

## LaFeO 3 thin films as relevant models for the surface investigation of 3-way catalysts

Shreya Nandi, Dimitri Blanck, Thomas Carlier, Marie-Hélène Chambrier, Anne-Sophie Mamede, Martine Trentesaux, Pardis Simon, Nicolas Nuns, Pascal Roussel, Anthony Ferri, et al.

#### ▶ To cite this version:

Shreya Nandi, Dimitri Blanck, Thomas Carlier, Marie-Hélène Chambrier, Anne-Sophie Mamede, et al.. LaFeO 3 thin films as relevant models for the surface investigation of 3-way catalysts. Surface and Interface Analysis, 2018, 50 (11), pp.1018-1024. 10.1002/sia.6450. hal-02309188

HAL Id: hal-02309188

https://hal.science/hal-02309188

Submitted on 9 Oct 2019

**HAL** is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

# LaFeO<sub>3</sub> thin films as relevant models for the surface investigation of three-way catalysts

Shreya Nandi<sup>†</sup>, Dimitri Blanck<sup>†</sup>, Thomas Carlier<sup>‡</sup>, Marie-Hélène Chambrier<sup>‡</sup>, Anne-Sophie Mamede<sup>†</sup>, Martine Trentesaux<sup>†</sup>, Pardis Simon<sup>†</sup>, Nicolas Nuns<sup>§</sup>, Pascal Roussel<sup>†</sup>, Anthony Ferri<sup>‡</sup>, Jean-François Paul<sup>†</sup> and Elise Berrier<sup>†,\*</sup>

† Univ. Lille, CNRS, ENSCL, Centrale Lille, Univ. Artois, UMR 8181,UCCS - Unité de Catalyse et Chimie du Solide, F-59000 Lille, France

‡Univ. Artois, CNRS, Centrale Lille, ENSCL, Univ. Lille, UMR 8181, Unité de Catalyse et Chimie du Solide (UCCS), F-62300 Lens, France

§IMMCL Chevreul, Institut des Molécules et de la Matière Condensée, F-59650 Villeneuve d'Ascq, France

★ Corresponding author: Elise.Berrier@univ-lille.fr

Note: This is the peer reviewed version of the following article: Nandi, S., Blanck, D., Carlier, T., Chambrier, M. H., Mamede, A. S., Trentesaux, M., Simon, P., Nuns, N., Roussel, P., Ferri A., Paul, J. F. and Berrier E. (2018). LaFeO $_{3\pm\delta}$  thin films as relevant models for the surface investigation of 3-way catalysts. Surface and Interface Analysis, 50(11), 1018-1024., which has been published in final form at https://doi.org/10.1002/sia.6450. This article may be used for non-commercial purposes in accordance with Wiley Terms and Conditions for Use of Self-Archived Versions.

**Abstract:** Lanthanum orthoferrite, a highly potential three-way catalyst, shaped as apolycrystalline thin film has been comprehensively analysed by combining bulk and surface characterization techniques. The possibility to accomplish unprecedented surface information has been presented, thanks to the combined use of LEIS, XPS and ToF-SIMS. The structural, morphological and surface properties at nano-metric scale make such thin films indistinguishable from powdered solids. Thus, the relevance of using such model materials for advanced surface investigations of LaFeO $_{3\pm\delta}$ -based three-way catalysts has been demonstrated.

**Keywords:** perovskite, LaFeO<sub>3</sub>, Raman, XPS, ToF-SIMS, LEIS

#### 1 Introduction

10

15

LaFeO $_{3\pm\delta}$  (LFO) solids showing ABO $_3$  perovskite structure have been demonstrated to be smart support materials in three-way catalysis (TWC) for the simultaneous abatement of CO, NO and unburnt hydrocarbons in the catalytic converters of automotive engines. Their application in such a system is highly suitable owing to their structural thermal stability under redox conditions [1], the possibility to fine-tune their composition by substitution in A and/or B site(s) [2] and their consistent oxygen storage capacity (OSC) [3].

In heterogeneous catalysis, only the outermost atomic layers of the active material, being in contact with the reactants, can directly participate in the reaction. On the other hand, the subsurface can help in maintaining and/or regenerating the active phase, which is the starting point of the key concept of *self-regeneration* [4, 5]. Moreover, it was recently shown that the composition of the perovskite oxide plays a key role in the regeneration process. In return, the versatile composition of oxide perovskite materials can help improving this crucial benefit [6]. In the context of reducing the amount of critical raw materials while maintaining a high activity in TWC, lanthanum perovskite materials with extremely low to zero noble metal content nowadays appear as quite a promising entry to the formulation of future 3-way catalysts [7]. Quite recently, we have reported on an *in situ* Raman study of LaFeO $_{3\pm\delta}$  and proposed that the surface composition and reconstruction played a crucial role in TWC performances [8]. More precisely, the mobility of FeO $_6$  moieties in and out the LFO lattice upon redox cycling was proposed to take part in the activity enhancement of La-deficient materials [9]. Following this hypothesis lies the capacity of iron-based moieties to move reversibly from the sub-surface to the surface mediated by the precipitation of small Fe $_2$ O $_3$  nanoparticles.

Shaping LaFeO $_{3\pm\delta}$  as Si-supported, polycrystalline thin films is expected to yield few layers of nanocrystallites and hence, provide a chance for a better description of the active phase located on the catalyst's (sub)surface. In practice, such thin films are also likely to favour a more suitable surface analysis. For example, perovskites shaped as thin films make it possible to afford an elemental depth profiling using ToF-SIMS, which is not as efficient with the corresponding powders. Distantly adapted from the well-accepted concept of investigating monocrystalline surfaces as model active materials [10], the flat model catalyst approach we apply here, however, does not involve such a material gap [11, 12]. To the best of our knowledge, the application of this method is original in the investigation of catalytically relevant LaFeO $_3$  thin films. Such a model material paves the way to a clear depiction of the surface and sub-surface that determine the catalytic behaviour of active bulk materials. However, an obvious prerequisite to the use of LaFeO $_{3\pm\delta}$  thin films as convenient model shapes for in-depth surface analyses lies in the structural and textural analogy with lab-scale active materials, such as powdered catalysts. The present work aims at clarifying the relevance of polycrystalline thin films in being used as a model surface/sub-surface system for investigating LaFeO $_{3\pm\delta}$ -based three-way catalysts.

#### 2 Materials and Methods

## 2.1 Preparation of polycrystalline LFO thin films

The preparation of the precursor sol was largely based on the one of Liu *et al.* for preparing alumina supported LaFeO<sub>3± $\delta$ </sub> films[13]. 0.65g of lanthanum nitrate hexahydrate (99.9 %, Aldrich) and 0.61 g of iron nitrate nonahydrate (99.9 %, Aldrich) were diluted in millipore water (15.0 M $\Omega$ cm at 17 °C) for preparing 5 mL of bimetallic solution having a metal concentration of [Fe]=[La]=0.3 mol L<sup>-1</sup>. The solution was transferred in a round flask and vigorously stirred. Citric acid (99.5 %, Sigma-Aldrich) was

then added in order to reach a concentration of  $0.6\,\mathrm{mol\,L^{-1}}$ , thus leading to a citric acid:(Fe+La) molar ratio of 1. Further,  $25\,\mu\mathrm{L}$  ethylene glycol (99.5 %, Fluka) and  $700\,\mu\mathrm{L}$  aqueous ammonium hydroxide (25%, Fluka) were added to facilitate the polymerization and adjust the pH, respectively. The solution was then heated under reflux at  $50\,^{\circ}\mathrm{C}$  overnight. Subsequently, the precursor sol was further heated at  $55\,^{\circ}\mathrm{C}$  in air until the total volume of the sol was decreased to one half of its initial value.

Si(100) substrates (Sil'Tronix) were cleaned by successive immersion in acetone (10 minutes) and isopropanol (10 minutes) in an ultrasonic bath. After rinsing with millipore water, the substrates were immersed in freshly prepared piranha solution (vol. 1:1  $H_2O_2/H_2SO_4$ ) for 20 minutes and rinsed well with millipore water and stored in the same for transportation. Before deposition, the substrate was characterized confirming a 1-2 nm of amorphous silica layer covering Si(100).

The sol was then deposited at room temperature on the dried  $SiO_2/Si(100)$  substrates by spin coating with a typical speed of 2000 rpm. The layers were further heated up to  $150\,^{\circ}$ C for 2 hours (ramp:  $0.5\,^{\circ}$ C min<sup>-1</sup>) and subsequently heated up to  $650\,^{\circ}$ C for 4 hours (ramp:  $1.0\,^{\circ}$ C min<sup>-1</sup>) in a muffle oven. A thin film appearing transparent, slightly orange-coloured and showing only few, tiny cracks was thus

obtained.

55

60

65

70

75

80

#### 2.2 X-Rays Diffraction (XRD)

The X-ray diffraction structural characterization of the LaFeO<sub>3± $\delta$ </sub> film was carried out using a 9 kW Rigaku Smartlab rotated anode X-ray diffractometer using Cu K<sub> $\alpha$ </sub> (1.5418 Å) wavelength, operated in Bragg-Brentano reflexion geometry. To avoid any saturation of the detector due to the (004) peak of the oriented Si substrate, the sample was intentionally disoriented by 2°, once it was verified that the deposition was not epitaxial but polycrystalline.

#### 2.3 Raman Analyses

Raman spectra were recorded using an  $Ar^+$  (488 nm) laser for excitation, the power at the sample was adjusted using neutral density filters for the purpose of the experiments. The laser beam was focused at the surface of layered samples using a 100X microscope objective (NA 0.9). The Raman signal was collected in backscattering mode using the same objective through a confocal hole of  $100\,\mu m$  diameter. The scattered light was dispersed via a spectrometer equipped with a 1800 grooves grating (typical spectral resolution at 488 nm:  $2\,cm^{-1}$ ) and finally analysed using a Peltier-cooled CCD (Horiba Labram HR).

### 2.4 Surface properties

**Atomic Force Microscopy (AFM) analysis** The surface morphology of the films was locally investigated using a commercial atomic force microscope (AFM) (MFP-3D, Asylum Research) working in AC

-or tapping- mode under ambient conditions. Si tips (Nanoworld Arrow-NC probes) with cantilever stiffness of  $40~\rm Nm^{-1}$  were used as local probes.

Time of Flight -Secondary Ion Mass Spectrometry (ToF-SIMS) analysis ToF-SIMS depth profiling was performed in positive mode using a TOESIMS5 instrument (ION-TOF, GmbH Germany), equipped with a bismuth liquid metal ion gun. Pulsed Bi<sup>+</sup> primary ions have been used for analysis (25 keV, 1 pA) while the sputtering was performed using a Cs<sup>+</sup> (1 keV, 80 nA) ion source running in non-interlaced mode [14]. The analysed surface was  $100 \, \mu m \, X \, 100 \, \mu m$  centred over the  $300 \, \mu m \, X \, 300 \, \mu m$  sputtered crater. Charging effects were compensated using pulsed low energy electrons (20 eV).

X-Ray Photoelectron Spectroscopy (XPS) analysis XPS characterization was carried out using the Kratos AXIS Ultra<sup>DLD</sup> spectrometer using a monochromatic Al K<sub> $\alpha$ </sub> X-ray source (10 mA, 12 kV). C 1s, O 1s, Fe 2p and La 3d high resolution spectra were obtained using a 20 eV pass energy while keeping a pressure in the analysis chamber below  $5 \times 10^{-9}$  mbar during analysis. All spectra have been calibrated by giving the adventitious C 1s spectral component (C-C, C-H) a binding energy (BE) of 284.8 eV. A Shirley background [15] was subtracted from the spectra for quantification. The analysed area was about 300 μm X 700 μm.

**Low-Energy Ion Scattering (LEIS) analysis** LEIS experiments were run using a Qtac100 instrument (ION-TOF GmbH) under a working pressure range of  $1 \times 10^{-8}$  mbar. The samples were analysed using a  $3 \, \text{keV}^4 \text{He}^+$  primary ion beam directed perpendicular to the target surface. The beam current was typically set to 5 nA. The experiments have been performed at various ionic doses, the analysed area was kept  $1000 \, \mu \text{m} \, \text{X} \, 1000 \, \mu \text{m}$ . The samples were treated *in situ* under atomic oxygen plasma prior to LEIS analyses.

#### 3 Results and Discussion

85

90

95

100

105

110

115

## 3.1 Surface morphology of LFO films

The 100X magnification picture of LFO, presented in Fig. 1a shows leaf-like patterns with orange ridges and dark blue gullies. The AFM image shown in Fig. 1b revealed that the ridges are consistent with zones featuring thicker deposits while gullies are characteristic of thiner LaFeO<sub>3± $\delta$ </sub> layers. The peak-to-valley roughness reaches 160 nm while the average level of thickness is equal to 70 nm±10 nm.

Fig. 1c shows that the LFO film is made of nano-sized aggregates similar in size and shape to those observed in powders prepared from the citrate-route [16]. The root mean squared roughness (RMS),  $R_q$ , defined by  $R_q = \sqrt{\frac{1}{n}\sum_{i=1}^n y_i^2}$ , where  $y_i$  is the level of the  $i^{th}$  of n data points, was found to reach about 15.6 nm. Interestingly, the average radius of LaFeO<sub>3± $\delta$ </sub> nanocrystallites from Figure 1c reaches 68 nm ± 3 nm, which matches well the depth of the deposit. This suggests that a single

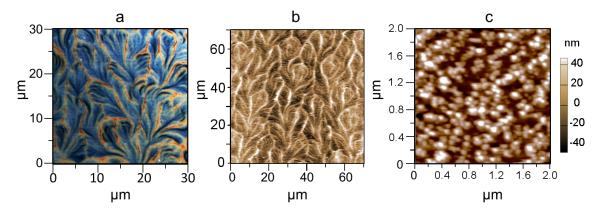


Figure 1: a: 100X magnification optical microscopy image, b: large-scale AFM image and c: AFM image on smaller area of LFO film.

layer of LaFeO<sub>3 $\pm\delta$ </sub> nanocrystals was deposited. It is to be noticed from Fig.1c, however, that between theLaFeO<sub>3 $\pm\delta$ </sub> nanocrystals nanosized voids are present.

#### 3.2 Structure of LFO layers

#### 3.2.1 X-Ray diffraction (XRD)

120

The diffraction pattern of the LFO layer displayed in Fig. 2 shows every reflection of the polycrystalline perovskite LaFeO<sub>3± $\delta$ </sub> crystallised in orthorhombic lattice with lattice constants a=5.553 Å, b=5.563 Å and c=7.867 Å (space group: Pbnm) [17].

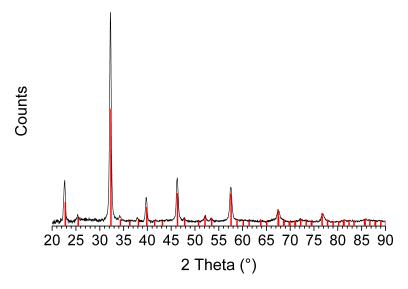


Figure 2: X-ray diffractogram of LaFeO $_{3\pm\delta}$  thin film; red stick diagram: diffraction pattern of LaFeO $_{3\pm\delta}$  (PDF 74-2203)

No reflection of any other crystalline phase such as La<sub>2</sub>O<sub>3</sub>, La(OH)<sub>3</sub> or Fe<sub>2</sub>O<sub>3</sub> is observed.

#### 3.2.2 Raman spectroscopy

125

130

135

140

The typically analysed sample depth is approximately  $1\,\mu m$ , which is one order of magnitude higher than the film's thickness. As a result, the Raman spectra recorded by focusing the laser beam at the surface of the layers are largely dominated by the signature of crystalline silicon found in bare  $SiO_2/Si(100)$ , as can be seen in the spectra presented in Fig. 3, bottom.

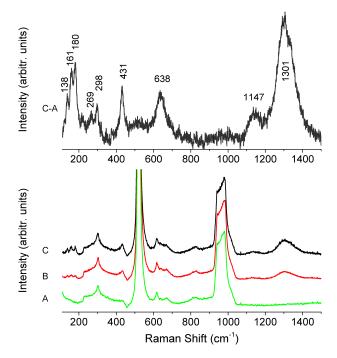


Figure 3: Bottom: Raman spectra of bare  $SiO_2/Si(100)$  (A) and of LFO thin film recorded in either a gully (B) or a ridge region (C). Top: Difference spectrum (C-A);  $\lambda_{exc.}$ =488 nm, acquisition time: 30 s, three scans were averaged.

However, the signature of polycrystalline lanthanum orthoferrite (LFO) can be handily retrieved by subtracting the Raman spectrum of the LFO thin film by the one of pure Si recorded under the same conditions. The difference spectrum, presented on top of Figure 3 C-A is similar in every respect to the Raman profile of the very same materials shaped as powders [9]. The assignment of the Raman bands of LFO have been already largely reported [18–20]. In brief, the modes below  $200\,\mathrm{cm}^{-1}$  are related to La motions, the doublet of bands observed at 269 and  $298\,\mathrm{cm}^{-1}$  are oxygen octahedral tilt (T) modes while the band at  $431\,\mathrm{cm}^{-1}$  is assigned to the bending vibration of  $FeO_6$  octahedra. The band at  $638\,\mathrm{cm}^{-1}$  is assigned to Fe-O and La-O stretch vibrations. Broad bands of relatively high intensity are also detected at  $\sim 1147$  and  $\sim 1300\,\mathrm{cm}^{-1}$ , which were previously assigned to second order scattering of IR-active longitudinal optic (LO) phonon modes. To sum up, in spite of the relatively low share of LFO phase as compared to the volume probed by Raman spectroscopy, LFO bands of high enough intensity are detected, which paves the way to further *operando* Raman analyses.

#### 3.3 Surface analysis of LFO thin films

145

150

155

160

**X-ray Photoelectron Spectroscopy (XPS)** The XPS spectra of the LaFeO<sub>3± $\delta$ </sub> thin film for C 1s, La 3d, Fe 2p, and O 1s are presented in Fig. 4. The C 1s photopeak presents an aliphatic peak together with weaker features at higher binding energies (BE) assigned to hydroxyl and carbonyl surface contaminations. The peak at BE of 289.2 eV with 11% of the total carbon peak area is attributed to the presence of adsorbed carbonate species at the surface of the LaFeO<sub>3± $\delta$ </sub> film, as was previously also observed in the lanthanum-based corresponding powders [8].

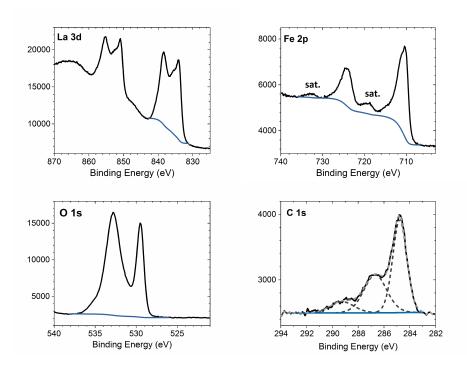


Figure 4: La 3d, Fe 2p, O 1s and C 1s spectra of LaFeO<sub>3 $\pm\delta$ </sub> thin film

The high-resolution spectrum of La 3d exhibits two peaks localized at 834.3 eV and 851.0 eV corresponding to the spin-orbit splitting of La  $3d_{5/2}$  and La  $3d_{3/2}$  of La<sup>3+</sup> ion in its oxide form [21–24]. Each of the spin-orbit peak is further split by multiplet splitting components. The multiplet structure associated to La  $3d_{5/2}$  peak shows a magnitude of 4 eV. This value corresponds to the presence of lanthanum hydroxide [21]. Indeed, owing to the reactive nature of La compounds which react in ambient air with water and  $CO_2$  to form La hydroxides and carbonates, it is possible that this La 3d component is consisting of contributions from surface hydroxides and carbonates besides the mixed oxide. The binding energies of Fe  $2p_{3/2}$  and Fe  $2p_{1/2}$  are 710.5 eV and 724.2 eV, respectively, which is in line with the BE of Fe<sup>3+</sup> ions in their oxide form. This is further confirmed by the 8.6 eV BE separation between the main Fe  $2p_{3/2}$  peak and its satellite (sat.), characteristic of Fe<sup>3+</sup> [25]. La 3d and Fe 2p core-level spectra thus revealed that both La and Fe stand in +3 oxidation state and their respective areas allows to calculate an atomic La:Fe ratio of 1.8. Only the La  $3d_{5/2}$  peak was considered for quantification keeping in mind the overlapping of La MNN Auger peak with the La  $3d_{3/2}$  component [21]. Compared to the

nominal composition, the LaFeO $_{3\pm\delta}$  thin film exhibits an excess of lanthanum at the surface whose value is close to what has been already reported on powdered samples [8].

The high-resolution spectrum of O 1s of the perovskite film is dominated by two intense components at  $529.5 \, \text{eV}$  and  $532.8 \, \text{eV}$ . The first one is in line with previous studies of contribution of bulk oxygen lattice involved in lanthanum oxide and iron oxide groups [21, 24, 26]. The second contribution centered at  $532.8 \, \text{eV}$  is assigned to oxygen from hydroxyl groups ( $\sim 531 \, \text{eV}$ ), organic oxygen ( $\sim 532 \, \text{eV}$ ) and silica due to the existence of a silica over the Si substrateappearing from the voids arising from inhomogeneity of the film at both micro- and nanoscale, as discussed previously.

165

**Low-Energy Ion Scattering (LEIS)** The composition of the outermost surface of the LaFeO<sub>3± $\delta$ </sub> thin film was investigated by means of LEIS and the corresponding spectra are presented in Fig. 5. Under static conditions, as in Fig. 5a, both La and Fe feature low-intensity scattering peaks detected at 2671 and 2313 eV, respectively, which confirms that the outermost LaFeO<sub>3± $\delta$ </sub> surface contains both La and Fe species. The coverage of the LaFeO<sub>3± $\delta$ </sub> nanocrystals over the silica/silicon surface induced interparticle voids, see Fig. 1c, which can explain the detection of the low intensity Si peak at 1760 eV. Upon increasing the ionic dose (see, for example, Fig 5c, d and e), both Fe and La peaks were found to gradually and jointly increase in intensity.

170

175

180

185

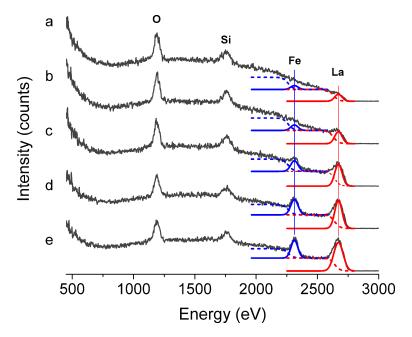


Figure 5:  $^4\text{He}^+$  at 3 keV LEIS spectra of LaFeO $_{3\pm\delta}$  thin film recorded using an ionic dose of a:  $2.0\times10^{14}~\text{cm}^{-2}$ , b:  $1.5\times10^{16}~\text{cm}^{-2}$ , c:  $5.9\times10^{16}~\text{cm}^{-2}$ , d:  $1.1\times10^{17}~\text{cm}^{-2}$  and e:  $1.5\times10^{17}~\text{cm}^{-2}$ ; Note that the spectra are shifted by a constant offset for clarity

An attempt for getting more information on the composition of the extreme surface of the film as probed by LEIS was made by considering the areas of Gauss LEIS surface peaks of Fe and La, further denoted as  $A_{Fe}$  and  $A_{La}$ , as a function of the ionic dose applied. The LEIS peaks areas presented in Fig. 6.

The extrapolated values of the raw areas of both La and Fe surface peaks at zero ionic dose suggest that both lanthanum and iron species are found in the extreme surface of the perovskite film. Besides, the  $\frac{A_{La}}{A_{Fe}}$  ratio, plotted against the ionic dose in Fig. 6, approaches a constant value close to 2. This suggests that, within the sample depth investigated by LEIS, the lanthanum excess evidenced by XPS translates into a relatively stable La/Fe ratio.

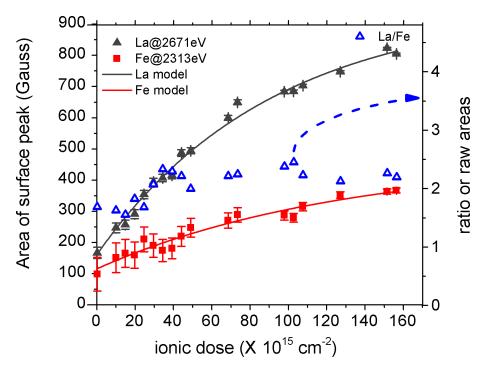


Figure 6: Evolution of the areas of Gauss profiles respectively assigned to Fe (red squares) and La (black triangles) surface peaks and the ratio of La:Fe raw areas (blue triangles) as a function of the applied ionic dose

#### 3.3.1 Time of flight - Secondary Ions Mass Spectrometry (ToF-SIMS)

190

195

200

205

210

The elemental depth profile was built using the relative SIMS intensities of the  $Cs_2O^+$  (281.80 a.m.u),  $CsFe^+$  (188.84 a.m.u),  $CsLaO^+$  (287.80 a.m.u) and  $Si^+$  (27.98 a.m.u) fragments, which are representative for O, Fe, La and Si-containing moieties, respectively. A constant sputter rate of  $0.08\,\mathrm{nm\,s}^{-1}$  was estimated on the basis of the AFM data, thus making it possible to convert sputter time into sample depth. Fig. 7 shows the evolution of the normalised intensities of the most relevant fragments, as detailed above.

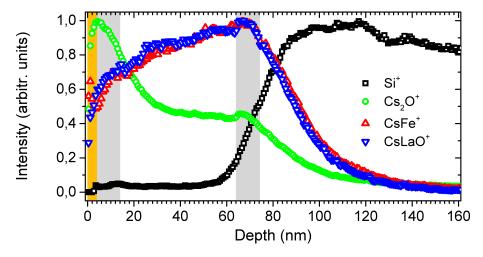


Figure 7: SIMS depth profile of Cs-enriched relevant fragments

A first layer of approximately 10 nm exhibits highest relative content in oxygen species, which can be related to contributions of carbonates species such as La<sub>2</sub>(CO<sub>3</sub>)<sub>3</sub> as well as hydroxyls from La, as suggested by our XPS results. Between 20 and 60 nm, both the concentrations of La- and Fe-containing moieties gently increase, as the contribution of oxygen declines. Having in mind that the LaFeO<sub>3+ $\delta$ </sub> phase is confirmed by XRD and Raman, we assume here that this moderate shift in the contribution of La, Fe and O-containing fragments does not reflect a genuine composition gradient, but is induced by an experimental bias. As a result, we propose that this region corresponds to a pure LaFeO<sub>3± $\delta$ </sub> phase which does not exhibit significant carbonate content. Between 60 and 75 nm, the depth profile features the coexistence of La, Fe, Si and oxygen species, which corresponds to the LaFeO<sub>3+ $\delta$ </sub>/SiO<sub>2</sub>/Si interface. Finally, only pure silicon is detected when the depth exceeds 120 nm. On the basis of the presented XPS and structural results, one would expect the depth profile to show a marked excess of La-containing fragments at the surface and a subsequent convergence of the normalised contributions of Fe- and La-containing species when entering the sample bulk. However, the depth profile shown here does not comply with this expectation at the outermost surface. Indeed, besides the skin effect and surface contaminants blurring the first five nanometers of SIMS intensities, both La- and Fe-moieties seems to evolve similarly.

An elemental mapping was also run over a wide sample surface and is presented in Fig. 8. It can be seen that La and Fe species are found in similar areas and that no obvious sizeable segregated domains

are detected within the investigated range.

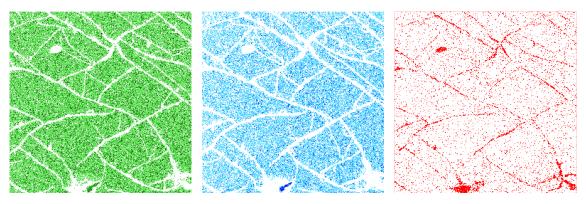


Figure 8:  $500\,\mu m\ X\ 500\,\mu m$  maps of Fe<sup>+</sup> (left), LaO<sup>+</sup> (middle) and Si<sup>+</sup>(right) fragments distribution as afforded from ToF-SIMS experiments

#### 4 Conclusion

215

220

225

A polycrystalline thin film of LaFeO $_{3\pm\delta}$  was prepared by spin coating over SiO $_2$ /Si(100) using the polymerisable complex method. The LaFeO $_{3\pm\delta}$  layer was shown to retain the same structural and surface properties as the powdered solids, which makes the as-prepared thin film a relevant model catalyst for investigating bulk catalysts based on LaFeO $_{3\pm\delta}$ . On the other hand, shaping LaFeO $_{3\pm\delta}$  into a thin film made it possible to attain unprecedented combined surface analyses. Moreover, the present work establishes the possibility to investigate such thin films using *in situ* Raman spectroscopy as well as advanced surface characterisation techniques in quasi-*in situ* or near-ambient pressure (NAP) XPS. This can enable us to establish relationships between the physico-chemical properties of perovskite model surfaces and the reactivity of the conventional catalysts.

## Acknowledgements

The present work is part of the PARTIAL-PGMs project funded by the European Commission under the grant agreement # 686086. *Chevreul Institute (FR 2638), Ministère de l'Enseignement Supérieur et de la Recherche, Région Nord - Pas de Calais* and *Fonds Européen de Développement Régional (FEDER)* within the *Pôle Régional d'Analyses de Surface* and the *Contrat de Plan Etat-Région (CPER)* project "Chemistry and Materials for a Sustainable Growth" are acknowledged for supporting and fundings. T. Carlier is grateful to the *Région Nord-Pas de Calais* for its financial support to carry out this work.

#### References

230

245

250

260

- [1] Pecchi G, Campos C, Peña O. Thermal stability against reduction of  $LaMn_{1-y}Co_yO_3$  perovskites. Mater Res Bull. 2009;44(4):846–853.
- [2] He H, Dai H, Au C. An investigation on the utilization of perovskite-type oxides  $La_{1-x}Sr_xMO_3$  (M=  $Co_{0.77}Bi_{0.20}Pd_{0.03}$ ) as three-way catalysts. Appl Catal B: Environ. 2001;33(1):65–80.
- 235 [3] Levasseur B, Kaliaguine S. Effects of iron and cerium in  $La_{1-y}Ce_yCo_{1-x}Fe_xO_3$  perovskites as catalysts for VOC oxidation. Appl Catal B: Environ. 2009;88(3):305–314.
  - [4] Nishihata Y, Mizuki J, Akao T, Tanaka H, Uenishi M, Kimura M, et al. Self-regeneration of a Pd-perovskite catalyst for automotive emissions control. Nature. 2002;418(6894):164–167.
- [5] Katz MB, Graham GW, Duan Y, Liu H, Adamo C, Schlom DG, et al. Self-regeneration of Pd– LaFeO3 catalysts: new insight from atomic-resolution electron microscopy. J Am Chem Soc. 2011;133(45):18090–18093.
  - [6] Kuc J, Zhang Y, Erni R, Yoon S, Karvonen L, Weidenkaff A, et al. Composition dependent self-regenerative property of perovskite-type oxides. Phys Status Solidi (RRL). 2015;9(5):282–287.
  - [7] Glisenti A, Pacella M, Guiotto M, Natile M, Canu P. Largely Cu-doped  $LaCo_{1-x}Cu_xO_3$  perovskites for TWC: Toward new PGM-free catalysts. Appl Catal B: Environ. 2016;180:94–105.
  - [8] Schön A, Dujardin C, Dacquin JP, Granger P. Enhancing catalytic activity of perovskite-based catalysts in three-way catalysis by surface composition optimisation. Catal Today. 2015;258:543–548.
  - [9] Blanck D, Schön A, Mamede AS, Dujardin C, Dacquin JP, Granger P, et al. *In situ* Raman spectroscopic evidence of surface sites potentially involved in the enhanced activity of La-deficient materials in 3-way catalysis (TWC). Catal Today. in press;
    - [10] Goodman DW. Model studies in catalysis using surface science probes. Chem Rev. 1995;95(3):523–536.
- [11] Gunter PL, Niemantsverdriet J, Ribeiro FH, Somorjai GA. Surface science approach to modeling supported catalysts. Catal Rev. 1997;39(1-2):77–168.
  - [12] Klisinska A, Mamede AS, Gaigneaux EM. Spin-coating of mixed citrate complexes as a versatile route to prepare films of transition metal multi-element oxide model catalysts with controlled formulation and crystalline structure. Stud Surf Sci Catal. 2006;162:745–752.
  - [13] Liu X, Ji H, Gu Y, Xu M. Preparation and acetone sensitive characteristics of nano-LaFeO<sub>3</sub> semiconductor thin films by polymerization complex method. Mat Sci Eng B-Solid. 2006;133(1):98–101.
    - [14] Iltgen K, Bendel C, Benninghoven A, Niehuis E. Optimized time-of-flight secondary ion mass spectroscopy depth profiling with a dual beam technique. J Vac SciTechnol A. 1997;15(3):460–464.
    - [15] Shirley DA. High-Resolution X-Ray Photoemission Spectrum of the Valence Bands of Gold. Phys Rev B. 1972 Jun;5:4709–4714.

- [16] Murade P, Sangawar V, Chaudhari G, Kapse V, Bajpeyee A. Acetone gas-sensing performance of Sr-doped nanostructured LaFeO<sub>3</sub> semiconductor prepared by citrate sol–gel route. Curr Appl Phys. 2011;11(3):451–456.
  - [17] Marezio M, Dernier P. The bond lengths in LaFeO 3. Mater Res Bull. 1971;6(1):23–29.
- [18] Andreasson J, Holmlund J, Rauer R, Käll M, Börjesson L, Knee CS, et al. Electron-phonon interactions in perovskites containing Fe and Cr studied by Raman scattering using oxygen-isoptope and cation substitution. Phys Rev B. 2008;78:235103.
  - [19] Popa M, Frantti J, Kakihana M. Lanthanum ferrite LaFeO<sub>3+ $\delta$ </sub> nanopowders obtained by the polymerizable complex method. Solid State Ionics. 2002;154-155:437–445.
- [20] Dubroka A, Humlíček J, Abrashev MV, Popović Z, Sapina F, Cantarero A. Raman and infrared studies of  $La_{1-y}Sr_yMn_{1-x}M_xO_3$  (M= Cr, Co, Cu, Zn, Sc or Ga): Oxygen disorder and local vibrational modes. Phys Rev B. 2006;73(22):224401.
  - [21] Sunding M, Hadidi K, Diplas S, Løvvik O, Norby T, Gunnæs A. XPS characterisation of in situ treated lanthanum oxide and hydroxide using tailored charge referencing and peak fitting procedures. Journal of Electron Spectroscopy and Related Phenomena. 2011;184(7):399–409.
- 280 [22] Phokha S, Pinitsoontorn S, Maensiri S, Rujirawat S. Structure, optical and magnetic properties of LaFeO3 nanoparticles prepared by polymerized complex method. J Sol-Gel Sci Technol. 2014;71(2):333–341.
  - [23] Su H, Jing L, Shi K, Yao C, Fu H. Synthesis of large surface area LaFeO3 nanoparticles by SBA-16 template method as high active visible photocatalysts. J Nanopart Res. 2010;12(3):967–974.
- 285 [24] Thirumalairajan S, Girija K, Mastelaro VR, Ponpandian N. Surface morphology-dependent room-temperature LaFeO<sub>3</sub> nanostructure thin films as selective NO<sub>2</sub> gas sensor prepared by radio frequency magnetron sputtering. ACS Appl Mater Interfaces. 2014;6(16):13917–13927.

290

- [25] Biesinger MC, Payne BP, Grosvenor AP, Lau LWM, Gerson AR, Smart RSC. Resolving surface chemical states in XPS analysis of first row transition metals, oxides and hydroxides: Cr, Mn, Fe, Co and Ni. Applied Surface Science. 2011;257(7):2717 2730.
- [26] Ramana C, Vemuri R, Kaichev V, Kochubey V, Saraev A, Atuchin V. X-ray photoelectron spectroscopy depth profiling of La<sub>2</sub>O<sub>3</sub>/Si thin films deposited by reactive magnetron sputtering. ACS Appl Mater Interfaces. 2011;3(11):4370–4373.