desal GK and Pleiade membranes, and it was thus chosen to work with this lower

284 applied voltage in order to minimize its impact and ensure that variations are mainly due to

285 polymer deposits. All the following results were thus obtained with an applied voltage of 7 kV.

286 **Table 2:** Intrinsic permeability  $L_p$  and mean pore radius  $r_p$  of the two investigated membranes, before  
 287 and after electrospray of water with a voltage of 12 kV.

Membrane	PLEIADE		DESAL GK	
	Pristine	After water ES	Pristine	After water ES
$L_p (10^{-14} m)$	8.4	3.3	0.6	1.2
$r_p (nm)$	2.5	1.8	1.8	1.8

### 288 3.2. Deposit highlighting

289 Before studying the influence of electrospray conditions, the presence of deposit was checked

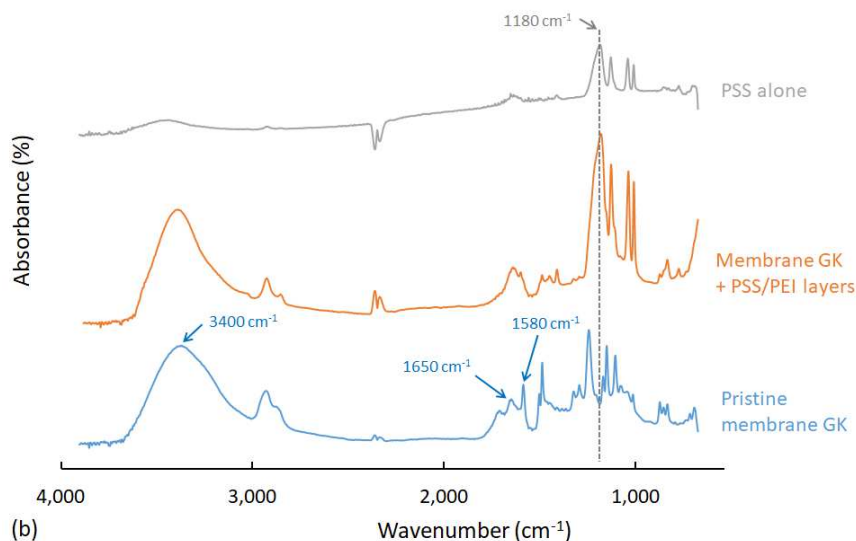
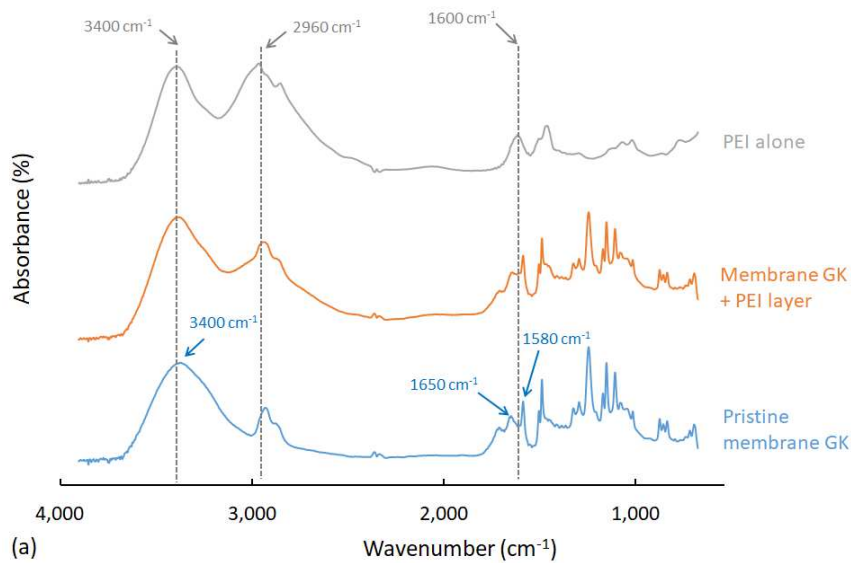
290 by attenuated total reflection – Fourier-transform infrared (ATR-FTIR) spectroscopy, and SEM

291 and AFM images. The FTIR spectra obtained with pristine membrane and membrane modified

292 by PEI and PSS/PEI depositions are given in Fig. 3.

293

294



**Fig. 3:** FTIR spectra of pristine and modified membrane surfaces, as well as polymers alone, for electro spray deposition of a PEI layer (3a) and PSS/PEI layers (3b).

The usual peaks characteristic of polyamide polymer obviously appeared in the spectra of pristine membrane. For instance, the bands at 3400, 1650 and 1580  $\text{cm}^{-1}$  refer to the stretching of N–H, stretching of C=O, bending of N–H bonds of the amide group, respectively. Bands characteristic of amine groups of PEI, namely stretching of N–H, stretching of N–H (amine salt) and bending of N–H appear at wavelength of 3400, 2960 and 1600  $\text{cm}^{-1}$ , respectively. These bands are also observable on the spectra of pristine and modified membranes (in Fig. 3a), which can be explained by the similarity of amide and amine groups. Moreover, it is likely that



306 the pristine membrane contains residues of amine groups from polymerization process, which  
 307 is suggested by the band referring to stretching of N–H (amine salt) at 2960 cm<sup>-1</sup> (Fig. 3a). This  
 308 can explain why specific bands of amine groups from PEI are not discernible on modified  
 309 membrane by electro spray of PEI. Oppositely, the specific band referring to stretching of S=O  
 310 of sulfonate group from PSS (which appears at 1180 cm<sup>-1</sup> in Fig. 3b) can be observed on the  
 311 spectrum of the membrane modified by PSS/PEI, whereas this band does not appear on the  
 312 spectrum of pristine membrane. The presence of this specific band of PSS (and others) clearly  
 313 proves that the PSS was deposited on the PEI layer. Hence, this also shows that PEI was coated  
 314 on membrane surface, even if its presence was not demonstrated from FTIR spectra.  
 315 To conclude, FTIR study clearly demonstrates the effective coating of membrane surface by  
 316 polymers during electro spray.

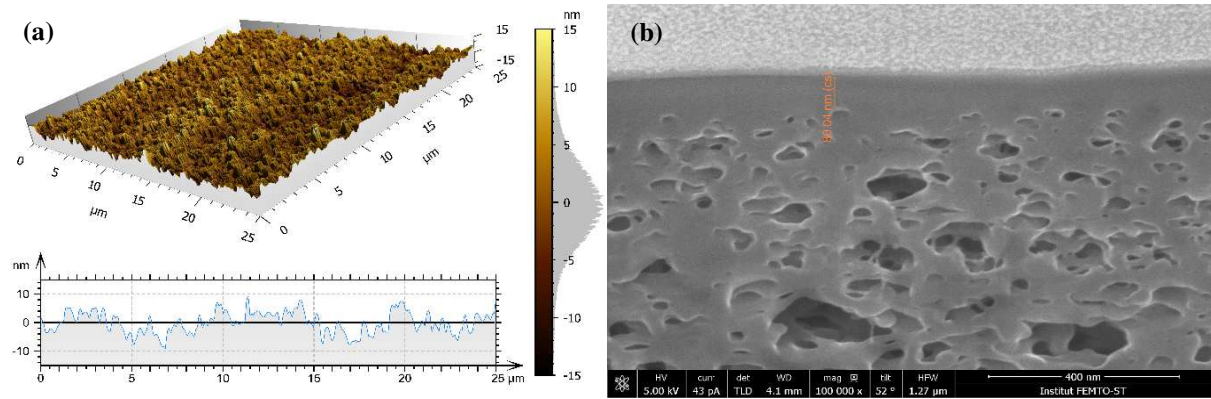
317 Fig. 4 presents the surface morphology obtained by AFM images and membrane cross-section  
 318 observed by FIB-SEM for pristine membrane (a, b) and after PEI (c, d) and PSS/PEI (e, f)  
 319 electro spray depositions. The corresponding values of roughness and thickness of the active  
 320 layer alone or with PEI and PSS/PEI deposits are summarized in table 3.

321  
 322 Table 3. Arithmetical mean height ( $S_a$ ), root mean square height ( $S_q$ ) and layer thickness of the  
 323 pristine membrane and membrane modified by electro spray deposition of PEI and PSS/PEI.

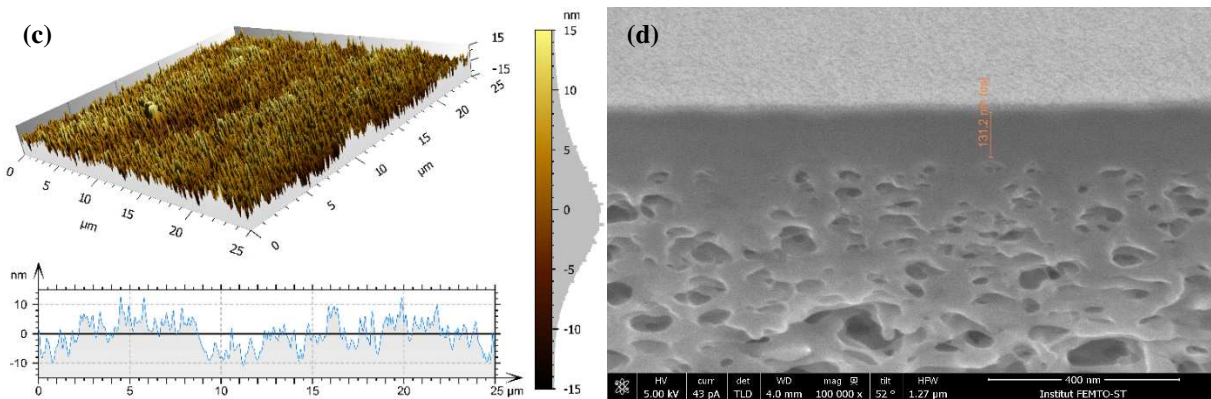
Membranes	$S_a$ (nm)	$S_q$ (nm)	Thickness (nm)
Active layer of pristine membrane	4.0	5.0	80
Active layer + PEI deposition	3.6	4.6	130
Active layer + PSS/PEI deposition	4.4	5.6	470

324  
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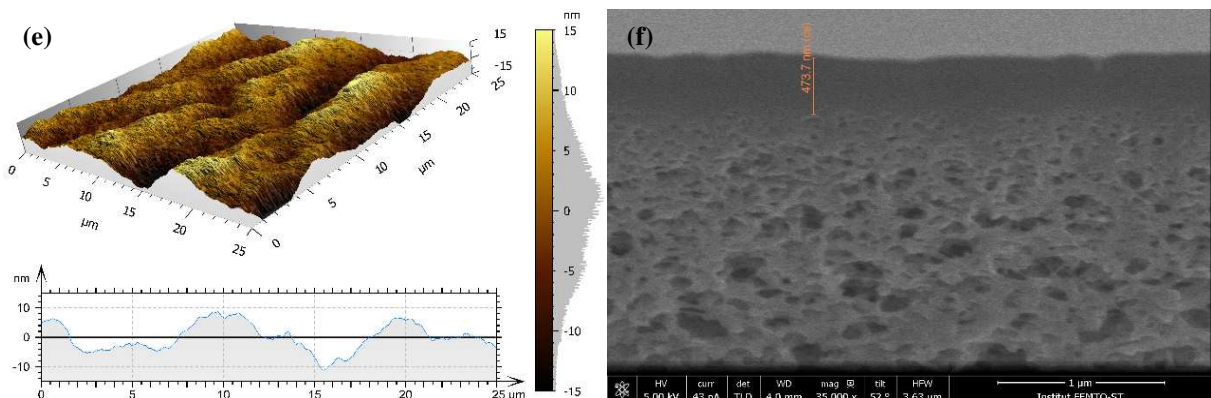
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328



329 **Fig. 4:** AFM images of membrane surface (a, c, e) and membrane cross-section observed by FIB-SEM  
330 (b, d, f) for pristine membrane (a, b), and after electro spray deposition of PEI (c, d) and PSS/PEI (e, f).  
331

332 AFM images provided in Fig. 4 (a, c, e) show that PEI deposition seems to have a weak impact  
333 on roughness and surface morphology. The roughness is identical and only a few tightening of  
334 peaks and valley is observable. Oppositely, the terminating layer of PSS (i.e. PSS/PEI  
335 deposition) was found to strongly modify surface morphology. Indeed, with this bi-layer  
336 deposit, the surface is smoother even if roughness remains similar, and the unevenness is more  
337 spread leading to larger peaks and valleys.

338 SEM images of the membrane cross-section (Figs. 4b, 4d, 4f) show that the thickness of the  
339 thin surface layer (given in Table 3) slightly increases when PEI is sprayed (from 80 to 130  
340 nm). When a PSS terminating layer is deposited on the PEI layer, the thickness is largely  
341 increased up to 470 nm. This confirms that PEI layer have a weak impact on membrane  
342 structure, whereas PSS deposit strongly modify the morphological properties of the overall  
343 membrane.

### 344 **3.3. Influence of spray parameters on membrane physicochemical properties**

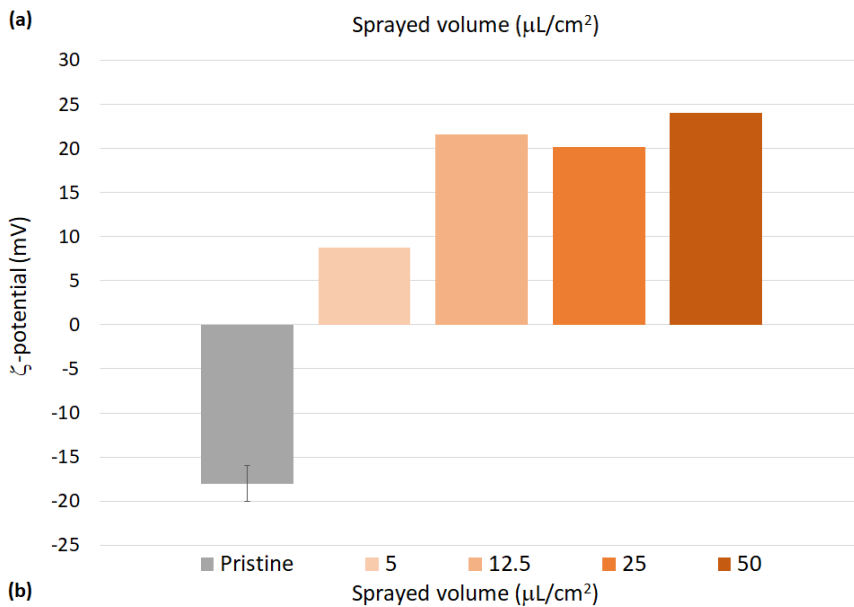
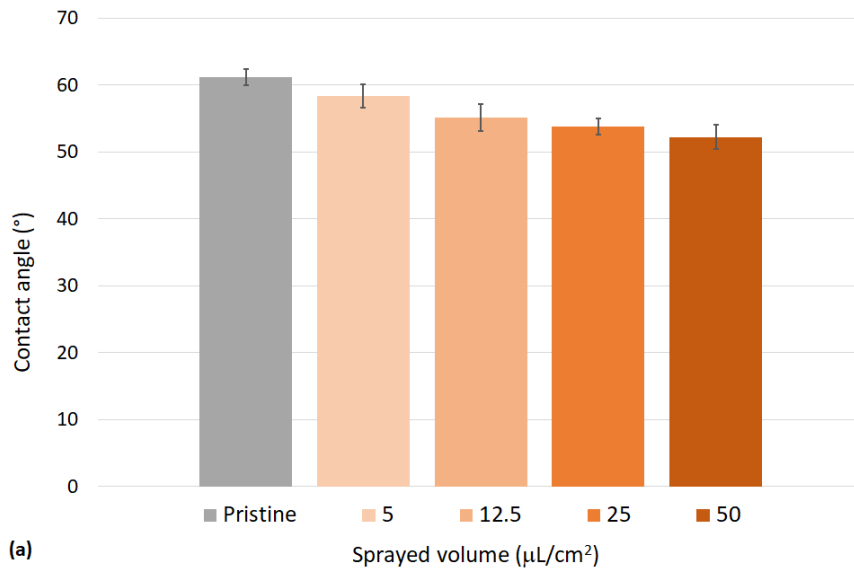
345 It is known that hydrophilicity is an important property of UF membranes, which tends to  
346 increase permeation flux but also reduce its fouling by various hydrophobic pollutants. So, the  
347 impact of the polymer deposition on hydrophilicity was investigated by contact angle  
348 measurements on the pristine and the PEI-modified membranes. Similarly, ion rejection and  
349 separation selectivity are governed by electrostatic interactions between the charge of ions and  
350 that of membrane surface. Hence, the impact of polymer deposition on the value of  $\zeta$ -potential  
351 was also investigated, which was estimated by tangential streaming current measurements.

#### 352 **3.3.1. Influence of sprayed volume / spray duration**

353 The main asset of electrospray deposition lies in the small quantities of polymer solution  
354 required for coating. However, the volume of sprayed solution can have an influence on surface  
355 properties. The sprayed volume of PEI solutions was varied from 2.5 to 50  $\mu\text{L}/\text{cm}^2$  and the  
356 impact on both contact angle and  $\zeta$ -potential was investigated. Each volume was sprayed at a  
357 constant flow-rate of polymer solution ( $3 \times 10^{-1} \text{ mol L}^{-1}$  of PEI amine group), which was fixed  
358 at 1 mL/min. The duration of pulverization was also dependent on the membrane sample area.  
359 Evolution of contact angle and  $\zeta$ -potential for the various sprayed volumes is provided in Fig.  
360 5.

361

362



**Fig. 5:** Evolution of the surface contact angle (5a) and  $\zeta$ -potential (5b) with the volume of sprayed solution containing  $3 \times 10^{-1} \text{ mol L}^{-1}$  of PEI

Fig. 5a shows that the contact angle between the water drop and the surface decreases monotonously when the volume of sprayed polymer solution increases. This means that the deposit of PEI makes the surface more hydrophilic, which can have a positive impact on flux and membrane fouling. The fact that contact angle seems to decrease more slightly for sprayed volumes larger than  $12.5 \mu\text{L}/\text{cm}^2$  tends to show that the membrane is probably partially covered by PEI for low sprayed volumes. It should be noted that this decrease in hydrophobicity is moderated, probably because PEI is slightly more hydrophilic than membrane surface, the hydrophilic properties of which come from exposed polar amine groups.

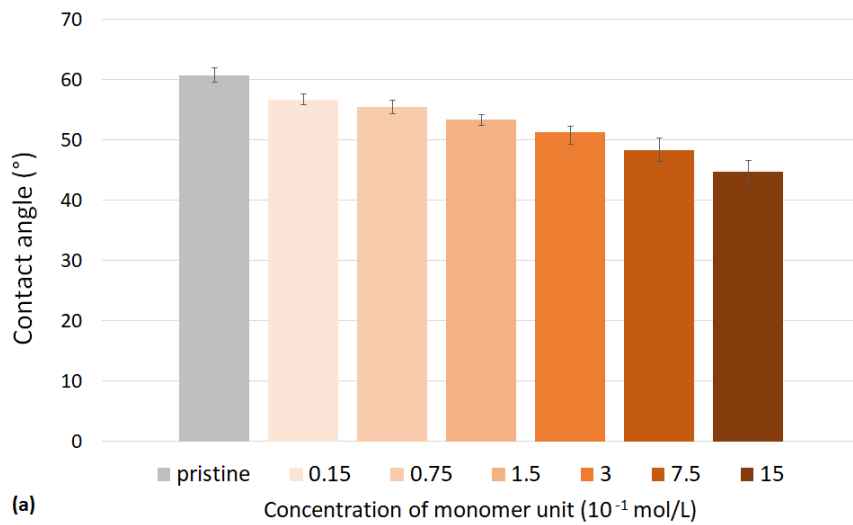
375  $\zeta$ -potential values given in Fig. 5b confirm the presence of a PEI deposit at the membrane  
376 surface, which allows a reversal of the sign of the surface charge, as illustrated in literature [43].  
377 From Fig. 5b, it can be observed that the membrane surface becomes positive after electro spray  
378 deposition of PEI and the positive charge is higher when the volume of sprayed solution is  
379 larger. It should be stressed that for volumes higher than  $12.5 \mu\text{L}/\text{cm}^2$ , the membrane charge no  
380 longer seems to be affected by the sprayed volume since differences in zeta potential from  $12.5$   
381  $\text{to } 50 \mu\text{L}/\text{cm}^2$  are within the confidence interval, which was assessed with several pristine  
382 membranes. This trend suggests that the membrane surface is probably almost covered by  
383 polymer above this volume.

### 384 3.3.2. Influence of polymer concentration

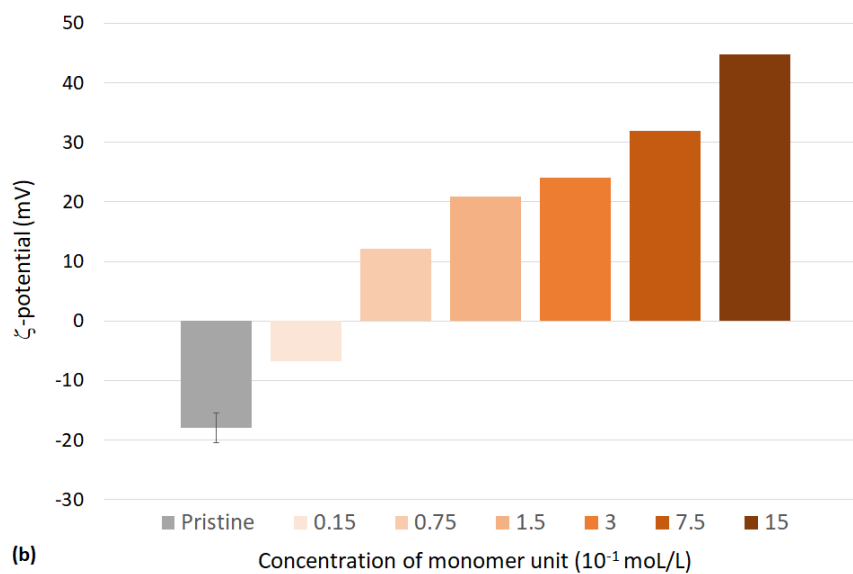
385 The influence of PEI concentration was investigated on both contact angle and  $\zeta$ -potential  
386 values. The evolutions with concentration of monomer unit of PEI (amine groups) are provided  
387 in Fig. 6 for a sprayed volume of  $50 \mu\text{L}/\text{cm}^2$ .

388 Fig 6 shows that polymer concentration also plays a major role on both surface hydrophilicity  
389 and electrical properties. It even seems that the impact of concentration is more striking than  
390 that of sprayed volume. For instance, contact angle is found to monotonously decrease when  
391 concentration of PEI increases up to  $45^\circ$  for a concentration of  $15 \times 10^{-1} \text{ mol/L}$  of PEI monomer  
392 unit. It can be seen in Fig. 6b that a too low amount of polymer does not allow a reversal of the  
393 charge sign ( $C < 0.75 \times 10^{-1} \text{ mol/L}$ ) but tends to decrease the negative charge. For  
394 concentrations higher than  $0.75 \times 10^{-1} \text{ mol/L}$ , the positive charge monotonously increases with  
395 concentration, up to  $45 \text{ mV}$ , without reaching a plateau. It should be noted that for  
396 concentrations higher than  $15 \times 10^{-1} \text{ mol/L}$ , the high viscosity of polymer solution clearly  
397 complicates the pulverization.

398



(a)



(b)

399

400

401 **Fig. 6:** Evolution of the surface contact angle (6a) and  $\zeta$ -potential (6b) with the concentration of PEI  
 402 monomer unit for a sprayed volume of  $50 \mu\text{L}/\text{cm}^2$

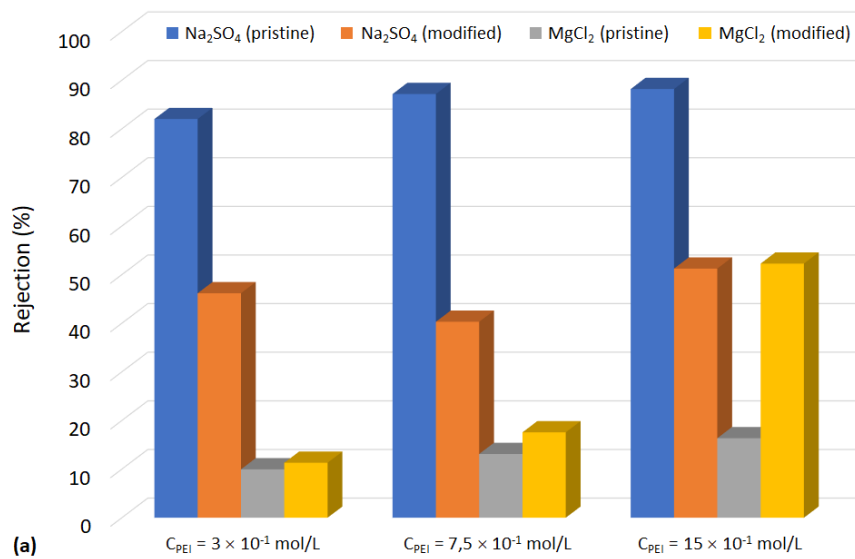
403 These trends are consistent with literature in which it was already highlighted that a higher  
 404 polymer concentration leads to larger polymer deposition rates, which obviously results in more  
 405 polymer adsorption on the membrane surface [44]. These outstanding modifications of  
 406 physicochemical properties with electrospray deposition of PEI should have a notable impact  
 407 on filtration performances.

### 408 3.4. Impact of PEI electrospray deposition on filtration performances

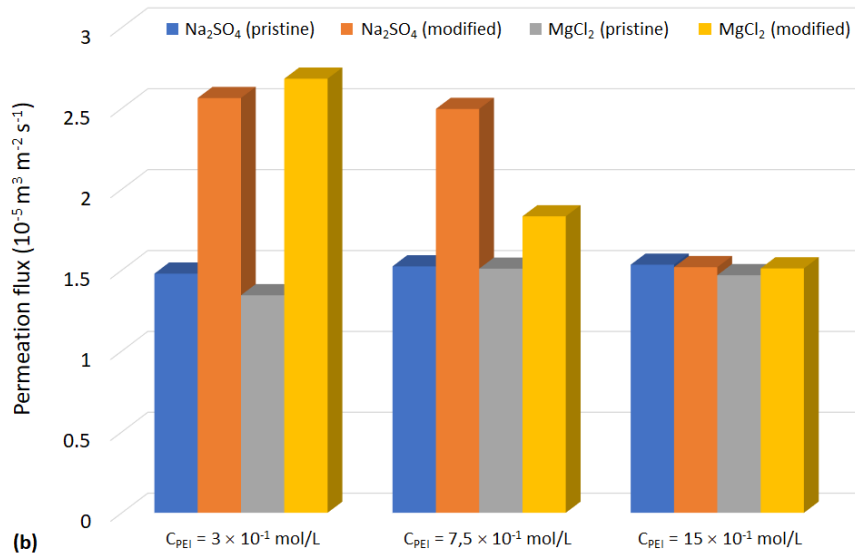
409 The main aim of PEI deposition is to increase electrostatic interactions between divalent ions

410 and membrane surface. Hence, the impact of the PEI concentration on filtration performances  
 411 is firstly investigated on  $\text{MgCl}_2$  and  $\text{Na}_2\text{SO}_4$  rejections. A salt concentration of  $10^{-3}$  mol/L was  
 412 chosen for filtration to keep a low ionic strength and avoid the screening of electrostatic  
 413 interactions between PEI and membrane surface as well as between PSS and PEI. Indeed, at  
 414 high ionic strength, charges of polyelectrolytes become neutralized by salt counter-ions, which  
 415 leads to weaker electrostatic attraction and results in thicker modified layers [45].

416



417



418 **Fig. 7:** Rejection (7a) and permeation flux (7b) at 15 bar of  $\text{MgCl}_2$  and  $\text{Na}_2\text{SO}_4$  measured with both  
 419 pristine and modified membranes for three concentrations of PEI monomer units  
 420  $\{V_{\text{sprayed}} = 12.5 \mu\text{L}/\text{cm}^2\}$   
 421

422 Salt rejection by pristine and modified membranes measured at 15 bar are provided for the three  
423 highest concentrations of PEI monomer unit in Fig. 7. The volume sprayed was chosen at 12.5  
424  $\mu\text{L}/\text{cm}^2$  (beginning of the plateau for  $\zeta$ -potential) to limit the quantity of polymer at the  
425 membrane surface and its potential impact on permeation flux.

426 It can be observed in Fig. 7a that electrospray deposition of PEI has a different impact on the  
427 rejections of divalent cations and anions. Indeed, a sharp decline of sulfate rejection is obtained  
428 irrespective of the PEI concentration in the sprayed solution. Unlike many studies found in  
429 literature [43], magnesium rejection is almost not enhanced, until a high concentration of PEI  
430 is sprayed. This means that the quantity of polyelectrolyte is enough to screen the primitive  
431 negative charge of the membrane but not enough to induce a strong repulsive interaction  
432 between positive PEI coated at the membrane surface and divalent cations. This conclusion is  
433 not in accordance with the values of zeta-potential provided in Fig. 6b since it seems that  
434 membrane surface charge is strongly positive even for lower PEI concentrations. In fact, this  
435 unexpected behavior can probably be explained by the high shear stress condition induced by  
436 cross-flow filtration, and especially with spacer-filled retentate channel, compared with the  
437 smoother laminar conditions occurring during streaming current measurements. It was shown  
438 in literature that a transition to unsteady flow occurs at relatively low Reynolds numbers ( $\text{Re} <$   
439  $50$ ) in spacer-filled channels. Indeed, spacer acts as a turbulence promoter leading to the  
440 development and separation of boundary layers, vortices formation, creation of high shear  
441 regions and recirculation zones [46], which obviously tends to break down deposits. Hence, it  
442 seems that the electrostatic interactions between the membrane and PEI are probably not  
443 sufficient to keep the whole deposit at the membrane surface under these harsh conditions.  
444 Harsh conditions are few investigated in studies reported in literature for which most of  
445 filtrations are implemented in dead-end mode [47, 48] or with smoother cross-flow conditions  
446 [49, 50]. However, for a high concentration such as  $15 \times 10^{-1} \text{ mol/L}$  of monomer unit, the strong

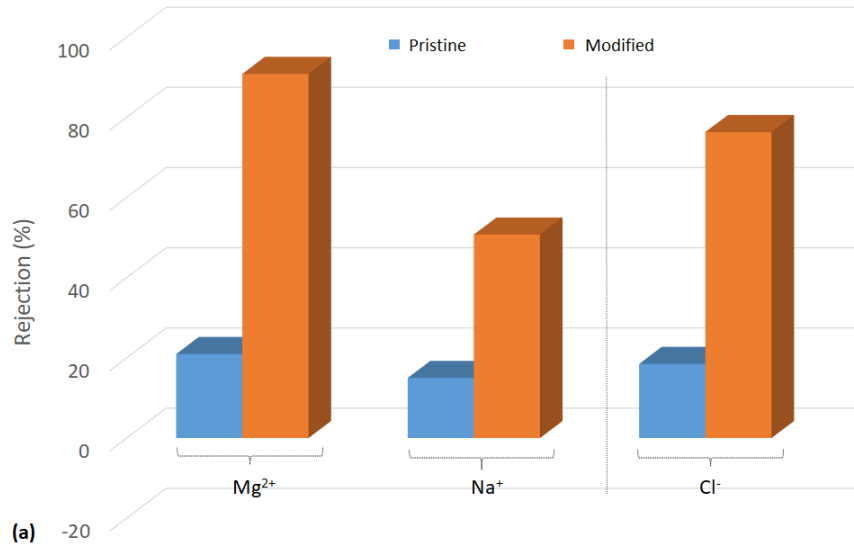


447 increase in solution viscosity probably leads to stronger adhesion of the polymer and a decrease  
448 of Reynolds values, which makes it more relevant for real cross-flow conditions.

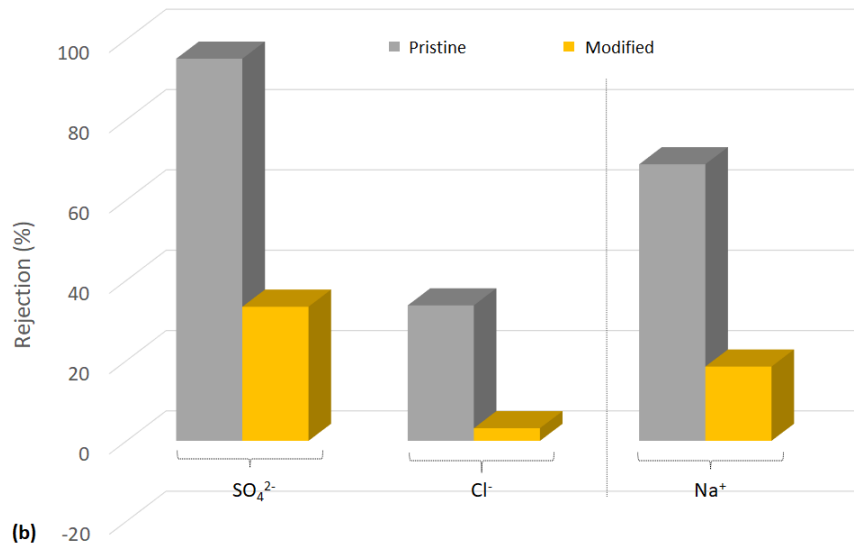
449 A cross-linking procedure leading to covalent bonds would probably have enhanced the  
450 stability of PEI coating, especially with high salt contents for which electrostatic interactions  
451 are screened.

452 Concerning the performances in terms of permeation, it can be concluded from Fig. 7b that the  
453 flux increases after electrospray deposition, as it was previously observed with water spray.  
454 However, this increase was found to be progressively reduced when PEI concentration  
455 increases. This trend can be attributed to the accumulation of polymer at the membrane surface,  
456 which obviously tends to increase overall membrane resistance to permeation. At  $15 \times 10^{-1}$   
457 mol/L, the increase in permeability induced by electrospray is fully balanced by the presence  
458 of the coated polymer layer and the resulting permeation flux is therefore not impacted by the  
459 modification compared with pristine membrane. This groundbreaking result is of prime  
460 importance for a potential use since it was proved that this method allows an improvement of  
461 cation rejection without negative impact on the permeation flux.

462 Enhancement in terms of divalent cation rejection was highlighted in this study. However,  
463 treatment of real solution implies to investigate separation selectivity for ion mixtures that  
464 mimic real-life conditions. To deeply investigate the impact of electrospray deposition on the  
465 filtration performances with ionic mixtures, the rejection curves for ternary and quaternary ion  
466 solutions are provided in Fig. 8 and 9 for the highest PEI concentration ( $15 \times 10^{-1}$  mol/L of  
467 monomer unit).



468

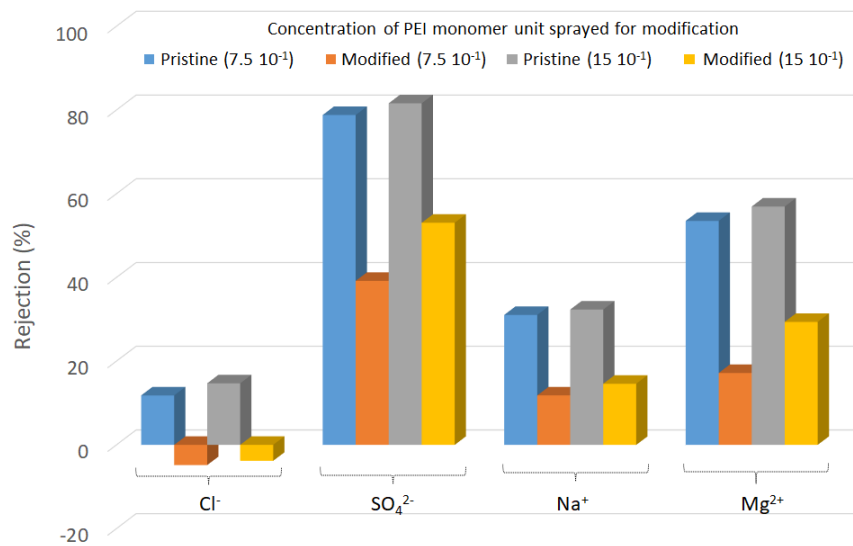


469

470 **Fig. 8:** Ion rejections at 15 bar for ternary mixtures (6a: MgCl<sub>2</sub>-NaCl and 6b: Na<sub>2</sub>SO<sub>4</sub>-NaCl) measured  
 471 with both pristine and modified membranes  
 472 {C<sub>PEI</sub> = 15 × 10<sup>-1</sup> mol/L and V<sub>sprayed</sub> = 12.5 μL/cm<sup>2</sup>}

473 Trends observed with the single salt solutions are confirmed by curves obtained with ternary  
 474 solutions. Indeed, it can be seen in Fig. 8 that the electro spray deposition of PEI clearly  
 475 increases rejection of cations and more intensely that of divalent cations, which leads to a better  
 476 selectivity between mono- and divalent cations. Similarly, rejection of anions and especially  
 477 divalent anions are decreased by the PEI deposition, which leads to a strong decrease of the  
 478 selectivity between mono- and divalent anions. It should be stressed that rejections of the  
 479 common ion Cl<sup>-</sup> for MgCl<sub>2</sub>-NaCl mixture and Na<sup>+</sup> for Na<sub>2</sub>SO<sub>4</sub>-NaCl mixture are not relevant

480 for discussion since their variation is imposed by the two other ions and their value is always  
 481 intermediate between mono- and divalent non-common ions, due to electroneutrality condition.  
 482 These trends are highly valuable for the potential application of heavy metal removal.  
 483 Electrospray deposition can be implemented to enhance the removal of multivalent metal ions  
 484 (e.g.  $\text{Cu}^{2+}$ ,  $\text{Co}^{2+}$ ,  $\text{Ni}^{2+}$ , etc.) without retaining monovalent ions. Rejection of ions from  
 485 quaternary mixtures (shown in Fig. 9) is also interesting but the impact of modification is less  
 486 notable, and rejection must be compared at two concentrations to understand how modification  
 487 govern selectivity of complex ion mixtures. Fig 9 provides the rejection of the four ions with  
 488 the pristine membrane and the same membrane modified by spraying  $7.5$  and  $15 \times 10^{-1}$  mol/L  
 489 of PEI monomer unit.



490

491 **Fig. 9:** Ion rejections measured at 15 bar before and after electrospray deposition of PEI at two  
 492 concentrations of monomer unit  $\{V_{\text{sprayed}} = 12.5 \mu\text{L}/\text{cm}^2\}$

493 From Fig. 9, it can be concluded that rejection of both anions and cations, and especially  $\text{Mg}^{2+}$   
 494 and  $\text{SO}_4^{2-}$ , decreases due to PEI deposition, which was unexpected considering results observed  
 495 with ternary mixtures. However, it appears that  $\text{SO}_4^{2-}$  rejection strongly decreases due to the  
 496 presence of polymer for  $7.5 \times 10^{-1}$  mol/L of PEI. This fall results in a strong decline of  $\text{Mg}^{2+}$   
 497 rejection due to electroneutrality condition,  $\text{Na}^+$  and  $\text{Cl}^-$  being weakly impacted by

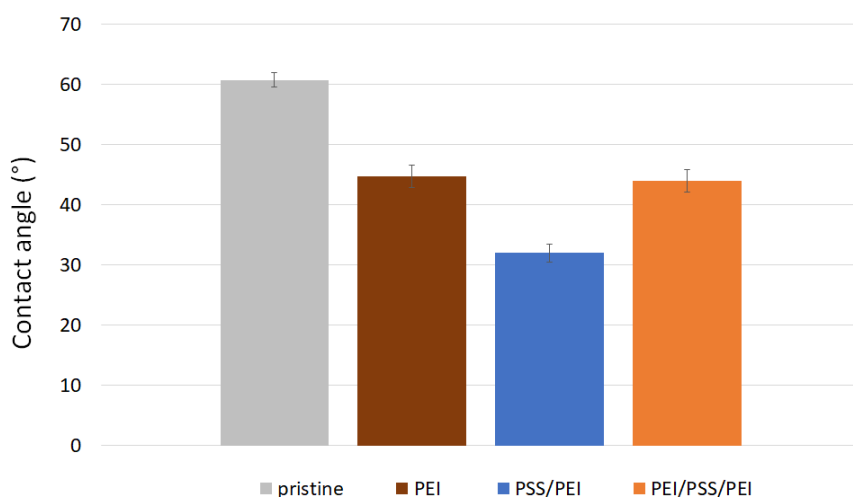
498 modification. In the case of the solution containing  $15 \times 10^{-1}$  mol/L of PEI monomer unit, it can  
499 be seen that the changes in electrostatic interactions caused by PEI deposit tends to reduce the  
500 decline of  $\text{Mg}^{2+}$  rejection (due to repulsion with positive PEI), thereby reducing the decline of  
501  $\text{SO}_4^{2-}$  rejection to comply with electroneutrality condition. **In ion mixtures, the transport of the**  
502 **monovalent anion is favored to balance the high rejection of divalent anion and the low rejection**  
503 **of cations. Sulfate ion being highly rejected, the transfer of cations forces the transfer of**  
504 **chloride ion to maintain electroneutrality, which can lead to negative rejections.** It can be noted  
505 that the very low monovalent rejections and especially the negative rejection of chloride  
506 obtained after PEI electrospray are particularly attractive for applications that aim at enriching  
507 oligo elements (multivalent ions) by desalting (NaCl removal) the retentate stream.

### 508 **3.5. Impact of layer-by-layer electrospray deposition**

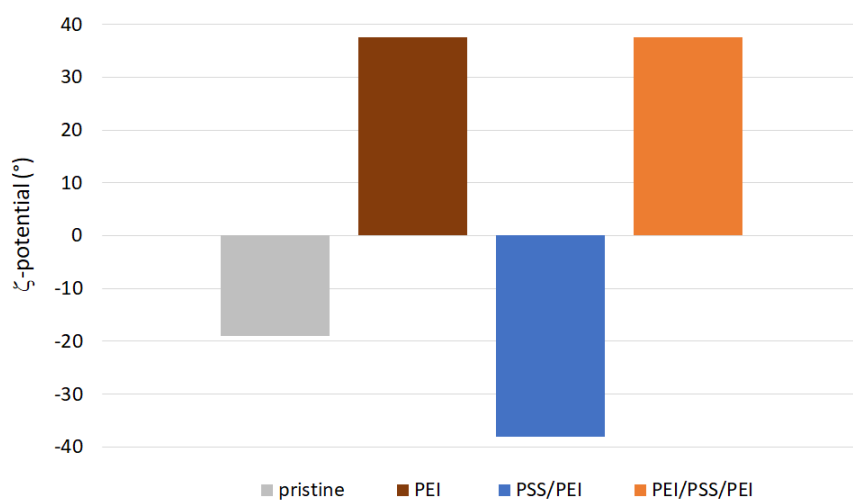
509 Various layers were deposited by electrospray of polymer solution containing  $15 \times 10^{-1}$  mol/L  
510 of monomer unit by alternating (positive and negative) polymers, *i.e.* PEI, PSS/PEI,  
511 PEI/PSS/PEI. Indeed, layer-by-layer assembly is mainly driven by electrostatic interactions and  
512 a negatively charged PSS layer cannot be coated on negative membrane surface. It should be  
513 stressed that even if electrostatic interactions are the main forces governing layer-by-layer  
514 assembly, it is not the sole mechanism and additional forces such as hydrophobic interactions  
515 and entropy gain have an impact on deposition by restructuring and release of water molecules  
516 and counter-ions [29].

#### 517 ***3.5.1. Impact on surface properties***

518 The impact of layer-by-layer electrospray on contact angle and zeta-potential are provided in  
519 Fig. 10.



(a)



(b)

520

521

522 **Fig. 10:** Evolution of the surface contact angle (10a) and ζ-potential (10b) with the various deposited  
 523 layers { $C_{PEI} = 15 \times 10^{-1}$  mol/L and  $V_{sprayed} = 50 \mu\text{L}/\text{cm}^2$ }

524 Fig. 10a shows that all deposits make the surface more hydrophilic than that of pristine  
 525 membrane. It can also be seen that the PSS layer seems clearly more hydrophilic than the PEI  
 526 layer. Hydrophilicity enhancement by PEI deposition can mainly be attributed to exposed polar  
 527 amine groups at the membrane surface, whereas that obtained by PSS is induced by hydrophilic  
 528 sulfonate groups. Contact angle of modified membrane surface is similar for terminating layers  
 529 made of PEI when PEI is deposited either on PSS (*i.e.* PEI/PSS/PEI layers) or on membrane  
 530 surface (PEI layer only). This means that the sublayers have a negligible impact on the

531 hydrophilicity of the modified membrane surface, confirming that the surface is almost fully  
532 covered by the various layers.

533 Values of  $\zeta$ -potentials provided in Fig. 10b clearly show that it is possible to obtain positive or  
534 negative surface charge by alternatively spraying positive (PEI) and negative (PSS) polymers,  
535 as it was previously highlighted in literature with other polymers [51]. Spraying PSS on a PEI  
536 layer even leads to a more negative surface charge than that of the pristine membrane. Finally,  
537 the deposition of a second PEI layer on the PSS/PEI layers leads to the same membrane charge  
538 than that achieved with only one PEI layer.

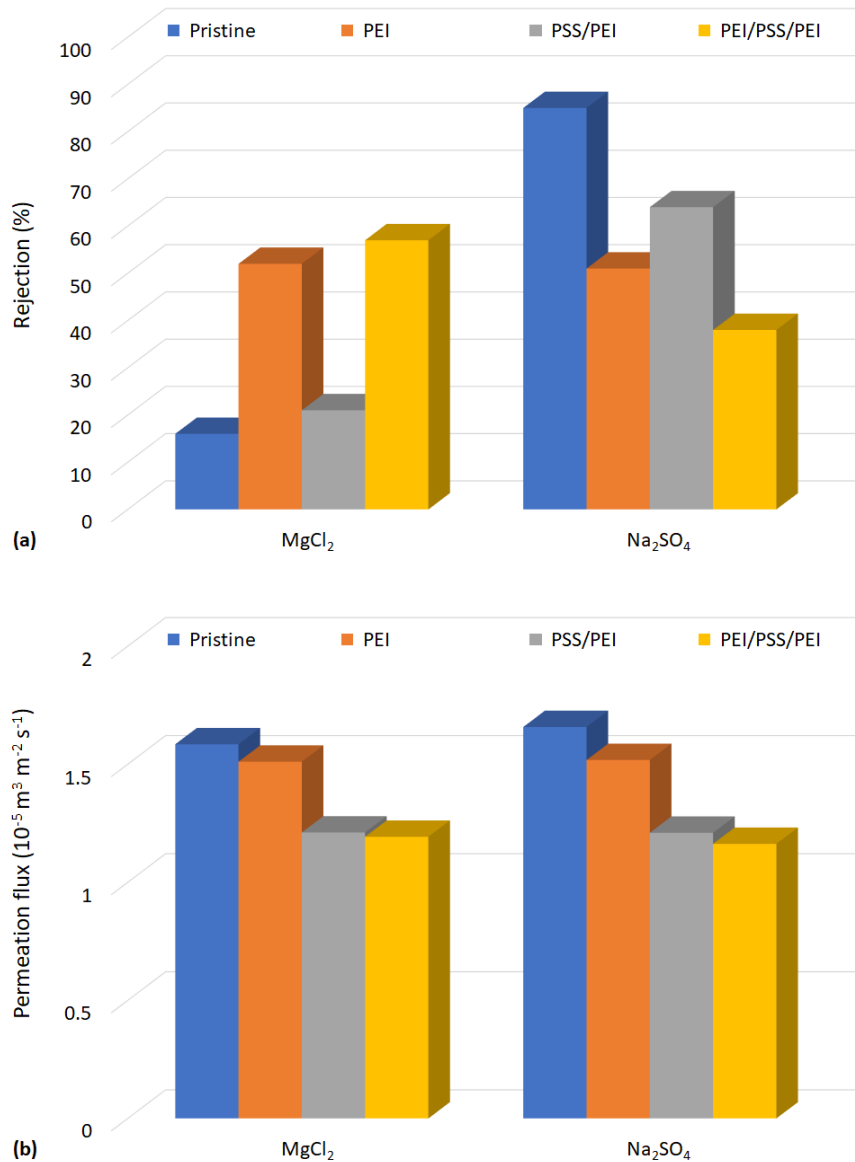
539 Moreover, superimposing alternatively more than 1 layer of PEI and PSS does not seem to  
540 present a substantial interest in terms of physicochemical properties, even if some changes in  
541 growth mechanisms and selectivity performances with the number of layers have been already  
542 highlighted in literature [52].

### 543 ***3.5.2. Impact on filtration performances***

544 Rejections and permeation flux obtained with pristine membrane and membranes modified by  
545 alternating PEI and PSS layers are provided in Fig. 11.

546 The filtration performances measured with various layers of PEI and PSS show that the external  
547 (terminating) layer mainly governs the rejection of salts. Indeed, the rejection of  $\text{MgCl}_2$  is  
548 highly increased when PEI is the external layer (*i.e.* PEI or PEI/PSS/PEI) and in the same order  
549 of magnitude. Similarly, the rejection of  $\text{Na}_2\text{SO}_4$  is strongly decreased in the latter case. These  
550 trends perfectly meet expectations and results provided in literature [53], and can be explained  
551 by electrostatic interactions. Oppositely, the rejection of  $\text{MgCl}_2$  is slightly increased and that of  
552  $\text{Na}_2\text{SO}_4$  notably decreased when external layer is made of PSS. These impacts are surprising  
553 since it is expected that the higher negative surface charge of the membrane modified by  
554 PSS/PEI compared with pristine membrane (shown in Fig. 10b) should lead to a decrease in  
555  $\text{MgCl}_2$  rejection and an increase in  $\text{Na}_2\text{SO}_4$  rejection. This could be explained by the fact that

556 rejection is probably not merely governed by the superficial layer but also partially by the  
 557 deeper deposited sublayers and membrane body. This impact could perhaps also be attributed  
 558 to the fact that the membrane surface is not fully covered when  $12.5 \mu\text{L}/\text{cm}^2$  of polymer solution  
 559 is sprayed, some amino groups being available for electrostatic interactions with divalent ions.



560

561

562 **Fig. 11:** Rejection (11a) and permeation flux (11b) at 15 bar of MgCl<sub>2</sub> and Na<sub>2</sub>SO<sub>4</sub> measured with  
 563 pristine and membranes modified by layer-by-layer electro spray deposition of PEI and PSS (PEI,  
 564 PSS/PEI, PEI/PSS/PEI) { $C_{\text{polymer}} = 15 \times 10^{-1} \text{ mol/L}$  and  $V_{\text{sprayed}} = 12.5 \mu\text{L}/\text{cm}^2$ }  
 565

566 Concerning flux, it can be concluded that permeation is slightly more hindered when several  
 567 layers are deposited, which has already been reported in literature [54]. However, it seems that

568 the decline involved by PSS deposition seems to affect more intensively permeation flux than  
569 PEI deposition. Although surface hydrophilicity increases with PSS deposit, the decline in  
570 permeation flux is in agreement with the notable increase of surface layer thickness observed  
571 after electrospray deposition of PSS (Fig. 4f) and the corresponding increase in hydraulic  
572 resistance.

573 It should be mentioned that the bulk permeability of the overall polymer film can show distinct  
574 changes depending on the terminating layer, which is known as odd-even effects [45] and could  
575 perhaps have an influence on observed permeability. However, a deeper investigation of this  
576 phenomenon should have required to study a larger number of polymer layers. Finally, it should  
577 be mentioned that ES depositions were implemented in the presence of KCl  $10^{-3}$  mol/L to  
578 comply with the same conditions as those required for electrokinetic characterization of the  
579 surface. Nevertheless, it is well-known that background ionic strength strongly governs  
580 polyelectrolyte layer structure and thus selectivity and permeability [55]. This specific point  
581 appears to be an attractive prospect to extend this study.

582



## 583 **IV. Conclusion**

584 In this study, it was shown that electrospray deposition of polyelectrolytes is a suitable  
585 technique to tailor physicochemical properties of nanoporous membranes, and thus ion  
586 separation selectivity. This study highlighted that the deposition does not induce flux loss, and  
587 can even lead to a slight enhancement in specific conditions. It was also proved that this  
588 technique allows a notable adjustment of ion rejection, provided that the quantity of polymer is  
589 sufficient to induce a high solution viscosity, helping the deposit to withstand the shear stress  
590 inherent in cross-flow filtration. In particular, the charge of membrane surface can be strongly  
591 modified to either highly positive values of  $\zeta$ -potential by spraying PEI or more negative values  
592 by spraying PSS (from -40 to +40 mV). This variation in membrane charge can lead to strong  
593 modification of separation selectivity induced by changes in electrostatic interactions between  
594 ions and membrane surface. It was especially shown that a positively charged superficial layer  
595 (PEI) induced an increase in rejection of divalent cation and a decrease in that of divalent  
596 anions. Superimposing many layers seems not to bring a substantial improvement compared  
597 with only one or two layers. Finally, the fact that surface hydrophilicity is notably increased by  
598 ES deposition of polymer, and especially PSS (from 60 up to 30°), bodes well for a real  
599 application since that tends to notably mitigate the membrane fouling by organic matter.

## 600 **Acknowledgements**

601 The authors would like to thank the region of Bourgogne Franche-Comté (Grant number:  
602 2016Y-04563) for its financial support and especially for financing the Ph.D. thesis of Elizaveta  
603 Korzhova. The Institut de Science des Matériaux de Mulhouse (IS2M, UMR CNRS 7361), and  
604 especially Simon Gree and Aissam Airoudj, are acknowledged for performing IR spectroscopy.  
605 **MIMENTO platform (FEMTO-ST, UMR CNRS 6174) and especially Roland Salut are also**  
606 **acknowledged for FIB-SEM images. Authors also thank Sandrine Monney (UTINAM**  
607 **platform) for performing AFM images and roughness analysis.**

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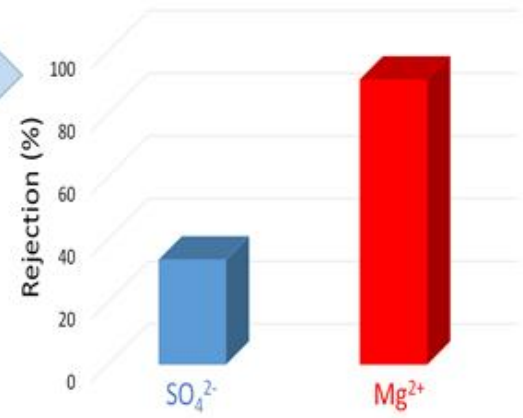
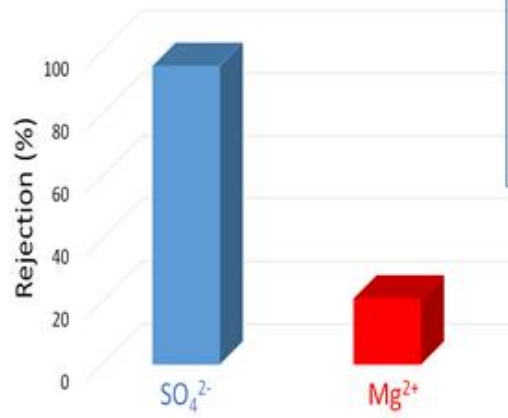
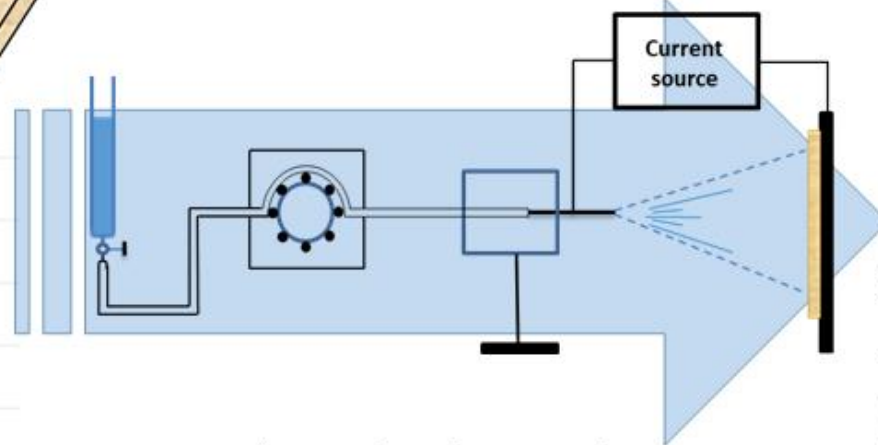
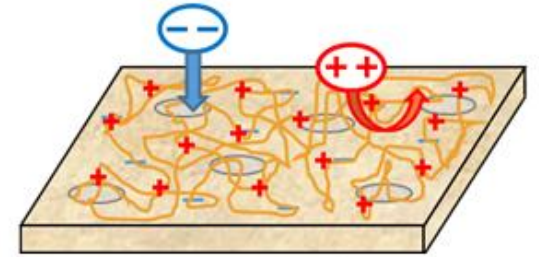
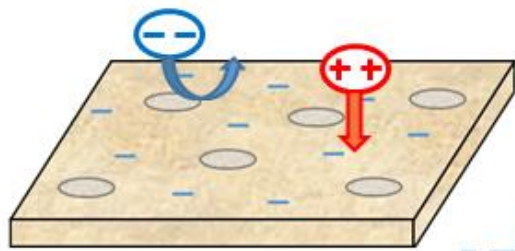
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# Electrospray deposition



of polyelectrolytes