

Gas storage

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Chapter 9: Gas Storage

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ABSTRACT

The continuous increase of energy demands based on fossil fuels in the last years have lead to an increase of greenhouse gases (GHG) emission which strongly contribute to global warming. The main strategies to limit this phenomenon are related to the efficient capture of these gases and to the development of renewable energies sources with limited environmental impact. Particularly, carbon dioxide (CO₂) and methane (CH₄) are the main constituents of greenhouse gases while hydrogen (H₂) is considered an alternative clean energy source to fossil fuels. Therefore, tremendous research to store these gases has been reported by several approaches and among them the physisorption on activated carbons (AC) have received significant attention. Their abundance, low cost and tunable porous structure and chemical functionalities with an existing wide range of precursors that includes bio-wastes make them ideal candidates for gas applications. This chapter presents the recent developments on CH₄, CO₂ and H₂ storage by activated carbons with focus on biomass as precursor materials. An analysis of the main carbon properties affecting the AC's adsorption capacity (i.e. specific surface area, pore size and surface chemistry) is discussed in detail herein.

Key words: Gas storage; activated carbons; biomass; methane; carbon dioxide; hydrogen;

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Introduction

Gases play an important role in our daily life. They are usually found in the form of gas mixtures with the one most known being atmospheric air which is constitute of approximate volumes of 78.08% nitrogen (N₂), 20.95% oxygen (O₂), 0.93% argon (Ar), 0.03% carbon dioxide (CO₂) and traces of methane CH₄, helium (He), neon (Ne), krypton (Kr), xenon (Xe) and ozone (O₃)(Encyclopedia Britannica, 2018). Amongst them, CO₂, CH₄and nitrogen oxides (NO_x) are a product of incomplete burning of fossil fuels (coal, oil, diesel, natural gas) and are the major contributors of global warming (GW) as they are emitted in higher amounts than other greenhouse gases (GHG) such as hydrofluorocarbons (HFCs), perfluorocarbons (PFCs) and hexafluorocarbon (F₆C).As it can be seen in Figure 1, the atmospheric concentrations of these three gases have dramatically increases since the pre-industrial years (≈1850)thus the reduction of their emissions has become a main target arising a need for new technologies for the selective adsorption and storage of these pollutants (Core Writing Team et al., 2014).

Carbon dioxide is a major compound in the gaseous effluent of many industrial processes, the fossil fuel power plants being the most important source (40% of the total CO₂). In order to reduce its emission, the main issues include increasing the power plants efficiency, ,the change towards nuclear and renewable energies and the implementation of carbon capture and storage (CCS) systems (Alonso et al., 2017). CCS can be used as a promising strategy to reduce not only the power plants emissions but it can also be employed in many industrial sectors such as the petrochemical industry, iron and steel refineries and cement production.

There are many separation technologies available for the separation and storage of CO₂: amine scrubbing, water scrubbing, membranes separation, cryogenic separation, biological and physical adsorption. Most of the commercial CCS systems use amine or water scrubbing but they have important drawbacks such as production of undesirable by-products, high energy regeneration, high corrosion and low cost efficiency. In this context, physical adsorption has the advantages of having low

energy requirements, cost advantage, no use of water or additional chemicals and being easy to scale to a wide range of temperatures and pressures (Ben-Mansour et al., 2016).

For the case of methane, the principal anthropogenic sources are thought to be the production of livestock, natural gas production and distribution, emissions from landfills and the coal mining (Alonso et al., 2017). Due to methane's low volumetric energy density, its storage in order to use as a clean energy source involves the use of high pressures to obtain compressed natural gas (CNG) and low temperatures, down to 120 K, for liquefied natural gas (LNG). These methods have a very high energy output as well as various security issues that limit their use. The storage of methane by adsorption in a solid material, known as adsorbed natural gas (ANG) presents advantages over LNG and CNG in terms of energy efficiency and safety. An ANG storage system would have the ability of storing great quantities of methane in small volumes under atmospheric conditions. To compete with other fuels (and other methane storage forms), the department of energy (DOE) of the U.S. has established a CH₄ storage capacity of 180 v/v (volume of gas adsorbed at standard temperature and pressure: 298 K, 0.1 MPa per volume of the storage vessel)(Choi et al., 2016; Ma et al., 2008).

In order to mitigate climate change a number of strategies are being implemented in addition to CO₂ emission reduction and the use of methane as a clean energy source. Hydrogen H₂ has a heating value three times higher than oil and its use as an energy results is completely free of pollution making it one of the most promising alternatives to the burning of fossil fuels. Unfortunately, similar to methane, the transportation and storage of hydrogen have proven to be a very challenging task. In order to be liquefied, it needs to be compressed below 33.2 K (H₂critical temperature)in a high energy demand process, and even when the compression process can be carried out the storage capacity of hydrogen is as low as 2 wt.% even at high pressures (20 MPa). A H₂ storage target for use on vehicles of 5.5 wt.%/40 g L⁻¹ by 2017 and 7.5 wt.%/70 g L⁻¹ ultimate was established by the DOE(Office of Energy Efficiency & Renewable Energy, n.d.). Aiming at this target the storage of hydrogen using porous materials is one of the most promising techniques (Liu et al., 2016).

The development of porous adsorbent materials compatible with the adsorption of gases such as CH₄, CO₂ and H₂has become then a target issue for researchers, with the goals of finding cheap and *eco-friendly* precursor materials and implementing the synthesis procedures that will produce better yielding and the required properties of the material. There are a variety of porous materials used for the adsorption of gases such as zeolites (Cavenati et al., 2004; Chen and Ahn, 2014; Kacem et al., 2015), carbon nanotubes (Gui et al., 2013; Lee et al., 2006), metal organic frameworks (MOFs) (Casco et al., 2015a; Llewellyn et al., 2008; Zhou et al., 2007) and activated carbons(Buss, 1995; Byamba-Ochir et al., 2017; Munusamy et al., 2015). Among these materials, activated carbons are the most widely used in the industry, due to their ease of synthesis as well as the possibility of their pore tailoring and chemical functionalization. In addition, activated carbons can be produced from biomass as the carbon source with the advantages of low cost, high availability, valorization of agricultural wastes and development of high specific surface area adsorbent (Feroldi et al., 2018).

In this chapter, the use of activated carbons for the adsorption of key gases with a focus on biomassderived activated carbons is discussed. A particular attention is paid on the effect of the textural (like surface area and pore volume) and chemical properties (surface groups) upon the activated carbons adsorption behavior.

9.1 Methane Storage

Methane (CH₄) accounts for about 10% of the total greenhouse gases emissions. Over 60% of total CH₄ is generated by human activities such as energy production and, organic waste and the raising of livestock while the other 40% is emitted by natural sources like natural wetlands (EEA, 2014; EPA, 2016). As part of the global efforts to mitigate its environmental impact, methane is considered as a clean energy in the forms of natural gas and biomethane. The global increase of the main greenhouse gases from 1750 to 2011is shown in Figure 1.



Fig. 1 Observed changes in atmospheric greenhouse gas concentrations between the years 1750 to 2011. Atmospheric concentrations of carbon dioxide (CO₂, green), methane (CH₄, orange) and nitrous oxide (N₂O, red). Data from ice cores (symbols) and direct atmospheric measurements (lines) are overlaid. Reprinted from (Core Writing Team et al., 2014).

Natural gas is considered an alternative fuel due to its environmental benefits; its combustion is relatively cleaner than that of other fossil fuels. Natural gas use results in the production and emission of less pollutants, 29% less carbon dioxide per joule delivered than oil and 44% less than coal (NaturalGas.org, n.d.). As a vehicle fuel, methane emits 15 to 20 percent less greenhouse gases (GHG) than gasoline. The main component of natural gas is methane (approximately 90% and higher in most cases, it also contains a small portion of heavier hydrocarbons such as ethane (C_2H_6), propane (C_3H_8) and butane (C_4H_{10}). On the other hand, biogas is the product of the anaerobic fermentation of organic matter which main constitutes are CH₄ (50-75%) and CO₂ (25-45%) (see table 1). Biogas can be directly

burned to generate electricity in a combined heat and power plant (CHP) or alternatively it can be purified to produce biomethane (96% CH₄) (Feroldi et al., 2018).

Compound	Natural Gas	Biogas
Methane (CH ₄)	88.1 %	50-75 %
Carbon dioxide (CO ₂)	0.8 %	25-45 %
Ethane (C_2H_6)	4.2 %	*
Nitrogen (N ₂)	5.2 %	0-3%
Hydrogen (H ₂)	*	0-2%
Hydrogen sulfide (H ₂ S)	*	0-1%

Table 1. Composition of natural gas and biogas in volume percentage (Feroldi et al., 2018).

* None detected

The utilization of biomethane as an alternative energy source also presents a variety of environmental benefits including decrease of greenhouse gas emissions, cheap organic waste recycling and the formation of nitrogen rich products that can be use as fertilizers. In addition, the similar properties of biomethane and methane make biomethane an alternative to replace or to be added to natural gas for vehicular engines. However, despite the benefits of using methane-rich fuels, their use is limited due to transportation and storage constraints. At normal temperature and pressure methane is in the gaseous state which results on a volumetric energy density of only 0.12% of that of gasoline. Table 2 shows the physical properties of methane.

Table 2. Physical Properties of methane (National Center for Biotechnology Information, n.d.).

Molar mass (g mol ⁻¹)	Boiling point (K)	Critical	Critical pressure	Critical	density
		temperature (K)	(MPa)	(g cm ⁻³)	
16.043	111.65	190.90	4.64	0.16	

Currently, CH₄ storage methods are performed by means of compressed natural gas (CNG) or liquefied natural gas (LNG). These methods involve high cost processes (cryogenics and compression), special storage tanks. In addition, these methods present risks to safety due to the use of high pressure and/or the refueling of cryogenic liquids that complicate their use in the vehicle sector. A promising alternative that will reduce the costs and safety risks, consists in the storage of natural gas (or biomethane) at low pressure and normal temperature in the adsorbed form (ANG – adsorbed natural gas).

According to the US Department of Energy (DOE), in order to commercialize the ANG technology a target of CH₄ storage capacity of 180 v/v at 35 MPa (volume of gas at standard pressure and temperature conditions per volume of the storage vessel) was established (Burchell and Rogers, 2000). In this context, the search of efficient adsorbent materials that can meet the DOE standards has become a major challenge.

The established value of 180 v/v corresponds to the energy density of CNG at a pressure of 16.3 MPa (Table 3). This value is considered as a reference target when searching for new adsorbents. However, the U.S. department of Energy under the program "Methane Opportunities for Vehicular Energy (MOVE)" demands an energy density for the ANG systems equivalent to CNG under 25 MPa (266 v/v) (Energy, U.S. Department of Energy: Washington, 2012).

Table 3. Energy Densities of Methane and Conventional Fluids (Kumar et al., 2017a).

Pressure (MPa)

Volumetric storage Energy density (MJ/L)

CNG (15 MPa)	168	5.80
CNG (20 MPa)	222	7.68
CNG (25 MPa)	266	9.2
LNG (110 K and 0.1 MPa)	600	22.2
Gasoline	*	34.2
Diesel	**	37.3

capacity (v/v) at STP

1 L of CH₄ = 0.0345 MJ.

*Required density of methane in a CNG tank to make it equivalent to gasoline

**Required density of methane in a CNG tank to make it equivalent to diesel: 1.0785 g/cm3

It is of general consensus that an efficient adsorbent for the storage of methane must have a high specific surface area (from 1000 m²/g up to 3000 m²/g), with a pore size distribution that ranges around 1.0 to 2.0 nm with a micropore volume of at least 85% of total pore volume (Biloe et al., 2001). Different class of porous materials has been studied aiming to reach the DOE target including activated carbons, ordered carbonaceous materials, zeolites and MOFs.

Among the available adsorbent materials, activated carbons present advantages that make them suitable materials for gas storage such as high surface area, large pore volume and light weight. In particular, for the methane storage process they have the benefits of not being sensitive to humid conditions, they tend to have reasonable prices as well as good adsorption properties at atmospheric pressure.

The adsorption capacity of activated carbons (ACs) is highly related to the textural properties. They present a wide range of pore sizes that goes from the micropore to macropore region and a surface area between 400 and 3000 m²g⁻¹. Furthermore, their textural properties can be controlled by various activation factors.

The CH₄ molecule is symmetrical and therefore it does not present a dipole or quadripole. It is difficult then, to increase the storage capacity of the adsorbent by surface modification. However, the gas storage capacity of ACs can be enhanced by controlling the pore structures. For the methane adsorption, it has been found that the presence of high microporosity with no macropore volume is optimal since the size of the methane molecules are equivalent to the size of micropore. An average pore volume of at least 0.8 nm (diameter greater than two molecules of methane) is wished (Antoniou et al., 2014; Cracknell et al., 1993). In fact, since the interaction of the methane molecules with activated carbons is governed by van der Waals interactions, the attractive force of the pore walls of the adsorbent are a function of the distance *z* between the center of the methane molecule and the center of a carbon atom on the pore wall, this interaction can be described by the Lennar-Jones



potential (Fig. 2).

Fig.2 Potential energy, u(z), for a spherical methane Lennard-Jones (LJ) site interacting with walls of a slit pore of width H=1.2 nm in terms of the distance z (distance between the center of the methane

molecule and the center of a wall carbon atom).Reprinted with permission from(Kumar et al., 2017a). Copyright 2017 American Chemical Society.

The attraction force becomes stronger when *z* is equivalent to one methane molecular diameter (0.36 nm), at longer distances the attraction energy becomes too week and thus the formation of a second adsorbed layer is impeded, the rest of the methane molecules will filled the empty pore space with a low gas-like density.

A linear relationship between the micropore volume (up to 1.6 cm³ g⁻¹) and the methane adsorption capacity of different activated carbons at 35 bars and 298 K Fig. 3) could be established. However, a large portion of pores that have an optimal size of around 0.8 nm plays a major role in the adsorption of methane (Lozano-Castelló et al., 2002).



Fig.3 Relationship of the methane adsorption capacity at 35 bar and 298 K with the micropore volume for a series of activated carbons calculated by N_2 adsorption isotherms. Reprinted from (Lozano-Castelló et al., 2002) with permission from Elsevier.

Furthermore, it has been reported that the presence of pores larger than 0.8 nm is needed in order to facilitate the access and adsorption of the gas molecules. For example, García-Blanco et al. showed that the Maxorb activated carbon, with a surface area of 3090 m²g⁻¹ and a pore size distribution from

pores between 0.5 and 3 nm showed a micropore filling of only 59% when adsorbing methane at 298 K and 3500 kPa. However, even if the micropore filling was low, the narrow micropore (< 0.8 nm) filling is total, indicating that methane is not only adsorbed in the narrow micropores but also of pores of larger size (García Blanco et al., 2016).

This need of larger pores for the methane adsorption was also evidenced by Casco et al. when comparing a set of activated carbons with different pore volumes (Fig .4) (Casco et al., 2015b). Samples with pure micropororous network (F400 and Maxsorb) showed a maximum adsorption capacity at moderate working pressures (6 – 8 MPa).In contrary, samples that combine micropores with mesopores (pore size < 2 nm) such as the activated carbons RGC30, LMA738, LMA405 and LMA726, there was a continuous increase of the adsorbed quantity. Furthermore, they found that the correct design of activated carbons can result in materials that meet both the DOE and DOE MOVE requirements set by the U.S. department of energy.



Fig. 4 Methane storage capacity for different activated carbons at 298 K and up to 10 MPa. MOF sample HKUST-1 has been included for the sake of comparison (closed symbols represent adsorption data, open symbols represent desorption data).Reprinted with permission from(Casco et al., 2015b). Copyright 2015 American Chemical Society.

As it was previously mentioned, the feasibility of using ANG instead of CNG and LNG relays on the ability of the adsorbent to store a high amount of gas inside the fuel tank, which means that the porous adsorbent must present a high packing density ensuring that the storage capacity and the energy density, on volumetric basis is high. Since the microporosity of ACs is created by removal of carbon atoms during the activation process, after a certain microporosity volume further activation is followed by the creation of macropores translating into a useless volume for gas storage and thus a decrease of the packing density. Hence, carbons with high CH₄ uptake might result in a lower volumetric storage capacity of methane than some with lower surface area. In this sense, it is useful to express the adsorption capacities in terms of volumetric quantities instead of gravimetric basis (Fig. 5) (Menon and Komarneni, 1998).



Fig 5. Effect of expressing the methane adsorption capacity in terms of gravimetric surface (m²g⁻¹) area vs volumetric surface area (m²cm⁻³) for different porous materials at 3.447 MPa and 298 K. A suited adsorbent for the ANG storage presents a maximum volumetric surface area. Reprinted by permission from Springer (Menon and Komarneni, 1998). Copyrights 1988.

Along with a high micropore volume, the activated carbon is required to have a high packing density, thus presenting a high energy density on volumetric basis. Optimization of the packing density can be done by means of improving the bulk density, monolith preparation and particle size design. It is strongly related to the activation degree of the adsorbent, the higher the surface area and pore volume are the lower the packing density is. Byamba-Ochir et al. showed the importance of the activation conditions for the preparation of ACs for methane storage. Authors found that the BET surface area increased with an increase of the ratio of activating agent and the Mongolian raw anthracite, however, the packing density gradually decreases and therefore does the volumetric capacity (Table 3) (Byamba-Ochir et al., 2017).

Table 4. Effect of activating agent/Mongolian raw anthracite radio in the volumetric methane adsorption capacity (Byamba-Ochir et al., 2017)

Carbon monoliths	SBET (m ² /g)	v/v (cm ³ _{CH4} /cm ³ _{activated carbon})
PMAC2/1-C3-65	622	43.6
PMAC1/1-C3-65	845	60.1
PMAC1/2-C3-65	1460	162.2
PMAC1/3-C3-65	1757	147.8

To further investigate the effect of the adsorbent design parameters and the volumetric storage capacity, Kumar et al. plotted the volumetric adsorption capacity versus the product of the packing density (PD) and the specific surface area of the adsorbents (Fig. 6) finding a logarithmic relationship. They determined that in order to achieve the volumetric methane adsorption capacity of 180 v/v, the adsorbents must have a surface area of 2463 m²cm⁻³ and packing density of about 1.2 gcm⁻³. This are just guidelines since consideration of pore volume and distribution must be made (Kumar et al., 2017b).

One important advantage of using activated carbons as adsorbent materials for gas separation and storage is the possibility of using different organic materials rich in carbon contents as precursors with coal and wood being the most employed ones. However, in recent years there has being an increasing interest on the production of AC from agricultural wastes. The use of this type of materials as precursors can reduce the pressure on mines and forests while valorizing products that otherwise result in waste production with economic and environmental impacts (Mozammel et al., 2002).



Fig. 6 Volumetric adsorption capacity (v/v) of carbon structures versus the product of the BET specific surface area (SSA) and packing density. Reprinted with permission from (Kumar et al., 2017a). Copyright 2017 American Chemical Society.

Examples of such precursor materials include rubber seed coat, olive seed, coconut shell, oil pal fiber and almond shell (Arami-Niya et al., 2012). Some of the results obtained using low-cost lignocellulosic biomass activated carbons for methane adsorption and storage are described in table 4.

Table 5. Biowaste precursors used to prepare activated carbons along with their specific surface area and CH₄ storage (González-García, 2018).

Precursors	Activating Agent	Surface Area (m ² g)	Storage Capacity
Sugarcane molasses	КОН	2202	197.23 mg g ⁻¹
Oil palm Shell	ZnCl ₂	870	13 cm ³ g ⁻¹
Corncob	КОН	3227	273.4 cm ³ g ⁻¹
Corn cobs	КОН	1320	120 v/v
Palm shell	H ₃ PO ₄		25.7 cm ³ g ⁻¹
Olive stones	H ₃ PO ₄	1014	4.69 mmol g ⁻¹

Activated carbons can be optimized for methane storage by testing several activation conditions, such as activation method, activation temperature, activation agent, activation agent to char ratio. Bagher and Abedi. studied the effect of activation conditions on a corn cobs based AC, showing that the methane adsorption capacity can be increased from a low 25 v/v to up to 120 v/v at 298 K and 3.44 MPa when using optimal activation conditions: activation temperature ok 823 K for 60 minutes and impregnation with KOH/char on a 1:1 ratio (Bagheri and Abedi, 2011). Using a similar optimization process Policicchio et al. were able to obtain a CH₄ adsorption capacity of 150 v/v at 298 K and 3.5 MPa for a cellulose based AC (Policicchio et al., 2013). This optimization relays on the change of the textural parameters of the adsorbent (such as BET surface area and pore size distribution) upon different activation conditions. Ruiz et al. tested the effect of different activation temperatures, activation agent/precursor weight and inner flow gas on the pore size distribution (Fig. 7) (Ruiz et al., 2017). Overall the best methane storage capacities were found when the activation method produced samples with high BET surface area. In the case of physical activation methods, a higher activation degree resulted in a development of the porosity thus resulting in a higher surface area. In the case of chemical activation, a low activation temperature (between 700 and 800 °C) and a higher activation ratio/precursor resulted in a higher narrow micropore (< 0.8 nm) volume.

All the hitherto present studies show that the new DOE MOVE requirements are difficult to achieve by activated carbons and its feasibility is questionable. Furthermore, using Monte Carlo simulations a maximum methane storage capacity of activated carbons has been predicted to be 209 v/v at 3.4 MPa (Matranga et al., 1992). Thus, this result implies for researchers working in the design of new materials for the methane storage to reconsider this targets and/or changing the operating conditions (i.e. using higher pressures).



Fig. 7 Micro-mesoporosity volume distributions (in percentages) within the chestnut shell activated carbons. Reprinted from (Ruiz et al., 2017). Copyright (2017), with permission from Elsevier.

9.2 Carbon Dioxide Storage

Reducing carbon dioxide (CO₂) emissions is a key factor to combat global warming since among the greenhouse gases CO₂ is the one emitted in higher volume. It is estimated that in 2006, CO₂ alone accounted for 81.6% of the total GHG emissions in the US (United States Environmental Protection Agency (EPA), n.d.; Yamasaki, 2003). The CO₂ atmospheric concentration has increased from a preindustrial value of 280 ppm to 408 ppm in 2018 resulting in an increase of 1°C of the global temperature (Intergovernmental Panel on Climate Change (IPCC), 2007; NASA, 2018).

The main source of carbon dioxide is the fossil fuels combustion followed by land-use change. Annual fossil fuels emissions increased from an average 6.4 GtC in the 1990s to 7.2 GtC average per year in 2000-2005 (1 gigatonne of carbon or GtC = 1 billion tonnes of carbon) (Plaza et al., 2008). Fossil fuel combustion is the principal source of anthropogenic carbon dioxide emissions (58% of global GHG emissions (International Energy Agency, 2017).

Carbon Capture and Storage (CCS) has been proposed one of the most promising technologies to reduce carbon dioxide emissions in the short-term while carbon-free processes are implemented. It is believed that by the year 2050 CCS can reduce by 20% the carbon dioxide emissions (Global CCS Institue, 2011). CSS is a three step process:

- i) Separation of CO₂ from other emissions
- ii) CO₂ transportation to the storage site
- iii) Permanent storage

The transportation and the storage step are already developed technologies. It is the first stage of the CCS that has made difficult its implementation. The separation of the CO₂ accounts for approximately two thirds of the total cost of a CCS (Intergovernmental Panel on Climate Change (IPCC), 2005; Li et al., 2011). In this sense, between the proposed carbon dioxide capture and separation options, post-

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combustion capture has been appointed as the most feasible to implement in power plants as it can be retrofitted to existing plants without modification of the already existing systems (Lee et al., 2012).

The process of post-combustion can be done by many separation technologies such as physical adsorption, chemical adsorption, cryogenic separation and membranes. Most of the carbon dioxide capture systems commercially available employ chemical absorption with alkaline amines. Despite of the method availability, remarkable disadvantages such as high energy requirements, oxidative degradation of the adsorbents, high corrosion and low cost efficiency are noted. Other possible technologies are cryogenic separation and membrane separation. However, the cryogenic separation has the disadvantage of using a great quantity of energy that makes it too expensive and the use of membranes is still in a very young developing phase (Oh, 2010).

Adsorption technology involves two steps, on the first step the carbon dioxide adsorption onto a porous adsorbent by the formation of chemical bonds (chemisorption) or by physical interactions such as van der Waals forces (physisorption). The second stage is the regeneration of the adsorption column after the system equilibration. The regeneration can be done by reducing the pressure (Pressure Swing Adsorption - PSA) or by increasing the temperature (Temperature Swing Adsorption - TSA).

The adsorption process using physisorbent materials is considered as the most cost-effective option for carbon dioxide separation due to its low energy requirements. In addition, it is a very noble technology that can be easily scale to different plant sizes and used on a wide range of temperatures and pressures. The success of an adsorption separation technology relays on the quality of the employed adsorbent. It has been described that a successful adsorbent for the carbon dioxide capture and storage should satisfy five main requirements (Sevilla et al., 2012):

- i) Large CO₂ uptake
- ii) High sorption rate
- iii) Good selectivity between CO₂ and other competing gases
- iv) Easy regeneration

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v) Low cost and high availability

The search for new adsorbent materials that better fulfill these requirements has become an active field of research. Adsorbents that have been studied for this purpose include activated carbons, zeolites, MOFs and polymers.

It is important to mention here that development of better adsorbents for the carbon dioxide separation would not be only beneficial to the CCS systems but also for the upgrading of biogas and natural gas. In this context, activated carbons (CAs) are considered as a superior material for the CO₂ capture and separation due to their hydrophobic character, lower cost, high thermal stability and low regeneration energy requirements (Plaza et al., 2010).

Similar to the adsorption process of methane, the textural properties of activated carbons play a major role on the adsorption behavior. CO₂ adsorption is enhanced by the presence of micropores of up to 1 nm. By comparing a number of porous carbons, Presser et al. showed that the volume of pores larger than 1 nm is not a major contributor to the CO₂ adsorption (Fig. 8c). In fact, it is the volume of ultramicropores (size smaller than 0.8 nm) that has the higher influence upon CO₂ adsorption as proven by a higher correlation coefficient (R² of 0.9958) (Fig. 8d). Presser et al. reported that upon very low CO₂ partial pressures such as 0.01 MPa it is the pores with a size smaller than 0.5 nm that are the main contributors (Presser et al., 2011). The dependency of the adsorption of carbon dioxide on the ultramicropore volume was also observed by Guojun et al. at 273, 283 and 293 K and pressures between 0.01 MPa and 0.10 MPa (Yin et al., 2013).



Fig. 8. CO₂ uptake at 273 K 0.1 MPa for the volume of pores a) Pore Volume < 0.5 nm, b)) Pore Volume < 0.8 nm 1.5 nm and c) Pore Volume > 1.0 nm. Reproduced from (Presser et al., 2011)with permission of The Royal Society of Chemistry.

If ultramicropore volume has proven to influence the adsorption capacity of the activated carbons no clear relationship between the BET surface area and the adsorption capacity has been reported for pressures up to 0.1 MPa (Presser et al., 2011; Serafin et al., 2017; Yin et al., 2013). The potential theory indicates that CO₂ adsorption happens with micropore filling followed by capillary condensation and liquefaction with increasing pressure, it is expected then that at high pressure the BET surface area and total pore volume gain influence the CO₂ adsorption. Drage et al. reported that at a pressure of 4.1 MPa a linear relationship between the CO₂ adsorption capacity of activated carbons and the BET surface area can be found (Drage et al., 2009)

Another important factor to be considered when designing a carbonaceous porous material for carbon dioxide adsorption is the source of the raw material from which it will be produced. One of the main advantages of activated carbons is their versatility, any carbonaceous material can be converted into activated carbon with the main requirements being a high carbon content and low in ash, this arises the possibility of producing environmental friendly activated carbons based on lignocellulosic biomass. In this context, several efforts have been made to produce CAs from lignocellulosic resources with the aim of CO₂ adsorption (table 6).

Table 6. Biowaste precursos used to prepare activated carbons along with their activation method, specific surface area and CO₂ storage capacity (González-García, 2018).

Precursors	Activating Agent	Surface Area (m ² g)	Storage Capacity
Oil cake/walnut	CO ₂	1207	34.72 ml g ⁻¹
Coconut shell	CO ₂	371	1.8 mmol g ⁻¹

Almond-shell	CO ₂	862	2.7 mmol g ⁻¹
Olive stone	CO ₂	1215	3.1 mmol g ⁻¹
Coffee residue	CO ₂	593	2.4 mmol g ⁻¹
Rice husk	ZnCl ₂	927	1.3 mmol g ⁻¹
Peanut shell	КОН	956	4.0 mmol g ⁻¹
Sunflower seed	КОН	1790	4.6 mmol g ⁻¹
Bamboo	КОН	1846	4.5 mmol g ⁻¹
Palm stone	H ₃ PO ₄	924	2.7 mmol g ⁻¹

The selection of the raw materials prior to the activated carbon production is of main concern for the gas storage suitability of the adsorbent. Olivares-Marín et al. showed the importance of the careful selection of the precursor. The Authors compared the CO₂ one sample of pre-consumer carpet and two samples of post-consumer sample under the same activation conditions finding that the presence of inorganic binders in the precursor can result in a lower micropore volume lowering then the CO₂ adsorption capacity between the samples from 13.8 wt% to 3.7 wt%. Hao et al. obtained similar findings when comparing activated carbons obtained from grass cuttings, house manure, organic waste from beer production and bio-sludge from a wastewater treatment plant. The activated carbons produced from grass cutting and horse manure showed higher CO₂ adsorption capacities due to a larger volume of ultramicropores (Hao et al., 2013). Munusamy et al. showed that using carboxyl methyl cellulose sodium salt and soluble starch as binders to convert a powdered mango seed shell derived activated carbon into extruded form they could enhance the maximum CO₂ equilibrium adsorption capacity from 14.3 wt.% to 21.3 wt.% at 0.1 MPa and 273 K explained by the formation of surface functionality by the binders on the extruded AC (Munusamy et al., 2015).

In addition to the precursor nature, the optimal operating conditions of the activation process need to be determined since they can significantly affect the adsorption capacity of the AC. There are two possible activation paths for producing ACs, chemical activation and physical activation. Chemical activation using KOH, H₃PO₄ or ZnCl₂ is usually preferred since it results in hierarchical porous structures with high surface areas it also consumes less energy as the chemical activation temperature is between 600 and 800°C versus 800 to 1000°C for physical activation. Chemical activation has the additional advantage that it can be done in one stage where carbonization and impregnation of the precursor are done simultaneously, while physical activation involves a first step of carbonization and a second step where the material is activated in the presence of oxidants such as air, CO₂ and steam (Singh et al., 2017; Wang and Kaskel, 2012).

Song et al. compared different activation methods and conditions for corn stalk based ACs. The ACs produced by physical activation at different activation concentrations or activation times presented surface areas lower than 500 m²g⁻¹ while chemical activation resulted in slightly higher surface areas (up to 639.8 m²g⁻¹), additionally the chemical activated carbons also showed higher micropore volumes. These two factors resulted in higher CO₂ uptakes of chemical activated carbons than physical activated ones (Fig. 9). Between all the samples a maximum carbon dioxide adsorption capacity of 7.33 wt.% was obtained (Song et al., 2015). Alabadi et al. were able to obtained highly porous activated carbons by chemical activation with KOH of a starch and gelatin 1:1 mix, a CO₂ capture capacity of 3.8 mmol g⁻¹ at 298 K and 0.1 MPa was obtained (Alabadi et al., 2015). Using also KOH activation to produce a pine nut-shell-derived activated carbon a high CO₂ uptake of 5.0 mmol g⁻¹ at 273 K and 0.1 MPa was determined by Deng et al. (Deng et al., 2014). A CO₂ adsorption of 5.22 mmol g⁻¹ at 293 K and 0.1 bar, one of the highest reported CO₂ adsorption uptakes for carbonaceous materials, was obtained by KOH chemical activation of Coca-Cola as a source of biomass by Boyjoo et al. (Boyjoo et al., 2017).

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Fig. 9 Effect of a) BET surface area and b) t-plot micropore area on CO₂ adsorption capacity at 298 K for chemical activated (squares) and physical activated (triangles) activated carbons from corn stalk Reprinted from(Song et al., 2015), Copyright (2015) with permission from Elsevier.

Typical activated carbons show weak affinity towards carbon dioxide with an adsorption capacity of surrounding 5 wt.% at 298 K and a pressure of 0.01 MPa (nearly the CO₂ partial pressure of post-combustion gas) (Lee and Park, 2015). This value however, can be improved by chemical modification of the AC's surface. One possibility is to introduce Lewis bases onto the activated carbon surface which favors the CO₂ adsorption due to its acidic properties. Nitrogen enrichment has been reported to be an effective way for introducing basic groups on the AC. Meng et al. reported a CO₂ adsorption capacity of up to 17.7% wt. at 298 K and 0.01 MPa when treating a polypyrrole activated carbon with NaOH, they reported that the adsorption capacity was both a function of the nitrogen content and the micropore volume (Meng and Park, 2014). Other works reported the same dependence between the carbon dioxide capacity and both the textural properties and N content (Tiwari et al., 2017; Yaumi et al., 2018).

When introducing nitrogen functionalities the presence of oxygen surface groups is desirable, since they will act as anchoring sites for the nitrogen groups. Caglayan et al. studied the effect of different oxidation and heat treatments on the formation of N surface groups. They found that the CO₂ adsorption capacity was enhanced by Na₂CO₃ impregnation of the air oxidized and nitric acid oxidized AC. They reported that sodium atoms (Na) have CO₂ sorption ability and that the oxidation with HNO₃provided anchoring sites for Na atoms (Caglayan and Aksoylu, 2013). Shafeeyan et al. compared the CO₂ adsorption capacity of a heat-treated under nitrogen AC without pre-oxidation and amination of pre-oxidized AC (Shafeeyan et al., 2011). The samples that were not pre-oxidized presented a higher BET surface area, pore volume and micropore volume. This behavior is due to the thermal decomposition of functional groups inside the pores. Furthermore, when the sample is previously oxidized there is a blockage of the micropores entrances by oxygen and also a collapse of some pore walls leading to a decrease of the BET surface area and the micropore volume. However, if the oxidation step is done at high temperature (800 °C) the oxygen surface groups are decomposed and the textural properties are recovered (Fig. 10, sample OXA-800). They found a clear relationship between both micropore volumes and nitrogen content with the CO₂ adsorption capacity.



Fig. 10 Relationship between the CO₂ capture capacity and a) micropore volume, b) nitrogen content (N%). OXA-400 and OXA-800: Pre-oxidized samples aminated at 400 and 800 °C, HTA-400 and HTA-800: Heat treated samples aminated at 400 and 800 °C without pre-oxidation. Reprinted from Publication (Shafeeyan et al., 2011), Copyright (2011), with permission from Elsevier.

Seema et al. showed that upon Sulphur (S) doping of microporous carbonaceous materials the CO₂ adsorption capacity can be increased. They found a correlation between the BET surface area, micropore volume and oxidized Sulphur surface groups with the carbon dioxide adsorption capacity. A maximum adsorption capacity of 4.5 mmol g⁻¹ was reached (Seema et al., 2014). Xia et al. also

reported an increased on the CO_2 adsorption capacity after S-doping of microporous carbons, they obtained a maximum adsorption capacity of 2.46 mmol g⁻¹ when a S content of 6.56% was present and of 1.69 mmol g⁻¹when there was no Sulphur atoms on the carbonaceous material.

It is then that even if the CO₂ adsorption capacity of activated carbons is greatly influenced by the BET surface area and the volume of narrow micropores (< 0.8 nm), the surface chemistry can also play an important role in the search of better materials for the CO₂ separation and storage. Different functionalities can be added to the surface of the adsorbent that will participate in the adsorption process trough acid-base interactions and polarizability of the carbon dioxide molecules.

In addition to nitrogen and sulphur doping, activated carbons can be treated with metal oxides resulting in chemical reactions between the CO₂ molecules and the metal oxides at elevated temperatures. Published works include impregnation of activated carbons with alkaline metals such as Mg and Ca, and transition metals like Cu, Co, Ni, Fe, Cr and Zn (Somy et al., 2009; Son et al., 2005; Yong and Mata, 2001). Son et al. determined the CO₂ adsorption capacity of metal impregnated activated carbons to decrease in the order of Mg>Ca>Co=Cu>Ni (impregnated metal) (Son et al., 2005). Using Zn²⁺ as a promotor (Fig. 11), Somy et al. tested the adsorption behavior of activated carbons impregnated by Fe₂O₃ and Cr₂O, finding a 20% increase of the CO₂ adsorption capacity upon Cr₂O impregnation (compared with raw AC); meanwhile Fe₂O₃ impregnation was not effective (Somy et al., 2009). It has been reported that an acidic treatment prior to the metal impregnation can further increase the carbon dioxide adsorption, the acid groups will oxidize the samples and provide oxygen surface groups that can form metal complexes with the metal ions (Hosseini et al., 2015).



Fig. 11 CO_2 adsorption isotherms of Zn^{+2} –metal oxide samples prepared by solution and slurry impregnation methods. Reprinted from (Somy et al., 2009), Copyright (2009), with permission from Elsevier.

9.3 Hydrogen Storage

One way to reduce the greenhouse gas emissions can be achieved by designing suitable adsorbents to efficiently remove gases like CO₂ and CH₄ from the atmosphere. In addition, renewable clean energy sources (water, wind, solar) are currently developed to replace at long term the fossil fuels. However, an important disadvantage of renewable energy sources is their intermittence which requires efficient energy storage systems to store and release energy when needed. Hydrogen, is one of the renewable resources that can help in addressing the growth of energy demand without contributing to the global climate change. Hydrogen is an abundant element in fossil hydrocarbons and all organic matter. In its pure molecular form, H₂, hydrogen is a colorless, odorless, non-toxic gas which produces only water vapor when used in to a fuel cell, making it a clean fuel. H₂ has the advantage to posses the highest energy density among common fuels by weight.; Unfortunately, gaseous H₂ has the lowest energy density by volume and present a risk of explosion in contact with the air. Therefore, developing safe,

efficient and inexpensive hydrogen storage system with acceptable gravimetric and volumetric energy densities is still an objective for researchers to achieve (Andrews and Shabani, 2014).

At present, two methods are mainly used for the storage of large quantities of hydrogen. The storage in the gaseous state is done at a high pressure (700 bars) in bottles which is rather heavy and bulky. The storage in the liquid state requires a temperature of 20K and therefore expensive cooling devices along with significant losses by evaporation (boil-off phenomenon). Both methods present significant problems related to the storage under pressure and the high explosive potential of hydrogen in air (above 4% by volume). Third solution is solid hydrogen storage, with significant safety and volume capacity advantages.

US Department of Energy (DOE) targeted a minimum gravimetric and volumetric requirement for onboard hydrogen storage systems. However, the current market and technology scenario have led to some revisions and changes in targets. In 2010, the targeted storage system has a gravimetric density of 4.5 wt% (1.5 kWh/kg) and volumetric density of 28 g/L (0.9 kWh/L), while for 2015, the targeted gravimetric density was changed to 5.5 wt% (1.8 kWh/kg), and the volumetric density was revised to be 40 gH₂/L (1.3 kWh/L) (Md Arshad et al., 2016).

As highlighted in Figure 12(Durbin and Malardier-Jugroot, 2013), the current hydrogen storage technology do not meets the DOE standards. This may be related to fundamental limitations of each particular storage method: i) the gaseous H₂ requires a large volume to be stored, ii) the liquid hydrogen requires low temperatures and evaporates easily, iii) metal hydrides are heavy and present thermodynamic hydrogen release limitation and IV) adsorbent materials present low adsorption capacities at room temperature and pressure. Therefore, significant efforts are made to improve these storage techniques and to develop new storage methods.

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Fig. 12 Ranking of current hydrogen storage technologies by volumetric and gravimetric capacities showing the DOE 2015 and ultimate targets.Reprinted from (Durbin and Malardier-Jugroot, 2013), Copyright (2013), with permission from Elsevier.

A great interest has been focused on hydrogen physisorption on different adsorbents including zeolites, metal organic framework, and carbon-based materials. Particularly, carbon adsorbents exhibit several advantages compared to other adsorbents, i.e., low cost, low densities, fast kinetics, and complete hydrogen reversibility(V et al., 2010). Moreover, the hydrogen sorption capacity can be comparable or even higher than MOFs and zeolites (Builes et al., 2011). Among the carbonaceous materials, activated carbons (ACs) have been widely investigated as H₂ storage materials due to their easiest preparation process combined with more developed porous structure when compared to nanostructured carbons (carbon nanotubes, carbon nanofibers, template carbons).

Biomass is widely employed as precursor to obtain the carbon mainly to their eco-friendly nature, widespread availability and high adsorption/desorption rates (Agarwal et al., 1987; Masika and Mokaya, 2013). Several biomass precursors have been explored, i.e., cellulose, corncob, rice husk, olive bagasse, starch etc. (see Table 7), however, simple pyrolysis of biomass results in low surface area carbon materials and in most of the cases an activation step is usually used in order to develop the porosity. Mainly, two types of activation procedure based on physical (CO₂ or water steam) and

chemical activation (KOH, H₃PO₄...) are used. In particularly, as revealed by Table 7-9 gathering several literature data, chemical activation with KOH is the most used since it leads to highly microporous activated carbons with well developed surface areas (up to 3700 m²·g⁻¹). In addition, the chemical activation requires lower activation temperatures and shorter reaction time than physical activation, but on the contrary it requires a supplementary washing step in order to remove the impurities created during the activation process.

The development of the porosity is a consequence of carbon oxidation (etching) due to the redox reaction between potassium hydroxide and carbon with the formation of carbonates, release of volatiles (H_2 , CO_x) and intercalation of K, according to the following equations(Choi and Park, 2015; Wang et al., 2009):

 $6KOH + 2C => 2K + 3H_2 + 2K_2CO_3$ (1) $K_2CO_3 + C => K_2O + 2CO$ (2) $K_2O + C => 2K + CO$ (3)

Other parameters such as the carbon precursor type, the activation agent type/ amount, and the activation temperature/time using during the carbon preparation are the main factors determining the porosity development and the surface chemistry of the materials. Both parameters are of prime importance when come to design efficient adsorbents for hydrogen storage applications and will be discussed further in more details.

9.3.1 Influence of adsorption conditions (T, P)

The hydrogen storage studies are mainly focused at 77K, due to the weak interaction between molecular hydrogen and solids. The amount of hydrogen stored at room temperature is very low, while it can consistently increase at low temperature, typically 77K, that is, the temperature of liquid nitrogen. At elevated temperatures such as room temperature, the storage capacity was found to vary linearly with pressure, following Henry's law, while at 77K the adsorption isotherm of all samples could be satisfactorily explained using the Langmuir model (Figure 13).



Fig. 13Adsorption isotherms of H₂ on the activated carbons AX-21 at different temperatures. Reprinted from (Chahine and Bose, 1994), Copyright (1994), with permission from Elsevier .

Therefore, at room temperature the hydrogen adsorption capacity is very low. This trend can be also seen as well in Table 7, where several hydrogen adsorption capacities of several biomasses derived activated carbon are shown. It can be observed that the maximum achieved capacity at room temperature and high pressure is only slightly superior to 1 wt% even if the surface area of the carbon is very high (~ $3000 \text{ m}^2 \cdot \text{g}^{-1}$).

Therefore, much more studies are made in cryogenic conditions, i.e. 77K and different pressures. It is therefore worth to discuss the influence of the porosity on the adsorption capacity in such conditions.

Table 7. Activated carbon derived from different biomasses along with their textural properties and hydrogen adsorption capacity at room temperature (398-303 K) and different pressures.

				H₂ ads.			
Precursor	Activating BET SS agent (m²/g	BET SSA Vμ	Capacity T(K)	T(K) P(bar)	Refs		
		(m-/g)	(cm³/g)	(wt%)			

Cellulose	КОН	3771	-	0.5	298	20	(Blankenship li et
							al., 2017)
							<i></i>
Olive	КОН	1269	0.48	1.22	298	204	(Heo and Park,
pomace							2015)
lute fibers	кон	1224	0 43	12	303	40	(Ramesh et al
	Kerr	122 1	0.15	1.2	505	10	2017)
							2017)
Tamarind	кон	1784	0.64	1.36	303	60	(Ramesh et al.,
seeds							2015)
							/
Coffee	КОН	3149	-	0.91	298	140	(G. Li et al., 2016)
shell							
Olive	CO ₂	1185	0.45	0.63	298	200	(Bader and
bagasse							Abdelmottaleb,
							n.d.)

9.3.2 Influence of porosity

Table 8 present the hydrogen adsorption capacities at 77K and 1 bar, for activated carbons issued from several biomasses (beer lees, rice husks, starch hemp stem etc) activated with CO_2 or KOH. At a first glance, it can be noticed that the hydrogen capacities are comprised between 1.97 and 3.2 wt.%, for activated carbon materials possessing specific BET surface area ranged between 687 and 3771 m²·g⁻¹ and micropore volumes between 0.3 and 1.19 cm³·g⁻¹. In order to evidence any correlation between the hydrogen adsorption capacity and the SSA and the micropore volume, several data were collected from the references listed in Table 8 and plotted in Figure 14.

Table 8. Activated carbon derived from different biomasses along with their textural properties and hydrogen adsorption capacity at 77 K and 1 bar.

Precursor	Activating agent	BET SSA (m²·g⁻¹)	Vμ (cm³·g ⁻¹)	H ₂ ads. Capaci ty (wt%)	Т (К)	P (bar)	Refs.
Fruit bunch	CO ₂	687	0.297	1.97	77	1	(Md Arshad et al., 2016)
Beer lees	КОН	1927	0.70	2.92	77	1	(Balathanigaimani et al., 2018)
Rice husks	КОН	2682	0.792	2.85	77	1	(Heo and Park, 2015)
Cellulose	КОН	3771	-	3.9	77	1	(Andrews and Shabani, 2014)
Starch	CO ₂	3281	1.1	2.3	77	1	(Y. Li et al., 2016)
Fungi	КОН	2137	0.87	2.4	77	1	(Wang et al., 2014)
Corncob	КОН	3708	1.14	3.2	77	1	(Liu et al., 2014)
Chitosan	КОН	2840	1.06	2.88	77	1	(Wróbel-Iwaniec et al.,
Chitosan	КОН	2919	1.19	2.71	77	1	2015) (Wang et al., 2016)
Surcrose	КОН	1552	0.72	2.5	77	1	(Choi and Park, 2015)
Hemp Stem	КОН	3078	1.13	3.18	77	1	(Zhang et al., 2017)

As can be noticed, the capacity increases rather linearly from 1.5 to 3.2 wt% when increasing the surface area from 468 to 3078 m²·g⁻¹. Similar trend of capacity improvement is observed with the increase of the microporous volume, however, in this case the data point are more scattered. This behavior maybe due to the different pore size or surface chemistries of these materials which induce also an effect on the adsorption capacity as it will be presented latter in the chapter.

D	Activating	BET SSA	Vμ	H₂ ads.	т	Ρ	Refs.
Precursor	agent	(m²/g)	(cm³/g)	Capacity	(К)	(bar)	
				(wt%)			
Corncob	КОН	3708	0.59	5.80	77	40	(Wang et al., 2009)
Cellulose	КОН	3771	-	8.1	77	20	(Blankenship li et al., 2017)
Olive	КОН	1269	0.48	3	77	25	(Bader and
pomace				6.11	77	209	Ouederni, 2017)
Activated	КОН	3190	1.09	7.08	77	20	(Wang et al., 2009)
Carbon							
Olive	CO ₂	1185	0.45	2.59	77	25	(Bader and
bagasse				3.34	77	200	Abdelmottaleb, n.d.)
Starch	CO ₂	3281	1.1	6.9	77	20	(Y. Li et al., 2016)
Fungi	КОН	2137	0.87	5.0	77	20	(Wang et al., 2014)
Corncob	КОН	3708	1.14	5.5	77	20	(Liu et al., 2014)
Chitosan	КОН	2840	1.06	5.01	77	40	(Wróbel-Iwaniec et
		2919	1.19	6.77	77	20	al., 2015)
							(Wang et al., 2016)
Melaleuca bark	КОН	3170	1.07	3.18	77	10	(Xiao et al., 2014)

Table 9. Activated carbon derived from different biomasses along with their textural properties and hydrogen adsorption capacity at low temperature, 77 K and different high pressures.



Fig. 14Correlation between the hydrogen adsorption capacity at 77K and 1bar, and the specific surface area (a) and microporous volume (b) (literature data are extracted from references listed in Table 8).

The pressure as mentioned has an important influence on the hydrogen adsorption capacity. In Table 3, several literature data are collected at 77K a high pressures (20 - 209 bars) for several activated carbons. The adsorption capacities are much higher than at atmospheric pressure and may reach 8.1 wt.% by combining a high surface area as well. Nevertheless, if one plot the values form Table 3 references (Figure 4) it can be clearly seen a linearly relationship between the adsorption capacity and the specific surface area and the microporous volume. The capacity increases from 4.2 to 8.1 by increasing the BET surface from 1452 to 3771 m²·g⁻¹.



Fig. 15 Correlation between the hydrogen adsorption capacity at 77K and 20 bars, and the specific surface area (a) and microporous volume (b) (literature data are extracted from references listed in Table 9).

Indeed the experimental conditions (T, P) and the textural properties of carbon materials are the main important parameters influencing the hydrogen adsorption capacity. Both the specific surface area and the microporous volume affect the adsorption capacity, and generally an increase of the capacity with the increase of the textural values is observed but a linear correlation is difficult to obtain when gathering many different carbon materials. This may be mainly due to the fact that the carbon materials may exhibit different pore size and surface chemistries. If one focus on one series of activated carbons derived from the same precursor for instance chitosan biopolymer, Wrobel-Iwaniectry et al. (Wróbel-Iwaniec et al., 2015) correlate the H₂ adsorption capacity at 77K and 1 bar with BET surface area (S_{BET}), total pore volume (V_T) and micropore volume (V_{DRN2}) and linear relationship was observed with a correlation coefficient (R^2) being the highest for the microporous volume (0.916). The same trend was observed at high pressures (40 bars) and in this case the correlation coefficient (R^2) is even higher than at low pressure (1bar), i.e., 0.961 for microporous volume. This is in line with the observations made in Figure 14 and 15 where several activated carbon hydrogen sorption capacities were presented. Similar observations showing the importance of micropores (pores smaller than 2 nm)



Fig. 16 The variation in hydrogen adsorbed at 77 K and 1 bar pressure with a) BET surface area, b) total pore volume and c) total micropore volume for activated carbons derived from chitosan. Reprinted from (Wróbel-Iwaniec et al., 2015), Copyright (2015), with permission from Elsevier.

V_T (cm³ g⁻¹)

0.5

ŕ

1.5

0

 $R^2 = 0.862$

4000

3000

Ť

1.5

0

0.5

V_{DRN2} (cm³g⁻¹)

 $R^2 = 0.916$

1.5

 $R^2 = 0.820$

1.5

ŕ

1.5

0

1000

2000

S_{BET} (m² g⁻¹)

Further investigation lead to the conclusion that H₂ storage capacity correlates better with the ultramicropore volume (volume of pore having size < 0.7 nm) than the surface area of carbon materials. For example, Texier Mandoki *et al.*(Texier-Mandoki et al., 2004) have shown that the hydrogen adsorption capacity of activated carbons is directly related to its CO₂ micropore volume (Figure 17 a). This work emphasizes the importance of the presence of a homogeneous narrow microporosity (dpores 0.7nm) for the hydrogen adsorption and the necessity to complete the N₂ adsorption characterizations of carbon materials by the CO₂ adsorption measurements at 273 K, in order to obtain a better knowledge of the whole microporosity range of activated carbons. However, in most of the available literature data, the N₂ adsorption is systematically used while the CO₂ adsorption is only marginally employed. To get more deep understanding in the contribution of porosity to the hydrogen capacity, Shafer *et al.*(Schaefer et al., 2016) gather the adsorption capacity of different activated carbon at room temperature and 77K and different pressure. As seen in Figure 17b, the main contribution to H₂ uptake is due to the narrowest micropores (0.5-0.7nm). When increasing the pore width the hydrogen uptake gradually decreases. The authors pointed also that the pore diameter through the overlap of the adsorption potentials is significantly less important at 77 K than at 298 K, temperature at which the adsorption potential needs to be very high to allow hydrogen adsorption at the carbon surface.



Fig. 17 H₂ adsorption capacity at 77K and 10 bar versus N₂ and CO₂ micropore volume. Adapted from (Texier-Mandoki et al., 2004), Copyright (2004), with permission from Elsevier.(a) and contribution of each pore diameter to the hydrogen up-take at different temperatures and pressures. Adapted from (Schaefer et al., 2016), Copyright (2016), with permission from Elsevier.

The main conclusions about these studies is that smaller pores ~ 0.7 nm are more convenient for hydrogen storage. Therefore, several studies focused on the effect of pore size on hydrogen uptake in high surface area porous carbons and the following optimum pore diameters for H₂ storage. However, Sethia and Sayari (Sethia and Sayari, 2016) and Masika and Mokaya (Masika and Mokaya, 2013) pointed out that in many studies the pore size vary in the same time with the pore volume or surface area, therefore, ambiguities may exist when studying the effect of pore size for such materials. In this sense, Sethia and Sayari (Sethia and Sayari, 2016)concluded that the high hydrogen uptake is associated for the activated carbon having the most important cumulative volume of ultra-micropores (<0.7). Two materials exhibited identical total pore volume (0.64 cm³/g vs. 0.65 cm³/g) and BET surface area (1312 m²/g vs. 1342 m²/g) has been prepared and it has been determined that they exhibited 21.8% difference in the hydrogen uptake. The authors attributed this significant decrease in hydrogen uptake to the occurrence of a broader PSD with larger micropores (0.63 vs. 0.59 nm) together with

28.6% decrease in ultra-micropore volume (0.21 vs 0.27 cm³/g). Masika and Mokaya (Masika and Mokaya, 2013) showed that the excess and total hydrogen uptake (at 77K and 20 bar) of activated carbons with identical surface areas of 3340 m²/g(Fig 18 left), increased from 3.7 and 5.4 wt % (31 Å pore size sample), to 4.8 and 6.3 wt % (23 Å pore size sample), and to 6.3 and 7.3 wt % for a 12 Å pore size sample. The authors concluded that the excess hydrogen storage density (μ mol H₂·m⁻²) of the carbons decreases linearly with pore size (Figure 18 right).



Fig. 18Pore size distribution curves (left) of porous carbon materials along with their hydrogen adsorption up-take vs. pore size (right).Reprinted with permission from (Masika and Mokaya, 2012) Copyright (2012) American Chemical Society.

9.3.3 Influence of surface chemistry

The interaction between adsorbed hydrogen and carbons is very weak and can be improved via functionalization of the surface with heteroatoms. The researches have demonstrated that the surface chemical properties of activated carbons have an important role in their sorption capacity. The amount of hydrogen adsorbed on activated carbons depends on the surface groups present on activated carbons; however, the presented literature results are somehow contradictory. Agarwal et al. (Agarwal et al., 1987)prepared activated carbons with modified amount of oxygen surface groups by thermal

oxidation at different temperatures and for different time periods. The nature and the amount of formed oxygen complexes were determined by Boehm titration, and it was found that the surface oxygen complexes affect the amount of hydrogen adsorbed on the surface, i.e., the amount of hydrogen adsorbed increases with the amount of surface acidic complexes. Concerning the influence of basic complexes groups the authors did not found any trend with the amount of hydrogen adsorbed. Georgakis et al (Georgakis et al., 2007)used a theoretical approach via molecular dynamic simulation in order to understand the impact of oxygen functional groups on the hydrogen adsorption capacity. Six types of commons oxygen groups found on porous carbon have been used (carboxyl, carbonyl, lactones, anhydrides, phenols and quinones). They end up to the conclusion that the oxygen functional groups induce a decrease in hydrogen adsorption capacity and this effect is more pronounced with the narrow pores (Figure 19a). They accounted this behavior to mainly three factors, i.e., increase of weight due to the oxygen, steric hindrance effects and weaker oxygen–hydrogen interactions compared to carbon–hydrogen (Figure 19b).



Fig. 19 Correlation between the adsorbed hydrogen amount for basic carbon (no oxygen functional groups and oxygenated model (contains 6 types of oxygen groups) vs. the pore size obtain by molecular dynamics simulation (left); Hydrogen adsorption snapshots for the pore size of 0.7 nm (a) purely carbonaceous material; (b) oxygenated materials when the adsorption simulation is complete in the basic model (right). Readapted (left) and reprinted (right) from (Georgakis et al., 2007), Copyright (2007), with permission from Elsevier.

From experimental point of view, Huang et al. (Huang et al., 2010) found the same trend, i.e. hydrogen uptake decreases when the carbons present oxygen functional groups. The authors used activated carbon derived from litchi wood which was post-modified by nitric acid or hydrogen peroxide at various temperatures to enable the formation of oxygen functional groups. In some cases however, the drastic oxidation conditions lead not only to the introduction of functional groups but also to the porosity collapsing and therefore, to the hydrogen uptake decrease (Figure 20 a). The authors found that the hydrogen capacity is not depending on the amount of the basic groups, but rather on the amount of acidic groups. It has been observed as well that the hydrogen capacity is significantly suppressed when increasing the acidic group amounts more than 0.8 mmol/g (Figure 20a).



Fig. 20Comparison of hydrogen capacity of the pristine activated carbon and oxidized activated carbon (left) and effect of surface acidic groups on hydrogen capacity measured at different conditions (right). Reprinted from (Huang et al., 2010), Copyright (2010), with permission from Elsevier.

Many other authors found that the amount of hydrogen adsorbed remarkably decreased by oxidation of the carbons (Bleda-Martínez et al., 2008; Takagi et al., 2004; Zhao et al., 2005) and they accounted this to the decrease accessibility to aromatic C–C bonds due to the pore blockage with the functional groups (Schimmel et al., 2003; Takagi et al., 2004). Some authors prove that the oxygen groups do not affect the hydrogen up-take since no significant differences were detected by comparing their capacity with that of pristine carbons(Llorens and Pera-Titus, 2009; Tellez-Juarez et al., 2014). However, recent works performed by Blankenship II et al.(Blankenship Ii et al., 2017) claim an opposite behavior. By using hydrothermally treated cellulose further activated with KOH at different temperatures (600-800°C), highly porous carbon (2000-3700 m² g⁻¹) containing high amounts of oxygen (17.9-20.6 wt.%) were prepared. This combination of high surface area and high amount of functional oxygen groups was reached at 700°C of activation (CA-4700) and allow improving the hydrogen up-take (Figure 21 a) significantly comparing to the reference activated carbon AX21. To explain this behavior, the isosteric heat of hydrogen adsorption (Qst) was measured (Figure 21 b) allowing an estimation of the strength of interaction between the hydrogen and carbon surface. As can be noticed, the isosteric heat of hydrogen adsorption is significantly much higher for the cellulose-derived activated carbons, CA (8 – 10 KJ/mol) compared to that of reference activated carbon, AX21 (5-6 KJ/mol). The authors accounted this difference not only to the porosity but also to the presence of oxygen functional groups.



Fig. 21Hydrogen storage capacity at room temperature (a) and isosteric heat of hydrogen adsorption (b) of activated carbons derived from cellulose acetate and a commercially reference activated carbon AX21. Reprinted form (Blankenship li et al., 2017).

Taking therefore into consideration all these works showing contradictory results, one can be stated that the effect of oxygen groups on the hydrogen adsorption still need to be investigated in order to better understand it. Moreover, when other functional groups are present on the carbon surface, such nitrogen-based groups, the effect of these groups on hydrogen adsorption is still much difficult to assess. However, there are only few studies dedicated to the effect of nitrogen atoms on hydrogen storage performance. Wang et al. (Wang et al., 2016)used chitosan precursor, containing nitrogen in its skeleton to obtain N-doped porous carbon through an activation process. The amount of nitrogen could be varied from 6.5 to 0.53 wt% by increasing the activation temperature from 600 to 800°C. The authors found that the high N-doping is beneficial for hydrogen adsorption at low pressure but it is detrimental at high pressure. It is worth to mention that high nitrogen contents are always accompanied by significant oxygen contribution. However, the authors did not took into consideration this aspect which may contribute as well to the observed hydrogen storage behavior.

Fonctionalisation of carbon with metal-based nanoparticles such as (Pd, Pt, Ni...) or with metal hydrides (MgH₂, LiBH₄, LiAlH₄) may further improve the hydrogen up-take by chemisorptions and/or spill-over effects (Figure 24), however, this is a large topic which will not be developed here but can be overviewed in a recent review (Ren et al., 2017).



Fig. 24 Binding energies and the associated sorption mechanisms in different materials. Reprinted from (Dillon AC, et al.).

9.4 Conclusions

From environmental and economic point of value the use of biomass to prepare carbon materials with tunable features is demonstrated in many works. Many sources of biomass or biomass waste were used and optimized activation conditions with physical or chemical agents afford to obtain materials with high specific surface area, high porous volume and tunable pore size characteristics, and various surface chemistries, highly desired for gas storage applications. For the three gases evaluated herein (CH_4 , CO_2 and H_2), the adsorption capacities on carbon adsorbents show many similar trends. As demonstrated in many works, the adsorption capacity depends firstly on the available porosity, namely the specific surface area and in particularly microporous volume, the pore size with narrow distributions (0.5 -0.8 nm) being likely more favorable for achieving higher gas adsorption up-takes. In addition to the porosity, the carbon functional groups are also influencing the gas adsorption capacity. The oxygen, nitrogen and sulfur were widely studied and their influence on the gas up-take is not the same for all the gases. For CO₂adsorption, oxygen seems beneficial to increase the gas up-take while for hydrogen adsorption oxygen proved to be detrimental. Regarding the other heteroatoms (N, S), their impact was found more difficult to precisely demonstrate due to the complexity to design materials with similar texture and oxygen contents, which usually interferes and accounts also in the adsorption process. The impact of the morphology on the gas adsorption is not deeply addressed and deserves further investigations to elucidate its impacts on the materials density, gas diffusion and volumetric gas capacity. Beside the carbon characteristics that strongly influences the gas adsorption capacity, the analysis condition such as temperature and pressure has significant importance. Due to the week interactions of the gases with the carbon structure, the gas adsorption capacities at room temperature and atmospheric pressure or even high pressure, remain very small to meets the requirements of DEO departments. They can be however strongly improved at low temperature (77K) and high pressure achieving in some cases the capacities required by DEO. Nevertheless, such solution involving low temperature/high pressures remains expensive and not feasible for real life application.

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