



**HAL**  
open science

## Conducting Anilate-Based Mixed-Valence Fe(II)Fe(III) Coordination Polymer: Small-Polaron Hopping Model for Oxalate-Type Fe(II)Fe(III) 2D Networks

Suchithra Ashoka Sahadevan, Alexandre Abhervé, Noemi Monni, Cristina Sáenz de Pipaón, José Ramón Galán-Mascarós, João C Waerenborgh, Bruno Vieira, Pascale Auban-Senzier, Sébastien Pillet, El-Eulmi Bendeif, et al.

### ► To cite this version:

Suchithra Ashoka Sahadevan, Alexandre Abhervé, Noemi Monni, Cristina Sáenz de Pipaón, José Ramón Galán-Mascarós, et al.. Conducting Anilate-Based Mixed-Valence Fe(II)Fe(III) Coordination Polymer: Small-Polaron Hopping Model for Oxalate-Type Fe(II)Fe(III) 2D Networks. *Journal of the American Chemical Society*, 2018, 140 (39), pp.12611-12621. 10.1021/jacs.8b08032 . hal-02356983

**HAL Id: hal-02356983**

**<https://hal.science/hal-02356983v1>**

Submitted on 9 Jan 2024

**HAL** is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

# Conducting Anilate-Based Mixed-Valence Fe(II)Fe(III) Coordination Polymer: Small-polaron Hopping Model for Oxalate-Type Fe(II)Fe(III) 2D Networks

Suchithra Ashoka Sahadevan,<sup>†,‡</sup> Alexandre Abhervé,<sup>†</sup> Noemi Monni,<sup>‡</sup> Cristina Sáenz de  
Pipaón,<sup>||</sup> José Ramón Galán-Mascarós,<sup>||,¶</sup> João C. Waerenborgh,<sup>×</sup> Bruno J. C. Vieira,<sup>×</sup> Pascale  
Auban-Senzier,<sup>‡</sup> Sébastien Pillet,<sup>Σ</sup> El-Eulmi Bendeif,<sup>Σ</sup> Pere Alemany,<sup>\*,#</sup> Enric Canadell,<sup>\*,‡</sup>  
Maria Laura Mercuri<sup>\*,‡</sup> and Narcis Avarvari<sup>\*,†</sup>

<sup>†</sup> Laboratoire MOLTECH-Anjou UMR 6200, UFR Sciences, CNRS, Université d'Angers, Bât. K, 2 Bd. Lavoisier, 49045 Angers, France

<sup>‡</sup> Dipartimento di Scienze Chimiche e Geologiche, Università degli Studi di Cagliari, I-09042 Monserrato (Cagliari), Italy

<sup>||</sup> Institut Català d'Investigació Química (ICIQ), The Barcelona Institute of Science and Technology (BIST), Avda. Països Catalans 16, 43007 Tarragona, Spain

<sup>¶</sup> ICREA, Passeig Lluís Companys 23, 08010 Barcelona, Spain

<sup>×</sup> Centro de Ciências e Tecnologias Nucleares, Instituto Superior Técnico, Universidade de Lisboa, 2695-066 Bobadela LRS, Portugal

<sup>‡</sup> Laboratoire de Physique des Solides, UMR 8502, Bât. 510, CNRS- Université Paris-Sud, 91405 Orsay, France

<sup>Σ</sup> Université de Lorraine, CNRS, CRM2, F-54000 Nancy, France

<sup>#</sup> Departament de Ciència de Materials i Química Física and Institut de Química Teòrica i Computacional (IQTCUB), Universitat de Barcelona, Martí i Franquès 1, Barcelona 08028, Spain

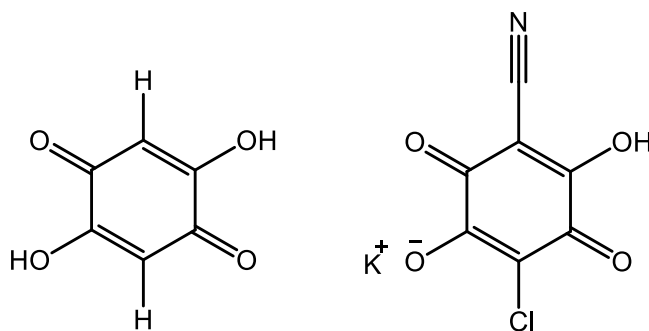
<sup>\*</sup> Institut de Ciència de Materials de Barcelona (CSIC), Campus de la UAB, E-08193 Bellaterra, Spain

*S. A. S. and A. A. contributed equally to this work*

1  
2  
3 **ABSTRACT:** The mixed-valence Fe<sup>II</sup>Fe<sup>III</sup> 2D coordination polymer formulated as  
4 [TAG][Fe<sup>II</sup>Fe<sup>III</sup>(CICNAn)<sub>3</sub>](solvate) **1** (TAG = tris(amino)-guanidinium, CICNAn<sup>2-</sup> = chloro-  
5 cyanoanilate dianionic ligand) crystallized in the polar trigonal space group *P*3. In the solid  
6 state structure, determined both at 150 K and 10 K, anionic 2D honeycomb layers  
7 [Fe<sup>II</sup>Fe<sup>III</sup>(CICNAn)<sub>3</sub>]<sup>-</sup> establish in the *ab* plane, with intralayer metal-metal distance of 7.860 Å,  
8 alternating with cationic layers of TAG. The similar Fe-O distances suggest electron  
9 delocalization and an average oxidation state of +2.5 for each Fe center. The cation imposes  
10 its C<sub>3</sub> symmetry to the structure and engages in intermolecular N-H...Cl hydrogen bonding with  
11 the ligand. Magnetic susceptibility characterization indicates magnetic ordering below 4 K and  
12 the presence of a hysteresis loop at 2 K with a coercive field of 60 Oe. Mössbauer  
13 measurements are in agreement with the existence of Fe(+2.5) ions at RT and statistic charge  
14 localization at 10 K. The compound shows semiconducting behavior with the in-plane  
15 conductivity of 2.10<sup>-3</sup> S/cm three orders of magnitude higher than the perpendicular one. A  
16 small-polaron hopping model has been applied to a series of oxalate type Fe<sup>II</sup>Fe<sup>III</sup> 2D  
17 coordination polymers, providing clear explanation on the much higher conductivity of the  
18 anilate based systems than the oxalate ones.  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

## Introduction

Molecular materials combining conducting ( $\pi$ -type, delocalized) and magnetic ( $d$ -type, localized) electrons have attracted major interest in molecular science since they can exhibit the coexistence of two distinct physical properties, furnished by the two independent networks, or novel and improved properties when they interact.<sup>1</sup> In this context, heterobimetallic oxalate-bridged compounds have been thoroughly used as magnetic lattice of multifunctional magnetic materials.<sup>2</sup> They are formed by anionic networks  $[M^II M^III(ox)_3]^-$  ( $ox = oxalate$ ) with magnetic ions linked through bis-bidentate bridging oxalate ligands. The second property is provided by the charge-compensating cation, thus combining the long-range magnetic ordering of the oxalate network with paramagnetism,<sup>3</sup> photochromism,<sup>4</sup> electrical conductivity,<sup>5</sup> proton conductivity,<sup>6</sup> ferroelectricity,<sup>7</sup> chirality,<sup>8</sup> or single-molecule magnet behavior.<sup>9</sup> In the last years, coordination polymers based on the 3,5-disubstituted-2,6-dihydroxy-1,4-benzoquinone ( $H_2dhdq$ , see Scheme 1), also called anilate ligand in its dianionic form ( $dhdq^{2-}$  or  $X_2An^{2-}$ ), have been thoroughly developed.<sup>10</sup> They usually present 2D layered<sup>11</sup> or 3D extended networks<sup>12</sup> with larger cavities than those obtained in oxalate-based coordination polymers due to the larger size of the bridging ligand. More recently, Miyasaka *et al.* have been able to increase the magnetic ordering temperature in the previously reported ferrimagnet  $(NBu_4)[Mn^II Cr^III(Cl_2An)_3]$  ( $Cl_2An^{2-} = chloranilate$ ) from 10 to 40 K.<sup>13</sup> Taking advantage of the porosity of the material, they have inserted  $Li^+$  ions into the pores of the 2D network in order to generate a radical  $Cl_2An^{\cdot-}$  ( $S=1/2$ ) and produce a new exchange interaction between the radical ligand and the metal centers, thus giving a further proof of the potential of the anilate ligand to enhance the magnetic coupling in the extended network. In this context, coexistence of electrical conductivity and magnetic ordering in  $Fe^II$  anilate-based coordination polymers have been reported simultaneously by two research groups in 2015.<sup>14</sup> In both cases, the conducting properties were attributed to the presence of radical anilate bridging ligand species. The role played by the  $Fe^II/Fe^III$  and  $L^{2-}/L^{3-}$  mixed valency was further discussed by Robson *et al.* with a more recent result based on an interpenetrated 3D network of formula  $(NBu_4)[Fe_2(F_2An)_3]$  ( $F_2An^{2-} = fluoranilate$ ).<sup>12d</sup> Another advantage of the anilate ligand is that it can be substituted on the 3 and 5 positions of the aromatic ring by a large variety of substituents.<sup>15</sup> Recently, the asymmetric chlorocyananilate ( $ClCNA^{2-}$ , see scheme 1)<sup>16</sup> ligand has been combined to the redox-active molecule bis(ethylenedithio)-tetrathiafulvalene (BEDT-TTF) to prepare an organic semiconductor with formula  $[HCNA^{2-}]_2[BEDT-TTF]$ ,<sup>17</sup> and to lanthanide ions in a series of coordination polymers of general formula  $[Ln_2(ClCNA)_3(DMF)_6] \cdot (CH_2Cl_2)_x$  ( $Ln = Yb, x = 0$ ;  $Ln = Nd$  or  $Er, x = 1$ ) which present NIR emission properties.<sup>18</sup> In this work, we have been investigating this non-symmetric anilate ligand to prepare the mixed-valence coordination polymer of formula  $[TAG][Fe^II Fe^III(ClCNA)_3] \cdot (solvate)$  (**1**), where TAG is the  $C_3$  symmetric tris(amino)-guanidinium cation, never used so far in such coordination polymers. Thorough structural characterization and study of the magnetic and conducting properties of this crystalline material are described. Since mixed-valence  $Fe^II Fe^III$  oxalate-based coordination polymers were previously reported to present a very poor conductivity, we have also investigated the origin and mechanism of the transport properties in mixed-valence  $Fe^II Fe^III$  networks based on oxalate related bridging ligands, i.e. oxalate, squarate,  $dhdq^{2-}$  and  $ClCNA^{2-}$ .



**Scheme 1.** 2,6-dihydroxy-1,4-benzoquinone ( $H_2dqb$ ) and potassium chlorocyananilate ( $KHClCNA$ ).

## Experimental Section

Tris(amino)-guanidinium chloride ( $TAGCl$ )<sup>19</sup> and potassium chlorocyananilate ( $KHClCNA$ )<sup>16</sup> were prepared according to the reported procedures. All other chemicals were commercially available and used as received without further purification.

### Synthesis of $[TAG][Fe^II Fe^III(CICNA)_3] \cdot (solvate)$ (**1**)

An aqueous solution (8 mL) of  $KHClCNA$  (64 mg, 0.27 mmol) was first placed in the bottom of a test tube, then was carefully layered a solution of  $TAGCl$  (42 mg, 0.3 mmol) in a mixture of water (2 mL) and THF (2 mL) in the middle, and then a solution of  $Fe(ClO_4)_2 \cdot xH_2O$  (51 mg, 0.2 mmol) in acetone (3 mL) on the top. After one week black hexagonal crystals suitable for XRD measurement appeared at the interface. Elemental Anal.  $C_{22}H_{67}N_9O_{41}Cl_3Fe_2$  calcd. C, 19.84; H, 5.07; N, 9.47 % found: C, 19.32; H, 5.23; N, 9.22 %. FT-IR ( $\nu_{max}/cm^{-1}$ ): 2221 ( $\nu_{C\equiv N}$ ), 1631, 1492 ( $\nu_{C-O} + \nu_{C-C}$ ), 869 ( $\delta_{C-Cl} + \nu_{C-O}$ ).

### Structural characterization

Data collection was performed at 150 K on an Agilent Supernova with  $CuK\alpha$  ( $\lambda = 1.54184 \text{ \AA}$ ). A single crystal of **1** was mounted on a glass fiber loop using a viscous hydrocarbon oil to coat the crystal and then transferred directly to the cold nitrogen stream for data collection. The structure was solved by direct methods with the SIR97 program and refined against all  $F^2$  values with the SHELXL-97 program using the WinGX graphical user interface. All non-hydrogen atoms were refined anisotropically except for the C and N atoms from the cationic entity  $CN_6H_9^+$ , and hydrogen atoms were placed in calculated positions and refined isotropically with a riding model. The program SQUEEZE from PLATON was used to calculate the potential solvent accessible void volume and the nature of the disordered solvent molecules. It has indicated a total void space of  $885 \text{ \AA}^3$  and 293 electrons/cell. This corresponds to 29 molecules of water that have been inserted in the formula of the compound. A summary of crystallographic data and refinement results are listed in Table 1.

Data collection was performed at 10 K on a SuperNova Microfocus diffractometer equipped with a two-dimensional ATLAS detector, using Mo K $\alpha$  radiation ( $\lambda = 0.71073 \text{ \AA}$ ) and a Helijet He open-flow cryosystem. The unit-cell determination and data reduction were performed using the CrysAlisPRO program suite (Rigaku Oxford Diffraction, 2017) on the full data set. The data have been indexed using the trigonal setting with cell parameters  $a = b = 13.5493(16) \text{ \AA}$ , and  $c = 9.484(3) \text{ \AA}$ . In addition, the diffraction pattern showed the presence of very weak ( $1/3 \ 1/3 \ 1/3$ ) superstructure reflections, which were not taken into account in the structure determination. The corresponding crystal structure was refined on  $F^2$  by weighted full matrix least-squares methods using the SHELXL program (Sheldrick, 2008). All non-hydrogen atoms were refined anisotropically except for the C and N atoms from the cationic entity  $\text{CN}_6\text{H}_9^+$ , and hydrogen atoms were placed in calculated positions and refined isotropically with a riding model. The program SQUEEZE from PLATON was used to calculate the potential solvent accessible void volume and the nature of the disordered solvent molecules.

CCDC-1858728 (150 K) and 1858527 (10 K) contain the supplementary crystallographic data for this paper.

**Table 1.** Crystallographic data for compound **1** at 150 K and 10 K.

	150 K	10 K
Empirical formula	$\text{C}_{22}\text{H}_{67}\text{N}_9\text{O}_{41}\text{Cl}_3\text{Fe}_2$	$\text{C}_{22}\text{H}_{67}\text{N}_9\text{O}_{41}\text{Cl}_3\text{Fe}_2$
Fw	1331.90	1331.90
Crystal color	black	black
Crystal size ( $\text{mm}^3$ )	$0.20 * 0.20 * 0.05$	$0.22 * 0.19 * 0.06$
Wavelength ( $\text{\AA}$ )	1.54184	0.71073
Crystal system, Z	Trigonal, 1	Trigonal, 1
Space group	$P3$	$P3$
$a$ ( $\text{\AA}$ )	13.616(2)	13.5493(16)
$b$ ( $\text{\AA}$ )	13.616(2)	13.5493(16)
$c$ ( $\text{\AA}$ )	9.430(4)	9.484(3)
$\alpha$ ( $^\circ$ )	90	90
$\beta$ ( $^\circ$ )	90	90
$\gamma$ ( $^\circ$ )	120	120
$V$ ( $\text{\AA}^3$ )	1514.1(8)	1507.8(6)
$\rho_{\text{calc}}$ ( $\text{g}\cdot\text{cm}^{-3}$ )	1.461	1.467
$\mu(\text{CuK}\alpha)$ ( $\text{mm}^{-1}$ )	5.994	0.653
$\vartheta$ range ( $^\circ$ )	3.75–73.74	2.15–26.36
Data collected	3465	8947
Data unique	2610	3976
Data observed	1360	1612
R(int)	0.0459	0.0604
Nb of parameters / restraints	140/4	125/2
$R1(F)$ , <sup>a</sup> $I > 2\sigma(I)$	0.0395	0.0715
$wR2(F^2)$ , <sup>b</sup> all data	0.1258	0.2391
$S(F^2)$ , <sup>c</sup> all data	0.891	1.095

<sup>a</sup> $R1(F) = \sum ||F_o| - |F_c|| / \sum |F_o|$ ; <sup>b</sup> $wR2(F^2) = [\sum w(F_o^2 - F_c^2)^2 / \sum wF_o^4]^{1/2}$ ; <sup>c</sup> $S(F^2) = [\sum w(F_o^2 - F_c^2)^2 / (n+r-p)]^{1/2}$ .

## Raman Measurements

Raman spectrum of **1** was carried out at room temperature on single crystals by using a micro Raman spectrometer (Horiba Labram 300) equipped with a He-Ne laser ( $\lambda = 632.81$  nm) laser in the 80-2000  $\text{cm}^{-1}$  range, with a 20 LWD objective ( $<0.25$   $\text{mW}/\mu\text{m}^2$  on the crystal). A  $180^\circ$  reflective geometry was adopted. The samples were mounted on a glass microscope slide, and the scattering peaks were calibrated against a Si standard ( $n = 520.7$   $\text{cm}^{-1}$ ). A typical spectrum was collected with a 300 s time constant at  $<1$   $\text{cm}^{-1}$  resolution and was averaged over three scans. KHClCNA FT-Raman spectrum was carried out at room temperature on a capillary tube by using a Bruker RFS/100 FT-Raman spectrometer equipped with a Nd-YAG laser ( $\lambda = 1064$  nm), on a back-scattering geometry. No sample decomposition was observed during the experiments. The choice of the FT-Raman spectrometer was required because of the fluorescence of the ligand under He-Ne laser.

## Magnetic measurements

Magnetic measurements were carried out on polycrystalline samples with a Quantum Design MPMS-XL-7T SQUID magnetometer (Quantum Design, Inc, San Diego, CA, USA). Magnetic measurements (dc) were carried out under an applied field of 1000 Oe. Zero-field cooled/field cooled/remanent magnetization (ZFC/FC/RM) were collected under an applied field of 25 Oe. AC susceptibility measurements were carried out with an alternating magnetic field of 3.95 Oe, in the 1-1500 Hz frequency range.

## Mössbauer spectroscopy

Mössbauer spectra were collected in transmission mode using a conventional constant-acceleration spectrometer and a 25 mCi  $^{57}\text{Co}$  source in a Rh matrix. The velocity scale was calibrated using  $\alpha$ -Fe foil. Isomer shifts, IS, are given relative to this standard at room temperature. The low temperature spectrum was collected in a bath cryostat with the sample in He exchange gas. The absorber was obtained by gently packing the powdered sample into a perspex holder. The spectra were fitted to Lorentzian lines using a non-linear least-squares method.

## Single-crystal transport measurements

Electrical transport measurements were performed on hexagonal-shaped single crystals. Gold wires (17  $\mu\text{m}$  diameter) were glued with silver paste either on two edges or on both faces of the crystals (for conductivity measurements parallel and perpendicular to the 2D planes, respectively). Two-probe DC measurements were performed applying a constant voltage in the range 0-5V and measuring the current using a Keithley 6487 Picoammeter/Voltage Source. Capacitance measurement was performed applying a constant voltage of 5V at a frequency of 2 kHz, using a RLC bridge. Low temperature was provided by a homemade cryostat equipped with a 4 K pulse-tube.

## Theoretical calculations

Density functional theory (DFT) based calculations were carried out adopting the hybrid TPSSh functional<sup>20</sup> which has been shown to give good high spin-low spin relative energies for spin-crossover complexes involving iron<sup>21</sup> and the standard double- $\zeta$  + polarization basis set 6-31G(d).<sup>22</sup> Geometries were optimized forcing a  $D_3$  symmetry and a high-spin configuration using the Gaussian 09 code.<sup>23</sup>

## Results and Discussion

### Synthesis

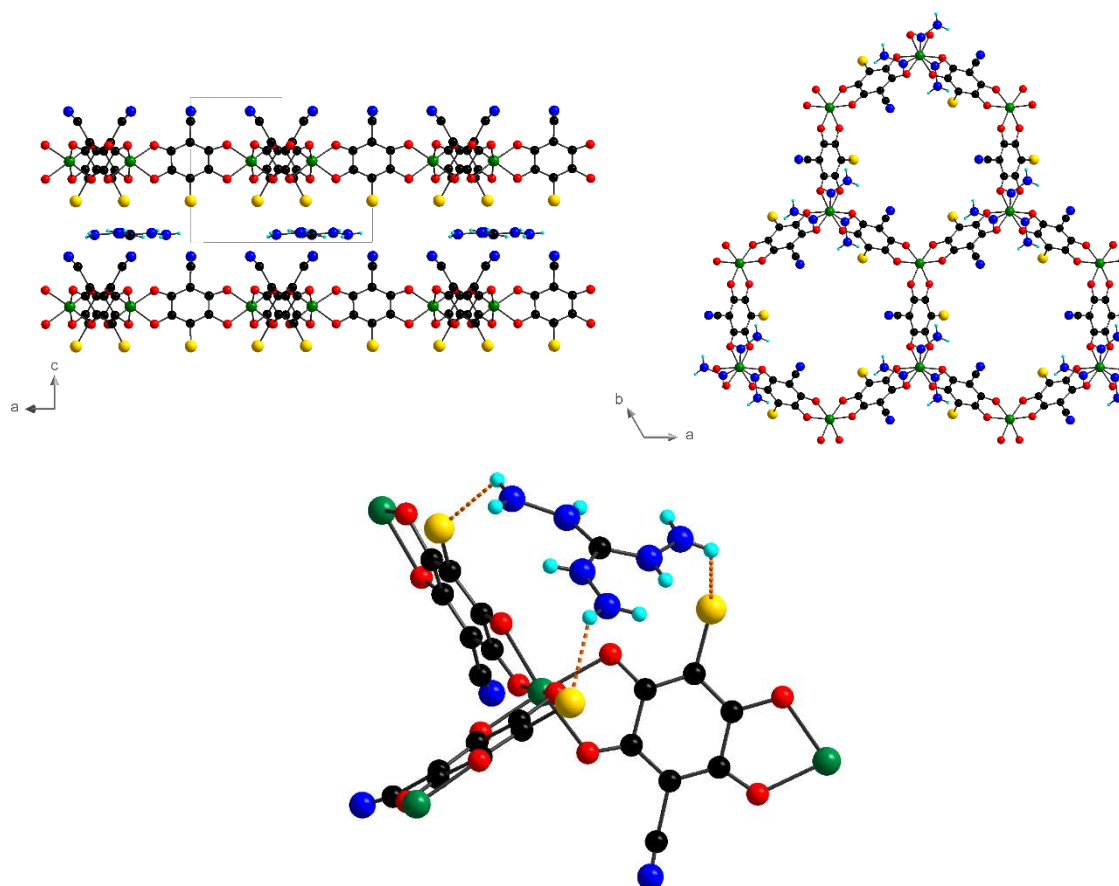
The synthesis of the mixed-valence compound **1** differs from that of the heterobimetallic oxalate and anilate-based coordination polymers. In such cases, the tetrabutylammonium salt of (tris-oxalato)metal(III) or (tris-anilato)metal(III) complex was first prepared and isolated. The precursor thus obtained was then reacted with the second metal salt by diffusion techniques to grow the bimetallic extended network. Here we have slowly diffused a solution of  $\text{Fe}^{\text{II}}\text{ClO}_4 \cdot x\text{H}_2\text{O}$  and a solution of tris(amino)-guanidinium chloride (TAGCl) into a solution of  $\text{KHClCNA n}$ . Due to the partial oxidation of the  $\text{Fe}^{\text{II}}$  ions under aerobic conditions, black hexagonal crystals of the  $\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}$  compound **1** were obtained after one week.

### Crystal Structure

Compound **1** crystallizes in the trigonal polar space group  $P3$ . The structure is formed by anionic 2D layers of formula  $[\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}(\text{ClCNA n})_3]^-$  in the  $ab$  plane, alternating with cationic layers of TAG. The anionic layer presents the well-known honeycomb structure, which is similar to other extended oxalate<sup>24</sup> and anilate-based 2D networks.<sup>11a-e</sup> It consists of a hexagonal layer with  $\text{Fe}^{\text{II}}$  and  $\text{Fe}^{\text{III}}$  ions linked through the anionic bis-bidentate  $\text{ClCNA n}^{2-}$  ligands (Figure 1, S1 and S2). As usual for this type of 2D networks, the two crystallographically independent metal centers present alternated chirality ( $\Delta$ -configuration for Fe1 and  $\Lambda$ -configuration for Fe2 in the crystal used to solve the structure). The intralayer metal-metal distance is 7.860 Å. The average Fe-O distances are very similar between both metal centres (2.037(12) Å for Fe1 and 2.047(13) Å for Fe2, see Table 2), which may indicate an electron delocalization and an average oxidation state of +2.5 for each Fe center (*vide infra*). The cationic layer is formed by one crystallographically independent TAG cation and water molecules. The TAG cation has an occupancy of 1/3, which is half of the Fe atoms, and is located only on half of the vertices of the hexagons. The structure of the cation is planar, with the C8-N2 and N2-N3 distances (1.394(15) and 1.558(17) Å respectively, see Table S1) in agreement with distances reported in the literature for other TAG based compounds.<sup>25</sup> Anionic and cationic layers present intermolecular H-bonding interactions between the terminal amino groups of the TAG cation and the chloro substituents of the anilate ligands (Figure 1). The distance between two anionic layers corresponds to the value of the  $c$  parameter (9.430(4) Å). The cationic and anionic 2D layers are eclipsed, leading to hexagonal channels along the  $c$  axis which are filled only by



1  
2  
3 solvent molecules. When compared to the previously reported anilate-based layered  
4 coordination polymers,<sup>11a-h</sup> here the use of the smaller cation results in a drastic increase of  
5 the void space inside the hexagonal channels (885 Å<sup>3</sup>). This represents 58% of the total volume  
6 thus increasing the porosity of the 2D material (Figure S3). As a consequence, the compound  
7 shows a fast release of the solvent molecules after filtration, and the nature of these solvent  
8 molecules couldn't be attributed without ambiguity. In order to reach a good reliability factor,  
9 and since 293 electrons per hexagonal cavity were determined by the SQUEEZE program, 29  
10 molecules of water have been integrated in the empirical formula (see Table 1). However, TGA  
11 analysis couldn't confirm the precise nature of the solvent molecules (Figure S4). Since a  
12 mixture of three solvents have been used during the synthesis (water, THF and acetone), the  
13 formula of the compound should be defined as [TAG][Fe<sup>II</sup>Fe<sup>III</sup>(CICNAn)<sub>3</sub>](solvate).  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50



**Figure 1.** Structure of **1** in the ac plane (top left) and in the ab plane (top right). H-bonding intermolecular interactions (dashed lines) between the cationic and anionic layers (bottom). Colour code: C, black; H, cyan; O, red; N, blue; Cl, yellow; Fe, green.

51 By comparison with the 150 K crystal structure, the overall honeycomb 2D structural  
52 architecture in the trigonal *P3* space group is retained at 10 K. Accordingly, the asymmetric  
53 unit still contains two symmetrically independent iron sites (Fe1 and Fe2). The unit cell  
54 parameters and unit cell volumes are slightly lower at 10 K, owing to usual thermal contraction  
55 effects. The intralayer metal-metal distance (7.8243(7) Å) and average Fe-O distances  
56 (2.020(14) Å for Fe1 and 2.043(14) Å for Fe2) are not significantly different from the 150 K  
57 values. Accordingly, the 10 K crystal structure does not evidence a specific ordering of the Fe<sup>II</sup>  
58 and Fe<sup>III</sup> ions on the symmetrically independent Fe1 and Fe2 sites. We can therefore consider  
59  
60

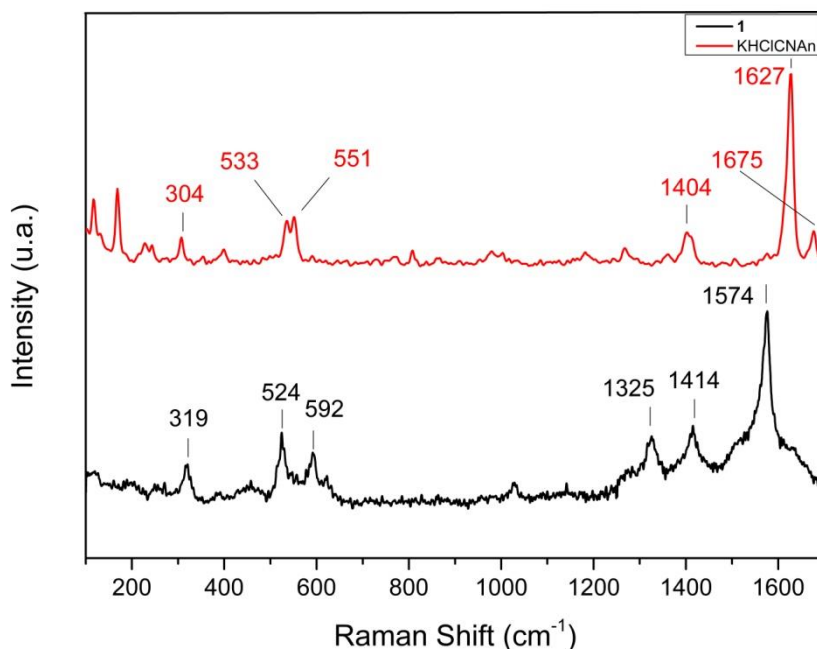
that in this description, the Fe<sup>II</sup> and Fe<sup>III</sup> ions are spatially distributed and disordered over the two sites. As mentioned in the experimental section, very weak superstructure reflections were detected on the diffraction pattern, which indicates that the exact structural ordering may be more complex than this description in the *P3* space group. However, the quality of the X-ray diffraction data, and especially the weakness of the superstructure reflections, does not allow going further. The current description leads to two Fe<sup>II</sup> and two Fe<sup>III</sup> different local environments in the crystal, which is consistent with the results from Mössbauer spectroscopy (*vide supra*).

**Table 2.** Selected Fe-O bond distances (Å) in compound **1**.

Fe-O distances	150 K (Å)	10 K (Å)
Fe1—O1	2.000(6)	1.937(7)
Fe1—O2	2.086(6)	2.103(7)
Fe2—O3	2.073(7)	2.117(7)
Fe2—O4	2.018(6)	1.974(7)

### Raman Spectroscopy

Raman spectra are valuable probes to investigate the oxidation state of coordinated benzoquinone derivatives.<sup>26</sup> Therefore In order to confirm the oxidation state of the bridging ligand a Raman study at room temperature was performed and a comparison between the Raman spectra of **1** and the free KHClCNA<sub>n</sub> ligand is reported in Figure 2.



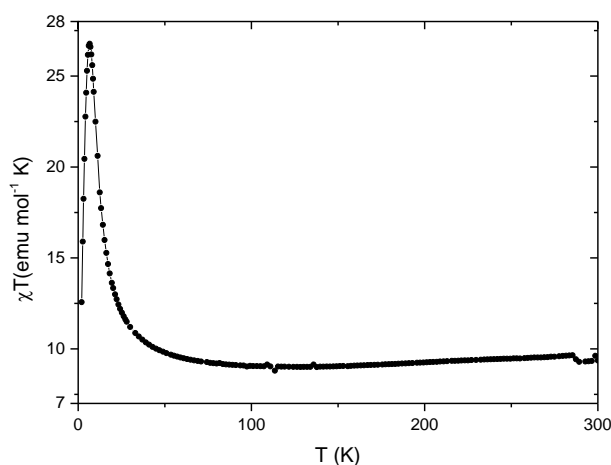
**Figure 2.** Comparison of Raman spectra of **1** and the KHClCNA<sub>n</sub> ligand, performed at room temperature with a He-Ne laser ( $\lambda = 632,81$  nm) and a Nd-YAG ( $\lambda = 1064$  nm), respectively.

The strong and broad band centered at ca.  $1574$   $\text{cm}^{-1}$  can be assigned to a  $\nu(\text{C}=\text{C}) + \nu(\text{C}=\text{O})$  combination band and the significant observed downshift from  $1617$   $\text{cm}^{-1}$  for the free ligand

1  
2  
3 can be attributed to a weakened double bond character of these terminal groups because of  
4 the coordination with the metal ion; the weak band centered at ca.  $1675\text{ cm}^{-1}$  present in the  
5 free ligand spectrum can be assigned to the  $\nu(\text{C}=\text{O})$  vibration mode for the uncoordinated  $\text{C}=\text{O}$   
6 groups of the free ligand and in fact is not observed in **1**. The two bands observed in the  $1400\text{--}$   
7  $1250\text{ cm}^{-1}$  region are assigned to the  $\nu(\text{C}-\text{C}) + \nu(\text{C}-\text{O})$  combination band and  $\nu(\text{C}-\text{C})$  vibration,  
8 respectively, and according to Harris *et al.*<sup>14c</sup> confirm the assignment of ligand oxidation state  
9 as dianionic  $\text{ClCNAn}^{2-}$ , supporting structural findings. The observed band at  $594\text{ cm}^{-1}$  can be  
10 assigned to  $\nu(\text{Fe}-\text{O}) + \nu(\text{C}-\text{C})$  combination stretching mode, as already found in previously  
11 reported dianionic anilate-based honeycomb-like networks.<sup>14a</sup> No bands can be  
12 unambiguously assigned to  $\text{Fe}^{\text{II}}-\text{O}$  and  $\text{Fe}^{\text{III}}-\text{O}$  vibrational modes thus supporting extensive  
13 electron delocalization between the Fe centers in **1** at room temperature, as clearly shown by  
14 Mössbauer spectra (*vide infra*).  
15  
16  
17  
18  
19  
20  
21

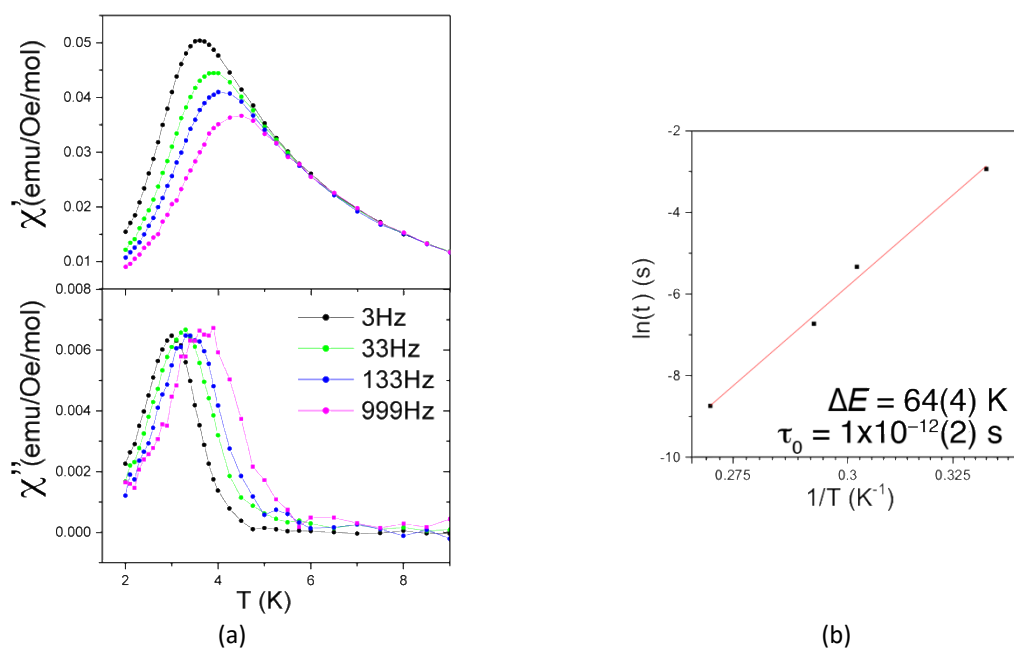
## 22 Magnetic Properties

23 The magnetic properties were measured on a polycrystalline sample of **1**. The product of the  
24 molar magnetic susceptibility times the temperature ( $\chi_{\text{m}}T$ ) presents a value of  $9.7\text{ emu}\cdot\text{K}\cdot\text{mol}^{-1}$   
25 at 300 K, which corresponds to the expected spin only value ( $7.38\text{ emu}\cdot\text{K}\cdot\text{mol}^{-1}$ ) with a  $g =$   
26  $2.2$  (Figure 3). When the temperature is lowered,  $\chi_{\text{m}}T$  slightly decreases, suggesting weak  
27 antiferromagnetic interactions between paramagnetic centres through the anilate bridges.  
28 Below 50 K,  $\chi_{\text{m}}T$  increases until reaching a value of  $\approx 27\text{ emu}\cdot\text{K}\cdot\text{mol}^{-1}$  at 7 K, followed by a sharp  
29 decrease at lower temperatures. This suggests a magnetic ordering, which was confirmed by  
30 zero-field-cooled/field-cooled (ZFC/FC) and remnant magnetization measurements under a  
31 very low magnetic field (Figure S5). The ZFC and FC plots diverge below 3.5 K, indicating the  
32 appearance of an irreversibility or memory effect. The remnant magnetization becomes non-  
33 negligible at the same temperature, confirming the existence of spontaneous magnetization  
34 below this temperature.  
35  
36  
37  
38  
39  
40  
41



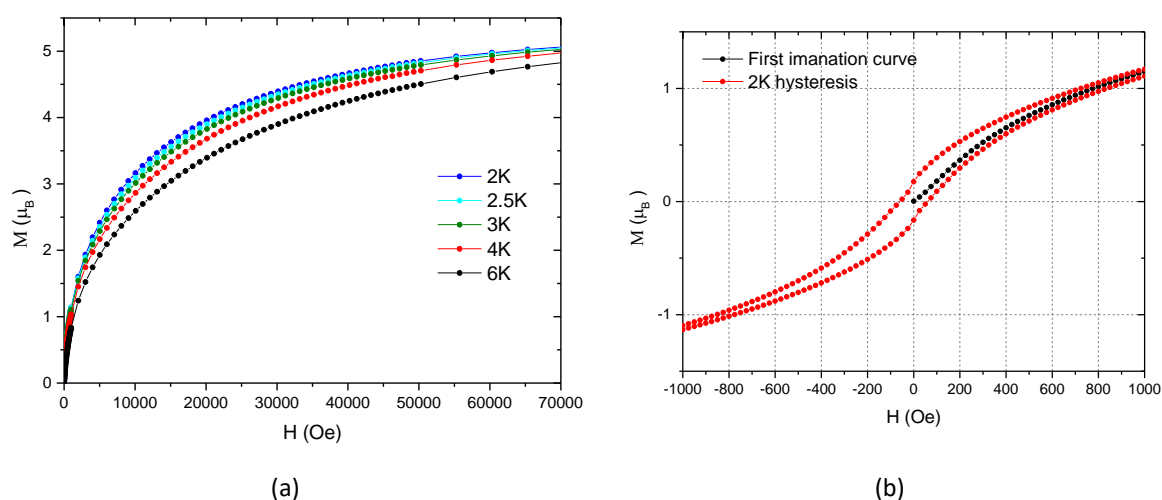
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60  
**Figure 3.** Temperature dependence of the product of the molar magnetic susceptibility times the temperature ( $\chi_{\text{m}}T$ ) of **1** under an applied field of 1000 Oe.

The magnetic ordering was also confirmed by susceptibility measurements under alternating magnetic field (AC susceptibility, Figure 4a). This shows the appearance of an out-of-phase signal and an ordering temperature  $T_c$  slightly dependent on the frequency, which can be observed for both superparamagnets and spin glasses.<sup>27</sup> The fitting of this frequency dependent behavior to a simple Arrhenius model (Figure 4b) yields parameters with no physical meaning, including a  $\tau_0 = 10^{-12}$  very different to what is found in superparamagnets (or single molecule magnets, with values of  $\tau_0$  between  $10^{-8}$  and  $10^{-10}$  s).<sup>28</sup> This  $\tau_0$  value falls within the range reported for magnetic spin-glass systems ( $10^{-12}$  -  $10^{-14}$  s)<sup>29</sup> and is then consistent with a glassy magnet behavior as observed in many 2D magnetic materials.<sup>30</sup>



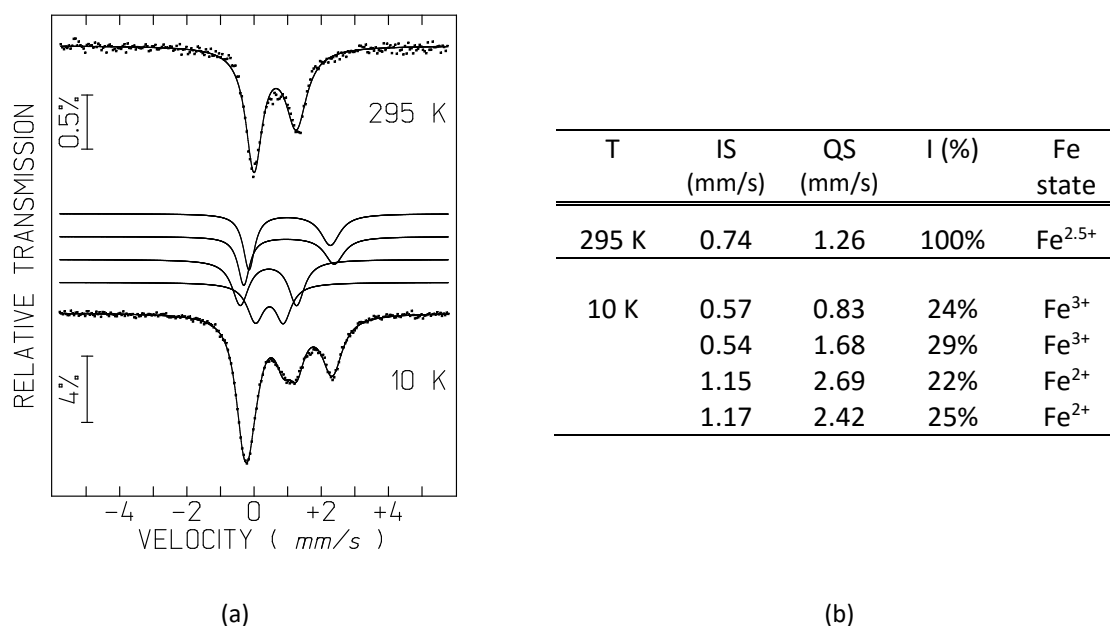
**Figure 4.** (a) Temperature dependence of the in-phase ( $\chi'$ ) and out-of-phase ( $\chi''$ ) AC susceptibility of **1** and (b) Arrhenius plot for the frequency dependence of the position of the peak in  $\chi''$  vs  $1/T$ .

Isothermal magnetization measurements at low temperatures show a fast increase of the magnetization at low fields that become more gradual at higher fields (Figure 5). The sharp increase at low fields ( $H < 1000$  Oe) also supports the appearance of spontaneous magnetization due to strong interactions between metal centers. The magnetization saturates at higher fields reaching  $\approx 5 \mu_B$ , far from the expected  $9 \mu_B$  for parallel alignment of spin carriers. This confirms the ferrimagnetic nature of the spontaneous magnetization that stabilizes a ground state with an intermediate spin and is characteristic of a glassy ferrimagnet as suggested by the initial decrease in the  $\chi_m T$  at high temperatures. An additional proof of the magnetic ordering is the presence of a hysteresis loop at 2 K with a coercive field of 60 Oe.



**Figure 5.** (a) Isothermal magnetization of **1** at different temperatures and (b) magnetic hysteresis loop at 2K.

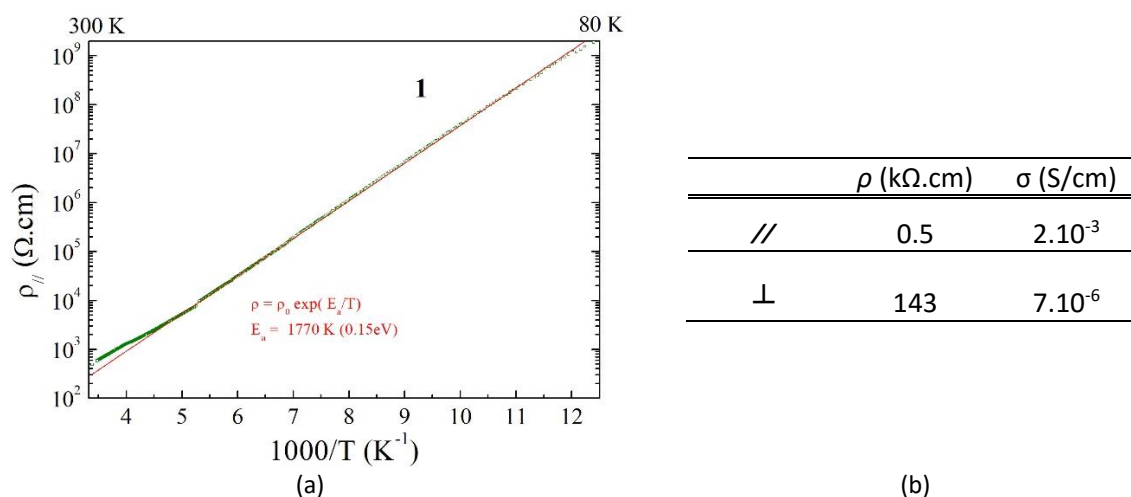
Mössbauer spectroscopy has been used to confirm the oxidation state of the Fe metal centers (Figure 6). At 10 K three broad absorption peaks are observed. They may be interpreted by two quadrupole doublets. However due to the large width of the absorption peaks a significantly better fit is obtained with four quadrupole doublets. The estimated isomer shift, IS, and quadrupole splitting, QS, are consistent with the presence of high-spin  $\text{Fe}^{3+}$  and high spin  $\text{Fe}^{2+}$  in octahedral coordination by anionic oxygen atoms.<sup>31</sup> The estimated relative areas indicate that approximately half of the Fe cations are in the +3 state and the other half in the +2 state. The two doublets observed for each oxidation state are consistent with the occupation of Fe(1) and Fe(2) crystallographic sites by both  $\text{Fe}^{2+}$  and  $\text{Fe}^{3+}$ . The room temperature spectrum shows only one asymmetric doublet with IS and QS consistent with an average oxidation state of +2.5.<sup>32</sup> The temperature dependence of the Mössbauer spectra of the anilate is similar to the behavior observed for other mixed valence iron compounds, namely molecular complexes.<sup>33</sup> The intermediate isomer shift at room temperature corresponds to a charge delocalized state on the Mössbauer spectroscopy time scale of  $\sim 10^{-7}$  s, i.e. a charge transfer frequency  $\geq 10^8 \text{ s}^{-1}$ . As the temperatures decreases, the frequency of charge delocalization gradually decreases and at 10 K the  $\text{Fe}^{2+}$  and  $\text{Fe}^{3+}$  states are localized when compared to the Mössbauer effect time window (i.e. the life-time of the  $\text{Fe}^{2+}$  and  $\text{Fe}^{3+}$  states becomes longer than  $10^{-7}$  s).



**Figure 6.** (a) Mössbauer spectra of **1** taken at 295 and 10 K and (b) estimated parameters from the spectra. The lines over the experimental points are the calculated functions. On the spectrum taken at 10 K this function is the sum of four quadrupole doublets shown slightly shifted for clarity. IS, isomer shift relative to metallic  $\alpha$ -Fe at 295 K; QS, quadrupole splitting; I, relative areas. Estimated errors are  $< 0.02$  mm/s for IS, QS,  $< 2\%$  for I.

### Transport properties

Since mixed-valence state and electron delocalization have been evidenced in **1**, we could expect this coordination polymer to present transport properties, therefore electrical conductivity measurements have been carried out on single crystals. The hexagonal shape of the crystals allows determining the direction of the {001} plane and then to measure the conductivity parallel ( $\sigma_{//}$ ) and perpendicular ( $\sigma_{\perp}$ ) to the 2D layers, corresponding to the  $ab$  plane. The temperature dependence of the resistivity indicates that **1** is a semiconductor (Figure 7). The parallel room-temperature conductivity value  $\sigma_{//}$  is about  $2 \cdot 10^{-3}$  S/cm, almost three orders of magnitude higher than the perpendicular room-temperature  $\sigma_{\perp}$  ( $7 \cdot 10^{-6}$  S/cm).

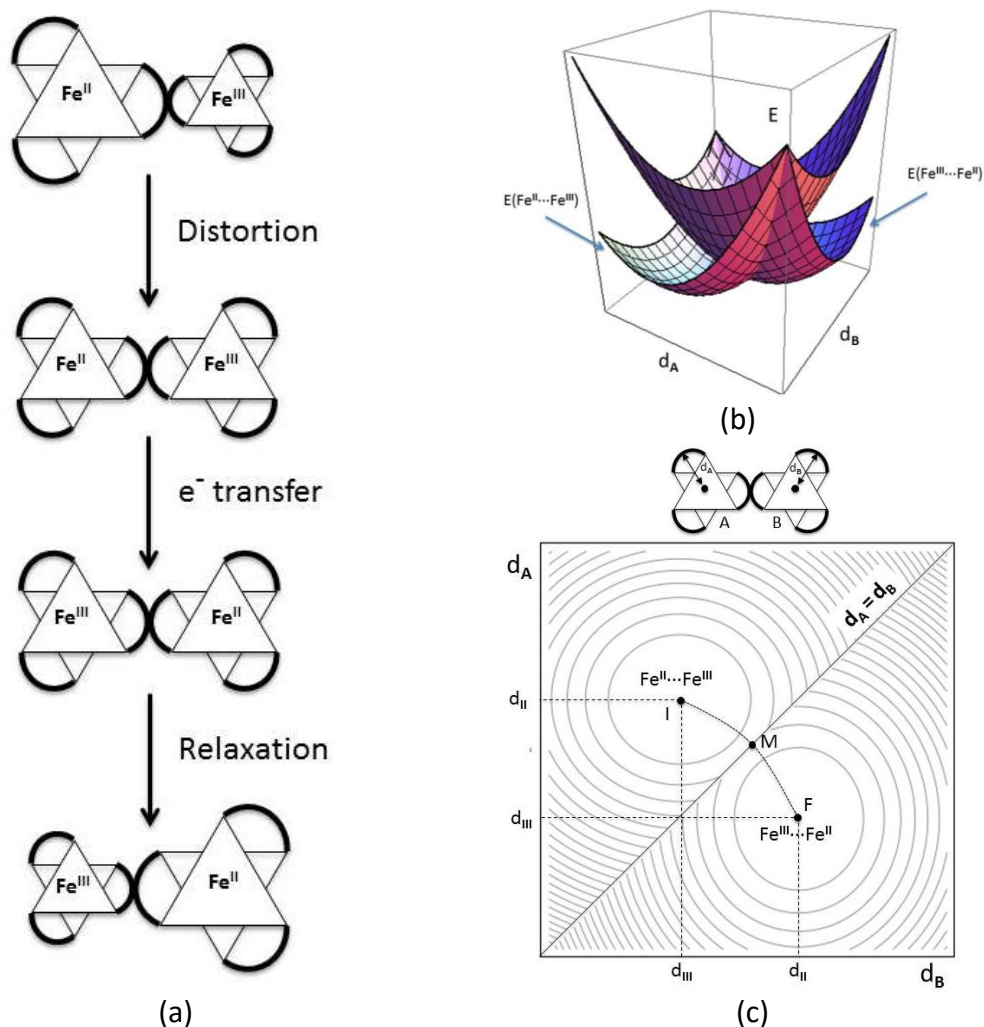


1  
2  
3 **Figure 7.** (a) Electrical resistivity  $\rho_{//}$  plotted as  $\log \rho_{//}$  versus the inverse temperature, measured with a  
4 4V voltage applied in the *ab* plane. The red line is the fit to the data with the law  $\rho = \rho_0 \exp(E_a/T)$  giving  
5 the activation energy  $E_a$ . (b) Room-temperature resistivity ( $\rho$ ) and conductivity ( $\sigma$ ) values measured  
6 along ( $//$ ) and perpendicular ( $\perp$ ) to the 2D layers.  
7  
8  
9

### 10 Relationship between the electrical conductivity and the nature of the bridging ligand

11  
12 The fairly good conductivity of compound **1** as well as of the few recently reported anilato-  
13 based Fe<sup>II</sup>Fe<sup>III</sup> coordination polymers<sup>11f,14b</sup> is in sharp contrast with the low conductivity found  
14 for the oxalate-based ones.<sup>34</sup> To gain an insight into the origin of the good conductivity in our  
15 anilate-based coordination polymer, and to point out the crucial role of the bridging ligand,  
16 we have undertaken a theoretical study on the electron transfer in 2D Fe<sup>II</sup>Fe<sup>III</sup> networks based  
17 on bis(bidentate) oxalate-type ligands in which we consider the conductivity dominated by  
18 thermally activated small-polaron hopping. We are interested in a simple, pragmatic  
19 approach, highlighting the role of the bridging ligand and in this comparative work, we have  
20 decided to focus on four ligands, oxalate, squarate (C<sub>4</sub>O<sub>4</sub><sup>2-</sup>, dianion of 3,4-dihydroxycyclobut-  
21 3-ene-1,2-dione), dnbq<sup>2-</sup> and ClCNAn<sup>2-</sup>, with different electron delocalization capabilities.  
22  
23  
24  
25

26 The basic reasoning behind the small-polaron hopping approach to the electron transfer  
27 process in either discrete or extended mixed-valence systems<sup>35,36</sup> is qualitatively illustrated in  
28 Figure 8. Consider a system formed by two separated high-spin Fe<sup>II</sup>L<sub>6</sub> ( $t_{2g}^4 e_g^2$ ) and high-spin  
29 Fe<sup>III</sup>L<sub>6</sub> ( $t_{2g}^3 e_g^2$ ) complexes in close proximity. Although the electron transfer between them is  
30 between levels of the Fe  $t_{2g}$  orbital set, which are formally non-bonding, such transfer leads,  
31 in general, to an increase/decrease of the M-L distances in the Fe<sup>II</sup>/Fe<sup>III</sup>L<sub>6</sub> units due to the  
32 expansion/contraction of the electron cloud. As a consequence of the fact that electrons move  
33 much faster than nuclei, the much faster electron transfer occurs in such a way that the  
34 geometry cannot change during the process and the system cannot exchange thermal energy  
35 with the surroundings. In other words, before the electron transfer can actually take place a  
36 Fe<sup>III</sup> species with the Fe<sup>II</sup> geometry (and viceversa) must be created. This rearrangement is  
37 however energetically unfavorable, so that the electron transfer will only occur when as a  
38 result of some vibrational process the two Fe centers reach equal coordination geometries  
39 (see Figure 8a). Thus, to understand the differences in thermally activated conductivity one  
40 must focus on the evaluation of the energetic cost of such “equalization” of the two sites.  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60



**Figure 8.** (a) Schematic representation of the three consecutive steps during the electron transfer with two neighboring pseudo-octahedral Fe(chelate)<sub>3</sub> centers shown along the C<sub>3</sub> axis. (b) 3D plot of the energy surface  $E_{tot}$  as a function of  $d_A$  and  $d_B$ . (c) Contour plot of the bottom surface from the  $E_{tot}$  plot displayed in (b).

A simple yet usually very insightful analysis relies on the assumption that the structural distortion around each center may be described by a simple harmonic oscillator  $E_i = \frac{1}{2} k_i (d_i - d_0)^2$ , where  $k_i$  is the force constant,  $d_i$  the Fe-L distance and  $d_0$  its equilibrium value. Within this approximation, the energy of the whole system is:  $E_{tot} = k_{III}(d_A - d_{III})^2 + k_{II}(d_B - d_{II})^2$  where for the sake of simplicity the  $\frac{1}{2}$  factors have been included in the force constants. Here let us assume that complex A is in the Fe<sup>III</sup> state and complex B in the Fe<sup>II</sup> state, but there is a totally equivalent expression interchanging A and B. If we plot the two energy surfaces  $E_{tot}(d_A, d_B)$  we see that they cross along the line  $d_A = d_B$  (Figure 8b). The initial and final configurations (respectively top and bottom configurations in Figure 8a) correspond to minima in the lower surface, and the structures for which  $d_A = d_B$  to the crossing seam. The situation is more conveniently analyzed by using a contour plot of the bottom surface (Figure 8c), obtained by joining the two bottom halves of the two intersecting surfaces. The two minima correspond



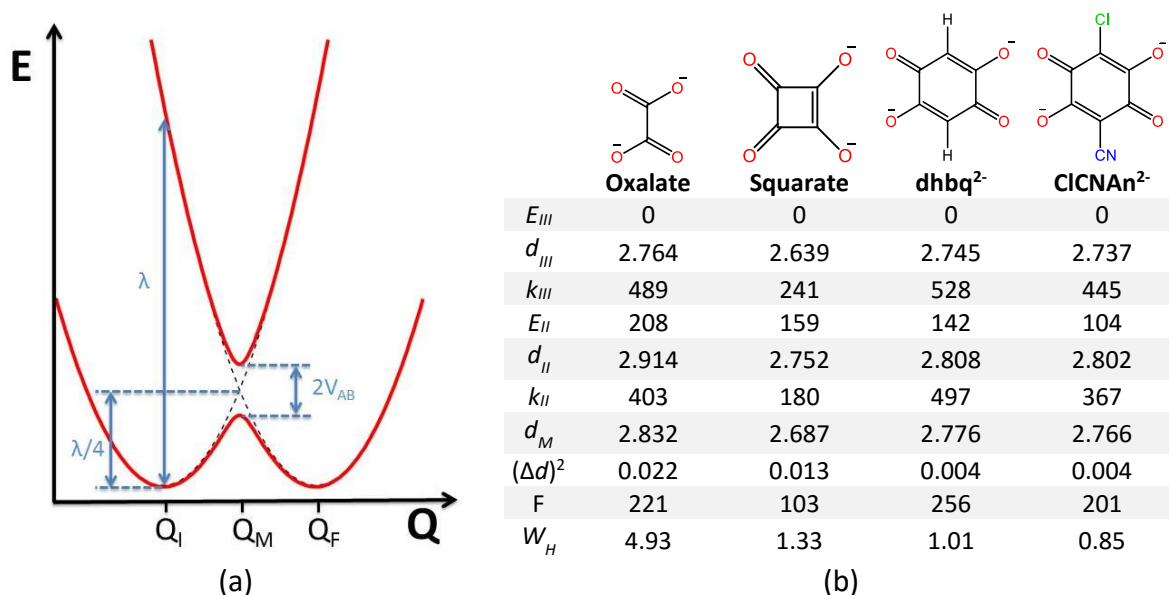
1  
2  
3 to the configurations  $\text{Fe}^{\text{III}}_{\text{A}} \cdots \text{Fe}^{\text{II}}_{\text{B}}$  (zone noted as F in Figure 8c) and  $\text{Fe}^{\text{II}}_{\text{A}} \cdots \text{Fe}^{\text{III}}_{\text{B}}$  (noted as I).  
4 The point on the  $d_{\text{A}} = d_{\text{B}}$  line for which the total energy is minimal (M) is found by taking the  
5 derivative of the energy function with  $d_{\text{A}} = d_{\text{B}} = d_{\text{M}}$  and equating it to zero. It is found that  $d_{\text{M}}$   
6  $= (k_{\text{II}}d_{\text{II}} + k_{\text{III}}d_{\text{III}}) / (k_{\text{II}} + k_{\text{III}})$  and the corresponding energy  $E_{\text{M}} = (k_{\text{II}}k_{\text{III}} / (k_{\text{II}} + k_{\text{III}}))(d_{\text{II}} - d_{\text{III}})^2$ . In  
7 other words, the minimal energy required to make the coordination environments on both Fe  
8 centers equal, that is, the height of the barrier for thermally activated electron transfer,  
9 depends basically on the difference squared between the radii of the  $\text{Fe}^{\text{III}}$  and  $\text{Fe}^{\text{II}}$  coordination  
10 environments in their equilibrium geometries.  
11  
12  
13

14 The model can be more simply depicted by using a cut through the two surfaces in which we  
15 plot the energy along the dotted path joining the two minima through the M point, which is  
16 taken as the origin (Figure 9). This gives two parabolas corresponding to the energy of the  
17 whole system with either an  $\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}$  or an  $\text{Fe}^{\text{III}}\text{Fe}^{\text{II}}$  configuration and crossing at the M point.  
18 Note that when the two complexes are not totally isolated from their surroundings, the energy  
19 necessary to distort the complexes has an additional contribution from the environment. This  
20 is particularly important in the case of 2D lattices where the distortion of one site contributes  
21 to the distortion of all its neighboring sites. This contribution is usually included considering a  
22 generic parameter  $\lambda$ , which is the vertical ionization energy from one minimum to the other  
23 curve, or in other words, the energy necessary to transfer the electron from A to B without  
24 considering a previous structural equalization of both centers. The barrier for thermal electron  
25 transfer in the absence of interaction (i.e. the energy difference between the minima and the  
26 crossing point of the two curves) is then simply  $\lambda/4$ . In the real case there is always some  
27 degree of electronic interaction between the two sites and a gap,  $2V_{\text{AB}}$ , is opened at point M  
28 and the barrier for thermal electron transfer is consequently lowered to  $\lambda/4 - V_{\text{AB}}$ .  
29 Unfortunately, there is no simple way to evaluate  $V_{\text{AB}}$  without having recourse to long and  
30 costly computations, but since we are here interested only in looking for trends when the  
31 bridging ligand is changed, it seems safe to consider that  $\lambda/4$  will be the leading term in the  
32 energy barrier, so that we can make our comparisons neglecting the effects of the nature of  
33 the ligands on  $V_{\text{AB}}$ .  
34  
35  
36  
37  
38  
39  
40

41 It is now easy to relate the main parameters (mainly structural) of this simple model to the  
42 transport measurements. The diagram in Figure 9 is completely general and applicable to any  
43 electron transfer process. In an extended system, when the dimensions of the zone where the  
44 necessary atomic rearrangement controlling the electron transfer occurs are of the order of  
45 the coordination sphere of a single site as in the present case, one talks about a system with  
46 small-polarons.<sup>36</sup> The polaron energy,  $W_{\text{P}}$ , is the energy gained when the system relaxes after  
47 addition of one electron. In this small-polaron scenario the conductivity is dominated by  
48 thermally activated electron hopping with the mobility given by the equation  $\mu = \mu_0 e^{-W_{\text{H}}/k}$ ,  
49 where  $W_{\text{H}}$  is the electron hopping barrier, i.e. the energy cost to reach the geometry under  
50 which the electron transfer is possible. As far as it is assumed that the variation of energy is a  
51 quadratic function of the structural parameters, the energy cost per site to reach the  
52 "equalization" geometry is  $1/4 W_{\text{P}}$  and, taking into account that there are two sites involved  
53 in the transfer, it follows that  $W_{\text{H}} = 1/2 W_{\text{P}}$ . Since according to the Franck-Condon principle  
54 the energy to optically excite one electron from one to the other site (i.e.  $\lambda$  in Figure 9) is twice  
55 the polaron energy it follows that  $W_{\text{H}} = 1/2 W_{\text{P}} = \lambda/4$ . Therefore  $W_{\text{H}} = E_{\text{M}} = F(d_{\text{II}} - d_{\text{III}})^2$ , with F  
56  
57  
58  
59  
60

$= k_{II}k_{III} / (k_{II} + k_{III})$  and  $d_M = (k_{II}d_{II} + k_{III}d_{III}) / (k_{II} + k_{III})$ . In that way it is possible to correlate the transport ( $W_H$ ) and structural ( $k_{II}$ ,  $k_{III}$ ,  $d_{II}$  and  $d_{III}$ ) features for a series of compounds.

How are these parameters tuned by the nature of the bridging ligand? The values of  $k_{II}$ ,  $k_{III}$ ,  $d_{II}$  and  $d_{III}$  can be evaluated from density functional calculations (DFT). We carried out structural optimizations of  $\text{Fe}^{II}\text{L}_3$  and  $\text{Fe}^{III}\text{L}_3$  complexes with the four ligands oxalate, squarate,  $\text{dhbq}^{2-}$  and  $\text{ClCNAN}^{2-}$  using the TPSSh functional, which is known to give good high spin-low spin relative energies for spin-crossover complexes involving iron.<sup>21</sup> The geometries were optimized forcing a  $D_3$  symmetry and a high-spin configuration. In that way we obtained the  $d_{M-L}$  parameter which is the distance between the Fe atom and the midpoint to the closest C–C bonds (between the C atoms bonded to the coordinating O atoms) which corresponds to  $d_{II}$  or  $d_{III}$  of the above discussion. Once the optimal  $d_{M-L}$  parameter for each  $\text{Fe}^{II}$  and each  $\text{Fe}^{III}$  complex was found the values of  $k_{II}$  or  $k_{III}$  were obtained by reoptimizing the structure for fixed  $d_{M-L} \pm \delta$  values and fitting the energy to a second order polynomial. The results for the four ligands are reported in Figure 9b.



**Figure 9.** (a) Energy plot for a  $\text{Fe}^{II}\text{Fe}^{III}$  to  $\text{Fe}^{III}\text{Fe}^{II}$  electron transfer process. (b) Calculated parameters for the four studied bridging ligands. Distances are expressed in Å, energies in  $\text{kcal}\cdot\text{mol}^{-1}$ ,  $k$  and  $F$  in  $\text{kcal}\cdot\text{mol}^{-1}\cdot\text{Å}^{-2}$ .

According to the data in Figure 8b  $W_H = 0.214, 0.057, 0.044$  and  $0.037$  eV for oxalate, squarate,  $\text{dhbq}^{2-}$  and  $\text{ClCNAN}^{2-}$ , respectively. Despite the simplicity of the approach, these numbers are very reasonable when compared with  $W_H$  estimations based on experimental data for  $\text{Fe}^{II}\text{Fe}^{III}$  solids like magnetite ( $\sim 0.15$  eV)<sup>36</sup> and partially substituted ferrites ( $\sim 0.11$  eV)<sup>37</sup> which are in between those calculated for oxalate and the other three ligands. Thus, we believe that the simple approach captures the essence of the electron transfer process although the activation energies are smaller than the experimental ones. This is not unexpected since the Fe sites are immersed in a quite rigid 2D network, which must lead to a noticeable increase in the parameter  $\lambda$ , and hence in the barrier estimated here by  $\lambda/4$ . According to the values in Figure 8b electrons should be much less mobile in oxalate networks than in anilate ones, which is in good agreement with experimental observations.<sup>34,11f,14b</sup> What is the reason for this

1  
2  
3 finding? The  $F$  values for the oxalate ligands, are in between those for both the  $\text{d}^{\text{I}}\text{d}^{\text{II}}$  and  
4  $\text{ClCNAn}^{2-}$  ones, yet  $W_H$  is five times larger. It is clear that the factor determining the large  
5 difference is  $(\Delta d)^2$ , i.e. the square of the difference in the  $d$  values for Fe in the two oxidation  
6 states. Because of the quadratic dependence, this term has an overwhelming influence on the  
7 final value of  $W_H$ . Comparing the  $d_{\text{II}}$  and  $d_{\text{III}}$  distances for the oxalate,  $\text{d}^{\text{I}}\text{d}^{\text{II}}$  and  $\text{ClCNAn}^{2-}$   
8 ligands in Figure 9b it is clear that whereas the  $d_{\text{III}}$  values are fairly similar,  $d_{\text{II}}$  for the oxalate  
9 ligand is markedly larger than those for the  $\text{d}^{\text{I}}\text{d}^{\text{II}}$  and  $\text{ClCNAn}^{2-}$  ligands. Thus, the determining  
10 factor for the lower conductivity of the oxalate systems lies in the difficulty to cope with the  
11 extra electron of the high spin  $\text{Fe}^{\text{II}}$  situation. The occurrence of extensive delocalization  
12 through the benzene ring of the  $\text{d}^{\text{I}}\text{d}^{\text{II}}$  and  $\text{ClCNAn}^{2-}$  acts as a buffer for the electronic  
13 rearrangement needed by the presence of the extra electron. Note that when the ligands are  
14 more similar, the  $F$  factor depending on the force constants may become the key factor.  
15 According to Figure 8b this is for instance the case when comparing the  $\text{d}^{\text{I}}\text{d}^{\text{II}}$  and  $\text{ClCNAn}^{2-}$   
16 ligands. The squarate ligand is also associated with a small  $W_H$  value, not much different from  
17 those of the  $\text{d}^{\text{I}}\text{d}^{\text{II}}$  and  $\text{ClCNAn}^{2-}$  ligands. However, the  $(\Delta d)^2$  term is intermediate between  
18 those of the oxalate and  $\text{d}^{\text{I}}\text{d}^{\text{II}}$  and  $\text{ClCNAn}^{2-}$  pair of ligands. Clearly, in that case the  $F$  term  
19 plays an important role. Both the geometrical constraints imposed by the four-member ring  
20 as well as the delocalization in the central part of the ligand influence the electron transfer  
21 tendency.  
22  
23  
24  
25  
26  
27  
28

29 We thus conclude that the present approach provides a simple, yet insightful, model to  
30 rationalize the transport results for these  $\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}$  mixed-valence coordination polymers and  
31 that the polymers based on the  $\text{d}^{\text{I}}\text{d}^{\text{II}}$  and  $\text{ClCNAn}^{2-}$  ligands rank among the most effective  
32 ones in promoting the electron delocalization through a small-polaron hopping mechanism.  
33  
34  
35  
36

## 37 Conclusions

38  
39 The mixed-valence  $\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}$  2D coordination polymer  $[\text{TAG}][\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}(\text{ClCNAn})_3]$  based on the  
40 asymmetric chloro-cyanoanilate ligand and containing, for the first time in such 2D networks,  
41 the tris(amino)-guanidinium (TAG) cation has been synthesized and crystallized in the polar  
42 non-centrosymmetric space group  $P3$  thanks to the  $C_3$  symmetry of the cation and its ability  
43 to engage in intermolecular hydrogen bonding with the chlorine atoms of the ligand. Magnetic  
44 susceptibility measurements in combination with Mössbauer spectroscopy are indicative of a  
45 spin-glass behavior with magnetic ordering below 4 K and the presence of intermediate  
46  $\text{Fe}^{(+2.5)}$  oxidation state at RT and charge localization  $\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}$  at 10 K, with a statistic  
47 occupational crystallographic site according to the 10 K Mössbauer spectra, also confirmed by  
48 the X-ray structure at 10 K. Single crystal electron transport measurements in the 2D plane  
49 and perpendicular on it show semiconducting behavior of the material, with a rather high RT  
50 value of  $2 \cdot 10^{-3}$  S/cm for the in-plane conductivity, much higher than the one reported in the  
51 oxalate based 2D  $\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}$  coordination polymers. In order to shed light on this difference and  
52 on the electron transport mechanism on these fast developing multifunctional families of 2D  
53 coordination polymers, the small-polaron hopping approach to the electron transfer process  
54 has been applied to a series of mixed-valence  $\text{Fe}^{\text{II}}\text{Fe}^{\text{III}}$  oxalate related coordination polymers  
55  
56  
57  
58  
59  
60

1  
2  
3 containing as bridging ligands oxalate<sup>2-</sup>, squarate<sup>2-</sup>, dnbq<sup>2-</sup> and ClCNAn<sup>2-</sup>. The results are clearly  
4 indicative of a much lower electron hopping barrier in the anilate complexes than in oxalates.

5  
6 These results open the way towards the use of the TAG cation in such multifunctional  
7 binuclear/bimetallic transition metal or lanthanide coordination polymers, possibly endowed  
8 with multiferroic properties thanks to the crystallization in polar space groups. Moreover, the  
9 mechanism of electron transport in the mixed-valence Fe<sup>II</sup>Fe<sup>III</sup> coordination polymers with  
10 bridging oxalate type ligands has been disclosed, pointing out the higher “elasticity” of the  
11 anilate network compared to the oxalate one.  
12  
13  
14  
15  
16

## 17 ASSOCIATED CONTENT

18  
19 **Supporting Information.** Full experimental section including description of the synthesis and  
20 characterization of all new materials and all the techniques employed in the research reported  
21 here. This information is available free of charge via the internet at <http://pubs.acs.org>.  
22  
23  
24  
25  
26  
27

## 28 AUTHOR INFORMATION

### 29 Corresponding Authors

30  
31 [p.alemany@ub.edu](mailto:p.alemany@ub.edu)

32  
33 [canadell@icmab.es](mailto:canadell@icmab.es)

34  
35 [mercuri@unica.it](mailto:mercuri@unica.it)

36  
37 [narcis.avarvari@univ-angers.fr](mailto:narcis.avarvari@univ-angers.fr)  
38  
39  
40  
41  
42  
43  
44  
45

### 46 Notes

47  
48 The authors declare no competing financial interest.  
49  
50  
51  
52  
53

## 54 ACKNOWLEDGMENTS

55  
56 The work in France was supported by the CNRS, the University of Angers, the Erasmus program  
57 (mobility grant to N.M.), and the RFI Regional project LUMOMAT (grant to A.A., project ASCO  
58 MMM). This work was supported in Italy by the Fondazione di Sardegna-Convenzione  
59  
60

1  
2  
3 triennale tra la Fondazione di Sardegna e gli Atenei Sardi, Regione Sardegna-L.R. 7/2007  
4 annualità 2016-DGR 28/21 del 17.05.2015 “Innovative Molecular Functional Materials for  
5 Environmental and Biomedical Applications” and INSTM. Work in Spain was supported by the  
6 Spanish Ministerio de Economía y Competitividad (Grants FIS2012-37549-C05-05, FIS2015-  
7 64886-C5-4-P, CTQ2015-64579-C3-3-P and CTQ2015-71287-R) and Generalitat de Catalunya  
8 (2014SGR301, 2017SGR797 and XRQTC, and the CERCA program). E.C. acknowledges support  
9 of the Spanish MINECO through the Severo Ochoa Centers of Excellence Program under Grant  
10 SEV-2015-0496. J.C.W. acknowledges Fundação para a Ciência e a Tecnologia (FCT, Portugal)  
11 through the project UID/Multi/04349/2013.  
12  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23

## 24 References

- 25  
26  
27 <sup>1</sup> (a) Coronado, E.; Day, P. *Chem. Rev.* **2004**, *104*, 5419–5448. (b) Enoki, T.; Miyazaki, A. *Chem. Rev.*  
28 **2004**, *104*, 5449–5478. (c) Kobayashi, H.; Cui, H. B.; Kobayashi, A. *Chem. Rev.* **2004**, *104*, 5265–5288.  
29 (d) Kurmoo, M.; Graham, A. W.; Day, P.; Coles, S. J.; Hursthouse, M. B.; Caulfield, J. L.; Singleton, J.;  
30 Pratt, F. L.; Hayes, W.; Ducasse, L.; Guionneau, P. *J. Am. Chem. Soc.* **1995**, *117*, 12209–12217. (e)  
31 Martin, L.; Turner, S. S.; Day, P.; Mabbs, F. E.; McInnes, E. J. L. *Chem. Commun.* **1997**, *15*, 1367–1368.  
32 (f) Rashid, S.; Turner, S. S.; Day, P.; Howard, J. A. K.; Guionneau, P.; McInnes, E. J. L.; Mabbs, F. E.; Clark,  
33 R. J. H.; Firth, S.; Biggs, T. *J. Mater. Chem.* **2001**, *11*, 2095–2101. (g) Uji, S.; Shinagawa, H.; Terashima,  
34 T.; Yakabe, T.; Terai, Y.; Tokumoto, M.; Kobayashi, A.; Tanaka, H.; Kobayashi, H. *Nature* **2001**, *410*, 908–  
35 910. (h) Fujiwara, H.; Fujiwara, E.; Nakazawa, Y.; Narymbetov, B. Zh.; Kato, K.; Kobayashi, H.; Kobayashi,  
36 A.; Tokumoto, M.; Cassoux, P. *J. Am. Chem. Soc.* **2001**, *123*, 306–314. (i) Coronado, E.; Falvello, L. R.;  
37 Galán-Mascarós, J. R.; Giménez-Saiz, C.; Gómez-García, C. J.; Lauhkin, V. N.; Pérez-Benítez, A.; Rovira,  
38 C.; Veciana, J. *Adv. Mater.* **1997**, *9*, 984–987. (j) Madalan, A. M.; Canadell, E.; Auban-Senzier, P.;  
39 Brânzea, D.; Avarvari, N.; M. Andruh, M. *New J. Chem.* **2008**, *32*, 333–339. (k) Pop, F.; Allain, M.; Auban-  
40 Senzier, P.; Martínez-Lillo, J.; Lloret, F.; Julve, M.; Canadell, E.; Avarvari, N. *Eur. J. Inorg. Chem.* **2014**,  
41 3855–3862.  
42 <sup>2</sup> Clemente-León, M.; Coronado, E.; Martí-Gastaldo, C.; Romero, F. M. *Chem. Soc. Rev.* **2011**, *40*, 473–  
43 497.  
44 <sup>3</sup> (a) Clemente-León, M.; Galán-Mascarós, J. R.; Gómez-García, C. J. *Chem. Commun.* **1997**, 1727–1728.  
45 (b) Coronado, E.; Galán-Mascarós, J. R.; Gómez-García, C. J.; Martínez-Agudo, J. M. *Adv. Mater.* **1999**,  
46 *11*, 558–561. (c) Coronado, E.; Galán-Mascarós, J. R.; Gómez-García, C. J.; Ensling, J.; Güttlich, P. *Chem.*  
47 *Eur. J.* **2000**, *6*, 552–563.  
48 <sup>4</sup> (a) Bénard, S.; Yu, P.; Audière, J. P.; Rivière, E.; Clément, R.; Ghilhem, J.; Tchertanov, L.; Nakatami, K.  
49 *J. Am. Chem. Soc.* **2000**, *122*, 9444–9455. (b) Aldoshin, S. M.; Sanina, N. A.; Minkin, V. I.; Voloshin, N.  
50 A.; Ikorskii, V. N.; Ovcharenko, V. I.; Smirnov, V. A.; Nagaeva, N. K. *J. Mol. Struct.* **2007**, *826*, 69–74.  
51 <sup>5</sup> (a) Coronado, E.; Galán-Mascarós, J. R.; Gómez-García, C. J.; Lauhkin, V. *Nature* **2000**, *408*, 447–449.  
52 (b) Alberola, A.; Coronado, E.; Galán-Mascarós, J. R.; Giménez-Saiz, C.; Gómez-García, C. J. *J. Am. Chem.*  
53 *Soc.* **2003**, *125*, 10774–10775. (c) Galán-Mascarós, J. R.; Coronado, E.; Goddard, P. A.; Singleton, J.;  
54 Coldea, A. I.; Wallis, J. D.; Coles, S. J.; Alberola, A. *J. Am. Chem. Soc.* **2010**, *132*, 9271–9273. (d)  
55 Coronado, E.; Galán-Mascarós, J. R.; Gómez-García, C. J.; Martínez-Ferrero, E.; Van Smaalen, S. *Inorg.*  
56 *Chem.* **2004**, *43*, 4808–4810. (e) Zhang, B.; Zhang, Y.; Zhu, D. *Chem. Commun.* **2012**, *48*, 197–199.  
57  
58  
59  
60

- 1  
2  
3
- 
- 4 <sup>6</sup> (a) Okawa, H.; Shigematsu, A.; Sadakiyo, M.; Miyagawa, T.; Yoneda, K.; Ohba, M.; Kitagawa, H. *J. Am.*  
5 *Chem. Soc.* **2009**, *131*, 13516–13522. (b) Pardo, E.; Train, C.; Contard, G.; Boubekour, K.; Fabelo, O.;  
6 Liu, H.; Dkhil, B.; Lloret, F.; Nakagawa, K.; Tokoro, H.; Ohkoshi, S.-I.; Verdaguer, M. *J. Am. Chem. Soc.*  
7 **2011**, *133*, 15328–15331. (c) Sadayiko, M.; Okawa, H.; Shigematsu, A.; Ohba, M.; Yamada, T.; Kitagawa,  
8 H. *J. Am. Chem. Soc.* **2012**, *134*, 5472–5475. (d) Okawa, H.; Sadakiyo, M.; Yamada, T.; Maesato, M.;  
9 Ohba, M.; Kitagawa, H. *J. Am. Chem. Soc.* **2013**, *135*, 2256–2262.
- 10 <sup>7</sup> (a) Endo, T.; Akutagawa, T.; Noro, S. I.; Nakamura, T. *Dalton Trans.* **2011**, *40*, 1491–1496. (b) Pardo,  
11 E.; Train, C.; Liu, H.; Chamoreau, L.-M.; Dkhil, B.; Boubekour, K.; Lloret, F.; Nakatani, K.; Tokoro, H.;  
12 Ohkoshi, S.-i.; Verdaguer, M. *Angew. Chem., Int. Ed.* **2012**, *51*, 8356–8360.
- 13 <sup>8</sup> (a) Andrés, R.; Gruselle, M.; Malézieux, B.; Verdaguer, M.; Vaissermann, J. *Inorg. Chem.* **1999**, *38*,  
14 4637–4646. (b) Andrés, R.; Brissard, M.; Gruselle, M.; Train, C.; Vaissermann, J.; Malézieux, B.; Jamet,  
15 J. P.; Verdaguer, M. *Inorg. Chem.* **2001**, *40*, 4633–4640. (c) Clemente-León, M.; Coronado, E.; Dias, J.  
16 C.; Soriano-Portillo, A.; Willett, R. D. *Inorg. Chem.* **2008**, *47*, 6458–6463. (d) Train, C.; Gheorghe, R.;  
17 Krstic, V.; Chamoreau, L. M.; Ovanesyan, N. S.; Rikken, G. L. J. A.; Gruselle, M.; Verdaguer, M. *Nat.*  
18 *Mater.* **2008**, *7*, 729–734. (e) Train, C.; Nuida, T.; Gheorghe, R.; Gruselle, M.; Ohkoshi, S. *J. Am. Chem.*  
19 *Soc.* **2009**, *131*, 16838–16843. (f) Gruselle, M.; Li, Y.; Ovanesyan, N.; Markhaev, V.; Shilov, G.;  
20 Mushenok, F.; Train, C.; Aldoshin, S. *Chirality* **2013**, *25*, 444–448.
- 21 <sup>9</sup> Clemente-León, M.; Coronado, E.; Gómez-García, C. J.; López-Jordá, M.; Camón, A.; Repollés, A.; Luis,  
22 F. *Chem.-Eur. J.* **2014**, *20*, 1669–1676.
- 23 <sup>10</sup> (a) Kitagawa, S.; Kawata, S. *Coord. Chem. Rev.*, **2002**, *224*, 11–34. (b) Mercuri, M. L.; Congiu, F.;  
24 Concas, G.; Ashoka Sahadevan, S. *Magnetochemistry*, **2017**, *3*, 17 (upon invitation for the *Special Issue*  
25 *Magnetism of Molecular Conductors*, Guest Editor: M. Almeida).
- 26 <sup>11</sup> (a) Atzori, M.; Benmansour, S.; Minguez Espallargas, G.; Clemente-León, M.; Abhervé, A.; Gómez-  
27 Claramunt, P.; Coronado, E.; Artizzu, F.; Sessini, E.; Deplano, P.; Serpe, A.; Mercuri, M. L.; Gómez García,  
28 C. J. *Inorg. Chem.* **2013**, *52*, 10031–10040. (b) Atzori, M.; Pop, F.; Auban-Senzier, P.; Gómez García, C.  
29 J.; Canadell, E.; Artizzu, F.; Serpe, A.; Deplano, P.; Avarvari, N.; Mercuri, M. L. *Inorg. Chem.* **2014**, *53*,  
30 7028–7039. (c) Abhervé, A.; Clemente-León, M.; Coronado, E.; Gómez-García, C. J.; Verneret, M. *Inorg.*  
31 *Chem.* **2014**, *53*, 12014–12026. (d) Abhervé, A.; Mañas-Valero, S.; Clemente-León, M.; Coronado, E.  
32 *Chem. Sci.* **2015**, *6*, 4665–4673. (e) Atzori, M.; Pop, F.; Auban-Senzier, P.; Clérac, R.; Canadell, E.;  
33 Mercuri, M. L.; Avarvari, N. *Inorg. Chem.* **2015**, *54*, 3643–3653. (f) Benmansour, S.; Gómez García, C. J.  
34 *Polymers* **2016**, *8*, 89. (g) Benmansour, S.; Abhervé, A.; Gómez-Claramunt, P.; Vallés-García, C.; Gómez-  
35 García, C. J. *ACS Appl. Mater. Interfaces* **2017**, *9*, 26210–26218. (h) Palacios-Corella, M.; Fernández-  
36 Espejo, A.; Bazaga-García, M.; Losilla, E. R.; Cabeza, A.; Clemente-León, M.; Coronado, E. *Inorg. Chem.*  
37 **2017**, *56*, 13865–13877. (i) Kingsbury, C. J.; Abrahams, B. F.; D'Alessandro, D. M.; Hudson, T. A.;  
38 Murase, R.; Robson, R.; White, K. F. *Cryst. Growth Des.* **2017**, *17*, 1465–1470. (j) DeGayner, J. A.; Jeon,  
39 I.-R.; Sun, L.; Dinca, M.; Harris, T. D. *J. Am. Chem. Soc.* **2017**, *139*, 4175–4184. (k) Benmansour, S.; Pérez-  
40 Herráez, I.; López-Martínez, G.; Gómez García, C. J. *Polyhedron* **2017**, *135*, 17–25. (l) Benmansour, S.;  
41 Hernández-Paredes, A.; Gómez-García, C. J. *J. Coord. Chem.* **2018**, *71*, 845.
- 42 <sup>12</sup> (a) Imaz, I.; Mouchaham, G.; Roques, N.; Brandès, S.; Sutter, J.-P. *Inorg. Chem.* **2013**, *52*,  
43 11237–11243. (b) Benmansour, S.; Vallés-García, C.; Gómez-Claramunt, P.; Minguez Espallargas, G.;  
44 Gómez García, C. J. *Inorg. Chem.* **2015**, *54*, 5410–5418. (c) Sumida, K.; Hu, M.; Furukawa, S.; Kitagawa,  
45 S. *Inorg. Chem.* **2016**, *55*, 3700–3705. (d) Murase, R.; Abrahams, B. F.; D'Alessandro, D. M.; Davies, C.  
46 G.; Hudson, T. A.; Jameson, G. N. L.; Moubaraki, B.; Murray, K. S.; Robson, R.; Sutton, A. L. *Inorg. Chem.*  
47 **2017**, *56*, 9025–9035.
- 48 <sup>13</sup> Taniguchi, K.; Chen, J.; Sekine, Y.; Miyasaka, H. *Chem. Mater.* **2017**, *29*, 10053–10059.
- 49 <sup>14</sup> (a) Jeon, I.-R.; Negru, B.; Van Duyne, R. P.; Harris, T. D. *J. Am. Chem. Soc.* **2015**, *137*, 15699–15702.  
50 (b) Darago, L. E.; Aubrey, M. L.; Yu, C. J.; Gonzalez, M. I.; Long, J. R. *J. Am. Chem. Soc.* **2015**, *137*,  
51 15703–15711. (c) DeGayner, J. A.; Jeon, I.-R.; Sun, L.; Dincă, M.; Harris, T. D. *J. Am. Chem. Soc.* **2017**,  
52 *139*, 4175–4184.
- 53 <sup>15</sup> (a) Rehwoldt, R. E.; Chasen, B. L.; Li, J. B. *Anal. Chem.* **1966**, *8*, 1018–1019. (b) Atzori, M.; Pop, F.;  
54 Cauchy, T.; Mercuri, M. L.; Avarvari, N. *Org. Biomol. Chem.* **2014**, *12*, 8752–8763.

- 1  
2  
3  
4  
5  
6  
7  
8  
9  
10  
11  
12  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60
- <sup>16</sup> Atzori, M.; Artizzu, F.; Marchiò, L.; Loche, D.; Caneschi, A.; Serpe, A.; Deplano, P.; Avarvari, N.; Mercuri, M. L. *Dalton Trans.* **2015**, *44*, 15786–15802.
- <sup>17</sup> Ashoka Sahadevan, S.; Monni, N.; Abhervé, A.; Auban-Senzier, P.; Canadell, E.; Mercuri, M. L.; Avarvari, N. *Inorg. Chem.* **2017**, *56*, 12564–12571.
- <sup>18</sup> Ashoka Sahadevan, S.; Monni, N.; Abhervé, A.; Marongiu, D.; Sarritzu, V.; Sestu, N.; Saba, M.; Mura, A.; Bongiovanni, G.; Quochi, F.; Avarvari, N.; Mercuri, M. L. *Chem. Mater.* **2018**, *under revision*.
- <sup>19</sup> Weiss, S.; Krommer, H. *Chem. Abstr.* **1986**, *104*, 206730.
- <sup>20</sup> Tao, J. M.; Perdew, J. P.; Staroverov, V. N.; Scuseria, G. E. *Phys. Rev. Lett.*, **2003**, *91*, 146401.
- <sup>21</sup> Jensen, K. P. *Inorg. Chem.* **2008**, *47*, 10357–10365.
- <sup>22</sup> Hariharan, P. C.; Pople, J. A. *Theor. Chim. Acta* **1973**, *28*, 213–222.
- <sup>23</sup> Gaussian 09, Revision B1, Frisch, M. J.; Trucks, G. W.; Schlegel, H. B.; Scuseria, G. E.; Robb, M. A.; Cheeseman, J. R.; Scalmani, G.; Barone, V.; Mennucci, B.; Petersson, G. A.; Nakatsuji, H.; Caricato, M.; Li, X.; Hratchian, H. P.; Izmaylov, A. F.; Bloino, J.; Zheng, G.; Sonnenberg, J. L.; Hada, M.; Ehara, M.; Toyota, K.; Fukuda, R.; Hasegawa, J.; Ishida, M.; Nakajima, T.; Honda, Y.; Kitao, O.; Nakai, H.; Vreven, T.; Montgomery Jr., J. A.; Peralta, J. E.; Ogliaro, F.; Bearpark, M.; Heyd, J. J.; Brothers, E.; Kudin, K. N.; Staroverov, V. N.; Kobayashi, R.; Normand, J.; Raghavachari, K.; Rendell, A.; Burant, J. C.; Iyengar, S. S.; Tomasi, J.; Cossi, M.; Rega, N.; Millam, J. M.; Klene, M.; Knox, J. E.; Cross, J. B.; Bakken, V.; Adamo, C.; Jaramillo, J.; Gomperts, R.; Stratmann, R. E.; Yazyev, O.; Austin, A. J.; Cammi, R.; Pomelli, C.; Ochterski, J. W.; Martin, R. L.; Morokuma, K.; Zakrzewski, V. G.; Voth, G. A.; Salvador, P.; Dannenberg, J. J.; Dapprich, S.; Daniels, A. D.; Farkas, Ö.; Foresman, J. B.; Ortiz, J. V.; Cioslowski, J.; Fox, D. J. Gaussian, Inc., Wallingford CT, **2009**.
- <sup>24</sup> (a) Tamaki, H.; Zhong, Z. J.; Matsumoto, N.; Kida, S.; Koikawa, M.; Achiwa, N.; Hashimoto, Y.; Okawa, H. *J. Am. Chem. Soc.* **1992**, *114*, 6974–6979. (b) Clemente-León, M.; Coronado, E.; López-Jordà, M.; Mínguez Espallargas, G.; Soriano-Portillo, A.; Waerenborgh, J. C. *Chem. Eur. J.* **2010**, *16*, 2207–2219.
- <sup>25</sup> Okaya, Y.; Pepinsky, R. *Acta Cryst.* **1957**, *10*, 681–684.
- <sup>26</sup> Hartl, F.; Stufkens, D. J.; Vlček, A. *Inorg. Chem.* **1992**, *31*, 1687–1695.
- <sup>27</sup> Tholence, J. L. *Sol. State Comm.* **1980**, *35*, 113–117.
- <sup>28</sup> (a) Novak, M. A. *J. Magn. Magn. Mater.* **2004**, *272-276*, e707. (b) Novak, M. A.; Folly, W. S. D.; Sinnecker, J. P.; Soriano, S. J. *Magn. Magn. Mater.* **2005**, *294*, 133–140.
- <sup>29</sup> (a) Gunnarsson, K.; Svedlindh, P.; Nordblad, P.; Lundgren, L.; Aruga, H.; Ito, A. *Phys. Rev. Lett.* **1988**, *61*, 754. (b) Laiho, R.; Lähderanta, E.; Salminen, J.; Lisunov, K. G.; Zakhvalinskii, V. S. *Phys. Rev. B* **2001**, *63*, 094405. (c) Mattsson, J.; Jonsson, T.; Nordblad, P.; Aruga Katori, H.; Ito, A. *Phys. Rev. Lett.* **1995**, *74*, 4305.
- <sup>30</sup> (a) Sellers, S. P.; Korte, B. J.; Fitzgerald, J. P.; Reiff, W. M.; Yee, G. T. *J. Am. Chem. Soc.* **1998**, *120*, 4662–4670. (b) Kaul, B. B.; Durfee, W. S.; Yee, G. T. *J. Am. Chem. Soc.* **1999**, *121*, 6862–6866. (c) Buschmann, W. E.; Enslin, J.; Gütlich, P.; Miller, J. S. *Chem.–Eur. J.* **1999**, *5* (10), 3019–3028. (d) Clérac, R.; O’Kane, S.; Cowen, J.; Ouyang, X.; Heintz, R.; Zhao, H.; Bazile, M. J.; Dunbar, K. R. *Chem. Mater.* **2003**, *15*, 1840–1850. (e) Krishnamohan Sharma, C. V.; Chusuei, C. C.; Clérac, R.; Möller, T.; Dunbar, K. R.; Clearfield, A. *Inorg. Chem.* **2003**, *42*, 8300–8308. (f) Zhang, X.-M.; Peng, L.; Gao, W.; Liu, J.-P.; Gao, E.-Q. *Dalton Trans.* **2015**, *44*, 511–514. (g) Gosh, S.; Roy, S.; Liu, C.-M.; Mohanta, S. *Dalton Trans.* **2018**, *47*, 836–844.
- <sup>31</sup> Greenwood, N.N.; Gibb, T.C. "Mössbauer Spectroscopy" Chapman and Hall, Ltd. Publishers, London, **1971**.
- <sup>32</sup> Shilov, G. V.; Nikitina, Z. K.; Ovanesyan, N. S.; Aldoshin, S. M.; Makhaev, V. D. *Russ. Chem. Bull.* **2011**, *60*, 1209–1219.
- <sup>33</sup> e.g. (a) Lupu, D.; Barb, D.; Filoti, G.; Morariu, M.; Tarin, D. *J. Inorg. Nucl. Chem.* **1972**, *34*, 2803–2810. (b) Maeda Y. *J. Nucl. Radiochem. Sci.* **2006**, *7*, R13–R18.
- <sup>34</sup> Bhattacharjee, A.; Bhakat, D.; Roy, M.; Kusz, J. *Physica B* **2010**, *405*, 1546–1550.
- <sup>35</sup> For a very pedagogic presentation of the basic methodology see: Launay, J.-P.; Verdaguer, M. "Electrons in Molecules: From Basic Principles to Molecular Electronics", Oxford University Press, Oxford, **2014**, chapter 3.
- <sup>36</sup> Austin, I. G.; Mott, N. F. *Adv. Phys.* **1969**, *18*, 41–102.

1  
2  
3  
4  
5  
6  
7  
8  
9  
10  
11  
12  
13  
14  
15  
16  
17  
18  
19  
20  
21  
22  
23  
24  
25  
26  
27  
28  
29  
30  
31  
32  
33  
34  
35  
36  
37  
38  
39  
40  
41  
42  
43  
44  
45  
46  
47  
48  
49  
50  
51  
52  
53  
54  
55  
56  
57  
58  
59  
60

---

<sup>37</sup> Tsuda, N.; Nasu, K.; Yanase, A.; Siratori, K. "Electronic conduction in Oxides", Springer Series in Solid-State Sciences, Springer Verlag, Berlin Heidelberg, **1991**, pp. 163–168.