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Graphical Abstract





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Review Article Progress in all-organic rechargeable batteries using cationic and anionic configurations: Toward low-cost and greener storage solutions?

Philippe Poizot^{1,2,*}, Franck Dolhem^{3,4} and Joël Gaubicher¹

- Our entry into the Fourth industrial revolution since the turn of 1
- the century is set to revolutionize our daily life notably with the 2
- blooming of digital technologies such as communications, 3
- artificial intelligence, technologies related to the Internet of 4 5
- Things, 3-D printing or nano/biotechnologies. It is however 6
- hoped this new paradigm shift will integrate sustainable 7 development goals and actions to address the critical damage
- 8
- caused by the previous industrial revolutions especially the threat of global warming. We have to be particularly aware 9
- there remains the urgent need for cleaner energy technologies
- 10
- which calls for a radical change in the energy mix to favor 11
- renewable energy and environmentally responsible energy 12
- storage solutions. Organic materials should provide 13
- opportunities to further improve existing energy storage 14
- technologies while offering sustainable, versatile and 15
- potentially low-cost energy storage devices. This review seeks 16
- to provide an update on all-organic battery assemblies 17
- 18 reported to date as well as some perspectives we can expect
- 19 in the future notably for stationary applications.

20 Addresses

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Introduction

A few years ago, we outlined a personal view about the 33 tricky questions of energy supply, its storage and conver-34 sion in the early 21st century and underlined the impor-35 tance of developing efficient, safe but also low-polluting 36 electrochemical storage solutions [1]. To date, commercial 37 batteries exclusively include inorganic electrode materi-38 als notably 3d transition metals which are scarce, expen-39 sive and energy greedy [2]. In contrast, organic materials 40 enable access to low cost and possibly greener compounds 41 because composed of naturally abundant elements (i.e., 42 C, H, O, N or S) moreover they are easier to recycle. In 43 addition, they offer high structural designability through 44 the well-established principles of organic chemistry and 45 notably access to both n- and p-type electrochemical stor-46 age mechanisms [3] making various cell or electrode con-47 figurations possible (Figure 1). 48

In 10 years, tremendous progress has been made to pro-49 mote organic compounds in various rechargeable storage 50 devices giving rise to nearly 15 published review articles 51 especially for applications in non-aqueous (metallic) Li or 52 Na-based batteries; for very recent examples, the reader 53 could refer to refs. [5–10]. Notwithstanding this abundant 54 literature, there appeared to be a lack of a comprehen-55 sive summary dedicated to all-organic cells that are also 56 increasing in number thanks to this rapid progress on or-57 ganic electrode materials. Herein, we have attempted to 58 fill the gap by thoroughly reporting prototype examples of 59 all-organic batteries investigated until now including the 60 pioneering examples studied in the mid-80s. One partic-61 ular exciting option is the true possibility of storing elec-62 tricity through cell reactions devoid of metals making the 63 concept of molecular-ion batteries possible as pointed out 64 by Yao's group in a recent visionary article [11^{••}]. In fact, 65 the shuttling ion can be either protons or ammonium-type 66 cations but also anions. In the latter case, a large choice of 67 chemical structures is accessible from atomic to molecular 68 anions. Moreover, anions tend to show higher limiting mo-69 lar conductivity values in ordinary organic solvents due to 70 lower solvation effects. In aqueous electrolytes, promising 71 organic assemblies based on the use of low-cost chemical 72 compounds could also pave the way for innovative local 73 stationary electrochemical storage devices. 74

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2 Batteries and Supercapacitors

Figure 1



Schematic of fundamental cell configurations obtained by playing with both n- and p-type organic electroactive materials shown during the discharge process. (a)-All-organic cell incorporating n-type electrode materials only (cationic rocking-chair). (b) Symmetric situation with p-type electrode materials (anionic rocking-chair). (c) Mixing both n- and p-type electrode materials (dual-ion configuration). Note that the use at the electrode level of mixed n-/p-type moieties can be envisaged too [4]. In addition, a given organic skeleton bearing a redox-active p-type moiety exhibits as a rule a formal potential higher than that of the corresponding n-type counterpart.

75 Advances in non-aqueous all-organic 76 batteries

All-organic batteries are naturally characterized by very 77 different assemblies depending on the selected active 78 material (p- and/or n-type) as well as the chemical na-79 ture of the charge carriers (cationic or anionic charge bal-80 ance). Therefore, we opted for a summary table for the 81 sake of comparison and to make the discussion easier. 82 Hence, Table 1 lists the possible organic cell configura-83 tions, the chemical nature of the used electrode materials¹ 84 and electrolyte as well as some performance metrics. First 85 of all, it seemed to us relevant to start our overview with 86 battery examples coupling inorganic and organic elec-87 trodes. In this area, since aromatic carboxylates proved to 88 be interesting candidates as negative material [7], Toy-89 ota Laboratories reported in 2014 [12[•]] very good elec-90 trochemical performance by coupling the high voltage 91 spinel LiNi_{0.5}Mn_{1.5}O₄ with dilithium 2,6-naphthalene di-92 carboxylate leading to 3.9 V Li-ion cells and demonstrated 93 the possible construction of 8V-bipolar laminated Li-ion 94 batteries (LIBs) delivering high specific power and en-95 ergy values (Table 1, #1). Preliminary data were also pub-96 lished by Mou *et al.* [13] using a composite anode made 97 of calcium terephthalate ball-milled with graphite face to 98 LiCoO₂ as the cathode material (Table 1, #2). Medabalmi 99 et al. [14] made a Na-ion coin-cell prototype operating at 100 \sim 3.2 V by using the sodiated form of 2,6-naphthalene di-101 carboxylate and Na₃V₂O₂(PO₄)₂/rGO (rGO standing for 102 reduced graphene oxide) for the cathode side but lim-103

ited stabilities upon cycling were observed (Table 1, #3). 104 Changing for a p-type cathode, Fan et al. [15] recently re-105 ported a potassium-based dual-ion full battery (PDIBs) 106 based on graphite anode, polytriphenylamine cathode, 107 and KPF_6 -based electrolyte (Table 1, #4) that shows 108quite good cycling stability over 500 cycles. Kang and co-109 workers [16] examined in a parallel research the perfor-110 mance of a lithium-based dual-ion full batteries (LDIBs) 111 with the N, N-substituted phenazine/Li₄Ti₅O₁₂ assembly 112 but the cyclability was not discussed at alk(Table 1, #5). 113

The second section of Table 1 concerns all-organic bat-114 teries incorporating n-type materials only (cationic con-115 figuration). Our group [17] was the first to report an 116 all-organic Li-ion cell based on renewable raw materi-117 als thanks to the amphoteric redox property of $Li_4C_6O_6$ 118 which makes the design of a cell exhibiting ~ 1 V as out-119 put voltage (Table 1, #6). Later we tried to go further by 120 investigating dilithium (2,5-dilithium-oxy)-terephthalate 121 $(Li_4-p-DHT)$ as another dual-function electrode material 122 deriving from biomass [18]. However, Chen's group re-123 ported the best performance with this material (Table 1, 124 #7) at the condition to be prepared as nanosheets [19]. 125 This second prototype of organic LIBs exhibits an aver-126 age operation voltage of ~ 1.8 V and an energy density of 127 about 130 W h/kg together with long cycling life (1000 cy-128 cles) when supported on graphene [20^{••}]. A biomolecule-129 based full LIB was even proposed by Hu et al. [21] us-130 ing a naturally occurring quinone (emodin) as the cath-131 ode and lithium humates as the anode. Nevertheless, a 132 fast capacity fading was observed probably due to the 133 poor stability of the electrode materials (emodin bears 134 three OH groups whereas lithium humates cannot be well 135

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¹ Note that the following terms "cathode" and "anode" refer to positive and negative electrodes, respectively.

Tabl	e 1							
Organic batteries.								
#	Cell configuration (ionic carriers)	Positive material or "cathode"	Negative material or "anode"	Electrolyte	Output voltage (V)	Cycling stability: retention, cycles, rate or current density	Specific capacity (mAh/g), Specific energy (Wh/kg), Final coulombic efficiency, Loading (mg/cm ²)	Re
	n-type inorganic ca	athode assembled with n-type organic	anode					
1	Li-ion	$LiNi_{0.5}Mn_{1.5}O_4$	LiO ₂ C CO ₂ Li	1 M LiPF ₆ in EC:DMC: EMC 3:4:3 (v/v/v)	3.9	96%, 100, 1 C	77 ^(c) , 300 ^(c) , -, -	[1]
2	Li-ion	LiCoO ₂	Ca _{1/2} O ₂ C-CO ₂ Ca _{1/2} /10% Graphite	1 M LiPF ₆ in EC:DMC 1:1 (v/v)	2.75	82%, 50, 1 C	138 ^(a) , -, 92%, -	[13
3	Na-ion	$Na_3V_2O_2(PO_4)_2/rGO^{(d)}$	NaO ₂ C	1 M NaClO ₄ in PC	3.2	68%, 20, 13 mA/g	150 ^(a) , -, 33% ^(g) , 2	[14
4	Dual-ion (K ⁺ /PF ₆ ⁻)		Graphite	0.8 M KPF ₆ in EC:DC 1:1 (v/v)	3.23	75.5%, 500, 50 mA/g	60 ^(b) , 105, -, 1.2	[15
5	Dual-ion (Li⁺/TFSI)		${\rm Li}_4{\rm Ti}_5{\rm O}_{12}$	1 M LiTFSI in TEGDME	1.9	-, -, 50 mA/g	163 ^(b) , -, -, -	[16
	n-type organic cat	hode assembled with n-type organic a	node					
6	Li-ion			1 M LiPF ₆ in EC:DMC 1:1 (v/v)	1.0	70%, 50, C/20	5555	[17
7	Li-ion	LiO ₂ C Li OLi	LiO ₂ C LiO ₂ C OLi	1 M LiPF ₆ in EC:DMC 1:1 (v/v)	1.8	91%, 20, 48.2 mA/g	225 ^(a) , 130, 99.7%, 1-1.5	[19
8	Li-ion	Me OH	Electrochemically pre-reduced lithium humates	1 M LiPF ₆ in EC:DMC 1:1 (v/v)	1.5	50%, 100, 100 mA/g	127 ^(b) , -, 99%, -	[2]
9	Li-ion	$\{ N_{i} \in \mathbb{C}^{N_{i}} \}_{n} \in \mathbb{C}^{N_{i}} \cap \mathbb{C}^{N_{i}} \cap \mathbb{C}^{N_{i}} \}_{n}$	$(\text{LiO})_2 C = -CO_2 \text{Li}$ $\text{LiO}_2 CCO_2 \text{Li}$ $\text{LiO}_2 CCO_2 \text{Li}$ (pre-reduced)	1 M LiPF ₆ in EC:DMC:EMC 1:1:1 (v/v/v)	1.2	81%, 200, 11.3 mA/g	78 ^(b) , 32, 99,6%, 0.8 - 2	[22
10	Na-ion	NaO ₂ C O ₂ Na	NaO ₂ C NaO ₂ C Na	1 M NaClO ₄ in EC:MC 1:1 (v/v)	1.8	76%, 100, 19 mA/g	198 ^(a) , 65, 99%, 1-2	[2:
							(continued on	next

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#	Cell configuration (ionic carriers)	Positive material or "cathode"	Negative material or "anode"	Electrolyte	Output voltage (V)	Cycling stability: retention, cycles, rate or current density	Specific capacity (mAh/g), Specific energy (Wh/kg), Final coulombic efficiency, Loading (mg/cm ²)	Ref.
11	Na-ion		Nao Nao Nao Nao Nao Nao Nao Nao Nao Nao	$1~{\rm M~NaPF_6}$ in PC	1.35	75%, 20, 50 mA/g	73, -, -, 1.4	[24]
12	Na-ion	$\downarrow \qquad \qquad$	NaO ₂ C	1 M NaClO₄ in tetraglyme	1.1	62%, 30, 130 mA/g	210,-, 99.2%, 1	[25]
13	K-ion	KO L CO KO L CO	кодок	1.25 M KPF_6 in DME	1.1	-, 10, 25 mA/g	70 ^(b) , 35, -, -	[26]
14	R ₄ N-ion	<i>t</i> -Bu <i>t</i> -Bu		0.1 M <i>n</i> -Bu ₄ NCIO ₄ +0.01 M <i>n</i> -Bu ₄ NOH in CH ₃ CN	0.6	91%, 250, 150 C	29, -, -, -	[27]
15	H-ion		OH OH OH	2-fluoropyridinium triflate: 2-fluroropyridine 1:1	0.5	66%, 100, 400 mA/g	103 ^(a) , -,99.9%, -	[28]
	p-type organic cat	hode assembled with n-type organic ano	de					
16	Dual-ion (ClO ₄ ⁻ / <i>n</i> -Bu ₄ N ⁺ ; BF ₄ ⁻ / <i>n</i> -Bu ₄ N ⁺)	↓ ∽→ ,	↓ ∽→ ↓ _n	0.5 M <i>n</i> -Bu ₄ NClO ₄ in PC; 0.3 M <i>n</i> -Bu ₄ NBF ₄ in THF	3.7 2.5			[29]
17	Dual-ion (BF4 ⁻ /n-Bu4N ⁺)	$\{ s \}_n$	$\{ s \}_n$	0.2 M <i>n</i> -Bu ₄ NBF ₄ in CH ₃ CN	3.1	-, -, 24 mA/g	-, 93, -, -	[30]
18	Dual-ion (ClO₄ ⁻ / <i>n</i> -Bu₄N ⁺)		t-Bu , O + Bu + Bu + Bu	0.5 M <i>n</i> -Bu ₄ NClO ₄ +0.01 M <i>n</i> -Bu ₄ NOH in CH ₃ CN	0.66	57%, 250, 10 C	32, 22, -, -	[31]
							(continued on next	page)

Table 1 (continued)

Ref.

[24]

[25]

[26]

[27]

[28]

[29]

[30]

[31]

Ref	Specific capacity (mAh/g), Specific energy (Wh/kg), Final coulombic efficiency, Loading	Cycling stability: retention, cycles, rate or current	Output voltage (V)	Electrolyte	Negative material or "anode"	Positive material or "cathode"	Cell configuration (ionic carriers)	#
[27]	(mg/cm*) 44, -, -, -	density 68%, 250, 60 C	1.3	0.5 M <i>n</i> -Bu ₄ NCIO ₄ +0.01 M <i>n</i> -Bu ₄ NOH in CH ₃ CN			Dual-ion (ClO ₄ ⁻ /n-Bu ₄ N ⁺)	19
[32]	200 ^(a) , 92, 99, -	85%, 500, 8 C	1.8	NaPF ₆ saturated in DME:DOL 1:1 (v/v)	o - - - - - - - - - - - - -		Dual-ion (PF ₆ ⁻ /Na ⁺)	20
[33]	200 ^(a) , -, -, -	80%, 50, 50 mA/g	1.7	5% _{mol} NaClO₄ in plastic crystal electrolyte	o b b b b b b b b b b b b b b b b b b b	P(AN-NA) ^(f)	Dual-ion (ClO4 ^{-/} Na ⁺)	21
[34]	250 ^(a) , -, -, -	-, -, 40 mA/g	2.35	1 M LiPF ₆ in EC:DMC:EMC 1:1:1 (v/v/v)	n-C ₆ H ₁₃ n-C ₆ H ₁₃		Dual-ion (PF ₆ ⁻ /Li ⁺)	22
[35]	153 ^(a) , -, -, -	63%, 60, 40 mA/g	3	1 M LiPF ₆ in EC:DMC:EMC 1:1:1 (v/v/v)	+	$+ \mathbb{C} + \mathbb{C}$	Dual-ion (PF ₆ ⁻ /Li ⁺)	23
[36]	105 ^(b) , -, -, -	67%, 250, 1C	1.35	1 M LiClO4 in EC:DMC 3:7 (v/v)		s S S S S S	Dual-ion (ClO4 ⁺ /Li ⁺)	24
[37]	90 ^(a) , 33, -, 1	-, -, -	1.2	2 M LiTFSI in ethyl acetate (55°C ≤ <i>T</i> ≤−70°C)	$\left[\begin{array}{c} & & \\ & $		Dual-ion (TFSI/Li ⁺)	25
page)	(continued on next							

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Progress in all-organic rechargeable batteries using cationicand anionic configurations Poizot, Dolhem and Gaubicher 5 ARTICLE IN PRESS

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#	Cell configuration (ionic carriers)	Positive material or "cathode"	Negative material or "anode"	Electrolyte	Output voltage (V)	Cycling stability: retention, cycles, rate or current density	Specific capacity (mAh/g), Specific energy (Wh/kg), Fina coulombic efficiency, Loading (mg/cm ²)
	p-type organic ca	thode assembled with p-type organic a	node				
26	ClO ₄ -ion	$(pPy')n \begin{bmatrix} & & \\ $	$(CH_2)_0 \xrightarrow{\bullet} N \xrightarrow{\bullet} N \xrightarrow{\bullet} M_0$	0.2 M LiClO ₄ in CH ₃ CN	1	70%, 100, 0.5 mA/cm ²	16, 15, -, - (thin films)
27	PF ₆ -ion		$ \begin{bmatrix} * \\ N \\ PF_{6} \end{bmatrix}^{+} - \begin{bmatrix} N^{+}(CH_{2})_{5} \\ PF_{6} \end{bmatrix}^{+}_{n} $	1 M <i>n</i> -Bu ₄ NPF ₆ in PC	1.8	65%, 100, 100 mA/g	100 ^(b) , -, -, -
	Organic batteries	with aqueous electrolytes					
28	Li-ion	${ m LiMn_2O_4}$		Li ₂ SO ₄ 2.5 M (pH 7)	1.13	80%, 3000, 280 mA/g	81, 92%, -, 2
29	NH ₄ -ion	(NH4)1.47Ni[Fe(CN)6]0.88	HN STATISTICS	(NH ₄) ₂ SO ₄ 1 M (pH 6)	1	67%, 1000, 0.120 mA/g	35 ^(c') , 35 ^(c') , 97.6%, 2-5
30	Cl-ion			NaCl 0.1 M (pH 7)	1.3	80%, 2000, 10.5 A/g	165 ^(a) , 214 ^(a) , - (100 nm to 1 μm thin films)
31	BF4-ion	Me NH Me Ne Me	$ \underbrace{ \{ \stackrel{\bullet}{\textbf{N}}_{\vec{n}} \\ \vec{Br} \\ \vec{Br} \\ \vec{Br} \\ \vec{Br} \\ \vec{Br} \\ \vec{Dr} \\ \vec{Br} \\ \vec{Dr} \\ $	NaBF₄ (pH 7)	1.2	75-80%, >2000, 60 C (67 μA/cm²)	104 ^(b) , 125 ^(b) , 95 (100 nm thin films)
32	Dual-ion (TFSI/Li ⁺)			LiTFSI 21 M (pH 7)	1	85%, 700, 0.5 A/g	105 ^(b) , 53 ^(b) , ~100, 1

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characterized). Note that a pre-discharge step of the an-136 ode was necessary prior to final assembly since emodin 137 (quinone form) is a non-lithiated compound (Table 1, #8). 138 Similarly, Gutel and co-workers [22] assembled a LIB 139 able to show an average cell voltage of ~ 1.2 V and good 140 cyclability over 200 cycles from pre-reduced perylenete-141 tracarboxylate face to a polyimide as the cathode (Table 142 1, #9). Taking benefit of the successful Li-ion cell based 143 on Li_4-p -DHT, Chen's group [23] investigated the sodi-144 ated counterpart material (Na₄-p-DHT) and published 145 the first all-organic sodium-ion battery (SIB), which gives 146 147 an average operation voltage of $\sim 1.8 \,\mathrm{V}$ for a specific energy of about 65 W h/kg (Table 1, #10). Shaijumon and 148 co-workers [24] proposed as SIB the coupling between 149 *N*,*N*-diamino-3,4,9,10-perylenetetracarboxylic polyimide 150 as the cathode with the disodium terephthalate (Na₂TP) 151 152 as the anode. Again the lack of inserted Na⁺ in the pristine 153 polyimide forced them to electrochemically pre-reduce Na₂TP before assembling (Table 1, #11). The cell deliv-154 ered an initial capacity of 73 mA h/g for an average cell 155 voltage of ~ 1.35 V but with a limited cycling stability. Very 156 recently, Li et al. [25^{••}] have gone beyond by associating 157 Na₂TP with sodiated poly(2,5-dihydroxy-*p*-benzoquinoyl 158 sulfide)/rGO composite material (Table 1, #12). In the 159 continuity of our former works on $Li_4C_6O_6$ [17], in 2016 160 Chen's group constructed the first example of organic K-161 ion batteries based on $K_4C_6O_6/K_2C_6O_6$ system [26] which 162 displayed an operation voltage of ~ 1.1 V and an energy 163 density of 35 W h/kg (Table 1, #13). Other cations than al-164 kali ones were also tested as ionic carriers. Interestingly, 165 when using pre-reduced poly(galvinoxylstyrene) with the 166 tetrabutylammonium ion as the cathode together with 167 168 poly[4(nitronylnitroxyl)styrene)] as the anode, Nishide and co-workers [27] constructed the first n-type full poly-169 mer battery free of metal. The test cell certainly exhibited 170 a limited output voltage (~ 0.6 V) but achieved impressive 171 rate performances with 90% of the original capacity main-172 tained at 150C rate (Table 1, #14). More recently Sjödin 173 174 and co-workers [28] reported a proof-of-principle study on an all-organic proton battery also devoid of metals using 175 poly(3,4-ethylenedioxythiophene) (PEDOT) functional-176 177 ized either with *p*-benzoquinone (cathode) or dihydroxvanthraquinone (anode) and working thanks to an original 178 protonated pyridinium triflate-based non-aqueous elec-179 trolyte (Table 1, #15). 180

The third section of Table 1 concerns the dual-ion cell 181 configuration employing naturally (for potential reason) a 182 p-type electrode material for the cathode side (Figure 1); 183 in this cell configuration the electrolyte is the reservoir 184 185 of ions for the charge compensation within electrode materials. There are more examples reported in the 186 literature for this type of assembly, the first one dating 187 back to the 80s following the discovery of conducting 188 polymers. The first completely organic rechargeable 189 storage battery of this kind were described by Mac-190 Diarmid and co-workers in 1981 by taking benefit of 191

the reversible n- and p-type electrochemical doping of 192 polyacetylene [29]. However, neither cycling curves nor 193 electrochemical performance data were mentioned in the 194 article (Table 1, #16). Then, polythiophene-based full 195 cells were proposed by Inuishi and co-workers exhibiting 196 an energy density of $\sim 90 \text{ W}$ h/kg [30] (Table 1, #17). 197 Fifteen years ago, a new class of polymers emerged 198 consisting of a polymeric chain with stable radical pen-199 dant groups leading to the development of the so-called 200 organic radical batteries (ORBs). Such systems possess 201 the right properties to allow the construction of organic 202 dual-ion cells as well. Nishide and co-workers [31] pro-203 posed the poly(galvinoxylstyrene) as an n-type redox 204 active polymer and used the poly(TEMPO-substituted 205 norbornene) as the p-type one (Table 1, #18). The totally 206 organic polymer-based radical battery thus obtained 207 (also devoid of metals) gave an interesting power rate 208 capability since it retained 60% of its initial capacity 209 after 250 cycles at a 10C rate. Later, the same group 210 [27] using poly[4(nitronylnitroxyl)styrene)] assembled a 211 symmetric (poleless) cell exhibiting good cycle life (more 212 than 250 cycles) at a 60C rate (Table 1, #19). Deng et 213 al. [32] assembled a low-cost all-organic dual ion battery 214 (PF_6^{-}/Na^{+}) with poly(triphenylamine) as the p-type 215 material and poly(anthraquinonyl sulfide) as the n-type 216 one (Table 1, #20). The same anode was also used by 217 Cao and co-workers [33] together with a poly(aniline/o-218 nitroaniline) and a plastic crystal electrolyte (Table 1, 219 #21). Subsequently, Yang's group proposed to assemble 220 poly(triphenylamine) with poly(3,4-dihexylthiophene) 221 [34] (Table 1, #22). In another study, this group [35] ex-222 ploited the gap in potential between the p- and n-doping 223 processes occurring in poly(paraphenylene) (Table 1, 224 #23). However, compared to ORBs, such dual-ion cells 225 exhibit featureless cycling curves which resemble more 226 supercapacitor electrochemical profiles. Another dual-ion 227 all-organic battery consisting in poly(2-vinylthianthrene) 228 as the positive material and poly(2-methacrylamide-229 tetracyanoanthraquinodimethane) as the negative was 230 also investigated by Schubert's group [36]. Although 231 the output voltage was only 1.35 V, the cell was able 232 to sustain almost 70% of its initial capacity after 250 233 cycles (Table 1, #24). Very recently, Dong et al. [37] were 234 able to cycle at very low temperature (up to -70 °C) a 235 dual-ion cell based on poly(triphenylamine) and 1,4,5,8-236 naphthalenetetracarboxylic dianhydride (NTCDA)-237 derived polyimide thanks to an ethyl acetate-based 238 electrolyte (2 M LiTFSI) which exhibits sufficiently high 239 ionic conductivity at low temperature (Table 1, #25). 240

Lastly, the fourth section of Table 1 recaps the few 241 examples of cell combining two p-type electrode materials. In fact, such cells are scarcer because p-type 243 compounds are naturally characterized by high formal 244 redox potentials except the single family of viologenrelated materials. Note that Lee *et al.* studied in 246 the 90s polypyrrole/polyaniline (PAni) then PAni/PAni 247

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8 Batteries and Supercapacitors

full cells but reported data were poor [38]. Pal-248 more and co-workers [39] have prepared a polypyrrole 249 with covalently bonded viologen (4,4'-bipyridine) moi-250 eties as the anode and polypyrrole doped with 2,2'-251 azino-bis-(3-ethylbenzothiazoline-6-sulfonic acid) for the 252 cathode side, both releasing or accepting a perchlo-253 rate anion during the redox processes. Experimentally 254 this battery exhibited a very low capacity (Table 1, 255 #26). Yao's group proposed also such a molecular ion-256 based "rocking-chair" type battery [11**] with poly(N-257 vinylcarbazole) as the cathode and also a viologen poly-258 259 mer as the anode (poly(1,1'-pentyl-4,4'-bipyridinium dihexafluorophosphate)) (Table 1, #27). The performance 260 of the battery was much better with this last configura-261 tion giving rise to ~ 1.8 V as output voltage coupled with 262 capacity of 100 mA h/gcathode. 263

Promise and challenges of aqueous organicbatteries

Organics can also operate in aqueous electrolytes. Al-266 though at the expense of energy density, aqueous rocking-267 chair batteries constitute a novel and promising technol-268 ogy as they are inherently safe, minimize cost and envi-269 ronmental impact by comparison to other battery tech-270 nologies [5,40-42], which is particularly relevant for pro-271 moting low-cost energy storage solutions. This approach 272 is still affiliated however, with relatively low energy den-273 sity, below 50 W h/kg_{18650-cell}², which puts even more pres-274 sure on material costs and durability in order to ensure 275 economic viability. It has been recently shown that the 276 electrochemical window of superconcentrated aqueous 277 electrolytes can be expanded to 3V thanks to the for-278 mation of electrolyte-electrode interphase and unusual 279 water molecule coordination environment [40,43]. How-280 ever, it should be stressed as far as energy cost (\$/kW h) 281 is concerned this voltage gain is counterbalanced by the 282 mass ($\sim 2-5$ times that of a 1 M electrolyte) and the price 283 of these additional salts. Indeed, a bulk energy storage 284 unit will only be implemented if the cost per unit of en-285 ergy falls below 0.03 \$/kW h, a value lower than the cost 286 of electricity from conventional power sources. This im-287 288 plies that the device must be able to deliver many thousands of charge-discharge cycles over many years (which 289 for the time being rules out zinc-based systems) to in-290 sure storage cost remains in the vicinity of 100 \$/kW h. 291 Moreover, aqueous batteries are intended for both do-292 mestic and large-scale applications and therefore the enor-293 mous scale of the required energy transition places limits 294 on poorly abundant, non-uniformly distributed as well as 295 monopolized metal resources. To sum up, the relatively 296 narrow electrochemical window available in aqueous me-297 dia together with costs and abundance issues makes even 298 more challenging the development of appropriate host 299

materials with optimal potential as well as high chemical 300 and electrochemical stability. 301

One of the emerging approaches followed by several 302 groups consists also in considering organic active materi-303 als to substitute inorganic ones with the promise of abun-304 dancy in elements, lower costs and high structural des-305 ignability. Note that only five inorganic materials were 306 identified to design aqueous batteries (i.e., LiMn₂O₄ 307 [44], Na₃Ti₂(PO₄)₃ [44], Fe and Mn-based Prussian Blue 308 derivatives [45] and Na₃MnTi(PO₄)₃ [46]). Another de-309 cisive advantage is related to the potential access to su-310 perior specific capacities as organics benefit from multi-311 electron redox reactions. Lastly, the low volumetric den-312 sity of organic compounds is obviously not as detrimen-313 tal for stationary application as it is for mobile ones. 314 Despite these benefits, several key issues remain and re-315 cent literature proves the design of economically viable 316 full aqueous batteries based on organic materials is still 317 a challenging and exciting prospect. First, as mentioned 318 earlier for stationary storage the most important criterion 319 remains the overall cost of the storage device. Consider-320 ing a plausible voltage of 1.2 V and capacity of 150 mA h/g 321 for both the positive and the negative materials, a hypo-322 thetical 18,650 full cell should enable an energy density 323 of about 56 W h/kg_{18650-cell} based on 15 mA h/cm² elec-324 trodes containing 80% of active material. In these con-325 ditions, costs of goods should stay in the vicinity of 5-326 10 \$/kg and therefore any organic chemistry involved for 327 material design cannot exceed one or two steps (as a rule 328 of thumb, one step corresponds to \sim 5 \$/kg in the pig-329 ment industry). This also serves to reiterate that neutral 330 pH and molar range salt concentration of the electrolyte 331 should be preferred to minimize production costs and cor-332 rosion issues. The second bottleneck arises from the cor-333 relation between the potential range of the organic mate-334 rials depending on the p- or n-type character. For aque-335 ous batteries plethora of n-type organic materials can be 336 designed as anode materials allowing the access to poten-337 tials below -0.3 V vs. SCE (2.95 V vs. Li⁺/Li at neutral 338 pH) thanks to the carbonyl/enolate redox moiety. How-339 ever, with the aim to fabricate at least 1 V cell, it is quite 340 challenging to reach sufficiently high working potentials 341 with n-type materials for the cathode side (>0.4 V vs.)342 SCE or 3.65 V vs. Li⁺/Li at neutral pH). Inversely, ex-343 cept for the viologen group for which redox potentials can 344 fit the negative side, organic p-type redox centers match 345 better the positive side. Consequently, cationic rocking-346 chair aqueous batteries only exploit hybrid cells with inor-347 ganic(cathode)/organic(anode) combination. Yao's group 348 [47^{••}] (Table 1, #28) nicely illustrate the advantages of 349 organic active materials in this field. Indeed, their study 350 demonstrated polypyrene-4,5,9,10-tetraone (PPTO) can 351 store ~ 220 mA h/g (two fold what can be achieved by 352 best inorganic materials) at ~ -0.3 V vs. SCE enabling a 353 full cell with LiMn₂O₄ (LMO) as the cathode material 354 to sustain ~90 W h/kgmaterials for more than 3000 cycles at 355

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 $^{^2}$ The 18650 (18 mm by 65 mm) battery is a size classification of lithium-ion batteries.

0.23 A/g (1C \Leftrightarrow 3500 h cycling) with near to 100% coulom-356 bic efficiency. As a comparison, similarly loaded cells ex-357 ploiting LMO and LiTi₂(PO₄)₃ (LTP) lead to similar en-358 ergy density and cyclability but with somewhat higher 359 estimated cost since PPTO would be $\sim 10-15$ \$/kg and 360 LTP higher than 20 \$/kg [47]. Another interesting hy-361 brid cationic full cell from Ji's group [48] consisted in 362 using an ammonium inserted Ni-based Prussian white 363 as the cathode against a 3,4,9,10-perylenetetracarboxylic 364 diimide in a 1 M $(NH_4)_2SO_4$ (Table 1, #29). This is 365 the first aqueous cell exchanging a non-metal cationic 366 charge carrier. Although the capacity retention is mod-367 erate (67% upon 1000 cycles at 3C rates) it enables up 368 to 43 W h/kg_{materials} with 1 V of voltage at 1.5C based on 369 a two-fold excess of positive electrode. The sole full or-370 ganic aqueous cells to date incorporate n-type materials 371 372 only (anionic configuration). To our knowledge only two systems have been reported since 2012: Nishide and co-373 workers considered thin film batteries (up to 1 mm thick) 374 fabricated using a TEMPO derivative, the poly (2,2,6,6-375 tetramethylpiperidin-4-yl) acrylamide (PTMA) as the 376 cathode coupled to two different polyviologen deriva-377 tives, either highly cross-linked polyviologen hydrogel 378 (poly-(tripyridiniomesitylene)) [49**] which enable an av-379 erage voltage of ~1.3 V over 2000 cycles (Table 1, #30), or 380 to the poly(N-4,4'-bipyridinium-N-decamethylene dibro-381 mide) (Table 1, #31) which sustains more than 2000 cy-382 cles with 1.2 V average voltage [50]. Recently, Dong et al. 383 [51] proposed a full organic dual-ion cell based on p-type 384 polytriphenylamine and n-type-polynaphthalene diimide 385 polymers at the positive and negative electrode, respec-386 tively (Table 1, #32). Despite the cell requires the use of 387 388 a 21m LiTFSI water-in-salt electrolyte to prevent water oxidation, the authors showed near to 53 W h/kgmaterials 389 and 32 kW/kg_{materials} can be obtained for 1 mg/cm² elec-390 trodes. Another direction has been recently proposed by 391 our group by coupling p-type bipyridinium and n-type 392 naphthalene diimide redox moieties into one of a new 393 394 family of non-soluble oligomer for negative electrodes. The latter was shown to exchange both cations and an-395 ions simultaneously on cycling, therefore paving the way 396 to the design of a new type of dual cation-anion where 397 the salt concentration does not vary on cycling. The syn-398 ergistic coupling of the two redox units enables to reach 399 competitive capacities ranging from 60 to 90 mA h/g in 400 both neutral Na⁺ and Mg²⁺ electrolytes of molar range 401 concentration [4]. 402

403 Conclusions and outlook

The peculiar field of organic batteries has seen significant 404 progress these last few years with promising research di-405 rections attracting positively more and more interest from 406 the energy storage community. The unique features of 407 organics including flexibility, processability, structure di-408 versity as well as the true possibility of being prepared 409 from renewable resources and eco-friendly processes are 410 today substantive arguments even if practical energy den-411

sity values remain low. Let us recall that NEC group an-412 nounced ORBs close to reaching the market in 2012 [52]. 413 However, improvements are still needed to push forward 414 organic batteries especially to get a better stability upon 415 cycling. In fact, several organic materials are notably prone 416 to solubility issues, including some polymers. Thus NEC 417 have recently reported that the use of cross-linked PTMA 418 gels enable very good electrochemical performance com-419 pared to linear PTMA with ~ 100 mA h/g_{PTMA} for more 420 than 500 cycles [53]. This short review was also the oc-421 casion to underline that the richness of the redox or-422 ganic chemistry enables the development of both vari-423 ous innovative electrode materials and cell configurations. 424 Aqueous organic batteries appear notably as promising 425 devices for stationary electricity storage at the condition to 426 have low production costs and long cycling stabilities. In 427 this regard, the high ionic conductivity of aqueous elec-428 trolytes that allows in principle to use ultra-thick elec-429 trodes should enable to pull the price per unit of energy 430 even lower [54]. This aspect could be all the more impor-431 tant that the price of actual organic active materials re-432 mains too high (>5-10 \$/kg). Although it was beyond the 433 scope of this article, it is worth noting that very promis-434 ing results have also been reported in regard to the use 435 of organic redox materials for the redox flow technology 436 [55-57]. 437

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