

Cellulose phosphorylation comparison and analysis of phosphorate position on cellulose fibers

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1	Cellulose phosphorylation optimization and analysis of phosphorus							
2	position on cellulose fibers							
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32 Highlights

- 33 Cellulose phosphorylation was studied in order to improve the final charge content
- 34 The effect of phosphate salt constant of solubility was studied
- 35 The position and penetration of the phosphate groups on the fibers were evaluated

Abstract. Chemical modifications of cellulose fibers as pretreatment for cellulose nanofibrils 36 (CNF) production have been investigated to improve the production process and the quality of 37 obtained cellulosic nanomaterial. In this study, phosphorylation of cellulose fibers was 38 optimized in anticipation of a future nanofibrillation. Different phosphate salts, namely 39 NH₄H₂PO₄, (NH₄)₂HPO₄, Na₂HPO₄, NaH₂PO₄ and LiH₂PO₄ with different constants of 40 solubility (Ks) were used to increase the efficiency of the modification. Phosphorylated 41 cellulose pulps were analyzed using elemental analysis, solid-state ¹³C and ³¹P NMR, or 42 conductimetric titration method. No effect of Ks was observed whereas a counterion effect 43 was pointed out. The study also reported the effect of pH, cellulose concentration, 44 temperature and urea content in phosphorylation efficiency. 45 Finally, chemical functionalization and penetration of phosphorus in the cellulose fibers were evaluated using 46 47 XPS, SEM-EDX, ToF-SIMS and solid-state NMR.



48 **1. Introduction**

Cellulose, the most abundant polymer on Earth, can be nanofibrillated since 1980. 49 Cellulose nanofibrils (CNF) are described in many books and reviews (Dufresne, 2013; 50 Klemm et al., 2011; Lavoine, Desloges, Dufresne, & Bras, 2012; Rol, Belgacem, Gandini, & 51 52 Bras, 2018). They are sustainable, biodegradable and biocompatible materials. CNF films are transparent, barrier and with high mechanical properties which make them useful in many 53 54 applications such as packaging (Lavoine et al., 2012; Saini, Yücel Falco, Belgacem, & Bras, 2016), paper (Patent No. WO 2014118466 (A1), 2014; Brodin, 2014) or nanocomposites. 55 56 However, the use of CNF is still limited due to the high energy consumption during the production. To overcome this issue, chemical pretreatments of cellulose fibers are used since 57 58 2000's to weaken the hydrogen bonds between fibers, and make the nanofibrillation easier. 59 Moreover, CNF produced via a chemical pretreatment usually present better homogeneity 60 and/or new functionalities as recently reviewed (Rol et al., 2018). TEMPO (2,2,6,6-Tetramethylpiperidine 1-oxyl) oxidized CNF(Isogai, Saito, & Fukuzumi, 2011) are classically 61 62 produced and present higher quality mainly due to their charges which create repulsion and facilitate the nanofibrils liberation. Cationic CNF can also be produced via cellulose fibers 63 64 reaction with EPTMAC (2,3-Epoxypropyl)trimethylammonium chloride). In this case, they display antimicrobial properties thanks to the amine quaternary grafted (Chaker & Boufi, 65 2015; Littunen et al., 2016; Saini et al., 2016). However, the main drawback of those classic 66 67 chemical pretreatments resides in the use of toxic chemicals and complex procedures still 68 difficult to industrialize.

Since 2013, new pretreatments have been developed in order to further increase the quality, 69 with low energy consumption, produce greener CNF and impart new properties. For example, 70 Ghanadpour et al. (Ghanadpour, Carosio, Larsson, & Wågberg, 2015) introduced negative 71 72 charges on cellulose fibers thanks to the phosphorylation. Sulfite softwood pulp was 73 impregnated with ammonium hydrogen phosphate and urea at 1 wt%, filtered to a solid 74 content of 10 wt%, dried at 70°C and then cured at 150°C. Maximal charge of 1,8 mmol/g was measured and CNF of high quality with flame-retardant properties were produced via a 75 76 microfluidizer. Phosphorylation of cellulose fibers for nanofibrillation is a very recent process which can lead to high-quality CNF, easily produced at industrial scale. Today, few 77 78 publications are available. Naderi et al. (Naderi et al., 2016), used similar protocol without filtration step and using NaH₂PO₄. They showed that microfluidization is impossible under a 79 80 degree of substitution of 0.13. Another protocol was proposed in 2017 by Noguchi et al. (Noguchi, Homma, & Matsubara, 2017) and allows reaching a charge content of 2.2 mmol/g. 81

3

Softwood pulp sheets were soaked in solution of $NH_4H_2PO_4$ and urea and then cured in hot air at 165°C for 10 min. CNF of 3-4 nm wide and high transparency suspensions were produced thanks to a homogenizer. The phosphorylation does not lead in change in crystallinity or degree of polymerization (DP) of the cellulose fibers. Such good results have been even transfer to industry with first production announcement in Japan (Yajima, 2018).

Even if phosphorylated CNF have appeared recently, cellulose phosphorylation is an old 87 and well-known process. Phosphorylated cellulose is used in many fields as for example 88 orthopaedics (Petreus et al., 2014), biomedical (Granja et al., 2001; Mucalo, Kato, & 89 90 Yokogawa, 2009), textiles (Davis, Findlay, & Rogers, 1949) or bio-chemical separation (Suflet, Chitanu, & Popa, 2006). Phosphate groups can be introduced to cellulose during 91 homogeneous or heterogeneous reaction. In many papers, cellulose phosphorylation is done 92 on organic solvent such as pyridine (Reid & Mazzeno, 1949; Reid, Mazzeno, & Buras, 1949) 93 94 or dimethylformamide (Katsuura & Mizuno, 1966) using phosphorylating agents such as phosphorus pentoxide, phosphoric acid or phosphate salt (Aoki & Nishio, 2010; Coleman et 95 96 al., 2011; Inagaki, Nakamura, Asai, & Katsuura, 1976; Reid et al., 1949). Microcrystalline cellulose was also phosphorylated in molten urea as solvent using phosphorus acid (Suflet et 97 98 al., 2006). Cellulose can also be phosphorylated using water, urea and phosphoric acid/salts. 99 For example, Gallagher (Gallagher, 1965) soaked cotton fibers in phosphate salt solution 100 before heating at a temperature superior to 100°C for less than 15 min. Ichiwaka et al. (Patent No. JP54138060A, 1979) used almost the same protocol but the curing step was longer (1-3h) 101 102 for obtaining fire retardant materials. Nuessle et al. (Nuessle, Ford, Hall, & Lippert, 1956) phosphorylated cellulose by immerging fibers in a water/urea/diammonium phosphate system 103 104 followed by a curing step at 150°C for 10 min. In many papers, urea is used to enhance fiber swelling and to prevent the degradation of cellulose during the curing step. Phosphorylation 105 106 of cellulose fibers in water based urea system using phosphate salts appears as the easier 107 process and it is the one classically used to produce CNF (Ghanadpour et al., 2015; Naderi et 108 al., 2016; Noguchi et al., 2017). However, only few papers tried to optimize this process and 109 understand the role of salts solubility parameter, their penetration and functionality.

The aim of this work is to optimize the phosphorylation of cellulose fibers in aqueous media to increase the phosphorus content. For that, adsorption of different phosphate salts on cellulose fibers was studied and phosphate salts with different constants of solubility were tested. Position of phosphate group on the anhydroglucose unit and phosphorus penetration on the fibers were also studied using ToF-SIMS, NMR and SEM-EDX.

115 **2. Materials & methods**

116 **2.1 Materials**

Cellulose used throughout this work is eucalyptus bleached kraft pulp Fibria T35. All the 117 materials and chemicals were used as received from the producers: urea (99.5%, Roth), 118 ammonium phosphate monobasic (NH₄H₂PO₄, >99%, Sigma Aldrich), ammonium phosphate 119 dibasic ((NH₄)₂HPO₄, >99%, Sigma Aldrich), lithium phosphate monobasic (LiH₂PO₄, 99%, 120 Sigma Aldrich), sodium phosphate monobasic (NaH₂PO₄, >99%, Sigma Aldrich), sodium 121 phosphate dibasic (Na₂HPO4, 99%, Sigma Aldrich), sodium hydroxide (NaOH, 99%, Sigma 122 123 Aldrich), hydrochloric acid (HCl, 37%, Sigma Aldrich). Distilled water was used for all the 124 preparations.

125 **2.2 Methods**

Refining cellulose. Cellulose fibers were dispersed in water at 2 wt% and refined using a pilot
disk beater (Matech, France) until reaching a Schopper-Riegler degree (°SR) of about 90. At
least three measurements of °SR were done following standard ISO 5267.

129 Phosphorylation of cellulose fibers. Refined cellulose suspensions at different concentration (1 wt%, 2 wt% or 5 wt% solid content) were mixed with different amount of urea and 130 phosphate salts (NH₄H₂PO₄, (NH₄)₂HPO₄, NaH₂PO₄, Na₂HPO₄, LiH₂PO₄) for 30 min. The 131 132 different ratio of AGU/ HPO₄/urea used are 1/1.2/4.9, 1/0.32/1.29, 1/1.6/6.58, 1/3.22/12.88, 1/4.83/19.75 and it corresponds to a $(NH_4)_2$ HPO₄ concentration of respectively 0.0744, 0.02, 133 0.1, 0.2, 0.3 mol/L. After that, suspensions at 1 and 2 wt% were filtered until 10 wt% and 134 dried at 105°C according to the protocol of Ghanadpour et al., 2015). When 135 the test was made with a cellulose concentration of 5 wt%, fiber suspension was not filtered 136 but directly dried at 105°C. Finally, dried fibers were cured at 150°C for 1h in an oven. After 137 the curing step, fibers were soaked in water at 2 wt%, redispersed and washed several times 138 139 with boiling water and water under filtration with filter of 1µm.

Phosphorus content in the filtrate by pH titration. When cellulose suspension (at 1 or 2 wt%) was filtered until 10 wt% before the drying step, the concentration of phosphate groups in the filtrate was measured. 200 mL of the filtrate was titrated with HCl at 0.5M. The phosphate component reacts with acid to form phosphoric acid.

144 The concentration of phosphate salt in the filtrate is evaluated and the quantity of phosphate 145 ion adsorbed in mmol per gram of cellulose is then deduced. At least duplicates were 146 performed.

147 *Total charge content of phosphorylated fibers.* After the curing step and the washing step, the 148 phosphorylated pulp was titrated by conductivity with NaOH. 0.5 g (equivalent dry) of 149 modified cellulose fibers were dispersed in 200 mL of distilled water and the pH was reduced 150 to 3 to convert the hydroxyl groups in their acidic form. The total charge content is measured 151 by titration using NaOH 0.05M. At least triplicates were performed.

Elemental analysis & ICP analysis. Carbon, hydrogen, oxygen and phosphorus contents were measured for the different phosphorylated cellulose pulps using a Vario MICRO cube CHNS-O elemental analyzer (Elementar). Phosphorus total content was measured after micro waves mineralization in presence of acid and followed by inductively coupled plasma (ICP). At least two measurements were done for each sample. The degree of substitution was calculated on the basis of the phosphorus content using the following equation:

158
159

$$DS = \frac{162 * \% P}{n_{P \text{ in grafted mol.}} * M(P) - M(\text{grafted mol.}) * \% P}$$

where %P is the P content determined by ICP, M(P) is the molar mass of phosphore (i.e. 31 g/mol), M(grafted mol.) is the molar mass of the grafted molecule (i.e. 80 g/mol) and $n_{P \text{ in}}$ _{grafted mol.} is the number of P element in the grafted molecule.

163 Degree of polymerization (DP) of cellulose fibers. DP was evaluated using the ISO 5351:2010 164 standard. 250 mg (equivalent dry) of phosphorylated cellulose fibers was dissolved in copper 165 (II) ethylenediamine and intrinsic viscosity ($\eta_{int.}$) was measured at least three times. The 166 degree of polymerization (DP) was calculated using the Mark-Houwink-Sakurada equation 167 and the intrinsic viscosity as follows:

- 168 $DP^{0.905} = 0.75 \eta_{\text{int.}}$
- 169 At least duplicates were performed.

170 *Crystallinity Index (CI).* CI of the modified celluloses was obtained using wide-angle X-ray
 171 diffraction (XRD) spectra. A PANanalytical X'Pert PRO MPD diffractometer, equipped with

172 an X'celerator detector was used and X-rays were generated with a copper anode (K α 173 radiation $\lambda = 1.5419$ Å). The cellulose CI (%) was determined according to the peak height 174 method, where I₀₀₂ represents the diffraction intensity of the main crystalline peak at $2\theta \approx$ 175 22.5° and I_{am} is the intensity at $2\theta \approx 18.7^{\circ}$. All the samples were tested at least in duplicate.

176
$$CI = \frac{(I_{002} - I_{am}) * 100}{I_{002}}$$

177 Optical microscopy images. Cellulose suspensions were diluted to 0.5 wt% and optical 178 microscopy images were taken. A Carl Zeiss Axio Imager M1 (Germany) optical microscope 179 was used in a transmission mode. A minimum of five pictures were taken for each sample and 180 the most representative is shown here.

181 Scanning Electron Microscopy – Energy Dispersive X-ray analysis (SEM-EDX). Scanning 182 electron microscopy equipped with an energy-dispersive X-ray spectroscopy module (LEO 183 Stereoscan 440). Detector Si (Li) EDAX-10 mm² was used to evaluate the position of 184 phosphate groups in the fiber. Tension was reduced at 6 keV and the acquisition time was 185 2h30 in order to increase the resolution.

186 *X-Ray photoelectron spectroscopy (XPS).* Chemical analysis was also established for the 187 modified and unmodified samples. The test was carried out using a K α apparatus modulated 188 with a monochromatic Al K α X-ray source at 14,875 eV. The substrates were positioned at an 189 angle of 90° under an ultrahigh vacuum of less than 10⁻⁷ Pa. Spectrum NT was used to 190 decompose the signal and the overall spectrum was shifted to fix the contribution of <u>C</u>-C in 191 the C_{1s} spectra at 285.0 keV.

192 The O/C ratio was calculated using the following equation:

193
$$\frac{O}{C} = \frac{I_O}{S_O} * \frac{S_C}{I_C}$$

where I_0 and I_c are the intensity of the oxygen and carbon peaks respectively. S_c , S_0 and S_P are equal to 0.00170, 0.00477 and 0.0014 for carbon, oxygen and phosphorus respectively. It characterizes the atomic sensitivity factor. P/C ratio was also calculated using this method.

Nuclear Magnetic Resonance (NMR). Solid-state NMR experiments were performed on
 Bruker Avance III HD 11.75 T spectrometer equipped with a 3.2 mm triple resonance low
 temperature MAS probe and Bruker Avance Neo 21.14 T spectrometer equipped with a 1.3
 mm triple resonance MAS probe in double resonance mode. The ¹³C detected ¹H-¹³C CP, ¹H-

¹³C HETCOR, 1D ¹H CRAMPS and 2D ¹H DQ-CRAMPS experiments for all the samples were carried out on Bruker Avance III HD 11.75 T spectrometer using magic angle spinning (MAS) of 12.5 kHz. The proton detected ¹³C-¹H HETCOR and ¹H DQMAS experiments were performed for all the samples on Bruker Avance Neo 21.14 T spectrometer using MAS rate of 62.5 kHz. The proton and carbon spectra were referenced externally by using L-alanine methyl proton as 1.38 ppm and methyl carbon as 20.5 ppm. All the experiments were performed at room temperature.

Time of Flight-Secondary Ion Mass Spectrometry (ToF-SIMS). This technique makes possible 208 the molecular structures identification at the top surface of materials (information depth 209 limited to a few monolayers i.e. about 2 to 5nm) and was used here to gain more insight on 210 the grafting mode of the phosphorylation at the surface of the fibers. Measurements were 211 carried out on a TRIFT III ToF-SIMS instrument from Physical Electronics operated with a 212 pulsed 22 keV Au⁺ ion gun rastered over a 300 x 300 μ m² area. Data were analyzed using the 213 214 WinCadence software. Mass calibration was performed on hydrocarbon secondary ions. The relative intensity for a given ion was calculated by dividing the intensity of that ion by the 215 216 total integrated intensity minus the H intensity which is barely reproducible.

217 **3. Results & discussions**

3.1 Effect of the filtration step on the final charge content of the modified fibers

The objectives of this part are to check if there is adsorption of phosphate ions and to evaluate the influence of the filtration step on the final charge content.

Cellulose fiber suspensions at 1 wt% were mixed with (NH₄)₂HPO₄ and urea in quantity described in Methods part and then filtered until 10 wt%. The filtrate was pH titrated in order to measure the non-absorbed phosphate groups and so deduced the charge adsorbed.



Figure 1: Phosphorylation of cellulose fibers (a) adsorption of phosphore depending on mass of phosphate salt added and (b) depending on cellulose concentration and phosphate salt

Salt adsorption increases with the concentration of phosphate salt as shown in Figure 1a. However, the mass adsorbed was very low compared to the mass added. Cellulose fibers which are slightly negatively charged (0.06 mmol/g) due to residual hemicelluloses or cellulose oxidation during bleaching constitute heterogeneous surface and are not favorable to adsorb anionic phosphate salt. Phosphate compound is poorly adsorbed on cellulose fibers and the filtration step before the drying step eliminates most of the salt.

Cellulose concentration and the effect of the filtration step were then studied as reported in 230 Figure 1b. The final charge content increases a lot with the cellulose concentration and was 231 really higher when the filtration step was eliminated whatever the phosphate salts. When 232 cellulose pulp was at 2 wt% solid content, phosphate groups were "closer" to the fibers than at 233 234 1 wt% and the adsorption step before the filtration might be improved and so the final charge content. Another explanation is the fibers entanglement. At 2 wt% entanglement was 235 increased and phosphate salts were trapped easily in the fibrous mat during the filtration. Both 236 options result in higher modification rate. 237

When phosphorylation occurs, fiber swells and may even be dissolved as reported by 238 Ghanadpour et al. (Ghanadpour et al., 2015) and confirmed by optical microscopy in Figure 239 S4. We supposed that phosphorylation starts at the fiber surface and then thanks to the 240 241 swelling, phosphate salts can reach the core of the fibers. When phosphate ion was just adsorbed on the surface, charge content was lower. On the contrary, when filtration step was 242 243 eliminated, excess of phosphorus was present and the phosphorylation can go further from the surface to the core. This fact is confirmed by Noguchi et al., 2017) who 244 245 obtained higher charge content using high cellulose concentrated medium and lower phosphate salt concentration (ratio 1/0.63/3.22 compared to a ratio of 1/1.2/ 4.9 with 246 247 filtration). Eliminating the filtration step constitutes a good option to increase the charge content without increasing the chemicals consumption. 248

249 **3.2** Urea role during the phosphorylation of cellulose fibers

Urea has already been used in many protocols to prevent the degradation of cellulose (Ghanadpour et al., 2015; Noguchi et al., 2017; Nuessle et al., 1956; Suflet et al., 2006), to act as a catalyst (Lagier, Zuriaga, Monti, & Olivieri, 1996; Yurkshtovich, Yurkshtovich, Kaputskii, Golub, & Kosterova, 2007) or to enhance the fibers swelling and hence increase the phosphate groups penetration. Urea effect was investigated as reported in Table 1.

Cellulose	Molar ratio AGU/	Charge content		Presence of
concentration	(NH ₄) ₂ HPO ₄ / Urea	[mmol/g]	Centriose color	aggregate
2	1/0/0	0.06 ± 0.01	White	No
2	1/0/4.9	0.17 ± 0.03	White	No
2	1/1.2/0	0.37 ± 0.01	Black	Yes
2	1/1.2/4.9	1.18 ± 0.06	Brown	Yes
2	1/1.2/7.2	2.37 ± 0.05	Yellow	Yes

Without urea, the phosphorylation only slightly occurs with a charge content of 0.37 256 257 mmol/g whereas the cellulose became totally black. It is now clear that urea is necessary to phosphorylated cellulose but also to prevent cellulose degradation. One reactional mechanism 258 259 adapted from Kokol et al.(Kokol, Božič, Vogrinčič, & Mathew, 2015) is presented in Figure 2 with several phosphorylated cellulose structure possibilities. Urea has an active role in the 260 261 reaction but also in the protection of the cellulose. Presence of nitrogen basic elements in the 262 phosphorylation process seems primordial (Katsuura & Mizuno, 1966; Nuessle et al., 1956; 263 Reid & Mazzeno, 1949; Reid et al., 1949).

When urea content increased, charge content increased and the coloration of cellulose decreased. Yurkshtovich et al.(Yurkshtovich et al., 2007) showed that phosphorus content increased with the concentration of urea up to 3.33M (i.e. 4 times higher than what we used here). However side reaction can occur and creates carbamate. This is not a problem from a nanofibrillation point of view, because it also weakens the H-bonds between cellulose nanofibrils.

270 **3.3 Influence of phosphate salt solubility and pH on the final charge content**

Up to our knowledge, no study dedicated to phosphorylated CNF has compared various phosphate salts. In this study, higher modification rates were expected with an higher constant of solubility (Ks). Unfortunately, no relationship between the degree of substitution and the Ks was observed even if various phosphorylation rates are clearly obtained, depending on the phosphate salt Ks as reported on Table 2. However, if the Ks is very low (for example Na₂HPO₄), the grafting was very low. Indeed, the counterion was closer due to lower dissociation and there was some hindrance and less reactivity.

For a same counterion, the total charge content increased if the Ks increased, meaning that counterion have more impact than Ks. In each case, the counterion ammonium (NH_4^+) led to

higher DS than lithium (Li⁺) and than sodium (Na⁺). For now, counterion effect in the 280 chemical reaction is still under discussion (Manet, 2007). Some studies reported that in 281 concentrated sodium hydroxide media, NH₄⁺ have a lower affinity with hydroxyl groups and 282 will not surround the anhydroglucose units. Moreover, the ammonium counterion has a lower 283 affinity for water than sodium ion. The more hydrophobic character of NH₄⁺ would produce 284 more pronounced swelling of the cellulose as the non-activated anhydroglucose unit will be 285 repelled (Ott, Spurlin, Grafflin, Bikales, & Segal, 1954). In water, lithium ion is more 286 hydrated than sodium followed by ammonium. Counterion electronegativity can also 287 influence the reaction. Lithium and sodium with an electronegativity of respectively 0.98 and 288 0.93 are much less electronegative than nitrogen (3.04) and so the quaternary ammonium. The 289 distance between NH_4^+ and HPO_4^{2-} is longer than the one between Li⁺ or Na⁺ and HPO_4^{2-} due 290 to lower difference of electronegativity. NH4⁺ is more distant of HPO4²⁻ than the other 291 counterion and so salts are less stable (or more reactive) which leads to higher grafting. 292

Table 2: Influence of phosphate salt constant of solubility (Ks) and pH, on cellulose
 phosphorylation using AGU/Phosphate salt/ urea ratio of 1/1.2/4.9 and cellulose fiber
 concentration of 5 wt% without filtration. *pH was modified by adding NaOH or HCl

Dhosphoto solt	Ka	Cellulose	пЦ	Charge content	Equivalent	DS based on P
r nospitate sait	KS	concentration [%]	рп	[mmol/g]	DS	content [AE]
LiH ₂ PO ₄	126	5	9*	2.15 ± 0.31	0.35	-
LiH ₂ PO ₄	126	5	5	2.44 ± 0.07	0.39	0.18
NaH ₂ PO ₄	94.9	5	10*	0.48 ± 0.12	0.08	-
NaH ₂ PO ₄	94.9	5	4	1.48 ± 0.12	0.24	0.29
(NH ₄) ₂ HPO ₄	68.9	5	8	2.93 ± 0.14	0.47	0.45
(NH ₄) ₂ HPO ₄	68.9	5	4*	4.40 ± 0.50	0.71	-
NH ₄ H ₂ PO ₄	37.4	5	4	2.21 ± 0.10	0.36	-
Na ₂ HPO4	12	5	8	0.41 ±0.01	0.07	-

The pH effect was also investigated. For every phosphate salt, the reactivity and the final charge content was higher when acidic medium was used as confirmed by Lim and Seib on wheat starch (Lim & Seib, 1993). For example, charge content passes from 0.5 to 1.48 mmol/g for NaH₂PO₄ or from 2.93 to 4.40 for (NH₄)₂HPO₄. Indeed at acidic pH, two OH groups are available on the phosphate salt and the reaction is improved. Moreover, in acidic medium, negative charges of cellulose are reduced and anionic phosphates are not repelled.

302 **3.4 Chemical structure of the grafted cellulose fibers**

As shown in the Figure 2, different cellulose structures can be obtained after the 303 phosphorylation. Hydroxyl groups of cellulose can react with phosphate salt and form 304 phosphate groups (Cell-O-P(O)(OH)₂), phosphite groups (Cell-O-P(OH)₂) or combination. 305 Crosslinking can also occur (Noguchi et al., 2017). Some aggregates are formed during the 306 curing step and are really difficult to destroy and might correspond to crosslinked structures. 307 Moreover, position of phosphate groups on the cellulose units and its penetration on the fiber 308 is not so evident. Different methods were investigated to determine which form is the 309 310 preferential one and how fibers are modified.



Figure 2: Different grafting modes for phosphorylated cellulose as proposed by Kokol et
al.(Kokol et al., 2015)

313 **3.4.1 Conductivity titration**

The first method that might give information about the cellulose structure is the 314 conductimetric titration (Figure S1). The degree of substitution measured by elemental 315 analysis is equal to the one obtained by conductimetric titration as reported in Table 2 and 316 confirmed by Naderi et al., 2016). During the titration only one pH jump is 317 318 observed meaning that only one hydroxyl group is titrated. This can suppose that the phosphate group form is the one in Figure 2a, (second form) or that there is crosslinking like 319 in scheme Figure 2c. However Kokol et al. (Kokol et al., 2015) reported pKa of 6.8 and 12.5 320 for (Cell-O-PO₃H₂/ Cell-O-PO₃H⁻) and cell-CH₂-O-PO₃^{2-.} Most of the time the titration was 321 done between pH 2 and 11 meaning that the second hydroxyl groups of the phosphate group 322 is not measured which means that all the form presented in Figure 2 could be obtained. 323

However, even when the pH was increased until 13 for some trials, only one pH jump was reported.

326 3.4.2 Solid-State NMR

327 High-resolution solid-state NMR at 11.74 Tesla

Native and phosphorylated cellulose fibers were analyzed at 11.74 T (500 MHz for ¹H 328 resonance) using ¹³C, ³¹P and ¹H NMR analyses as reported in Figures 3 and 4 to determine 329 where the phosphorus is located. The ¹³C NMR spectrum is shown in Figure 3a. The 330 respective spectra of cellulose and phosphorylated cellulose are consistent with the data 331 reported in the literature. The peak in the 58-68 ppm region corresponds to the C6 of the 332 cellulose ring, in 68-80 ppm region to C2, C3 and C5 carbon, from 80 to 91 ppm to the C4 333 and finally from 101 to 109 ppm to the C1 (Atalla & Vanderhart, 1984). Note that signals at 334 84 ppm and 63 ppm are respectively assigned to C4 and C6 of amorphous domains of 335 336 celluloses (Atalla & VanderHart, 1999; Montanari, Roumani, Heux, & Vignon, 2005). As their spectra appear to be very alike, it indicates that the carbon structure of the 337 phosphorylated cellulose fibers with a charge content of 2.10 mmol/g might be not 338 significantly modified compared to the carbon structure of the pristine cellulose. Ghanadpour 339 340 et al.(Ghanadpour et al., 2015) also reported that all the regions of modified fibers are slightly changed and thus no region selectivity is observed. In our case, it could indicate that the 341 phosphate grafting was homogenously distributed on the cellulose fibers which do not lead to 342 specific changes in the ¹³C chemical shift. 343

The presence of phosphorus on the cellulose fibers after a strong washing step is confirmed by an intense signal in the HPDEC ³¹P NMR spectra as shown in Figure 3b. The phosphorus components appeared in the region -10 to 10 ppm it is mainly made of an intense signal at 0 ppm overlapping a broader low intensity signal which spreads down -15 ppm.



Figure 3: Influence of the phosphorylation on the carbon structure of the cellulose: (a) ¹³C solid-NMR spectra and (b) HPDEC ³¹P

At a 11.7 T field high-resolution solid-state ¹H NMR spectra of both pristine and 350 phosphorylated cellulose were obtained with DUMBO sequence and reported in Figure 4a. 351 352 Very interestingly, contrarily to their carbons counterparts the proton spectra present some significant differences. We can observe a low shift of the maximum intensity from pristine to 353 grafted celluloses. Moreover, the spectra of the grafted cellulose present two large and intense 354 shoulders between 5 and 7 ppm. Similar signals are also observed in the pristine cellulose but 355 356 much less intense. It means that new type of protons appeared during the phosphorylation. Those new protons can be due to either (i) hydroxyl protons at the phosphate groups grafted 357 358 on cellulose or (ii) modifications of the hydrogen bond network. This latter statement can be supported by the number of oxygens, brought by the phosphorylation, that are good candidate 359 to modify the hydrogen bond network. As protons resonances are shifted to the high 360 361 frequencies we can infer that the amount of proton involved in hydrogen bond network is increased. 362

363 To have more information concerning the proton network of the phosphorylated and 364 pristine celluloses, dipolar proton-carbon correlation experiments were carried out and 365 reported in Figure 4b.

The 2D maps (normalized related to the intensity of signals of crystalline C4) look very similar, however some differences can be noticed. In Figure 4b, showing the dipolar interactions between C and H, the proton resonances of the phosphorylated cellulose are shifted to the higher frequency except the protons in interaction with C4 of crystalline domains. Due to the normalization, it appears that the amount of C4 in amorphous domain is increased (larger correlation). In a similar way the correlation assigned to C2,3,5 appears to be broader in the phosphorylated cellulose that in the pristine one. The correlation at C1 is not significantly affected.

On the base of the expected chemistry of phosphorylation, grafting should occur mainly at C6. Therefore the broadening that we mentioned previously of C2,3,5 correlations at around be assigned not to chemical shift changes of theses carbons but mainly to the effect of substitution on C6 which leads to a low field effect and consequently to the observed signal broadening of C2,C3,C5 correlation. Moreover grafting mainly appears on amorphous domain (increase of the amorphous C4).



Figure 4: Solid-NMR spectra. Effect of the phosphorylation on the proton network of cellulose fibers. (a) Hetcor experiment and (b) DUMBO experiment

382 High-resolution solid-state NMR at 21.1 Tesla

The spectra recorded at 21.1 T with MAS rate of 62.5 KHz bring some complementary 383 information. On an experimental point of view, to the fact that very high spinning rates are 384 used, some consequences are expected. For ¹H NMR, spectra can be acquired either with 385 direct excitation or with spin echo experiment. For the dipolar ¹H-¹³C correlation experiments 386 (proton detected at this field), it has to be mentioned that due to the high spinning rate of the 387 388 samples the molecular domains with small proton carbon dipolar interactions can be difficult to detect. In that case the experiment enhances the crystalline parts of pristine or 389 390 phosphorylated celluloses. Taking into account these observations the following conclusions can be drawn. 391

The ¹H NMR recorded at 21.1 T and shown in Figure 5,a confirms the results obtained at 11.7 T, namely the presence of new signals at 5 and 7 ppm. Note that for the pristine cellulose the signal at 4.2 ppm is assigned to water molecules which can be identified by adding small quantity of water to the sample and which are very sensitive to the temperature.



Figure 5: (a)¹H NMR recorded at 21.1T for pristine cellulose and phosphorylated cellulose;
(b) and (c) Proton-carbon correlation recorded at 21.1 T. Influence of the phosphorylation on
the cellulose structure

Contrary to the 2D ¹H-¹³C correlation experiments recorded at 11.7 T the 2D spectra recorded at 21.1 T are very similar, as shown in Figure 5b,c. It has to be reminding that in that case we mainly see the crystalline domain of the materials. However the C1 are correlated to protons that appear to be slightly shifted at higher frequency in the phosphorylated cellulose compared to pristine.

404 At such field, due to the fast spinning rate and the low amount of material that are analyzed 405 in the order of a few mgs, direct detection of 13 C NMR is impossible within reasonable 406 experiment time. The only way to obtain an equivalent 1D spectrum is to calculate the sum of 407 the columns (carbon in the indirect dimension). The calculated spectra are shown in Figure 6 408 and contrary to the 13 C spectrum acquired at 11.7 T the spectra present some differences. The feature of the calculated spectrum of phosphorylated cellulose appears to be more crystalline than the pristine one. It is a strong indication that the phosphorylation of cellulose fibers lead to amorphous domains which are not seen in the 2D experiments recorded under these conditions, i.e., 21.1 T and 62500 Hz of spinning rate.



Figure 6: Reconstituted ¹³C NMR 1D spectrum of pristine cellulose and phosphorylated
 cellulose

To conclude, the NMR analysis cannot help to determine which exact structure of the phosphorylated cellulose fiber is formed insofar as the carbon structure is not affected and the proton network is entirely modified. However, the phosphorylation is clearly visible regarding the ³¹P spectrum or with the change in the proton network.

419 **3.4.3** Time of Flight-Secondary Ion Mass Spectrometry

ToF-SIMS analyses were performed at the surface the phosphorylated cellulose fibers to try to determine the preferential grafting mode of phosphorylation at the extreme surface of cellulose fibers. Results are displayed in Figures 7 and S2 and relative intensities of specific signatures are given in Figure S3.

The phosphate salt, $(NH_4)_2HPO_4$, used for the phosphorylation of cellulose was analyzed in its original form and several characteristics peaks were found: in the negative mode, PO_2^- (m/z = 62.96), PO_3^- (m/z = 78.96) and $H_2PO_4^-$ (m/z = 96.97) which are the typical peaks detected for phosphorylated compounds but also $HP_2O_6^-$ (m/z = 158.93), $H_3P_2O_7^-$ (m/z = 176.93) and $H_5P_2O_8^-$ (m/z = 194.95) and in the positive mode, $NH_3H_3PO_4^+$ (m/z = 116.01). The peaks at m/z = 120.96 and 142.95 in the positive mode correspond to NaH₃PO₄⁺ and Na₂H₂PO₄⁺, respectively, and indicate a cationization effect related to Na (a significant intensity is also detected for Na⁺). The detection of HP₂O₆⁻ (m/z = 158.93), H₃P₂O₇⁻ (m/z = 176.93) and H₅P₂O₈⁻ (m/z = 194.95) indicates a dimer alike based emission.

In the spectra of phosphorylated fibers, there was no detection of high mass molecular
peaks combining cellulose and the grafted moieties. On the other hand, no new phosphorusbased secondary ions were detected.

In the positive mode spectra of phosphorylated fibers, the peak at m/z = 116.01corresponding to NH₃H₃PO₄⁺ was not detected, which can confirm the grafting as well as an efficient washing step. Furthermore, urea displays a characteristic peak around at m/z = 61.03, as shown in Figure S2, and this peak is not detected on the phosphorylated fibers meaning that the washing step was efficient.

The spectra of phosphorylated fibers in the negative mode show some signatures of the reference powder. PO_2^- (m/z = 62.96), PO_3^- (m/z = 78.96) and $H_2PO_4^-$ (m/z = 96.97) are detected in the modified cellulose fibers but their relative intensities are different for the various samples (see Figure 7 and S3). Hence, a detailed study of the relative intensity of the above described secondary ions was undertaken.

The relative intensities of PO₂, PO₃ and H₂PO₄ are similar for the phosphorylated pulps 446 with charge contents of 2.93 and 1.18 mmol/g (taking into account standard deviations) but 447 448 they are significantly different for the charge content equal to 0.54 mmol/g. This indicates a different extent in the way of grafting, observed as a function of the different quantity of 449 phosphate salt added in the suspension. On the other hand, intensities are also different in the 450 spectra of the modified celluloses and in the spectra of the reference powder, which could 451 then help understanding what is the probable referred grafting mode among those displayed in 452 453 Figure 2. One must still be careful as no reference sample of the proposed grafting modes could be available for ToF-SIMS analysis. 454



Figure 7: ToF-SIMS spectra of phosphate salt powder, pristine cellulose and phosphorylated
 cellulose fibers in the positive in the 100-150 m/z range and in the negative modes in the 50 100 and 150-200 m/z range

458 PO_3^- (m/z = 78.96) and H₂PO₄⁻ (m/z = 96.97) are the most preponderant secondary ions for 459 the modified fibers and their normalized intensity (according to PO₂ intensity) decreased 460 when grafting rate decreases. This can confirm that most of the phosphorylated fibers are in 461 the form of Figure 2a.

Moreover, the peaks at HP₂O₆⁻ (m/z = 158.93), H₃P₂O₇⁻ (m/z = 176.93) are detected in the phosphorylated cellulose fibers (the peak at m/z = 194.95 m/z exhibits a very low relative intensity) but their relative intensities are really different, even between the samples with charge contents of 2.93 and 1.18 mmol/g. The presence of this species increases with the charge content from 0.54 to 2.93 mmol/g whereas the intensities of PO₂, PO₃ and H₂PO₄ increase for a charge content from 0.5 to 1.1 and then tend to stabilize.

Those species can be present due (i) to the grafting of the impurities of the phosphate salt powder on the fibers, (ii) due to the grafting of two HPO_4^{2-} ions on the cellulose, (iii) due to the grafting of this species present in the powder on the cellulose fibers or due to (iv) possible recombination during the analysis.

Their intensities in the functionalized fibers are different and this can be explained by 472 different reasons. First, if HPO4²⁻ is grafted on the fibers, the combinations are less possible 473 and those species cannot be formed. HPO_4^{2-} is not free and cannot make combination. 474 However, in this case, if all the HPO₄²⁻ were grafted, the peaks at 158.93 and 176.93 m/z 475 would have totally disappeared. The presence of these two peaks at 158.93 and 176.93 m/z 476 can also relate the grafting of those species present in the powder or the combination of two 477 478 HPO₄²⁻ and formation of di or tri ester as reported in Figure 2b. The formation of the cellulose 479 structure described in Figure 2c should lead to new peaks and none are visible.

480 To conclude, no characteristic peak of the phosphorylated cellulose has been clearly 481 identified by ToF-SIMS analysis but some hypothesis of grafting was determined. It seems 482 that the phosphorus is present on the cellulose fiber with the $H_2PO_4^-$ form and in a fewer 483 extent HP₂O₆. The presence of crosslinked cellulose structures has been discarded insofar as 484 no signal was detected. This technique analyzes the extreme surface whereas some phosphate 485 salt can penetrate. That is why an intermediate surface technique (XPS) was performed to486 carry on investigations.

487 **3.4.4 X-ray photoelectron spectroscopy**

The phosphorylated pulp with a charge content of 2.10 mmol/g and the pristine cellulose 488 were analyzed as reported in Figure S4. Total spectra of native and phosphorylated cellulose 489 490 are presented in Figure S4a. Distinct peaks of C_{1s} and O_{1s} are present on the native cellulose 491 as well as minor peaks characteristics of surface contamination. Additional peaks are present 492 on the phosphorylated cellulose. The peaks characteristics of the phosphorus appeared at 493 134.4 eV and 190.5 eV and are assigned to P_{2p} and P_{2s} (Božič, Liu, Mathew, & Kokol, 2014; 494 Ghanadpour et al., 2015; Pasqui, Rossi, Di Cintio, & Barbucci, 2007), confirming the 495 phosphorylation. The peak at 401.4 eV corresponds to the presence of nitrogen (Bourbigot, Le 496 Bras, Gengembre, & Delobel, 1994) N_{1s} which is probably assigned to NH_4^+ groups of the phosphate salt. 497

498 The scan spectra of C_{1s} of the native cellulose and the phosphorylated cellulose are 499 presented in Figure S4c and d. The spectra are decomposed in 4 peaks characteristics of the 500 cellulose, namely, C1(C-C), C2 (C-O), C3 (C=O) and C4 (O-C-O) at respectively, 285.0, 501 286.6, 287.8, 289.2 eV. The two spectra present high differences especially regarding the C1 502 and C2 peaks. Indeed, the grafting of phosphate groups can change the chemical environment of the carbon and thus increases the C-C peak while decreasing the C-O peak (Božič et al., 503 504 2014). The intensity of C-C peak increases and this can be due to the presence of phosphate groups covering all of the outer surface of fibers (Ghanadpour et al., 2015; Li, Wang, Liu, 505 506 Xiong, & Liu, 2012).

507 The relative surface concentration of C, O and P atoms were calculated by integrating the 508 peaks of each element and results are presented in Table S1. Unfortunately after 509 phosphorylation the O/C ratio does not increase whereas a lot of oxygen is supposed to be 510 added on the cellulose with the phosphorus grafting. Indeed, one free hydroxyl groups of the 511 pristine cellulose is substituted by two new hydroxyl groups and one phosphoryl group (Božič 512 et al., 2014; Granja et al., 2001).

Regarding the O_{1s} scan of phosphorylated cellulose, in Figure S4f, a very small asymmetry is detected and the peak can be decomposed in two peaks: 532.7 eV for O1(C-<u>O</u>) and 531.5 eV for O2 (P-<u>O</u>) which confirms once again the phosphorylation (Bourbigot et al., 1994). The 516 DS of the phosphate groups on the cellulose fibers is estimated as follow according to Li et 517 al.(Li et al., 2012). The obtained value is reported in Table S1.

518
$$DS = \frac{O2}{O1 + O2}$$

The tested phosphorylated pulp presents a charge content of 2.10 mmol/g which 519 corresponds to a DS of 0.32. Here, the obtained DS is three times lower as reported in Table 520 S1. Indeed, XPS measured the surface grafting whereas the titration measure the bulk 521 grafting. The difference can be explained by different points: (i) some phosphorus can be 522 523 embedded in the fibers network and unseen by XPS, however, we considered that fibers are well washed, (ii) the fibers washing is more efficient in surface than in the inner structure of 524 525 the cellulose which results in lower grafting, (iii) the grafting is heterogeneous and the analysis was done on a poorly grafted part of the fibers surface. 526

527 To conclude, phosphorus was detected by XPS analysis but no information on the 528 phosphorylated cellulose structure is obtained using this technique.

529 **3.4.5 SEM EDX**

The phosphate group position on the fiber was then evaluated using SEM-EDX on a sectional view of phosphorylated cellulose paper with a charge content of 2.1 mmol/g as reported in Figure 8. Phosphorus is present in huge quantity and covered entirely the sample. Even if the image scale is important compared to the fiber diameters some fibers are visible in sectional view and framed in red. It seems that phosphate groups penetrate the fiber diameter and are homogeneously distributed.



Figure 8: Evaluation of phosphorus position on the cellulose fibers (2.93 mmol/g) using EDX
images with (a) MEB images and (b) coloration of phosphorus group in green

538 **3.5 Influence of the phosphorylation step on cellulose fibers**

539 Fiber degradation traditionally occurs during cellulose chemical pretreatment. Cellulose 540 degradation due to the phosphorylation is evaluated measuring the degree of polymerization 541 and the crystallinity index as reported in Figure S5.

542 Partial swelling of fibers was observed when the charge content increased as well as a small 543 decrease in the crystallinity index (CI). Cellulose phosphorylation induces a fiber swelling 544 which can lead to some dissolution. A ballooning effect is visible for the sample containing 545 2.9 mmol/g of phosphate group but fibers morphologies are still maintained.

However, despite the high charge content of 2.9 mmol/g, cellulose I form was maintained and suitable CI was obtained. Noguchi et al.(Noguchi et al., 2017) reported no crystallinity change when the charge content varied from 0 to 2.2 mmol/g. Increasing the charge content will drastically improve the nanofibrillation step without affecting too much the cellulose fibers structure and morphology. This is confirmed by the absence of modification of the carbon spectra in NMR analysis.

552 The degree of polymerization (DP) of the phosphorylated cellulose pulp with a charge 553 content of 2.1 mmol/g was measured. The phosphorylated pulp presents a DP of 628 ± 5 554 whereas the refined cellulose pulp presents a DP of 1,200. Phosphorylation as other chemical 555 pretreatments used to weaken cellulose fibers, decreases the DP of the fibers. Rol at al.(Rol et 556 al., 2017) reported a DP decrease of 50 and 75% for respectively enzymatic pretreatment and TEMPO-oxidation of cellulose fiber. As observed in Figure S5 phosphorylation can lead to 557 558 fiber partial swelling which can result in some dissolution and thus induce a DP decrease. 559 Plus, during the phosphorylation, acid can be released and lead to DP decrease.

560 To conclude, phosphorylation leads to some fiber DP decreases and can lead to some fibers 561 swelling which will further help the nanofibrillation but preserves the cellulose crystalline 562 structure.

563 4 Conclusion

The phosphorylation of cellulose fibers was studied and optimized to increase the final charge content. Different phosphate salts were tested and no effect of the Ks was reported whereas the presence of ammonium counterion improves the modification rate. A poor adsorption of phosphate ions on the cellulose fibers was reported and the filtration step should be eliminated to strongly increase the charge content. However, the grafting should be limited to avoid fiber dissolution. The grafted molecule position in the cellulose fibers and the chemical structure of the phosphorylated fibers were investigated using several high level

- 571 techniques such as NMR, XPS, ToF-SIMS or SEM EDX. NMR analysis confirms that the 572 phosphorylation impacted all the protons network of the cellulose and that the reaction mainly 573 occurs in amorphous area of the cellulose fibers. Finally, it seems that phosphate groups do
- 574 not affect significantly the physical properties of cellulose fibers.

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