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$K_xNa_{1-x}NbO_3$ perovskite thin films grown by pulsed laser deposition on R-plane sapphire for tunable microwave devices

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Abstract

$K_xNa_{1-x}NbO_3$ thin films with $x = 0.5$ and $x = 0.7$ were deposited by pulsed laser deposition onto R-cut sapphire substrates to be suitable for microwave applications. The 500-800 nm-thick films present a preferential (100) orientation. The ω -scans show a weak mosaicity (full-width-at-half-maximum equal to 0.36° and 0.60° for $x = 0.5$ and $x = 0.7$, respectively). In addition to this texture, the in-plane ordering evidenced by X-ray diffraction φ -scan for the (100) orientation is in agreement with an epitaxial-like growth in spite of the high lattice mismatch between $K_xNa_{1-x}NbO_3$ and sapphire. The dielectric characteristics and the frequency tunability at microwave frequencies were obtained from coplanar waveguide devices (transmission lines and stub resonators). For the $K_{0.5}Na_{0.5}NbO_3$ and $K_{0.7}Na_{0.3}NbO_3$ compositions, high dielectric permittivity ϵ_r values of 360 and

250, and loss tangent $\tan\delta$ values of 0.36 and 0.43 without biasing were retrieved from the transmission lines measurements at 10 GHz, respectively. Frequency tunabilities of 15 % and 12 % have been assessed under 80 kV/cm biasing from stub resonator measurements for the $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ and $\text{K}_{0.7}\text{Na}_{0.3}\text{NbO}_3$ compositions, respectively. $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ composition is therefore a promising solution for miniaturized tunable devices at microwave frequencies.

Keywords: ferroelectrics; thin films; tunable devices; microwave frequencies.

1. Introduction

From the last decade, a great interest is devoted to multifunctional oxides that present a wide range of properties, and offer new opportunities in the development of efficient devices. In this frame, ferroelectric materials, which exhibit also piezoelectric behavior, have been extensively studied due to their numerous applications in sensors, micro-electro-mechanical systems (MEMS) and memory devices [1–4]. Nowadays, wireless applications require advanced miniature and tunable microwave devices [5–7]. Besides the standard reconfigurable devices based on radiofrequency micro-electro-mechanical systems (RF-MEMS) or solid-state components, integration of ferroelectric materials brings an alternative solution. Indeed ferroelectric thin films exhibit a high dielectric permittivity ϵ_r which can be controlled by an external static electric field, promoting both miniaturized and reconfigurable microwave devices. Besides the most popular ferroelectric materials, namely $\text{Ba}_x\text{Sr}_{1-x}\text{Ti}_x\text{O}_3$ (BST) [8, 9] and $\text{PbZr}_{1-x}\text{Ti}_x\text{O}_3$ (PZT) [10, 11], the lead-free potassium niobate family presents relevant features [12–14].

Currently, among the lead-free ferroelectric materials, a specific attention is paid to the perovskite-type oxide $\text{K}_x\text{Na}_{1-x}\text{NbO}_3$ (KNN). The KNN compositions belong to a solid solution between the ferroelectric KNbO_3 (orthorhombic, space group (SG): $Bmm2$, JCPDS file # 01-071-2171) and the antiferroelectric NaNbO_3 (orthorhombic, SG: $P2_1ma$, JCPDS file # 01-077-0873) [15, 16]. Through the solid-solution, the KNN compounds present various crystalline symmetries. Only

2 % atomic of K is sufficient to modify the structure of NaNbO_3 [17]. At room temperature, for x lying between 0.48 and 0.83, the KNN compounds have a primitive monoclinic cell with $a_m \approx c_m \approx 4 \text{ \AA} > b_m$, and β slightly higher than 90° . The phases can also be described by a larger A-centered orthorhombic cell (i.e. $a_0 \approx c_0 \approx 4\sqrt{2} \text{ \AA}$) [18, 19]. Tellier *et al.* and Handoko *et al.* have shown that the lattice parameters a_m and c_m increase with the potassium content. This increase is explained by the larger K^+ radius (1.33 \AA) against the Na^+ one (0.97 \AA) [18, 19].

The KNN oxide exhibits (i) high Curie temperature ($T_c \approx 400^\circ\text{C}$) depending upon its composition [17, 20], (ii) high values of dielectric permittivity ϵ_r (between 100 and 500 at 100 kHz [15]), and (iii) high piezoelectric coefficient d_{33} (up to 300 pC/N [21]). This range in T_c value is valuable to ensure the strong thermal stability of the targeted microwave devices operating at room temperature. Recent studies were reported at microwave frequencies on KNN ceramics and thin films [22–26]. As expected, the dielectric permittivity varies according to the bulk composition from $\epsilon_r = 92$ for $\text{K}_{0.2}\text{Na}_{0.8}\text{NbO}_3$ to $\epsilon_r = 124$ for $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ at 10 GHz [23]. On one hand, Peddigari *et al.* reported values of $\epsilon_r = 287$ and loss tangent $\tan\delta = 0.01$ at 10 GHz on $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ thin films [25]. On the other hand, Kim *et al.* reported the value $\tan\delta = 0.23$ at 20 GHz on $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ thin films (ϵ_r value not available) with a capacitance tunability of 22 % at 20 GHz under an external biasing field $E_{bias} = 200 \text{ kV/cm}$ [26]. According to these preliminary studies, compositions with the lowest potassium/sodium ratio are not identified as promising candidates for microwave applications. This explains why the $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ composition has been first selected here. Moreover, its proximity with the reported morphotropic phase boundary [27], defined as continuous-phase transitions between two crystalline structures, augurs well for interesting ferroelectric properties at microwave frequencies [28]. A second composition with a higher potassium/sodium ratio ($\text{K}_{0.7}\text{Na}_{0.3}\text{NbO}_3$) has also been studied here.

The main objective of the present work is to investigate the microwave response of tunable devices based on KNN thin films. Two compositions $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ and $\text{K}_{0.7}\text{Na}_{0.3}\text{NbO}_3$ were

investigated in detail. The microstructure and the crystalline characteristics of the thin films grown by pulsed laser deposition on R-plane sapphire substrates were determined and the related KNN-based tunable device performances at microwave frequencies were compared.

2. Experimental details

2.1. Elaboration of KNN thin films by pulsed laser deposition (PLD)

The ceramic targets were prepared from KNN powders first synthesized by a solid state reaction at 940°C for 4 hours in air from sodium carbonate Na_2CO_3 (RP NORMAPUR 99.8%), anhydrous potassium carbonate K_2CO_3 (ACROS Organics 99 %) and niobium oxide Nb_2O_5 (Alfa Aesar 99.5%) powders mixed in a planetary ball-mill for 1 hour at a 400 rotations per minute speed (PM100, Retsch). In a second step, the powders were K-enriched with a potassium nitrate KNO_3 powder, to balance the loss of potassium during the laser deposition due to its volatility. Afterwards the powders were uniaxially pressed into 25 mm-diameter targets and annealed up to 500°C in air for 3 hours.

The KNN thin films were deposited by the pulsed laser deposition (PLD) technique from two KNN targets. By controlling the K/Na ratio of the home-made targets, two thin film compositions were prepared: $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ (KNN50/50) and $\text{K}_{0.7}\text{Na}_{0.3}\text{NbO}_3$ (KNN70/30). Each thin film was grown on R-plane sapphire ($1\bar{1}02$) Al_2O_3 (10 mm × 10 mm × 0.5 mm) single crystal previously cleaned in acetone and isopropanol ultrasonic baths for 5 min each. Despite the strong mismatch with the perovskite-like structures, sapphire substrate fits well within microwave applications, thanks to its convenient dielectric characteristics ($\epsilon_r \approx 10$ and $\tan\delta \approx 10^{-4}$ at 10 GHz and room temperature). Moreover sapphire substrate is highly stable in air and moisture atmosphere, then well suitable for further lithography and wet etching processes. A KrF excimer laser (Coherent company, pulse duration 20 ns, $\lambda=248$ nm) set at an energy of 210 mJ, a fluence of 2-3 $\text{J}\cdot\text{cm}^{-2}$ and a 4 Hz working frequency, was used to ablate the KNN ceramic targets. The KNN films were grown onto

sapphire substrates heated at 650°C and located at 55 mm in front of the selected target, under a 0.3 mbar oxygen pressure. The expected deposition rate being close to 15 nm/min, a deposition time of 40 minutes was used to obtain a thin film thickness of about 600 nm.

2.2. KNN thin film characterizations

The structural characteristics were investigated by X-ray diffraction (XRD) by using a θ - 2θ diffractometer (D8 Advance, BRUKER) with a monochromatized $Cu K_{\alpha 1}$ wavelength ($\lambda=1.54056 \text{ \AA}$) and a texture diffractometer (D8 Discover, BRUKER) giving access to the θ - 2θ , ω -scan and φ -scan modes. The surface morphology was observed at 10 kV by scanning electron microscopy (SEM, *JSM 7100F*, JEOL) coupled with an energy dispersive X-ray spectroscopy detector (EDS, *SDD X-Max 50mm²*, Oxford Instruments) to determine the thin film composition. A 5 nm-thick carbon layer was deposited before SEM observation to ensure optimal element detection and especially the potassium element detection that we have observed to be highly sensitive to charge accumulation effect. Each film thickness was measured by SEM on a transverse section obtained from a fracture along an edge.

2.3. KNN microwave devices

The dielectric characteristics and the frequency tunability of the KNN thin films were investigated using coplanar waveguide (CPW) devices designed with a 3-D commercial electromagnetic software (Ansys HFSSTM). Three 50- Ω transmission lines (8 mm, 5 mm and 3 mm long) with a width $w_l = 40 \text{ }\mu\text{m}$ surrounded by a gap $g_l = 50 \text{ }\mu\text{m}$ (Figure 1) were implemented to retrieve the KNN dielectric characteristics, namely the dielectric permittivity ϵ_r and the loss tangent $\tan\delta$, from the microwave measurements. A CPW quarter-wavelength open-ended stub resonator was also designed to operate in X-band (Figure 1). Its resonance frequency F_r depends upon the stub length L_r and the effective permittivity ϵ_{eff} of the heterostructure, as follows:

$$F_r = \frac{c}{4L_r\sqrt{\varepsilon_{eff}}} \quad (1)$$

where c is the speed of light in vacuum. The devices were fabricated from a 2 μm -thick silver overlayer / 5 nm-thick titanium underlayer deposited by RF magnetron sputtering at room temperature. The silver thickness is three times larger than the skin depth value ($\delta = 0.64 \mu\text{m}$) at 10 GHz. Note that the ultra-thin titanium film is used here only to ensure the strong adhesion of the silver metallization onto the ferroelectric oxide. Standard photolithography and wet etching processes were implemented to pattern the devices. Gold wire bondings (15 μm -diameter and 250 μm -length) were used to enforce the equipotential condition on both CPW grounds and prevent from the excitation of parasitic slotline mode.

Microwave measurements at room temperature were carried out through a probe station coupled to a vector network analyzer (*VNA 8510C, Hewlett Packard*) from 1 GHz to 20 GHz under an external *DC* bias voltage range from 0 V to 150 V, which corresponds to a maximal static electric field E_{bias} equal to 80 kV/cm for the stub resonator and 27 kV/cm for the transmission lines. The measured gap dimensions ($g_r \approx 18 \mu\text{m}$ and $g_l \approx 54 \mu\text{m}$) are slightly larger than the nominal ones (Figure 1) due to the inherent overetching of the wet etching process used. Further details are reported elsewhere [29].

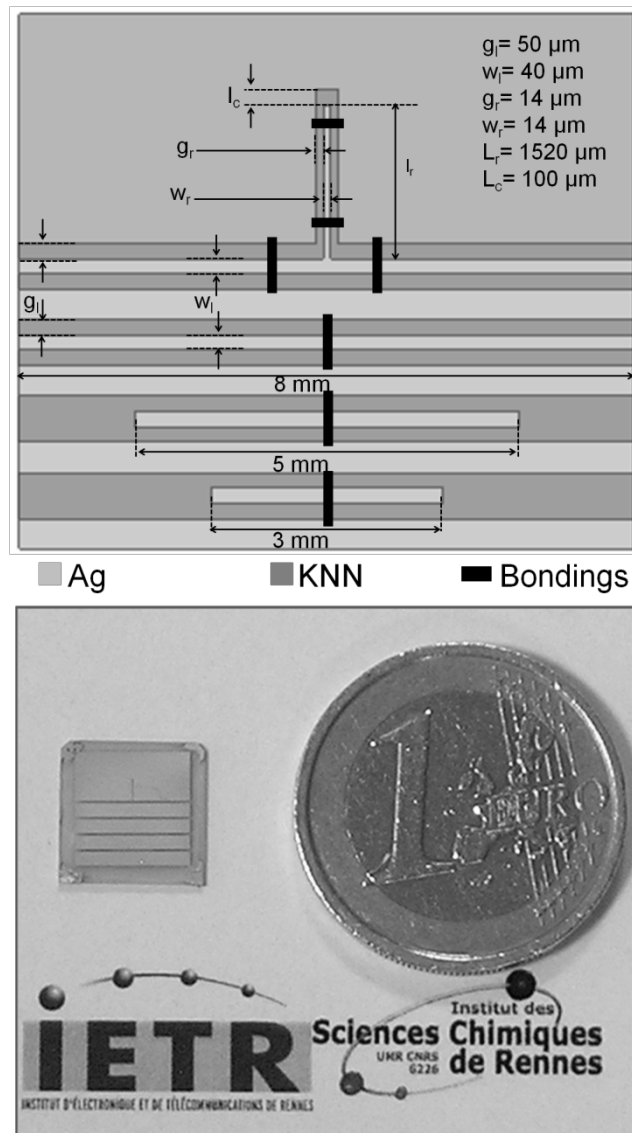


Fig. 1 Layout, dimensions and picture of the CPW devices

3. Results and discussion

3.1. Composition, microstructure and structural properties of the KNN thin films

The volatility of potassium, which is higher than that of sodium [30], greatly influences the global stoichiometry of the thin films at the end of the deposition process. Consequently, as described in the experimental details section, the two home-made targets have been K-enriched. The EDS analyses of the targets and of the related thin films (Table 1) depict this volatility. Even after a strong enrichment of the target, the $K_{0.5}Na_{0.5}NbO_3$ thin film may exhibit a slight lack of

alkaline elements with a ratio of (K+Na)/Nb typically in the range ~ 0.95 -1. This lack of alkaline elements may lead to an oxygen deficiency for charge compensation. The related oxygen vacancies may be responsible for decreasing the piezoelectric and dielectric properties in some ferroelectric oxides [31]. Here, the reported compositions are average values obtained from three measurements performed on different areas of each sample. The achievement of a standard deviation lower than 2% is affording evidence tending to prove the good homogeneity and reproducibility of the deposition method used. Nevertheless the overall accuracy of EDS technique itself is usually considered close to 10%, especially in the case of resistive materials in thin films. These results enable us to confirm the so-called KNN50/50 thin film composition close to $K_{0.5}Na_{0.5}NbO_3$ and that of the KNN70/30 thin film close to $K_{0.7}Na_{0.3}NbO_3$.

Figure 2 displays the regular morphology of the thin films. The homogeneous shape of the grains and their arrangement suggest a preferred orientation. Some outgrowths currently observed at the surface of thin films deposited by PLD, especially with not highly densified targets, are visible with a lateral size that is still submicronic. The KNN50/50 sample exhibits a thickness slightly lower than the targeted value: 510 nm against ~ 600 nm (Figure 2a) in contrast with the KNN70/30 sample which presents a thickness equal to 780 nm (Figure 2b). The thickness difference between the two compositions has been systematically obtained with other samples (not shown here).

Sample	Composition ratio of the targets (K : Na : Nb)	Composition ratio of the thin films (K : Na : Nb)	Thin film thickness (nm)	Cell parameters (\AA)
KNN50/50	0.9 : 0.5 : 1	0.51 : 0.47 : 1	510	$a_m \approx c_m \approx 4.00$ $b_m \approx 3.95$
KNN70/30	1.4 : 0.3 : 1	0.71 : 0.29 : 1	780	$a_m \approx c_m \approx 4.02$ $b_m \approx 3.96$

Table 1 Composition, thickness and measured cell parameters of the thin films versus the target composition

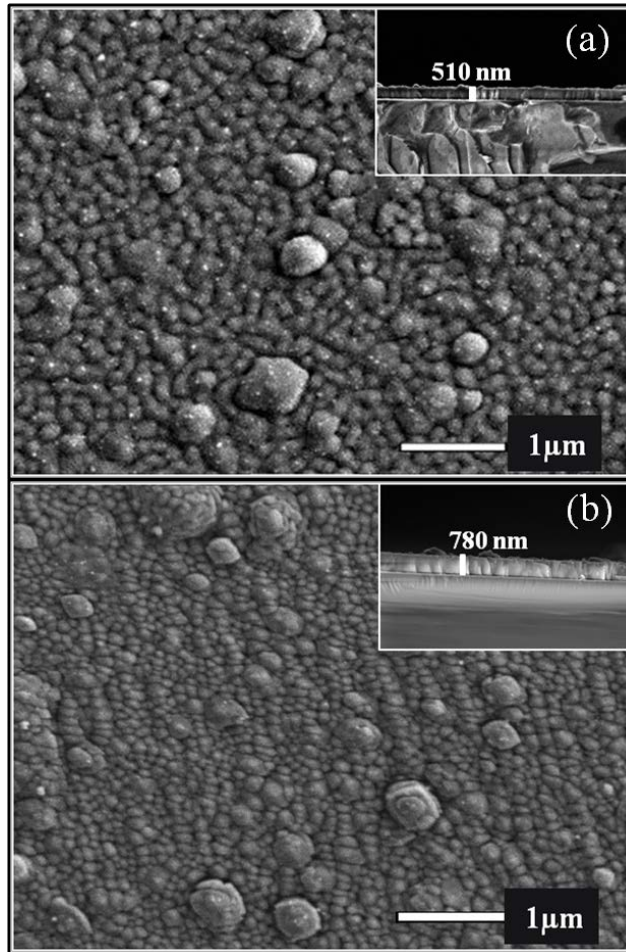


Fig. 2 SEM images of the surface morphology and cross-sections (in inset) of (a) KNN50/50 and (b) KNN70/30 thin films

The XRD patterns of the KNN50/50 and KNN70/30 samples are depicted in Figures 3a and 3b using the monoclinic indexation and displayed with a logarithmic intensity scale. A powder XRD pattern available from the database (JCPDS file # 00-061-0315) is displayed as reference in Figure 3c [18]. It is worth noting that deficiency in alkaline elements often leads to the presence of the unwanted $K_4Nb_6O_{17}$ phase (JCPDS file # 01-076-0977), leading to a peak at $2\theta \approx 10^\circ$ on XRD patterns [32]. For our samples, all the peaks are purely indexed in the KNN phase, demonstrating the absence of such secondary phases. The two peaks at $2\theta \approx 22^\circ$ correspond to the diffraction of the (100) and (010) KNN planes. A strong (100) orientation is observed on both KNN50/50 and KNN70/30 films with a second weaker (110) orientation at $2\theta \approx 32^\circ$. The relative intensities of the (100), (010) and (110) peaks are similar for both samples. The orientation factors of the (100) peak,

computed by the Lotgering method [33], comparing the relative intensity of the peaks from the experimental XRD patterns (Figures 3a and 3b) to the powder XRD patterns available from the database, are equal to 0.93 and 0.90 for the KNN50/50 and KNN70/30 films, respectively. Therefore both samples can be considered as structurally alike. The slight shift of the KNN70/30 2θ peaks towards large angles is attributed to the lattice parameters increase with K amount increase in the compounds (Table 1). These results are in agreement with the cell parameter values reported by Tellier *et al.* and Hondoko *et al.* of $a_m = 4.004 \text{ \AA}$ and $c_m = 3.944 \text{ \AA}$ for $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$, and $a_m = 4.01 \text{ \AA}$ and $c_m = 3.97 \text{ \AA}$ for $\text{K}_{0.7}\text{Na}_{0.3}\text{NbO}_3$ [18, 19].

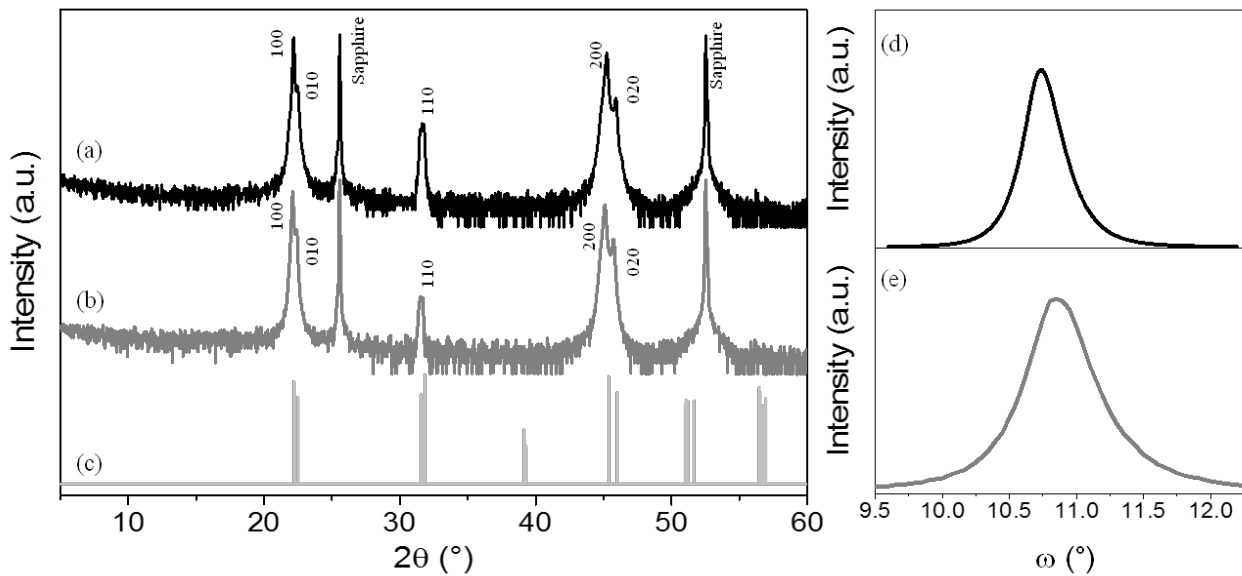


Fig. 3 θ - 2θ X-ray diffraction patterns displayed with a logarithmic intensity scale of (a) KNN50/50 thin film, (b) KNN70/30 thin film (c) $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ powder retrieved from the database JCPDS file # 00-061-0315 [18]; ω -scans on the (100) peak of (d) KNN50/50 and (e) KNN70/30 thin films

To further investigate the crystalline quality of the samples, ω -scans on the (100) peak (Figures 3d and 3e) were recorded. The full-width-at-half-maximum (FWHM) is equal to 0.36° and 0.60° for the KNN50/50 and KNN70/30 samples, respectively. Those values are a good indication of the well-textured thin films. The ω -scan on the (1 $\bar{1}$ 02) sapphire substrate gives $\text{FWHM} = 0.02^\circ$ for comparison. The φ -scans were also performed on the KNN {110} planes (Figure 4). They present four distinctive peaks at 90° apart from one another, which indicate an in-plane ordering

which refers to an epitaxial-like growth for this orientation, besides a significant background noise corresponding to the texture. This result is in good agreement with the microstructure previously observed by SEM (Figures 2a and 2b). Though examples of epitaxial growth of perovskite on sapphire exist [34, 35], such crystalline growth was unexpected due to the large mismatch between the KNN perovskite phase and the R-plane sapphire ($\sim 14\%$) and, at our knowledge, has not been reported yet.

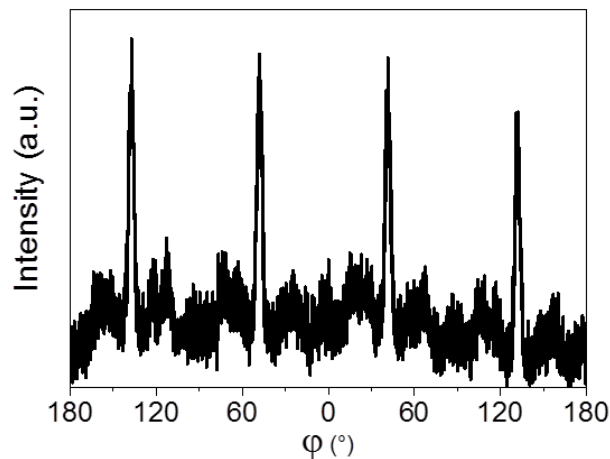


Fig. 4 φ -scans performed on the KNN {110} planes typical of KNN50/50 and KNN70/30 thin films deposited on sapphire substrate

3.2. KNN dielectric characteristics

The dielectric characteristics (ϵ_r , $\tan\delta$) of the KNN thin films were retrieved from the measured reflection (S_{11}) and transmission (S_{21}) coefficients of the CPW transmission lines (Figure 5) [29, 36]. Without biasing, KNN50/50 exhibits a permittivity value $\epsilon_r = 360$ at 10 GHz against 250 for KNN70/30. Permittivity value of KNN50/50 is higher than those previously reported on Dy₂O₃-doped KNN thin films ($\epsilon_r = 307$ at 10 GHz [37]) and on pure K_{0.5}Na_{0.5}NbO₃ thin films grown on Pt/Ti/SiO₂/Si ($\epsilon_r = 287$ at 10 GHz [25]). The higher permittivities measured in the present study can be attributed to the crystalline orientation, *i.e.* epitaxial-like growth, of our samples, compared with the KNN polycrystalline growth mentioned by Peddigari *et al.* [25]. The loss is also

higher than that expected for this kind of materials: $\tan\delta = 0.36$ for KNN50/50 and $\tan\delta = 0.43$ for KNN70/30 without biasing at 10 GHz, whereas Peddigari *et al.* reported $\tan\delta = 0.014$ at 10 GHz for the Dy₂O₃-doped KNN thin films [37] and $\tan\delta = 0.010$ at 10 GHz for the annealed KNN thin films [25]. As PLD technique induces outgrowths into KNN thin films (Figures 2a and 2b), the interface quality between the metallic transmission lines and the KNN films may have an impact on the retrieved $\tan\delta$ values. This contribution to the measured loss, not present in the split post dielectric resonator (SPDR) [38] technique, could explain the $\tan\delta$ values measured in the present study. Indeed our results can be compared with those reported by Kim *et al.* [26], *i.e.* $\tan\delta = 0.23$ at 20 GHz measured on a coplanar interdigitated capacitor. Under $E_{bias} = 27$ kV/cm, the dielectric permittivity decreases from 360 to 320 for KNN50/50 and from 250 to 220 for KNN70/30 at 10 GHz, demonstrating the tunability of such compositions under biasing. The important decrease of the dielectric permittivity from 1 to 5 GHz with and without biasing is attributed to the freezing of the polar domain motion, whereas $\tan\delta$ fluctuations in this frequency range are attributed in part to the metal skin depth loss [29].

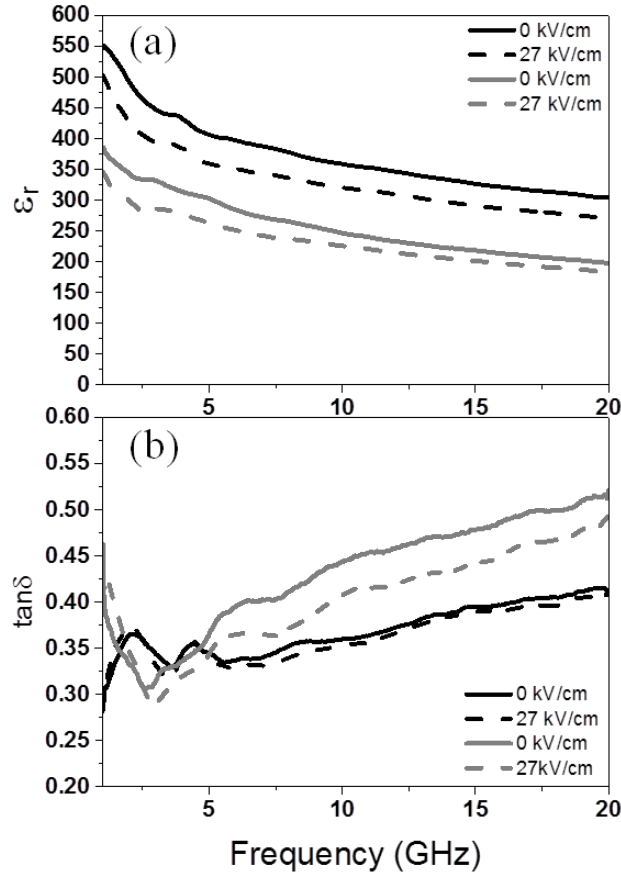


Fig. 5 Measured frequency dependence of (a) dielectric permittivity ϵ_r and (b) intrinsic loss $\tan\delta$ with and without biasing; Black line: KNN50/50; Grey line: KNN70/30

Resonance frequencies F_r (frequency value at the minimum S_{21} , see Figure 6) of the CPW stub resonators printed on the KNN50/50 and KNN70/30 samples were measured under various bias voltage values. The fine tuning of the KNN dielectric permittivity value under bias voltage controls F_r . The frequency tunability T (Figure 7) is computed from the F_r values under E_{bias} , as follows:

$$T(\%) = \frac{|F_r(E_{bias}=0) - F_r(E_{bias})|}{F_r(E_{bias}=0)} \times 100 \quad (2)$$

The maximum external DC electric field applied is $E_{bias} = 80$ kV/cm. The shift of the resonance frequency from 12.7 GHz (without biasing) to 14.6 GHz (E_{bias} max.) and from 12.0 GHz (without biasing) to 13.4 GHz (E_{bias} max.) leads to a frequency tunability equal to 15% and 12% for the KNN50/50 and KNN70/30 samples, respectively. More detailed results on tunability values under

various bias voltages are reported in Figure 7. Clearly the highest tunability is reached with the KNN50/50 sample, especially when the applied bias voltage is high.

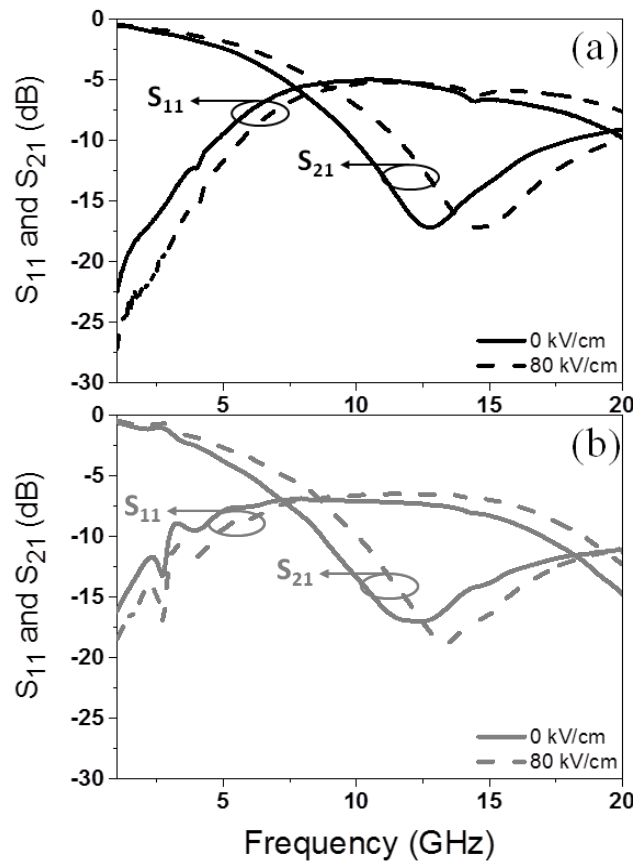


Fig. 6 Measured reflection coefficient S_{11} and transmission coefficient S_{21} with and without biasing on the stub resonator printed on (a) KNN50/50 and (b) KNN70/30

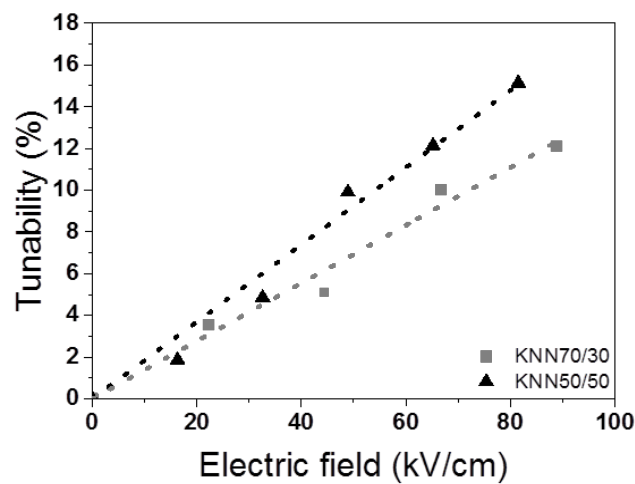


Fig. 7 Frequency tunability under biasing. \blacktriangle KNN50/50; \blacksquare KNN70/30; the black and grey dotted lines are guide eyes

The global loss (GL) of the resonant devices was also computed from S_{11} and S_{21} magnitude values at their specific resonance frequency F_r , as follows:

$$GL = 1 - |S_{11}|^2 - |S_{21}|^2 \quad (3)$$

GL is plotted in Figure 8 from 1 to 20 GHz with and without E_{bias} for both samples. The computed values without biasing for KNN50/50 ($GL= 0.59$) and KNN70/30 ($GL= 0.76$) are of the same order of magnitude, with a slightly higher value for KNN70/30. The decrease of GL under biasing (Table 2) is consistent with the related loss $\tan\delta$ decrease for the KNN 70/30 sample. Regarding the KNN50/50 one, $\tan\delta$ steadiness under biasing explains the non-variation of the GL .

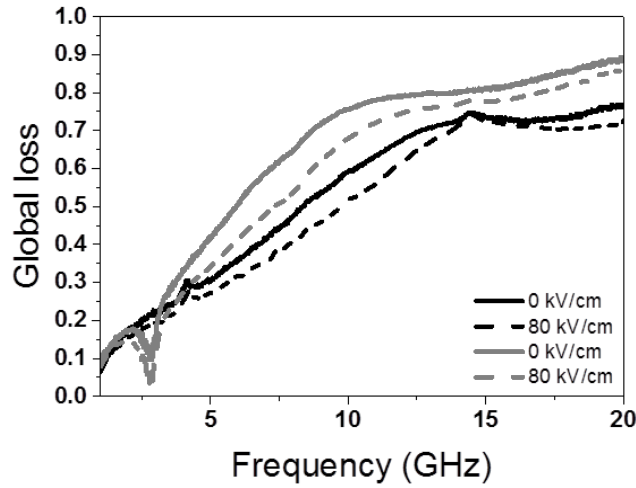


Fig. 8 Global loss variation versus frequency with and without biasing; Black line: KNN50/50; Grey line: KNN70/30

Sample	$E_{bias} = 0 \text{ kV/cm}$			$E_{bias} = 27 \text{ kV/cm}$		$E_{bias} = 80 \text{ kV/cm}$	$E_{bias} = 80 \text{ kV/cm}$
	ϵ_r	$\tan\delta$	GL	ϵ_r	$\tan\delta$	GL	T (%)
KNN50/50	360	0.36	0.59	320	0.35	0.51	15
KNN70/30	250	0.43	0.76	220	0.40	0.68	12

Table 2 Dielectric permittivity ϵ_r , loss tangent $\tan\delta$, and tunability T of the KNN50/50 and KNN70/30 samples at 10 GHz; global loss GL is computed at the stub resonator resonance frequency

An additional factor, other than composition, could induce the higher global loss exhibited by the KNN70/30 sample. Indeed, we have observed in a previous work that the use of a thicker ferroelectric film increases comparatively the global loss [39]. Therefore the global loss achieved on

the 780 nm-thick KNN70/30 sample is consistent with that of the 510 nm-thick KNN 50/50. On the other hand, the frequency tunability is usually enhanced by a thicker film [39], contrasting with the results obtained here, which supports an effect due to the composition. The similar crystalline characteristics of both samples (see section §3.1) point also the discussion at the KNN composition itself. KNN50/50 with a composition close to the morphotropic phase boundary provides advanced performance, especially higher dielectric permittivity and higher frequency tunability at microwave frequencies ($\epsilon_r = 360$ at 10 GHz and $T = 15\%$). This value of tunability, while lower than that reported by Kim *et al.* ($T = 22\%$ [26]) is relevant in regards to the moderate biasing used ($E_{bias} = 80$ kV/cm against 200 kV/cm in [26]). This tunability can also be compared with that reported in the $\text{KTa}_x\text{Nb}_{1-x}\text{O}_3$ (KTN) system, which from a chemical point of view, remains close to that of the KNN system. Even though lower, the result is still relevant considering the KNN Curie temperature ($T_c \approx 400^\circ\text{C}$) and therefore the greater thermal stability at room temperature that is expected. For comparison, the highest frequency tunability of the KTN-based devices ($T = 44\%$ under $E_{bias} = 70$ kV/cm at ~ 10 GHz [29]) was obtained with a composition marked by a Curie temperature close to room temperature ($T_c \approx 90^\circ\text{C}$), improving thereby its reconfigurability but also its sensitivity to temperature variation.

Nevertheless, some attempts need to be undertaken to decrease further KNN loss. Previous solutions have already been investigated on ferroelectric oxide materials, such as doping [40], crystalline quality improvement by the use of seed layer [41], extrinsic loss decrease by the use of ferroelectric/dielectric multilayers [42], as well as confining the ferroelectric layer in the efficient regions of the microwave devices [43].

4. Conclusion

This study presents the elaboration, the dielectric characterization and the performance at microwave frequencies of tunable ferroelectric KNN devices based on two compositions, namely

$\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ and $\text{K}_{0.7}\text{Na}_{0.3}\text{NbO}_3$. It has been shown that structurally ordered thin films of KNN perovskite on R-plane sapphire are achievable and greatly reproducible with different compositions using PLD technique. The highly oriented films with some in-plane order suggest an epitaxial-like growth, despite the strong mismatch between the two lattices ($\sim 14\%$). The characterization of the KNN films at microwave frequencies demonstrates the higher performance at room temperature of the $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$ composition, close to the morphotropic phase boundary. A dielectric permittivity $\varepsilon_r = 360$ at 10 GHz without biasing associated with a frequency tunability $T = 15\%$ under a moderate bias voltage promotes the strong interest of such films as lead-free ferroelectric material for future tunable microwave devices.

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