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Near Unity Absorption in Nanocrystal Based Short Wave Infrared Photodetectors using Guided Mode Resonators

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Abstract: Nanocrystals appear as versatile building blocks for the design of low-cost optoelectronic devices. The design of infrared sensors based on nanocrystals is currently facing a key limitation: the short carrier diffusion length resulting from hopping transport makes that only a limited part of the incident light is absorbed. In order to enhance the device absorption, we use Guided Mode Resonance (GMR). The method appears to be quite versatile and is applied to both PbS and HgTe nanocrystals presenting respectively cut-off wavelengths at 1.7 and 2.6 μm . The designed electrodes present a large enhancement of the material responsivity around a factor of ≈ 250 , reaching external quantum efficiency of 86% for PbS and 340% for HgTe. This increase of the response can be deconvoluted in a factor of 3 for the enhancement of the absorption and a factor of 80 for the photocurrent gain. The method can also be suited to finely tune the cut-off wavelength of the material thanks to geometrical parameters at the device level. The obtained devices are now only limited by the material noise.

Keywords: nanocrystals, short wave infrared, infrared detection, plasmonic resonance, PbS, HgTe, guided mode resonance.

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Colloidal Quantum Dots (CQDs) have raised a significant interest thanks to their widely tunable optical features from the UV to the THz^{1,2} ranges. Their integration into optoelectronic devices raises another level of challenge because of the combination of optical properties and electronic transport. A key step in this direction relates to the development of ligand exchange procedures which allow to achieve stronger electronic coupling as well as tuning the CQD doping³⁻⁶. Thanks to ligand exchange, the mobility, which is a macroscopic measurement of the local CQD coupling, has been increased from $10^{-6} \text{ cm}^2 \cdot \text{V}^{-1} \cdot \text{s}^{-1}$ for CQDs capped with long ligands to 10^{-3} - $10^{-2} \text{ cm}^2 \cdot \text{V}^{-1} \cdot \text{s}^{-1}$ with short organic ligands and even above one with ionic/inorganic surface chemistry⁷⁻¹¹. However, in spite of these progresses, the carrier diffusion length remains limited typically below 100 nm, which is, at least, one order of magnitude smaller than the absorption depth^{12,13}. As a result, the design of CQD-based photodetectors results from a trade-off between the absorption and the charge conduction. Due to this short diffusion length, the thickness of the CQD-based absorbing layer remains below 500 nm which makes that only 10 to 50% of the incident light is absorbed and the amount of photocarriers that reach the electrodes is even lower. This is a major challenge for the design of CQD-based photodetector that needs to be tackled.

In the recent years, large progresses have been obtained for the design of short-wave infrared detectors^{14,15} using PbS¹⁶⁻²⁰ and HgTe²¹⁻²⁶ CQDs, as a possible low-cost alternative to InGaAs technology. A possible strategy to overcome the low absorption in CQD-based light detectors, due to the limited thickness of the CQD layer, is to concentrate the electromagnetic field compared to its plane wave propagation. This is in particular possible with the introduction of plasmonic resonances²⁷. While this field has been widely investigated for conventional semiconductor solar cells^{28,29} and infrared sensors³⁰⁻³³, far less works have been done on this question for CQD-based devices³⁴⁻⁴³.

In this paper, we focus on the short-wave infrared (SWIR, $\lambda < 1.7 \text{ } \mu\text{m}$) and extended short-wave infrared (E-SWIR, $\lambda < 2.5 \text{ } \mu\text{m}$) ranges using respectively PbS and HgTe CQDs as active material. We propose to implement Guided Mode Resonance (GMR)⁴⁴ as a strategy to achieve absorption close to unity while using thin films (200-300 nm) and to improve photocarrier collection. We demonstrate that not only we obtain good qualitative and quantitative agreement with electromagnetic simulation, but also that GMR is a viable path to boost device performance and tune its spectral response.

Discussion

We start by synthesizing PbS¹² and HgTe⁴⁵ nanocrystals with excitonic feature respectively at 1.55 μm (0.8 eV or 6000 cm^{-1} , later called PbS 6k) and 2.6 μm (≈ 0.5 eV or 4000 cm^{-1} , called HgTe 4k) using established methods from the literature, see spectra in Figure 1a and Transmission Electron Microscopy (TEM) image in Figure 1b and c. Both materials can be deposited onto interdigitated electrodes with 30- μm period to build a conductive device. In this case, in order to build thick films (typically 200 nm), a multi-layer deposition is required. For each layer, we perform a ligand exchange step from long alkyl chains, oleic acid and dodecanethiol for respectively PbS and HgTe, towards ethanedithiol to ensure a strong interparticle coupling. Both materials present a photoresponse as shown by the modulation of the I-V curve under illumination, see Figure 1d and e. For such films, the responsivity has been estimated to be around 1 $\text{mA}\cdot\text{W}^{-1}$ for the two materials, under 1 V of applied bias and for room temperature operation, see Figure S1 and S2. Such low responsivity is the result of (i) a weak absorption $\approx 12\%$ considering the coefficient of absorption $\alpha \approx 3 \times 10^3 \text{ cm}^{-1}$ ^{13,46} for HgTe and a film thickness of 200 nm and (ii) a poor charge collection due to the short diffusion length. These limitations come in addition with the presence of $1/f$ noise^{47–49}, see Figure S1 and S2, to finally lead to low signal over noise ratio (*i.e.* detectivity) around 10^7 Jones.

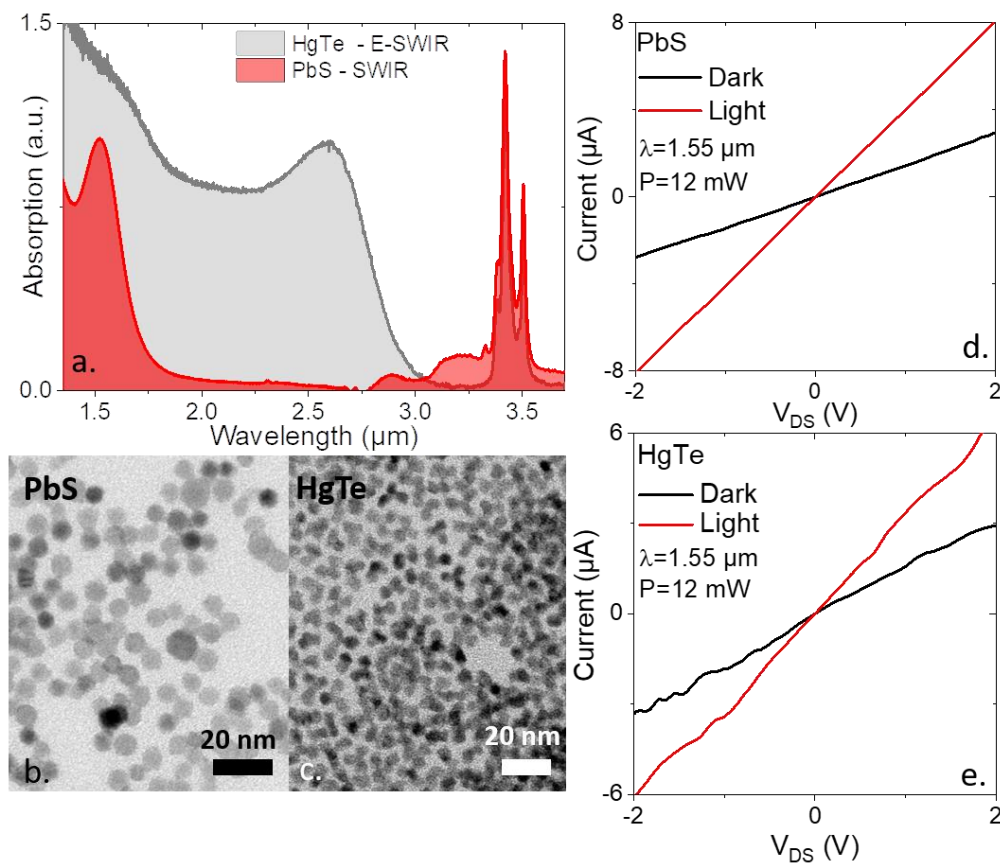


Figure 1 Material characterizations. a. Absorption spectra of PbS and HgTe nanocrystals used for this study. b. (resp. c.) Transmission electron microscopy images

of PbS (resp. HgTe) nanocrystals. d. (resp. e.) I-V curves under dark condition and under illumination ($\lambda=1.55 \mu\text{m} - P=12 \text{ mW}$) for PbS (resp. HgTe) nanocrystal 200-nm thick films measured at room temperature.

To address the difficulties resulting from both weak absorption and poor charge collection, we propose an architecture based on Guided-Mode Resonances (GMR). This structure not only allows the introduction of optical resonances to enhance the light-matter coupling but also relies on the shorter inter-electrodes spacing to reduce the transit time and thus achieve photoconductive gain^{50–52}. Compared to the work done on non-colloidal semiconductors, the main challenge is to design a structure which accounts for the fragility of the colloidal materials. PbS nanocrystals can easily be oxidized, while HgTe nanocrystals sinter upon annealing even at relatively low temperatures⁵³. Our approach is to design a structure in which all lithography steps are conducted before the CQD deposition, see Figure S3 for fabrication details.

The GMR structure consists in a thin film deposited onto a grating pattern. From an electromagnetic point of view, the grating, via diffracted waves, aims to couple incoming light within a guided-mode of the thin film. Depending on the materials and geometries, this coupling gives rise to resonant transmission, reflection or absorption of light, at a particular wavelength λ_R . Such GMR architecture has been widely used for band-pass filtering⁵⁴ or notch filtering⁵⁵, and more rarely for resonant absorption³¹. Its main advantage, particularly compared to Fabry-Perot resonators, stands in the fact that the resonant wavelength is tunable with the period of the grating (*i.e.* through a geometrical factor rather than by tuning a material parameter). Thus, multispectral patterns can be achieved on a sample with a single lithography step.⁵⁶ In our study, grating is made of gold and the waveguide is the CQD film.

Figure 2 provides a schematic of the three structures we studied: (i) the conventional interdigitated electrodes with a 30 μm period (Figure 2 a, d & g), (ii) the same electrode pattern with a back side-mirror (Figure 2b, e & h) and the micro-electrodes used to induce GMR (Figure 2c, f & i). The absorption of the conventional interdigitated electrodes corresponds mainly to one passing of the incident light through the CQD film (Figure 2g). The addition of the gold mirror allows reflection of the light and so, a second passing of the incident light through the film. Note that it is similar to the optical path in vertical geometry devices where metallic back electrode can also reflect the incident light towards the absorbing CQD film (Figure 2h). Last, Figure 2f illustrates light trapping in the CQD film thanks to diffracted orders. For sake of comparison, we calculate the predicted spectrum for the three types of electrodes. These calculations have been achieved with Reticolo® software⁵⁷ based on RCWA (Rigorous Coupled-Wave Analysis). Compared to conventional electrodes, the metallic mirror leads to a 45 % increase of the absorption (Figure 2j and k). This is less than a factor of two due to the already existing partial reflection of the light at the CQD/SiO₂ interface (Figure 2 g). With the GMR electrodes, the absorption reaches 70% (Figure 2l) of the total

incident light, which is 2.9 times more than the absorption obtained for conventional electrodes (Figure 2j and l). We will now explain the resonant mechanisms involved in GMR and the basic design rules.

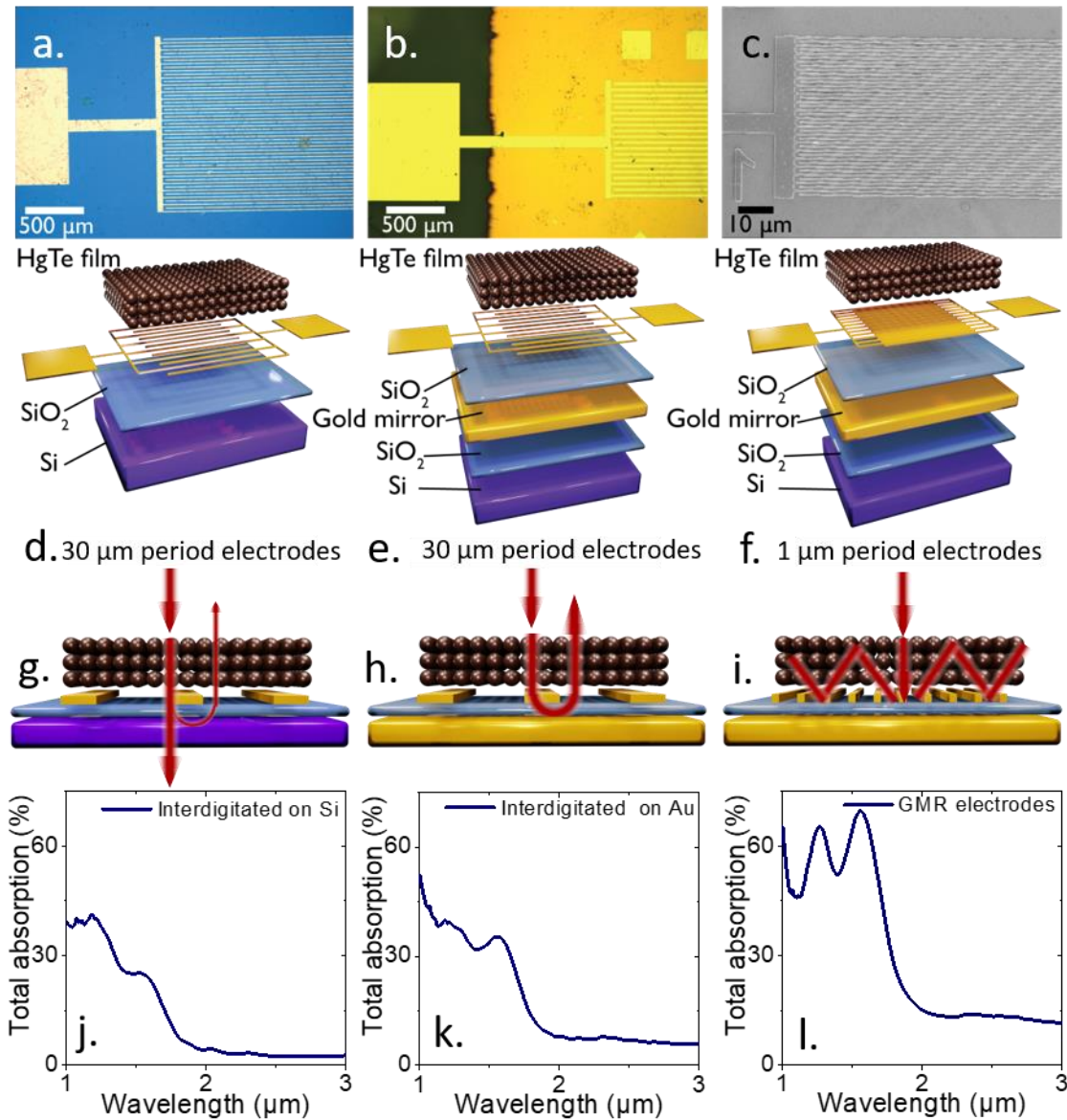


Figure 2: Description of used structures. a. Optical microscopy image of the conventional interdigitated gold electrodes deposited on a Si/SiO₂ substrate. b. Optical microscopy image of the interdigitated electrodes made of gold contact deposited on a Au/SiO₂ substrate. c. Scanning electron microscopy image of the interdigitated gold electrodes on a Au/SiO₂ substrate designed to induce GMR. d-f. Schematics of respectively d. the conventional interdigitated gold electrodes deposited on a Si/SiO₂ substrate, e. the interdigitated electrodes made of gold contact deposited on a Au/SiO₂ substrate and f. the interdigitated gold electrodes on a Au/SiO₂ substrate designed to induce GMR. g-i. Schematics of the cross section of g. the conventional interdigitated gold electrodes deposited on a Si/SiO₂ substrate, h. the interdigitated electrodes made of gold contact deposited on a Au/SiO₂ substrate and i. the interdigitated gold electrodes on an Au/SiO₂ substrate designed to induce GMR. Red arrows correspond

to light paths. *j-l*. Simulated absorption spectra for *j*. the conventional electrodes on Si/SiO₂, *k*. on Au/SiO₂ and for *l*. the electrodes on a Au/SiO₂ substrate designed to induce GMR. For figures *j*, *k* and *l* the incident light is not polarized.

The proposed GMR structure is given in Figure 3b. The latter is designed to achieve a quasi-unity absorption for the TM mode (Transverse Magnetic – magnetic field parallel with the slits of the grating) of a 200-nm thick film of PbS CQDs at $\lambda_R = 1.55 \mu m$. The determination of the geometrical parameters starts with the period p of the grating. It is set so that there is no diffracted orders reflected in air, but there must be diffracted orders in the CQD layer. Thus, the condition $p < \lambda_R < p \cdot n_{CQD}$ should be fulfilled under normal incidence (n_{CQD} is the refractive index of the CQDs). Then, the period value is set in accordance with the CQD layer thickness t_{CQD} , so that resonance occurs at the wavelength of interest. Finally, the thickness of the grating t_g and the width of its openings w are adjusted to maximize resonance. It is worth noting that only the grating is responsible for the resonant induce enhancement of the absorption (Figure S5). The back-side mirror is here to avoid loss of photon resulting from light transmission and induces a broad band enhancement of the light absorption.

Figure 3a shows calculated absorption spectra under normal incidence, for TM polarization (blue curve) and for transverse electric polarization (TE – electric field parallel with the slits – red curve) lights. The average of TM and TE curves is represented in black and corresponds to an unpolarized incident light. For a grating period of 1.0 μm and a CQD film thickness of $t_{CQD} = 200$ nm, a single and almost total absorption peak ($A_{tot}^{TM} = 99.3\%$) occurs for $\lambda_R = 1.55 \mu m$ for TM polarization, due to GMR. For TE polarization, a shallow peak also occurs at 1.55 μm , which is due to PbS absorption and a stronger peak ($A_{tot}^{TE} = 80.4\%$) occurs for $\lambda = 1.25 \mu m$, due to a second GMR.

The operation of the GMR electrodes (Figure 3b) is even better revealed by the map of electromagnetic absorption, see Figure 3c and d. Refractive indices of involved materials and resultant geometrical parameters of the GMR resonator are given in Supplementary information, see Figure S4 and Table S1. The antinodes of the TM mode are mostly located on the top of the gold finger, while for TE mode they are localized between the fingers. This second resonance is quite interesting since it is far away from the metal and makes that 89% of the absorption is spatially located within the CQD (against 77% for the TM mode) The remaining absorption (11%) occurs in the metal and leads to thermal losses and have consequently to be minimized. As a result, absorption in the CQDs are $A_{CQD}^{TE} = A_{tot}^{TE} \times 0.89 = 71\%$ and $A_{CQD}^{TM} = A_{tot}^{TM} \times 0.77 = 77\%$ for TE and TM modes, respectively.

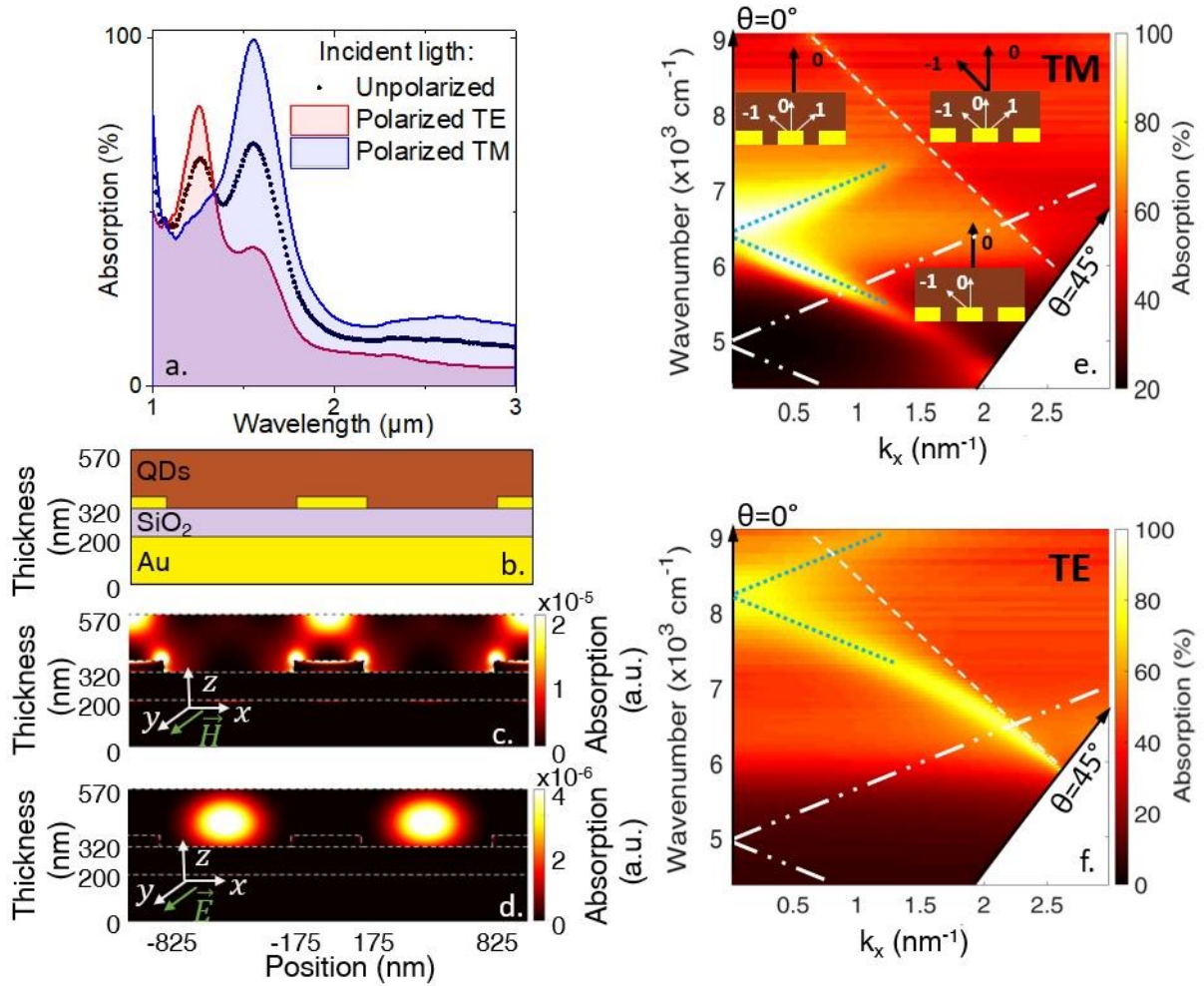


Figure 3 Electromagnetical properties of the GMR architecture. a. Simulated absorption spectra of the interdigitated electrodes on a gold/SiO₂ substrate designed to induce GMRs for an unpolarized (black), TM-polarized (blue) and TE-polarized (red) incident light. b. Cross section of the nanoelectrodes used to induce GMRs. c. Simulated cross section map of the absorption relative to the TM mode at 1.55 μm. 77% of the total absorption occurs in the CQDs. d. Simulated cross section map of the absorption relative to the TE mode at 1.25 μm. 89% of the total absorption occurs in the CQDs. All simulation are conducted using PbS 6k as absorbing material. e. (resp. f.) Angle-resolved specular TM (resp. TE) absorption simulation through interdigitated electrodes on a Au/SiO₂ substrate designed to induce GMRs, in linear color map as a function of $\sigma = 1/\lambda$, the wavenumber, and $k_x = 2\pi \cdot \sin(\theta) / \lambda$, the x component of the incident wave vector. White dashed line corresponds to the appearance/disappearance of one diffracted order in the air. White dot-dashed lines correspond to the appearance/disappearance of one diffracted order in the CQD layer. Blue dotted lines correspond to the absorption peaks. Sketches represent the existing diffracted orders which propagate in the CQD array (brown box) and in air (top of the brown box).

The exact mechanisms of resonances can be deduced from angular dependency and absorption map. Figure 3e and f show calculated absorption diagram, given in linear

color scale. We give on the y-axis the wavenumber and on x-axis k_x , which is the x-component of the incoming wavevector. k_x depends on incidence angle θ and on wavenumber σ : $k_x = 2\pi \cdot \sigma \cdot \sin(\theta)$. First of all, in the diagrams of Figure 3e and f, in white (dash or dash-dot) lines show the limits where first propagative orders are diffracted by the grating (in air or in CQDs, respectively). As expected, we thus see that resonance occurs when ± 1 orders are diffracted in CQD only. The wave vector of the guided mode in the CQD is written as $\beta = n_{eff} \cdot 2\pi/\lambda$, where n_{eff} is the effective index of the mode. For GMR, this mode couples with the first diffracted orders $k_x^{\pm 1}$, and the resonance condition can thus be expressed as $k_x^{\pm 1} = 2\pi \cdot \sigma \cdot \sin(\theta) \pm 2\pi/p \approx \beta = n_{eff} \cdot 2\pi \cdot \sigma$. At $\theta = 0^\circ$, we find $n_{eff} = \lambda/p = 1.55$. This dispersion relation is plotted in Figure 3e for $k_x^{\pm 1}$, and we see the superimposition with the resonant absorption, which demonstrates the GMR mechanism. In this diagram, we also highlight an absorption peak at $\sigma = 6250 \text{ cm}^{-1}$, who has no angular dependency. This absorption is due to intrinsic absorption of PbS at this wavelength (see Figure 1a).

We have tested the robustness of this design toward changes of geometrical parameters. An increase of the film thickness leads to a redshift of the transition, see Figure S6. This nevertheless remains quite limited since a change by 50 nm of the thickness only shifts the resonance peak by its linewidth. We have also estimated how the spectrum is affected by the film surface topology, see Figures S8 and S9. Qualitatively, the conclusion remains unchanged and only a small shift (smaller than the peak linewidth) can be observed. Finally, angular dependence of the structure is discussed in Figure S10.

A comparison between the simulated spectra and the measured photocurrent is given in Figure 4. Several periods close to the optimal values have been designed to balance possible spectral shift resulting from a change of the CQD film thickness. The first striking observation is that the spectral response is drastically affected by a change in the period compared to the one obtained with non GMR electrodes, see Figure S1. We observe a large increase of the absorption around $1.55 \mu\text{m}$, whose exact energy depends on the electrode period. The larger the period, the redder the resonance, see Figure 4a and b. While the conventional interdigitated electrodes display no polarization effect (Figure S1b), the GMR electrodes present a clear polarization effect (Figure 4d), in good agreement with the estimated spectra, see Figure 4c.

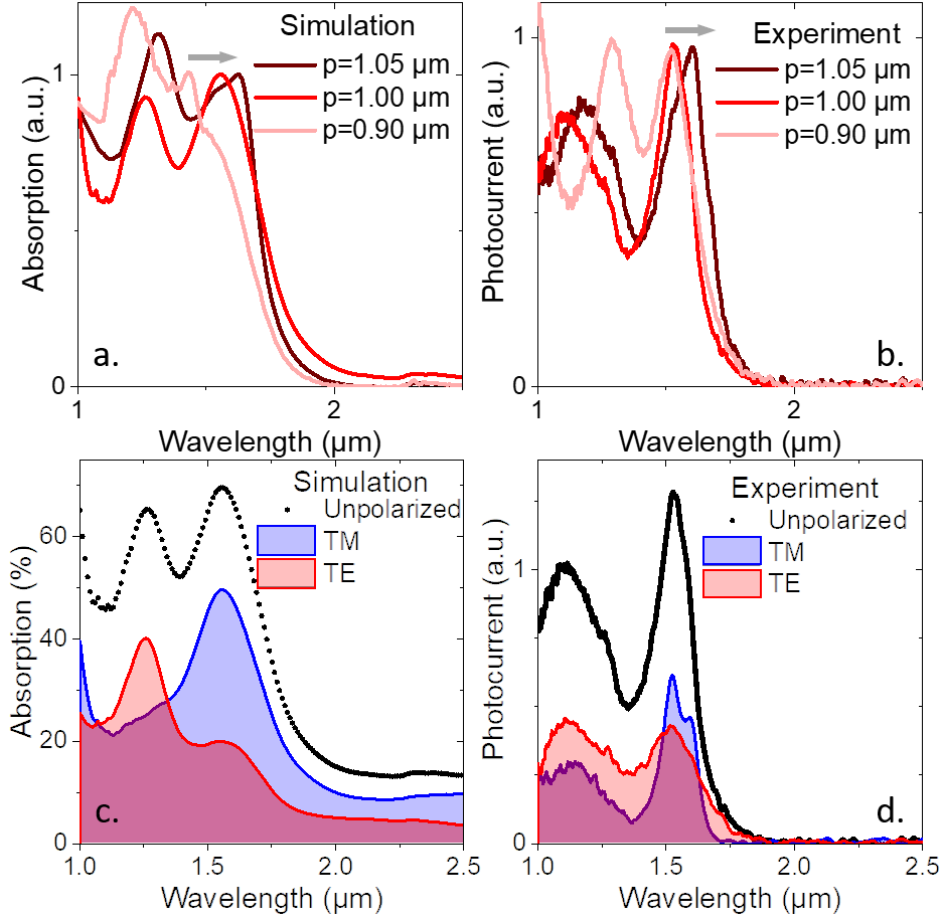


Figure 4 Optical simulations and measurements of GMR structures. a. Simulated absorption and b. experimental photocurrent spectra for PbS 6k nanocrystal-based devices with various periods. c. Simulated absorption and d. experimental photocurrent spectra relative to the total absorption (black dots) and its TM (blue) and TE (red) components for PbS 6k nanocrystal based devices with a 1 μm period.

We have then estimated how the GMR electrodes affect the detection performance of the CQD film. The responsivity of the GMR electrodes reaches $1 \text{ A}\cdot\text{W}^{-1}$ (corresponding to an external quantum efficiency of 86%), see Figure 5a. This responsivity is a factor of 250 higher than the one obtained for the same film (same thickness, same applied bias and same operating temperature) using conventional interdigitated electrodes. This increase of the responsivity is also much larger than the increase of the absorption (≈ 3). This clearly brings evidence that the short spacing of the electrodes leads to an increase of the photoconductive gain by a factor of $250/3 \approx 80$. Note that the time response of the electrodes remains almost unaffected with a 3 dB cut-off frequency around 1 kHz, see Figure 5b. The GMR electrodes also leave the nature of the noise unchanged, being still $1/f$ limited, see Figure 5c. Thanks to the improvement of the responsivity, the detectivity is dramatically enhanced and is close to 10^9 Jones at room temperature and is above 10^{10} Jones at 200 K, see Figure 5d. Regarding the effect of temperature, it is also worth mentioning that we observe a small redshift of the band

edge as the sample is cooled down, see Figure S11, but the energies of the resonance peaks remain unaffected. In this sense the GMR electrodes reduce the effect of temperature on the spectra.

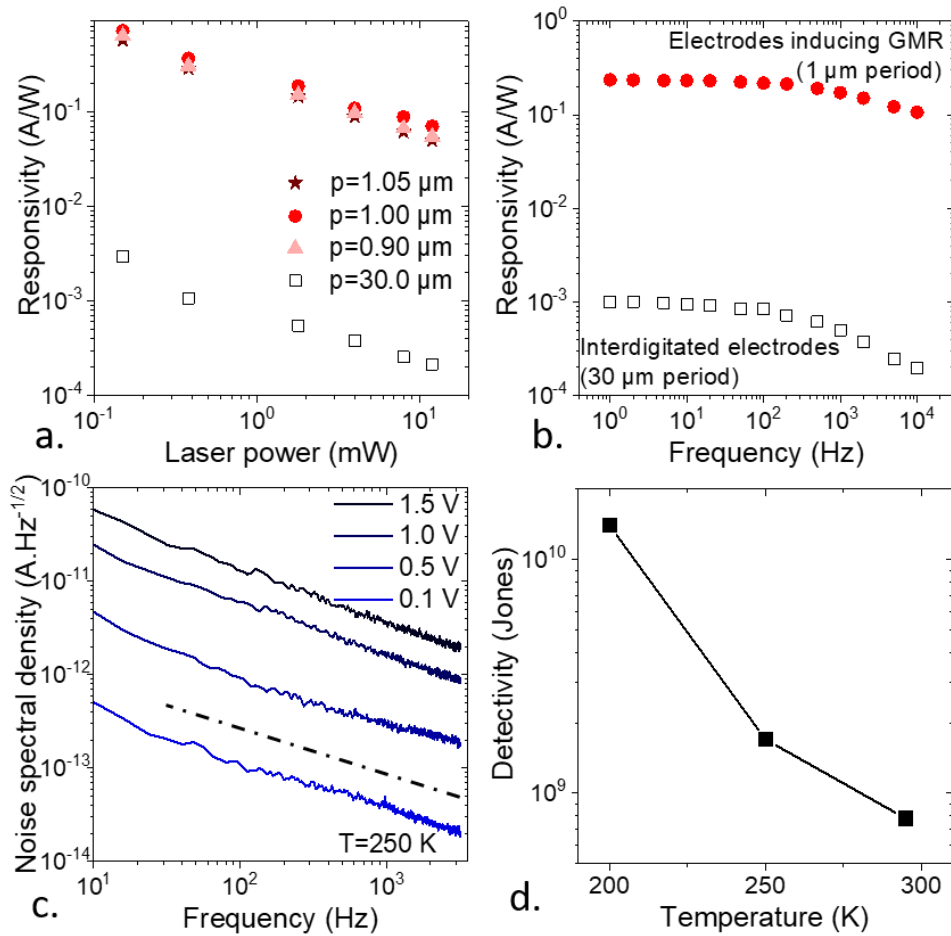


Figure 5 Electrical properties of GMR structures. a. Responsivity as a function of the incident laser beam power ($\lambda=1.55 \mu\text{m}$) for device based on PbS 6k nanocrystals using interdigitated electrodes made of gold contact deposited on a Si/SiO₂ substrate, and interdigitated electrodes on a Au/SiO₂ substrate designed to induce GMR with various period values. In all cases, the applied bias is 1 V. b. Responsivity as a function of the light chopping frequency for device based on PbS 6k nanocrystals using interdigitated electrodes made of gold contact deposited on a Si/SiO₂ substrate, and interdigitated electrodes on a Au/SiO₂ substrate designed to induce GMR with 1 μm period. c. Noise spectral density as a function of signal frequency for various applied biases at 250 K measured on the device with interdigitated electrodes on a Au/SiO₂ substrate designed to induce GMR with 1 μm period. Dash line corresponds to 1/f noise decay. d. Detectivity as a function of temperature measured on the device with interdigitated electrodes on an Au/SiO₂ substrate designed to induce GMR with 1 μm period.

In the last part of the paper we extend the obtained results on PbS in the SWIR range to HgTe nanocrystals in the E-SWIR range (2.6 μm), see Figure 6. Because the exciton

peak is shifted toward longer wavelengths, the period of the GMR is also increased, typically around 1.8 μm . Figure 6a shows a comparison of the absorption spectrum for the three types of electrodes proposed in Figure 2, using this time an HgTe CQD layer. Again, we obtain a very good qualitative and quantitative agreement between the simulated and experimental spectra, see Figure 6c and d.

This time, the enhancement of the absorption is even more dramatic, reaching a factor of ≈ 6 , see Figure 6a. The spectrum of the GMR device becomes driven by the shape of the resonance rather than the shape of the excitonic transition. As a result, the cut-off wavelength can be strongly tuned thanks to the geometrical parameters, see Figure 6e and f. Tuning the period of the GMR electrodes from 1.5 to 1.9 μm shifts the peak cut-off wavelength from 2.15 to 2.5 μm .

As for PbS, the introduction of the GMR electrodes dramatically enhances the responsivity by three orders of magnitude, see Figure 6b. Values above 1 $\text{A}\cdot\text{W}^{-1}$ are also reached for this material corresponding to an external quantum efficiency of 342 %. Such values above 100% are again a clear evidence that gain plays a critical role in the improvement of the device performance.

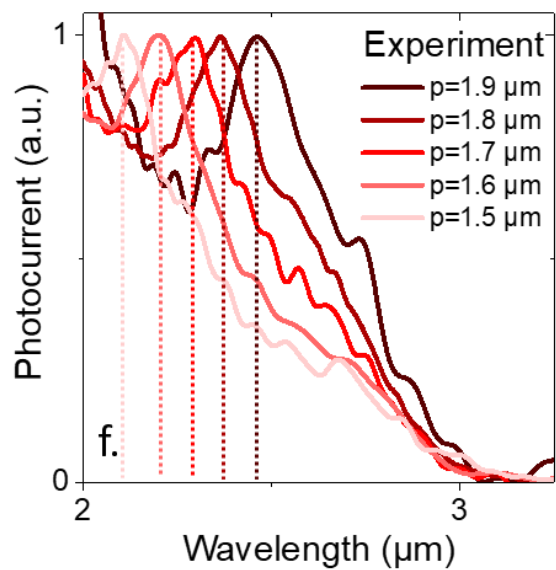
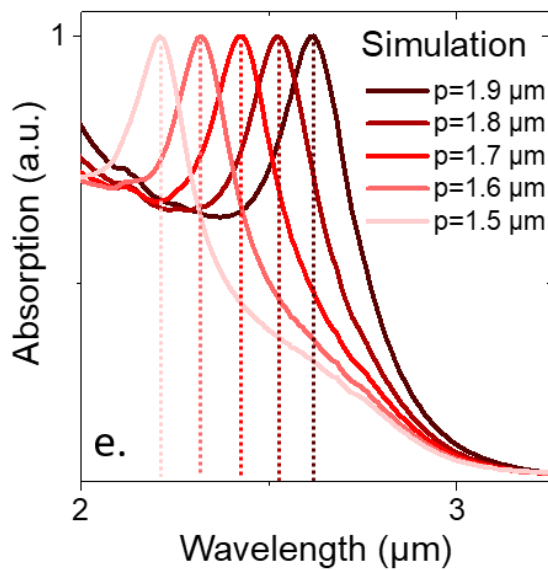
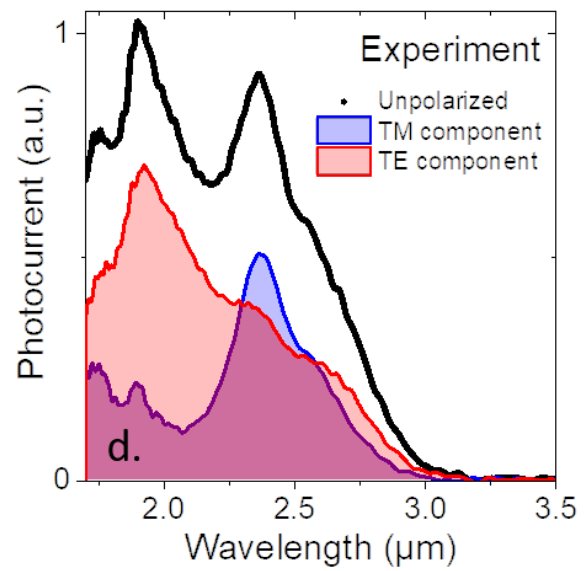
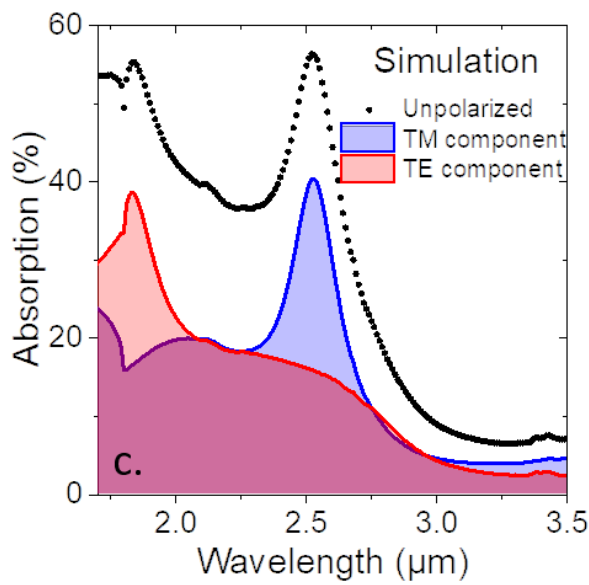
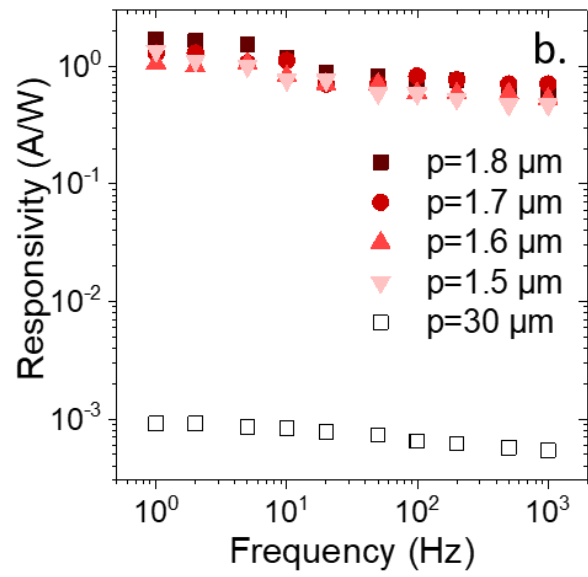
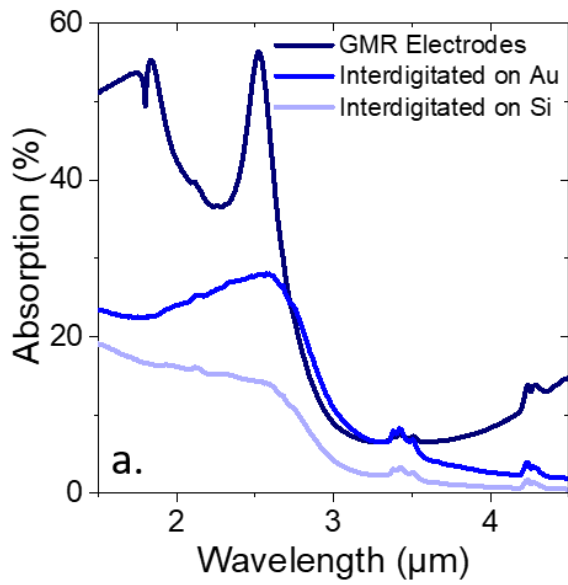


Figure 6: Study of a GMR structure operating at 2.6 μm using HgTe nanocrystals. a. Absorption spectra for device based on HgTe 4k nanocrystals using interdigitated electrodes made of gold contact deposited on a Si/SiO₂ substrate (dark blue), interdigitated electrodes made of gold contact deposited on a Au/SiO₂ substrate (blue) and interdigitated electrodes on a Au/SiO₂ substrate designed to induce GMR (light blue) with a period of 1.8 μm . b. Responsivity as a function of the light signal frequency for device based on HgTe 4k nanocrystals using interdigitated electrodes made of gold contact deposited on a Si/SiO₂ substrate, and interdigitated electrodes on a Au/SiO₂ substrate designed to induce GMR with various period values. The light source is a blackbody at 980 °C. The incident power is calculated to be 200 μW for conventional interdigitated electrodes and 1.7 μW for electrode inducing GMR. c. Simulated absorption and d. experimental photocurrent spectra relative to TE mode (red), TM mode (blue) and unpolarized (black) absorption for HgTe 4k nanocrystal based device with a 1.8 μm period. e. Simulated absorption and f. experimental photocurrent spectra for HgTe 4k nanocrystal based device with various periods.

To conclude, we have proposed a new geometry of electrodes for the design of infrared photodetectors based on CQD film in planar geometry. Thanks to the introduction of GMR, near-unity absorption can be achieved at least along one polarization. At the device scale, this corresponds to an increase of the absorption by a factor of 3 to 6 compared to what can be obtained with similar films deposited on conventional interdigitated electrodes. An excellent agreement between simulation and experiment is obtained thanks to a design of electrodes which takes into account the fragility of the CQDs. The enhancement of the responsivity can be as high as a factor of 1000 for HgTe CQDs. Beyond the increase of the absorption (x 3-6) the electrodes designed for the GMR introduce gain which is typically responsible for a factor of ≈ 100 in the increase of the response. This strategy is quite versatile and can be applied to several colloidal materials especially in the infrared range. The overall performance of the device becomes fully noise-limited.

Methods

Chemicals for nanocrystal synthesis. Mercury chloride (HgCl₂, Strem Chemicals, 99%), tellurium powder (Te, Sigma-Aldrich, 99.99%), trioctylphosphine (TOP, Cytek, 90%), oleylamine (OLA, Acros, 80-90%), dodecanethiol (DDT, Sigma-Aldrich, 98%), lead chloride (PbCl₂, Afla Aesar, 99%), sulfur powder (S, Afla Aesar, 99.5%), oleic acid (OA, Afla Aesar, 90%), 1,2-ethanedithiol (EDT, Fluka, 98.0%), lithium perchlorate (LiClO₄, Sigma-Aldrich, 98%), polyethylene glycol (PEG, M_w = 6 kg.mol⁻¹), chloroform (VWR), ethanol absolute anhydrous (VWR), methanol (Carlo Erba, 99.8%), acetone (VWR), n-hexane (VWR), toluene (Carlo Erba, 99.3%), N,N-dimethylformamide (DMF, Sigma Aldrich), mercaptoalcohol (MpOH, Merck, >99%), Methylisobutylketone (MIBK, VWR, >98.5 %). All chemicals are used as received, except oleylamine which is

centrifuged before use. **Mercury and lead salts are toxic, handle them with great care.**

1 M TOP:Te precursor. In a three-neck flask, 2.54 g of Te powder is mixed in 20 mL of TOP. The flask is kept under vacuum at room temperature for 5 min and then is heated up to 100 °C and kept under vacuum for 20 more minutes. The atmosphere is changed to Ar and the temperature is raised to 275 °C. The solution is stirred until a clear orange coloration is obtained. Then the flask is cooled down and the color switches to yellow. The flask is degassed again for 10 min at room temperature. This solution is finally transferred to an Ar-filled glovebox for storage.

PbS CQD Synthesis. In a three-neck flask, 300 mg of PbCl₂ and 7.5 mL of OLA are degassed, first at room temperature and then at 110 °C for 30 min. Meanwhile 30 mg of S powder is mixed with 7.5 mL of OLA until full dissolution and an orange clear solution is obtained. Then under Ar at 160°C, this solution of S is quickly added to the flask. After 15 minutes, the reaction is quickly quenched by addition of 1 mL of oleic acid and 10 mL of hexane. The nanocrystals are precipitated with ethanol and redispersed in toluene. This washing step is repeated one more time. The solution is then centrifuged to remove the unstable phase. The supernatant is precipitated with methanol and redispersed in toluene. Finally, the PbS CQD solution in toluene is filtered through a 0.2 µm PTFE filter.

HgTe CQD Synthesis. In a 100 mL three-neck flask, 543 mg of HgCl₂ and 60 mL of oleylamine (OLA) are degassed at 110 °C for an hour. The atmosphere is then switched to Ar and the solution is cooled down to 80 °C. A solution containing 2 mL of TOP:Te (1 M) and 10 mL of OLA is quickly added to the flask after warming it up with a heat gun. After 3 minutes, the reaction is quickly quenched by addition of a mixture of 1 mL of DDT and 9 mL of toluene. The nanocrystals are precipitated with ethanol and redispersed in toluene. As a second step of cleaning, the nanocrystals are precipitated with methanol and redispersed in toluene. The pellet is eliminated and the supernatant is washed one more time. Finally, the HgTe CQD solution in toluene is filtered through a 0.2 µm PTFE filter.

Infrared spectroscopy is conducted using a Fisher IS50 Fourier transform Infrared spectrometer. To measure CQD absorption, we use the spectrometer in ATR configuration. A drop of CQD solution is dried on the diamond cell. The source is a white light and the detector is a DTGS ATR. Spectra are typically acquired between 10 000 cm⁻¹ and 2000 cm⁻¹ with a 4 cm⁻¹ resolution and averaging over 32 spectra. To measure a film absorption, an extended InGaAs detector is used and the source is a white light. In case of extended SWIR device, the visible light is switch to a global source combined with a DTGS detector. Each spectrum is averaged 32 times and the resolution is set to 4 cm⁻¹. Photocurrent spectra are acquired as the sample is biased using a Femto DLPCA 200 current amplifier which role is also to magnify the current. The signal is then fed into the FTIR acquisition board.

For TEM pictures, a drop of a diluted CQD solution is drop-casted on a copper grid covered with an amorphous carbon film. JEOL 2010F is used at 200 kV for acquisition of picture. For device characterization, a FEI Magellan scanning electron microscope is used.

For SEM pictures, a Zeiss Supra 40 scanning electron microscope is used. The acceleration bias is set a 10 kV and the aperture at 20 μm .

DC Transport. The sample is connected to a Keithley 2634b, which controls the drain bias (V_{DS}) and measures the associated current (I_{DS}). This measure is carried out in the dark or under illumination using 1.55 μm laser diode.

Responsivity measurement. For PbS 6k, the source is a laser diode at 1.55 μm placed at 20 cm of the sample. The total spot power is measured to be 0.15 mW. The spot size is 1.6 mm^2 , corresponding to irradiance of 9.4 $\text{mW}\cdot\text{cm}^{-2}$. The flux is chopped from 1 Hz to 10 kHz. The photocurrent is measured using Zurich Instruments MFLI lock-in amplifier at 1 V bias. For HgTe 4k, the source is a black body at 980 $^{\circ}\text{C}$ placed at 20 cm of the sample. A germanium filter is used to suppress the high energy part of the blackbody spectrum. The total power is measured to be 200 μW for interdigitated electrodes (2.5 mm^2) and 1.7 μW for nanoelectrodes. The flux is chopped from 1 Hz to 1 kHz. The photocurrent is measured using Zurich Instruments MFLI lock-in amplifier at 1 V bias.

Noise measurement. Current from the device (at 1 V bias, kept in the dark) is amplified by a Femto DLPCA-200, then fed into a SRS SR780 signal analyzer.

Detectivity determination. The detectivity (in Jones) of the sample is determined using the formula: $D^* = \frac{R\sqrt{A}}{S_I}$, where R (in $\text{A}\cdot\text{W}^{-1}$) is the responsivity, S_I is the noise ($\text{A}/\sqrt{\text{Hz}}$) and A the area of the device (cm^2).

SUPPORTING INFORMATIONS

Supporting Information is available from ACS website. They include photoconduction characterization of the materials, details relative to fabrication of the GMR electrodes, input used for electromagnetic simulation and discussion about the robustness of the GMR structure.

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Conflict of interest

The authors declare no conflict of interest.

References

- (1) Chen, H.; Liu, H.; Zhang, Z.; Hu, K.; Fang, X. Nanostructured Photodetectors: From Ultraviolet to Terahertz. *Advanced Materials* **2016**, *28*, 403–433.
- (2) Goubet, N.; Jagtap, A.; Livache, C.; Martinez, B.; Portalès, H.; Xu, X. Z.; Lobo, R. P. S. M.; Dubertret, B.; Lhuillier, E. Terahertz HgTe Nanocrystals: Beyond Confinement. *Journal of the American Chemical Society* **2018**, *140*, 5033–5036.
- (3) Kroupa, D. M.; Vörös, M.; Brawand, N. P.; McNichols, B. W.; Miller, E. M.; Gu, J.; Nozik, A. J.; Sellinger, A.; Galli, G.; Beard, M. C. Tuning Colloidal Quantum Dot Band Edge Positions through Solution-Phase Surface Chemistry Modification. *Nature Communications* **2017**, *8*, 15257.
- (4) Koh, W. K.; Kuposov, A. Y.; Stewart, J. T.; Pal, B. N.; Robel, I.; Pietryga, J. M.; Klimov, V. I. Heavily Doped N-Type PbSe and PbS Nanocrystals Using Ground-State Charge Transfer from Cobaltocene. *Scientific reports* **2013**, *3*, 2004.
- (5) Robin, A.; Livache, C.; Ithurria, S.; Lacaze, E.; Dubertret, B.; Lhuillier, E. Surface Control of Doping in Self-Doped Nanocrystals. *ACS Appl. Mater. Interfaces* **2016**, *8*, 27122–27128.
- (6) Brown, P. R.; Kim, D.; Lunt, R. R.; Zhao, N.; Bawendi, M. G.; Grossman, J. C.; Bulović, V. Energy Level Modification in Lead Sulfide Quantum Dot Thin Films through Ligand Exchange. *ACS Nano* **2014**, *8*, 5863–5872.
- (7) Nag, A.; Kovalenko, M. V.; Lee, J.-S.; Liu, W.; Spokoyny, B.; Talapin, D. V. Metal-Free Inorganic Ligands for Colloidal Nanocrystals: S^{2-} , HS^- , Se^{2-} , HSe^- , Te^{2-} , HTe^- , TeS_3^{2-} , OH^- , and NH_2^- as Surface Ligands. *J. Am. Chem. Soc.* **2011**, *133*, 10612–10620.
- (8) Tang, J.; Kemp, K. W.; Hoogland, S.; Jeong, K. S.; Liu, H.; Levina, L.; Furukawa, M.; Wang, X.; Debnath, R.; Cha, D.; et al. Colloidal-Quantum-Dot Photovoltaics Using Atomic-Ligand Passivation. *Nature Materials* **2011**, *10*, 765–771.
- (9) Choi, J.-H.; Fafarman, A. T.; Oh, S. J.; Ko, D.-K.; Kim, D. K.; Diroll, B. T.; Muramoto, S.; Gillen, J. G.; Murray, C. B.; Kagan, C. R. Bandlike Transport in

- Strongly Coupled and Doped Quantum Dot Solids: A Route to High-Performance Thin-Film Electronics. *Nano Lett.* **2012**, *12*, 2631–2638.
- (10) Kovalenko, M. V.; Scheele, M.; Talapin, D. V. Colloidal Nanocrystals with Molecular Metal Chalcogenide Surface Ligands. *Science* **2009**, *324*, 1417–1420.
 - (11) Lhuillier, E.; Keuleyan, S.; Zolotavin, P.; Guyot-Sionnest, P. Mid-Infrared HgTe/As₂S₃ Field Effect Transistors and Photodetectors. *Advanced Materials* **2013**, *25*, 137–141.
 - (12) Moreels, I.; Lambert, K.; Smeets, D.; De Muynck, D.; Nollet, T.; Martins, J. C.; Vanhaecke, F.; Vantomme, A.; Delerue, C.; Allan, G.; et al. Size-Dependent Optical Properties of Colloidal PbS Quantum Dots. *ACS Nano* **2009**, *3*, 3023–3030.
 - (13) Lhuillier, E.; Keuleyan, S.; Guyot-Sionnest, P. Optical Properties of HgTe Colloidal Quantum Dots. *Nanotechnology* **2012**, *23*, 175705.
 - (14) Kershaw, S. V.; Susha, A. S.; Rogach, A. L. Narrow Bandgap Colloidal Metal Chalcogenide Quantum Dots: Synthetic Methods, Heterostructures, Assemblies, Electronic and Infrared Optical Properties. *Chem. Soc. Rev.* **2013**, *42*, 3033–3087.
 - (15) Lhuillier, E.; Guyot-Sionnest, P. Recent Progresses in Mid Infrared Nanocrystal Optoelectronics. *IEEE J. Selected Topics in Quantum Electronics* **2017**, *23*, 6000208.
 - (16) Hu, C.; Gassenq, A.; Justo, Y.; Devloo-Casier, K.; Chen, H.; Detavernier, C.; Hens, Z.; Roelkens, G. Air-Stable Short-Wave Infrared PbS Colloidal Quantum Dot Photoconductors Passivated with Al₂O₃ Atomic Layer Deposition. *Appl. Phys. Lett.* **2014**, *105*, 171110.
 - (17) Yakunin, S.; Dirin, D. N.; Protesescu, L.; Sytnyk, M.; Tollabimazraehno, S.; Humer, M.; Hackl, F.; Fromherz, T.; Bodnarchuk, M. I.; Kovalenko, M. V.; et al. High Infrared Photoconductivity in Films of Arsenic-Sulfide-Encapsulated Lead-Sulfide Nanocrystals. *ACS Nano* **2014**, *8*, 12883–12894.
 - (18) Thambidurai, M.; Jang, Y.; Shapiro, A.; Yuan, G.; Xiaonan, H.; Xuechao, Y.; Wang, Q. J.; Lifshitz, E.; Demir, H. V.; Dang, C. High Performance Infrared Photodetectors up to 2.8 μm Wavelength Based on Lead Selenide Colloidal Quantum Dots. *Opt. Mater. Express* **2017**, *7*, 2326–2335.
 - (19) Bi, Y.; Pradhan, S.; Gupta, S.; Akgul, M. Z.; Stavrinadis, A.; Konstantatos, G. Infrared Solution-Processed Quantum Dot Solar Cells Reaching External Quantum Efficiency of 80% at 1.35 μm and J_{sc} in Excess of 34 $\text{mA}\cdot\text{cm}^{-2}$. *Advanced Materials* **2018**, *30*, 1704928.
 - (20) Killilea, N.; Wu, M.; Sytnyk, M.; Amin, A. A. Y.; Mashkov, O.; Spiecker, E.; Heiss, W. Pushing PbS/Metal-Halide-Perovskite Core/Epitaxial-Ligand-Shell Nanocrystal Photodetectors beyond 3 μm Wavelength. *Advanced Functional Materials* **2019**, *29*, 1807964.
 - (21) Kovalenko, M. V.; Kaufmann, E.; Pachinger, D.; Roither, J.; Huber, M.; Stangl, J.; Hesser, G.; Schäffler, F.; Heiss, W. Colloidal HgTe Nanocrystals with Widely Tunable Narrow Band Gap Energies: From Telecommunications to Molecular Vibrations. *J. Am. Chem. Soc.* **2006**, *128*, 3516–3517.
 - (22) Tang, X.; Ackerman, M. M.; Chen, M.; Guyot-Sionnest, P. Dual-Band Infrared Imaging Using Stacked Colloidal Quantum Dot Photodiodes. *Nature Photonics* **2019**, *13*, 277.
 - (23) Jagtap, A.; Martinez, B.; Goubet, N.; Chu, A.; Livache, C.; Gréboval, C.; Ramade, J.; Amelot, D.; Troussset, P.; Triboulin, A.; et al. Design of a Unipolar Barrier for a

- Nanocrystal-Based Short-Wave Infrared Photodiode. *ACS Photonics* **2018**, *5*, 4569–4576.
- (24) Cryer, M. E.; Halpert, J. E. 300 nm Spectral Resolution in the Mid-Infrared with Robust, High Responsivity Flexible Colloidal Quantum Dot Devices at Room Temperature. *ACS Photonics* **2018**, *5*, 3009–3015.
- (25) Chen, M.; Yu, H.; Kershaw, S. V.; Xu, H.; Gupta, S.; Hetsch, F.; Rogach, A. L.; Zhao, N. Fast, Air-Stable Infrared Photodetectors Based on Spray-Deposited Aqueous HgTe Quantum Dots. *Advanced Functional Materials* **2014**, *24*, 53–59.
- (26) Chen, M.; Lu, H.; Abdelazim, N. M.; Zhu, Y.; Wang, Z.; Ren, W.; Kershaw, S. V.; Rogach, A. L.; Zhao, N. Mercury Telluride Quantum Dot Based Phototransistor Enabling High-Sensitivity Room-Temperature Photodetection at 2000 nm. *ACS Nano* **2017**, *11*, 5614–5622.
- (27) Tong, J.; Suo, F.; Ma, J.; Tobing, L. Y. M.; Qian, L.; Zhang, D. H. Surface Plasmon Enhanced Infrared Photodetection. *Opto-Electron. Adv.* **2019**, *02*, 180026.
- (28) Massiot, I.; Vandamme, N.; Bardou, N.; Dupuis, C.; Lemaître, A.; Guillemoles, J.-F.; Collin, S. Metal Nanogrid for Broadband Multiresonant Light-Harvesting in Ultrathin GaAs Layers. *ACS Photonics* **2014**, *1*, 878–884.
- (29) Massiot, I.; Colin, C.; Péré-Laperne, N.; Roca i Cabarrocas, P.; Sauvan, C.; Lalanne, P.; Pelouard, J.-L.; Collin, S. Nanopatterned Front Contact for Broadband Absorption in Ultra-Thin Amorphous Silicon Solar Cells. *Appl. Phys. Lett.* **2012**, *101*, 163901.
- (30) Montoya, J. A.; Tian, Z.-B.; Krishna, S.; Padilla, W. J. Ultra-Thin Infrared Metamaterial Detector for Multicolor Imaging Applications. *Opt. Express* **2017**, *25*, 23343–23355.
- (31) Verdun, M.; Portier, B.; Jaworowicz, K.; Jaeck, J.; Lelarge, F.; Guilet, S.; Dupuis, C.; Haïdar, R.; Pardo, F.; Pelouard, J.-L. Guided-Mode Resonator for Thin InGaAs P-i-N Short-Wave Infrared Photo-Diode. *Appl. Phys. Lett.* **2016**, *108*, 053501.
- (32) Jackson, E. M.; Nolde, J. A.; Kim, M.; Kim, C. S.; Cleveland, E. R.; Affouda, C. A.; Canedy, C. L.; Vurgaftman, I.; Meyer, J. R.; Aifer, E. H.; et al. Two-Dimensional Plasmonic Grating for Increased Quantum Efficiency in Midwave Infrared Nbn Detectors with Thin Absorbers. *Opt. Express* **2018**, *26*, 13850–13864.
- (33) Palaferri, D.; Todorov, Y.; Bigioli, A.; Mottaghizadeh, A.; Gacemi, D.; Calabrese, A.; Vasanelli, A.; Li, L.; Davies, A. G.; Linfield, E. H.; et al. Room-Temperature Nine-Mm-Wavelength Photodetectors and GHz-Frequency Heterodyne Receivers. *Nature* **2018**, *556*, 85–88.
- (34) Yifat, Y.; Ackerman, M.; Guyot-Sionnest, P. Mid-IR Colloidal Quantum Dot Detectors Enhanced by Optical Nano-Antennas. *Appl. Phys. Lett.* **2017**, *110*, 041106.
- (35) Le-Van, Q.; Le Roux, X.; Aassime, A.; Degiron, A. Electrically Driven Optical Metamaterials. *Nature Communications* **2016**, *7*, 12017.
- (36) Baek, S.-W.; Ouellette, O.; Jo, J. W.; Choi, J.; Seo, K.-W.; Kim, J.; Sun, B.; Lee, S.-H.; Choi, M.-J.; Nam, D.-H.; et al. Infrared Cavity-Enhanced Colloidal Quantum Dot Photovoltaics Employing Asymmetric Multilayer Electrodes. *ACS Energy Lett.* **2018**, *3*, 2908–2913.
- (37) Tang, X.; Wu, G. fu; Lai, K. W. C. Plasmon Resonance Enhanced Colloidal HgSe Quantum Dot Filterless Narrowband Photodetectors for Mid-Wave Infrared. *J. Mater. Chem. C* **2017**, *5*, 362–369.

- (38) Tang, X.; Ackerman, M. M.; Shen, G.; Guyot-Sionnest, P. Towards Infrared Electronic Eyes: Flexible Colloidal Quantum Dot Photovoltaic Detectors Enhanced by Resonant Cavity. *Small* **2019**, *15*, 1804920.
- (39) Chen, M.; Shao, L.; Kershaw, S. V.; Yu, H.; Wang, J.; Rogach, A. L.; Zhao, N. Photocurrent Enhancement of HgTe Quantum Dot Photodiodes by Plasmonic Gold Nanorod Structures. *ACS Nano* **2014**, *8*, 8208–8216.
- (40) Tang, X.; Ackerman, M. M.; Guyot-Sionnest, P. Thermal Imaging with Plasmon Resonance Enhanced HgTe Colloidal Quantum Dot Photovoltaic Devices. *ACS Nano* **2018**, *12*, 7362–7370.
- (41) Prins, F.; Kim, D. K.; Cui, J.; De Leo, E.; Spiegel, L. L.; McPeak, K. M.; Norris, D. J. Direct Patterning of Colloidal Quantum-Dot Thin Films for Enhanced and Spectrally Selective Out-Coupling of Emission. *Nano Lett.* **2017**, *17*, 1319–1325.
- (42) De Leo, E.; Cocina, A.; Tiwari, P.; Poulikakos, L. V.; Marqués-Gallego, P.; le Feber, B.; Norris, D. J.; Prins, F. Polarization Multiplexing of Fluorescent Emission Using Multiresonant Plasmonic Antennas. *ACS Nano* **2017**, *11*, 12167–12173.
- (43) Baek, S.-W.; Molet, P.; Choi, M.-J.; Biondi, M.; Ouellette, O.; Fan, J.; Hoogland, S.; Arquer, F. P. G. de; Mihi, A.; Sargent, E. H. Nanostructured Back Reflectors for Efficient Colloidal Quantum-Dot Infrared Optoelectronics. *Advanced Materials* **2019**, *31*, 1901745.
- (44) Sharon, A.; Rosenblatt, D.; Friesem, A. A. Resonant Grating–Waveguide Structures for Visible and near-Infrared Radiation. *Journal of the Optical Society of America A* **1997**, *14*, 2985.
- (45) Keuleyan, S.; Lhuillier, E.; Guyot-Sionnest, P. Synthesis of Colloidal HgTe Quantum Dots for Narrow Mid-IR Emission and Detection. *J. Am. Chem. Soc.* **2011**, *133*, 16422–16424.
- (46) Martinez, B.; Ramade, J.; Livache, C.; Goubet, N.; Chu, A.; Gréboval, C.; Qu, J.; Watkins, W. L.; Becerra, L.; Dandeu, E.; et al. HgTe Nanocrystal Inks for Extended Short-Wave Infrared Detection. *Advanced Optical Materials* **2019**, *7*, 1900348.
- (47) Lai, Y.; Li, H.; Kim, D. K.; Diroll, B. T.; Murray, C. B.; Kagan, C. R. Low-Frequency 1/f Noise in Nanocrystal Field-Effect Transistors. *ACS Nano* **2014**, *8*, 9664–9672.
- (48) Liu, H.; Lhuillier, E.; Guyot-Sionnest, P. 1/f Noise in Semiconductor and Metal Nanocrystal Solids. *Journal of Applied Physics* **2014**, *115*, 154309.
- (49) De Iacovo, A.; Venettacci, C.; Colace, L.; Scopa, L.; Foglia, S. Noise Performance of PbS Colloidal Quantum Dot Photodetectors. *Appl. Phys. Lett.* **2017**, *111*, 211104.
- (50) Wang, H.; Lhuillier, E.; Yu, Q.; Zimmers, A.; Dubertret, B.; Ulysse, C.; Aubin, H. Transport in a Single Self-Doped Nanocrystal. *ACS Nano* **2017**, *11*, 1222–1229.
- (51) Konstantatos, G.; Sargent, E. H. PbS Colloidal Quantum Dot Photoconductive Photodetectors: Transport, Traps, and Gain. *Appl. Phys. Lett.* **2007**, *91*, 173505.
- (52) Lhuillier, E.; Dayen, J.-F.; Thomas, D. O.; Robin, A.; Doudin, B.; Dubertret, B. Nanoplatelets Bridging a Nanotrench: A New Architecture for Photodetectors with Increased Sensitivity. *Nano Lett.* **2015**, *15*, 1736–1742.
- (53) Chu, A.; Ferre, S.; Noguier, V.; Gréboval, C.; Livache, C.; Martinez, B.; Prado, Y.; Goubet, N.; Cruguel, H.; DuDy, L.; et al. HgTe Nanocrystals for SWIR Detection and Their Integration up to Focal Plane Array. *ACS Appl. Mater. Interfaces* **2019**, asap.

- (54) Bierret, A.; Vincent, G.; Jaeck, J.; Pelouard, J.-L.; Haïdar, R.; Pardo, F. Field Extension inside Guided-Mode-Resonance Filters under a Focused Beam. *Opt. Lett.* **2017**, *42*, 4187–4190.
- (55) Kondo, T.; Ura, S.; Magnusson, R. Design of Guided-Mode Resonance Mirrors for Short Laser Cavities. *J. Opt. Soc. Am. A* **2015**, *32*, 1454–1458.
- (56) Bierret, A.; Vincent, G.; Jaeck, J.; Pelouard, J.-L.; Pardo, F.; Barrière, F. D. L.; Haïdar, R. Pixel-Sized Infrared Filters for a Multispectral Focal Plane Array. *Appl. Opt.* **2018**, *57*, 391–395.
- (57) Hugonin, J.-P.; Lalanne, P. Reticolo Software for Grating Analysis. *Reticolo software for grating analysis, Institut d'Optique, Orsay, France* **2005**.

TOC graphic

