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1 The mixed impact of nanoclays on the apparent diffusion coefficient of additives in

2 biodegradable polymers in contact with food

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12 Abstract

In face of growing environmental concerns, biodegradable and bio-sourced plastic nanocomposites are emerging as a new class of materials, especially for the food packaging sector. However, their use in food contact raises new issues in term of consumer safety.

The aim of this study was to determine the impact of nanoclays on the apparent diffusion coefficient (D_{app}) of selected additives from biopolymers into fatty food simulants. For the most part, nanoclay addition has a non-conventional impact. To understand this, the following parameters were studied: (i) the exfoliation state of nanoclay platelets in the polymer matrix, (ii) the sorption of food simulant by the polymer matrix, and (iii) the crystallinity of the materials. At first glance, solvent uptake and crystallinity agree with the results of diffusivity, however these parameters cannot explain the extreme differences between D_{app} values.

23 Keywords: nanoclays, biopolymer, apparent diffusion coefficient (D_{app}), additives, food
24 packaging

25 **1 Introduction**

The growing global production and consumption of plastic materials engenders serious 26 environmental concerns (Kirwan and Strawbridge, 2003), primarily the need to implement eco-27 efficient end-of-life treatments. For many years, alternative materials based on bio-sourced and 28 biodegradable polymers have been considered as innovative and promising materials for future 29 food packaging applications (Tharanathan, 2003; Sorrentino et al., 2007). However, the 30 substitution of plastic in all its various uses in food packaging remains difficult given the broad 31 spectrum of functionality offered by petroleum-based polymers. In addition to their high water 32 sensitivity, the properties of biodegradable materials make them uncompetitive in comparison 33 with those of conventional plastic polymers, particularly in terms of barrier properties. The 34 35 addition of reinforcing fillers such as clay nanoparticles, giving rise to the formation of a bio-36 nanocomposite material, is widely considered as the solution to this problem (Perumal et al., 2018). 37

38 The introduction of nanotechnologies to the field of food packaging raises new issues related to consumer safety. Research has focused on the exposure of consumers to nanoparticles 39 through contamination of the food in contact with nanocomposite materials. There is a 40 41 consensus in the research on the migration of nanoclay that its limited diffusion under a nanoform in a polymeric matrix prevents its migration into food (Šimon et al., 2008). Like any 42 material in contact with food, nano- and bio-nanocomposite materials are subject to the 43 European framework regulation UE 1935/2004. In the absence of specific regulations covering 44 their use in contact with food, the suitability for food contact applications of nanocomposite 45 packaging are assessed according to the recommendations established for plastic materials 46 specified in the regulation EU/10/2011. 47

48 Given the impact of nanoclays on the transfer properties of materials (Muñoz-Shugulí et 49 al., 2019), it is logical to investigate their impact on the migration of property-enhancing

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plastics additives such as plasticizers, UV-stabilizers, anti-oxidants, and anti-static agents 50 (Figge, 1980). The impact of nanoparticles on the potential migration of low molar mass 51 additives has received little attention, the only studies having focused on plastic-based 52 nanocomposite materials. These studies, performed on nanocomposite materials, specifically: 53 PETs (Farhoodi et al., 2016), polyamides (Pereira de Abreu et al., 2010), and polyolefins 54 (Otero-Pazos et al. 2016; Nasiri et al. 2016; Nasiri et al. 2017), conclude that nanoparticles 55 decrease the diffusivity of low molar mass substances. Such results suggest the applicability of 56 the Piringer model in a migration prediction approach based on an overestimated value of 57 diffusivity. 58

59 From a safety standpoint, the barrier effect provided by the nanostructure of packaging film against additives could reduce the exposure of consumers to toxic compounds and their 60 potential adverse health effects, depending on the individual barrier properties of polymers. 61 62 Although nanoclays have a positive effect on high barrier polymers such as PET, the decrease in the diffusion coefficient has no effect on the migration value for low-barrier polymers, for 63 64 which the migration values are more related to the partition coefficient (Farhoodi et al., 2016). The reasons why the presence of nanoparticles decreases the mobility of the diffusing molecules 65 are multiple. While the tortuosity effect is generally cited (Paul and Robeson, 2008; Duncan, 66 2011), the mechanisms of diffusion in nanocomposite matrices imply additional essential 67 factors. In the case of polymer-clay nanocomposites, the interactions of organic molecules with 68 the clay's minerals are likely to generate a decrease in their D_{app} by sorption or adsorption 69 mechanisms (Nasiri et al. 2016). In addition to their high cation exchange capacity, clays can 70 71 exchange other chemical bonds, such as van der Walls interactions, or hydrogen bonding, with migrants, delaying their transport in the nanocomposite structure (Aguzzi et al., 2007). It is also 72 well established that the presence of clay nanoparticles structurally modifies polymer networks, 73 having nucleating effects, and their ability to increase the crystallization rate of semi-crystalline 74

thermoplastic polymers (Ke et al., 1999; Yuan et al., 2006). At a finer scale, crystallinity 75 gradients in the vicinity of nanoparticles can have a modulating effect on the transfer properties 76 of plastic-based nanocomposites (Wurm et al., 2010). This combination of effects emphasized 77 nanocomposite bioplastics (Charlon et al. 2015b), but to our knowledge, no research has 78 measured and characterized migration from bio-nanocomposite packaging. Such materials, by 79 definition unstable, may behave differently than synthetic polymers in contact with food, and 80 their inertial properties remain undemonstrated. This study aims to characterize the food contact 81 suitability of bio-nanocomposite packaging films. In view of their emergence on the market and 82 their environmental benefits (Rajan et al., 2018; Hongsriphan and Pinpueng, 2019), two distinct 83 polymers: poly(butylene succinate-co-butylene adipate) (PBSA) and poly(hydroxybutyrate-co-84 hydroxyvalerate) (PHBV) were selected on the basis of their differing transfer properties, in 85 order to cover different behaviors in contact with food simulants, the PHBV being a high-barrier 86 87 polymer (Crétois et al., 2014) contrary to PBSA (Phua et al., 2013). Organo-modified montmorillonite was added to these materials, which were spiked with a panel of target 88 89 migrating substances and submitted to migration tests following the protocol for testing 90 conventional plastic packaging. The approach aims to measure and identify the influence of nanoclays on the diffusion properties of these emerging packaging materials. 91

92

2 Materials and Methods

93 2.1 Materials

Poly(butylene succinate-co-butylene adipate) (PBSA, PBE 001, density 1.24 g.cm⁻³) and
Poly(hydroxybutyrate-co-hydroxyvalerate) (PHBV, PHI 002, density 1.23 g.cm⁻³) pellets were
commercially procured from Natureplast, France.

97 Cloisite 30B (C-30B), an organo-modified montmorillonite clay containing a methyl bis98 2-hydroxyethyl ammonium quaternary salt with a cation exchange capacity (CEC) of 90

meq/100 g, were supplied by BYK additives & instruments, Germany. C-30B was chosen
because of its favorable interaction with PBSA (Sinha Ray et al., 2008) and PHBV (Carli et al.,
2011; Iggui et al., 2015).

An array of solid and liquid additives was chosen. These molecules represent a range of categories of chemical compounds with a variety of chemical and physical properties as described in FDA and EFSA regulations (Food and Drug Administration, 2006; European Food Safety Authority, 2011). The additives chosen include: volatile polar organic substances, volatile non-polar organic substances, non-volatile polar organic substances, and non-volatile non-polar organic substances. Three high molar mass additives and eight low molar mass additives were procured from Sigma-Aldrich (Table 1).

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2.2

Preparation of bio-nanocomposite films

PBSA and PHBV films with 5 wt% C-30B were synthetized by a melt extrusion process using an EuroLab 16XL co-rotating twin-screw extruder (Thermo ScientificTM, Germany) with a L/D ratio of 40 and a screw diameter of 16 mm. The extruder produced films using a calendar die. Polymer pellets were dried at 60°C for 15 hours before use to remove residual moisture. Nanoclays were used without drying, because hydrated nanoclay platelets intercalate better with the polymer matrix (Tenn et al., 2013).

PBSA films were produced in three stages (Charlon et al. 2015a). First, a masterbatch of 15 wt% C-30B was prepared. The polymer pellets and nanoclays were introduced in the extruder at a flow rate of 0.85 kg.h⁻¹ and 0.15 kg.h⁻¹, respectively. The feed to die temperature profile was 120°C to 160°C. At the extrusion die, the emerging polymers were cooled in a water bath and pelletized. The masterbatch was then diluted with neat polymer to obtain 5 wt% nanoclay pellets. Finally, films of 180 \pm 10 μ m for PBSA and 220 \pm 25 μ m for PBSA nanocomposite (PBSA NCP) were produced from these pellets using a flow rate of 1.0 kg.h⁻¹ and a 100°C to 135°C temperature profile. All steps were processed at a screw speed of 200
rpm.

125 PHBV films were produced in two stages (Iggui et al., 2015). A masterbatch of 15 wt% 126 C-30B was prepared in the same way as for PBSA, however with a temperature profile of 140°C 127 to 180°C. The masterbatch was diluted with neat polymer to obtain PHBV films containing 128 5 wt% nanoclay. The films of $145 \pm 20 \,\mu\text{m}$ for PHBV and $140 \pm 10 \,\mu\text{m}$ for PHBV NCP were 129 produced using a screw speed of 210 rpm and a flow rate of 1.0 kg.h⁻¹. PBSA and PHBV pellets 130 or films were oven dried at 60°C for 15 hours after each stage and before storage.

131 The control samples of neat PBSA and PHBV were prepared following the same protocol132 used for nanocomposites in order to respect the same thermal processes.

133 2.3 Spiking of neat and nanocomposite films

The procedure used for the spiking of films depends on additive molar mass. High molar mass additives were introduced to films during the extrusion process because of their excellent thermal stability (Mauricio Iglesias, 2009). The theoretical additive amount was 500 ppm. Low molar mass additives were introduced to films through contamination at 40°C for 7 days under rotary agitation. The theoretical additive amount was 500 ppm for all low molar mass additives apart from toluene at 1000 ppm.

140 **2.4** Food simulants

Two different food simulants were chosen to evaluate the D_{app} of the selected additives: 95% ethanol, and iso-octane, which both simulate fatty foods behaviors according to the European Commission for food contact materials (EU 10/2011) (Commission, 2011). The experiments were also performed using 3% acetic acid to simulate acidic food behavior, but the migration of high molar mass additives in this solvent was less than the limit of detection.

146 **2.5 Desorption tests**

Desorption tests were carried out by putting contaminated films in contact with food simulants in a surface/volume ratio of 6 dm².L⁻¹. The bottles were stored under magnetic agitation at 40°C for 10 days, except for PHBV films contaminated by high molar mass additives, for which storage was increased to 30 days. Over specific periods of time, an aliquot of each food simulant was collected and analyzed by chromatography (GC-FID for low molar mass additives or HPLC for high molar mass additives). The samples of iso-octane were evaporated under nitrogen and re-dissolved in 100% ethanol before HPLC analysis.

154

2.6 Small angle X-ray diffraction (SAXS)

SAXS experiments were performed using an "in-house" setup to study the degree of 155 exfoliation of the nanoclays in the polymer matrix. A high brightness low power X-ray tube, 156 coupled with aspheric multilayer optic (GeniX^{3D} from Xenocs) delivered an ultralow divergent 157 beam (0.5 mrad; flux: 20 MPhotons.s¹; $\lambda = 1.5418$ Å; size at sample: 0.6*0.6 mm). A 158 transmission configuration was used and the scattered intensity determined by a 2D pixel 159 "Pilatus" detector at a distance of 1.9 m from the sample. Glass capillary support was used in 160 the case of nanoclay powder, while for films, the beam passed through five stacked film 161 thicknesses. The obtained intensities were corrected by transmission and the empty cell 162 contribution was subtracted. The interlayer spacing relative to nanoclay platelets was 163 determined at a diffraction angle in the range of 0.5° to 10° . 164

165

55 **2.7 Transmission electron microscopy (TEM)**

166 A JEOL 1200EX2 transmission electron microscope with 100 kV acceleration voltage 167 equipped with an EMSIS Olympus camera was used to analyze film samples and observe nanoclay dispersion and exfoliation in the polymer matrix. For that, the film samples wereincluded in a resin and cut with a Leica UC 7 ultramicrotome.

170 **2.8** Differential scanning calorimetry (DSC)

184

A differential scanning calorimeter (DSC series Q200, TA Instruments) was used to study 171 the crystallinity of the films. Experiments were carried out on neat and nanocomposite films, 172 before and after 10 days of contact with 95% ethanol or iso-octane. All measurements were 173 174 performed under a nitrogen atmosphere. Specimens of 5-8 mg, weighed using a microbalance (Sartorius micro PRO 11), were placed in aluminum sample pans. The thermal protocol was 175 defined according to the polymer. The PBSA samples were: heated from ambient temperature 176 to 40°C, cooled to - 35°C, heated to 150°C, maintained at 150°C for 5 min, cooled to - 35°C, 177 heated to 150°C, and finally cooled to 40°C. In the case of PHBV, samples were: heated from 178 179 ambient temperature to 40°C, cooled to - 30°C, heated to 200°C, maintained at 200°C for 5 min, cooled to - 30°C, reheated to 200°C, and finally cooled to 40°C. Temperatures were 180 adjusted at a rate of 10°C.min⁻¹. The apparent melting enthalpy (Δ Hm) was determined from 181 the DSC curves using the Universal Analysis 2000 software by TA Instruments. The 182 crystallinity degree (X_C) of neat and nanocomposite polymers was determined by the equation: 183

$$X_{C}(\%) = \frac{\Delta H_{m}}{\Delta H_{m}^{0}} \times \frac{100}{w}$$
(1)

185 Where: Δ Hm is the melting enthalpy of the polymer matrix, w is the polymer weight fraction 186 in the sample and Δ H⁰_m is the theoretical melting enthalpy of the polymer assumed to be 100% 187 crystalline, Δ H⁰_m = 146.0 J.g⁻¹ for PHBV (Barham et al., 1984) and Δ H⁰_m = 116.9 J.g⁻¹ for PBSA 188 (Nikolic and Djonlagic, 2001).

189 2.9 Sorption of food simulants

195

The food simulant uptake by the films was determined by gravimetric measurements using a precision balance (10⁻⁴ g) before and after subjecting films to contact with food simulants at 40°C for 10 days under magnetic agitation. At the end of this procedure, excess food simulant was removed with tissue paper before re-measuring. The percentage of solvent uptake was obtained by the following equation:

Sorption =
$$\frac{m_t - m_i}{m_i} \times 100$$
 (2)

196 Where: m_i and m_t are weights of samples before and after sorption, respectively.

197 **2.10** Gas chromatography with flame ionization detector (GC-FID)

An Agilent 7890A GC equipped with a 7693A Automatic Sampler and a flame ionization 198 detector was used to determine the concentration of low molar mass additives ($< 400 \text{ g.mol}^{-1}$) 199 200 in food simulants. The components were separated using a HP-5 (5%-201 phenyl)methylpolysiloxane capillary column of 32 mm ID and 30 m length. The thermal protocol used for the oven was as follows: an initial temperature of 40°C maintained for 5 min, 202 heated at a rate of 6°C.min⁻¹ to 270°C, maintained for 15 min. The injector temperature was 203 250°C. An external calibration was made in each food simulant in the range of 1.25 -204 40 mg.L^{-1} . 205

206 2.11 High performance liquid chromatography (HPLC)

207 An Alliance HPLC system (Waters) equipped with an Alltima C18 column (250 mm × 208 2.1 mm, 5- μ m) protected with a Alltima C18 (7.5mm × 2.1 mm, 5- μ m) column guard was used 209 to determine the concentration of high molar mass additives in food simulants. Compounds 210 were separated with 55% acetonitrile at a flow rate of 0.3 mL.min⁻¹ at 50°C. Absorbance was recorded at 230 nm to quantify the additive concentration. An external calibration was made in each food simulant with concentrations of 1.25 to 20 mg.L⁻¹.

213 2.12 Estimation of the Dapp

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2.12.1 Fick's second diffusion equation

The D_{app} of additives from the polymer films into food simulant was calculated from experimental desorption kinetic curves using Fick's second law. The films are considered as one-dimensional infinite plane sheets, given that diffusion via the borders of the films is negligible. Applying Fick's second law results in the following one dimensional diffusion equation:

220
$$\frac{\partial C}{\partial t} = D_{app} \left(\frac{\partial^2 C}{\partial x^2}\right)$$
(3)

Where: C is the additive concentration, t, the contact time, x, the position in the film and D_{app},
the apparent diffusion coefficient.

223 The Eq. (3) was solved using the following initial (Eq. 4) and boundary (Eq. 5) conditions:

224
$$C(t = 0, x) = C_0 \forall x \in [-L, L]$$
 (4)

225

$$C(t, x = \pm L) = C_s(t) \forall t \ge 0$$
(5)

226 Where: $C_s(t)$ is the additive concentration in food simulant at time t and L, the film half 227 thickness.

This diffusion model can be used provided that the following assumptions are verified (Helmroth et al., 2002): (i) additive contamination of the film is homogeneous, (ii) there is no noticeable swelling caused by interaction between the food simulant and the polymer, (iii) there is no concentration gradient in the food simulant, and (iv) the film thickness is homogeneous. In the case of an infinite plane sheet suspended in a stirred solution of limited volume, the Eq. (3) can be solved analytically as described by Crank (Crank, 1975):

234
$$\frac{M_t}{M_{\infty}} = 1 - \sum_{n=0}^{\infty} \frac{2 \alpha (1+\alpha)}{1+\alpha + \alpha^2 q_n^2} \exp\left\{-\frac{D q_n^2 t}{L^2}\right\}$$
(6)

235 With:

236

$$\alpha = \frac{1}{K_{P,F}} \frac{V_F}{V_P} \tag{7}$$

Where: Mt is the total amount of additives in food simulant at time t and M_{∞} is the total amount of additives in food simulant at the steady state, V_P is the polymer volume and V_F the food simulant volume, $(q_n)_n$, the positive roots of the equation $\tan q = -\alpha q$ and $K_{P,F}$, the partition coefficient of the additive in the polymer/food simulant system.

The numerical simulations were carried out using Matlab[®] software and its *lsqnonlin* function to estimate the D_{app} . To evaluate the correlation between experimental and estimated data, the percentage of the root mean-square error (% RMSE) was calculated using the following equation:

245
$$RMSE = \frac{1}{M_0} \sqrt{\frac{1}{N} \sum_{i=1}^{N} ((M_t)_{experimental} - (M_t)_{predicted})^2} \times 100$$
(8)

246 Where: M_0 is the initial mass of additive in the biopolymer film and M_t is the mass of additive 247 in the food simulant at time t.

248 **2.12.2 Piringer correlation**

The Piringer correlation (1994) estimates the worst case additive diffusivity in a given polymer by correlating the diffusion coefficient with the molar mass of additives and the temperature. D_P (Eq. 9) can be refined to not depend on experimental data (Brandsch et al., 2002).

253
$$D_{\rm P} = D_0 \exp\left[A_{\rm P} - 0.1351 \,\,{\rm M}^{\frac{2}{3}} + 0.003 \,\,{\rm M} - \frac{10454}{\rm T}\right] \tag{9}$$

254 With: $D_0 = 10^4 \text{ cm}^2 \text{.s}^{-1} = 1 \text{ m}^2 \text{.s}^{-1}$ and $A_P = A'_P - \frac{\tau}{T}$

Where the parameter A_P is linked to the polymer and describes the basic diffusion behavior of the polymer matrix in relation to the migrants, M is the molar mass of the additive, T is the absolute temperature and τ is a polymer specific "activation temperature" increment. The apostrophe ' indicates the parameter is temperature independent.

The parameters A'_P and τ are given in Table 2 (Begley et al., 2005). In the absence of specific parameters for PBSA and PHBV, those for LDPE and PET were used for, respectively, low and high barrier property polymers.

262 **3 Results and discussion**

263 **3.1** Microstructure of composite material

Because the interaction between polymers and nanoclays determines the formation of exfoliated nanocomposite materials, clay modification has been extensively studied (Muñoz-Shugulí et al., 2019). Exfoliated nanoparticles reduce diffusion by creating tortuosity effect and results in higher transfer properties (Nasiri et al. 2016).

The degree of exfoliation of nanoclay platelets in biopolymer matrices was measured by 268 XRD (Fig. 1 A) and TEM (Fig. 1 B). As expected, the analysis of the C-30B powder by XRD 269 revealed a diffraction peak at 4.9°, corresponding to an interlayer spacing of 1.8 nm (Bharadwaj 270 et al., 2002). The analysis of neat biopolymers exhibited no diffraction peaks in the studied 2Θ 271 angle range, while the analysis of nanocomposites revealed one diffraction peak with a shift 272 towards the smallest angles compared to C-30B powder, 2.7° for PBSA NCP and 2.2° for 273 PHBV NCP, which respectively corresponds to a d-spacing of 3.3 nm and 4.0 nm. Similar 274 values were reported for PBSA NCP (Charlon et al., 2015a) and PHBV NCP (Bordes et al., 275 2008). This increase in the interlayer distance reflects the formation of intercalated structures, 276 277 confirmed by TEM observations which indicated both intercalated and exfoliated structures for nanocomposite materials. 278

279

3.2 Migration curves of additives from studied films into food simulants

The desorption kinetic of each additive from polymers into food simulants was measured over time, until 10 days, or 30 days in the case of PHBV contaminated with high molar mass additives. An example of desorption kinetics is reported in Fig. 2, for methyl stearate additive, from neat and nanocomposite PBSA into 95% ethanol food simulant.

For all molecules, a good correlation was observed between migration experimental data and simulated curves using Eq. (6), with a RMSE inferior to 6%, calculated using Eq. (8).

Desorption curves can be analyzed in their growth phase, from which the diffusion coefficient can be estimated, and in their state of equilibrium, from which the partition coefficient can be estimated. The addition of nanoclays in the polymer matrix has varying effects on both phases according to the nature of the diffusing additives (Nasiri et al. 2016).

290 **3.3** Comparison of the D_{app} of additives in neat and nanocomposites materials

The estimation of the D_{app} of each additive from biopolymer into food simulant is reported in Fig. 3, with the respective impacts of: polymer type, food simulant, and the effect of nanoparticle incorporation.

The differences in the D_{app} in the various systems, illustrated by the differing scales of the histograms, reflect the influences of the polymer type and the food simulant in contact. As in previous works that highlight the poor gas barrier properties of PBSA (Phua et al., 2013), the diffusivity values measured in PBSA are between five and twenty times lower than those in PHBV, which normally has good barrier properties (Crétois et al., 2014).

299 Contact with 95% ethanol seems to have a dramatic impact: producing diffusion values 300 nearly ten times higher than for iso-octane contact. The very peculiar behavior of polyesters in 301 contact with ethanol was already reported for PET (Begley and Hollifield, 1990), PLA 302 (Jamshidian et al., 2012), and PHBV (Chea et al., 2015). For PBSA and PHBV matrices with nanoparticles in contact with 95% ethanol and isooctane, the D_{app} varies according to additives. The impact of nanoparticles is not significant if compared to the impact of the biopolymer type or food simulant in contact. For PBSA in contact with 95% ethanol, nanoparticles decreased the D_{app} for all additives. In contact with iso-octane, nanoparticles increased the D_{app} , especially for low molar mass additives. For PHBV, nanoparticles increased the D_{app} significantly whatever the food simulant in contact.

309 These results contradict previous conclusions on conventional synthetic materials. For polyurethane, there is a negative correlation between C-30B (concentrations 0 - 50 wt%) and 310 the diffusivity of three volatile organic compounds: toluene, decane, and butanol (Herrera-311 312 Alonso et al., 2009). Nasiri et al. (2016) measured the diffusivities in LLDPE with a wide range of additives exposed to four food simulants: Dapp decreased with the addition of Cloisite 313 regardless of the nature of the diffusing substance. Therefore, whereas plastic-based 314 315 nanocomposite materials meet inertia criteria regarding the migration of low molar mass substances, the behavior of biodegradable materials raises new concerns relating to compliance 316 317 with regulations and consumer safety. Given the limited available data, it is difficult to establish behavioral laws to describe the specific impact of nanoclays on the evolution of the D_{app} of low 318 molar mass additives. In addition to the tortuosity effect demonstrated in section 3.1, these 319 results suggest that nanoparticles modify polymer structures (size, shape, or arrangement of 320 spherulites, and degree of crystallinity) and thus their transfer properties. 321

322 **3.4** Predicting the D_{app} of additives according to molar mass

323 The D_{app} of additives is related to their specific volume and, consequently, to their molar 324 mass: $D \sim M^{-\alpha}$ with a scaling parameter α that can be related to the transport mechanism 325 (Lodge, 1999). Fig. 4, below, shows the D_{app} of additives in function of their molar mass in neat and nanocomposite PBSA (Fig. 4 A) and PHBV (Fig. 4 B) in contact with 95% ethanol and isooctane food simulants. The linear dependence of Log (D) = f (M) is clearer for PBSA than PHBV. The dependence of the diffusion coefficient on size has already been demonstrated in natural rubber and in a glassy synthetic polymer (Chern et al. 1985; Reynier et al. 2001; Martinez-Lopez et al., 2016). The inclusion of nanoparticles does not contradict this relationship given their weak influence on additive diffusivity.

Log D being correlated with the molar mass of additives, a Piringer correlation can be applied to PBSA and PHBV biopolyesters. This empirical model would enable verification of accordance with European regulations of suitability for food contact of materials on the basis of an overestimation of the diffusion coefficient of plastic additives.

Lacking the specific parameters, A'_p and τ , for PBSA and PHBV, the regression was 337 performed using the values for LDPE and PET, these two polymers being selected as references 338 for their low and high barrier properties respectively (Arora and Padua, 2010). The curves 339 obtained from the Piringer model using key parameters of LDPE and PET are presented in 340 Fig. 4. Migration modelling can provide the upper migration values, necessary to predict a 341 342 reliable worst-case scenario for bio-polyesters. While the PET Piringer model underestimates the diffusivity values measured of bio-polyesters, the LDPE model is better adapted, slightly 343 overestimating diffusivity (except for high molar mass additives), especially for neat and 344 nanocomposite PBSA in contact with 95% ethanol. The values of A'_p and τ , although based on 345 the integration of large quantities of experimental data are insufficient to provide Piringer 346 347 parameters guaranteeing systematic overestimation of migration levels from PHBV and PBSA based packaging. However, the results of this study suggest that the incorporation of nanoclay 348 349 does not modify these recommendations.

350

351 **3.5** Impact of the sorption of food simulant

The contact of polymers with food simulating liquids can modify their transport properties (Figge, 1972), and the presence of nanoparticles is likely to impact these interactions. The sorption of food simulant by the semi-crystalline biopolymer usually induces swelling in the biopolymer, which accelerates additive migration and increases the diffusion coefficient value (Reynier et al., 2001; Barnes et al., 2007).

The solvent uptake of PBSA, PHBV and their corresponding nanocomposites after 10 days of contact with 95% ethanol and iso-octane at 40°C were summarized in Table 3. The sorption of ethanol by both tested packaging films could explain the high D_{app} values measured during the contact with this food simulant (Fig. 3).

In the case of PBSA based materials, significant ethanol uptakes of 3.5% (neat) and 4.7% (nanocomposite) were recorded, although there was no measurable change in film thickness. The 95% ethanol sorption value of PHBV is in accordance with the value reported by Chea et al. (2015) on the same matrix/food simulant system. As mentioned in section 3.3, the peculiar interaction of 95% ethanol with polyesters was already proven with PET (Begley and Hollifield, 1990; Widén et al., 2004; Kim and Lee, 2012), PLA (Jamshidian et al., 2012) and PHBV (Chea et al., 2015).

For neat and nanocomposite PBSA in contact with iso-octane, no solvent uptake was detected. Neat and nanocomposite PHBV in contact with iso-octane had a < 1.5%, solvent uptake.

The incorporation of nanoclays in the polymer matrix leads to different behaviors depending on the nature of the food simulant. 95% ethanol uptake increased with nanoclays whatever the polymer. In PHBV, addition of nanoclay reduced iso-octane uptake. Nanoclays are hydrophilic and therefore soluble in ethanol (Ho and Glinka, 2003) which leads to a greater

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solvent uptake. However, we cannot attribute the increase in D_{app} values observed for all
additives solely to solvent sorption.

377 3.6 Impact of the semi-crystalline structure of bio-nanocomposites

Bio-nanocomposite transfer properties are influenced by factors such as the degree of crystallinity of the polymer: high crystallinity limits movement and therefore limits the diffusion of migrating substances (Hedenqvist et al., 1996).

The degrees of crystallinity of neat and nanocomposite biopolymers evaluated by DSC experiments on films without contact and after 10 days in contact with food simulants at 40°C were summarized in Table 4.

PHBV has a high degree of crystallinity (Chea et al., 2015): almost twice that of PBSA,
which may explain the difference between D_{app} values for these two materials.

The degree of crystallinity of polymers proved to be food simulant dependent. In contact 386 with 95% ethanol, the degree of crystallinity decreases by 1.3% for PHBV, and by 3.9% for 387 PBSA. These results are similar to previous research conducted on these polymers and food 388 simulants (Chea et al., 2015; Siracusa et al., 2015). The decrease in crystallinity could be 389 attributed to the degradation of polymer chains due to hydrolysis reactions induced by the 95% 390 ethanol. However, in contact with iso-octane, there was no change in the degree of crystallinity 391 for PHBV, a 1.3% increase for PHBV NCP and a close to 2% increase for neat and 392 nanocomposite PBSA. This result confirms the findings of Chea et al. (2015): there was no 393 394 structural change to PHBV in contact with iso-octane.

C-30B decreases the degree of crystallinity of the two polymers by nearly 10%. This has previously been observed for PBSA/C30B (Sinha Ray et al., 2005), and for PHBV/OMMT (Wang et al., 2005). It can be explained by the full exfoliation of nanoclays platelets in the polymer matrix (see section 3.1), which restricts the mobility of the polymer chains and thus prevents them from crystallizing (Krikorian and Pochan, 2003). This decrease results in an increase of the amorphous phase in which the diffusivity of additives is facilitated, this tendencyis observed in all cases apart from PBSA in 95% ethanol (Fig. 3).

Although the degree of crystallinity of materials is known to impact the transport properties of semi-crystalline polymers, strict correlation between crystallinity and diffusivity cannot be concluded in the case of biodegradable packaging materials. For instance, D_{app} of additives in PBSA are three times higher in contact with 95% ethanol than with iso-octane, but the crystallinity of both materials is similar. The relationship between diffusion and crystallinity on a macroscopic scale remains obscure. Nanoclays generate localized crystalline gradients (Wurm et al., 2010) which could modify the transport properties of materials.

409 4 Conclusion

The incorporation of nanoclays in biodegradable materials raises new questions related 410 to their food contact suitability. While the few studies carried out on synthetic plastic materials 411 412 agree that the inclusion of nanoclays decreases diffusivity, this work demonstrates that this is not the case for biodegradable materials. The presence of nanoparticles affects diffusion 413 414 variously, depending on the nature of the migrating substance and particularly depending on the nature of the food simulant. As previously observed for biopolyester based materials, 415 contact with ethanol, whose diffusion activation mechanisms remain unclear, is the worst-case 416 scenario. This particular sensitivity of biopolyesters to ethanol, especially when they 417 incorporate inorganic nanoparticles such as nanoclays, underscores the problem of applying 418 419 existing recommended testing conditions for the evaluation of conventional plastic materials to 420 these new materials. Ethanol sorption, which promotes the crystallization of synthetic polyesters and thus prevents the mobility of low molecular weight molecules, appears to be 421 422 enhanced by the incorporation of clay nanoparticles. Being necessarily cautious we cannot, with sufficient accuracy, predict migration levels from bio-nanocomposite packaging using diffusion 423 models generally applied to food contact material considered as continuous homogeneous 424

425 phase. However, Piringer's empirical model remains an applicable way to overestimate 426 migration and guarantee consumer safety. This model would be more effective with the creation 427 and inclusion of database of the diffusion coefficients of low molecular weight molecules in 428 these materials.

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611







TEM image

Fig. 1. XRD spectra of neat and nanocomposite polymers (A) and TEM image (B) of PHBV NCP.



Fig. 2. Experimental (symbols) and simulated (curves) desorption kinetics for the first 10 hours for PBSA

(grey) and PBSA NCP (black) into 95% ethanol for methyl stearate.



Fig. 3. Comparison of D_{app} of additives in neat (grey bars) and nanocomposite (black bars) PBSA (Fig. A and B) and PHBV (Fig. C and D) films in contact

with food simulants: 95% ethanol (Fig. A and C) and iso-octane (Fig. B and D).



Fig. 4. D_{app} of additives in function of additives molar mass for neat and nanocomposite PBSA (A) and PHBV (B). Neat PBSA or PHBV in contact with 95% ethanol (\blacksquare) or iso-octane (\bullet) and PBSA NCP or PHBV NCP in contact with 95% ethanol (\Box) or iso-octane (\circ). Piringer estimations in LDPE (-) and PET (---) at 40°C.

			CAS	М
Category	Code	Physical properties	number	(g.mol ⁻¹)
High molar	Uvitex OB	Non-volatile	7128-64-5	430.56
mass	Irganox 1076	Non-volatile polar	2082-79-3	530.86
contaminants	Irganox 1010	Non-volatile	6683-19- 8	1177.63
	Toluene	Volatile non-polar	108-88-3	92.14
	Chlorobenzene	Volatile polar	108-90-7	112.56
	Methyl salicylate	Non-volatile polar	119-36-8	152.15
Low molar	Biphenyl	Volatile non-polar	92-52-4	154.21
mass contaminants	Phenyl cyclohexane	Non-volatile non- polar	827-52-1	160.26
	Benzophenone	Non-volatile polar	119-61-9	182.22
	Methyl stearate	Non-volatile non- polar	112-61-8	298.50
	DEHA	Non-volatile polar	103-23-1	370.57

Table 1: List of selected additives.

Polymer	A' _P	τ
LDPE	10	0
PET	2.2	1577

Table 2: LDPE and PET specific parameters.

	Solvent up	take (%)
Food simulant Sample	95% ethanol	Iso-octane
PBSA	3.7 ± 0.5	0
PBSA NCP	4.7 ± 0.1	0
PHBV	3.5 ± 0.1	1.3 ± 0.1
PHBV NCP	3.8 ± 0.2	0.5 ± 0.1

Table 3: Solvent uptake of films after 10 days of contact with food simulant at 40°C.

Table 4: Degree	of crystallinity	of film samples	without contact and	after 10 days of c	ontact
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Degree of crystallinity (X _C) (%)		
Without contact	95% ethanol	Iso-octane
-38.2 ± 0.0	36.7 ± 0.1	39.0 ± 0.1
34.9 ± 0.4	34.1 ± 0.3	35.7 ± 0.1
62.8 ± 1.6	62.0 ± 0.1	62.9 ± 0.1
55.0 ± 0.4	53.6 ± 0.3	55.7 ± 0.1
	Degree Without contact 38.2 ± 0.0 34.9 ± 0.4 62.8 ± 1.6 55.0 ± 0.4	Degree of crystallinity (X Without contact 95% ethanol 38.2 ± 0.0 36.7 ± 0.1 34.9 ± 0.4 34.1 ± 0.3 62.8 ± 1.6 62.0 ± 0.1 55.0 ± 0.4 53.6 ± 0.3

with 95% ethanol or iso-octane at 40° C.