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1	A fast response and recovery H_2S gas sensor based on free-standing
2	${\rm TiO}_2$ nanotube array films prepared by one-step anodization method
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13	
14	Abstract
15	Free-standing TiNT array film was successfully synthesized by a one-step
16	anodization method. The characteristic techniques including scanning electron
17	microscopy (SEM), transmission electron microscopy (TEM) and X-ray diffraction
18	(XRD) were adopted to characterize the morphology and chemical composition of the
19	TiNT array film. Subsequently, gas sensor based on the TiNT array film was
20	fabricated and its sensing performance toward H_2S was investigated. The results
21	showed that the optimum operating temperature to detect H_2S gas was 300°C, the
22	detection range to H_2S gas was 1-50 ppm with the response value of 4.5-26.2, and a

23	good linearity of sensing characteristics could be observed. Meanwhile, the response
24	and recovery time of the sensor to 50 ppm H_2S gas were as low as 22 s and 6 s,
25	respectively. In addition, mechanisms of the development of the free-standing TiNT
26	array film and the sensor towards H_2S were discussed. In conclusion, its outstanding
27	sensing properties and readily fabrication of the TiNT array film sensor presented the
28	potential industrial applications.

30 Keywords: TiO₂ nanotube array film; Free-standing; Anodization; H_2S ; Gas sensor

31

32 **1. Introduction**

Hydrogen sulfide (H_2S) is one of the main pollutants produced in various industrial processes. It is colorless, highly toxic, extremely flammable and corrosive in nature [1,2]. Humans exposed to H_2S at low concentrations suffer eye irritation, olfactory fatigue, and damage to the lungs and nervous system. The inhalation of 320 ppm H_2S may collapse the heartbeat leading to sudden death [3,4]. Thus, from a safety point of view, the monitoring of H_2S is crucial in many industrial areas.

Recently, there have been many reports on the H₂S sensors based on metal oxide semiconductors, such as: Fe₂O₃ [5], CuO [6], ZnO [7], WO₃ [8] etc. In the past decades, due to its interesting physical and chemical properties, titanium dioxide (TiO₂) has been considered to be one of the important multifunctional materials with photocatalytic, photoelectrochemical and gas sensor applications. In particular, being similar to other metal oxide semiconductors, TiO₂ also has shown responses to several kinds of gases [9-11]. Although the detection results have been reported so far, the research on the H_2S sensors with TiO₂ is relatively new and their gas-sensing performance needs to be improved [12,13]. Thus, developing new sensing material with high response, good selectivity and quick response is extremely important for real-time monitoring of H_2S .

Recently the improved surface reactivity of TiO2 systems has been reported, 50 where the nanowires, nanorods and nanotubes have been used [14-16]. One of the 51 main features of these nanostructures is the large specific surface area that makes 52 53 them attractive for use as the sensitive films for gas sensors, which can improve the gas sensitive performance effectively. Although the TiO₂ nanotube (TiNT) arrays have 54 been successfully synthesized with the electrochemical anodization method, most 55 56 sensitive performance were just obtained on the generated nanotube array without assembling, which hinders the practical application of TiNT [16-18]. 57

Several methods for the preparation of the independent TiO₂ nanotube arrays 58 59 have been reported, which are classified into physical methods and chemical methods. 60 The physical methods mainly included solvent evaporation film separation [19] and ultrasonic oscillation [20], whose main drawback was that it was difficult to obtain the 61 large complete independent TiO₂ thin film, and the morphology of the titanium tube 62 was prone to be destroyed. The chemical methods mainly included the secondary 63 anodic oxidation [21] and the dissolution method in chemical solution [22]. In other 64 65 words, the secondary treatments were obligatory during the preparation process of the TiNT array membrane and the thick film restricted their feasibility for gas sensing 66

67 applications.

In this work, in Section 3.1, the process of obtaining a complete free-standing 68 69 TiNT array membrane by the anodic oxidation method followed by annealing without any secondary treatments is presented. In Section 3.2, the morphology of the TiNT 70 71 membrane obtained with this cost-effective and time-saving method is studied. In Section 3.3, obtained free-standing array film of TiNT to fabricate a gas sensor is 72 explained and its gas sensing performances to H₂S are carefully investigated. In 73 Section 3.4, the gas sensing mechanisms of the developed free-standing TiNT array 74 75 film are also analyzed in the paper.

76

77 2. Materials and methods

78 **2.1 Synthesis of TiO₂ nanotube array**

Prior to the anodization experiments, the titanium foils (99.7% purity, 0.01 mm thickness) were ultrasonically cleaned for 10 min in acetone, alcohol and deionized water in sequence, finally dried in air. The anodization was conducted in a two-electrode electrochemical cell with a platinum foil as cathode and the titanium foil as anode at a constant potential. All the anodization experiments were carried out at room temperature with stirring.

The anodizing was carried out at 30 V for 2 h using a direct-current stabilizer (TASI-1305, Suzhou TASI electronic CO., LTD.). The growth of the nanotube arrays was obtained using a glycol solution with 0.55 wt% ammonium fluoride and 20 wt% deionized water. After that, the anodized titanium foil was rinsed with deionized water,

89	and dried in air. The samples were heated up to 100°C, 200°C, 300°C, 400°C for 10 min
90	orderly, at the rate of 10°C/min, and then annealed at 450°C for 2 h in the air.

92 **2.2** Characterization of the nanotube array film

The morphologies of the free-standing TiNT array film samples were characterized by the scanning electron microscopy (SEM) and the energy-dispersive X-ray spectroscopy (EDS) (Merlin, Zeiss, Germany).

The X-ray diffraction (XRD) patterns were recorded at the room temperature
with Cu Kα radiation of 0.15418 nm in an X-ray diffractometer (D8 ADVANCE,
Bruker, Germany), using a generator voltage of 40 kV and current of 40 mA. The data
were collected for scattering angles (20) ranging from 10° to 85° with a step of 0.02°
for 2 s per point. The transition electron microscope (TEM) was performed with the
transmission electron microscopy (JEM-2100F, Japan).

102

103 2.3 Sensor fabrication

For fabrication of the gas sensor, the alumina plate (1.0×1.5 mm) with the gold electrodes and heating layer processed by the method of screen painting printing [23, was used as the substrate. In order to remove the stain, the substrates were immersed in acetone, ethanol and deionized water, successively, for 10 min.

A mixture of 10 mL of titanium tetraisopropoxide solution in ethanol with terpilenol in volume ratio of 6: 3: 2 was painted onto the side of the substrate for facilitation the later adhesion for the free-standing TiNT array membrane. Then, the 111 sensor attached with the nanotube layer was heated at 350° C for 30 min. After heating 112 treatment, the adhesive was transformed into the TiO₂ interlayer, which let the 113 connection between the substrate and TiNT array film became more compact. 114 Because of the little amount of TiO₂, the contribution of the TiO₂ interlayer to the 115 sensing performance can be considered negligible.

The schematic diagram of the sensor structure is shown in Fig. 1 for the sensing measurements. To improve its stability and repeatability, the gas sensor was welded on the pedestal, followed by aging at 450°C for 72 h in air.

119

120 2.4 Gas sensing measurements

The measurement of the gas sensing performances of the sensor were conducted by a WS-30A Gas Sensing Measurement System (Weisen Electronic Technology Co., Ltd., China). Fig. 2 displays a schematic diagram the whole measuring process. The final concentration of the H_2S inside the test chamber was controlled by mass flow controllers (MFC) connected to calibrated bottles of N_2 and H_2S . The relative humidity (RH) was about 50% in the whole testing process.

As shown in the measuring electric circuit (Fig. 2), in the sensing test process, an appropriate bias voltage (V_t =5 V) is applied. The working temperature of the sensors could be controlled from room temperature to 500°C by the heating voltage V_h . R_s is a resistor of the sensor and R_l is a load resistor connected in series with the gas sensor.

131 The gas response was defined as the ratio of the stationary electrical resistance of 132 the sensor in the air R_a to the resistance in the test gas R_g , that is, Response= R_a/R_g .

133	The response and recovery times of the films were calculated from the response
134	curves automatically. The response and recovery times were defined as the time
135	required for a change of the response value to reach 90% of the equilibrium value and
136	falls to 10% of its maximum sensitivity after injecting and removing the detected gas,
137	respectively.

139 **3. Results and discussions**

140 **3.1 Development of the free-standing TiNT array film**

The formation of the free-standing TiNT array film by electrochemical anodization is a complex process, its growth mechanism can be represented by Fig. 3. The overall process of oxide formation was divided into four steps [25-27]:

144 (1) Formation of the TiO_2 via oxidation

At the beginning of this stage, the concentration of Ti^{4+} was raised by the electrochemical etching of the Ti, then the amorphous TiO_2 film was formed through the reaction between Ti^{4+} and adsorbed O^{2-} on the surface of the electrode.

148 (2) Formation of the pores

After the formation of the initial TiO_2 layer, fluorine ions could attack the oxide leading to the formation of the pores through the produced water-soluble $[TiF_6]^{2-}$ species. With the extension of the oxidation time, the ordered pores were formed on the surface of the TiO_2 layer. Under the effect of electrochemical etching, the depth of the pores would be increased gradually.

154 (3) Formation of the TiO_2 nanotube array

The electric field intensity on the concave surface of the pores is lower than outside the pores, hence the corrosion rate of titanium dioxide on these sites is the fastest [28]. With the migration of the charge density to the bottom of the nanotubes, the oxidation film was dissolved along the axial direction of the tubes, gradually forming the tubular structure film layer. However, as shown in Fig. 4 (a), as time goes on, the dissolving speed decreases, so that the thickness of the nanotubes wall gradually increased from top (13 nm) to bottom (36 nm).

Moreover, the metallic region between the tube holes also undergoes a similar transition leading to the formation of the array of relatively independent nanotubes (Fig. 4).

165 (4) Formation of the free-standing TiNT array film

166 The tube length is governed by competition between anodic oxide formation and its chemical dissolution. While oxygen and fluorine ions around the titanium are 167 gradually consumed, both the fluorine ion etching rate and the TiO₂ forming rate 168 169 begin to decrease until reaching an equilibrium. As shown in Fig. 5, when the reaction lasts 4 hours, the Ti substrate had been completely oxidated to TiNT array. However, 170 as presented in Fig. 4 (b-c), after 4 h, the pore mouths have been eroded. When the 171 reaction duration is increased to 6 h, the tube holes are more dissolved and begin to 172 173 collapse.

With the progress of oxidation time, the length of the TiNT increases leading to an increased strain strength between the TiNT array film and Ti substrate. When the reaction time attains 2 hours, the compressive stress reaches the critical debongding stress. Moreover, annealing modifies the interface between them. Hence, the nanotubearray film can be desquamated off naturally after heating treatment [29].

In order to ensure the integrity of nanotube mouth and membrane, the oxidationtime was taken as 2 hours in the subsequent experiments.

181

3.2 Structural and morphological characteristics

Vertically oriented, high aspect ratio TiNT arrays grown by anodic oxidation of 183 Ti foil were flaked off naturally after the annealing treatment. The transparent TiNT 184 185 peeled off the Ti foil is shown in Fig. 6 (a). Fig. 6 (b-d) present the SEM images with side, top and bottom views of the free-standing TiNT array film. The average inner 186 diameter of the nanotube was around 110 nm with a wall thickness of 13 nm and the 187 188 tube length was 3.8 µm. The hollow structure and the wall of the TiNT can be seen clearly in Fig. 7. The enlarged TEM image shows a fringe pattern, where the resolved 189 spacing between the two parallel neighboring fringes is 0.35 nm, corresponding to the 190 191 interplanar distance of the $\{101\}$ lattice planes in anatase TiO₂ (Fig. 4b) [30,31].

To better understand the crystalline form and the surface element distribution, the XRD and EDS were performed. Fig. 8 shows the X-ray diffraction (XRD) patterns of the TiNT array calcined at 450°C for 2 h. It can be seen that the phase present was anatase (JCPDF No. 21-1272). The diffraction peaks at 2θ = 25.1°, 37.4°, 47.8°, 53.8°, 54.9° and 62.8° are identified to be the (101), (004), (200), (105), (211) and (204) crystal faces, respectively [12,32]. These crystalline peaks indicate that the crystal structure of TiNT film is the anatase phase. Moreover, Fig. 9 reveals the

199	stoichiometric TiO_2 composition of the film measured by the energy dispersive
200	spectroscopy method. Hence, the titanium nanotubes are mainly composed with the
201	titanium and oxygen element, which proved that the compound was titania.
202	
203	3.3 Gas sensing properties
204	From the results mentioned above, we had obtained the free-standing TiNT array
205	film prepared by one-step electrochemical anodization followed by thermal annealing.
206	
207	3.3.1 Optimum operating temperature for gas sensing
208	In order to evaluate the optimum operating temperature, the sensor was exposed
209	to 50 ppm H_2S at temperatures ranging from room temperature to 400°C. The response
210	of the sensor to H_2S with respect to its operating temperature is presented in Fig. 10.
211	Over this temperature range, the response of the sensor increased with the
212	temperature up to 300°C and then started to decrease. At temperatures below 300°C, an
213	increase in the operating temperature promoted $\mathrm{H}_2 \mathrm{S}$ chemisorption and the reaction
214	between the adsorbed H_2S molecules and the adsorbed oxygen species, leading to an
215	enhanced response to H_2S gas. When the operating temperature was 300°C, the
216	response value was 26.2. In contrast, at temperatures higher than 300°C, the response
217	to H_2S was reduced due to desorption of H_2S , which decreased the amount of H_2S
218	adsorbed on the sensor surface [5,33]. Therefore, the optimal operating temperature
219	was 300° C for the TiO ₂ sensor.

221 **3.3.2** Response and recovery time

The response and recovery time are two important parameters for a gas sensor in practical application [34]. Fig. 11 is an image of the real-time gas sensing transient to 50 ppm H₂S at 300°C drawn to show the moments of gas input and gas stop. The response value increased rapidly when the TiNT sensor was exposed to H₂S gas and returned to the initial value when the H₂S gas supply was stopped and air was introduced. The response and recovery times of the sensor are measured from Fig. 11 and presented in Fig. 12.

229 From Fig. 12, it can be seen that the sensing films have short response-recovery times, suggesting that the sensors respond quickly to both the injection and removal of 230 the H₂S gas in air. Moreover, the response and recovery times do not dependent on the 231 232 concentration of H₂S at 1-50 ppm. However, the response times of the sensor exposed to H_2S at 10-50 ppm are longer than at 1-10 ppm. The phenomenon reveals that the 233 response time increases with the increasing concentration of H₂S. When exposed to 234 235 50 ppm H_2S at 300°C, the response time and recovery time of the sensor were 22 s and 6 s, respectively. 236

237

238 **3.3.3 Effects of gas concentration**

The sensitive performance of the sensor was further investigated by exposing the sensors to different concentrations of H_2S gas at the operating temperature of 300°C. The curves of gas response versus time for 1-10 ppm of H_2S gas sensed by the TiNT array film at 300°C are shown in Fig. 13 (a), where the response amplitude of the sensor increases with H_2S concentration. Fig. 13 (b) shows that the sensitivity for the TiO₂ sensor is improved with increasing H_2S concentration in the range of 1-50 ppm. The sensor responses toward 1 and 10 ppm are found about 4.5 to 12.4, respectively. It could be clearly seen that the response increases linearly with the concentrations in this range, indicating a good linearity of sensing characteristics. Moreover, when the concentration of H_2S exceeds about 10 ppm, the sensitivity still increases but at a lower rate.

250

251 **3.3.4 Selectivity studies**

The gas sensing selectivity of the TiNT array film was further tested at 300°C by exposing the sensors to 50 ppm potentially interfering gases including methanol, formaldehyde, benzene, toluene, xylene and n-dodecane vapors, and the results are depicted in Fig. 14. Clearly, the sensor based on the TiNT is the most sensitive to H_2S , whereas it presents a low response to the other interfering gases at the same temperature, implying that the sensor exhibits a better selectivity to H_2S than to other gases.

259

260 **3.3.5 Stability and repeatability**

Thermal stability and repeatability of the sensor had been investigated by repeating experiments for 50 ppm H₂S at 300°C. Fig. 15 displays the measured values for the sensor response versus the storing time. As displayed in the figure, the response results of the sensor possess a variation of $\pm 1.4\%$ from the initial result (26.2)

265	in one month. The free-standing TiNT array film demonstrated a good stability and
266	reproducibility with regard to 50 ppm H_2S gas at 300°C, which indicated a stable
267	morphology and good crystallinity of the fabricated sensing layer.

269 3.4 Gas sensing mechanism

It is well known that the sensing mechanism of TiO₂-based gas sensors belongs 270 to the surface-controlled type, which is based on the change in conductance of the 271 semiconductor [35,36]. When the sensor is exposed to air, electrons in the surface 272 273 region of the TiO_2 nanotube walls are consumed by the oxygen molecules that are adsorbed on the TiO₂ surface. Then a depletion region is created on the wall of the 274 TiNT, leading to an increase in the electric resistance of the TiNT array film [37]. 275 276 Another point to be noted is that the type of adsorbed oxygen species depends on the working temperature of sensor. The changing process of oxygen molecules on the 277 nanotube surface is illustrated in Eqs. (1-4) [38,39]: 278

$$279 \qquad O_2(g) \to O_2(ads) \tag{1}$$

280
$$O_2(ads) + e^- \rightarrow O_2^-(ads) \quad (T \le 147^{\circ}C)$$
 (2)

281
$$O_2^{-1}(ads) + e^{-1} \rightarrow 2O^{-1}(ads) \quad (147^{\circ}C \le T \le 397^{\circ}C)$$
 (3)

282
$$O^{-}(ads) + e^{-} \rightarrow O^{2-}(ads) \quad (T \ge 397^{\circ}C)$$
 (4)

When the sensor was exposed to H_2S gas, which was a reducing gas, these chemisorbed oxygen species played a crucial role in enhancing the H_2S gas sensitivity as explained by the schematic diagram in Fig. 16. The H_2S reacted with these oxygen species and got dissociated after releasing trapped electrons to the TiNT via the speculated reactions given in Eqs. (5-6) [4,12]:

288
$$H_2S(g) \rightarrow H_2S(ads)$$
 (5)

289
$$2H_2S(g) + 6O^-(ads) \rightarrow 2H_2O(g) + 2SO_2(g) + 6e^-$$
 (6)

This leaded to an increase in the carrier concentration on the TiO_2 nanotube walls and a decrease in the surface depletion layer width. Hence, there is an increase in the response value as shown in Fig. 11.

Table 1 summarizes the results of gas response to H_2S using various forms of TiO₂. The response and recovery time with most of the H_2S sensors prepared by TiO₂ materials were long, or a modification method was necessary to improve its gas sensitive performances. However, in this work, the response value, response and recovery times of the purified TiNT array film sensor toward H_2S gas were comparable to those of the most other TiO₂ nanomaterials.

The following physical and chemical effects might contribute partly to the enhanced H_2S sensing properties of the TiNT array film sensor: (1) electrons are conducted through the wall of TiNT because of its particular structure, which reduces the recombination of electrons and holes; (2) the TiNT that presents a large surface area due to their tube array structure results in a better sensitivity.

304

4. Conclusion and perspectives

The free-standing TiNT array film was successfully synthesized by a one-step anodization method. Then the film was fabricated and tested for H_2S gas sensing at low concentrations.

309	The H_2S sensing properties including operating temperature, reversibility,
310	relationship between sensitivity and H ₂ S concentration, selectivity and stability were
311	studied. The results showed that the optimum operating temperature to detect H_2S gas
312	was 300°C, the detection range to H_2S gas was 1-50 ppm with the response value of
313	4.5-26.2, and a good linearity of sensing characteristics could be observed. It is
314	generally accepted that metal oxide gas sensors possess relatively longer response and
315	recovery times. However, the response and recovery time of the sensor to 50 ppm H_2S
316	gas were as low as 22 s and 6 s, respectively. Moreover, the growth mechanism and
317	gas sensing mechanism of the TiNT array film were studied.
318	It can be seen that the response values, response and recovery times of the sensor

It can be seen that the response values, response and recovery times of the sensor were outstanding throughout the tests, which demonstrated that the fabricated TiNT array film sensor presented the potential industrial applications. Moreover, this electrochemical anodization method based on the Ti foil is relatively simple and, hence, induces lower costs and makes it be attractive for developing gas sensors devices.

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No.	Morphology	Crystal Structure	Response $(R=R_a/R_a)$	Concentration (ppm)	Temperature	Response time (s)	Recovery time (s)	Reference
1	Ag-doped TiO ₂ nanofiber	Anatase & rutile	8.5	100	350	-	-	Ma et al., 2016
2	TiO ₂ nanoparticle-d ecorated Fe ₂ O ₃ nanorods	anatase	7.4	200	300	150	157	Kheel et al., 2016
3	TiO ₂ -Al ₂ O ₃	rutile	38.7	1000	650	390	480	Arafat et al., 2017
4	CuO doped TiO ₂ nanoparticle	anatase	1.77	50	Room temperature	14	22	Chaudhari and Mishra, 2016
5	TiO ₂ nanoplates	anatase	4.8	10	300	10	-	Guo et al., 2016
6	TiO ₂ nanowires	rutile	11	80	140	-	-	Munz et al., 2013
7	TiO ₂ pellet	rutile	275	10	100	150	2500	Jagadale et al., 2015
	Free-standing							
8	TiNT array	anatase	26	50	300	22	6	Our work
	film							

Table 1. Comparison of the sensor responses to H_2S with different forms of TiO_2 .



Figure 1. (a) Photographs of the blank sensor. (b)A schematic illustration of the sensor coated with the sensing material. (c) Top view and (d) side view of the fabricated sensor.



Figure 2. Experimental setup and measuring electric circuit for gas sensing performance.



Figure 3. Schematic diagram of the growth mechanism of free-standing TiNT array film.







Figure 4. Top-view SEM images of TiNT array film at different oxidation time.



Figure 5. Side-view SEM image of the TiNT array at oxidation time of 4h.



(a)(b)Image: State Sta

Figure 6. (a) Photographic image of the detached TiNT-array film, (b) Top-view SEM, (c) Side-view SEM, (d) Bottom-view SEM image of the TiNT array with the oxidation time of 2h..



Figure 7. Wide-field TEM and HRTEM images of a single TiNT.



Figure 8. XRD peaks of the TiNT.



Figure 9. Energy dispersive spectroscopy spectrum of the TiNT arrays.



Figure 10. Response values of the free-standing TiNT array film to 50 ppm H₂S at various operating temperatures.



Figure 11. The real-time response curve to 50 ppm H_2S of the sensor at 300°C.



Figure 12. Summary of response time and recovery time of the sensor exposed to H_2S at different concentrations at 300°C.



Figure 13. (a) Real-time response curve of the TiNT array film sensor to H₂S with increased concentration at a working temperature of 300°C; (b) Relationship between the sensitivity and H₂S concentration.



Figure 14. Comparison of sensor response obtained for different gases of 50 ppm.



Figure 15. Variation of the sensor response to 50 ppm H₂S at 300 °C for different duration times.



Figure 16. Band diagrams and schematic images of the surface reactions under different atmospheres: (a) exposed in the air (b) in the presence of H₂S gas.