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### 3D-SEM height maps series to monitor materials corrosion and dissolution

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ARTICLEINFO	A B S T R A C T
<i>Keywords:</i> Dissolution Corrosion 3D reconstruction AFM Microscopy SEM	An accurate method for the characterization of the topographic evolution of materials during dissolution at the submicrometric scale is reported. This method is based on the recording of Environmental SEM tilted images of a selected zone at the surface of a sample for various dissolution times. After each observation, a 3D surface reconstruction was made leading to a series of height maps of the selected zone of interest. One 3D reconstruction was compared to an AFM image of the same zone in order to estimate the accuracy of the heights determined using stereoscopic images. The maximum achievable resolution along the z-axis on the 3D reconstructions was found to be 38 nm. Several micro-structural parameters were extracted from the 3D-image series and their evolutions were related to the dissolution process. This method offers new insights for the monitoring of the changes occurring at the surface of a sample at the micrometer scale during dissolution or corrosion of the material.

#### 1. Introduction

It is of strong interest to study and characterize the dissolution and/or corrosion mechanisms at the micrometer scale to evaluate materials' ageing or recycling. On the one hand, the materials durability is mainly limited at room temperature by corrosion mechanisms that yield to the degradation of materials by several well-known and often complex mechanisms. On the other hand, material recycling at the end of its lifecycle generally requires dissolution step in acidic or basic media to further separate the elements of interest. Both corrosion and dissolution of materials are controlled by surface reactions that are generally studied at the macroscopic scale by evaluating the element releases in solutions for various conditions. Nevertheless, the understanding of these processes generally requires data recorded at the microscopic scale. Indeed, it is now admitted that for several kinds of materials, the main weakness during dissolution/corrosion/leaching tests is associated to grain boundaries, triple junctions or pores which can be observed at this scale. While other defaults which are present in the material (stacking faults, dislocations...) are not directly observable at this scale, their consequences in terms of behavior during dissolution/corrosion/leaching (formation of voids, pitting...) can be observed by scanning electron microscopy (SEM). Thus, as most of the events occurs at the microscopic scale, there is a real need to observe directly the corrosion/dissolution of materials at the micrometer/submicrometer scale in order to characterize precisely the modifications occurring at the solid/liquid

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interface and to determine accurately the associated mechanisms.

Only few studies have been performed in order to evidence the evolution of materials surface at the micrometer scale. In this field, recent works [1–4] reported the description of a ceramic material dissolution in aqueous media. However, if the images recorded gave new insights in the changes occurring at the surface, no precise determination of the quantity of matter dissolved was done, due to the lack of information in the z-direction.

Various methods were used to characterize the surface microstructure at the microscopic scale: Atomic Force Microscopy (AFM) [5,6], Scanning Tunneling Microscopy (STM) [7] or Vertical Scanning Interferometry (VSI) [8-10]. Each technique exhibits its own advantages and limits that will be shortly detailed hereafter. AFM images are characterized by a very good zresolution (close or higher than 1 nm) but the technique is limited to the observation of relatively smooth samples [11,12] due to tip radius effects. The apparatus can be used directly when the sample is in contact with a solution. However, it can be very difficult (and time consuming) to find (and recover) the Region of Interest (noted hereafter ROI). Moreover, the size of the ROI is always relatively small compared to the size of the sample (maximum size of a few square micrometers). Last, the time to record an image can be very long when a high resolution is required. VSI images are characterized by a sub-nanometer resolution in the z-direction but also by a limited x- and y-resolution (in the order of 1 µm) due to the light illumination of the sample [13]. The ROI is relatively easy to find as large zones can be observed and measured (few square millimeters). Last,

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STM images are very accurate but difficult to record and limited to very small ROI (few square nanometers) [7].

The use of stereoscopic images recorded with an Environmental Scanning Electron Microscope (ESEM) to reconstruct and compare height maps of the surface of a material during its dissolution has not been reported previously. However, it appears as an interesting alternative to already reported methods. Indeed, the calculation of 3D surfaces from SEM tilted image series is now a well-established method. Several commercial softwares (Mountains®, Alicona Mex) are available and usually used in research and industry. This method is also sometimes directly implemented in SEM driving systems (Zeiss, Tescan SEM). Furthermore, the continuous development of optimized calculation procedures for 3D surface reconstructions by several research groups is also an active field of research, clearly evidencing the real possibilities offered by this technique [14-18]. Indeed, the reconstruction of 3D surfaces using this technique only requires a set of 2 (or 3) tilted SEM images. The recording of these images only last for few minutes and the 3D calculation process last for 5 to 30 min (depending on the computer power as well as on the image size).

In the present work, the experimental protocol for the acquisition of stereoscopic SEM images allowing 3D reconstruction is first reported. Then, the image processing developed using height maps is detailed and the obtained results are compared with AFM images in order to validate the method. Last, different ways to use the 3D-SEM height maps series to characterize materials corrosion and dissolution are described.

#### 2. Materials and methods

#### 2.1. Materials

The material used as reference in the present study was  $UO_2$  pellet doped with 3 mol% metallic particles whose relative density was close to 95%. The microstructure of the obtained pellet consisted in  $UO_2$  grains of  $0.2-5\,\mu m$  in size, mixed with 50–500 nm diameter metallic particles randomly located in the grains and at the grain boundaries, at the surface but also in their bulk of the pellets.

The aim of the experiment was to follow the modifications of the sample surface during its dissolution in 0.1 M HNO<sub>3</sub> at 60 °C. Two particular zones of the pellet, called hereafter "Region Of Interest" (ROI) were selected. Images of these zones were recorded at different magnifications prior the dissolution test, in order to find "easily" the ROI after the dissolution steps (Fig. 1).

#### 2.2. Methods

#### 2.2.1. Dissolution of sample

The pellet was put in contact with 25 mL of 0.1 M HNO<sub>3</sub> solution at 60 °C under stirring in a PTFE dissolution reactor. At regular intervals, the sample was removed from the dissolution device, washed with water three times and finally the excess of water was removed with an absorbent paper, in order to stop the dissolution process. Then, the sample was introduced in the ESEM chamber equipped with a Peltier stage. The Peltier stage was cooled down to 2 °C prior the introduction of the sample introduction. A great caution was paid to the pumping sequence in order to avoid any dehydration of the sample. This pumping sequence consisted in 8 differential pumping steps between 500 and 750 Pa of H<sub>2</sub>O. Finally, the water vapor pressure in the chamber was adjusted to 300 Pa which corresponded to a relative humidity of 42.5%. With this experimental procedure, the sample was never dried. The use of the ESEM under wet conditions prevented the dissolution experiment from perturbations induced by the sample observation.

After recovering the region of interest and recording the image series, the sample was put again in contact with the solution. This sequence was repeated several times up to the end of the dissolution test. Thus, the full experiment corresponded to a multistep observation sequence.

#### 2.2.2. Recording of ESEM images

The ESEM used during this study was a Quanta 200 ESEM FEG (commercialized by the FEI Company). After the sample was introduced in the ESEM chamber, the images were recorded using a specific gaseous secondary electron detector [19]. The imaging conditions were defined in order to reduce the consequences of irradiation of the sample by the electron beam [20,21]. Indeed, the irradiation of the sample could have strongly modified locally the reactivity of the sample surface and finally generated unexpected local modifications of the dissolution process (i.e. increase or decrease of the local dissolution kinetics). Thus, this effect had to be avoided - or limited - as much as possible. The optimized imaging conditions (that are optimized for the Quanta 200 ESEM FEG used in this study) are the following:

o Working distance = 6 mmo  $E_0 = 6 \text{ kV}$ 

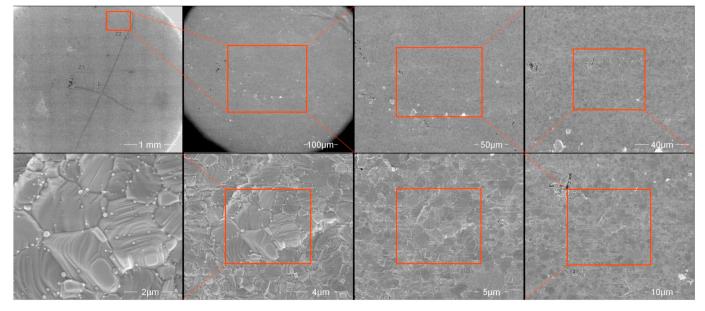


Fig. 1. Series of images recorded at different magnifications prior the dissolution test that are used to find the same Region Of Interest (ROI) from one dissolution time to the next one.

- o Spot size = 3-4
- o Diaphragm diameter =  $20\,\mu m$
- o Image acquisition conditions
- 2048 × 1887 pixels
- 1 µs per pixel
- Accumulation of 16 images with drift correction
- o Magnification = 20,000, 10,000, 5000 times (corresponding to  $\{6.35 \,\mu\text{m}; 5.48184 \,\mu\text{m}\}, \{12.7 \,\mu\text{m}; 10.9737 \,\mu\text{m}\}$  and  $\{25.4 \,\mu\text{m}; 21.9273 \,\mu\text{m}\}$  field of views respectively)
- o The images were recorded with tilt angles corresponding to  $-10^\circ, 0^\circ$  and  $10^\circ.$

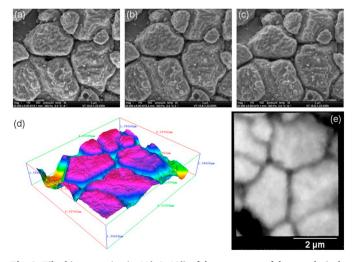
The image recording conditions resulted from a compromise between the final image quality required to reconstruct the sample surface (in terms of signal/noise ratio and limitation of the image deformation) and the irradiation of the sample surface by the electron beam (no observable sample surface modification of the ROI compared to the other parts of the sample).

#### 2.2.3. 3D-SEM surface image reconstruction and height map

The tilted image series were pre-aligned using Fiji software and the particular SIFT method [22]. The image surface reconstructions were performed using the commercial Alicona Mex software with the three images recorded at the same magnification [23] (Fig. 2a, b, c). Then the image of the sample surface was calculated, the 3D image was aligned by comparison with a reference plane which was the plane cutting the image in two parts with the same quantity of matter on the top and on the bottom sides (Fig. 2d). Thus, 16 bits images were extracted from the 3D surface reconstructions using a scale of 1 Å per pixel (corresponding to the grey scale). In these conditions, the maximum assessable scale was 6.5535 µm (Fig. 2e). Obviously, this scale must be modified and adapted by adjusting the length associated to a pixel, depending on the size of the studied sample. These images (hereafter called "height maps") contain the coordinates of each point of the sample surface in a three dimensional Cartesian system. They are the raw material that will be used to characterize the topographic changes of the surface sample as a function of the dissolution time.

#### 2.2.4. Atomic Force Microscopy (AFM)

The sample was also characterized by Atomic Force Microscopy using a MULTIMODE 8 AFM apparatus (with a lateral resolution of approximately 8 nm and an optimal z resolution of 1 Å) equipped with a Nanoscope 5 controller from Bruker Germany. The observed sample



**Fig. 2.** Tilted image series  $\{-10^\circ; 0; 10^\circ\}$  of the same zone of the sample (a, b, c) leading to the 3D surface image reconstruction using Alicona Mex software (d) and the corresponding extracted 16 bits height map (e). The features of the grains are recognizable on the 3 different representations of the surface.

was put in contact with the dissolution medium for 1064 h prior the microscopic characterization. The same Region of Interest (ROI) was also observed by ESEM before and after AFM.

AFM imaging was performed in the peak force mode, under dry conditions using SNL tips (silicon tip on silicon nitride cantilever) from Bruker Company with a spring constant of  $0.24 \,\mathrm{N}\cdot\mathrm{m}^{-1}$ . This mode has been used with specific parameters defined by the operator in order to minimize the force and to use the same tip for all the samples. Furthermore, the peak force mode has been used in order to minimize the energy that is dissipated at the sample surface during the AFM image recording with respect to their fragility (thin hydrated layers probably covers the sample). Last, a 16 bits height map with a 0.4 Å per pixel resolution was also extracted from the AFM topographic image.

#### 3. Validation of the height maps

#### 3.1. Mex height map versus AFM height map

The same region of interest (ROI) of the sample surface was observed using two types of microscopies (Fig. 3b and d) in order to compare both datasets and to validate the height values obtained with Mex by comparison with the AFM measurements.

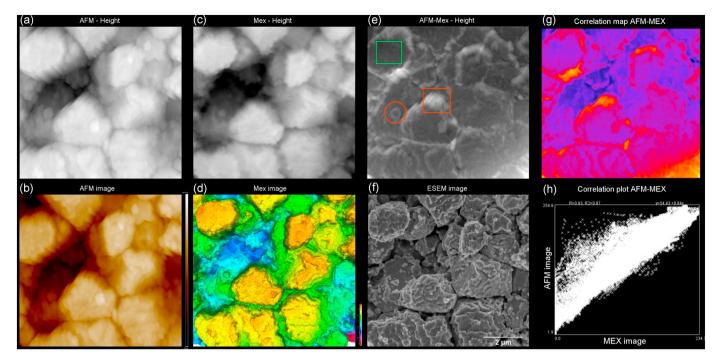
The main difference between both techniques was the time required to record an image of the ROI. Concerning AFM, one day was necessary to recover the ROI (i.e. a  $10 \,\mu\text{m} \times 10 \,\mu\text{m}$  zone on a pellet of 5 mm in diameter with marks). Then, the time necessary to record an image with the desired resolution was 1.5 h. The precision in the height values was approximately 1 nm. Regarding the surface reconstruction, finding the ROI and recording the ESEM image series required 20 min for the 3 different magnifications. Then, it took 15 min to reconstruct the topographic image of the surface using the Mex software. The precision in the determination of the height values was approximately 30–40 nm (the determination of this value will be detailed later).

The height maps extracted from the Mex reconstruction and the AFM image were thus compared. Since AFM technique provided the most accurate measurements of the surface sample heights, these values were used to determine the correctness of the heights retrieved using the Mex software. Thus, the AFM height map was used as a reference to describe the real sample surface. Both AFM and Mex height maps of the same ROI (with the same scale) are reported in Fig. 3a and c and the image showing the difference between AFM and Mex height maps is reported in Fig. 3e. A more accurate comparison of both images was obtained by plotting the correlation map (Fig. 3h) between both images and by determining the cross-correlation coefficient [24]. In this case, the figure of merit  $(R^2)$  was equal to 0.85. This value reflects the very good correlation between the two height maps but also the defects present in both images and that are inherent to each type of characterization technique. The correlation map showed the defects of the AFM image as well as those of the Mex image.

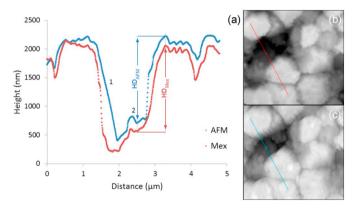
The usual artifacts were observed on AFM images obtained on samples with large topography:

- The image of the AFM tip was observed on the steep edges of the grains. This defect was highlighted in Fig. 3e by a red rectangle in the image difference between AFM and Mex height maps and referenced as Zone 1 in Fig. 4a.
- The deformation of the image arising from the high radius of curvature of the tip used relative to the geometry of the feature to be imaged (apparent increase of the sample size called tip artifact) [25]. This defect was highlighted in Fig. 3e by a red circle on the image difference between AFM and Mex height maps and referenced as Zone 2 in Fig. 4a.

Artifacts of Mex images due to the reconstruction process were also observed:



**Fig. 3.** Validation of the Alicona Mex surface reconstructions. Observation of the sample surface using several types of microscopies: (a) AFM height map; (b) AFM view of the sample; (c) Mex height map (using the same scale for the z pixel than for AFM height map); (d) 3D Mex reconstruction; (e) image of difference between AFM and Mex height maps; (f) SEM image; (g) correlation map between AFM and MEX height maps and (h) correlation plot between AFM and MEX height maps. (It must be noted that the ratio between the contrast and the brightness of the "Image of difference between AFM and Mex height maps" was enhanced in order to highlight the differences). (For interpretation of the references to color in this figure, the reader is referred to the web version of this article.)



**Fig. 4.** Height profiles of the same zone (a) extracted from the Mex height map (b) and AFM height map (c). The size of the images was  $7.73 \,\mu\text{m} \times 8.26 \,\mu\text{m}$ . The HD<sub>Mex</sub> and HD<sub>AFM</sub> values corresponded to the height differences measured at the same locations from the Mex and AFM data, respectively.

- The main one corresponded to the limit of the reconstruction resolution along the z-axis (estimated to 30–40 nm) which can be seen on the difference image on the top of the grains (shown by a red rectangle on Fig. 3e). There is much more topographic information at the nanometer scale on the AFM image than in the Mex image.
- The general deviation of the baseline (that remained limited on this example, but still visible on the correlation map Fig. 3g). It was related with a discrepancy of the reference planes between both height maps.

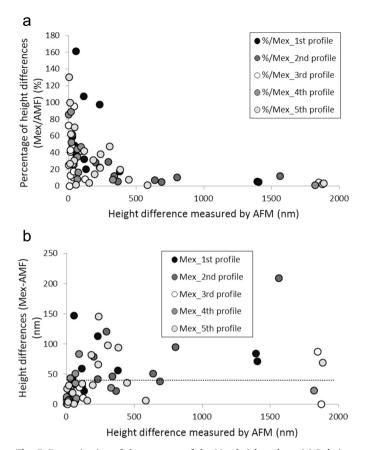
At this stage, the direct comparison between Mex and AFM height maps revealed a very good agreement between both datasets. This agreed with the conclusions of Raspanti et al. who showed the excellent visual match between Mex and AFM height maps recorded on biological materials [26]. In addition, profiles were drawn at the same location using both AFM and Mex height maps. The data and the position of the profiles on the selected ROI are reported in Fig. 4. Using these data, the accuracy of the Mex height values was determined by comparison with the AFM height values. The procedure used is described hereafter. First, the height difference between two points of the Mex profile was measured (HD<sub>Mex</sub>). Second, the height difference between the same points was measured from the AFM height profile (HD<sub>AFM</sub>) (Fig. 4a). Third, the percentage of height difference between AFM and Mex measurements was determined as follows:

$$\% HD = 100 \times ((HD_{Mex} - HD_{AFM})/HD_{AFM})$$
(1)

The %HD value is the relative difference between Mex reconstruction and AFM data.

This procedure was repeated several times on this profile in order to determine the %HD values for several HD<sub>AFM</sub> values. This set-up was also applied on four other profiles determined from the same AFM and Mex height maps. The obtained %HD values are reported as a function of the height difference determined using the AFM image, HD<sub>AFM</sub>, in Fig. 5a whereas the absolute values of the difference |HD<sub>Mex</sub> – HD<sub>AFM</sub>| are plotted versus the height difference determined using the AFM image, HD<sub>AFM</sub>, in Fig. 5b. This dataset allowed determining the accuracy of the height values measured from the Mex reconstructed image as a function of the height difference. As an example, when the height difference between two details was equal to 30 nm, the accuracy reached  $\pm$  60%. The accuracy equaled  $\pm$  20% and 5% when the height difference was 130 nm and 1500 nm, respectively. In the conditions used for the present study, the average height difference between the HD<sub>Mex</sub> and HD<sub>AFM</sub> measurements was 38 nm. This value is represented by the dashed line in Fig. 5b. It corresponds to the resolution of the Mex image in the height direction.

As a conclusion, there was a good accuracy of the relative z positions determined as well as the  $\{x,y\}$  coordinates (depending on the initial SEM image magnification). This result indicates that the image was not significantly deformed by the 3D surface reconstruction process



**Fig. 5.** Determination of the accuracy of the Mex height values. (a) Relative height difference (%HD) measured between Mex and AFM data at the same (x, y) coordinates and (b) absolute values of the height differences between Mex and AFM measurements ( $|HD_{Mex} - HD_{AFM}|$ ). The line reported in (b) represents the average value of all the height differences measured (i.e. 38 nm).

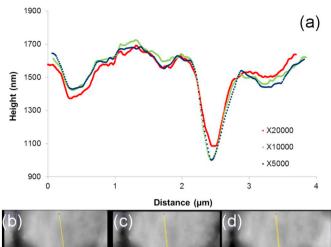
using the Mex software. This conclusion is also in good agreement with the results already reported in the literature [26].

#### 3.2. Mex height maps recorded at different magnifications

Height maps of the same region of interest were reconstructed with Mex software at several magnifications. Three profiles determined on the same zone and the corresponding images are reported in Fig. 6a. The width of the line (integration zone) was adjusted as a function of the image magnification and it always corresponded to 33 nm (Fig. 6b, c, d).

A general good agreement between the three reported profiles was observed. On the basis of Fig. 6a, the baseline of the profile obtained from the image reconstructed at 20 kX was slightly shifted compared to the profiles obtained at 5 and 10 kX. This discrepancy was attributed to the automatic determination of the reference plane by the Mex software. Indeed, the image obtained after the reconstruction process could be tilted and straightened with respect to a reference plane. In this case, the reference plane was fixed in order to set one half of the volume below the plane and the second half of the volume under the plane. As a consequence, this plane was not necessarily the same for the two images (containing the same ROI) recorded at different magnifications. Thus, this yielded to a slight shift of the baselines for the profiles reported in Fig. 6a.

Another discrepancy was observed at the position of the grain boundary. The depth of the grain boundary measured from the 5 kX and 10 kX reconstructed images was higher than that obtained from the 20 kX 3D reconstruction. It probably corresponded to the presence of a crystallite that was not always exactly positioned at the same location



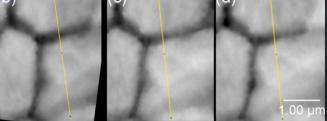


Fig. 6. (a) Height profiles determined on the same ROI using Mex 3D-reconstructions of SEM images recorded at different magnifications; (b) height map and corresponding profile reconstructed from  $\times$  20,000 magnification SEM images (line integration on 10 pixels); (c) height map and corresponding profile reconstructed from  $\times$  10,000 magnification SEM images (line integration on 5 pixels); (d) height map and corresponding profile reconstructed from  $\times$  5000 magnification SEM images (line integration on 3 pixels). NB: The width of the integration line was adjusted as a function of the magnification and corresponded to 33 nm. The size of the images reached 2.82 µm  $\times$  3.78 µm.

depending on the magnification of the SEM image series. Nevertheless, the very good agreement between the 3 sets of data can also be considered as a proof of the quality of the surface reconstructions.

#### 4. How to use the height map series?

#### 4.1. Height maps coupled with SEM images

From the 3D surface reconstructions, the surface area of the solid/ solution interface was determined precisely. Thus, this method allowed calculating this parameter in the case of materials presenting low specific surface area that were not accessible by the usual techniques such as BET method using Kr adsorption. In addition, using stack of 3D reconstructions allowed monitoring of the evolution of the surface area of the sample during dissolution.

The Abbott-Firestone curve (or Bearing Area Curve) was used to describe the surface texture of the sample (for more details, please refer to Fig. 4 reported by Loberg et al. [27]). From this curve, several parameters of interest for the kinetics of local dissolution processes were determined [28]:

- o Core void volume ( $V_{vc}$ ) expressed in cm<sup>3</sup>·m<sup>-2</sup>
- o Core material volume ( $V_{mc}$ ) expressed in cm<sup>3</sup>·m<sup>-2</sup>
- o Valley void volume (V<sub>vv</sub>) expressed in cm<sup>3</sup>·m<sup>-2</sup>
- o Height of the core material (Sk) expressed in µm

By considering that the surface of the sample was flat before the dissolution test, the value of  $V_{vc}$  directly represented the quantity of matter that was dissolved (assuming that the highest points of the reconstructed surface where included in the initial surface of the sample) (Fig. 7).

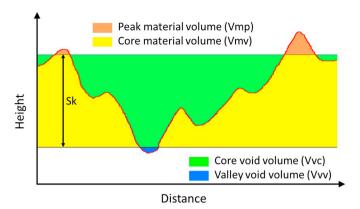


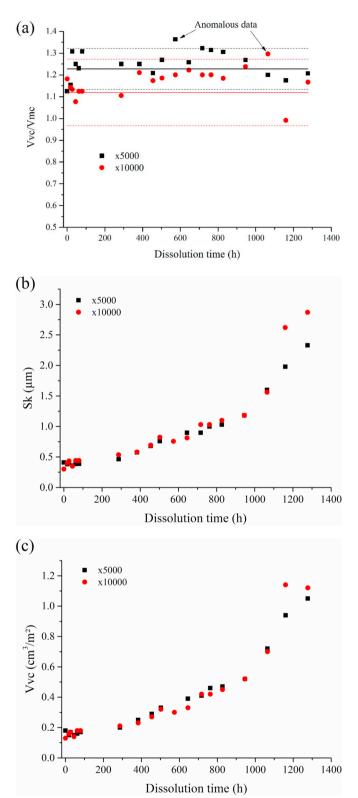
Fig. 7. Abbott-Firestone curve and associated parameters considered to characterize the sample texture.

The evolution of the parameters associated with Abbott-Firestone curve extracted from the 3D reconstructions of the solid/solution interface of the sample during dissolution is reported in Fig. 8. First, the ratio between the core void volume ( $V_{vc}$ ) and the core material volume ( $V_{mc}$ ) was plotted for the stack of 3D reconstructions in order to detect anomalous reconstructions. From Fig. 8a, it appeared that the ratio between the volume of matter and the core void volume of the reconstruction was independent of the dissolution times, but slightly dependent on the magnification (i.e. the dimensions of the reconstructed zone). The average value and associated standard deviation of the ratio was 1.23  $\pm$  0.09 and 1.12  $\pm$  0.15 at  $\times$ 5000 and  $\times$ 10,000, respectively. The anomalous data that are evidenced from Fig. 8a correspond to images where detached grains are deposited on the sample surface. The evolution of the height of the core material and the void volume of the core are plotted in Fig. 8b and c, respectively. Sk and  $V_{VC}$  were found to increase with the dissolution time. This result showed that the depth of the "valleys" dug during the dissolution of the pellet increased. This was a strong indication that the dissolution did not occur by normal retreat of the surface, which would lead to constant Sk and Vyc values.

#### 4.2. Height map series

During the study of the dissolution of materials, several reconstructions of the same ROI were obtained for various different dissolution times. As explained before, a height map was extracted for each reconstruction (where one grey level corresponds to 1 Å). Thus, these height maps were combined together into a stack. The images of the stack were aligned twice using alternatively the "Linear Stack Alignment with SIFT" [22] and the "StackReg" plugins with Fiji software [29]. One limit of this technique was the absence of invariant points during the dissolution process allowing an accurate alignment of all the images in the z direction, and thus the evaluation of the absolute dissolved volumes. However, this limit was overcome by considering one detail that remained recognizable from one image to the next one. Even if the error was not determined precisely, it was considered to remain relatively low (few nanometers to ten nanometers from one image to the next one).

The aligned stack of images offered the possibility to describe quantitatively the dissolution process at the surface of the material as a function of time, with images that were spatially resolved in the 3 directions.



**Fig. 8.** Evolution of the parameters associated to the Abbott-Firestone curve determined from the 3D reconstruction of the topography of the sample during dissolution: (a) ratio between core void volume and core material volume (detection of anomalous reconstruction); (b) height of the core material; (c) core void volume used to calculate the amount of dissolved material.

#### 4.3. Profiles in the height map series

b

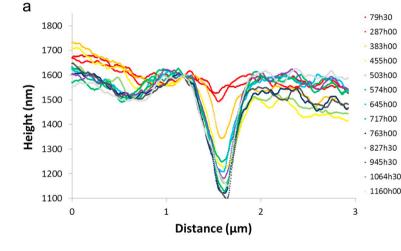
Profiles in the height map series were drawn to quantify the local height variations as a function of the dissolution time. Two profiles were extracted from the stacks of images recorded with various magnifications (Fig. 9). The profiles reported in Fig. 9a and c correspond to 20 kX and 5 kX height map image series, respectively. They were aligned in order to keep the position of one point (point I) invariant in height with the dissolution time. From the high magnification images stack (Fig. 9a), the preferential dissolution at the grain boundaries was clearly evidenced. This preferential dissolution both corresponded to the increase of the depth and of the width of the grain boundary. From the low magnification images stack (Fig. 9c), the detachment of grains during the dissolutions that were observed from one time to the next one.

The whole (or part of the) profile was integrated on a selected width of matter corresponding to the width of the line used to extract the profile. Thus local quantities of matter removed during the dissolution/ corrosion process were determined accurately. Using the data from the whole image series, the dissolution rates were evaluated at the microscopic scale.

#### 4.4. Differences of height maps

The determination of differences of height maps can be useful to locate the zones of the sample where the dissolution process was predominant. This first required to reconstruct 3D images of the same zone at two different dissolution times and then to extract height maps. The difference between two aligned height maps was calculated using the Fiji software. The result was an image highlighting the preferential dissolution/corrosion zones at the surface of the material. Reconstructed surfaces after 287 h and 645 h of dissolution are reported in Fig. 10a and c, respectively, whereas the corresponding height maps are reported in Fig. 10b and d, respectively, and the difference map is shown in Fig. 10e (3D view) and Fig. 10f. From Fig. 10f, it is clear that a preferential dissolution occurred at the grain boundaries and at the triple junctions of the ceramic. As the x-, y- and z-scales were expressed in metric units, it was possible to derive the quantity of matter that was dissolved locally between 287 h and 645 h of dissolution.

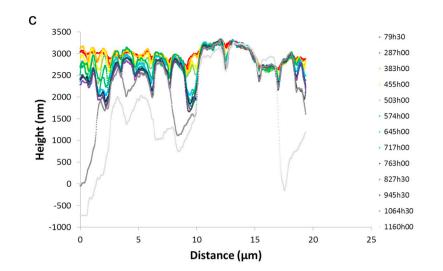
The main limit of the comparison of two images obtained by 3D reconstruction method from SEM images was the necessity to have a reference plane, or "reference points", which remained at the same location in both images. In other words, this implies that a zone of the sample remained unaffected by the dissolution process. In the studies



 79h30
 287h
 383h
 455h
 503h
 574h

 1 μm
 717h
 763h
 827h30
 945h30
 1064h30

Fig. 9. Height profiles determined on the same ROI for stack of height maps (a) and (c) recorded at (b) 20 kX and (d) 5 kX magnifications, respectively.



d

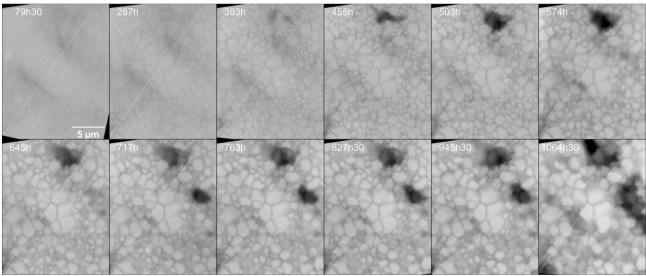


Fig. 9. (continued)

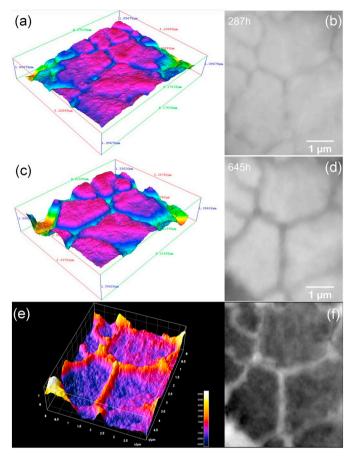
already reported, authors used a shadowing method to hinder the contact of the corrosive liquid with the sample in a specific zone. As example, platinum masks could be incorporated in the sample prior the dissolution test [30], a part of the sample could be protected with a coating [10] that could not be dissolved or a particular assembly jig could be used to create a non-wetted reference zone [9,31]. However, these techniques require to record large sized 3D images of the sample (from few hundreds of micrometers to millimeter scale) and they are impossible to implement with the reconstruction method reported herein (as the final image size only represents few square micrometers). As a consequence, when the dissolution mechanism occurred by normal retreat of the surface (i.e. general decrease of the sample thickness), no reference points were found on both images and the quantity of dissolved matter was not accurately determined. This limit was probably overcome by implementing reference points (or reference plane) in the sample that would remain unaltered during the dissolution experiment. This will be part of future technical developments.

More generally, if the dissolution process is not too fast, some unaltered zones could be found despite the dissolution progress. These invariant points could be used to align the different height maps by assigning the same absolute height (i.e. grey level) in both height maps.

#### 5. Conclusions

An accurate method for the characterization of materials dissolution at the micrometric scale was reported. This method was based on (E) SEM tilted image series of the same zone at the sample surface after several exposition times to a corrosive solution. After each observation, a 3D surface reconstruction was obtained and these topographic images were compared to AFM images in order to validate the height values obtained. The quality of the 3D reconstruction from (E)SEM images was close to the AFM image one as the figure of merit of the correlation map between both images reached 0.85.

The main advantages of this characterization technique are the high quality of the 3D surface reconstructions, the fast acquisition of the data, the possibility to observe wet samples (by using an Environmental Scanning Electron Microscope) and the limited beam effects. This technique can be applied to the observation of topographic changes occurring at the surface of numerous materials due to various mechanisms such as dissolution and corrosion, formation of coatings or corrosion layers, cracking under mechanical stress... It could be also extended to the observation of high temperature transformations such as glass-ceramic crystallization, grain sintering, grain growth, etc.



**Fig. 10.** Difference of height maps. 3D reconstruction of the solid/solution interface after 645 h (a) and 287 h (c) of dissolution; corresponding height maps (z-scale: 1 pixel = 0.1 nm) obtained using the Fiji software ((b) and (d), respectively) and 3D (e) and 2D (f) views of the difference of height maps obtained between 287 and 645 h.

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