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# In situ X-ray tomography densification of firn: the role of mechanics and diffusion processes.

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# Abstract

One of the most efficient proxy methods for paleoclimatology consists of obtaining data previously preserved within polar ice cores. Models for past climate reconstruction are based in particular on the characterization of entrapped gases in ice closed pores. Improving the temporal accuracy of these models requires a better understanding of firn densification mechanisms. In particular, the interplay between viscoplastic deformation and diffusion processes for pore closure is not well understood. In this work, we describe the first in situ laboratory densification experiments on polar firn retrieved from Antarctica with live characterization by X-ray tomography. Our in situ tests allow for the first time to approximately access the process of pore closure in ice, which takes thousands of years to occur in Antarctica, from visualizations and quantitative analyses of short time laboratory experiments. The parameters of pore separation and closure and the microstructural changes that accompany them are monitored. We show that densification of polar

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firn and pore closure could be replicated at higher strain rate and warmer temperature. Experiments allow the viscoplastic part of the firn deformation to be decoupled from the diffusion mechanisms that occur at high temperature. Our results show that density alone is not sufficient to predict the close-off density at which gases get entrapped. More generally, the method laid out here may find useful application in the domain of high temperature powder compaction, for which pore closure and grain growth are significant process parameters.

Keywords: firn, X-ray tomography, pore closure, in situ compaction

#### <sup>1</sup> Introduction

Polar firn contains a large part of the history of atmospheric air in its pores [1]. Thanks to the progressive densification of snow to ice, gases are entrapped in firn, and progressively buried down in the ice core. This process occurs by gradual pore closure with the age of gases being distributed for a given depth. Thus, dating a particular atmospheric composition, and relating it to changes in past climate temperature is challenging. In particular, improving the evaluation of the gas-ice age difference requires a better understanding of firn densification and gas-trapping mechanisms.

From a climate point of view, the firn column can be divided into three zones [2]. Top layers constitute a convective zone for air as snow is highly permeable and subjected to wind. Below, gas transport occurs by molecular diffusion and gravitational fractionation within the firn. When reaching the lock-in depth (LID), gravitational fractionation stops [1], and gases enter the lock-in zone (LIZ). Along these three zones, porosity is decreasing, which leads to progressive closure and separation of pores to form individual closed
pores. At the close-off depth (COD), these pores are fully isolated from the
surface, i.e. air can no longer be pumped out from the firn.

These changes in gas transport mechanisms originate from different den-19 sification mechanisms. First, snow undergoes grain rearrangement and sin-20 tering until a density  $\rho = 550 \, \text{kg.m}^{-3}$  is reached. Further densification takes 21 place by viscoplastic deformation of grains. The order of magnitude of the 22 densification rate being typically between  $10^{-12}$  and  $10^{-9}$  s<sup>-1</sup> depending of 23 the polar sites, Maeno and Ebinuma [3] proposed that deformation proceeds 24 by a combination of dislocation creep, which is the dominant mechanism, 25 and diffusion creep (boundary or lattice diffusion depending on the temper-26 ature). The last stage of densification is porosity-related. When the density 27 exceeds about  $840 \,\mathrm{kg.m^{-3}}$ , pores are closed and further shrinking is driven by 28 the difference between their internal pressure and the larger external stress. 29

As pore closure is a very slow process, there is no direct observation of 30 this phenomenon in the field as it would take many decades to capture it. 31 However, ex situ characterizations of post mortem firm samples have been 32 performed, in particular with the use of X-ray tomography [4, 5, 6, 7]. This 33 enabled the amount of closed pores to be quantified and to relate it to gas 34 transport in the firm. The significant geometrical evolution of pores with 35 depth can also be characterized. Still, the closure process and its accom-36 panying microstructural evolution can only be assumed in ex situ observa-37 tions, while in situ observations in real time would give much more valuable 38 information on the phenomena at play. Therefore, in situ observations un-39 der laboratory conditions provide very promising tests to study densification 40

<sup>41</sup> mechanisms and pore closure within a reasonable time frame. To our best
<sup>42</sup> knowledge, no in situ compaction of firn under X-ray microtomography has
<sup>43</sup> been attempted yet. In other words, pore closure has not been followed in
<sup>44</sup> situ during mechanical tests.

Mechanical tests on firm are relatively scarce. Some laboratory experi-45 ments on dense snow and natural firm have been reported in the literature. 46 For instance, Landauer [8] performed uniaxial tests up to  $630 \,\mathrm{kg.m^{-3}}$  density 47 and recorded stresses of the order of 0.3 MPa for strain rates of  $4 \cdot 10^{-4} \text{ s}^{-1}$ . 48 Ebinuma and Maeno [9] carried out creep tests on denser laboratory snow 49 for various initial densities (above  $550 \,\mathrm{kg.m^{-3}}$ , similar to natural firn), and 50 went further than  $850 \,\mathrm{kg.m^{-3}}$ . They concluded that above a critical density, 51 which depends on the applied stress, dislocation creep governs deformation 52 [10]. Meussen et al. [11] offer a more recent and complete work on creep 53 experiments on natural firm with a large range of tested densities. Results 54 for densities of approximately  $850 \text{ kg} \cdot \text{m}^{-3}$  exhibited less than 1% of strain 55 for an eighty-hour experiment under 0.5 MPa and at -20 °C. For those large 56 densities (close to the close-off density), firm exhibited a similar behavior to 57 polycrystalline ice. In short, in these tests, applied stresses are about the 58 same order of magnitude than in the firn core (ranging from 0.1 to 0.7 MPa) 59 but those experiments were too lengthy to allow for X-ray tomography. These 60 results exemplify the range of mechanical responses of our material from the 61 extremely porous to the extremely dense firn and provide the necessary con-62 ditions for displacement-controlled tests. Indeed, decreasing by a factor 10 63 the time of the experiment will lead to significantly larger stresses. Disloca-64 tion creep is also expected to be the dominant microstructural mechanism 65

<sup>66</sup> under such conditions.

Rolland Du Roscoat et al. [12] were the first to perform in situ X-ray 67 diffraction contrast tomography on snow samples under slow creep, allow-68 ing grains to be differentiated from one another (with stresses lower than 69 0.14 MPa and initial densities lower than  $500 \text{ kg} \cdot \text{m}^{-3}$ . These authors fol-70 lowed the intragranular deformation of snow grains as small intragranular 71 disorientations were observed. These results emphasized that the anisotropic 72 viscoplasticity of ice crystals should be considered even for snow. Other com-73 paction tests on snow also revealed that deformation is mainly controlled by 74 pressure sintering, but is also dependent on the density and on grain spe-75 cific surface area [13, 14, 15]. Still, in situ studies of snow mostly consist of 76 following time lapse metamorphism for different types of snow [16] and with 77 controlled thermal gradients or other particular conditions [17, 18]. The ul-78 timate goal is usually to compute snow effective properties [19] or model 79 snow behavior [20, 21, 22]. Moreover, curvature change is a key feature of 80 snow metamorphism and its investigation reveals underlying processes at 81 work [23, 24]. In particular, rounding and smoothing of snow grains along 82 with grain growth and specific surface area decrease, demonstrate time and 83 temperature influence [25, 26]. 84

For firn typical densities (ranging from  $550 \text{ kg.m}^{-3}$  to  $912 \text{ kg.m}^{-3}$ ), the evolution of pore curvature, and time and temperature effects should be noticeable in laboratory conditions (day-long experiment at -10 °C). In this work, we investigate the effects of time, temperature and densification-rates on pore closure, microstructure evolution, and pore curvatures for large initial densities ( $\rho > 780 \text{ kg.m}^{-3}$ ). In particular, we make an attempt to discern the effects of dislocation creep and diffusion creep by using two relevant firm
parameters: the closed porosity ratio and the connectivity index. To that
end, in situ compaction tests are performed on natural firn characterized by
X-ray tomography.

# 95 Methods

# 96 Strategy

This work aims at monitoring, in a time frame of a few hours, the densi-97 fication of polar firm by X-ray micro-tomography. This means experimental 98 strain rates ( $\approx 10^{-6} - 10^{-5} \text{s}^{-1}$ ) six or seven orders of magnitude larger than 90 during natural densification in polar ice cap, as well as warmer temperature 100 (-10°C in laboratory instead of -55°C in the East Antarctica plateau for in-101 stance). Power law creep is the dominant mechanism for such strain rates at 102 high temperature. Diffusion mechanisms and processes such as sintering or 103 grain growth are not expected to occur in an eight-hour experiment. To as-104 sess the impact of the different mechanisms on firn, experimental tests and in 105 situ inspection were carried out. Ex situ characterization was first performed 106 and published in Burr et al. [7], thus providing a reference state for the firm 107 from Dome C (East Antarctica plateau)<sup>2</sup>. The ex situ data set is composed 108 of 29 samples extracted from the same ice core at different depths, above and 109 below the close-off depth. Ex situ samples were not subjected to any treat-110 ment. In contrast, we refer to in situ samples when samples extracted from 111 the ice core are thereafter mechanically tested or thermally treated. In situ 112

<sup>&</sup>lt;sup>2</sup>Volsol ice core, grant ANR NT09-431976-VOLSOL)



Figure 1: The experimental strategy developed in this work. Ice age against density for Dome C [27] together with the proposed paths in the experimental procedure: mechanical densification under oedometric compaction at -10 °C followed by ageing (heat treatment) at -2 °C. Creep tests on firm are also performed at -2 °C.

compacted samples are artificially densified at -10 °C by means of an oedo-113 metric compaction device, while being characterized by X-ray tomography. 114 Additionally, a few samples are kept at high temperature at -2 °C for five 115 months and analyzed afterward. A third set of in situ samples are submitted 116 to creep also at -2 °C for five months. This procedure is illustrated by the 117 evolution of the ice core age with density in Fig. 1. The compaction test den-118 sifies the material, while the heat treatment at -2 °C ages it. The creep tests 119 attempt to couple stress and temperature for densification. The accompany-120 ing microstructures of these samples are compared with ex situ samples from 121 the same density, corresponding to different firn depths: 79.87 m, 89.5 m and 122 94.5 m. 123

#### 124 Compaction device

A special cold cell was developed and adapted to a compaction device to 125 perform mechanical tests on the firm. A motor controls the vertical motion 126 of a piston via gears. A load cell is mounted under the piston and a software 127 enables the displacement of the piston to be controlled in a rather large 128 range of velocities (between 0.1 and  $10\,\mu m.s^{-1}$ ). The cold cell is fixed on the 129 compaction device. It consists of a cold air input and a cover that bears the 130 load (Fig. 2). The sample is set between the piston and the cover. The cover 131 was designed to follow several specifications. It should sustain a sufficient 132 load, be thermally insulated and should moderately absorb X-rays. 133

Figure 2 shows the apparatus inside the laboratory tomograph. Cold air 134 flows through the cell at -10 °C thanks to the combination of an air dryer and 135 a cryostat. The diameter of firn samples are adjusted to the PMMA socket 136 with a lathe, in order to impose oedometric conditions before subsequent 137 treatment. These conditions are required to avoid issues that could occur for 138 such a granular material in uniaxial compression (grains may detach easily 139 during the compaction and lead to catastrophic failure). An oedometric test 140 is a compaction test, which reproduces satisfactorily the strain state imposed 141 in ice caps and soils, and which models in this work an artificial densification 142 of firn. 143

#### 144 Tested samples

Ex situ and in situ samples are 12 mm in diameter and their height is in the 20–30 mm range. All samples are characterized at -10 °C in an Easytom XL nano tomograph from RX Solutions, at an acceleration voltage of 60 kV and a target current of 166  $\mu$ A. Scans have a helical geometry enabling the



Figure 2: The apparatus inside the laboratory tomograph with the compaction device highlighted in a red ellipse. The firn sample is shown inside its PMMA socket (oedometric conditions). Cold air flows in the cell at -10 °C.

sample to be imaged on its full length. Multiple scans on the same ex situ 149 sample have barely shown any effect of the X-rays and of the air flow on the 150 pore shape nor density [28]. Voxel size is 12 µm for ex situ conditions and 151 15 µm for in situ conditions. After 3D image reconstructions with the filtered-152 back projection algorithm, all image analyses and parameter determination 153 were carried out using the free software Fiji [29] combined with the plugin 154 Analysis 3D [30]. To comply with the method proposed in Burr et al. [7], a 155 cylindrical region of interest of 9.6 mm in diameter is extracted from volume 156

<sup>157</sup> image after thresholding of the ice phase.

In situ compacted samples were tested at different displacement rates. 158 Tests were interrupted to scan samples at different densities and ensure a 159 stable microstructure. After 10 minutes of viscoplastic relaxation, a 25-160 minute scan was performed, to avoid blurred 3D images. Interruptions of 161 tests roughly lead to a 30% stress drop (see SI for a stress-time curve). Four 162 samples were successfully scanned during in situ compaction. They originate 163 from the same slice of the Dome C ice core at 79.92 m depth and are detailed 164 in Table 1. The nomenclature of samples relates to their depth in the ice 165 core. When samples  $80_3$ -comp,  $80_4$ -comp,  $80_5$ -comp and  $80_6$ -comp were in 166 oedometric conditions, the displacement rate of the piston was kept constant 167 leading to strain rates in the range  $[2 \cdot 10^{-6} \text{s}^{-1}, 1.5 \cdot 10^{-5} \text{s}^{-1}]$ . Displacement 168 rates were set so that there were twice the one of  $80_3$ -comp for  $80_4$ -comp and 169  $80_6$ -comp, and three times larger for  $80_5$ -comp. Note that strain rates above 170  $\approx 4 \cdot 10^{-5} \mathrm{s}^{-1}$  lead to micro-cracks within the firn. Care was taken to ensure 171 a correct positionning of samples before the desired speed could be applied. 172 This delicate pre-positioning of samples could last up to 5 hours, leading to 173 a total compaction time of 8 to 10 hours. Digital volume correlation on grey 174 level images indicated that radial strain reached between 1% and 2.5% in 175 the first few hours of the test, corresponding approximately to an axial strain 176  $\varepsilon \approx 4\%$  [28]. 177

After compaction, samples are denoted with the suffix -comp. The particular sample  $80_6$  is submitted to two consecutive treatments: first a compaction ( $80_6$ -comp) and then a heat treatment ( $80_6$ -comp-temp). Table 1 lists the initial designations and densities of all tested samples as well as



Figure 3: Experimental strategy throughout this work. The red path denotes heat treatment with no applied stress. The green path denotes densification under mechanical compaction. The purple path denotes densification under creep for 5 months. Samples are denominated by their initial depth in the ice core. For example, sample  $80_6$  was drilled at 80 m depth, compacted ( $80_6$ -comp) and finally submitted to heat treatment ( $80_6$ -comp-temp).

their designations and densities after a 5 months heat treatment at -2  $^{\circ}\mathrm{C}$  (de-182 noted with a suffix -temp). These temperatures and duration were chosen to 183 keep a reasonable time frame. Note that samples  $80_2$  and  $90_2$  crept during 184 heat treatment as a load was applied on them (denoted with a suffix -creep, 185 axial stress  $\approx 0.3$ , and  $\approx 0.7$  MPa, respectively), in contrast to samples  $80_1$ 186 and  $90_1$  which only aged at elevated temperature without external stress. 187 The heat treatment was performed in two hermetic boxes with large hu-188 midity content to avoid sublimation (RH74% and RH99.9%, respectively). 189 Samples inside their PMMA socket were also wrapped in plastic foils to limit 190

<sup>191</sup> sublimation. Figure 3 illustrates the various treatments applied on samples<sup>192</sup> 80 and 90.

#### 193 Microstructure

After experiments, all in situ tomographic samples are cut in half (-comp, 194 -temp and -creep), and the thin section of each half is analyzed (typically 195  $\approx 25 \mu m$ ). Microstructures and grain orientations are revealed thanks to po-196 larized light using an Automatic Ice Texture Analyzer (AITA) [31]. Mean 197 grain size is then determined for each thin section by averaging the surface 198 area of all grains (using Fiji [29]). For ex situ samples, slices of ice core 199 retrieved at the same depth than in situ samples are analyzed with thin sec-200 tions of  $20x20 \text{ mm}^2$ . Therefore the mean grain size of ex situ samples is more 201 statistically representative than for in situ tomographic samples (algorithm 202 for image analyses and evolution with depth of the ex situ microstructure 203 and  $\mathbf{c}$  axis orientations are given in Burr [28]). 204

Ex situ samples			In situ samples					
Initial name	Depth (m)	$ ho_k$ (kg.m <sup>-3</sup> )	State 1	Final strain rate $(s^{-1})$	Final stress or dead load (MPa)	$ \begin{vmatrix} \rho_1 \\ (\text{kg.m}^{-3}) \end{vmatrix} $	State 2	$ ho_2$ (kg.m <sup>-3</sup> )
$95_{0}$	94.5	829.7						
$80_{1}$	79.87	786.3	$80_1$ -temp		0	784.8		
$80_{2}$	79.92	794.1	$80_2$ -creep		$\approx 0.3$	810.3		
803	79.92	797.8	80 <sub>3</sub> -comp	$3.4\cdot 10^{-6}$	$\approx 3$	825.1		
$80_{4}$	79.92	785.0	$80_4$ -comp	$7.6\cdot 10^{-6}$	$\approx 4.8$	867.8		
$80_{5}$	79.92	784.8	$80_5$ -comp	$9.3 \cdot 10^{-6}$	$\approx 5$	852.2		
806	79.92	-	80 <sub>6</sub> -comp	$\approx 4 \cdot 10^{-6}$	$\approx 5.7$	828.3	$80_6$ -comp-temp	825.6
$90_{1}$	89.5	822.3	$90_1$ -temp		0	823.6		
$90_{2}$	89.5	821.0	$90_2$ -creep		$\approx 0.7$	829.5		

Table 1: Nomenclature of samples. Ex situ samples are reference states, used as initial states of in situ samples (before any treatment). In situ samples are samples submitted to a treatment. Denotations -comp, -temp and -creep refer respectively to in situ compacted samples at -10 °C, in situ heat treated samples at -2 °C and samples that crept at -2 °C. Sample 80<sub>6</sub> is first compacted at -10 °C and then heat treated at -2 °C, hence is denoted -comp-temp. Samples  $80_{3,4,5}$  are scanned regularly throughout compaction and state 1 refers to their final strain step. Though strain rates and stresses evolve during compaction tests, only the final ones are shown (see supplementary file for curves). Samples come from a slice of Dome C ice core from depths in the range 80 to 95 meters.

## 205 Curvature

Curvature is supposed to be the driving force of snow metamorphism 206 (surface energy minimization). On 3D images, it is usually described with 207 the mean and Gaussian curvatures [32, 23]. Here we take advantage of two 208 other measures of local shape introduced by Koenderink and Doorn [33]. The 209 shape index is a measure of the local shape that also differentiates convexities 210 from concavities. It ranges from -1 to +1. The curvedness is a positive value 211 that quantifies the amount of curvature of the shape. It differentiates well 212 flat surfaces from acute shapes for instance. These measures of shape were 213 successfully used in the metallurgy community, to quantify dendritic growth 214 [34] or to compare dendrites depending on the solidification method used 215 [35]. 216

In the following, curvatures are investigated in the porosity reference framework, such that the surface of an entrapped spherical bubble is considered convex. This is important as we adopt the convention  $\kappa_1 > \kappa_2$  for principal curvatures. The shape index S writes :

$$S = \frac{2}{\pi} \arctan\left(\frac{\kappa_1 + \kappa_2}{\kappa_1 - \kappa_2}\right) \tag{1}$$

Regarding the curvedness C, it is equal to 0 for a perfectly planar surface (infinite curvature radius), and increases as the curvature radius decreases. Its formulation is:

$$C = \sqrt{\frac{\kappa_1^2 + \kappa_2^2}{2}} \tag{2}$$

Local principal curvatures are computed with the software Avizo [36]. The generation of the surface mesh implies the choice of a mesh size. It is achieved by selecting a smoothing parameter in Avizo (set to 3). For the

actual calculation of curvatures, the user also has to determine how many 227 neighbors are taken into account for the local curvature. Using the ten closest 228 neighbors for local curvatures leads to about 10 to 20 minutes of computation 229 time (Burr [28] for more details). The computation of the curvedness and 230 the shape index gives a 2D histogram which can be interpreted with a scaled 231 probability density P(S,C). In the following, we chose a normalization such 232 that  $P(S_{max}, C_{max}) = 1$  which means that the local surface characterized by 233 the corresponding values  $S_{max}$  and  $C_{max}$  is the most abundant in the sample. 234

#### 235 Initial state - ex situ samples

Figure 4 shows microstructural information for samples from 80 and 90 m 236 depths at their initial state. Figure 4 is used as a reference state for com-237 parison with in situ samples. The classification of shapes depending on the 238 value of S in the range [-1; 1] is also exemplified as layouts in Fig. 4. For 239 instance, a perfectly spherical pore (cap-like shape) would have a shape index 240 equal to 1, and a perfectly cylindrical pore (ridge-like shape) would have a 241 shape index equal to 0.5. Figure 4b1 shows more closed pores (illustrated in 242 red) than Fig. 4a1 as the sample was taken 10 meters deeper. In both cases, 243 pores are usually rounded with smooth surfaces (Fig. 4a1-b1), as depicted by 244 S-C maps in Fig. 4a2-b2. Indeed, their shape span through the saddle-ridge 245 and the dome-like shape (between a ridge and a cap). Looking at the most 246 abundant local surface in the sample, for which  $P(S_{max}, C_{max})=1$ , one can 247 also distinguish the small shift of  $S_{max}$  towards increasing S between samples 248 coming from 80 meters ( $S_{max} < 0.5$ ) and 90 meters depth ( $S_{max} > 0.5$ ). Vari-249 ation of the curvedness is relatively low, with the same horizontal shift of the 250 maximum curvedness than for  $S_{max}$ . Grains are quite equiaxed and depth 251

clearly impacts grain size: 90 m deep samples exhibit larger grains than 80 m deep samples (left hand side of Fig. 4a3-b3). In fact, firn grain size increases monotonously with depth in Dome C firn [28, 7]. Finally, orientations of c axes towards the ice core direction (loading direction) are already discernible for all samples from 80 m and 90 m (see right hand side of Fig. 4a3-b3).

# 257 **Results**

# <sup>258</sup> Closed porosity ratio and connectivity index

Pore closure evolution has been classically investigated by using the closed 259 porosity ratio [38, 39, 40, 41]. Recently, we have introduced a parameter 260 defined as the connectivity index, which was found more reliable than the 261 closed porosity ratio to predict the close-off depth and density [7]. The 262 connectivity index is defined by the ratio of the volume of the largest pore to 263 the total pore volume of a sample, and its quantification is less dependent on 264 the analyzed volume, heterogeneities and voxel size than the closed porosity 265 ratio. The closed porosity ratio is defined by the ratio of the closed pore 266 volume to the total pore volume. They both work reversely and vary between 267 0 and 100%, but the closed porosity ratio is fully dependent of the status 268 of pores (close or open), which is sensitive for the small volumes typically 269 obtained with X-ray tomography in this work. Its quantification is carried 270 out at the scale of the sample and depends on the cut-pore ratio. It does not 271 give values that can be used to characterize a firm layer. That being said, 272 both parameters are calculated and discussed here. 273

Figure 5a shows the evolution of the closed porosity ratio with density and Figure 5b shows the connectivity index for both types of samples. Ex



Figure 4: Typical initial microstructures and references for samples (a)  $80_k$  and (b)  $90_k$ (with k ranging from 1 to 6 for samples  $80_k$  and being 1 or 2 for samples  $90_k$ ). Layouts are shown as reading guides. Visualization of open pores (white) and closed pores (red) in a subvolume (a1),(b1) (side view  $6.72x6.72mm^2$ ). (a2),(b2) Curvature maps. (a3),(b3) Left hand side : thin sections observed by polarized light illustrating grain size. Right hand side : **c** axis reconstruction of the same thin section using the AITA python toolbox [37] (pores and reconstructed pixels with low quality are in white).



Figure 5: Evolution of the closed porosity ratio (a) and of the connectivity index (b) with density for ex situ and in situ samples. Blue dots depict ex situ samples [7], and green crosses, diamonds and squares depict all strain steps of in situ compacted samples  $80_{3,4,5}$ comp. Blue and green dashed lines serves only to guide the eyes. All other in situ samples are displayed by arrows that link their initial states to their final states. Different steps of compaction of sample  $80_4$ -comp are also visualized with open pores in white and closed pores in red (with  $\rho_A = 785.0 \text{ kg.m}^{-3}$ ,  $\rho_B = 839.8 \text{ kg.m}^{-3}$  and  $\rho_C = 867.8 \text{ kg.m}^{-3}$ ). See Fig. 3 and Table 1 for sample color and nomenclature, and supplementary for full in situ visualization of sample  $80_4$ -comp.

situ points from Burr et al. [7] and their trend have been added in blue, while 276 in situ compacted samples are displayed in green. The red and purple colors 277 correspond respectively to in situ -temp and -creep samples along with their 278 initial states. All points linked by an arrow are the same extracted subvol-279 umes at different states. However, to avoid superimposing arrows, samples 280  $80_{3,4,5}\text{-}\mathrm{comp}$  and their intermediary steps are displayed by green crosses, di-281 amonds and squares, respectively. When required for analyses, error bars 282 have been added. These were calculated by investigating the effects of image 283 processing, voxel size, sample size and pixel connectivities on the studied 284 parameters [7]. Relative errors are 0.15% for  $\rho$ , 3.1% for the closed porosity 285 ratio and 0.11% for the connectivity index. In situ points have been obtained 286 with very different strain rates, but interestingly they all fall on a master 287 curve (displayed by a green dashed line) for the closed porosity ratio and the 288 connectivity index. For any given density, the fraction of closed porosity is 289 larger for ex situ samples. For instance, at  $830 \,\mathrm{kg.m^{-3}}$ , approximately 24.4%290 of pores are closed for ex situ samples, whereas for in situ compacted samples 291 there are only 4.5% of closed pores. Accordingly, the connectivity index is 292 the largest for in situ compacted samples. In other words, pores are closing 293 at larger densities for in situ compacted samples than for ex situ samples. 294 Tomographic images Fig. 5 illustrate this pore closure process. Note that 295 the minimum value obtained for the connectivity index is 9.3% with a den-296 sity value of  $867.8 \text{ kg.m}^{-3}$  (sample  $80_4$ -comp was the most compacted one 297 at the end of the experiment). We have shown that the close-off depth and 298 its corresponding density could be predicted from the abrupt drop and the 299 final plateau of the connectivity index curve (see Burr et al. [7] for details 300

on the determination procedure). Here, the final plateau is not reached at 301  $840 \text{ kg} \cdot \text{m}^{-3}$  for in situ samples, meaning that percolation is still possible after 302 compaction up to at least  $870 \text{ kg} \cdot \text{m}^{-3}$ . This means that using density as the 303 only indicator of the close-off is not correct. Microstructure and densification 304 history also play an important role in determining close-off. It is likely this 305 microstructure effect is partly taken into account when using the temperature 306 of a polar site as a determination parameter for the air volume entrapped in 307 closed pores (e.g. Martinerie et al. [42]). 308

Figure 5 also displays the closure of pores after heat treatment (red ar-309 rows). In fact, a green arrow representing compaction, links  $80_6$  to  $80_6$ -comp, 310 illustrating a density increase without closing much pores. But, from  $80_6$ -311 comp to  $80_6$ -comp-temp, the closed porosity ratio increases from 3.3 to 6.7 % 312 while the density only slightly decreases. In contrast to the closed poros-313 ity ratio, the connectivity index drops from 74 to 63%. Depending on the 314 conditions they were submitted to, samples exhibit different trends. Indeed, 315 the density stayed constant within error bars between  $80_1$  and  $80_1$ -temp, or 316 between  $90_1$  and  $90_1$ -temp, but an increase of the closed porosity ratio is only 317 noticed between  $90_1$  and  $90_1$ -temp. Creep tests on samples  $80_2$  and  $90_2$  both 318 exhibit an increase in density. However, only the creep test on the deepest 319 sample do close some pores ( $90_2$ -creep), while they do not for sample  $80_2$ -320 creep. Consistently, the connectivity index decreases for sample  $90_2$ -creep 321 and stagnates for sample  $80_2$ -creep. Pore size being rather large for sample 322  $80_1$ , the 5 months heat treatment was not sufficient to have pore channels dis-323 connecting from one another, compared to sample  $90_1$ . In brief, compaction 324 tests lead to an increase in density without disconnecting much pores, while 325

<sup>326</sup> increasing temperature tends to separate pores for a given density.

Visual inspection of 3D views (see Fig. 4-5-6) gives valuable qualitative information on the porous structure. However, in the following section, a more precise evaluation of the morphological differences is realized by quantifying these changes in terms of curvatures of pores.

## 331 Curvature

Figure 6 shows a close-up view within the porosity networks for different 332 samples of approximately the same density ( $\rho = 828 \pm 2 \,\mathrm{kg.m^{-3}}$ ), but re-333 sulting from different process histories. The separation of globular portion 334 of pores is highlighted in blue for samples  $80_6$ -comp and  $80_6$ -comp-temp. 335 Blue ellipses point to sharp dead ends in Fig. 6b1 that appear smaller and 336 rounder in Fig. 6c1. Green ellipses highlight thin channels (Fig. 6b1) that 337 have disappeared after heat treatment (Fig. 6c1). The accentuated curvature 338 after compaction test is illustrated by zoom-ins on the pore network. Sam-339 ples  $95_0$  (Fig. 6a2) and  $90_2$ -creep (Fig. 6d2) do not exhibit marked curved 340 pores, whereas sample  $80_6$ -comp (Fig. 6b2) exhibits sharp pore shapes, char-341 acterized by a larger curvedness on pore edges. After heat treatment, sam-342 ple  $80_6$ -comp-temp exhibit smoother edges as emphasized by red ellipses in 343 Fig. 6c2. The curvedness clearly shows that the sharp shapes and pointy 344 dead-ends are starting to round. Concavities also seem to be reduced and 345 some pore channels are disappearing, thus indicating pore closure during heat 346 treatment. 347

This morphological evolution of pores is confirmed quantitatively by curvature maps in Figs. 6a3 to d3. As observed on 3D images, the ex situ sample  $95_0$  depth is very similar to  $90_2$ -creep, which was crept at -2 °C during five



Figure 6: Comparison of pore morphologies (rows 1-2) and local curvatures (rows 2-3) for samples of approximately equal density ( $\rho = 828 \pm 2 \text{ kg.m}^{-3}$ ) but resulting from different process histories. (a) 95<sub>0</sub>, (b) 80<sub>6</sub>-comp, (c) 80<sub>6</sub>-comp-temp and (d) 90<sub>2</sub>-creep. Blue and green ellipses highlight pore closure processes. Red ellipses indicate the smoothing of pores after heat treatment. White ellipses emphasize isolines around  $P(S,C) \approx 0.5$ . See Table 1 for sample nomenclature.

<sup>351</sup> months. Most of the local surfaces have shapes between the saddle-ridge and <sup>352</sup> the dome, with low curvedness. In comparison to Fig. 4a2 and Fig. 4b2, <sup>353</sup> the distributions of local shapes of those samples (95<sub>0</sub>, 80<sub>6</sub>-comp, 80<sub>6</sub>-comp-<sup>354</sup> temp and 90<sub>2</sub>-creep) are slightly shifted towards the cap-like shape (S = 1) <sup>355</sup> as samples come from a deeper slice (94.5 m instead of 79.92 m) or are denser <sup>356</sup> (829 kg.m<sup>-3</sup> instead of 785 kg.m<sup>-3</sup>).

In contrast to Fig. 4a2-b2, Fig. 6b3 displays a larger distribution of local 357 pore shapes after compaction. In addition, this curvature map  $(80_6$ -comp, 358 Fig. 6b3) looks very similar to all other in situ compacted samples  $(80_3$ -comp, 359  $80_4$ -comp and  $80_5$ -comp). Negative values are much more prevalent than for 360 Figs. 6a3  $(95_0)$  and 6d3  $(90_2$ -creep), indicating a non-negligible proportion 361 of concavities. This is due to the flattening of pores during the compaction. 362 Similarly to Fig. 6c3 ( $80_6$ -comp-temp), the dominant local shape of Fig. 6b3363 (80<sub>6</sub>-comp) is the cylinder as  $S_{max}$  is centered around S = 0.5. The distri-364 bution is more widespread for  $80_6$ -comp than for  $80_6$ -comp-temp, but the 365 most striking difference comes from the curvedness. The maximum curved-366 ness value of the isoline P(S,C)=0.5 is roughly 14% larger after compaction 367 than after heat treatment, as pointed out by white ellipses on Fig. 6. This 368 is a quantitative assessment of the evolution of the acute shapes observed 369 in Fig. 6b2 ( $80_6$ -comp) towards the smoother shapes of pores observed in 370 Fig. 6c2 (80<sub>6</sub>-comp-temp). 371

372 Grain size

Compaction tests also impacts ice grains according to microstructural analyses. In fact, Fig. 7 focuses on four microstructures representing different firn states (ex situ reference state, in situ -comp, -comp-temp and -creep).

Note that we have access to the grain size only at the end of the in situ 376 experimental tests. The ex situ sample  $95_0$  exhibits a larger mean grain 377 size than the in situ compacted sample  $80_5$ -comp (see Fig. 7a1 and 7b1). 378 Overall, the mean grain size of an ex situ sample is always larger compared 379 to the mean grain size of an in situ compacted sample with the same or larger 380 density. This is simply because of the duration of the compaction experiment, 381 which is too short to notice any grain growth. The mean grain size of sample 382  $80_3$ -comp is similar to its value before compaction (if compared to Fig. 4a3, 383  $\langle \bar{A} \rangle = 0.82 \,\mathrm{mm^2}$ ), but the final strain was small ( $\varepsilon = 5.90 \,\%$ ). It has a lower 384 value around  $0.40 \,\mathrm{mm^2}$  for samples  $80_4$ -comp and  $80_5$ -comp (with standard 385 deviations equal to the mean grain size for all samples). The decrease in 386 mean grain size for samples  $80_4$ -comp and  $80_5$ -comp could be consistent with 387 dynamic recrystallization which can be expected for the temperature and 388 strain-rate conditions imposed here [43]. Dynamic recrystallization would 389 also explain the many small grains observed in Fig. 7b1. Moreover, features 390 of dynamic recrystallization have been shown even for ex situ samples [28], 391 and more generally in some firm cores [44]. However, the size of the thin 392 sections (L = 18 mm, l = 8 mm) for these in situ samples was small with 393 less than a hundred grains, possibly affecting representativeness. In any case, 394 grain growth is excluded in these eight-hour experiments. The study of the 395 **c** axis orientations in the Dome C firm (ex situ samples) shows the initial 396 development of fabric with the  $\mathbf{c}$  axes becoming aligned with the vertical -397 core axis (see full fabric analysis for ex situ samples in Burr [28]). In the 398 case of the compaction test  $80_5$ -comp (Fig. 7b2), the fabric is a little more 399 pronounced than the ex-situ sample  $95_0$  (Fig. 7a2). This is coherent with a 400



Figure 7: Comparison of the microstructures and  $\mathbf{c}$  axis orientations of (a1)-(a2) 95<sub>0</sub>, (b1)-(b2) 80<sub>5</sub>-comp, (c1)-(c2) 80<sub>6</sub>-comp-temp and (d1)-(d2) 90<sub>2</sub>-creep.

<sup>401</sup> preferential mechanism of dislocation creep during in situ compaction test. <sup>402</sup> However, asserting sample texture is not appropriate when only a few grains <sup>403</sup> are observable as for the 90<sub>2</sub>-creep sample for instance (Fig. 7d1). In any <sup>404</sup> case, quantifying the rotation of **c** axes and the strength of the fabric is <sup>405</sup> beyond the scope of this work.

<sup>406</sup> Despite the rather high storage temperature at -2 °C, no noticeable dif-<sup>407</sup> ferences were observed when comparing samples 80<sub>1</sub>-temp and 90<sub>1</sub>-temp to

their reference states Fig. 4a3 and Fig. 4b3 respectively. Thus, there are 408 not shown. However samples  $80_2$ -creep and  $90_2$ -creep (Fig. 7d1) that were 409 crept at  $-2 \,^{\circ}$ C for five months, have a larger mean grain size than initially 410 (compare Fig. 7d with Fig. 4b3), but also larger than those only heat treated 411  $(80_1$ -temp and  $90_1$ -temp) or than ex situ sample of the same density  $95_0$ 412 (compare Fig. 7d1 and Fig. 7a1). As for sample 80<sub>6</sub>-comp-temp (Fig. 7c1), 413 features of recrystallization are hardly noticeable, and its grain size is similar 414 to the one shown in Fig. 4a3. It is difficult to evaluate in which proportion 415 dynamic recrystallization occurred. 416

# 417 Discussion

In situ characterization revealed large differences with ex situ results both 418 for pore morphology and for pore closure. In particular, during in situ com-419 paction tests, rate dependency was observed, with stresses about one order 420 of magnitude larger than in firn cores (from which the samples are taken). 421 The strain rates used in these tests are also large compared to polar firm 422 densification rates (about 6 orders of magnitude larger). The durations in-423 volved during in situ tests are much too short for slow thermally activated 424 processes to occur. The eight-hour in situ experiment represents about a 425 thousand years of densification in Dome C for the same reached density. It is 426 not possible to reproduce grain growth and sintering during this short time. 427 The differences in closed porosity ratio and connectivity index observed be-428 tween ex situ and in situ compacted samples come from this rate dependency. 429 Thus, our tests decoupled the effect of viscoplastic deformations but thereby 430 excludes also potential dynamic feedbacks between diffusion processes and 431

dislocation glide. According to the sharpening of pore morphology, texture observations and strain rates, dislocation creep was the dominant mechanism during compaction. It has been previously shown that such mechanism occurs essentially by basal glide for ice and is strongly anisotropic (see Burr et al. [45], Chauve et al. [46] for recent updates on that subject).

To counteract these differences, some in situ samples were kept at -2 °C. 437 Figures 5 and 6 prove that high temperature and time facilitate pore closure 438 and pore rounding. But, closure also depends on the initial microstruc-439 ture. Indeed, between samples  $90_1$  and  $90_1$ -temp, curvature variations are 440 small. These are much less important than for samples  $80_6$ -comp and  $80_6$ -441 comp-temp for instance. Therefore, surface minimization (defined by the 442 difference between the mean curvature and the local curvature), which is 443 one of the driving forces for pore closure is different for these samples. De-444 spite this difference, closure occurs for both  $90_1$ -temp and  $80_6$ -comp-temp 445 samples (Fig. 5). We have shown that closure was heterogeneous between 446 approximately 87 m and 98 m for ex situ firm due to the presence of small 447 pore channels [7]. This means that closure is particularly accelerated in a 448 certain range of depths, namely for a particular type of microstructure. More 449 samples should be tested to reach a firmer conclusion on this topic. 450

Sample 80<sub>6</sub>-comp-temp (Fig. 7c1), which was compacted and then heat treated, exhibits a grain morphology surprisingly close to  $80_1$ -temp (only heat treated), thus close to its reference state Fig. 4a3. However, in agreement with the anisotropic plasticity, the **c** axes of  $80_6$ -comp-temp are still more aligned with the loading direction (Fig. 7c2). For all compacted samples ( $80_3$ -comp,  $80_4$ -comp,  $80_5$ -comp and  $80_6$ -comp-temp), no clear sign of <sup>457</sup> recrystallization is noticed.

Indeed, in situ compaction experiments are rather fast  $(>10^{-6} \text{ s}^{-1})$  with 458 large stresses (from 1 MPa to 5 MPa, see SI) while the creep experiment 459 is closer to what Chauve et al. [47] carried out, with stresses smaller than 460 1 MPa. However, crystal nucleation along grain boundaries such as necklaces 461 of grains is not observed for any of our experiments contrary to what these 462 authors observed in columnar ice [47]. Capolo [48] also observed drastic 463 recrystallization after a few hours of creep under  $\approx 0.3 \,\mathrm{MPa}$  stress on ice 464 tricrystal. Serrated grain boundaries (evidence of grain boundary migration) 465 are also not noticed in contrast to observations from Montagnat et al. [49] or 466 Capolo [48]. However resolution of our images may hinder close inspection. 467 Examination of thin sections reveals that there are less small grains for sample 468  $80_6$ -comp-temp than for ex situ and in situ compacted samples. This could 469 mean an effective annealing of the sample excluding nucleation or rotation 470 recrystallization. Nevertheless, we would have expected grains to grow in 471 that case and we do not see a particularly marked increase in grain size. 472 Again, the small number of observable grains calls for caution. 473

Samples 80<sub>2</sub>-creep and 90<sub>2</sub>-creep reached vertical strains of  $\approx 6.6\%$  and 474 11.9%, respectively. Therefore secondary creep should be reached, and dy-475 namic recrystallization should take place [49, 50]. In addition, as the grain 476 size is larger than initially, it is likely that high temperature also activated 477 diffusion mechanism during creep. Meanwhile, according to the very few 478 grains observed, the c axes of these samples do show a quite strong girdle 479 textures (stronger for  $90_2$ -creep shown in Fig. 7d2 than for  $80_2$ -creep). This, 480 combined with the microstructure of Fig. 7d1 reminds of grain boundary mi-481

gration mechanism, which happens in the deepest part of ice cores when large grains interlock as dynamic recrystallization proceeds [51, 52, chap. 6]. Highresolution images and precise measurements of the orientation with the help of EBSD should enhance our comprehension of the underlying mechanisms [49, 47].

# 487 Conclusion

In this paper, we presented three different experiments in order to discern 488 the mechanisms of densification of polar firn. In situ oedometric compaction 489 tests of Dome C firn were carried out under X-ray at -10 °C during a few 490 hours. It was possible to characterize the evolution of pore curvature dur-491 ing densification and to follow the closure of pores. The connectivity index 492 appears to be a good indicator of pore closure phenomena as it is not depen-493 dent of cut pores at the sample boundary as is the closed porosity ratio [7]. 494 Since, by construction, lab experiments will never be able to reproduce the 495 evolution on climatological time scales, it is important to exploit the rate-496 temperature equivalence as much as possible for aiming at it. Therefore, 497 the second experiment consisted in studying the effect of temperature on firm 498 through heat treatment (ageing at -2 °C) during a few months. The third one 499 involved creep tests, i.e. compaction of firm performed at high temperature 500  $(-2 \,^{\circ}C)$  during 5 months to combine the two previous experiments. 501

It is reported here that in situ compaction at high strain rates (between  $2 \cdot 10^{-6} \text{ s}^{-1} < \dot{\varepsilon}_z < 1.5 \cdot 10^{-5} \text{ s}^{-1}$ , larger than natural strain rates in Antartica) results in densification with a reduced volume of closed pores compared to the natural conditions of the ice cap for the same firn density. In addition, the connectivity index decreases, but at higher densities than for the ex situ profile. On the contrary, heat treatments lead to pore closure without significantly changing the density. Creep tests allow to combine at the same time the densification and the closure of pores.

These experiments highlight that densification of firm in ice cap activates both physical mechanisms: it is an interplay between dislocation glides and diffusion processes. The dislocation glides lead to intragranular deformation with  $\mathbf{c}$  axis orientation evolution (the ice viscoplastic behavior is strongly anisotropic) and to an increase in density.

The second mechanism leads to atomic diffusion, especially on the sur-515 face, which makes it possible, for example, to smooth pores by minimizing 516 curvature and increasing the size of the grains. This last mechanism closes 517 also some pores but it does not seem to change density in the range investi-518 gated in our experiments. Therefore, microstructure evolution of polar firm 519 should depend on the history of each of the parameters influencing the local 520 flow of ice cap. In return, the microstructure and not only the density should 521 ultimately influence the capture of gases. In conclusion, we believe that the 522 anisotropy of ice viscoplasticity should be considered in polar firm densifi-523 cation models as it would better represent ice crystal deformation than the 524 common use of the isotropic power law of ice [e.g., 53], and may ultimately 525 give different past climate reconstructions. It is also noteworthy that pore 526 closure could be modeled by curvature minimization given appropriate diffu-527 sion coefficient. The physical model proposed by Flin et al. [24, 54] for snow 528 metamorphism could be adapted for firn. This would help understand coa-529 lescence or separation of pores. It would also offer an alternative approach to 530

the classical use of empirical parameterizations of the closed-porosity ratio when modeling gas transport in firm [55].

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