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Impact of Anion Vacancies on the Local and Electronic Structures of Iron-based Oxyfluoride Electrodes

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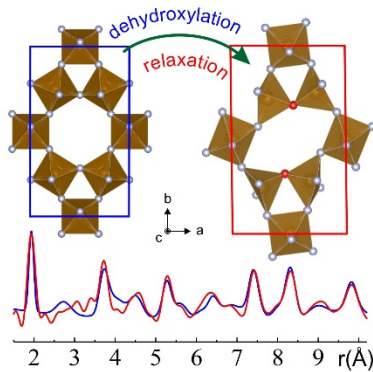
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ABSTRACT. The properties of crystalline solids can be significantly modified by deliberately introducing point defects. Understanding these effects, however, requires understanding the changes in geometry and electronic structure of the host material. Here we report the effect of forming anion vacancies, via dehydroxylation, in a hexagonal-tungsten-bronze-structured iron oxyfluoride, which has potential use as a lithium-ion battery cathode. Our combined pair-distribution function and density-functional-theory analysis indicates that oxygen vacancy formation is accompanied by a spontaneous rearrangement of fluorine anions and vacancies, producing dual pyramidal $(\text{FeF}_4)\text{-O-}(\text{FeF}_4)$ structural units containing five-fold-coordinated Fe atoms. The addition of lattice oxygen introduces new electronic states above the top of the valence band, with a corresponding reduction in the optical band gap from 4.05 eV to 2.05 eV. This band gap reduction relative to the FeF_3 parent material is correlated with a significant improvement in lithium insertion capability relative to defect-free compound.

TOC GRAPHICS



KEYWORDS. Defects, pair distribution function, density functional theory, band gap engineering.

The search for high capacity electrodes for rechargeable lithium-ion batteries has led to many materials being proposed as potential candidates. Iron fluorides contain earth-abundant iron as their redox-active species, and have a high theoretical capacity (237 mA h g^{-1} for 1 e) and a high voltage vs. Li^+/Li of 3.0 V–3.3 V.¹ Metal fluorides, however, are electronic insulators characterized by large bandgaps ($> 3 \text{ eV}$), which limits their electrochemical properties due to poor electronic conductivities. To overcome this limitation, several approaches have been proposed, including: (i) the synthesis of nanocomposites containing carbon;² (ii) the preparation of oxyfluorinated compounds with reduced band gaps, which can produce complex electrochemical behaviour;^{3,4,5} (iii) the stabilization of open-frameworks such as the hexagonal-tungsten-bronze⁶ (HTB) and pyrochlore⁷ type structures, that can be functionalized with single wall nanotubes (SWNTs)⁸; (iv) surface engineering to protect the electrode materials⁹; and (v) the introduction of anionic vacancies¹⁰ in an oxyfluoride HTB-type structure, which can be denoted $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$, with \square referring to anionic vacancies.

One proven strategy for improving the electrochemical performance of electrode materials is to deliberately introduce defects, such as vacancies.¹⁰⁻¹³ At present, a general understanding of this effect is absent. Building such an understanding requires a detailed characterization of the relevant defects, to allow the coupled changes in stoichiometry, crystal geometry, and electronic structure to be resolved. We have previously reported the stabilization of anionic vacancies in an iron oxyfluoride compound $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$, studied using x-ray diffraction analysis and Mössbauer spectroscopy.¹⁰ We previously reported that the substitution of fluoride by oxide anions is accompanied by the formation of anionic vacancies, producing undercoordinated iron atoms. The effect of fluorine substitution by oxygen and accompanying anion vacancy formation on the local structure has yet to be fully assessed. Here, we describe a combined experimental and computational study, performed to resolve the changes in crystal geometry and electronic structure induced by concomitant oxygen substitution and anion vacancy formation.

The vacancy-containing iron oxyfluoride compound $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ was prepared by a two-step thermal decomposition process of $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$: a compound that can be produced from spent liquor from stainless steel production.¹⁴ Using a self-generated atmosphere, $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$ was decomposed into a HTB-structured hydroxyfluoride compound $\text{FeF}_{3-x}(\text{OH})_x \cdot n\text{H}_2\text{O}$ ($n \leq 0.33$), denoted $\text{FeF}_{3-x}(\text{OH})_x$ hereafter.¹⁵ This sample was heat treated at 250 °C for 12 h, and 350 °C for 1 h to form $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ with $x \sim 0.4$. X-ray diffraction confirmed the phase purity of the vacancy-containing HTB structure (**Figure S1**). The presence of anionic vacancies was ascertained by ⁵⁷Fe Mössbauer spectroscopy through the characteristic signature of five-fold coordinated iron (**Figure S2**).¹⁰ In agreement with our previous studies,^{10,16} the implementation of anionic vacancies led to a significant improvement of the electrochemical properties as compare to the defect-free compound $\text{FeF}_{3-x}(\text{OH})_x$. Using a similar electrode processing, we showed that $\text{FeF}_{3-x}(\text{OH})_x$ could only

reversibly inserted 0.05 Li^+ per Fe while introducing vacancies improved the reversibility up to 0.6 Li^+ per Fe (**Figure S3**).

The thermal decomposition of the precursor $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$ is associated with a distinctive color change, from the light green color of $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$ to a brownish hue for $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$. To track the evolution of the optical band-gap, we collected diffuse reflectance UV-visible spectrophotometry data, which were translated into Kubelka-Munk¹⁷ absorption spectra. The resulting data for $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$, $\text{FeF}_{3-x}(\text{OH})_x$ and $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ are shown in **Figure 1**. On decomposition, the adsorption edge is slightly red-shifted from $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$ ($E_{\text{gap}} = 4.05 \text{ eV}$) to $\text{FeF}_{3-x}(\text{OH})_x$ ($E_{\text{gap}} = 3.75 \text{ eV}$). These band-gap energies indicate an electronic insulator behavior, which suggests that substituting F with OH does not significantly alter the electronic properties of this material.¹⁸ After a second thermal treatment to form anionic vacancies, however, the absorption band is more strongly red-shifted, giving an optical band-gap energy for $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ of $E_{\text{gap}} \approx 2.05 \text{ eV}$. This large band-gap reduction means this anion-vacant material may be considered a semiconductor, rather than an insulator. This distinction is usually associated with an increase in electronic conductivity, because of the relative ease with which charge carriers can now be thermally excited across the fundamental band gap.

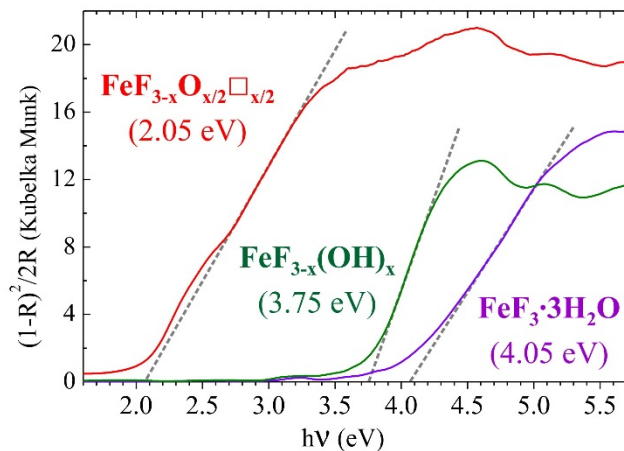


Figure 1. Kubelka-Munk absorption spectra of $\text{FeF}_3 \cdot 3\text{H}_2\text{O}$, $\text{HTB-FeF}_{3-x}(\text{OH})_x$ and $\text{HTB-FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$.

The atomic structure of $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ was probed by the pair distribution function (PDF) method obtained by Fourier transformation of the total scattering data.¹⁹ PDF describes structural features across different scales, giving information about local-, intermediate- and long-range order.²⁰ We refined the structure of $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ using the following procedure: (1) the structural model was refined against the PDF data with the exclusion of the short-range order, *i.e.*, in the 7 Å–50 Å range. In this region, the HTB-type structure (space group: $Cmcm$)²¹ provides an excellent fit to the data, with a reliability factor $R_w = 18.4\%$ (**Figure 2a**). (2) Including the local structure, *i.e.*, the 1 Å–7 Å range, increased the reliability factor to $R_w = 44.5\%$ (**Figure 2b**). While the ideal FeF_3 structure provides a good description of the average structure at intermediate–to–long range scales, over shorter distances the local structure shows significant deviations (cf. the difference curve, **Figure 2b**), which we attribute to local structural distortions due to anionic vacancies.

To more clearly visualize the local structure in $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$, differential PDF (d-PDF) was obtained by subtracting re-scaled PDFs of $\text{FeF}_{3-x}(\text{OH})_x$ and $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ (**Figure 2c**). The d-PDF allows identification of specific interatomic-distances that appear with the introduction of anionic vacancies at 1.83 Å–2.04 Å, 3.08 Å, 3.40 Å and 6.23 Å. The structural distortion introduced by forming anion vacancies therefore extends beyond the first coordination sphere of iron, which suggests the emergence of new structural motifs.

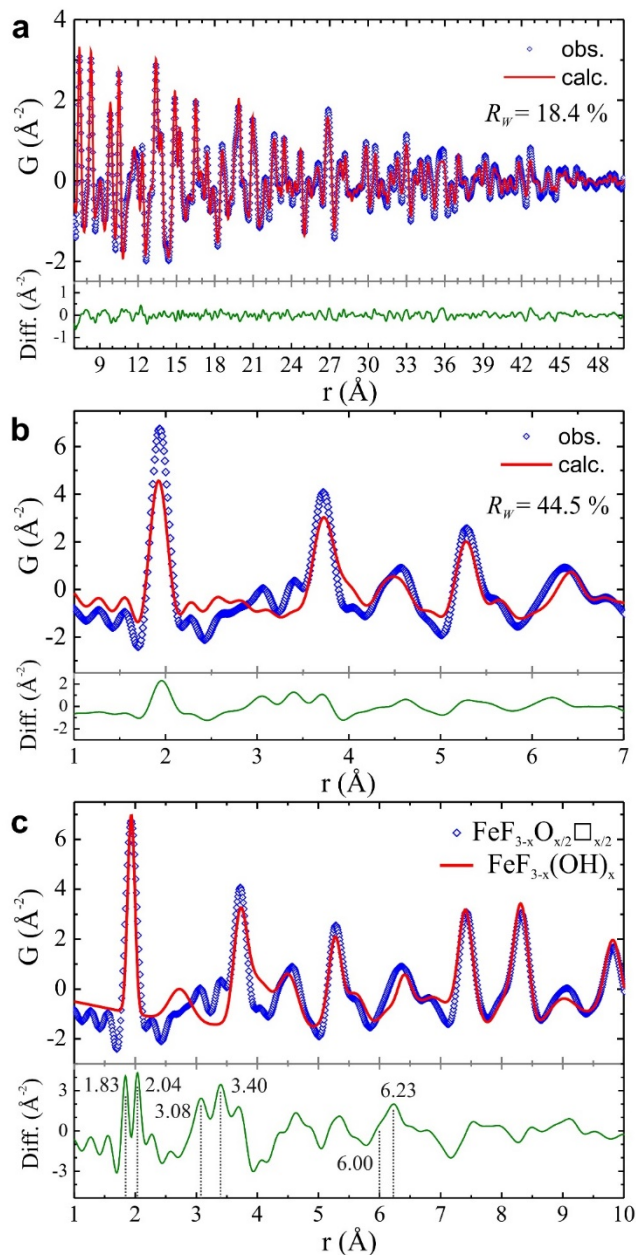


Figure 2. PDF refinement of $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ using the HTB-type structure. **a.** Intermediate-long range region (7 Å–50 Å). **b.** Short range region (1 Å–7 Å). **c.** Re-scaled PDFs of $\text{FeF}_{3-x}(\text{OH})_x$ and $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ and the corresponding differential PDF (d-PDF).

To better understand the structural changes that accompany dehydroxylation, we performed a series of density functional theory (DFT) calculations, modelling stoichiometric FeF_3 ($\text{Fe}_{12}\text{F}_{36}$),

hydroxylated FeF_3 ($\text{Fe}_{12}\text{F}_{36-x}(\text{OH})_x$), and dehydroxylated FeF_3 ($\text{Fe}_{12}\text{F}_{36-x}\text{O}_{x-y}\square_y$). We performed calculations at two defect concentrations: the first considered hydroxylation at two anion sites, followed by dehydroxylation to form a single $\square_{\text{F}}+\text{O}_{\text{F}}$ defect pair; the second considered hydroxylation at four anion sites, followed by dehydroxylation to form two $\square_{\text{F}}+\text{O}_{\text{F}}$ defect pairs. In both cases, geometry optimization of the dehydroxylated model gives a spontaneous reorganization of F^- anions, corresponding to $\text{FeF}_5\text{O} + \text{FeF}_5 \rightarrow \text{FeF}_4\text{O} + \text{FeF}_6$ changes in Fe coordination environments. Here, we discuss this relaxation in the double vacancy system, while the equivalent details for the single vacancy system are given in the SI. **Figure 3** shows the geometries of HTB- $\text{Fe}_{12}\text{F}_{36}$ and $\text{Fe}_{12}\text{F}_{32}\text{O}_2\square_2$. For HTB- $\text{Fe}_{12}\text{F}_{36}$, we show a starting unrelaxed structure (*i.e.*, at a corresponding $\text{Fe}_{12}\text{F}_{32}(\text{OH})_4$ geometry) and the final optimized structure. For $\text{Fe}_{12}\text{F}_{32}\text{O}_2\square_2$, we show only the final optimized geometry. Before relaxation, the anion vacancies in $\text{Fe}_{12}\text{F}_{32}\text{O}_2\square_2$ are on opposite sides of the hexagonal cavity. These vacancies give pairs of under-coordinated Fe sites with pure F-coordination, FeF_n ($n < 6$), and with mixed F/O-coordination, FeF_mO , separated by the anion vacancies. The under-coordinated FeF_n ($n < 6$) centers have formal oxidation states less than +3, which are balanced by six-coordinate FeF_5O centers with formal oxidation states of +3.5. Geometry optimization involves a structural rearrangement where F ions leave FeF_5O units and join previously under-coordinated FeF_n units. This produces a structure where every Fe now has a formal +3 oxidation state, and oxide anions link vertex-sharing pairs of square-pyramidal FeF_4O units. For the double vacancy system, the formation of defect pairs on opposite sides of the hexagonal cavity is characterized by a specific set of interatomic distances, which closely match those observed in the d-PDF (**Table 1**), allowing all distances to be assigned.

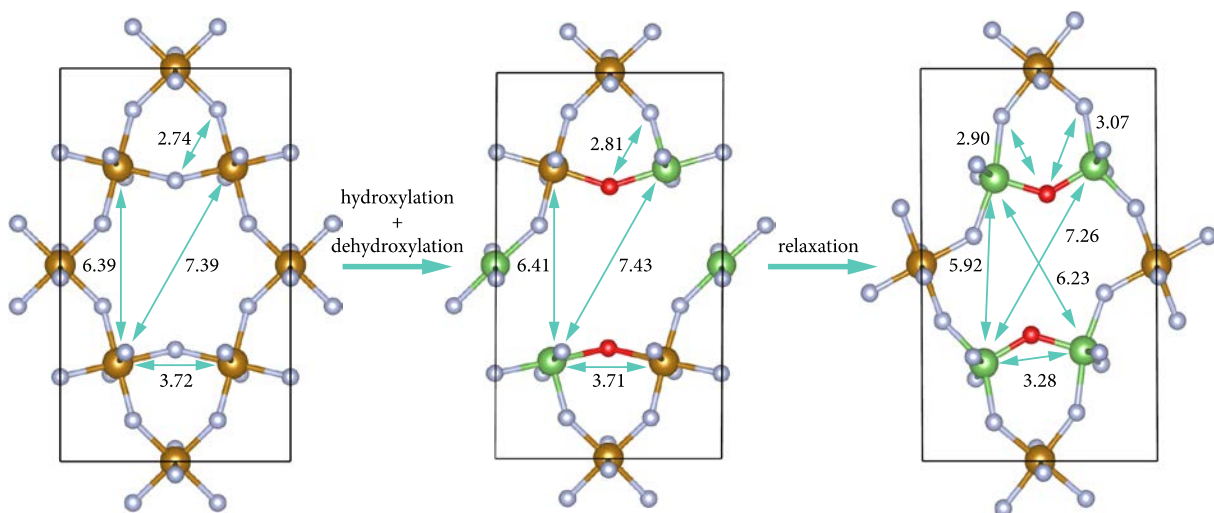


Figure 3. Left panel: optimized structure of the $\text{Fe}_{12}\text{F}_{36}$ cell. Middle to right panels: evolution of the structure of the $\text{Fe}_{12}\text{F}_{32}\text{O}_2$ cell before and after relaxation. Views are along the c -axis. Distances are given in Angstroms. The structure files used to generate this plot are available under CC-BY-4.0 licence as part of ref. 22.

Table 1. Interatomic distances obtained from the differential PDF (d-PDF), DFT calculations (double vacancy model) and corresponding assignments.

d-PDF (Å)	DFT (Å)	Assignments
1.83	1.79	Fe–O
2.04	1.97	Fe–F
3.08	3.07	O–F
3.40	3.28	Fe–Fe
6.23	6.23	Fe–Fe

The assignment of d-PDF distances to pairs of bipyramidal $\text{FeF}_4\text{O–FeF}_4$ units indicates a degree of defect clustering. The x-ray diffraction data, however, appear to not show any superstructure peaks, which suggests that this clustering does not lead to long-ranged ordering of anion vacancies.

The angular-momentum-projected densities of state (pDOS) from our DFT calculations provide information about how defects change the electronic structure of HTB-FeF₃ (**Figure 4**). In stoichiometric HTB-FeF₃ (top panel) the valence band is dominated by the F 2p states and the conduction band is dominated by Fe 3d states. Hydroxylation (middle panel) introduces new occupied O 2p defect states slightly above the top of the valence band. Dehydroxylation to form anion vacancies (bottom panel) pushes these O 2p states further into the gap. Compared to stoichiometric HTB-FeF₃, these O 2p states reduce the energy gap between highest-occupied and lowest-unoccupied states. The relaxed dehydroxylated structure contains both six-coordinate FeF₆ and five-coordinate FeF₄O units. By projecting the d-state orbital character of the charge density for these 5-coordinate and 6-coordinate Fe sites (**Figure S5**), we observe that d-states from the 5-coordinate Fe centres mix with the O 2p defect states above the conduction band maximum, and contribute to the effective band-gap reduction.

Calculated absorption spectra for these three systems show a sequential reduction in the onset of optical absorption in moving from HTB-FeF₃ to the hydroxylated system, and then to the dehydroxylated system (**Figure 4, bottom**). While the reduction in optical band gap predicted by these calculations is not as large as we observe in our experimental absorption spectra (**Figure 1**), the correct qualitative trend is reproduced, which supports our proposition that this change from insulating to semiconducting behavior is caused by the introduction of lattice oxygen defects.

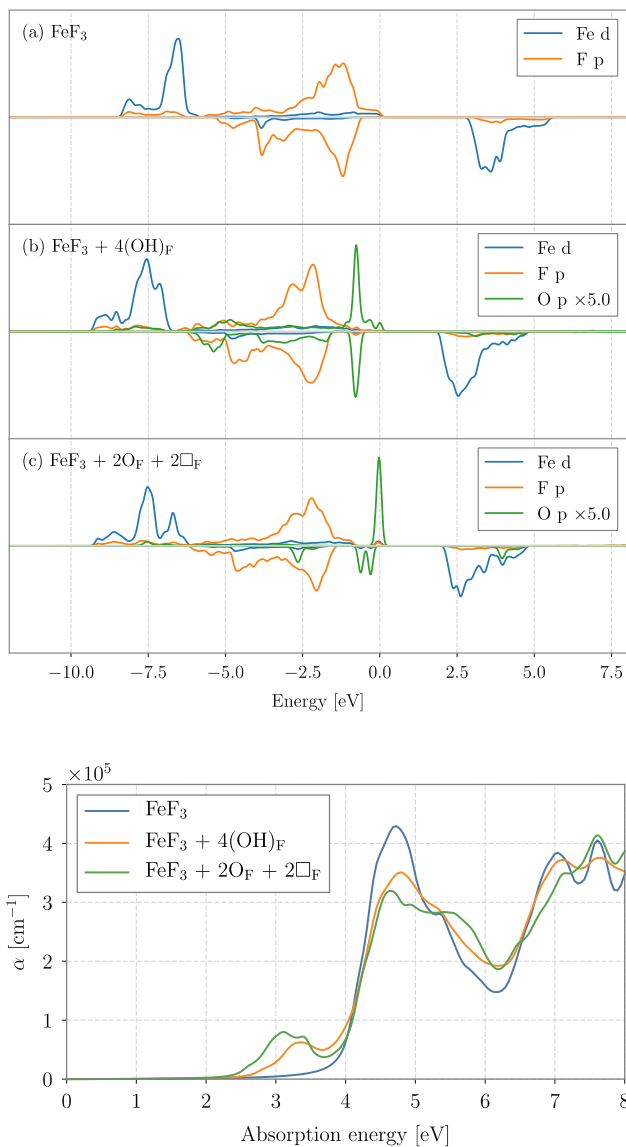


Figure 4. top. PBEsol calculated electronic densities of states (EDOS) for (a) HTB- FeF_3 , (b) hydroxylated HTB- FeF_3 , and (c) the dehydroxylated HTB system with two vacancies. **bottom.** PBEsol calculated absorption coefficients, α , for stoichiometric FeF_3 , hydroxylated $\text{FeF}_3 + 4(\text{OH})_{\text{F}}$, and dehydroxylated $\text{FeF}_3 + 2\text{O}_{\text{F}} + 2\text{□}_{\text{F}}$. *Source:* source data, figure files, and plotting scripts are available under the CC-BY-4.0 licence as part of ref. 22.

To conclude, we investigated the local structural changes induced by the stabilization of anionic vacancies in an iron oxyfluoride electrode material. The combination of the PDF and DFT-calculations used in this study made it possible to gain a clearer understanding of the local structure of anionic vacancies in HTB-FeF₃. The PDF enabled us to visualize the atomic distortion induced by the vacancies. It revealed that the structure locally re-arranges at a scale range of several angstroms. Differential PDFs clearly identified new interatomic-distances that appear after the introduction of anionic vacancies. DFT-calculations were used to model an HTB cell containing two anionic vacancies. After the cell relaxation, the under-coordinated Fe form dimers that induced new structural motifs in the structure with inter-atomic distances that match those observed by PDF, hence validating the proposed model. Moreover, we identified the presence of defect states of O 2p character above the valence band upon hydroxylation of HTB-FeF₃. These states shift further towards the conduction band upon the introduction of vacancies. As a result, vacancies drastically modify the electronic structure moving from electronic insulator to a semiconductor, which is correlated with the observed improvement of the electrochemical properties using this material as an electrode in a lithium-ion cell.

EXPERIMENTAL AND COMPUTATIONAL METHODS

Synthesis. The precursor FeF₃·3H₂O was synthesized according to the reference¹⁵. FeF_{3-x}O_{x/2}□_{x/2} was prepared by a two-steps thermal decomposition process of FeF₃·3H₂O. First, FeF₃·3H₂O was decomposed at 150 °C under air for two hours using the self-generated atmosphere method. A subsequent heat treatment was performed at 250 °C for 12h and 350 °C for one hour to form FeF_{3-x}O_{x/2}□_{x/2}.

X-ray diffraction (XRD) analysis. Powder XRD patterns were recorded on PANalytical X'Pert (Co-K α or Cu-K α_1 radiations) diffractometers in a Bragg–Brentano geometry (θ – 2θ).

Mössbauer spectroscopy. Measurements were performed at room temperature using a constant acceleration Halder type spectrometer with a room temperature ^{57}Co source in transmission geometry. The velocity was calibrated using pure α -Fe as the standard material. Mossbauer spectra were reconstructed using the WinNormos[®] software (*Wissenschaftliche Elektronik GmbH*).

UV-Visible absorption analysis. Diffuse reflectance spectra $R(\lambda)$ were recorded at room temperature using a Cary 17 spectrophotometer with an integration sphere in the wavelength range of 200-800 nm and a 1 nm step (bandwidth : 2 nm). Halon was used as white reference.

Pair Distribution Function (PDF) analysis. High energy X-ray data were collected at the 11-ID-B station at the Advanced Photon Source (Argonne National Laboratory). After corrections (background and Compton scattering), PDFs were extracted from the data using PDFgetX2 software.²³ Refinements of the PDF data were performed using PDFgui.²⁴

Electrochemical characterization. Electrochemical reactions with lithium were evaluated using coin cells comprising an iron-based fluoride and metallic lithium as electrodes. The composition of iron-based fluorides Electrodes was 75 wt.% active materials, 15 wt.% acetylene black as the conductive agent, and 10 wt.% polyvinylidene difluoride as the binder. The different components were hand milled with NMP solvent prior to deposit on an aluminum foil. The prepared slurry was spread by a Doctor-blade. The area of the electrode was 1.0 cm² with a typical mass loading around 3 mg. The electrolyte consisted of the commercially available LP30 (1.0M LiPF₆ dissolved in a mixture of ethylene carbonate and ethyl methyl carbonate (3:7 volume ratio). Cells were cycled between 2 and 4.2V at 50 mA/g current density.

Computational methods.

Our density functional theory (DFT) calculations were performed using the code VASP^{25,26} with valence electrons described using a plane-wave basis with a cutoff of 520 eV. Interactions between core and valence electrons were described using the projector augmented wave (PAW) approach²⁷, with cores of [Ar] for Fe, [He] for O, [He] for F, and [H⁺] for H. Calculations used the revised Perdew-Burke-Ernzerhof generalized-gradient approximation PBEsol functional²⁸, with a Dudarev +U correction applied to the Fe d states (GGA+U)^{29,30}, or the hybrid HSE06 functional³¹. For the PBEsol+U calculations, we used a value of $U=5.3$ eV, taken from the pymatgen MPRelaxSet.³² Calculations were performed on Fe₁₂Fe₃₆ cells, or anion-substituted equivalents. For each stoichiometry, we first calculated zero-pressure optimized lattice parameters, by performing a series of constant-volume geometry optimisations and fitting the resulting energy-volume data to the Murnaghan equation of state, before performing a final geometry optimization at these cell parameters. Individual calculations were deemed converged when all atomic forces were smaller than 0.01 eV Å⁻¹. All calculations were spin polarized, and we assumed ferromagnetic configurations. *k*-space sampling used a 4 × 2 × 4 Monkhorst-Pack grid. A data set containing all DFT calculation inputs and outputs is available at the University of Bath Data Archive²² published under the CC-BY-4.0 license. This dataset includes scripts used for post-processing and plotting our DFT data.

ASSOCIATED CONTENT

Supporting Information. X-ray diffraction and Mössbauer analyses, electrochemical characterizations, and additional DFT-calculations data.

AUTHOR INFORMATION

The authors declare no competing financial interests.

NOTES:

The DFT dataset supporting this study is available from the University of Bath Research Data Archive (ref 22, doi.org/10.15125/BATH-00576), published under the CC-BY-4.0 license. This dataset contains all input files and outputs for the VASP DFT calculations, and Python scripts for collating, processing, and plotting the relevant data used in our analysis.

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Supplementary Information

Impact of Anionic Vacancies on the Local and Electronic Structures of Iron-based Oxyfluoride Electrodes

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X-ray diffraction analysis

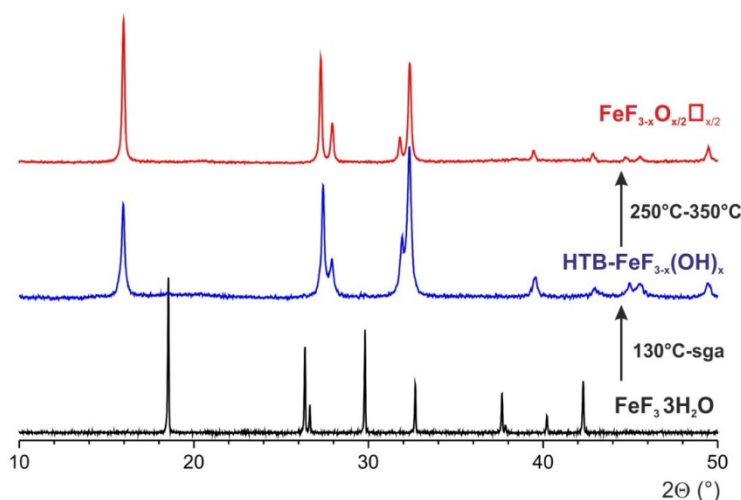


Figure S1. Powder x-ray diffraction patterns of $\beta\text{-FeF}_3 \cdot 3\text{H}_2\text{O}$, the hydroxyfluoride phase obtained by thermal decomposition of the tri-hydrate precursor at 130°C under self-generated atmosphere (sga) and $\text{FeF}_{3-x}\text{O}_{x/2}$ obtained by thermal treatment of $\text{FeF}_{3-x}\text{OH}_x$ at 250°C overnight and 350°C for 1 hour under vacuum.

Mössbauer spectroscopy

Figure S2 shows the ^{57}Fe Mössbauer spectra of $\text{FeF}_{3-x}\text{O}_{x/2}$ which was reconstructed using three contributions: $Fe(a)$, accounting for 34% in relative intensity, which is characterized by an isomer shift of 0.46 mm/s and a low quadrupolar splitting value of 0.15 mm/s, which was ascribed to Fe^{3+} ions in regular FeF_6 octahedra, $Fe(b)$ which is described by an isomer shift of 0.38 mm/s and a quadrupole splitting distribution that is associated to Fe^{3+} ions located in distorted $\text{FeF}_{6-x}(\text{OH})_x$ octahedra, finally $Fe(c)$, accounting for 28%, with an isomer shift at 0.34 mm/s and a large quadrupolar splitting distribution with an average value of 1.11 mm/s (**Table S1**). This last signal is due to five-fold coordinated Fe^{3+} ions and thus attests to the presence of anionic vacancies. Note that the observed differences in isomer shift values of $Fe(a)$ and $Fe(b)$ clearly evidenced a drastic change of the ionic character of the metal-ligand bonds through the fluorine substitution by OH^- or O^{2-} and anionic vacancies.

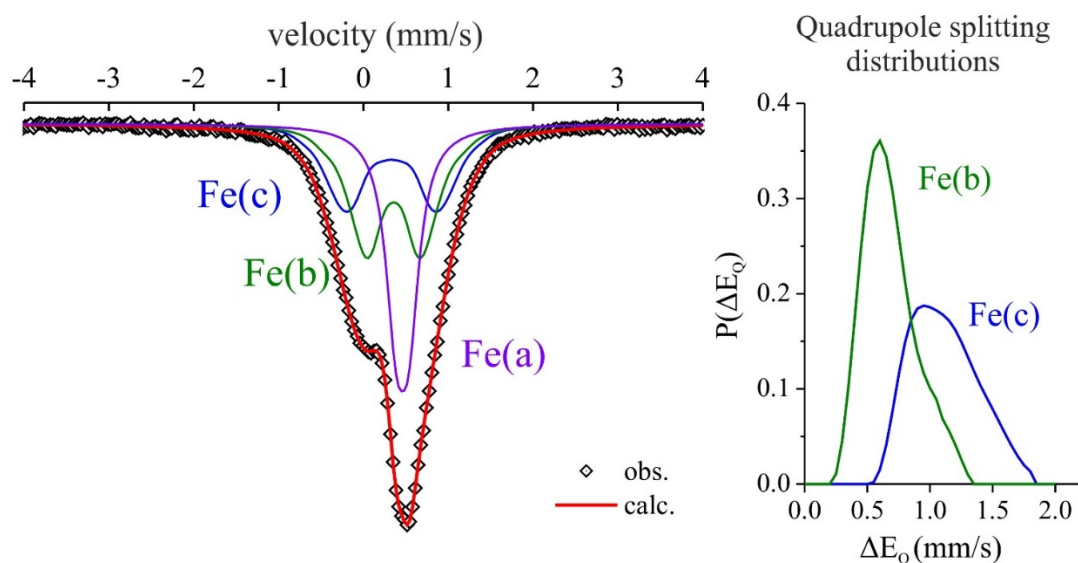


Figure S2. ^{57}Fe Mössbauer spectra of $\text{HTB-FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ obtained after a post-thermal treatment. Right panel: Quadrupole splitting distributions.

Table S1. ^{57}Fe Mossbauer parameters obtained from the reconstruction of spectra shown in Figure S2. *Average value.

	δ (mm/s)	ΔE_Q (mm/s)	Γ (mm/s)	Area (%)	Assignments	Environments
Fe(a)	0.46(1)	0.15(1)	0.30(1)	34	Six-fold coord. Fe^{3+}	FeF_6
Fe(b)	0.38(1)	0.68*	0.30(-)	38	Six-fold coord. Fe^{3+}	$\text{FeF}_{6-x}(\text{OH})_x$
Fe(c)	0.34(1)	1.11*	0.30(-)	28	Five-fold coord. Fe^{3+}	FeF_4O

Electrochemical properties

To probe the impact of the presence of anionic vacancies on the electrochemical properties, electrodes using $\text{FeF}_{3-x}(\text{OH})_x \cdot n\text{H}_2\text{O}$ and $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ as active materials were tested in lithium cells. Equal electrode processing (hand milling) and operating conditions ($50 \text{ mA} \cdot \text{g}^{-1}$, 2.0 V–4.2 V voltage window) were used. The galvanostatic discharge and charge curves for the first cycle are gathered in **Figure S3**. In agreement with our previous study, the $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ electrode shows a significant improvement in terms of lithium insertion compared to the vacancy-free compound.^{1,2} The $\text{FeF}_{3-x}\text{O}_{x/2}\square_{x/2}$ electrode showed high reversibility, with the insertion of up to 0.7 Li^+ per Fe (166 mAh/g) during the first discharge and 0.6 Li^+ (140 mAh/g) upon charging.

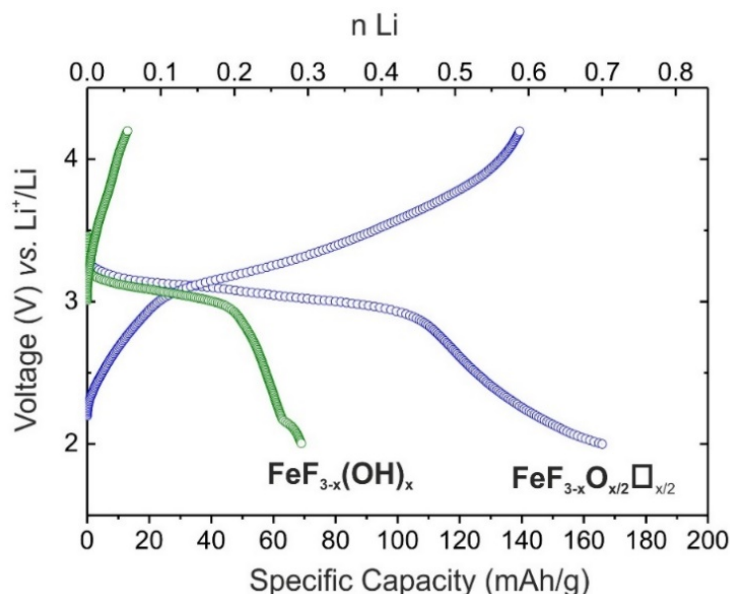


Figure S3. First discharge and charge curves for Li/FeF_{3-x}O_{x/2}□_{x/2} (blue) and Li/FeF_{3-x}OH_x (green) cells cycled under 50 mA.g⁻¹.

DFT calculations

Single-vacancy dehydroxylated FeF₃

To model the formation of a single anion vacancy in a hydroxyfluorinated cell, we performed a sequence of DFT calculations, starting with a configuration containing two OH⁻ substitutions: Fe₁₂F₄₆(OH)₂ (**Figure S4, left panel**). We then removed H + OH to form a dehydroxylated cell with stoichiometry Fe₁₂F₄₆O₁□₁, containing one Fe–O–Fe oxo-bridge and one anionic vacancy, Fe–□–Fe (**Figure S4, middle panel**). Performing a geometry optimization on this dehydroxylated structure, we identified a relaxation of the F anions close to the vacancy, corresponding to FeF₅ + FeF₅O → FeF₆ + FeF₄O change in the Fe coordination environments, and forming a FeF₄–O–FeF₄ bipyramidal unit (**Figure S4, right panel**). The five-fold coordinated Fe³⁺ have square-pyramidal coordination geometries. To estimate the relaxation energy associated with this reorganization we performed a second geometry optimization starting from the original dehydroxylated structure, ensuring that the original coordination environment was preserved. The resulting structure, with the oxygen vacancy in the original as-formed position, was 0.40 eV higher in energy than the fully relaxed structure.

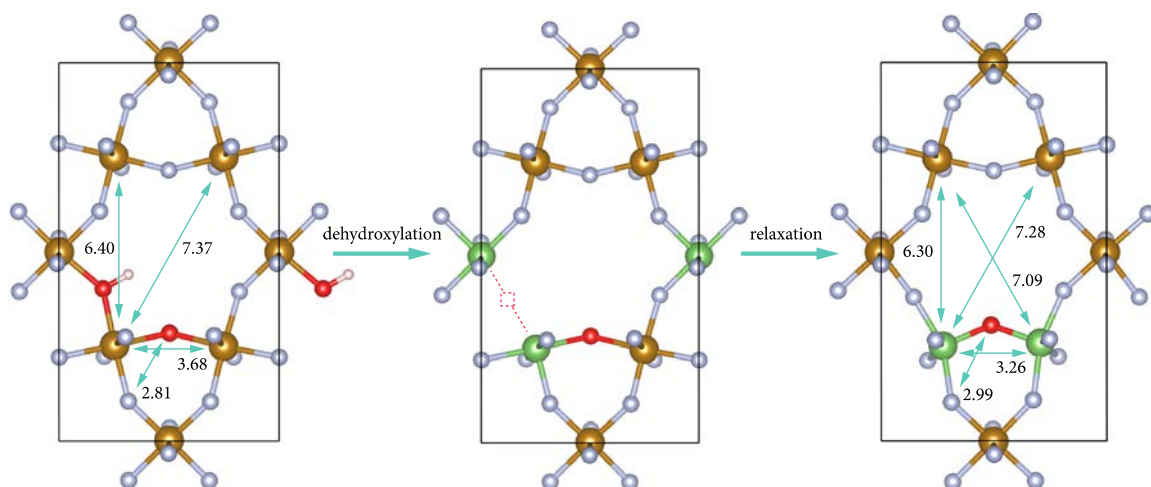


Figure S4. Left panel: optimized structure of the $\text{Fe}_{12}\text{F}_{34}(\text{OH})_2$ cell. Middle to right panels: evolution of the structure of the $\text{Fe}_{12}\text{F}_{34}\text{O}$ cell before and after relaxation. Views are along the c -axis. Distances are given in Angstroms. The structure files used to generate this figure are available under CC-BY-4.0 [3].

Fe coordination environment DOS analysis

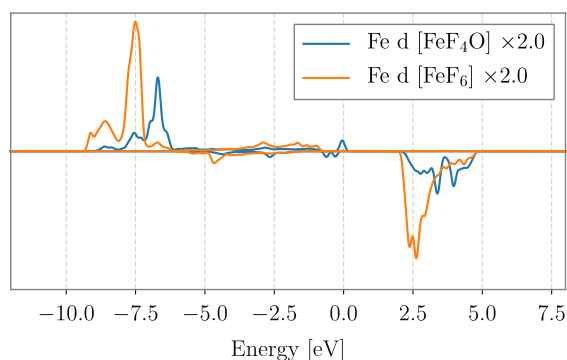


Figure S5. PBEsol projected densities of states for the Fe d states, differentiating between Fe centres with FeF_4O coordination (blue) and FeF_6 coordination (orange). *Source:* source data, figure files, and plotting scripts are available under CC-BY-4.0 [3].

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