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## Mechanical Response and Damage of Woven Composite Materials Reinforced with Fique

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**Abstract.** In this paper, we present the experimental and numerical modelling for the mechanical behaviour of woven composites reinforced with fique (*furcraea selloa*) fibre, for different fique fibre woven configurations embed in an R744 epoxy matrix. The woven configurations are taken from commercial models and their mechanical properties validated by experimental data. We perform experimental tests using ASTM D3039 for the tensile response. We obtain values for Young's modulus, ultimate strength, and deformation of unidirectional and woven reinforced composites. Scanning electron microscopy (SEM) is used for the fractographic analysis of the reinforced specimens. For the numerical model of the woven composite, we use the Texgen software to define the finite element voxel model and to estimate orthotropic mechanical parameters. Then, we determine the equivalent elastic properties of the composite, according to the materials and the fibrematrix relations. We compare and validate the numerical results with the experimental data. We obtain stress and strain fields for the yarns and the matrix. The objective of this work is to establish a baseline of the mechanical behaviour of these natural reinforced composites to propose applications for structural engineering.

#### Introduction

In recent decades, the use of natural fibres has emerged as a response to the need for sustainable composites materials. Composites reinforced with natural fibres have gained attention among the industrial an academic communities, and efforts to characterize and evaluate their mechanical behaviour have been done for several fibres, e.g., jute, sisal, fique, etc. [1]–[3], mostly for unidirectional laminates. Low environmental impact, low cost and a wide range of applications are some of their unique features [4], [5], [1], [6]. To optimize the performance of natural fibre composites, we can improve the adhesion between the fibres and the polymer matrix, or develop appropriate geometries and configurations of the reinforcement phase, which contribute to increase the mechanical properties of the composite [7]–[9].

Composites materials reinforced with woven fibres are a good alternative in many applications, and they can be found in several configurations depending on the textile used, e.g., 2D laminated textile, 2D braided textile, etc. In this regard, Wang et al. [10] have developed methods for woven fibres to use RVE with periodic boundary conditions for finite element analysis in micro and meso scales.

Reinforcement textiles can be recreated in different manners. Texgen software can reproduce the hierarchical structures and the geometric patterns of complex textiles [11], [12]. Numerical models using finite elements (FE) can be used to represent the mechanical response of composites with

textiles [13]–[16]. The mesoscale modelling of textile composites is used to homogenize mechanical properties by evaluating the stress and strain fields inside a representative unit cell. The conditions for failure start and failure development, associated with damage properties of the materials, elastic properties and geometric characteristics may be proposed based on unit cells [17], [18].

In this work, we investigate the mechanical response of a polymer-matrix composite reinforced with natural fibres of fique in the form of woven textile. The configuration of the natural textile was taken from commercial models and mechanical properties were obtained by ASTM tensile tests. Numerical models to predict the elastic constants are used, both considering the general rule of mixtures and finite element models, for different fibre configurations, showing good agreement with experimental results.

### Materials and methods

The classical linear elasticity problem is considered, given by the internal equilibrium equation, and Dirichlet and Neumann boundary conditions. We assume materials with linear elastic behaviour, under the generalized Hooke's law for anisotropic materials [19]. The textile architecture is modelled using the TexGen software [12], in two different configurations. We assume that both the yarn and the matrix behave as isotropic bodies.

Fique natural fibres (*furcraea andina*), supplied by Coohilados del Fonce Ltd., were used. The fique fibre has a density of 0,72 g/cm<sup>3</sup>, for a linear density of 31,32 tex, with an average cross-section of 0,176 mm<sup>2</sup> and a diameter of 0,473 mm [20]. Applying descriptive statistics, it was found that the yarn used is formed of approximately 29 fibres with an estimated diameter of 0.016 mm and an apparent thread diameter of 1.444 mm. Elastic modulus for fique is 22,06 GPa (measured with a fibre length of 250 mm), shear modulus 8,82 GPa and Poisson ratio 0,25, with an ultimate tensile stress of 780 MPa. These values were obtained experimentally for a single thread. The epoxy resin R744 was assumed as a homogeneous isotropic material, with an elastic modulus of 0,951 MPa. One of the textiles used is shown in Fig. 1. The dimensions of the models are 250x25x10 mm, the thread that forms the fabric is 1.44 mm in diameter.

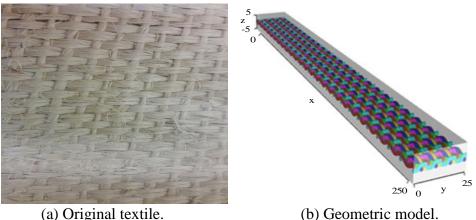


Figure 1. Model of the textile  $2 \times 1$ .

### Results

To determine the mechanical properties of the proposed composites, tensile tests were carried out on a universal MTS BIONIX testing machine with a load cell of 25 kN; 3 types of specimens were considered, unidirectional (UD), textile 1 and textile 2. The procedure and the size of the samples were made according to ASTM D3039, the specimens are shown in Fig. 2.

In Table 1, we show the results of the tensile tests after the corresponding data treatment for each type of specimens. Note the effect of the reinforcement on the properties of the material, even for low volume fractions. Ultimate stress,  $\sigma_u$ , and Young's modulus,  $E_1$ , increase with the number of threads, and the strain,  $\varepsilon_{max}$ , decreases, which is consistent with the expected material response. For the samples with reinforced woven fibres, it is observed that Young's modulus and ultimate stress increase, and the strain decreases, compared with the epoxy specimen.

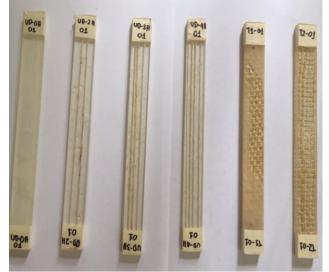


Figure 2. Specimens for the tensile tests.

Table 1. Mechanical properties from tensile tests for epoxy and composite configurations.

Specimen type	σu. [MPa]	ε <sub>max</sub> [mm/mm]	E <sub>1</sub> [GPa]
Epoxy	25,237 <u>+</u> 1,218	7,470 <u>+</u> 0,316	0,951 <u>+</u> 2,044
UD, 2 threads	28,803 <u>+</u> 1,135	2,975 <u>+</u> 0,022	1,417 <u>+</u> 1,440
UD, 3 threads	28,137 <u>+</u> 2,517	2,617 <u>±</u> 0,126	1,528 <u>+</u> 0,727
UD, 4 threads	30,477 <u>+</u> 1,518	2,783 <u>+</u> 0,015	1,658 <u>+</u> 1,181
Textile 1	28,150 <u>+</u> 1,603	2,938 <u>+</u> 1,438	1,587 <u>+</u> 0,797
Textile 2	30,254 <u>+</u> 1,155	3,420 <u>+</u> 0,285	1,509 <u>+</u> 0,561

For the fractographic analysis, images were obtained by scanning electron microscopy (SEM), using a Quanta FEG 650 microscope. Note in Fig. 3 that the failure mode for the tensile tests indicates a good adhesion between the fibre and the matrix, which improves the properties of the material [2]. Impregnation of the matrix between the fibres of the threads of the textile is optimal, which indicates an appropriate process of casting and curing [21]. The fibre pull-out of the warp could be due to the lack of fibre preparation treatments, which are used to detach the layers of hemicellulose and lignin from its surface, to improve the adhesion between fibre and matrix. Also, small marks left by the weft are observed. Despite the minimal extraction, it can be observed that the material failed uniformly, little evidence of delamination is observed [22], [23].

**Finite element model.** A conventional Galerkin formulation is used for the finite element model of the linear elastic problem. For the topological model, a voxelized mesh was used, which was obtained using TexGen [24]. The boundary conditions are consistent with the ASTM tensile tests.

**Elastic model of unidirectional and woven fibres**. Initially, we obtained the elastic constants for the composite material with unidirectional fibres: orthotropic elastic moduli  $E_1$ ,  $E_2$ , shear modulus  $G_{12}$ , and Poisson coefficients  $v_{12}$ ,  $v_{21}$ , Table 2.



Figure 3. SEM results for the textile 1 under tensile loads.

Table 2. Elastic constants for unidirectional fibres considering a different number of threads.

Threads	E <sub>1</sub> [GPa]	E <sub>2</sub> [GPa]	G <sub>12</sub> [GPa]	$v_{12}$	$\nu_{21}$
2	1,346	0,968	0,394	0,348	0,250
3	1,544	0,977	0,401	0,347	0,220
4	1,741	0,986	0,408	0,346	0,196

We use the rule of mixtures to estimate the elastic modulus for the different volume fractions. Table 3 shows the error between the experimental,  $E_{1,exp}$ , and the estimated value,  $E_{1,RM}$ . Then, the elastic modulus obtained by finite elements,  $E_{FE}$ , was evaluated for each model, including the textiles. The error with respect to the values obtained experimentally is small, as shown in Table 3.

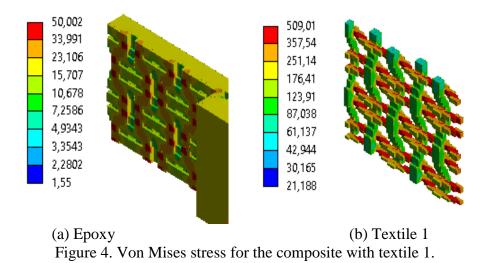
Specimen type	E <sub>1,exp</sub> [GPa]	E <sub>1,RM</sub> [GPa]	% error	E <sub>FE</sub> [GPa]	% error
UD de 2 hilos	1,417	1,346	5,01	1,376	2,89
UD de 3 hilos	1,527	1,544	1,11	1,461	4,36
UD de 4 hilos	1,657	1,741	5,06	1,629	1,69
Tejido 1	1,587	N/A	N/A	1,693	6,67
Tejido 2	1,509	N/A	N/A	1,457	3,44

Table 3. Comparison of elastic modulus for experimental, rule of mixtures and FE models.

With the voxelized FE model, a load of 8000 N was applied to obtain the stress distribution for each material. Fig. 4 shows the von Mises stress distribution for textile 1, displaying the reinforcement fabric and the epoxy matrix. As expected, the warp of the textile exhibits the highest stress, close to 500 MPa, while the transverse threads, weft, show compression stress close to the intersection with the warp. For the epoxy, the stress is high in the spaces between the longitudinal threads, which could cause separation of the matrix from the transverse threads. Considering that the cross-section of the model has an area of 250 mm<sup>2</sup>, the nominal stress is 32 MPa. Thus, the composite fails because the stress on the matrix exceeds the ultimate stress.

### Conclusions

Tensile tests were done according to ASTM D3039 for different composite configurations. Epoxy R744 was reinforced with woven fique fibres and an increase in Young's modulus up to 66.8% was evidenced. Using voxel FE models, we could approximate the stress distributions for the textile and matrix. Longitudinal fibres show higher values of stress, as expected. The intersection with transverse fibres shows high stress values for the epoxy. According to FE results, the failure mechanism might be associated with the areas where the matrix exceeds the allowable stress.



Numerical results using FE models presented differences below 6.8% with respect to the data obtained experimentally, for the elastic constants of the material. This indicates that the model has good agreement with experimental values, and could be used to study other composite configurations and volume fractions.

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