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Fe-facilitated dynamic active site generation on spinel CoAl_2O_4 with self-termination of surface reconstruction for water oxidation

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Abstract

The development of efficient and low-cost electrocatalysts for oxygen evolution reaction is critical for improving the water electrolysis efficiency. Here we report a strategy using Fe substitution to enable the inactive spinel CoAl_2O_4 to become highly active and superior to the benchmark IrO_2 . The Fe substitution is revealed to facilitate the surface reconstruction into active Co oxyhydroxides under OER conditions. It also activates the deprotonation on the reconstructed oxyhydroxide to induce negatively charged oxygen as active site, thus significantly enhancing the OER activity of CoAl_2O_4 . Furthermore, it promotes the pre-oxidation of Co and introduces great structural flexibility due to the uplift of the O 2p levels. This results in an accumulation of surface oxygen vacancy along with lattice oxygen oxidation that terminates as Al^{3+} leaches, preventing further reconstruction. We showcase a promising way to achieve tunable electrochemical reconstruction by optimizing the electronic structure for low-cost and robust spinel oxide OER catalysts.

1 Hydrogen has been proposed for long as an energy carrier for a sustainable and clean energy
2 infrastructure. However, such a hydrogen energy infrastructure has not been realized yet even so
3 many years have passed since its first proposal. One of the major reasons is the low efficiency and
4 high material cost of water electrolysis, which is a mean to sustainably produce hydrogen fuel from
5 water using the electrical energy generated by sustainable resources like solar panels.¹ The low
6 energy efficiency for such system is mainly caused by the sluggish reaction kinetics at the anode side
7 of water electrolyzer,² i.e. where water is oxidized and the oxygen evolution reaction (OER) occurs.
8 The benchmark anode electrocatalysts are noble metal-based oxides, such as IrO₂ and RuO₂,³⁻⁴ which,
9 however, aggravates the cost problem for water electrolysis. In recent years, great efforts have been
10 made to explore first-row 3d transition metal oxides as low-cost alternatives for OER.^{1,5} Toward that
11 goal, one important progress was made in identifying the active sites for some Co-containing oxides
12 as oxyhydroxides generated under operando condition.⁶⁻¹⁰ Specifically, many Co-based oxides were
13 reported to undergo surface self-reconstruction of Co sites to Co (III) oxyhydroxides, offering a
14 higher activity^{6, 9-10} arising from the di-μ-oxo bridged Co-Co sites.¹¹ However, despite revealing the
15 critical role of the surface reconstruction into oxyhydroxides, such reconstruction sometimes
16 compromises a stable OER catalysis due to instable surface chemistry.¹² In addition, how to properly
17 facilitate the surface reconstruction is still elusive.

18 Here, we report an approach to promote the surface reconstruction on inactive but low-cost CoAl₂O₄
19 (see Supplementary Note 1 for more about CoAl₂O₄) and to boost its OER performance by
20 substituting Al with a small amount of Fe. The partially substituted CoFe_{0.25}Al_{1.75}O₄ outperforms
21 IrO₂. We demonstrate here that a low level of Fe substitution is able to facilitate the surface
22 reconstruction of CoAl₂O₄ by activating the pre-oxidation of Co and optimizing the O 2p level of

oxide for greater structural flexibility. More importantly, a distinctive reconstruction behavior with self-termination has been revealed on $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, which enables a stable surface chemistry. In addition, we suggest that, on the reconstructed surface, the Fe substitution facilitates a two-step deprotonation process, which leads to the formation of active oxygen sites at a low overpotential and thus greatly promotes the OER. Finally, the electrolysis application of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ as OER catalyst was demonstrated in membrane electrode assembly (MEA) configuration.

Results

Crystal structure characterization

The $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0\sim 2.0$) oxides were synthesized by a sol-gel method. The crystal structures of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ oxides were characterized by powder X-ray diffraction (XRD). As displayed in Fig. 1a, the diffraction peaks of the as-prepared CoAl_2O_4 and CoFe_2O_4 match well with that of the standard cubic spinel (Fd-3m) oxides. $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ oxides with different Fe substitution amount remain in cubic spinel structure. Besides, the diffraction peak exhibits certain shift to lower angle with the increase of Fe substitution amount in the range of $0.25 < x < 2$ (Supplementary Fig. 1). Such peak shift could be ascribed to changes in lattice parameters induced by different ionic radius of Fe and Al cations and suggests a solid solution property. However, when the Co local atomic structure was investigated by EXAFS (Fig. 1b, Supplementary Fig. 2), certain isobestic points, such as at 4.5 \AA^{-1} , are remarkable, implying a complex property with Co in different components.²⁶ As further observed in Fourier transform (FT) Co K-edge EXAFS (Fig. 1c), two separated peaks, i.e. Peak II and Peak III, at $2.4\sim 3.1 \text{ \AA}$ are assigned to the features of Co_{Oh} (in octahedral site) and Co_{Td} (in tetrahedral site). At $x \leq 0.25$, no obvious changes are observed for peak II and peak III compared to pristine

44 CoAl_2O_4 , indicating that low-level Fe substitution does not change the occupation of Co. The obvious uplift of
45 these two peaks starts from $x=0.5$, which could be attributed to the segregation of a component that comprises
46 more octahedrally coordinated Co. This segregated component is inferred as CoFe_2O_4 which is in inversed
47 spinel structure where Co occupy the octahedral site.²⁷ Such inference is further evidenced by a Co K-edge
48 XANES linear combination fitting (LCF) with CoAl_2O_4 , CoFe_2O_4 and $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ as standard
49 (Supplementary Fig. 3). The fitting delivers an extremely low R factor, indicating that Co is primarily in
50 CoAl_2O_4 , CoFe_2O_4 and $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ components throughout the oxides.

51

52 In addition to the complex behavior of Fe-containing components throughout the oxides, the Fe substitution in
53 CoAl_2O_4 also alters the local atomic structure of Co. As observed in Co K-edge EXAFS (Fig. 1c), the
54 first-shell peak at ~ 1.5 Å, representative of the metal-oxygen bond, is weakened by Fe substitution, suggesting
55 lower metal-oxygen coordination number and more oxygen vacancy²⁸. Specifically, the EXAFS fitting result
56 (Supplementary Table 1) of the first-shell peak for $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ and CoAl_2O_4 reveals that Fe substitution
57 reduced the average coordination number by ~ 0.4 at maximum. Such effect, indicating an increase in oxygen
58 vacancy, is consistent with a decrease in cobalt valence state after Fe substitution which is evidenced by the
59 shift to lower energy of the Co K-edge XANES (Fig. 1d). The nominal Co valence states (Supplementary
60 Table 2) are indicated by the K-edge positions when compared to references CoAl_2O_4 and Co_3O_4 (see details
61 in SI). While Fe^{3+} is more electronegative than Al^{3+} , after Fe substitution, the K-edge energies of Co slightly
62 shift to lower absorption energy compared to CoAl_2O_4 , indicating a decrease in Co valence state. This
63 counterintuitive phenomenon could be rationalized by the creation of oxygen vacancies,⁵ which is also
64 supported by theoretical calculation (Supplementary Table 3). Indeed, the Fe substitution in CoAl_2O_4 was
65 found to reduce the formation enthalpy of oxygen vacancy, which indicates easier formation for oxygen

66 vacancy.

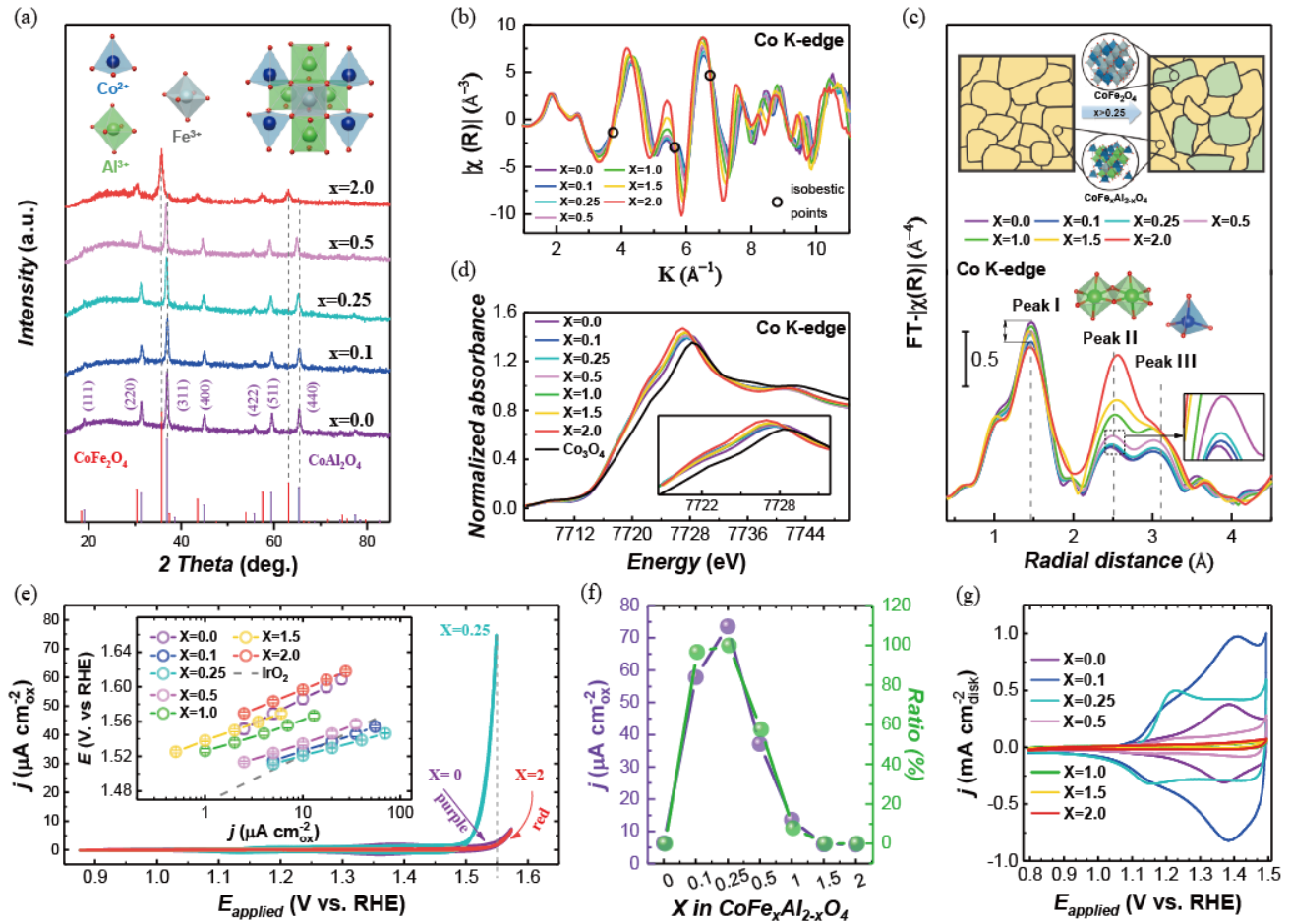


Fig. 1 | Structural characterizations and OER performances of as-prepared $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ catalysts. **a**, The powder X-ray diffraction (XRD) patterns of as synthesized $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0.0\sim 2.0$) oxides. **b**, The Co K-edge EXAFS spectra of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0.1\sim 2.0$). **c**, The Fourier transform (FT) $k^3\chi(R)$ Co K-edge EXAFS of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0.1\sim 2.0$). The inset is schematics of the compositions in the $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ oxide. **d**, The normalized Co K-edge XANES spectra of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ samples and the standard Co_3O_4 (Sigma Aldrich) as the references. The K-edge position is determined by an integral method²⁹ as described in Methods, and the details about edge positions and nominal valence state of Co and Fe are shown in Supplementary Table 2. **e**, The cyclic voltammetry curves of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0, 0.25$, and 2) in O_2 saturated 1 M KOH with a scan rate of 10 mV s^{-1} . The inset is the corresponding Tafel plots after oxide surface area normalization, capacitance correction, and iR correction. The error bars represent the standard deviation from three independent measurements. The grey dashed line shows the best IrO_2 OER performance reported to date.⁴ **f**, The OER current densities (left axis) of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ at 1.55 V vs. RHE . The composition ratio of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ component (right axis) in $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ oxides is plotted to show its correlation with the OER activity. **g**, The cyclic voltammetry curves of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x = 0\sim 2$) in O_2 saturated 1 M KOH with a scan rate of 10 mV s^{-1} between 0.8 and 1.5 V vs RHE . The upper limit of the potential window is capped to show the pseudocapacitive behavior preceding to OER.

86 OER activity

87 The spinel $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ oxides were then measured for their electrocatalytic OER performances under
88 alkaline condition (see details in Methods). Supplementary Fig. 4 shows their steady OER cyclic voltammetry
89 (CV) curves. The current density was normalized to the surface area of oxides to present the intrinsic activity,
90 with the surface area of oxides being determined by Brunauer-Emmett-Teller (BET) measurements
91 (Supplementary Fig. 5 and Supplementary Table 4). Ohmic drop iR correction was applied to compensate the
92 potential loss from the resistance of the electrolyte solution. As observed in Supplementary Fig. 4, among these
93 spinel oxides, $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ is the best-performing one. Comparable OER performance is also observed for
94 low-level Fe substituted oxides at $x=0.1$ and 0.5 . In contrast, when further increasing the Fe substitution,
95 significant drop in activity is found as x exceeds 0.5 in $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$. Fig. 1e shows the selected OER CV
96 curves for $x=0, 0.25$, and 2 . Much better OER activity is found for $x=0.25$ than for $x=0$ and $x=2$. As seen in
97 the Tafel plots given in the inset of Fig. 1e, the OER overpotential for $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ (at $10 \mu\text{A cm}^{-2}_{\text{oxide}}$) is
98 approximately 70 mV lower than that for CoAl_2O_4 and CoFe_2O_4 (Supplementary Fig. 6). Such activity contrast
99 indicates a special role of Fe in oxides at the low substitution level. In addition, IrO_2 previously benchmarked
100 is also shown in the Tafel plot for comparison.⁴ As observed, the electrocatalytic activity of the
101 best-performing $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ outperforms the activity of the benchmark IrO_2 catalyst. Long-term stability
102 tests (Supplementary Fig. 7) of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0, 0.25$ and 2) oxides were carried out at constant current
103 density of $10 \mu\text{A cm}_{\text{ox}}^{-2}$ in 1 M KOH for 10 h . All measured samples exhibit negligible potential loss, which
104 indicates a good stability for these spinel catalysts under OER conditions. The best-performing catalyst,
105 $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, was further carried Chronopotentiometry test for 48 hours under $20 \mu\text{A cm}_{\text{ox}}^{-2}$ ($\sim 3.5 \text{ mA}$
106 $\text{cm}_{\text{disk}}^{-2}$). As shown in the Supplementary Fig. 8 a, b, the $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ still shows negligible activity change
107 after 48-hour test, which demonstrates a superior stability of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ in alkaline for OER.

Composition effect on OER

We then investigate the OER promotion at low substitution level and the detrimental effect at higher substitution level. Firstly, the $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ component may primarily contribute to the OER activity according to the observed composition dependence. We then carried out LCF for the Co K-edge XANES using $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ as standard in order to study the composition throughout the substituted oxides (Supplementary Fig. 3 and Supplementary Table 5). As found by the LCF (Fig. 1f, green line), the composition ratio of the $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ component is abundant at low substitution level such as $x=0.1$ and 0.25 but become negligible as $x \geq 1$. This trend correlates well with the OER activities of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ oxides (Fig. 1f, purple line). Thus, it suggests that the $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ component is primarily responsible for the significant OER enhancement. Besides, the consistent trend is also observed for the pseudocapacitive charge in CV before OER region (Supplementary Fig. 9) which is quantified from CVs (Fig. 1g) by previously reported approaches.³⁰⁻³¹ The pseudocapacitive charge is particularly large at low substitution level but significantly decrease at high Fe substitution ($x \geq 1$). The pseudocapacitive charge indicates the redox of surface active sites,³¹ and its consistent trend with OER activities throughout the oxides suggests that the amount of active sites is a dominating factor for OER activity in our case. Thus, the strong correlation between OER activity and the composition ratio should be ascribed to the $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ component, which may govern the formation of active sites.

Active site identification

Attention was then paid on the dynamic changes at metal sites during the electrochemical process. Note that there is nearly no change in the valency of Co and Fe in $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ (Fig.1d, Supplementary Fig. 10, and Supplementary Table 2) and the local atomic structure of Fe site in $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ remains unchanged under

OER (Supplementary Fig. 11). The pre-oxidation of Co (II) in $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ during the potential sweeps is proposed to play a critical role in evolving active species (the preclusion of the pre-oxidation of Fe (III) to evolve active species is shown in SI). Earlier studies have revealed the importance of pre-oxidation of Co (II) in oxides for OER, where Co (II) is inclined to be oxidized to Co (III) or higher oxidation state, which is believed to be a critical step to generate active oxyhydroxide sites for OER.³²⁻³³ To study the pre-oxidation of Co (II) in CoAl_2O_4 and $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, their cyclic voltammetries (1st and 2nd cycles) are investigated (Fig. 2a). For both CoAl_2O_4 and $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, the 1st cycle display a larger pseudocapacitive charge than 2nd cycle and the CV profiles exhibit negligible changes during subsequent cycles (Supplementary Fig. 12). Such electrochemical behaviors suggest that the surface of catalysts might undergo an irreversible surface reconstruction into, as reported, the oxyhydroxide¹⁰, evolving a stable catalytic surface for OER. In addition, we found that the oxyhydroxide formation here displays different pseudocapacitive behaviors depending on the presence or not of Fe. To be specific, the anodic peak in 1st cycle appears at ~1.32 V for $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, while a much more anodic one is observed for CoAl_2O_4 (~1.41 V), suggesting a promoting effect of Fe on the pre-oxidation of Co (II) and facilitating the subsequent formation of Co oxyhydroxide. For $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, significant differences are observed between the 1st cycle and 2nd cycle in oxidation peak as well as the pseudocapacitive charge, indicating a change in surface chemistry, while no such marked contrast was observed for CoAl_2O_4 . Thus, a more thorough reconstruction may happen on the surface of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$. According to the aforementioned pseudocapacitive behaviors, such reconstruction should be ascribed to the presence of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ component. Then, as observed in the OER region, the overpotential for triggering the OER by $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ is greatly reduced at 2nd cycle while almost no such difference is detected between 1st and 2nd cycle for CoAl_2O_4 , indicating that the reconstruction process in the presence of Fe is a critical step for OER. In the 2nd and subsequent cycles, reversible redox peaks are observed for CoAl_2O_4 and

153 $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ in pseudocapacitive range, suggesting reversible redox reactions on the reconstructed
154 surfaces. For CoAl_2O_4 , the major anodic peak appears at ~ 1.38 V, which could be assigned to Co (III) / Co (IV)
155 transition as suggested in previous literature³⁴. However, the Co (III) / Co (IV) transition cannot be
156 rationalized without an earlier notable Co (II) / Co (III) anodic features (should appear at ~ 1.2 V vs RHE).
157 Besides, its high intensity also contradicts the fact that only a small portion of Co cation in Co-based oxides
158 could be reached and oxidized into Co (IV)^{33, 35}. Considering that the surface has reconstructed during the 1st
159 cycle, we believe that this anodic peak at ~ 1.38 V should be primarily attributed to the delayed Co (II) / Co
160 (III) transition. Since the anodic process on oxyhydroxide surface can be viewed as deprotonation process
161 with oxidation of metal cation,³⁶ the delayed Co (II) oxidation for CoAl_2O_4 suggests a difficult deprotonation
162 process. For $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, an obvious anodic wave is observed at ~ 1.2 V which is assigned to Co (II) / Co
163 (III) transition.^{9, 34} This observation suggests an easier deprotonation process and activated Co (II) oxidation
164 due to the Fe substitution. Such Co (II) / Co (III) redox is followed by a double layer charging response,
165 which may result from the large surface area of the reconstructed surface and suggests a diffusion of a
166 distribution of protons on the surface³⁷. After such anodic redox of Co species, $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ exhibit a much
167 lower overpotential required for triggering OER than that for CoAl_2O_4 (~ 70 mV lower as indicated in Fig. 1e).
168 Thus, it is clear that in the presence of Fe, highly active oxyhydroxides would be induced along with the
169 reconstruction of oxide surface. Therefore, the Fe substitution is inferred to activate the Co pre-oxidation at
170 low potential, facilitating both the surface reconstruction and the subsequent evolution of surface active sites.
171 Such inference is further substantiated by *in-situ* XANES and the active species as identified by *in-situ*
172 EXAFs.

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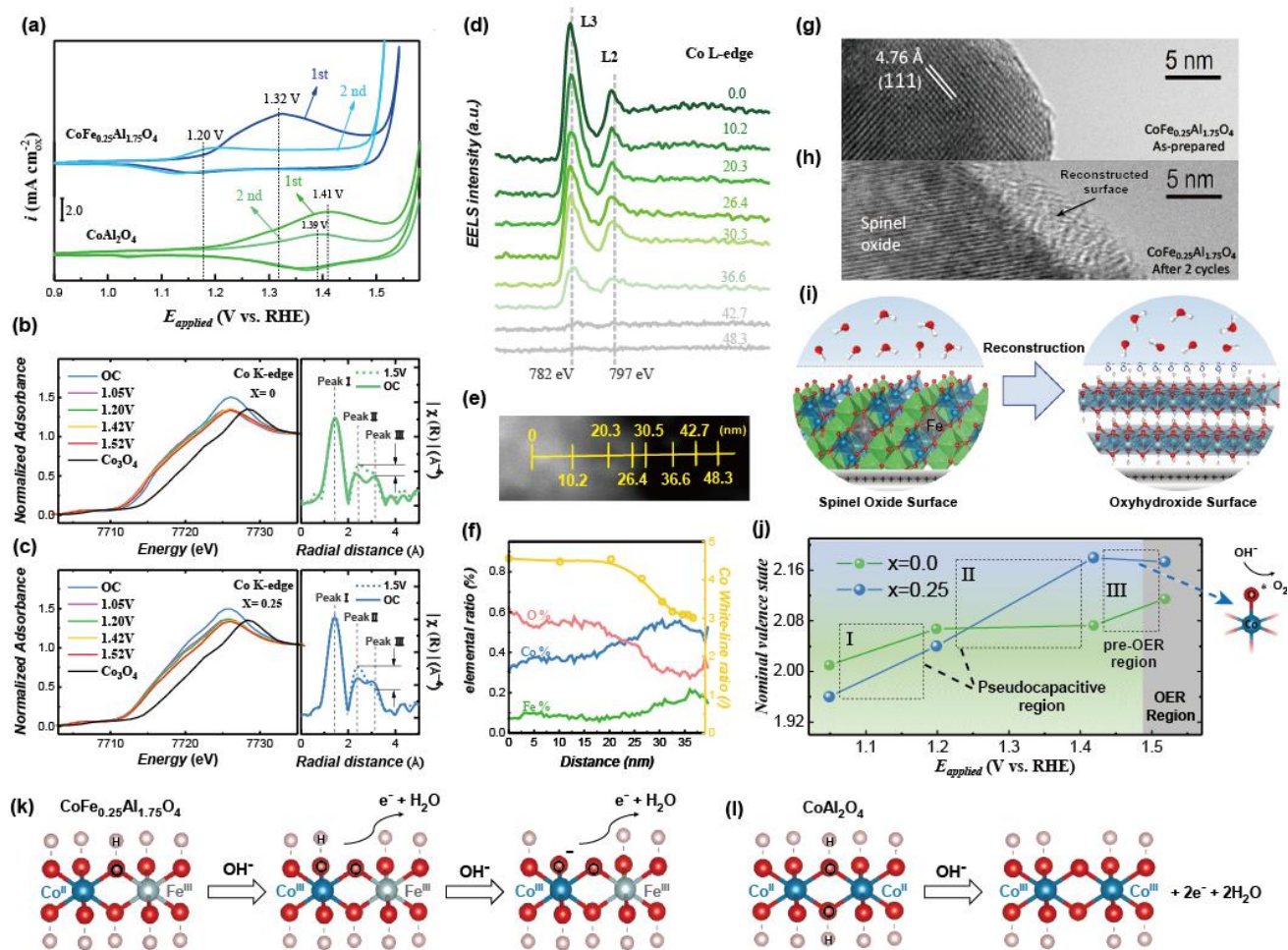


Fig. 2 | *in-situ* investigation on pre-OER behaviors of catalysts and schematic illustration of surface reconstruction and deprotonation process. **a**, The pseudocapacitive behavior in the 1st and 2nd cycles of CoFe_{0.25}Al_{1.75}O₄ and CoAl₂O₄ during the cyclic voltammetry cycling. **b,c**, The normalized in-situ Co K-edge XANES (left) under the different potentials of 1.05, 1.20, 1.42, and 1.52 V (vs. RHE) with Co₃O₄ (Sigma Aldrich) as the references, as well as the in-situ Fourier transform (FT) k₃χ(R) Co K-edge EXAFs (right) under open circuit (OC) and 1.5 V (vs. RHE): **(b)** CoAl₂O₄ and **(c)** CoFe_{0.25}Al_{1.75}O₄. The peak I, II, and III in FT-EXAFS are assigned to M-O, M_{OH}-M_{OH}, and M_{OH}-M_{Td} radial distance, respectively. **d**, The Co L-edge EELS spectra along with the line pathway as shown in **e**, scanning transmission electron microscopy (STEM) of CoFe_{0.25}Al_{1.75}O₄ after 100 cycles. The mark points denote the scanning distance along with the path way (nm): 0, 10.2, 20.3, 26.4, 30.1, 36.6, 42.7 and 48.3. **f**, The elemental ratio of O, Co, Fe (left axis) and the white-line ratio of Co L-edge (right axis) along with the line pathway. The white-line ratio is determined by the intensity of L3 and L2 peaks in EELS spectra.³⁸ **g,h**, The HRTEM images to show the surface regions for (g) as-prepared CoFe_{0.25}Al_{1.75}O₄; (h) CoFe_{0.25}Al_{1.75}O₄ after 2 cycles **i**, The reconstruction process from spinel CoFe_{0.25}Al_{1.75}O₄ into oxyhydroxide with activated negatively charged oxygen ligand. **j**, The *in-situ* Co oxidation state of CoFe_{0.25}Al_{1.75}O₄ and CoAl₂O₄ (i.e. x=0) under the different potentials of 1.05, 1.20, 1.42, and 1.52 V (vs. RHE). **k,l**, The proposed deprotonation mechanism before OER are shown for **(k)** CoFe_{0.25}Al_{1.75}O₄ and **(l)** CoAl₂O₄.

In Fig. 2b and 2c (right diagram), the *in-situ* Co K-edge Fourier transform (FT) EXAFS spectra without and

195 with an applied potential of 1.5 V (vs RHE) are shown for $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ and CoAl_2O_4 ($x=0$). At open circuit,
 196 the FT-EXAFS profiles for $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ and CoAl_2O_4 ($x=0$) are quite similar. The Peak I at ~ 1.5 Å is
 197 referred to an average metal-oxygen bond length. Two separated peaks, i.e. Peak II and Peak III at 2.4 Å and
 198 3.1 Å, respectively, are assigned to the radial distance of Co_{Oh} (Co in octahedral site) and Co_{Td} (Co in
 199 tetrahedral site) to their neighboring metal atoms. Because the Co fraction does not change in $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$
 200 with the Fe substitution, the ratio of Peak II to Peak III represents the composition ratio of Co_{Oh} and Co_{Td} in
 201 these oxides. Compared with the profiles collected at open circuit, the Peak II of Co K-edge in both
 202 $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ and CoAl_2O_4 increase with the applied potential of 1.5 V. Such increase of the Peak II
 203 indicates an accumulation of Co atoms in edge-sharing octahedral coordination⁸, which is attributed to the
 204 formation of Co oxyhydroxide which is in edge-sharing CoO_6 octahedral structure.³³ Notably, a much higher
 205 ratio of Peak II / Peak III is observed for $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ as compared to that for CoAl_2O_4 , suggesting a
 206 consistent conclusion as our observation in 1st cycle of CV that the reconstruction into oxyhydroxide is more
 207 thorough for $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$. Such reconstruction of Co on the surface for $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ is also evidenced
 208 by STEM-EELS (Fig. 2d and 2e) and HRTEM (Fig. 2g and 2h) at the reconstructed surface. Under
 209 STEM-EELS, a notable increase of elemental ratio of Co (Co%) at the near surface of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$
 210 particles is detected (Supplementary Fig. 13 and Fig. 2f), which would be a result of Co oxyhydroxide
 211 formation on surface. The white-line ratio for Co L edge (Fig. 2f) decreases and indicates increased oxidation
 212 state³⁹, which could be an effect of irreversible of electrochemical oxidation of Co (II) to form Co (III)
 213 oxyhydroxide during the 1st cycle. Under HRTEM, the generated oxyhydroxide can be observed on the
 214 reconstructed surface of oxide. Thus, the surface chemistry of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ is changed by reconstruction
 215 from oxide into oxyhydroxide (Fig. 2i). As reported that Co oxyhydroxides evolved as the active species for
 216 many Co-based oxide catalysts,^{7, 9-10, 32} we thus believe that the reconstruction facilitated by Fe substitution is

the most critical step for evolving active surface oxyhydroxide.

Fig.

In the cases of active oxyhydroxides, some studies proposed a so-called active oxygen species,^{21-22, 40} which is created during the deprotonation (anodic sweep) step, as the ultimate active site. Thus, considering the observed alternation of anodic peak by Fe substitution in 2nd CV cycle, the dynamic valence state of Co during the anodic sweep was then examined by *in-situ* XANES. Fig. 2b and 2c (left diagram) display the Co K-edge XANES of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ and CoAl_2O_4 ($x=0$) oxides recorded at 1.05, 1.20, 1.42, and 1.52 V (vs. RHE), respectively. The K-edges in XANES of both oxides shift to higher energy, indicating the oxidation of Co. The corresponding nominal valence states of Co in $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ and CoAl_2O_4 under each applied potential are plotted in Fig. 2j. The valency increment in the pseudocapacitive region could be primarily observed in region I (1.05 – 1.20 V) and region II (1.20 – 1.42 V), and the Co behavior in pre-OER stage can be found in region III (1.42 – 1.52 V). The increase of Co valency could be viewed as deprotonation process on the reconstructed surface of catalysts. As observed, in pseudocapacitive range (region I and region II), the $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ exhibits an increase of valency in both region I and region II. Whereas, the valency for CoAl_2O_4 increases only in region I, suggesting a limited deprotonation process. Its next deprotonation process is only observed at higher potential in the region III. Thus, the Fe substitution is likely to facilitate the deprotonation process at low potential to form active oxygen species on the surface, which accounts for the OER activity enhancement. Here, to illustrate the role of Fe in deprotonation process, we propose two proton/electron transfer processes for evolving active oxygen sites on $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ surface. As shown in the Fig. 2k, the first deprotonation process on $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ should start at the bridged OH linked to both Co and Fe center, which is responsible for the valency increment of Co (II) cation, and such process on bridged OH could be facilitated

239 by its neighboring Fe^{3+} center. The first deprotonation process is followed by another deprotonation process at
240 terminal OH linked to Co or Fe center. Similar deprotonation process was also reported for NiFe_xOOH ²².
241 Unlike NiFe_xOOH where the Fe substitution anodically shifts the Ni oxidation peak^{22, 36}, an opposite behavior
242 was observed here for Fe substituted CoAl_2O_4 . The suppressing effect of Fe on Ni oxidation was explained by
243 the kinetics barrier for the deprotonation of terminal OH linked to Fe center.³⁶ Thus, on the basis of the
244 activated Co oxidation in our study, we believe that the second deprotonation process should be at the terminal
245 OH linked to Co center, and the activated Co oxidation is ascribed to the reduced kinetic barrier for the proton
246 abstraction at Co site. The second process with one proton abstraction is not compensated by the metal
247 oxidation but rather by negatively charged oxygen ligand (O^*) that serves as active site. However, for
248 CoAl_2O_4 , during the corresponding pseudocapacitive range, it merely undergoes deprotonation process on
249 bridged OH (Fig. 2l). The following deprotonation for CoAl_2O_4 on the terminal OH is greatly delayed and
250 OER does not occur till the second deprotonation happens, suggesting this process as a prerequisite for OER.
251 Thus, the critical role of Fe is to facilitates the deprotonation process to generate active oxygen site at lower
252 potential on $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, thereby leading to a lower overpotential for OER. It should be noted that the
253 redox reaction on the reconstructed oxyhydroxide surface would be greatly affected by the reconstruction
254 process in the 1st cycle. As also observed in our pH dependence measurement (Fig. 3a), the redox peak is
255 greatly altered, not simply shifted, by changing the pH of electrolyte. Especially, the redox peak is even muted
256 at $\text{pH} \leq 13$. Clearly, such alternation is led by a pH-sensitive surface reconstruction, which results in forming
257 surface oxyhydroxide in different state with varied pH. Such pH-sensitive surface reconstruction suggests that
258 it may include certain decoupled proton/electron transfer process such as lattice oxygen oxidation⁴¹ which is
259 further demonstrated in the following section.

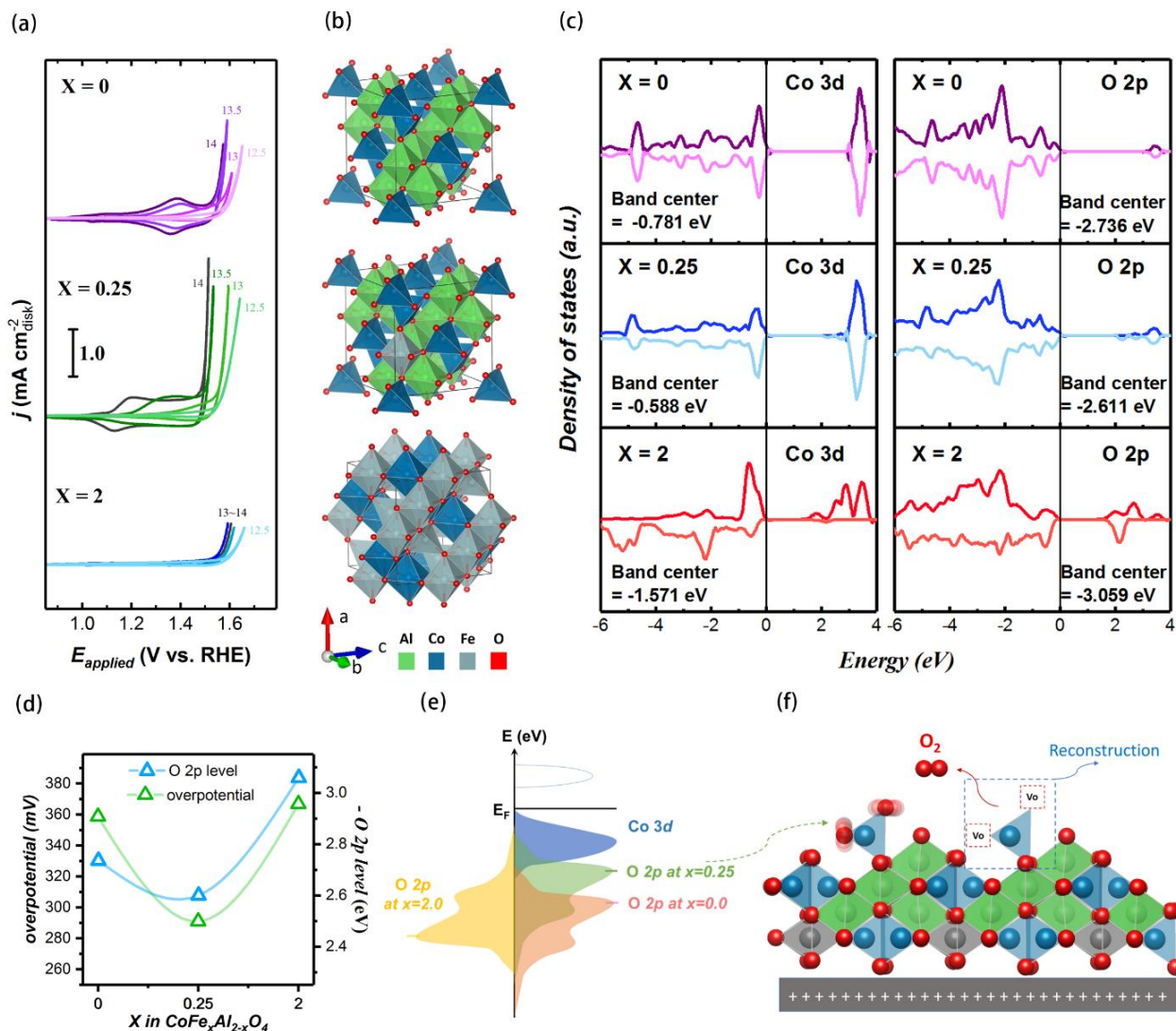


Fig. 3 | Electronic interpretation of the effect of Fe substitution on surface reconstruction. **a**, Cyclic Voltammetry (CV) of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0.0, 0.25$ and 2.0) scanned in O_2 -saturated KOH (pH= 12.5~14) at a scan rate of 10 mV s^{-1} . **b**, The computational models of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$: (top) $x=0$, (middle) $x=0.25$, (Bottom) $x=2$. **c**, The computed Co 3d, O 2p density of state (PDOS) of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0, 0.25$ and 2.0). **d**, The OER overpotential at $10 \mu\text{A cm}^{-2}_{\text{oxide}}$ and the O 2p band center relative to Fermi level ($-O 2p$) for $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0.0, 0.25$ and 2.0). **e**, The schematic band diagrams of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0, 0.25$ and 2.0). The Co 3d-band in the diagram represents the highest occupied state and the lowest unoccupied state. **f**, The schematic diagram of a surface reconstruction mechanism for $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$.

Interpretation of O 2p

As aforementioned, the active oxygen site is generated by the deprotonation on the oxyhydroxide surface and the reconstruction for forming surface oxyhydroxide serves as the prerequisite for efficient OER catalysis. We

274 have also mentioned that the Fe-facilitating reconstruction is dominated by the $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ component
 275 and may involve certain decoupled proton/electron transfer process. The inner driving force for the
 276 reconstruction is further studied by density functional theory (DFT) calculation. The electronic density of state
 277 (DOS) calculation is adopted to examine the alternation in electronic structure of oxides by Fe substitution.
 278 The computational models for $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0, 0.25$, and 2) are shown in Fig. 3b (see modelling and
 279 calculation details in SI). The projected density of state (pDOS) of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0, 0.25$, and 2) oxides and
 280 their band center energies are given in Fig. 3c (see more details in Supplementary Table 6). As found for band
 281 center energies, the Fe substitution in CoAl_2O_4 uplifts the O 2p-band center in energy closer to the Fermi level.
 282 However, as Al^{3+} is fully replaced by Fe^{3+} to form CoFe_2O_4 , it becomes an inverse spinel structure and the O
 283 2p-band center greatly moves down in energy. The PDOS results reveal that Fe substitution could either uplift
 284 or downshift the O 2p level depending on the substitution level. Moreover, the position of O 2p-band center
 285 relative to Fermi level shows consistent trend with the OER activity of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ oxides ($x=0, 0.25$ and 2)
 286 (Fig. 3d). It is also noteworthy that the OER activity of $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ oxides is greatly affected by their
 287 reconstruction process under OER conditions. Thus, the O 2p level is likely an influential factor to the surface
 288 reconstruction. Earlier studies have revealed that the O 2p-band level relative to the Fermi level was always
 289 associated with some activity-related structural parameters for perovskite oxides.^{14, 42-44} For example, a linear
 290 relationship was established between the O 2p-band level and the oxygen vacancy ($\text{V}_\text{O}^{\bullet\bullet}$) formation energy in
 291 some perovskite oxides, in which the low $\text{V}_\text{O}^{\bullet\bullet}$ formation energy could be predicted by high O 2p level close to
 292 Fermi level.^{14, 42} Given that spinel structure contains octahedral MO_6 unit as well as that of perovskite oxides,<sup>5,
 293 ⁴⁵ the uplifted O-2p band center by Fe substitution should also facilitate the $\text{V}_\text{O}^{\bullet\bullet}$ formation in spinel oxides.
 294 Consistent results have been revealed above in the structural analysis of substituted oxides where Fe
 295 substitution lowers the metal-oxygen coordination number and decrease the Co valence state, suggesting an</sup>

296 increased oxygen vacancy concentration. In addition to the influential role of O 2p level on bulk $V_O^{\bullet\bullet}$
 297 formation, it is also governing the lattice oxygen oxidation mechanism for oxides.^{41, 46-47} With the uplifted O
 298 2p center closer to Fermi level, the oxygen character in the antibonding state below Fermi level becomes more
 299 dominant (Fig. 3e). As an anodic potential is applied, the Fermi level shifts deeper into O 2p state and the
 300 holes in oxygen state are created as the O_2/H_2O redox potential is aligned with the O 2p state energy in the
 301 oxide, which leads to the oxidation of lattice oxygen.^{41, 48} Thus, credited to the uplifted O 2p level, the
 302 oxidation of lattice oxygen in $CoFe_{0.25}Al_{1.75}O_4$ should be more favorable than that in $CoAl_2O_4$ and $CoFe_2O_4$.
 303 Note that the lattice oxygen mediated OER mechanism should not dominate the OER here. This is also
 304 confirmed by a cycling test of $CoFe_{0.25}Al_{1.75}O_4$ oxides over 100 cycles (shown in Supplementary Fig. 12). The
 305 lattice oxygen-mediated OER is always featured with unstable oxide due to cation leaching on catalysts and
 306 thus leading to activity increase with cycling.⁴⁷ For example, $Ba_{0.5}Sr_{0.5}Co_{0.8}Fe_{0.2}O_{3-\delta}$ (BSCF) as well-identified
 307 oxygen active-perovskite catalysts exhibit ~4-fold current increase over 50 cycles.¹² In contrast, the
 308 $CoFe_{0.25}Al_{1.75}O_4$ did not exhibit marked OER current variation during CV cycling, suggesting that the
 309 reconstructed surface is stable and the involvement of lattice oxygen is not notable during the OER catalysis.
 310 This is further evidenced by the electrochemical study on active surface area and HRTEM (Fig. 2g, 2h and
 311 Supplementary Fig. 14) of $CoFe_{0.25}Al_{1.75}O_4$ after reconstruction and 100 cycles. The reconstruction in 1st cycle
 312 is found merely at limited depth (~5 nm) on the surface of $CoFe_{0.25}Al_{1.75}O_4$ and the reconstructed surface
 313 keeps quite stable during subsequent cycling. Please also see detailed discussion in Supplementary Note 2.
 314 Thus, the stable surface chemistry after reconstruction excludes the involvement of lattice oxygen in OER
 315 catalysis on the reconstructed surface, and also keeps the accuracy of the estimation of the specific activity
 316 normalized to BET surface area. However, we believe that the surface reconstruction should start with the
 317 lattice oxygen oxidation, which results in the aforementioned pH-sensitive reconstruction (Fig. 3a). According

318 to early report, high $V_O^{\bullet\bullet}$ (oxygen vacancies) concentration (especially $V_O^{\bullet\bullet}$ on the surface) would grant
319 structural flexibility for surface reconstruction on oxides.⁶ As a further step, even though the induced $V_O^{\bullet\bullet}$
320 grants certain structural flexibility in oxides, these $V_O^{\bullet\bullet}$ are actually stabilized in the bulk crystal. In another
321 word, such flexibility has to be triggered by additional perturbation such as the lattice oxygen oxidation. This
322 is also evidenced by our theoretical study and XANES result of CoFe_2O_4 , where CoFe_2O_4 exhibits the lowest
323 enthalpy for oxygen vacancy formation (Supplementary Table 3) and experimentally possessing the highest
324 oxygen vacancy concentration (Supplementary Table 2) compared to CoAl_2O_4 and $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$. However,
325 neither an obvious reconstruction-related current response in the first cycle nor a redox peak subjected to
326 oxyhydroxide during the second cycle can be observed for CoFe_2O_4 in CV (Supplementary Fig. 15). Besides,
327 compared to CoAl_2O_4 and $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, CoFe_2O_4 shows the weakest pH-dependent OER performance (Fig.
328 3a) and the lowest O 2p band center. Thus, we believe that the lattice oxygen oxidation at the pristine surface
329 to creating more surface $V_O^{\bullet\bullet}$ should be a critical trigger for the surface reconstruction. This is further
330 evidenced by the elemental ratio on the surface of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ after 100 cycles obtained by STEM-EELS
331 (Fig. 2f). The O% greatly decreases along with notable Co enrichment on the reconstructed surface, which
332 implies the critical role of creating surface oxygen vacancy on triggering Co reconstruction. Thus, it is
333 believed that, along with the lattice oxygen oxidation,⁴¹ a great structural instability emerges as the oxygen
334 vacancies further accumulate on the oxide surface and induce the surface reconstruction (Fig. 3f) into
335 oxyhydroxides that is more stable in alkaline.¹⁹⁻²⁰ As a result, such dynamic instability in electrochemical
336 process can be electronically indicated by the O 2p level.

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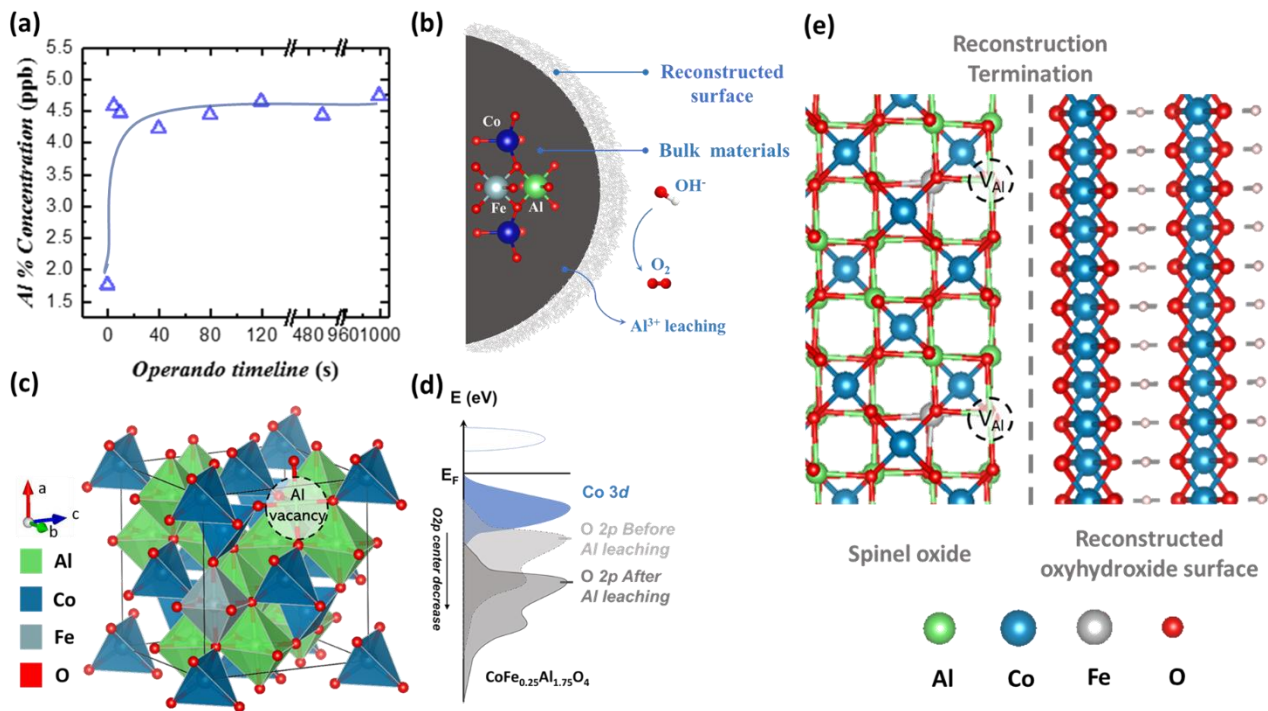


Fig. 4 | Reconstruction terminating mechanism with Al³⁺ leaching. **a**, ICP-MS test of the electrolyte for CoFe_{0.25}Al_{1.75}O₄ cycling under operation time of 0~1000s (in 1 M KOH under 20 $\mu\text{A cm}_{\text{ox}}^{-2}$). The dissolubility of Al in terms of $\text{Al}(\text{OH})_4^-$ is far beyond the concentration of Al³⁺ in our tested electrolytes. **b**, The schematic of Al³⁺ leaching along with surface reconstruction of spinel oxide. **c**, The computational model for CoFe_{0.25}Al_{1.75}O₄ after Al³⁺ leaching. The spinel structure beneath reconstructed surface was confirmed under HRTEM (Supplementary Fig. 14b). **d**, The schematic band diagrams of CoFe_{0.25}Al_{1.75}O₄ with and without Al³⁺ vacancy. **e**, The schematic of CoFe_{0.25}Al_{1.75}O₄ that terminates its surface reconstruction due to the termination of lattice oxygen oxidation.

Also importantly, while many reconstructable catalysts like Ba_{0.5}Sr_{0.5}Co_{0.8}Fe_{0.2}O_{3- δ} exhibit unstable surface chemistry and become notably amorphous after cycling¹², CoFe_{0.25}Al_{1.75}O₄ is distinguished for its stable surface chemistry after reconstruction as discussed above. The reconstruction triggered by lattice oxygen oxidation was terminated after 1st cycle and the reconstructed surface is highly active and stable in the subsequent cycles. To investigate the mechanism of such reconstruction termination, we carried ICP test on the electrolyte used for CoFe_{0.25}Al_{1.75}O₄ cycling. It was found that the leaching of Al cations was notable while Co and Fe cations both exhibited negligible leaching (Supplementary Table 7). Besides, observed under operando timeline, the Al leaching was found quickly finished as the OER happened, and no notable Al

leaching is found thereafter (Fig. 4a), which is consistent with the reconstruction process observed in CV. Hence, we believe that the leaching of Al is closely associated with the reconstruction process. As Al leached with the reconstruction at the very beginning, such leaching would alter the local electronic structures of oxide to prevent further reconstruction (Fig. 4b). We further employed DFT to study the local electronic structure for the lattice with Al vacancy (Fig. 4c). As illustrated in Fig. 4d and Supplementary Fig. 16, the O 2p level decreases in energy as Al vacancy is introduced in the lattice. As a result, the lattice oxygen oxidation would be terminated as O 2p level is low in energy, and the reconstruction thus stops accordingly as no more oxygen vacancies created (Fig. 4e). Such termination mechanism makes $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ a discernable catalyst which shows a stable surface chemistry after reconstruction and is capable to carry efficient and stable OER catalysis on reconstructed surface.

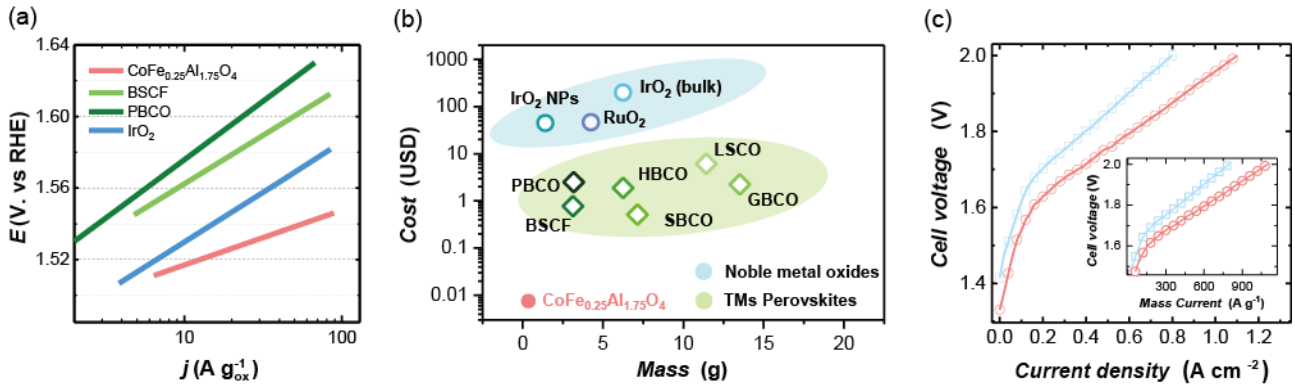


Fig. 5| Competitive potential in electrolyzers application. **a**, Mass activity of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ (BSCF)¹, $\text{Pr}_{0.5}\text{Ba}_{0.5}\text{CoO}_{3-\delta}$ (PBCO)⁴⁴ and IrO_2 nanoparticles (NPs)⁴. **b**, The material mass and cost for delivering a current of 10 A at overpotential of 0.3 V by $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$, $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ (BSCF)¹, $(\text{Pr}_{0.5}\text{Ba}_{0.5})\text{CoO}_{3-\delta}$ (PBCO)⁴⁴, $(\text{Sm}_{0.5}\text{Ba}_{0.5})\text{CoO}_{3-\delta}$ (SBCO)⁴⁴, $(\text{Gd}_{0.5}\text{Ba}_{0.5})\text{CoO}_{3-\delta}$ (GBCO)⁴⁴, $(\text{Ho}_{0.5}\text{Ba}_{0.5})\text{CoO}_{3-\delta}$ (HBCO)⁴⁴, IrO_2 nanoparticles (NPs)⁴, IrO_2 (bulk, Premetek Co.)⁴⁹ and RuO_2 ⁵⁰. The cost is evaluated by the cost of metal elements in oxides. **c**, The polarization curves of the electrolyzer with $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ (1 mg cm^{-2}) and IrO_2 (Premetek Co., 1 mg cm^{-2}) as anode catalyst and Pt/C (TKK 47.1 wt% Pt, 1 mg cm^{-2}) as cathode catalyst. Inset is the polarization curves of $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ and IrO_2 in mass current density. Cell temperature was maintained at 60 °C. The experimental details are shown in Methods.

With all advantages discussed above, $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ exhibits competitive potential in alkaline electrolyzer applications. Its mass activity (Fig. 5a) outperforms IrO_2 and the benchmarked transitional metal oxides (e.g. $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ ¹ and $\text{Pr}_{0.5}\text{Ba}_{0.5}\text{CoO}_{3-\delta}$ ⁴⁴). Its cost for given performance is lower than that of noble

metal oxides and other reported transition metal perovskites by orders of magnitude (Fig 5b). $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ was further examined in a homemade membrane electrode assembly (MEA) electrolysis cell with an anion exchange membrane as the solid electrolyte (Supplementary Fig. 17). The $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ exhibited a notable higher mass efficiency as well as higher areal activity than IrO_2 (Fig. 5c). Its performance in MEA is also better than that reported for $\text{Ba}_{0.5}\text{Sr}_{0.5}\text{Co}_{0.8}\text{Fe}_{0.2}\text{O}_{3-\delta}$ under similar condition (Supplementary Fig. 18).⁶

Conclusion

In summary, we show a promoted surface reconstruction on CoAl_2O_4 by Fe substitution. $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ is proven to be a critical component in the $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0.0\sim2.0$) series as it undergoes surface reconstruction. The surface reconstruction has been investigated by HRTEM, EELS, and XAFS. Evidenced by operando XAFS, Fe activates two deprotonation processes in reconstructed oxyhydroxides, leading to the formation of active oxygen species at a low overpotential. We interpret that the uplift of O 2p level by Fe substitution facilitates the creation of surface oxygen vacancies ($\text{V}_\text{O}^{\bullet\bullet}$) along with the lattice oxygen oxidation under OER condition and grants a greater structural flexibility for reconstruction. The O 2p moves down as Al^{3+} leaches at the beginning of reconstruction and further terminates the surface reconstruction for a stable surface chemistry. Under this strategy, $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ exhibits outstanding intrinsic activity, mass efficiency, and stability toward OER. Its high performance has been also demonstrated in MEA configuration. On the basis of this work, more alternative strategies would be explored to tune the active sites formation by adjusting its dynamic reconstruction to develop robust and low-cost OER catalysts.

Methods

Materials synthesis and characterization. $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0, 0.1, 0.25, 0.5, 1, 1.5, 2$) powders were prepared by the sol-gel method using citric acid as a chelating agent and urea as combustion agent. Firstly, cobalt acetate ($\text{Co}(\text{OAc})_2\cdot 4\text{H}_2\text{O}$), Iron(III) nitrate nonahydrate ($\text{Fe}(\text{NO}_3)_3\cdot 9\text{H}_2\text{O}$), aluminum nitrate ($\text{Al}(\text{NO}_3)_3\cdot 9\text{H}_2\text{O}$) in specific molar ratio were dissolved in diluted nitric acid, followed by the addition of citric acid and urea. The mixture was stirred and heated up at 80-100 °C to generate highly viscous gel. Then, the gel was then transferred to an oven to decompose and dry in the air at 170 °C for 12 hours. Finally, followed by the calcination at 400 °C for 6 hours, the spinel $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ ($x=0, 0.1, 0.25, 0.5, 1, 1.5, 2$) oxides were obtained. The High resolution TEM (HRTEM) and STEM-EELS were taken on a JEOL JEM- 2100F microscope at 200KV. The X-ray diffraction (XRD) patterns of

bulk $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ were recorded on Bruker D8 diffractometer at a scanning rate of 2° min^{-1} , using Cu-K_α radiation ($\lambda = 1.5418 \text{ \AA}$). The BET (Brunauer-Emitter-Teller) surface area was analyzed on ASAP Tristar II 3020 from single-point BET analysis performed after 12 h outgassing at 170°C (Supplementary Table 4).

Electrochemical characterization under three-electrode system. The working electrode was fabricated by drop casting method. The as-prepared catalysts which first mixed with acetylene black (AB) at a mass ration of 5:1, then were dispersed in isopropanol/water ($v/v=1:4$) solvent followed by the addition of Na^+ -exchanged Nafion as the binder and ultrasonicated for 20min to form homogeneous ink. The glassy carbon electrode was polished to a mirror finish with 50 nm $\alpha\text{-Al}_2\text{O}_3$ and ultrasonicated in IPA and water to completely clean up. At last, 10 μl of the as-prepared ink was dropped onto a glassy carbon (GC) electrode (0.196 cm^2) and dried overnight at room temperature to yield a final loading mass of $255 \mu\text{g}_{\text{ox}} \text{ cm}^{-2}$.

The electrochemical tests were carried out by three-electrode method using $\text{CoFe}_x\text{Al}_{2-x}\text{O}_4$ as working electrode, platinum plate ($1 \times 2 \text{ cm}^2$) as the counter electrode, Hg/HgO (1M KOH, aqueous, MMO) as the reference in O_2 -saturated 1.0 M KOH by using Bio-logic SP 150 potentiostat. All potentials are converted to RHE scale and iR corrected by the resistance of electrolyte. The conversion between the potentials vs. RHE and vs. MMO was performed by the following equation: $E \text{ (vs. RHE)} = E \text{ (vs. MMO)} + E_{\text{MMO}} \text{ (vs. SHE)} + 0.059 \times \text{pH}$. $E_{\text{MMO}} \text{ (vs. SHE)} = 0.098 \text{ vs. SHE}$ at 25°C . The Cyclic Voltammetry (CV) was obtained under potentials from 0.875V to 1.575V (vs. RHE) at a scan rate of 10 mV s^{-1} . Besides, the CV was also conducted for all samples under potentials from 0.695V to 1.495V (vs. RHE) to investigate the pseudocapacitive charge preceding the OER region. The chronopotentiometry measurement was performed by holding the specific current density of $10 \mu\text{A cm}^{-2}_{\text{oxide}}$ for 10 hours. Electrochemical impedance spectra (EIS) was recorded at 1.525V (vs. RHE) under 10 mV of amplitude from 100 KHz to 0.01Hz.

X-ray absorption spectroscopy. X-ray absorption near edge spectroscopy (XANES) and extended X-ray absorption fine structure (EXAFs) were performed under transmission mode at Singapore Synchrotron Light Source, XAFCA beamline. The Co and Fe K-edge position obtained by an integrate method²⁹ is shown in Supplementary Table 2. The Co nominal valence state is obtained by using as-prepared CoAl_2O_4 (+2.0, 7717.41 eV) and standard Co_3O_4 (+2.67, 7719.96 eV) as benchmark. The Fe nominal valence is obtained by using standard Fe_3O_4 (+2.67, 7121.71 eV) and standard Fe_2O_3 (+3, 7122.82 eV) as benchmark. Noted that the as-prepared CoAl_2O_4 well match with the standard CoAl_2O_4 , and Co valence state in standard CoAl_2O_4 is +2. The nominal oxygen vacancy concentration is calculated on the basis of the nominal valence state of Co, Al (III) and Fe. The in-situ XAS measurements were performed in fluorescence-transmission geometry, where the spectra of samples and references were measured in fluorescence mode. The catalysts were sprayed on carbon paper at a loading of 2 mg cm^{-2} as working electrode. The measurement was carried out under the same condition as OER measurement in a homemade cell. The *in-situ* XANES measurement was applied after one cycle. The *in-situ* XAFS measurement was taken on the as-prepared catalysts without any additional electrochemical treatment. Acquired XAFS data were processed in ATHENA program and analysed in ARTEMIS program integrated with IFEFFIT software package⁵¹.

Density functional theory calculations. The calculation was carried out by Vienna ab initio Simulation package (VASP) using spin-polarized density functional with the Hubbard model (DFT+U).⁵²⁻⁵³ The projector augmented wave (PAW) model with Perdew-Burke-Ernzerhof (PBE) function was used to describe the interactions between core and electrons, and the value of the correlation energy (U) was fixed at 3.3, 4.3 eV for the 3d orbits of Co and Fe, respectively.⁵⁴⁻⁵⁵ An energy cutoff of 500 eV was used for the plane-wave expansion of the electronic wave function. The Brillouin zones of all systems were sampled with Gamma-point centered Monkhorst-Pack grids. A $7 \times 7 \times 7$ Monkhorst Pack k-point setup were used for bulk geometry optimization, while $9 \times 9 \times 9$ for electronic structures calculation. The force and energy convergence criterion were set to 0.02 eV \AA^{-1} and 10^{-5} eV , respectively. All models are created by comparing the stability of spinel with normal or inversed structure in a unit cell, and the structures at lower energy are selected for study. The CoAl_2O_4 (Supplementary Table 7) prefer the normal spinel structure with Co^{2+} located on tetrahedral sites and the Al^{3+} located on octahedral sites ($[\text{Co}]_{\text{Td}}[\text{Al}]_{\text{Oh}}[\text{Al}]_{\text{Oh}}\text{O}_4$). The $\text{CoFe}_{0.25}\text{Fe}_{1.75}\text{O}_4$ model was established by replacing two Al^{3+} atom in CoAl_2O_4 unit cell with two Fe atoms. The CoFe_2O_4 (Supplementary Table 8) prefers inverse spinel with Co^{2+} at

octahedral sites and Fe^{3+} both at tetrahedral sites and octahedral sites ($[\text{Fe}]_{\text{Td}}[\text{CoFe}]_{\text{Oh}}\text{O}_4$).

Membrane electrode assembly (MEA) electrolyzer. A homemade MEA electrolysis cell with an anion exchange membrane was employed to evaluate the performance of the as-prepared $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ catalyst. The cell includes two titanium end plates, on which a single serpentine flow field (area: 6.25 cm^2 , 1.0 mm in width, 0.5 mm in depth, and 1.0 mm in rib) was machined. The titanium end plates were coated with a gold layer (thickness: 200 nm) to reduce the contact resistance. The $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ was first mixed with high surface area carbon (HSAC, Ketjen black EC-600J, carbonization treatment) with a weight ratio of 1:1. The composite powder was then mixed with PTFE suspension (60 wt% PTFE dispersion in water from Chemours), isopropyl alcohol and water to prepare the anode catalyst ink. The PTFE content in the catalyst layer was controlled with 10 wt%. The catalyst ink was ultrasonicated for 30 minutes at room temperature and manually spread onto a corrosion-resistant stainless-steel mesh (SSL mesh, #500). The catalyst loading on the as-prepared electrode was 1 mg cm^{-2} . The cathode Pt electrode was fabricated with the same procedure without the addition of HSAC. A catalyst ink, consisting of commercial Pt/C catalyst (TKK, TEC10EA50E 47.1 wt% Pt, 3.22 nm in mean particle size), PTFE suspension, isopropyl alcohol and water, was manually spread onto a carbon paper (Toray 060). The catalyst loading on the electrode was 1 mg cm^{-2} . The PTFE content in the catalyst layer was 10 wt%. The IrO_2 anode was fabricated with commercial IrO_2 powder (Premetek Co) by the same procedure as for preparing the $\text{CoFe}_{0.25}\text{Al}_{1.75}\text{O}_4$ electrode. An anion exchange membrane (A901, 11 μm , Tokuyama) was employed as the membrane. To reduce the contact resistance, the membrane sandwiched by the anode electrode and cathode electrode was pressed at a pressure of 1 MPa for 5 minutes at room temperature. 0.1 M KOH was pumped into the electrode channels by a peristaltic pump at a constant flow rate of 2 ml min^{-1} . The cell temperature ($60\text{ }^\circ\text{C}$) was maintained by an electric heating plate and measured by a thermocouple placed near the anode and cathode current collectors. The polarization curves were measured with an electrochemical workstation (Solartron 1470E). The water electrolysis was performed under constant current mode, in which the current was increased by 0.125 A (20 mA cm^{-2}) and retained for 5 minutes for each step until the cell voltage was reached 2.0 V. Before the water electrolysis, the as-fabricated MEA was activated by a potentiostatic mode with a scan rate of 0.5 mV s^{-1} and a terminal voltage of 2.0 V.

Data availability

The data related to this study is available from the authors upon reasonable request.

References

1. Suntivich, J., et al., A perovskite oxide optimized for oxygen evolution catalysis from molecular orbital principles. *Science* **334**, 1383-1385 (2011).
2. Grimaud, A.; Hong, W. T.; Shao-Horn, Y.; Tarascon, J. M., Anionic redox processes for electrochemical devices. *Nat. Mater.* **15**, 121 (2016).
3. Hong, W. T., et al., Toward the rational design of non-precious transition metal oxides for oxygen electrocatalysis. *Energy Environ. Sci.* **8**, 1404-1427 (2015).
4. Lee, Y., et al., Synthesis and Activities of Rutile IrO_2 and RuO_2 Nanoparticles for Oxygen Evolution in Acid and Alkaline Solutions. *J. Phys. Chem. Lett.* **3**, 399-404 (2012).
5. Chao, W., et al., Cations in Octahedral Sites: A Descriptor for Oxygen Electrocatalysis on Transition-Metal Spinel. *Adv. Mater.* **29**, 1606800 (2017).
6. Fabbri, E., et al., Dynamic surface self-reconstruction is the key of highly active perovskite nano-electrocatalysts for water splitting. *Nat. Mater.* **16**, 925-931 (2017).

497 7. Hsu, C.-S., et al., Valence- and element-dependent water oxidation behaviors: in situ X-ray diffraction, absorption and
 498 electrochemical impedance spectroscopies. *Phys. Chem. Chem. Phys.* **19**, 8681-8693 (2017).
 499 8. Risch, M., et al., Structural Changes of Cobalt-Based Perovskites upon Water Oxidation Investigated by EXAFS. *J.*
 500 *Phys. Chem. C* **117**, 8628-8635 (2013).
 501 9. Bergmann, A., et al., Reversible amorphization and the catalytically active state of crystalline Co₃O₄ during oxygen
 502 evolution. *Nat. Commun.* **6**, 8625 (2015).
 503 10. Wang, H.-Y., et al., In Operando Identification of Geometrical-Site-Dependent Water Oxidation Activity of Spinel
 504 Co₃O₄. *J. Am. Chem. Soc.* **138**, 36-39 (2016).
 505 11. Smith, R. D. L., et al., Spectroscopic identification of active sites for the oxygen evolution reaction on iron-cobalt
 506 oxides. *Nat. Commun.* **8**, 2022 (2017).
 507 12. May, K. J., et al., Influence of Oxygen Evolution during Water Oxidation on the Surface of Perovskite Oxide
 508 Catalysts. *J. Phys. Chem. Lett.* **3**, 3264-3270 (2012).
 509 13. Tan, Y., et al., Insight the effect of surface Co cations on the electrocatalytic oxygen evolution properties of cobaltite
 510 spinels. *Electrochim. Acta* **121**, 183-187 (2014).
 511 14. Mefford, J. T., et al., Water electrolysis on La_{1-x}Sr_xCoO_{3-δ} perovskite electrocatalysts. *Nat. Commun.* **7**, 11053
 512 (2016).
 513 15. Grimaud, A., et al., Activation of surface oxygen sites on an iridium-based model catalyst for the oxygen evolution
 514 reaction. *Nat. Energy* **2**, 16189 (2016).
 515 16. González-Flores, D., et al., Heterogeneous Water Oxidation: Surface Activity versus Amorphization Activation in
 516 Cobalt Phosphate Catalysts. *Angew. Chem. Int. Ed.* **54**, 2472-2476 (2015).
 517 17. Indra, A., et al., Unification of Catalytic Water Oxidation and Oxygen Reduction Reactions: Amorphous Beat
 518 Crystalline Cobalt Iron Oxides. *J. Am. Chem. Soc.* **136**, 17530-17536 (2014).
 519 18. Liu, W., et al., Amorphous Cobalt–Iron Hydroxide Nanosheet Electrocatalyst for Efficient Electrochemical and
 520 Photo-Electrochemical Oxygen Evolution. *Adv. Funct. Mater.* **27**, 1603904 (2017).
 521 19. Bajdich, M., et al., Theoretical investigation of the activity of cobalt oxides for the electrochemical oxidation of water.
 522 *J. Am. Chem. Soc.* **135**, 13521-13530 (2013).
 523 20. Chivot, J., et al., New insight in the behaviour of Co–H₂O system at 25–150°C, based on revised Pourbaix diagrams.
 524 *Corros. Sci.* **50**, 62-69 (2008).
 525 21. Trzeźniewski, B. J., et al., In Situ Observation of Active Oxygen Species in Fe-Containing Ni-Based Oxygen
 526 Evolution Catalysts: The Effect of pH on Electrochemical Activity. *J. Am. Chem. Soc.* **137**, 15112-15121 (2015).
 527 22. Yang, C.; Fontaine, O.; Tarascon, J. M.; Grimaud, A., Chemical Recognition of Active Oxygen Species on the
 528 Surface of Oxygen Evolution Reaction Electrocatalysts. *Angew. Chem. Int. Ed.* **56**, 8652-8656 (2017).
 529 23. Abreu-Sepulveda, M. A., et al., The Influence of Fe Substitution in Lanthanum Calcium Cobalt Oxide on the Oxygen
 530 Evolution Reaction in Alkaline Media. *J. Electrochem. Soc.* **163**, F1124-F1132 (2016).
 531 24. Duan, Y., et al., Tailoring the Co 3d-O 2p Covalency in LaCoO₃ by Fe Substitution To Promote Oxygen Evolution
 532 Reaction. *Chem. Mater.* **29**, 10534-10541 (2017).
 533 25. Zhu, Y., et al., A High-Performance Electrocatalyst for Oxygen Evolution Reaction: LiCoO₂. 8FeO. 2O₂. *Adv. Mater.*
 534 **27**, 7150-7155 (2015).
 535 26. Zhu, Y. G., et al., Unleashing the Power and Energy of LiFePO₄-Based Redox Flow Lithium Battery with a
 536 Bifunctional Redox Mediator. *J. Am. Chem. Soc.* **139**, 6286-6289 (2017).
 537 27. Walsh, A.; Yan, Y.; Al-Jassim, M. M.; Wei, S.-H., Electronic, Energetic, and Chemical Effects of Intrinsic Defects and
 538 Fe-Doping of CoAl₂O₄: A DFT+U Study. *J. Phys. Chem. C* **112**, 12044-12050 (2008).
 539 28. Terada, Y., et al., In Situ XAFS Analysis of Li(Mn, M)2O₄ (M=Cr, Co, Ni) 5V Cathode Materials for Lithium-Ion
 540 Secondary Batteries. *J. Solid State Chem.* **156**, 286-291 (2001).

541 29. Dau, H.; Liebisch, P.; Haumann, M., X-ray absorption spectroscopy to analyze nuclear geometry and electronic
542 structure of biological metal centers—potential and questions examined with special focus on the tetra-nuclear
543 manganese complex of oxygenic photosynthesis. *Anal. Bioanal. Chem.* **376**, 562-583 (2003).

544 30. Burke, L. D.; Murphy, O. J., Cyclic voltammetry as a technique for determining the surface area of RuO₂ electrodes.
545 *J. Electroanal. Chem.* **96**, 19-27 (1979).

546 31. Stoerzinger, K. A.; Qiao, L.; Biegalski, M. D.; Shao-Horn, Y., Orientation-Dependent Oxygen Evolution Activities of
547 Rutile IrO₂ and RuO₂. *J. Phys. Chem. Lett.* **5**, 1636-1641 (2014).

548 32. Yeo, B. S.; Bell, A. T., Enhanced Activity of Gold-Supported Cobalt Oxide for the Electrochemical Evolution of
549 Oxygen. *J. Am. Chem. Soc.* **133**, 5587-5593 (2011).

550 33. Kanan, M. W., et al., Structure and Valency of a Cobalt–Phosphate Water Oxidation Catalyst Determined by in Situ
551 X-ray Spectroscopy. *J. Am. Chem. Soc.* **132**, 13692-13701 (2010).

552 34. Nkeng, P., et al., Characterization of Spinel - Type Cobalt and Nickel Oxide Thin Films by X - Ray Near Grazing
553 Diffraction, Transmission and Reflectance Spectroscopies, and Cyclic Voltammetry. *J. Electrochem. Soc.* **142**, 1777-1783
554 (1995).

555 35. McAlpin, J. G., et al., EPR Evidence for Co(IV) Species Produced During Water Oxidation at Neutral pH. *J. Am.*
556 *Chem. Soc.* **132**, 6882-6883 (2010).

557 36. Görlin, M., et al., Tracking Catalyst Redox States and Reaction Dynamics in Ni–Fe Oxyhydroxide Oxygen Evolution
558 Reaction Electrocatalysts: The Role of Catalyst Support and Electrolyte pH. *J. Am. Chem. Soc.* **139**, 2070-2082 (2017).

559 37. Costentin, C.; Porter, T. R.; Saveant, J. M., How Do Pseudocapacitors Store Energy? Theoretical Analysis and
560 Experimental Illustration. *ACS Appl. Mater. Interfaces* **9**, 8649-8658 (2017).

561 38. Tan, H.; Verbeeck, J.; Abakumov, A.; Van Tendeloo, G., Oxidation state and chemical shift investigation in transition
562 metal oxides by EELS. *Ultramicroscopy* **116**, 24-33 (2012).

563 39. Wang, Z. L.; Bentley, J.; Evans, N. D., Valence state mapping of cobalt and manganese using near-edge fine
564 structures. *Micron* **31**, 355-362 (2000).

565 40. Zhang, M.; de Respinis, M.; Frei, H., Time-resolved observations of water oxidation intermediates on a cobalt oxide
566 nanoparticle catalyst. *Nat. Chem.* **6**, 362 (2014).

567 41. Grimaud, A., et al., Activating lattice oxygen redox reactions in metal oxides to catalyse oxygen evolution. *Nat. Chem.*
568 **9**, 457-465 (2017).

569 42. Lee, Y.-L., et al., Prediction of solid oxide fuel cell cathode activity with first-principles descriptors. *Energy Environ.*
570 *Sci.* **4**, 3966-3970 (2011).

571 43. Cheng, X., et al., Oxygen Evolution Reaction on La_{1-x}Sr_xCoO₃ Perovskites: A Combined Experimental and
572 Theoretical Study of Their Structural, Electronic, and Electrochemical Properties. *Chem. Mater.* **27**, 7662-7672 (2015).

573 44. Grimaud, A., et al., Double perovskites as a family of highly active catalysts for oxygen evolution in alkaline solution.
574 *Nat. Commun.* **4**, 2439 (2013).

575 45. Zhou, Y., et al., Superexchange Effects on Oxygen Reduction Activity of Edge-Sharing [Co_x Mn_{1-x} O₆] Octahedra
576 in Spinel Oxide. *Adv. Mater.*, 1705407 (2018).

577 46. Hong, W. T., et al., Charge-transfer-energy-dependent oxygen evolution reaction mechanisms for perovskite oxides.
578 *Energy Environ. Sci.* **10**, 2190-2200 (2017).

579 47. Rong, X.; Parolin, J.; Kolpak, A. M., A Fundamental Relationship between Reaction Mechanism and Stability in
580 Metal Oxide Catalysts for Oxygen Evolution. *ACS Catal.* **6**, 1153-1158 (2016).

581 48. Goodenough, J. B., Perspective on engineering transition-metal oxides. *Chem. Mater.* **26**, 820-829 (2013).

582 49. Jung, S., et al., Benchmarking nanoparticulate metal oxide electrocatalysts for the alkaline water oxidation reaction.
583 *Journal of Materials Chemistry A* **4**, 3068-3076 (2016).

584 50. Kim, N.-I., et al., Enhancing Activity and Stability of Cobalt Oxide Electrocatalysts for the Oxygen Evolution

Reaction via Transition Metal Doping. *J. Electrochem. Soc.* **163**, F3020-F3028 (2016).

51. Newville, M., IFEFFIT : interactive XAFS analysis and FEFF fitting. *Journal of Synchrotron Radiation* **8**, 322-324 (2001).

52. Kresse, G.; Hafner, J., Ab initio. *Physical Review B* **47**, 558-561 (1993).

53. Kresse, G.; Hafner, J., Ab initio. *Physical Review B* **49**, 14251-14269 (1994).

54. Kresse, G.; Joubert, D., From ultrasoft pseudopotentials to the projector augmented-wave method. *Physical Review B* **59**, 1758-1775 (1999).

55. Perdew, J. P.; Burke, K.; Ernzerhof, M., Generalized Gradient Approximation Made Simple. *Phys. Rev. Lett.* **77**, 3865-3868 (1996).

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Author contributions

T.W., S.S. A.G. and Z.X. conceived the original concept and initiated the project. T.W. prepared the materials and performed electrochemical and XRD measurements. S.S. help designed the set-up for in-situ XAS measurement. S.X. and T.W. carried out the XAS measurement. Y.D. and T.W. processed and analyzed the XAS data. J.S. worked on the DFT calculations and analysis. W.S., G.L., and B.C. carried HRTEM and STEM-EELs investigation. L.Z. conducted the measurement in MEA system. L.Z., H.W., H.L., and G.S. analyzed the MEA result. T.W. wrote the manuscript with the input from all authors, and Z.X., A.G., and S.S. revised manuscript.

Competing interests

The authors declare no competing interests.

Additional information

Supplementary information is available for this paper at