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► **To cite this version:**

Godefroy Leménager, Sandrine Tusseau-Nenez, Maud Thiriet, Pierre-Eugène Coulon, Khalid Lahlil, et al.. NaYF₄ Microstructure, beyond Their Well-Shaped Morphology. *Nanomaterials*, 2019, 9 (11), pp.1560. 10.3390/nano9111560 . hal-02345184

HAL Id: hal-02345184

<https://hal.science/hal-02345184>

Submitted on 7 Dec 2020

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Article

NaYF₄ microstructure, beyond their well-shaped morphology

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Version August 18, 2019 submitted to Nanomaterials

Abstract: Lanthanide doped nanoparticles are widely investigated for their optical properties. However, the sensitivity of the lanthanide ions to the local symmetry, useful when investigating structural environments, becomes a drawback for optimized properties in the case of poorly controlled crystallinity. In this paper, we focus on β -NaYF₄ nanorods in order to provide a detailed description of their chemical composition and microstructure. The combination of detailed XRD analysis and TEM observations show that strong variation may be observed from particles from a same batch of synthesis, but also when considering small variations of synthesis conditions. Moreover, also the nanorods observed by SEM exhibit a very nice faceted shape, they are far from being monocrystalline and present significant local deviation of crystalline symmetry and orientation. All these structural considerations, sensitively probed by polarized emission analysis, are crucial to be analyzed for the development of optimal systems toward the targeted applications.

Keywords: X-ray diffraction; crystalline structure; polarized luminescence

1. Introduction

The lanthanide doped NaYF₄ nanoparticles (NPs) are a unique class of luminescent nanoparticles that focus increasing attention due to their efficient up-conversion properties and the ability, through colloid chemistry, to nicely play on a variety of shapes[1–3], length[4,5], doping[6]. In the last decade, these particles have been considered for a large variety of innovative applications such as biological labeling, nanothermometers[7,8], rheometry[9], anti-counterfeiting [10,11].

As in all rare-earth doped luminescent compounds, optical properties (emission efficiency, spectral shapes) are known to result from the intrinsic properties of sensitizing ions modulated by their environment within the host matrix. The effect on spectral shape is of primary importance when d orbitals are involved (Eu²⁺, Ce³⁺), but remains significant for transitions implying only f-electron states. This is well documented in the case of Eu³⁺ doped compounds, this ion being considered as a very good probe of local environment, providing good indications on site symmetry through the hypersensitive ⁵D₀ – ⁷F₂ transition[12]. Much less is known and understood on structure/property relationships in the case of up-conversion compounds, which involve more complex emission scheme and energy levels [12]. NaYF₄, which is the emblematic compound for up-conversion, is known to exist in two different polymorphs: cubic α -NaYF₄ and hexagonal β -NaYF₄. The impact of the host crystal on optical properties is directly evidenced by the poor emission properties of the α -NaYF₄ as compared to the β -NaYF₄ [ref]. As revealed by a literature survey, the exact structure of the hexagonal β -NaYF₄ phase is not so clear [13–17] since three slightly different crystalline structures (see table 2)

32 have been reported, with small but significant variations of their cell parameters and structures, but
33 also different host site symmetries.

34 Following our recent study on $\text{LaPO}_4\text{:Eu}$ nanorods[18], the initial purpose of the present work
35 was to investigate structure/property relationship in the case of $\text{NaYF}_4\text{:Yb-Er}$ nanoparticles, with
36 the objective of understanding and optimizing emission properties including emission yield, spectral
37 shape of the emission lines and polarization properties. On this way, polarization characterization
38 rapidly provided evidences that such an investigation would be first limited by the fact that NaYF_4
39 nanorods are apparently not monocrystalline and present a significant microstructure. We found that
40 structural investigations of NaYF_4 nanoparticles have never been discussed in details despite the
41 numerous works done on this kind of particles, all obtained following similar routes. We attribute
42 this lack of interest to the very nice shape of the particles as revealed by SEM pictures, which suggest
43 an excellent quality of their structure thus hiding interest for such studies. Moreover, polarization
44 of the emission is not so commonly characterized, and appears to be a very sensitive probe of the
45 microstructure[18]. All these points motivated the present study, aiming in providing new insights on
46 the particles microstructure in order to understand and optimize optical properties that are the basis of
47 increasing number of very exciting applications [19–22].

48 2. Experimental section

49 2.1. Typical synthesis of the $\text{Er}^{3+}/\text{Yb}^{3+}/\text{Gd}^{3+}\text{-NaYF}_4$ nanorods

50 $\text{GdCl}_3\cdot 6\text{H}_2\text{O}$ (99.99%), $\text{YCl}_3\cdot 6\text{H}_2\text{O}$ (99.99%), $\text{YbCl}_3\cdot 6\text{H}_2\text{O}$ (99.99%), $\text{ErCl}_3\cdot 6\text{H}_2\text{O}$ (99.99%), NaOH
51 (98%), NH_4F (98%), NaF (98%) and oleic acid (90%) were all purchased from Sigma-Aldrich and used
52 as starting materials without any further purification.

53 Synthesis where adapted from the protocols of [23] and [24], which consists in solvothermal
54 precipitation of chloride lanthanide salts and NH_4F in a water/ethanol/oleic acid mixture.
55 Experimental conditions are shown on table 1 for samples labeled NRXX for nanorods sample N°XX.
56 In a typical experiment (eg sample NR11), 30 mmol (1.2 g) of NaOH in 5.6 mL of water were mixed
57 with 20 mL of ethanol (EtOH) and 20 mL of oleic acid (OA) under stirring. To the resulting mixture
58 were selectively added 0.75 mmol (228 mg) of $\text{YCl}_3\cdot 6\text{H}_2\text{O}$, 0.27 mmol (104.6 mg) of $\text{YbCl}_3\cdot 6\text{H}_2\text{O}$, 0.03
59 mmol (11.4 mg) of $\text{ErCl}_3\cdot 6\text{H}_2\text{O}$, 0.45 mmol (167 mg) of $\text{GdCl}_3\cdot 6\text{H}_2\text{O}$ and 7.7 mmol (278 mg) of NH_4F
60 dissolved in 12 mL of water. The solution was then transferred into a 75 mL autoclave and heated
61 at 200 ° C for 2 h under stirring. In some cases, the NPs were heated for a longer time (24h instead
62 of 2h) to remove the $\alpha\text{-NaYF}_4$ NPs and also a sealed glass tube has been used to test higher pressure
63 synthesis. After cooling down to ambient temperature, the resulting nanoparticles were precipitated
64 by addition of 50 mL of ethanol, collected by centrifugation, washed with water and ethanol several
65 times. They were finally dried and kept as a powder. For the optical experiments, we manipulate
66 the nanorods (NRs) as aqueous dispersion to ease the manipulation and observations. In this case,
67 a functionalization by ligand exchange is needed to ensure the good dispersion in water. About 20
68 mg of NaYF_4 @oleic acid NPs are sonicated and centrifuged several times with 2 mL aqueous citrate
69 solution (0.2 M), washed with EtOH and water to remove remaining oleic acid molecules, finally the
70 nanorods are well dispersed in water.

71 2.2. Confocal microscopy

72 The NPs were analyzed by a confocal microscopy system from Nikon (Nikon Eclipse Ti with
73 confocal module C2 Si). A nanomolar solution of citrate functionalized NPs in water is first sonicated 4
74 times 45s (450W Branson) to ensure the perfect dispersion of the NRs and then drop-casted on a glass
75 coverslip. A Tsunami Ti-Sapphire pulsed laser was focused on the NPs with a 60x oil objective. The
76 polarized resolved spectroscopy was performed by means of a spectrometer (IsoPlane SCT320 from
77 Princeton Instrument) coupled to CCD (Pixis-400-BX) and a motorized polarizer.

Sample Name	Ln amount			Solvent volume			Heating		Type of content
	Er	Yb	Gd	H ₂ O	EtOH	OA	Time	Temp	
NR00	2	18	30	17	20	20	2H	200° C	Autoclave
NR01	2	18	0	17	20	20	2H	200° C	Autoclave
NR02	2	18	0	17	20	20	2H	200° C	Autoclave
NR03	2	18	5	17	20	20	2H	200° C	Autoclave
NR04	2	18	5	17	20	20	20H	200° C	Autoclave
NR05	2	18	15	120	110	110	2H	200° C	Glass tube
NR06	2	18	15	45	50	50	2H	200° C	Glass tube
NR07	2	18	15	45	50	50	20H	200° C	Glass tube
NR08	2	18	30	45	50	50	2H	200° C	Glass tube
NR09	2	18	30	45	50	50	2H30	200° C	Glass tube
NR10	2	18	30	120	110	110	3H	200° C	Autoclave
NR11	2	18	30	17	20	20	2H	200° C	Autoclave
NR12	2	18	45	45	50	50	2H	200° C	Glass tube
NR13	2	18	60	45	50	50	2H	200° C	Glass tube

Table 1. Summary of all the synthesis parameters

78 2.3. TEM observation

79 Transmission Electron Microscopy (TEM) experiments were performed using a field emission gun
80 JEOL JEM-2010F microscope operating at 200 kV with a high-resolution HR polar piece ($C_s = 1.0$ mm,
81 $C_c = 1.4$ mm, point resolution = 0.24 nm, with a dose of ≈ 20 electrons \AA at 50,000 magnification and \approx
82 100 electrons \AA at 250,000 magnification). Chemical mapping was performed by XEDS (Cliff-Lorimer
83 methods) using a FEI Titan Themis probe-corrected microscope operating at 200kV and equipped with
84 a Scanning Transmission Electron Microscopy (STEM) module at 115,000 and 225,000 magnification
85 (probe size 1 \AA) and "Super-X" detectors (beam convergence angle : 24.6 mrad).

86 During the ions quantification, an amorphization of the NPs may be observed. The loss of the
87 crystalline structure can increase the ion mobility and change the doping repartition in the NP but not
88 the average doping of a NP. For this reason, we present only global measurements on whole NPs.

89 2.4. X-Ray diffraction: ESRF

90 The synchrotron X-Ray powder diffraction experiments were carried out at the Swiss-Norwegian
91 Beam-Lines (BM1A station) of the European Synchrotron Radiation Facility (ESRF), France. A
92 monochromatic beam of wavelength $\lambda = 0.7129$ \AA was focused onto the sample by sagittal bending of
93 the second crystal of a double-crystal Si(111) monochromator using additional slits of $272.26 * 327.34$
94 μm^2 . X-Ray Powder Diffraction (XRPD) data were collected in transmission geometry using a
95 pixel-array detector (PILATUS 2M, Dectris Ltd). Samples were sealed in 0.4 mm diameter glass
96 capillaries and rocked by 10° during data collection. The instrumental resolution was determined
97 using a LaB₆ NIST standard (Standard Reference Material 660a, cell parameter = 0.41569162 nm \pm
98 0.00000097 nm at 22.5°C).

99 The selected patterns for the identification of phases by XRPD and the crystallographic data (COD
100 database [25]) for the line profile analysis are summarized in Table 2.

101 3. Results and discussion

102 3.1. Confocal microscopy and polarized luminescence

103 With our confocal set-up, the polarized spectra of our NPs deposited by drop-casting of diluted
104 NaYF₄ NPs in water are measured. The low concentration coupled to a sonication of the solution just
105 prior deposition leads to dispersed NPs on the glass coverslip. Without reference, the orientation of the
106 NP cannot be known nor the expected orientation of the polarization. We measured the spectra for a

Phase	Space group	Cell parameters (Å)	Atomic positions	Occupancy	Re site symmetry	COD ID	ref
Hexagonal (β)	$P6_3/m$ 176	$a = b =$ 5.99276 $c =$ 3.52281	Y $2/3$ $1/3$ $1/4$ Na $2/3$ $1/3$ $1/4$ Na 0 0 z F x y $1/4$	0.75 0.25 0.25 0.25	C_{3h}^2	1517675	[26]
Hexagonal (β)	$P\bar{6}$ 174	$a = b =$ 5.99276 $c =$ 3.52281	Y $2/3$ $1/3$ $1/2$ Y 0 0 0 Na $2/3$ $1/3$ $1/2$ Na $1/3$ $2/3$ z F x y $1/2$ F x y 0	0.5 0.75 0.5 0.5 1 1	C_{3h}^2 C_{3h}^2	1517672	[26]
Hexagonal (β)	$P\bar{6}2m$ 189	$a = b =$ 5.9148 $c =$ 3.52281	Y $1/3$ $2/3$ $1/2$ Y 0 0 0 Na $1/3$ $2/3$ $1/2$ F x 0 0 F 0 $1/2$	0.25 1 0.75 1 1	C_{3h}^1 C_3^1	1517674	[26]
cubic (α)	$FM\bar{3}m$ 225	$a = b = c =$ 5.47000	Y 0 0 0 Na 0 0 0 F $1/4$ $1/4$ $1/4$	0.5 0.5 1	O_h	1517676	[27]

Table 2. Crystallographic data on synthesized phases of $NaYF_4$ for phase identification and rare-earth (Re) ion site symmetry

107 wide range of polarization angle. In these conditions, the axes of the two orthogonal electric dipoles will
 108 be given by the least brilliant and the brightest spectra. In Fig. 1.a we present the photoluminescence
 109 (PL) spectrum for two orthogonal polarizations (red and blue) and the difference between them in
 110 green. Moreover, by plotting the PL intensity versus the polarization angle (see Fig. 1.b), the typical
 111 dipole-like behavior expected from lanthanide ions appears to be in good approximation with the
 112 Malus law.

113 However, if we look in detail this dependence for several wavelengths, the extreme values are
 114 not obtained for the same angles as expected. Indeed, the polarization is due to the site symmetry
 115 of the lanthanide ion and the electrons can optically relax by emitting α , σ or π photons [28] with
 116 orthogonal polarization. Then a PL dependence can only present two angles for the PL optima values
 117 for all measured transitions, which is not the case: the 665.0 nm transition has a maximum shifted
 118 of 20° as compared to another transitions at 662.9 nm. This observation leads to conclude that the
 119 local symmetry orientation is not the same for all the emitters. To understand this observation, we
 120 investigated the microstructure of the particles in more details.

121 3.2. TEM observation and doping homogeneity

122 The doping of the $NaYF_4$ NPs (nature and concentration of the rare earth ions) plays an
 123 unquestionable part in the shape and structure of the NPs[3]. Several batches of NPs were synthesized
 124 with different nominal doping concentrations. The effective doping of NPs was measured by EDX
 125 analysis. The table 3 presents the expected and effective doping of particles obtained from different
 126 syntheses and shown in figure 2.

127 The three first samples were prepared under similar conditions (doping concentration, heating
 128 temperature and time). However, the effective dopings are found to be different: NRs are less rich in
 129 Y of about 20 mol%, more than 3 times more rich in Er whereas Yb and, in a minor way, Gd doping
 130 are close to the expected values. This shows the sensitivity of the synthesis to some parameters that
 131 can be associated to the volume of solvent and the type of container, that could affect for example the
 132 pressure during the synthesis.

133 TEM observations evidences two situations. On one hand, we sometime observe in the same
 134 batch two (or more) populations of NPs, each one having a relatively low size dispersion. However,

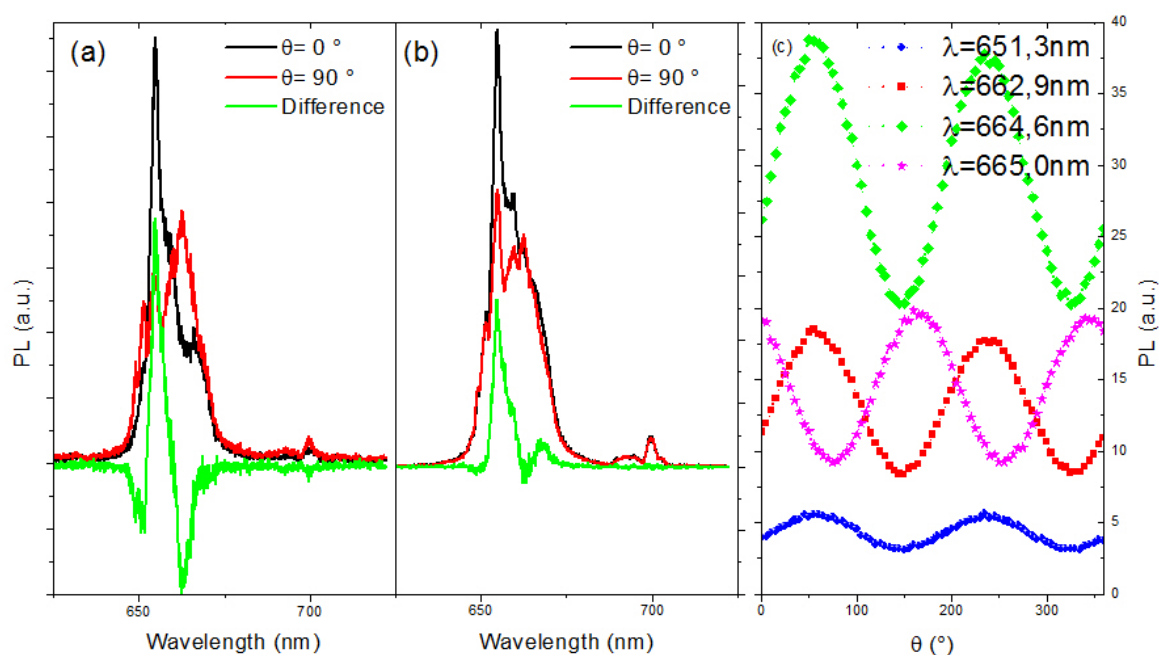


Figure 1. (a-b) Spectrum for two orthogonal polarizations (red and black) and the difference between them (green) for two different NPs of the same batch NR07. (c) PL intensity versus polarization for different wavelengths.

Sample name	TEM size (nm)		Aspect Ratio	Expected doping (mol%)				Measured doping (mol%)				Area in Fig. 2
	L	D		Y	Er	Yb	Gd	Y	Er	Yb	Gd	
NR00	300	50	6	50	2	18	30	40.2	9.4	18.2	32	
NR09	150	60	2.5	50	2	18	30	40.0	6.9	15.6	38.6	
NR11	200	30	6.6	50	2	18	30	41.5	8.4	17.5	32	1
	640	100	6.4					56	6.7	13.4	18	2
NR13	55	40	1.4	20	2	18	60	13.3	14.5	13.5	58	4
	130	35	3.7					23.9	14.6	13.9	47.6	3

Table 3. Expected and effective doping concentration of different NPs measured by EDX. Two lines in the tabular correspond to two population measured by EDX. The NPs sizes given come from the measured NPs and can differ from the mean values.

135 the size of each population can be very different as it is illustrated in Fig. 2 for two syntheses (NR11
 136 and NR13). The EDX measurements of the delimited area give us the chemical composition as reported
 137 in the table 3. For each population, we measure a strong difference in the gadolinium concentration
 138 (and in consequence the yttrium concentration) while erbium and ytterbium concentrations present
 139 only small variations. In Fig. 2.a, the NP in the area (1) delimited by the dotted green line area has a
 140 concentration of gadolinium of 32 mol% and its size is typical from this synthesis (50 * 260 nm), while
 141 the NP shown by the continuous red line (area 2) has a concentration of only 18 mol% and its size is
 142 twice the average size (110 * 650 nm) with a similar aspect ratio. Another example for the batch NR013
 143 is presented in Fig. 2.b where the same behavior is observed with a complete different aspect ratio.
 144 While the experimental conditions are exactly the same, one (NR00) or two populations of particle
 145 sizes (NR11) are observed, with identical aspect ratios (about 6). At this stage, no clear parameter can
 146 explain these results but this is probably an evidence that the two kind of particles formed at different
 147 time during the synthesis, probably because the kinetic of formation of Gd rich or Gd poor particles is
 148 different.

149 On the other hand, the expected doping and the effective doping can differ strongly. Depending
 150 on the synthesis, the gadolinium concentration expected to be around 30%, is found to vary from

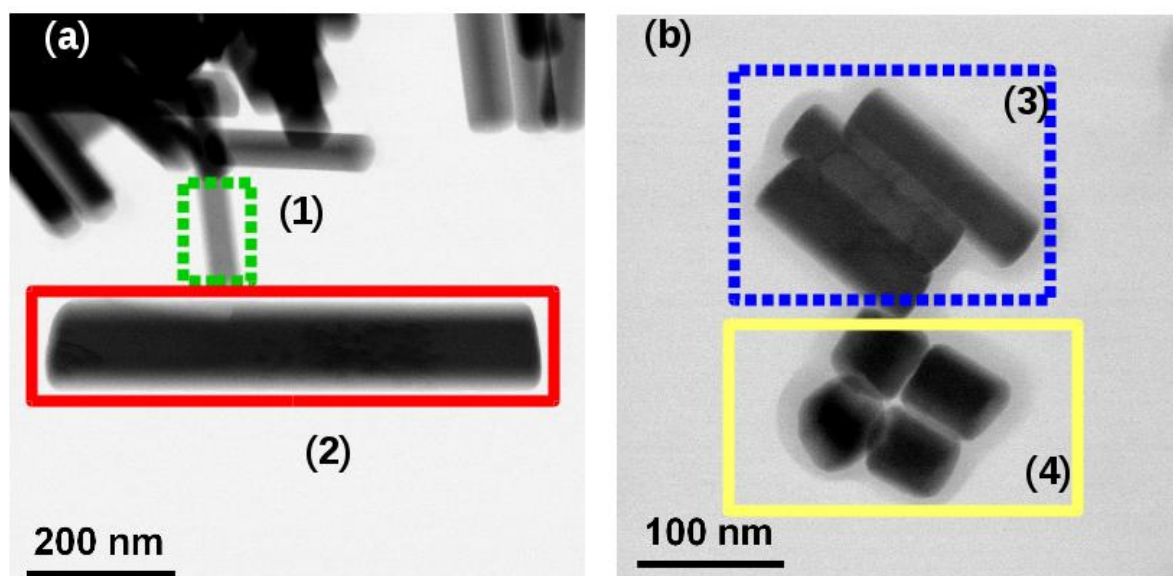


Figure 2. Two TEM images of NaYF₄ NPs from the two batches presenting two NPs populations ((a) NR11 and (b) NR13)

151 18% to 38%. Two samples share a population with the same doping while the first sample presents
 152 a different doping. The two last experiments were done with another solvent ratio compared to the
 153 first experiment resulting in a different pressure during the synthesis. These differences in pressure,
 154 solvent composition and the correlated solubility can explain the doping difference between the first
 155 and the two last syntheses.

156 3.3. Phase identification and doping dependence

157 In Fig. 3.b, we present one typical diffraction pattern obtained for a mixture of α -NaYF₄ and
 158 β -NaYF₄ phases. The mixture of these two types of structure can easily be studied independently
 159 since they exhibit clearly different diffraction patterns. Not all of our syntheses result in a mixture
 160 of α -NaYF₄ and β -NaYF₄ nanoparticles - we focused our work on β -NaYF₄ nanorods as this phase
 161 provides the interesting optical properties. As previously mentioned, the β -NaYF₄ NPs can adopt three
 162 different crystalline structures, with space groups depending on the synthesis parameters known to be
 163 P6₃/m, P $\bar{6}$ and P $\bar{6}2$ m (see Fig. 3) [26]. The main difference between the powder diffraction patterns
 164 lies in an extra peak from the (001) planes for the P $\bar{6}$ and P $\bar{6}2$ m compared with the P6₃/m patterns at
 165 $2\theta = 11.70^\circ$ in our experimental conditions. The intensity of this peak is very low, estimated around 2%
 166 of the maximum diffraction peak intensity and below 0.1% respectively for P $\bar{6}$ and P $\bar{6}2$ m. This small
 167 difference is usually not visible in XRPD done with a usual laboratory diffractometer and was at the
 168 origin of long discussions in the literature [13,14]. The P $\bar{6}$ and P-62m can only be distinguished by a
 169 quantitative analysis of the peak intensities. A synchrotron facility is required to properly determine
 170 the symmetry of NaYF₄ NPs.

171 For the β -NaYF₄ NRs, among the thirteen studied samples, only one of them presents a detectable
 172 specific peak of the P $\bar{6}$ or P $\bar{6}2$ m structure. Furthermore, the intensity of this peak is very low (0.1%
 173 of the maximum intensity) indicating that only a negligible portion or region of the particles was obtained
 174 with this structure. For these reasons, only the P6₃/m structure was considered in the following for
 175 our NPs.

176 After the phase identification, the Fullprof software [29,30] was used in profile matching mode
 177 with a constant scale factor in order to perform a microstructural analysis. The XRPD patterns were
 178 fitted with a Thompson-Cox-Hastings function [31] (a modified pseudo-Voigt function) to extract, for
 179 each phase (α and/or β -NaYF₄), the cell parameters and the width and breadth (full width at half

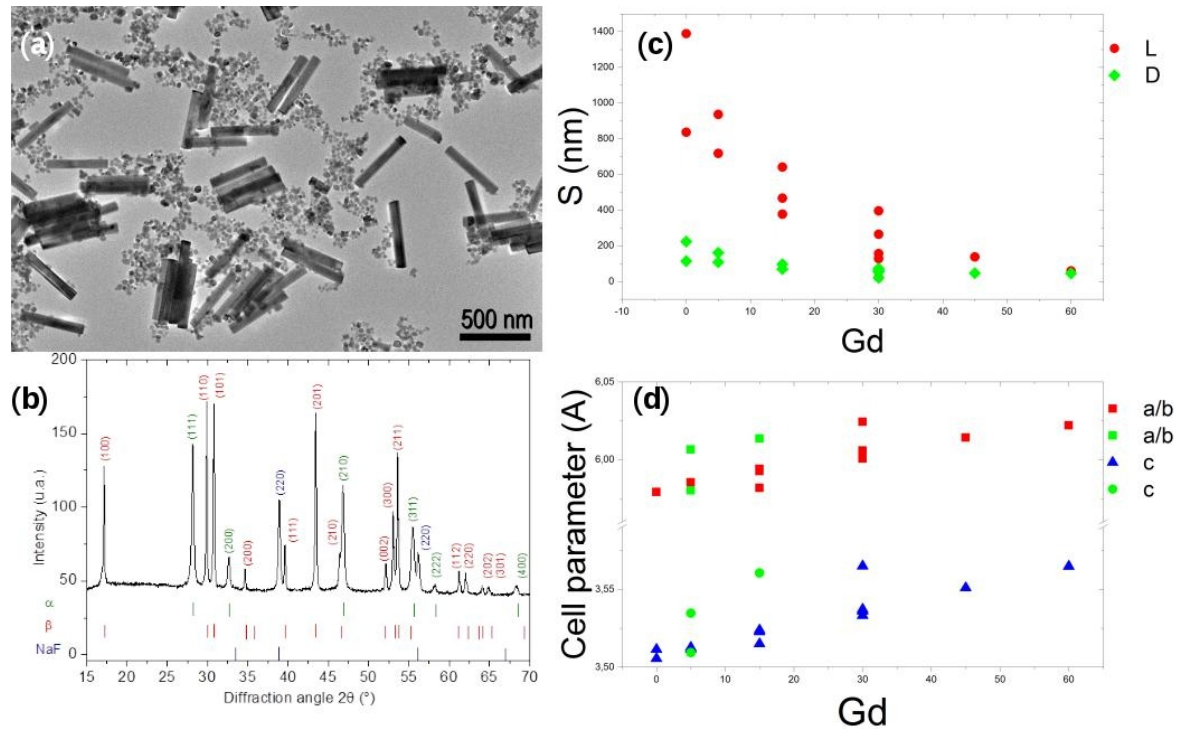


Figure 3. NaYF_4 nanoparticles (a) TEM image with small cubes ($\alpha\text{-NaYF}_4$) big nanorods ($\beta\text{-NaYF}_4$) (b) typical X-Ray pattern of these particles (c) TEM NPs sizes S (L = length, D = diameter) and (d) cell parameter dependences vs Gd nominal doping with blue triangles and red squares for the main set of cell parameters of each batch and green stars and green circles for the second set of cell parameters

180 maximum FWHM and integral breadth IB) of the Bragg peaks. Our study focuses on profile fitting
 181 and diffraction line broadening analysis. As said previously, the particles are doped with a constant
 182 nominal concentration of ytterbium (18 mol %) and erbium (2%) and we only changed the gadolinium
 183 concentration (from 0% to 60%). On the contrary to EDX focusing on single or few NPs, XRPD gives a
 184 global and averaged analysis of the sample.

185 Fig. 3.d presents the evolution of the cell parameters with the nominal gadolinium content of the
 186 $\beta\text{-NaYF}_4$ phase. In some samples (NR03, NR 04 and NR06), two $\beta\text{-NaYF}_4$ phases are identified, a main
 187 peak is observed at the expected position and a second one slightly shifted (shoulder or separated
 188 peak, associated to other set of cell parameters shown in Fig. 3.d with a green star or circle). It is not
 189 possible to associate the two phases observed here with the two populations previously observed with
 190 the EDX. However, the two results are similar and result reinforce themselves. Moreover, both sets
 191 present in parallel a linear dependence with the amount of gadolinium (even with the poor statistic
 192 of the second group). The second set of cell parameters cannot be explained by a population with a
 193 higher gadolinium concentration as observed previously. Yet, the linear dependency is expected due
 194 to the higher ionic radius of the gadolinium compared to yttrium.

195 3.4. Microstructural analysis

196 For all the 13 studied samples, all XRPD patterns exhibit peaks broader than the instrumental
 197 resolution (typically the FWHM for the samples was above $0.050^\circ 2\theta$ to be compared with $0.029^\circ 2\theta$
 198 for the LaB_6 standard pattern). This broadening is known to be due to size effect and/or microstrain
 199 in the system. From our previous fit, the shape factors (ϕ) can be defined by $\phi = IB_{v0}/FWHM_{v0}$
 200 with IB_{v0} the observed integral breath (IB) and the observed Full Width at Half-Maximum (FWHM).
 201 The Lorentzian and Gaussian limits for the shape factors are respectively $\phi = 2/\pi \approx 0.6366$ and
 202 $\phi = 2(\frac{\log_e(2)}{\pi})^{1/2} \approx 0.93949$ [32]. Shape factor with $\phi < 0.6366$ and $\phi > 0.9394$ can be referred as super

203 Lorentzian and super Gaussian respectively [32]. Otherwise, it implies that the profile is a convolution
 204 of both shapes and justifies the use of a pseudo-Voigt.

205 It is well admitted now that, for small geometrical effects of a single line instrument, the
 206 instrumental profile is approximately Lorentzian, whereas the profile arising from lattice strain is more
 207 Gaussian [33,34]. The broadening due to defaults or small crystallite size depends respectively on
 208 their nature of the shape and the size distribution of the crystallites, and is assumed to be Lorentzian.
 209 These components can be taken as Voigtian, the Lorentzian and Gaussian parts are the limiting cases.
 210 If two or more reflections are available, size and strain effects can be determined from the variation of
 211 Lorentzian and Gaussian contributions in the IB depending on the hkl Miller indices [35].

212 We followed the procedure recommended by Langford [32], computing the FWHM and IB of a
 213 pseudo-Voigt profile from the broadened profile to give the shape factor, calculating the breadths of
 214 the constituent profiles corrected by the instrumental broadening and analyzing the Williamson-Hall
 215 [36] and Halder-Wagner [37,38] plots. The empirical procedure derived by de Keijser *et al.*[35] for
 216 calculating the Lorentzian and Gaussian components (β_L , β_G) and the integral breadth (IB) has
 217 been used in the Fullprof software.

218 The figure 4.a presents the dependence of the shape factor for four typical samples (NR 04-07-09-10)
 219 and the most different one (NR13), all the peaks show a shape factor between Gaussian and Lorentzian
 220 limits. The shape factor may not correlated with the Gd concentration (Fig 4.b) To understand
 221 the origin of the broadening of our peaks, possibly due to both size and microstrain contribution,
 222 the Halder-Wagner method [37,38] is recommended with such profile parameters instead of a
 223 Williamson-Hall approach [36], as the peaks have not a pure Lorentzian shape. Note that the fit
 224 is not so dependent on the 2θ range, even if Halder-Wagner plot insists on the small 2θ values (however
 225 the most intense peaks).

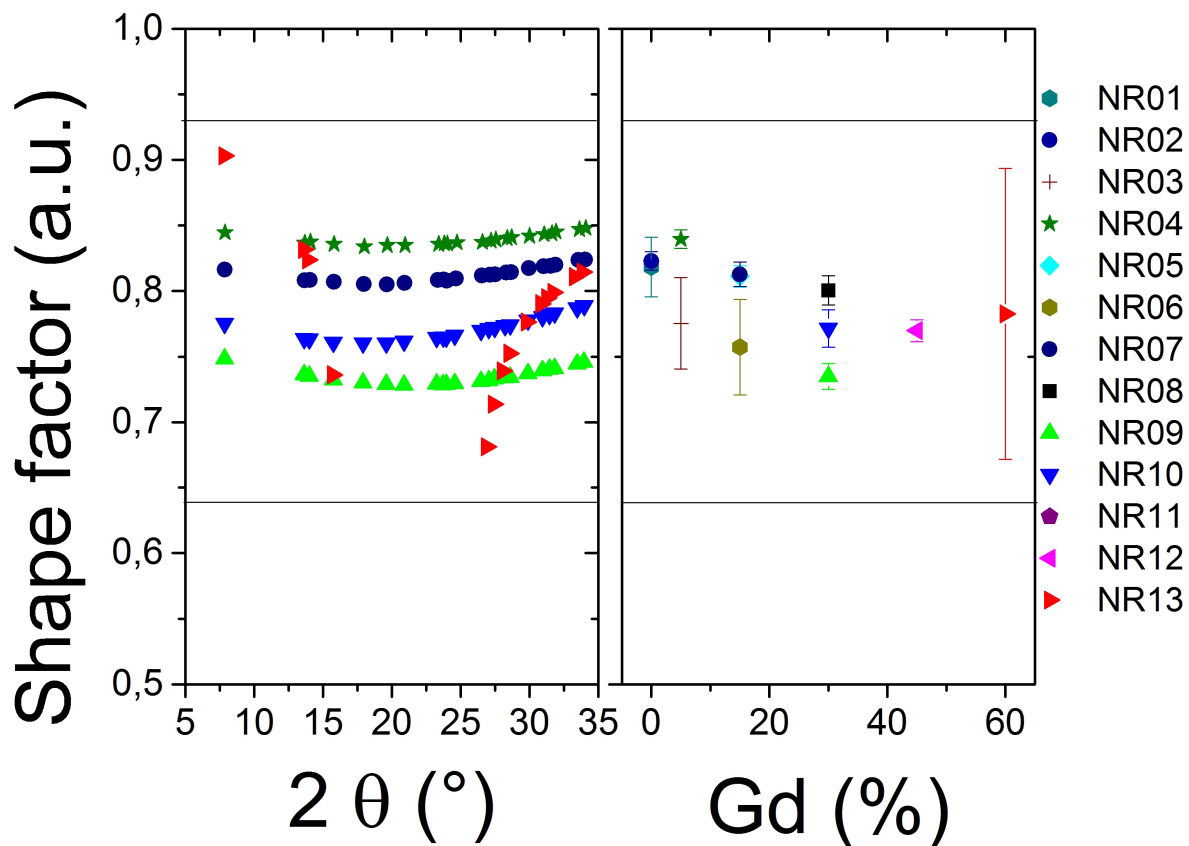


Figure 4. (a) Shape factor for each peak of four typical samples (NR04, NR07, NR09, NR10) and the most different sample (NR13) (b) Mean shape factor versus the expected gadolinium concentration

	TEM size (nm)		AR	Gd (mole %)	$y = ax + b$		shape factor	Crystallite size (nm)	microstrain
	L	D			a	b			
NR01	835	113	7.4	0	1.16	1.69	0.89	85	2.5e-4
NR02	1390	234	5.9	0	0.92	0.42	0.82	108	1.3e-4
NR03	718	108	6.6	5	1.35	0.29	0.77	74	1e-4
NR04	935	160	5.8	5	0.98	0.46	0.83	102	1.35e-4
NR05	375	70	5.4	15	1.69	0.79	0.81	59	1.7e-4
NR06	465	70	6.6	15	1.44	2.9	0.76	69	3.4e-4
NR07	640	100	6.4	15	1.3	0.8	0.81	77	1.8e-4
NR08	400	55	7.2	30	2.01	0.36	0.8	50	1.2e-4
NR09	156	62	2.5	30	2.01	-0.21	0.73	50	n.s.
NR10	130	70	1.9	30	2.04	0.53	0.77	49	1.5e-4
NR11	260	55	4.7	30	2.21	1.08	0.80	45	2e-4
NR12	137	45	3	45	2.83	0.72	0.77	35	1.7e-4
NR13	60	45	1.3	60	3.9	-6	0.78	26	n.s.

Table 4. Main parameters of the studied NPs (with L for Length, D for Diameter, AR for Aspect Ratio) and the parameters from the Halder-Wagner graph with the apparent size and microstrain obtained for each of them. Here only the expected gadolinium concentration is given. (n.s. negative microstrain even if the value is very close to zero)

226 To distinguish between size and microstrain respective contributions, the Halder-Wagner equation
 227 is written as $(b^*/d^*)^2 = IB^*/(d^*)^2$. By plotting this equation a linear behavior is observed for all
 228 our samples (Fig. 5). The intercept of the linear regression in the Halder-Wagner plot is related to
 229 the importance of microstrain in the material and, on the other hand, the slope is related to a mean
 230 apparent size (in volume) of the NPs.

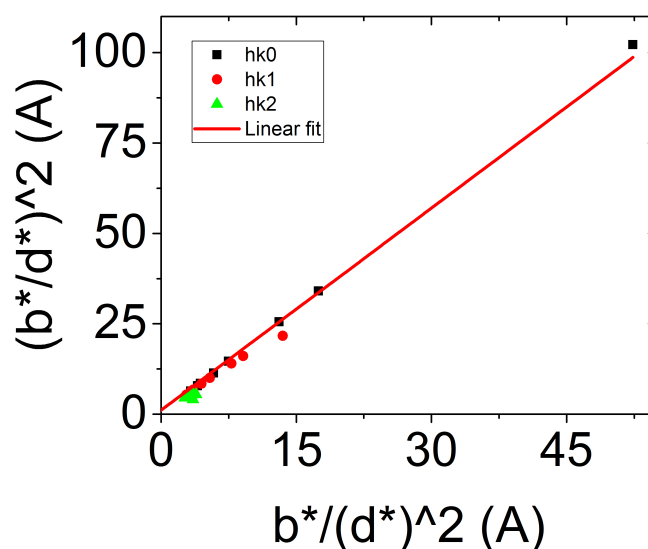


Figure 5. Halder-Wagner plot for a typical sample

231 In Fig. 5, here for NR08, a clear linear regression is observed, with an origin close to zero. This
 232 behavior is common to all our synthesized NPs, as summarized in table 4. Whatever the sample the
 233 microstrains are close to zero. For the rest of the study, the broadening of the Bragg peaks is then
 234 considered to be the broadening of a size effect only, and the microstrain effect can be neglected. One
 235 can also notice that the characteristic size of the NPs obtained from the Halder-Wagner plot is smaller
 236 than the one obtained by TEM imaging. In first approximation, it appears that the size obtained by
 237 XRD is close to the diameter of the NPs observed by TEM. This lead us to study more carefully the
 238 crystallinity of our NPs to understand the meaning of this characteristic size.

	TEM size (nm)		AR	Gd (mol%)	Crystallite size (nm)	
	L	D			min	max
NR01	835	113	7.4	0	77	100
NR02	1390	234	5.9	0	143	153
NR03	718	108	6.6	5	79	180
NR04	935	160	5.8	5	132	142
NR05	375	70	5.4	15	65	77
NR06	465	70	6.6	15	63	69
NR07	640	100	6.4	15	89	93
NR08	400	55	7.2	30	58	102
NR09	156	62	2.5	30	52	110
NR10	130	70	1.9	30	57	74
NR11	260	55	4.7	30	51	87
NR12	137	45	3	45	40	92
NR13	60	45	1.3	60	36	36

Table 5. Main parameters of the studied NPs (with L for Length, D for Diameter, AR for Aspect Ratio) and the apparent size obtained by the anisotropic model. Here only the nominal gadolinium concentration is given.

239 The previous observation was assuming an isotropic NPs shape, supposed to be spherical. Now
 240 a Scherrer-type analysis was used, where only the Lorentzian contribution is considered for each
 241 individual reflection in order to take into account eventual anisotropic broadening, to obtain the size
 242 of the crystallite, a needle-like model was applied - with a main axis along the c axis or [OOI] direction.
 243 Remember that by this method it is at best a mean value of the size of the coherent domains of a
 244 supposedly unimodal distribution of crystallites which is obtained for each (hkl) Bragg peak.

245 From this analysis, the three dimensions of the crystallites in our NPs can be extracted. All of them
 246 present a longer length along the [OOI] direction and a smaller size in the corresponding orthogonal
 247 [hk0] directions. Moreover, in most cases, the length along this direction [hk0] is slightly smaller than
 248 the apparent size observed by TEM. We summarized the crystallite sizes in the table 5 with again the
 249 apparent size for an easier comparison.

250 As previously said, the smallest length of the crystallites is associated to the directions orthogonal
 251 to c-axis. Moreover, for NPs with a diameter below 100 nm, this length is almost always slightly
 252 smaller of few nanometers than the diameter of the NPs. This fact was explained with the HRTEM
 253 observation. In Fig. 6.a or b, we observed an amorphous phase or a less crystallized shell around the
 254 NPs. Another observation, more surprising from our point of view, is the possibility to have a well
 255 crystallized phase in an orthogonal direction surrounding the core of the NPs. The inset of Fig. 6.a
 256 shows the FFT of the area delimited by the dotted green line. This FFT presents a rectangular signal
 257 characteristic from an hexagonal structure seen orthogonally of the c-axis. Even though the NPs in Fig.
 258 6.c is a rod well defined, the FFT of the area delimited by the continuous red line is characteristic from
 259 an hexagonal phase through the c-axis. Note that the crystallinity of the NPs presented is not altered
 260 by the electron beam of the TEM.

261 This observation is important if we want to analyze the luminescence of the NPs. It is well
 262 known that lanthanide ions have a specific luminescence correlated with their local environment. For
 263 a structure like $P6_3/m$, the doping ions have access to only one type of site as mentioned in table
 264 2. The luminescence and in particular the polarized properties will be depending of this site (here
 265 C_{3h}^2)[28]. But if the crystallinity of the NPs is not good meaning polycrystallines nanorods instead
 266 of monocrystalline and the lanthanide-site are not well aligned with a well-defined c-axis along the
 267 whole rod, the optical properties will be affected starting with a reduction of the polarization degree.
 268 Moreover, for singles NPs studies, due to statistical variation from one NP to another, it will be difficult
 269 to analyze the optical properties.

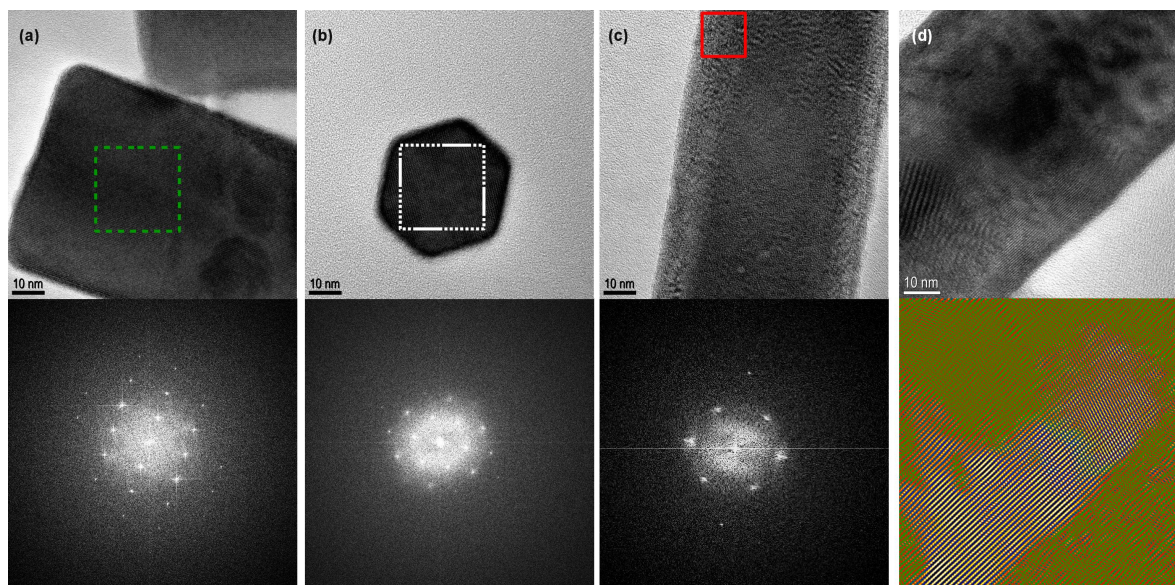


Figure 6. (a-c) HRTEM of NPs with below the FFT of the indicate area (a) NR09 with a typical FFT orthogonal to the c-axis (b) NR13 with a typical FFT parallel to the c-axis (c) NR11 with a FFT characteristic from an hexagonal phase trough the c-axis but observed orthogonally to the c-axis. (d) HRTEM of a NR06 with various defects below same image filtered with false color.

270 4. Conclusion

271 Many works have reported the synthesis, optical properties and application of Na(Gd)YF₄:Yb, Er
 272 nanoparticles. The large interest for this systems comes both from their highly appealing emission
 273 properties, and the ability to synthesis particles with different size and shapes. In particular, very nice
 274 faceted nanorods can be obtained with adjustable aspect ratio playing on the particles composition
 275 such as some Gd³⁺ ions in substitution to Y³⁺. Nevertheless, characterizations of polarized emission
 276 from single nanorods evidence that the particles are not monocrystals, which motivated this study to
 277 get a deeper insight in the particles microstructure. This work has been done using TEM and XRD
 278 analysis on a set of 13 samples obtained under various conditions. We could show that in a given
 279 batch of synthesis, and focusing on particles with the β -NaYF₄ structure, different populations of
 280 particles may coexist with different size and shape, but also with strong deviations from the nominal
 281 composition. XRPD analysis using the Halder-Wagner approach, combined with HRTEM point
 282 out an anisotropic polycrystalline structure of the nanorods with important local variations of the
 283 crystallites orientations. This is clearly at the origin of the polarization spectra, which is very sensitive
 284 to crystallites orientation.

285 As NaYF₄ NPs are now widely used for their interesting optical properties, these observations
 286 show that an exhaustive structural and chemical analysis is compulsory in the aim to understand and
 287 optimize this system for the targeted applications.

288 **Author Contributions:** Conceptualization, Godefroy Leménager and Sandrine Tusseau-Nenez; Data curation,
 289 Godefroy Leménager and Sandrine Tusseau-Nenez; Formal analysis, Godefroy Leménager and Sandrine
 290 Tusseau-Nenez; Funding acquisition, Thierry Gacoin; Investigation, Godefroy Leménager; Methodology, Godefroy
 291 Leménager and Sandrine Tusseau-Nenez; Project administration, Godefroy Leménager; Resources, Godefroy
 292 Leménager, Sandrine Tusseau-Nenez, Maud Thiriet, Pierre-Eugène Coulon, Khalid Lahlil, Eric Larquet and
 293 Thierry Gacoin; Software, Sandrine Tusseau-Nenez; Supervision, Godefroy Leménager; Validation, Godefroy
 294 Leménager, Sandrine Tusseau-Nenez and Thierry Gacoin; Visualization, Godefroy Leménager; Writing – original
 295 draft, Godefroy Leménager and Sandrine Tusseau-Nenez; Writing – review editing, Godefroy Leménager,
 296 Sandrine Tusseau-Nenez and Thierry Gacoin.

297 **Funding:** Funding for this project was provided by the French National Research Agency in the frame-work of
 298 the FiPlaNT project (ANR-12-BS10-012).

309 This work was supported by the French ANR research programm "Investissements d'avenir"
300 (ANR-10-EQPX-50).

301 **Acknowledgments:** We acknowledge the European Synchrotron Radiation Facility for provision of synchrotron
302 radiation facilities and we would like to thank Adeline Adam, Lucio Martinelli and Isabelle Maurin for the help
303 during the experiment.

304 **Conflicts of Interest:** The authors declare no conflict of interest.

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