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Tuning graphene transistors through ad hoc electrostatics induced by a nanometer-thick molecular underlayer

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We report on the modulation of the electrical properties of graphene-based transistors that mirrors the properties of a few nanometers thick layer made of dipolar molecules sandwiched in-between the 2D material and the SiO₂ dielectric substrate. The chemical composition of the films of quinonemonoimines zwitterions molecules adsorbed onto SiO₂ has been explored by means of X-ray photoemission and mass spectroscopies. Graphene-based devices are then fabricated by transferring the 2D material onto the molecular film, followed by the top source-drain electrodes deposition. The degree of supramolecular order in disordered films of dipolar molecules was found to be partially improved as a result of the electric field at low temperatures, as revealed by the emergence of hysteresis in the transfer curves of the transistors. The use of molecules from the same family and suitably designed to interact with the dielectric surface yields a disappearance of the hysteresis. DFT calculations confirm that the dressing of the molecules by an external electric field exhibits multiple energy minima landscape that explains the thermally-stabilized capacitive coupling observed. This study demonstrates that the design and exploitation of ad-hoc molecules as interlayer between a dielectric substrate and graphene represents a powerful tool for tuning the electrical properties of the 2D material. Conversely, graphene can be used as an indicator of the stability of molecular layers, by providing insight into the energetics of ordering of dipolar molecules under the effect of electrical gating.

Introduction

In the last decade graphene has emerged as an exciting new material displaying exclusive optical, electrical, thermal and mechanical properties, with potential impact on many realms of science and technology.¹ Among numerous outstanding characteristics, its highest surface-to-volume ratio makes it extremely sensitive to environmental changes at the nanoscale. The unique electrostatic sensitivity of graphene to nearby molecules is key since it allows to tune the properties of the 2D material for applications in hybrid molecules-graphene based devices.² It is indeed notorious that the operation and performance of graphene-based solar cells,³ organic electronics⁴ and chemical/gas sensors^{5,6} critically depend on the interface

between the surface of the electrically conducting 2D material and the (conjugated) organic molecular layers. The vast majority of studies⁷ reported so far have focused on the interactions between graphene flakes supported on a solid substrate and molecules adsorbed on its top in the form of single-molecules,⁸ self-organized thin films^{9,10} or supramolecular architectures.¹¹ By fabricating planar FET devices, the graphene-substrate electrostatic interactions can possibly impact the transistor properties, with the nearby charges locally doping the graphene layer and shifting the neutrality point. Such electrostatic influence of adsorbed molecules on top of graphene is well-documented.^{12,13}

However, what lies underneath graphene also matters. Recent findings revealed that graphene can be 'chemically transparent' since the chemical environment underneath its surface affects the properties of the material grown on its top.¹⁴ Nevertheless, studies on the inverse FET geometry, based on graphene lying on top of a molecular layer, are much scarcer. To our knowledge, they are limited to functionalized substrates for charge screening purposes, aiming at improving the performance of the transistors by shifting their electrostatically-induced doping (or related work function)^{15,16,17} or to ferroelectric molecular layers designed for driving a very large graphene doping.¹⁸

We propose to exploit graphene as a tool to explore the properties of an underlying molecular film grown on a dielectric substrate. It

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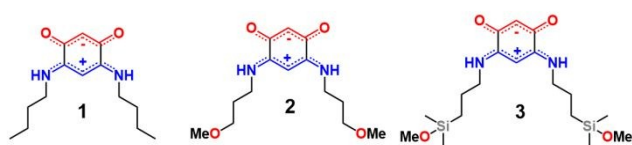
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is also fair to note that the characterization of molecular layers adsorbed onto dielectric surfaces is experimentally much more challenging than when conducting substrates are exploited. This is because surface imaging or spectroscopy tools, including scanning tunneling microscopy, (ultraviolet)photoemission or inverse photoemission spectroscopies, require substrates stable under electric charging.

Here, we investigate ultra-thin films of overall neutral yet dipolar molecules, which can be expected to influence the electronic properties of graphene, bypassing the complications related to the presence of counter-ions associated with charged species. We take advantage of our previous endeavor on the synthesis of a family of zwitterion molecules, their assembly from solution into ultra-thin films, and the study of their physical and chemical properties when adsorbed onto conductive substrates.^{19,20,21} Quinonemonoimines zwitterions (QZW, Scheme 1) are characterized by positive and negative charges on opposite sides of the six-membered ring that are electronically decoupled but chemically connected through carbon-carbon single bonds.^{22,23,24} The overall 12π electrons are partitioned in two 6π -electrons sub-units, with the positive charge delocalized between the nitrogen functions over four bonds involving 6π -electrons, while the negative charge is likewise spread between the oxygen atoms. This results in a remarkably large electric dipole of typically 10 D across the short planar six-membered ring structure of the benzoquinonemonoimine "core".²³ We previously detailed how these molecules are prone to undergo spontaneous self-assembly into molecular layer onto gold substrates and show unique electronic properties, making them of interest for enhanced metal-organic interfaces of relevance for organic electronics applications.²⁵ Chemical modification of the nitrogen-bound end groups represents a facile strategy for the tuning of their reactivity, packing and electronic properties when adsorbed on metals.^{19,20,26}



Scheme 1. Chemical structure of molecules **1**, **2** and **3**.

These dipolar molecules are therefore particularly suited to be interfaced with a graphene surface via electrostatic interactions. The specific choice of the molecular design, which takes full advantage of our developed knowledge, is tailored in order to meet the following characteristics:

- **1** is our best documented member of the family, which has been previously shown to assemble into compact nm-thick films over metallic surfaces.^{19,20,26}
- **2** showed better affinity for Au than SiO₂, as revealed by XPS mapping over patterned Au on SiO₂,²⁰ and is therefore expected to have small interactions with the

SiO₂ substrate.

- **3** is expected to interact with SiO₂ substrate, forming a QZW layer featuring a greater stability than **1** or **2**.

The molecular films have been investigated by combining X-ray photoelectron spectroscopy (XPS) and time of flight secondary ion mass spectroscopy (ToF-SIMS) analyses. While the former offers insight into the chemical composition of the adsorbed molecules, the latter provides information on the thickness, quality and integrity of ultra-thin molecular films supported on SiO₂ substrates. The electrical characteristics of graphene FET devices built over thin molecular films adsorbed onto such dielectric substrates are exploited as internal gauge to evaluate the stability of such molecular layer, thus using graphene as 'detector' of the underlying film.

Experimental

Devices construction

General procedure for the synthesis of the zwitterions.

Commercial 4,6-diaminoresorcinol dihydrochloride and functional amines were used directly without further purification. Solvents were freshly distilled under argon prior to being used. ¹H NMR spectra were recorded in CDCl₃ on a Bruker 500 MHz instrument, operating at 125 MHz for ¹³C spectra and 500 MHz for ¹H spectra. Chemical shifts are given in δ units, in parts per million (ppm) relative to the singlet at $\delta = 7.26$ for CHCl₃. The splittings were designated as s, singlet; d, doublet; t, triplet; m, multiplet; br, broad. Zwitterions **1** and **2** were readily prepared following published procedures.^{27, 20}

The synthesis of (6Z)-4-(3-(methoxydimethylsilyl)propan-1-amino)-6-(3-(methoxydimethylsilyl)propan-1-iminio)-3-oxocyclohexa-1,4-dien-1-olate (**3**) was performed from a suspension of the parent zwitterion **1** (0.100 g, 0.72 mmol) in freshly distilled methanol (10 mL) to which 3-(ethoxydimethylsilyl)propylamine (0.290 mL, 1.52 mmol) was added. The solution was heated under argon at 100 °C for 3 h (preheated oil bath). After cooling to room temperature, the reaction mixture was filtered and the solid collected was extracted with ethanol. The solution was concentrated under reduced pressure and addition of diethyl ether precipitated the zwitterion as a violet-pink solid which was washed successively with ether and pentane (0.147 g, 51%). ¹H NMR (500 MHz, CDCl₃): δ 0.12 (s, 12H, Si-CH₃), 0.65 (m, 4H, Si-CH₂), 1.77 (m, 4H, N-CH₂-CH₂), 3.36 (m, 4H, N-CH₂), 3.41 (s, 6H, Si-OCH₃), 5.14 (s, 1H, N-CH-CH), 5.43 (s, 1H, O-CH-CH), 8.28 (br s, 2H, NH). ¹³C{¹H} NMR (125 MHz, CDCl₃): δ -2.68 (s, Si-CH₃), 13.40 (s, Si-CH₂), 22.42 (s, N-CH₂-CH₂), 45.92 (s, N-CH₂), 50.47 (s, Si-OCH₃), 80.68 (s, N-CH-CH), 98.92 (s, O-CH-CH), 156.67 (s, N-CH), 172.43 (s, O-CH).

Graphene fabrication. Graphene was grown using low-pressure chemical vapor deposition (LPCVD) methods on a Cu foil (Alfa Aesar, Item No. 46986, 99.8%, cut into 6 cm strips) in a hot wall furnace consisting of a 4 in. fused silica tube. Prior to CVD, the foils were cleaned using a Ni etchant (Transene, TFB) for 5 min and then

rinsed with DI water 3 times. A typical growth process flow is as follows: (1) load the Cu foil, evacuate, heat to 1000 °C, and annealing for 20 min under a 100 sccm H₂ flow (at a pressure of about 70–80 mTorr); (2) introduction of 30 sccm CH₄ and 30 sccm H₂ for 40 min (60 mTorr); (3) cooling down the furnace to room temperature in vacuum, after exposure to CH₄ for 40 min. A PMMA solution (950 K, 4% by volume dissolved in chlorobenzene) was spin-coated onto the top of the sample at 2000 rpm for 30 s. The PMMA film (thickness of 200 nm) was then dried at 30 °C overnight. The Cu under the graphene film was etched using a copper etchant solution and washed with deionized water 3 times. The resulting PMMA/graphene layer was collected and transferred on a SiO₂ substrate functionalized with QZW molecules. Finally, the PMMA film was dissolved using acetone. Graphene etching is achieved by Ar ion bombardments that are accelerated at 400 V from the initial gas introduced with a flow rate of 3 sccm at a pressure of 10⁻⁴ mbar. In contrast, Ar ion etching, being more directed and less diffusive, resulted into functional patterned devices.

Molecular films growth. 10 mM solutions of dichloromethane solution of molecules **1–3** were processed at surfaces following two methods. In *method 1*, the SiO₂/Si substrate was rinsed with dichloromethane. The molecular film was prepared by spin-coating: a drop the QZW solution was applied onto the basal plane of the surface, then the sample was spun at 2000 rpm for 30 s. This process was repeated 3 times. In *method 2*, the SiO₂ substrate was cleaned with dichloromethane and immersed in the QZW solution for about 18 h in a sealed container. When the substrate was taken out, dichloromethane quickly evaporated and an uneven layer of zwitterion molecules was observed by optical microscopy. The functionalized substrate was then rinsed with a highly diluted QZW solution (conc = 0.5~1 mM) in dichloromethane to remove the excess zwitterion molecules, and then washed by sample spinning at 500 rpm for 10 s followed by 3000 rpm 30 s. This process was repeated twice. XPS analysis of the films of molecules **1, 2** prepared using the two different methods gave similar results. Atomic force microscopy imaging revealed covered substrates of limited roughness. More specifically, while the pristine SiO₂ substrates revealed a root-mean square roughness (R_{RMS}) of 0.15 nm as determined on areas of 2 x 2 μm^2 , upon absorption of the molecular layer the R_{RMS} was found being below 1 nm for molecules **1,2** and below 2 nm for molecules **3**.

Graphene transistors fabrication. The chosen graphene transistors architecture comprises a bottom-gate, based on Si-n⁺⁺ substrate covered by a 300 nm thick thermally grown SiO₂ layer, and top source/drain Au electrodes, 30 nm thick, evaporated through a shadow mask yielding channels lengths of several hundreds of μm and channel widths of 40 μm (Figure 1 a,b). For the purpose of investigating how the buried molecular underlayer impacts graphene transistor properties, it is of paramount importance to avoid heating and solvent processing when transferring the graphene layer. We therefore employed a recently

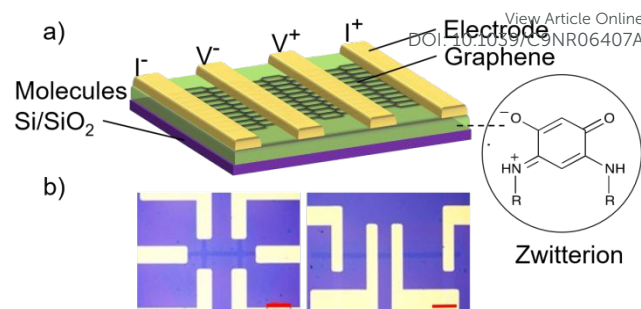


Figure 1. Design of the samples a): Schematics of the graphene-over-molecules device; b): Optical micrograph of typical devices investigated (scale bar is 200 μm long).

developed solvent-free approach.²⁸ To cross-check the quality of the transferred CVD graphene, we used Raman spectroscopy which revealed that our transistors comprise uniform graphene (mostly bilayer) and exhibit good electrical properties (mobility > 1000 $\text{cm}^2\text{V}^{-1}\text{s}^{-1}$, clean quantum hall plateau at low temperatures).²⁸

Surface spectroscopy characterization. In order to reveal and quantify the presence of thin molecular films adsorbed onto SiO₂, two complementary methods were used. **XPS Measurements** were performed by using the ThermoScientific K-Alpha XPS system (Al K α 1486.6 eV). All data were calibrated to a C-C bond energy of 284.9 eV. The beam spot size was focused to 400 μm^2 at the take-off angle of 90°. A pass energy of 20 eV was implemented and energy step size of 0.05 eV was used. The acquisition time was set to 200 ms and 10 scans of data were recorded, with measurements performed on two regions of the sample. All measurements were conducted at pressures < 6×10⁻⁸ Torr. XPS spectra were recorded on samples prior and after the deposition of the graphene films. **ToF-SIMS measurements** were performed with an instrument from IONTOF GmbH, Germany. Mass spectra were acquired using 25 keV Bi₃⁺ primary ions in high mass resolution mode (~5000) under the static limit (< 10¹² ions/cm²). Positive secondary ions in the mass range 1–800u were recorded on randomly selected areas of 200 x 200 μm^2 . To ensure reproducibility of the results, two measurements were performed on each sample and compared. Cross-check measurements with same operating parameters were carried out on pure bulk molecules (compressed powder) and on bare silicon substrates.

Results and discussion

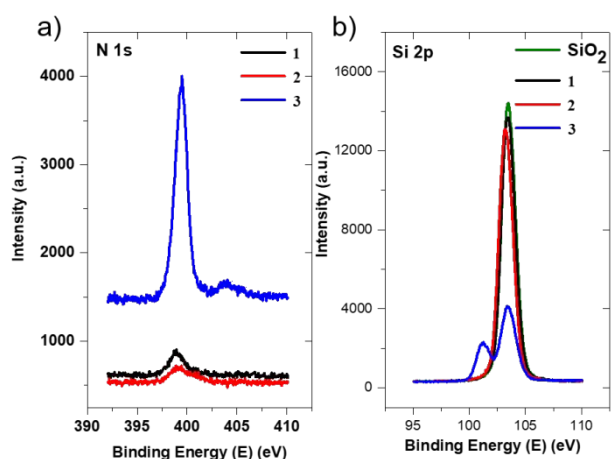


Figure 2. a): XPS N1s spectra of molecules **1**, **2**, **3** (black, red and blue), respectively over SiO₂ substrates. Data shown after the graphene over-layer transfer are identical to those prior to addition of graphene; b): XPS Si 2p spectra of molecules **1**, **2**, **3**, respectively over SiO₂ with the same color notation as left figure. The data in green color is taken on bare SiO₂ substrate.

We first took great care to ensure and quantify the presence of compact molecular thin films on the dielectric substrate. Figure 2 shows XPS spectra measured from all three molecules deposited on SiO₂ substrate. In contrast to bare SiO₂ and graphene/SiO₂ substrates exempt of detectable XPS signal of N, we observed that deposition of **1-3** on SiO₂ resulted in detectable N1s peak, which cannot be attributed to the solvent (Figure 2a). Successive rinsing with ethanol (up to 5 times with 10 s immersion in solvent) did not noticeably modify the XPS N peak, providing unambiguous evidence for the stability of the molecular films. Importantly, no noticeable difference in the N1s spectra were found on sample prior or after graphene deposition.

The Si2p background peak was also recorded for **1-3** on SiO₂ and compared to the data on as-prepared initial SiO₂ substrate (Figure 2b). As expected, the Si2p peak at 103.5 eV decreases when substrates are coated with the QZW films of molecules **1** and **2**, with a further decrease for **3**, also showing additional spectroscopic features, corresponding to the energy of bulk Si atoms, likely to result from those of molecule **3**.

From the literature XPS absorption of self-assembled monolayer, revealing an extinction length of 4 nm,²⁹ we estimate that molecules **1,2** form layers thinner than 1 nm, while molecules **3** make thicker films, around 4 nm in thickness.

Further insight into the presence of the molecules of interest on the support was gained by the use of ToF-SIMS that offers both elemental and molecular information of surfaces with ppm sensitivity. Hence, it enables to detect full molecular species of molecules **1**, **2** and **3** on SiO₂, demonstrating their presence and integrity on surfaces. Figure 3 (top) shows part of the positive spectrum recorded on a molecular film of **3**. Four characteristic peaks can be observed in the mass region of the full molecule. The

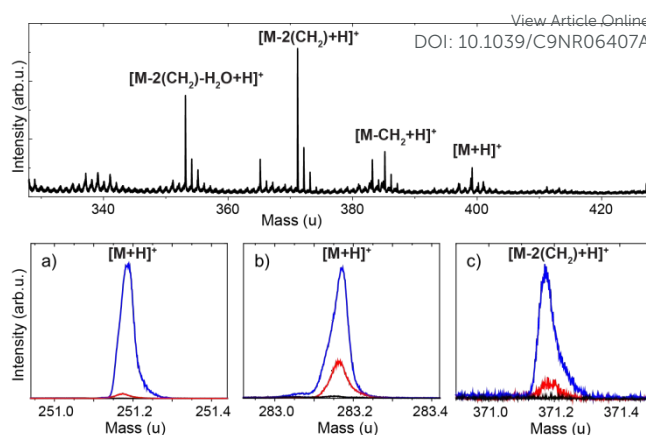


Figure 3. Top: ToF-SIMS positive mass spectrum of molecule **3** adsorbed on SiO₂, in the region of molecule **3** atomic mass ([M]). Target molecule with and without its methyl terminations and byproduct are observed. Bottom: ToF-SIMS positive mass spectra of molecules **1** a), **2** b) and **3** without the methyl terminations c), adsorbed on SiO₂ (red), as compared to the references spectra of bare SiO₂ (black) and bulk respective molecules (dashed blue).

far-left peak, at mass 399u, corresponds to the full molecule plus one hydrogen ([M+H]⁺). This confirms the presence of the molecule **3** on the substrate. Additionally, one can see another peak at mass 385u and a dominant peak at mass 371u. These correspond to the molecule **3** after the loss of one and both methyl termination(s) respectively. Remarkably, this confirms that the molecule **3** spontaneously releases the methyl groups to form Si-O-Si bonds with the substrate. Further, the relative intensities of these three peaks demonstrate that the latter binding scheme is the dominant one.

The bottom part of Figure 3 displays zoom-ins on the main molecular ion ([M+H]⁺) of the molecules **1** (a), **2** (b) and **3** (c). The results for SAM samples (red traces), bulk molecules (dashed blue traces) and substrate reference (black traces) are presented and compared. In all three cases, the peak is found to be characteristic (large intensity from the bulk molecule and no intensity from the substrate reference) and demonstrates the presence and integrity of the molecule on the SiO₂ substrate. Note that due to different ionization potential of different molecular species and the large dependence of matrix effect, the respective intensities cannot be compared nor estimated from the bulk situation.

Overall, the combination of XPS and ToF-SIMS confirms the presence of uniform films with thicknesses below 5 nm, covering the surface of the SiO₂ substrate for molecules **1-3**, with good confidence that the molecules remain intact after deposition, and a device construction methodology that ensures that the graphene "lid" on the top did not perturb significantly these assemblies.

Top-contacts bottom-gate graphene transistors supported on SiO₂ showed strongly p-doped characteristics, with significant hysteresis in the transfer curves, even though the gate voltage was slowly swept (90 mV/s, Figure 4). Hysteresis and

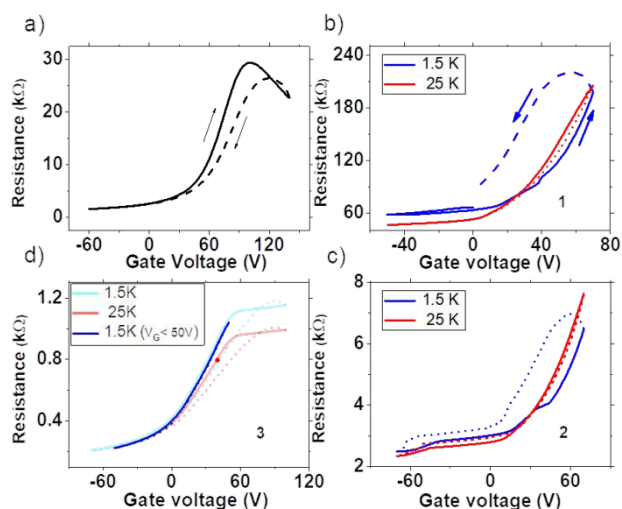


Figure 4. Transistor transfer curves of CVD graphene transferred on SiO₂ devices. a): sample transferred over bare SiO₂, measured at room temperature, nearly temperature-invariant; b) c) d): devices with substrates covered by QZW molecular layers made of molecules **1**, **2**, **3** prior to graphene transfer. Hysteresis of the transfer curve is roughly temperature-independent and small for bare SiO₂ and requires a larger gating bias to appear. Very large hysteresis occurs at lower temperatures for molecules **1**, **2**, while absent at higher temperatures under moderate gate bias. In contrast, hysteresis similar to bare SiO₂ case occurs for molecule **3**.

neutral point shift can be reduced by vacuum annealing treatment (down to 10⁻⁴ mbar, with the samples kept at 200 °C), a procedure previously shown to enhance significantly the quality of the transistors.²⁸ Here, we avoided this procedure by preventing heating in order to guarantee the integrity of the adsorbed molecules. We previously showed that CVD graphene devices of high quality can be obtained by proper substrate and post-treatment procedures,²⁸ and we traded off here the FET properties for better confidence in the chemical integrity and reproducibility of our devices. The observed deviations from ideal behavior relate to trapped molecules (solvent, oxygen, moisture) and electric-field-induced defect in the dielectrics that shift the local density of states of the graphene layer. The so-called *positive hysteresis* properties of the transfer curves, that shift the neutrality point to more positive gate voltage when sweeping back from large positive bias, persist in vacuum down to 1.5 K on samples with graphene transferred on bare SiO₂ substrates (Figure 4a). This observed hysteresis is commonly attributed to charge trapping in the substrate that imprints local changes in the graphene density of states.^{30,31} For applied gate voltage below 70 V, no hysteresis was observed in the temperature range 300 - 1.5 K for the long measurement time scale used.

The capacitive coupling between graphene and nearby charges yields a *negative hysteresis* behavior: when sweeping back from high positive voltages, the neutrality point is displaced to more negative voltage values.³⁰ Such negative hysteresis is usually

explained as resulting from (partial) freezing of the alignment of dipolar molecules in the immediate neighborhood of the active graphene layer under the electric field created by the gate voltage. This is for example observed for ion-gated graphene devices.³² Dipolar molecular adsorbates, in particular oxygen and moisture,

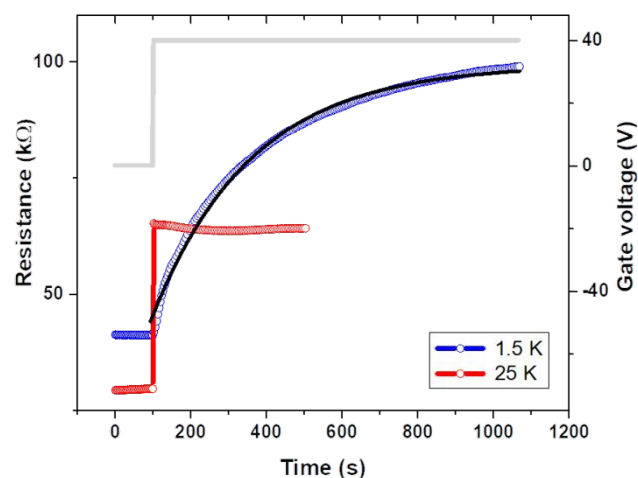


Figure 5. Time evolution of the graphene layer resistance over molecules **1**, after a 0 - 40 V voltage step. The resistance at 1.5 K varies exponentially, with a fitted time constant of 264 s.

can also induce negative hysteresis. This is well-known with interstitial water,³³ the hysteresis strongly diminishing below 0 °C, where the electric susceptibility of frozen water becomes negligible.

In our experiments, transistors comprising inter-layers of molecules **1** and **2** display the appearance of negative hysteresis (Figure 4 b, c). However, this occurs only if the temperature is low enough (a few K), under rather low gate voltage bias (several tens of V). Such hysteresis becomes negligible for temperatures greater than 10 K when a slow gate voltage sweep rate was used. A slow relaxation of the resistance correlates with the observed hysteresis (Figure 5). An exponential slow relaxation of the sample resistance is revealed in the lowest temperature range, with a time constant of typically 10²-10⁴ s (depending on the temperature and the history of the sample), while absent at higher temperature, where the hysteresis disappears.

One must emphasize that the temperature trend of the negative hysteresis is opposite to the one previously reported in literature.^{32,33} Clearly, the transition temperatures cannot correspond to freezing or a phase transition of (unlikely) solvent residues. Furthermore, negative hysteresis does not occur for the test molecule **3**, which was designed to strongly bind to the SiO₂ substrate. Indeed, transistors including molecules **3** exhibit a small positive hysteresis under larger voltage stress (Figure 4d), which resembles the behavior found on devices without molecules, even though the presence of molecules **3** on the substrate is validated by ToF-SIMS and XPS. In essence, low temperatures negative hysteresis unambiguously relates to dipolar QZW molecules weakly bound to the substrate.

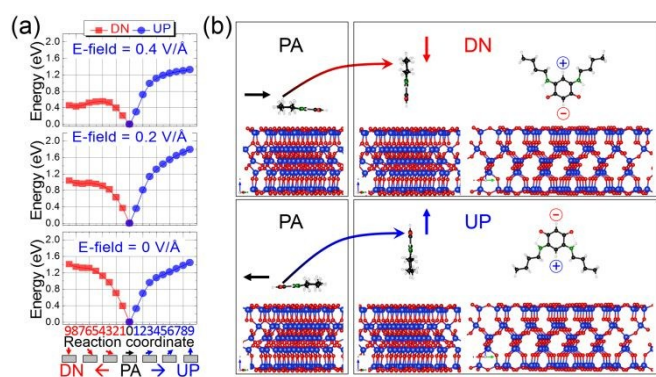


Figure 6. a): Nudged elastic band energy profiles; b): atomic geometries of the rotations of QZW molecule 1 on SiO_2 , under given applied electric field of 0, 0.2 and 0.4 $\text{V}/\text{\AA}$, for the paths from parallel configuration (PA) to 1st perpendicular configuration, where the molecule dipole points down (DN), and from PA to the other perpendicular configuration with dipole pointing up (UP). The red, blue, black, green, and white balls represent O, Si, C, N, and H atoms, respectively.

Quantitative insight into the energetic of QZW molecules (molecule 1) on SiO_2 , under applied electric field, was gained by calculations using the Nudged Elastic Band method (NEB).³⁴ We simplified our model system to the maximum level, to extract the key points explaining the experimental findings. The atomic geometries of all the model structures were calculated by using the Vienna ab initio simulation package.^{35,36} The exchange–correlation functional was approximated using the Perdew–Burke–Ernzerhof expression.³⁷ In particular, the optB86b-vdW functional, implemented in VASP by Klimeš *et al.* to account for weak van der Waals (vdW) interactions, was used to calculate the optimized structures.³⁸ Electron–ion interactions were modelled using the projector-augmented wave method.³⁹

The electronic wave functions were expanded in a basis set of plane waves with a kinetic energy cut-off of 400 eV.

To compute the initial and final structures needed to perform the NEB calculation, we prepared the reconstructed (001) alpha-quartz surface as the SiO_2 surface structure.⁴⁰ Geometry relaxation step in NEB calculation was repeated until the ionic forces were reduced to values below 0.1 eV/ $\text{\AA}$. The k-space Brillouin-zone integrations were carried out using only the gamma-point. For all the SiO_2 surfaces, the middle SiO_2 monolayer was fixed during structure optimization. In all models studied using this calculation, at least 20 $\text{\AA}$ -thick vacuum regions along the perpendicular direction (z) to the two-dimensional slab were included to minimize interactions between neighbouring image cells.

Figure 6 shows the energy profiles (a) and atomic geometries (b) upon rotating the QZW molecule on SiO_2 from a parallel (face-on) configuration ('PA'), with molecule lying flat on the substrate, to an orientation normal to the substrate with its dipole pointing upwards (configuration 'UP'), or pointing downwards (configuration 'DN'). Non-interacting QZW molecules on SiO_2 prefer to lay flat on the substrate, in agreement with previous

scanning probe imaging studies performed on low coverage films.^{41,21} Yet, intramolecular interactions in tightly packed layers formed by solution processing can exhibit a different behavior,²⁰ and it is clear that the adsorption potential energy minimum at low angle values (between molecule and substrate) cannot be applied to our experimental situation. Data of Figure 6 results from energy calculations under an applied electric field of 0, 0.2, and 0.4 $\text{V}/\text{\AA}$ respectively, for positive gate bias, estimated to correspond to the largest applied fields in the experiments shown in Figure 5. To maximize the electrostatic interactions, the negative pole of the

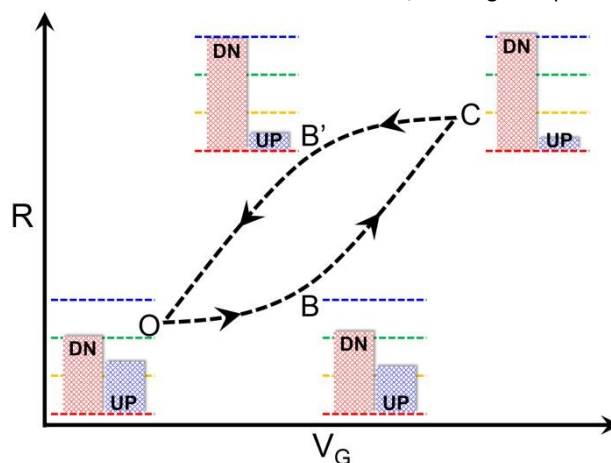


Figure 7. Evolution of dipole population due to the rotation of the QZW molecular on SiO_2 under applied electric field. Sweeping the gate voltage relates to the sequence $\text{O} \rightarrow \text{B} \rightarrow \text{C} \rightarrow \text{B}'$. In state **O**, the energy profiles are symmetrical (in Fig. 6) for the two directions of molecules rotations, under the hypothesis of a zero electric field. When increasing the electric field value, the **DN** orientation is energetically preferred (state **B**). Upon further increase of the gate voltage makes the **DN** orientation at nearly -90° a (meta) stable local minimum of energy. The related reverse energy barrier makes the high angle dressed molecules freeze when sweeping back the gate voltage (state **B'**). Estimates of the populations of molecules in **DN** and **UP** configurations are included in the figure. This makes the populations of **B** different from those of **B'**, the latter having a memory of the populations reached in **C**.

molecules prefers to point towards to positively biased substrate, explaining the lower energy values when the molecules point downwards.

We propose here that the observed negative hysteresis is due to a difference of the constrained orientation (or 'dressing') of the molecules under electric field. At negligible applied gate voltage (state **O** in Fig. 7), the energy configurations for the two rotations $\text{PA} \rightarrow \text{DN}$ and $\text{PA} \rightarrow \text{UP}$ are degenerate. When increasing the gate voltage (state **B** in Fig. 7), $\text{PA} \rightarrow \text{DN}$ is energetically more favorable, and a tendency to dress the molecules downwards emerges. At higher gate bias (state **C** in Fig. 7), a stable energy minimum develops for a large angle of the molecules, which can be considered energetically stable in the **DN** configuration, with an

energy barrier ΔE limiting their probability to escape. This stable statistical occurrence of a population of dipoles pointing down can persist when sweeping back the gate bias voltage, and makes the statistical distribution of the molecules orientations in the state **B'** differ from the previous state **B**. The energy barrier ΔE therefore creates a difference of the conductance state, between **B** and **B'**, related to the memory of the dressing in the **DN** configuration that occurred at the highest gate voltage applied.

Since the energy barrier ΔE is small, being estimated in our calculations as low as ~ 0.1 eV, it is only at the lowest temperature range that the 'reverse' transition path **DN** $\rightarrow$ **PA** becomes thermally impeded, and the related hysteresis develops. At 25 K, this property disappears, being nullified by the thermally driven motion of the molecules. One can even distinguish a slight overshoot of the resistance value, correlated with the small positive hysteresis observed at 25 K in Fig 5. Such thermodynamic 'viscosity' is well known in the field of (nano)magnetism, where a complicated multi-minima energy landscape describes equilibrium magnetic configurations.⁴² Here, we simplified our model to emphasize the two minima energy landscape that explains our findings.

Conclusions

In summary, our study shows how the electrical read out in graphene-based transistors is a unique tool to gain detailed insight into insulating molecular films deposited over a dielectric substrate. We illustrate this concept on ultrathin films of quinonoid zwitterion molecules **1-3** intercalated between the SiO₂ dielectric substrates and the graphene active layer. The transfer curves recorded at the lowest temperatures (< 25 K) reveal that the dipolar alignment of the molecules under gating electric field can be stabilized, for molecules **1,2** making ultra-thin layers with no indications of strong interactions with the substrate.

Temperature represents also a powerful remote control to modify the electrostatic effect of the molecular film on the device operation: when the temperature exceeds a few tens of K the thermal motions smears out the role played by the presence of the molecules on the transistor device output. However, low temperatures make possible molecular ordering stabilization under electric field, revealed by the graphene 'detector'. Here, the emergence of hysteresis in the transistor transfer curve indicates that such dipolar films made of quinonoid zwitterions can exhibit hysteresis in their polarizability, in analogy to ferroelectrics. Deeper insight into the energetics of the system is needed to clarify if this is a single molecule property or a collective effect, as well as better quantify how graphene plays an active stabilizing role.

Our data details unprecedented temperature dependence properties of graphene transistors; it therefore contributes to better pinpoint the origin of hysteresis in graphene FETs, being a key technological issue requiring better clarifications in the community. It also demonstrates that graphene can be

used as an electrical 'detector' of molecular thermal motions, with a potential for dynamics insight in a very wide frequency range. Finally, the construction of molecular layers-graphene hybrid systems made it possible to monitor the emergence of novel properties of molecular systems, which are highly relevant for applications in the field of information storage using multi-functional or multiferroic materials.

Conflicts of interest

There are no conflicts to declare.

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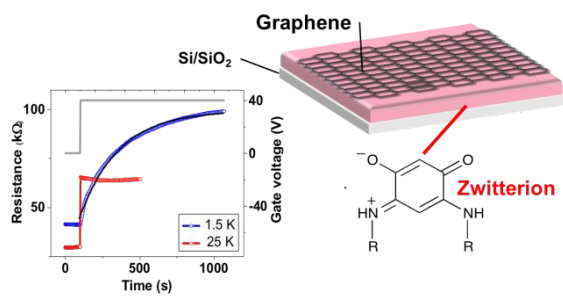
Contributions

J. L., C.Y. and B.D. conceived the experiment, performed by C.Y. and A. M. with S.J. handling the calculations and modeling. L.R., A.G. and P.B. synthesized the molecules, H. C. helped for the CVD and J.D. for the transport measurements. L.B., T.V. and P.S. performed the surface spectroscopy studies. All authors contributed to the manuscript writing.

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A graphene transistor can reveal ordering of dipolar molecules forming nm-thick underlayer, stabilized under electric field at low temperatures.