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# Composition and molecular scale structure of nanophases formed by precipitation of biotite weathering products

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11 Abstract

12

13 Because of their large surface area and reactivity, nanometric-size soil mineral phases 14 have a high potential for soil organic matter stabilization, contaminant sorption or soil 15 aggregation. In the literature, Fe and Al phases have been the main targets of batch-16 synthesized nanomineral studies while nano-aluminosilicates (Al and Si phases) have 17 been mainly studied in Andic soils. In the present work, we synthesized secondary 18 nanophases of Fe, Al and Si. To simulate a system as close as possible to soil conditions, 19 we conducted laboratory simulations of the processes of (1) biotite alteration in acidic 20 conditions producing a Al Si Fe Mg K leachate solution and (2) the following 21 neoformation of secondary nanophases by titrating the leachate solution to pH 4.2, 5 22 and 7. The morphology of the nanophases, their size, crystallinity and chemistry were 23 characterized by TEM-EDX on single particles and their local atomic structure by EXAFS 24 (Extended X-ray Absorption Fine Structure) at the Fe absorption K-edge. The main 25 nanophases formed were amorphous particles 10 to 60 nm in size whose composition 26 (dominated by Fe and Si) was strongly controlled by the pH conditions at the end of the 27 titration. At pH 4.2 and pH 7, the structure of the nanophases was dominated by the 28 polymerization of Fe, which was hindered by Al, Si, Mg and K. Conversely, at pH 5, the 29 polymerization of Fe was counteracted by precipitation of high amounts of Si. The 30 synthetized nanophases were estimated to be rather analogous to nanophases formed in 31 natural biotite-bearing soils. Because of their small size and potential high surface reactivity, the adsorption capacities of these nanophases with respect to the OM shouldbe revisited in the framework of soil C storage.

### 34 1 Introduction

35 Because of their large surface area, large percentage of surface atoms with 36 unbalanced charge and large number of surface functional groups per unit of mass, 37 nanometric-size mineral phases have a high potential for soil organic matter 38 stabilization (Levard et al., 2012; Qafoku, 2010), contaminant adsorption, (Levard et al., 39 2009) or soil aggregation (Asano and Wagai, 2014). These natural nanoparticles (NPs) 40 may occur either as « mineral NPs » (defined as minerals that can also exist in larger 41 sizes) or as « nanominerals » (defined as minerals that only exist at the nanometric 42 scale) (Hochella et al., 2008).

43 In surface environments, Fe and Al systems have been the main subjects of previous 44 studies due to soil observations (Kaiser and Guggenberger, 2000) as well as due to the poor solubility of Fe and Al species in most soil pH and redox conditions (pH above  $\approx 2$ 45 for Fe in oxidative conditions and pH above ≈5 for Al). However nanominerals other 46 47 than Al and Fe oxy-hydroxides are known to form in surface systems, one of the best known examples being the Al and Si imogolite/allophane phases (Basile-Doelsch et al., 48 49 2007; Basile-Doelsch et al., 2005; Basile-Doelsch et al., 2015; de Junet et al., 2013; Torn 50 et al., 1997).

51 Released by the continuous alteration of the primary minerals, Al, Si, Fe, etc. can form 52 nanominerals at the surface of the primary mineral or at a distance from the initial 53 altered surfaces. Direct formation of nanophases by the weathering of rock material has 54 been described in oxic waters (Faimon, 2003; Zänker et al., 2006). In presence of organic 55 matter, these newly formed nanophases can complex the organic compounds and 56 stabilize them for extended periods of time. The conceptual model proposed by Basile-57 Doelsch et al. (2015) represents this mechanism. To validate the model, it is necessary to show that (1) the products of the alteration of primary minerals are likely to form 58 59 nanophases and (2) these nanophases complex the OM. In this paper, we focus on step 60 (1). We characterized the chemical nature and the structure of the nanophases formed from the elements resulting from the alteration of a primary mineral in the absence of 61 62 OM. For this purpose, we considered biotite as a common soil primary mineral able to

63 provide a complex combination of elements (Fe, Al, Si, Mg and K) via weathering.

The formation of nanophases from a solution requires a polymerizing step by 64 65 hydrolysis of metallic cations like Fe or Al (Doelsch et al., 2000; Manceau and Drits, 1993; Rose et al., 1997). In oxic conditions and at pH higher than 3-4, Fe<sup>3+</sup> for instance, is 66 known to be insoluble and reactive. (Byrne et al., 2000; Dousma and Debruyn, 1976; 67 68 Rose and Waite, 2003). Polymerization steps start from elementary monomeric building blocks such as [Fe(H<sub>2</sub>O)<sub>6</sub>]<sup>3+</sup> octahedron and are accomplished via intramolecular 69 70 condensation that initially involves deprotonation (Combes et al., 1989; Rose et al., 71 1997). Formation of small clusters via edge and double corner sharing between Fe 72 octahedra is all included in the initial stage of "nucleation" which is followed by later 73 stages of growth and aggregation (Rose et al., 1997; Wells, 1984). Polymerization routes 74 have also been found for Al, with the formation of a transient  $Al_{13}$  polymer (Bottero et al., 1980). It has been shown that Si can strongly impair the polymerization stages in 75 76 both elements (Doelsch et al., 2000; Lartiges et al., 1997).

Our experimental protocol combined a weathering step that provided a complex 77 solution of Fe, Si, Al, K and Mg cations, followed by the polymerization of nanophases in 78 79 a range of pH corresponding to soil conditions (pH 4.2, 5 and 7). The atomic structures 80 of the nanophases were determined by Fe k-edge EXAFS (Extended X-ray Absorption 81 Fine Structure) spectroscopy and size, shape, crystallinity and chemical composition by 82 TEM-EDX. Three main questions are raised in this paper: (1) Does the proposed 83 experimental protocol make it possible to obtain nanophases? (2) Are the polymerized 84 nanophases homogeneous or of variable chemical composition? (3) Are the nanophases 85 obtained representative of phases that can form in soils?

### 86 2 Materials and Methods

#### 87 2.1 Experimental protocol

88 The biotite studied comes from Bancroft, Ontario Canada (Ward Science) (Turpault89 and Trotignon (1994)):

90 
$$K_{0.92} \Big( Fe_{1.06}^{(2+)} Mg_{1.66} \Big) \Big( Fe_{0.094}^{(3+)} Ti_{0.12} Mn_{0.046} \Big) \, [Si_3 AlO_{10}] (OH_{1.1}F_{0.9})$$
 Eq. (1)

Biotite was mechanically ground (< 50  $\mu$ m) in a zirconium oxide 'ball mill' setup, washed, sieved with ultra pure water and dried at 40<sup>o</sup>C.

93 *Dissolution step:* 33 g of < 50  $\mu$ m biotite particles were placed in 1L of pH=2 HNO<sub>3</sub> 94 solution (solid:liquid ratio of 1:30). After 29 days, the dissolved species were separated 95 from non-weathered biotite particles using Tangential Filtration flow (TFF) (Spectrum 96 Labs) at a cut-off size of 10 kD (pore size of  $\sim$ 2 nm). The filtered solution is hereafter 97 referred to as 'leachate solution'. Concentrations of dissolved species in the leachate solution were monitored with an inductively coupled plasma atomic emission 98 99 spectrophotometer (ICP-AES, Horiba Jobin-Yvon "Ultima C", Longjumeau, France). The 100 measurement uncertainty was < 10%. The preliminary experimental setup that 101 contributed to the design of the above dissolution protocol is detailed in supplementary 102 information (Figure A1 and Table A1).

103 Polymerization step: Nanophases were precipitated by increasing the pH of the 104 leachate solution. The initial pH (2) was raised by adding 0.2N NaOH (Sigma Aldrich) at 105 a constant rate of 70 µL/min (785 DMP Titrino). Pyrex beakers and stir bars were used. 106 Three final pH values were targeted: 4.2, 5 and 7. To determine the target pH values, a 107 preliminary titration curve of the leachate solution was performed (Figure A2). The 108 slope of this curve changed at pH 4.2 and 5. The variations in slope pointed to changes in 109 the nature of the hydrolyzing system, inferring correlative changes in the nature and/or 110 structure of the polymerized phases. At pH 7, the addition of NaOH no longer had an 111 effect on polymerisation. The durations of titration were 1.4, 1.5 and 1.67 hours at pH 112 4.2, 5 and 7 respectively. When the target pH was reached, a 1 ml aliquot of sample was 113 collected for TEM-EDX analyses and the remaining solution was ultra-centrifuged at 80 114 000 rpm for 2 h to separate precipitated nanophases from the dissolved species. The 115 settled particles were freeze-dried for EXAFS analyses. The whole experimental protocol 116 is summarized in Figure A3.

#### 117 2.2 TEM images and analyses

Images were obtained and chemical analyses of nanophases were performed using a transmission electron microscope coupled with an EDX analyzer (JEOL JEM 2011 TEM). For each sample, EDX analyses were performed on a set of around 40 nanophases of different sizes. The parameters were 50 000X magnification, an angular tilt of 20<sup>0</sup> toward the detector, energy range of 40 keV, and corrected counting time of 30 s. A constant beam density was set at ~63.5 pA.cm<sup>-2</sup>. Fe, Si, Al, K, and Mg were quantified using the Bruker AXS MET line mark data quantification procedure (Berthonneau et al.,
2014). The size of each EDX-analyzed nanoparticle was subsequently measured on the
corresponding micrograph.

#### 127 2.3 EXAFS at the Fe K edge

128 Extended X-ray absorption fine structure (EXAFS) spectroscopy at the Fe K-edge data 129 was performed at the ESRF synchrotron source on the FAME beam line (Grenoble, 130 France) and on beam line 11.1 at the ELETTRA synchrotron (Trieste, Italy). Spectra 131 were acquired using Si(220) and Si(111) double crystal monochromators at the ESRF 132 and ELETTRA for the samples and the references respectively. Spectra were acquired at 133 ambient conditions in both transmission and fluorescence modes (30-element solid-134 state Ge detector (Canberra, France)). All the samples were pressed into 5 mm diameter 135 pellets mixed with boron nitride as a dilution agent. Each spectrum comprised an 136 average of 3 to 5 individual spectra taken 107 eV below and 849 eV (14.75 Å<sup>-1</sup>) above the absorption edge of Fe (7112 eV) at a counting rate of 2-9 seconds until 14.75 Å<sup>-1</sup>. To 137 138 get representative spectra and minimize the risk of beam damage, each spectrum was 139 collected at a different pellet location. Data were treated using Athena software (Ravel 140 and Newville, 2005). All radial distances in the RDF (Radial Distribution Function) were 141 not corrected for phase shift (Figure 3b). According to the relation  $\Delta R = \pi/2k$ , the 142 minimum distance resolution of 0.11 Å was achieved.

143 A two-stage approach was used for the quantitative analysis of EXAFS data. The first 144 stage consisted of using linear combination fitting (LCF) based on EXAFS spectral library 145 of reference compounds. The second stage consisted of shell by shell fitting (SSF) to 146 investigate the atomic environment of Fe. The two quantitative methods were 147 performed at all three pH. The results of the two methods were in agreement. Only the 148 SSF method is described in the paper because this method provided additional 149 information on the local structure. Details of the LCF are presented in supplementary 150 information.

Goethite, Ferrihydrite, Fe short range order (Fe-SRO) phase and Fe dimers were used as references. Goethite provides the highest Fe polymerization level of all reference compounds with 4 Fe-Fe octahedra edge-sharing and 4 double corner-sharing interactions. Ferrihydrite is characterized by fewer edge interactions (2 to 2.5) and double corner interactions (~3-5) than goethite (Maillot et al., 2011; Michel et al., 2007). Fe-SRO is composed of Fe polymers formed in Si rich solutions. Fe polymers are characterized by 2 edge-sharing but only a 0.6 double corner interaction. Fe polymers were synthesized at pH 5 in presence of Si (Si/Fe=4), conditions in which there are no Si-O-Fe interactions (Doelsch et al., 2000; Doelsch et al., 2001). For Fe dimers only 1 edge-sharing coordination has been described (Rose et al., 1996)). (See Supplementary Information for details on reference samples).

162 Shell by shell fitting (SSF) was used to extract structural information such as the 163 coordination number (N<sub>i</sub>) and interatomic distances (R<sub>i</sub>) relative to the absorber atom 164 using the software code Artemis (Ravel and Newville, 2005). The contributions of the 165 different atomic shells were assessed using back-Fourier transform of the pseudo-radial 166 distribution function. These partial EXAFS functions were then least-squares fitted by a 167 theoretical function (in the single-scattering approximation) in order to determine the structural and chemical parameters of the atomic neighbors in the atomic shells around 168 169 Fe. Uncertainties for radial distances remained within a conservative ±0.02 Å range 170 typical in the determination of this parameter (Pokrovski et al., 2002). Varying the k 171 weight magnified or eliminated different sections of the EXAFS signal. A k weight of 3 172 was used for SSF techniques. To determine an appropriate value for E0 and amplitude 173 reduction factor (So2), the lepidocrocite spectrum with pre-confirmed structural 174 information was used to set those values to 2eV (±2eV) and 0.75 respectively. It is worth 175 noting that in complex media, various fits of similar quality can be obtained. It is 176 therefore important to recall that structural results are not necessarily univocal.

### 177 **3 Results**

# 178 3.1 Chemical composition of the leachate solution and conditions for the179 formation of nanophases

The chemical composition of the leachate solution is presented in Table 1. Fe, Mg and Si were the main contributors to the leachate solution, while Al and K contributed to a lesser extent. Mn and Ti concentrations were negligible (Figure A1) and were not included in the atomic proportions.

The preliminary titration experiment of the leachate solution (Figure A2) showed that pH 4.2, pH 5 and pH 7 were the values at which changes in the polymerization regime could be expected. Theoretically, nanophases formed at these three different pHshould thus reveal differences in chemical composition and/or atomic structure.

#### **188 3.2 TEM-EDX characterization of nanophases**

189 Polymerization experiments resulted in nanophases obtained at three final pH 4.2, 5 190 and 7. Collected masses of nanophases ranged between 11 to 15 mg. Regardless of the pH, particles were aggregated, forming distinct clusters of small and large particles 191 192 (Figure 1). The size distribution showed a bimodal Gaussian fit with maxima at  $6 (\pm 3.7)$ 193 nm and 30 (±15) nm. ~15 nm was the cut-off size between smaller and larger particles. 194 Volume estimates showed that the volume of larger particles was 50 times larger than 195 the volume of smaller particles. Electron diffraction analyses revealed a diffuse pattern 196 across all pH and particle sizes (Figure 1).

197 Fe and Si mainly composed the nanophases regardless of the pH and size range 198 (Figure 2 A and B). The proportions of Fe and Si were higher than in the leachate 199 solution, indicating preferential polymerization of Fe and Si compared with the other 200 elements. The proportions of Al in the nanophases were close to those in the leachate 201 solution. The contribution of Al was limited, even at  $pH \ge 5$  when polymerization of Al 202 was theoretically expected. The proportions of K and Mg in the nanophases were in the 203 same range as that of Al. However, they were significantly depleted in the nanophases 204 compared to in the leachate solution. Finally, in terms of general trends as a function of 205 pH, pH 5 represented a singular case in which the large particles were extremely Si 206 enriched and Fe and Al depleted.

#### **207 3.3 EXAFS** at the Fe K-edge: iron atomic range structure

#### 208 3.3.1 Comparison of raw data

The structural interactions between Fe octahedra are represented as signals at ~5.2 Å<sup>-1</sup> and ~7.5 Å<sup>-1</sup> on the EXAFS spectra (Figure 3a and Figure B1) (Mikutta, 2011; Mikutta et al., 2010; Pokrovski et al., 2003). These interactions were also observed as peaks in Rspace at radial distances of ~2.7 Å and ~3.2 Å corresponding to edge and double corner sharing interactions respectively (Figure 3b) (Rose et al., 1997). In transiting from the 'goethite and 2L ferrihydrite' to 'Fe-SRO and Fe dimers' references, i.e. with decreasing Fe octahedra polymerization, the weak intensity in k-space at ~5.2 Å<sup>-1</sup> converted into a faint shoulder and the intensity of the  $\sim$ 7.5 Å<sup>-1</sup> feature decreased. The above change in crystalline order was also seen in R-space with the double corner peak at  $\sim$ 3.2 Å disappearing and peak at  $\sim$ 2.7 Å decreasing in intensity. In the reference standards, the locations of these Fe-Fe markers changes from a shoulder to a fully formed peak as the Fe polymerization increases from Fe dimers to Goethite. This increase in Fe polymerization is accompanied by shift in location of the signals from lower to higher wavenumber values.

Features at  $\sim 5.2$  Å<sup>-1</sup> and  $\sim 7.5$  Å<sup>-1</sup> on the experimental EXAFS spectra showed that iron 223 224 polymers larger than dimers were formed during the precipitation steps at each pH 225 (Figure 3). Phases formed at the pH 4.2 produced a spectrum for the above-mentioned 226 Fe-Fe markers whose intensity was between that for goethite and 2L ferrihydrite. These phases produced distinct structural signals with the highest intensity of all the samples 227 228 (Figure 3). For pH 5 and 7 phases, collective comparison of k and R-space spectra gave 229 an average between 2L ferrihydrite and Fe-SRO references. The intensities of these two 230 samples were weaker than that of the pH 4.2 sample.

#### 231 3.3.2 Quantitative analysis of EXAFS data using shell by shell fitting

232 In all samples, the first coordination sphere around iron was composed of oxygen atoms at a distance of 1.96±0.02 Å (Table 2). The coordination number varied from 6.3-233 234 6.5±1.3 for pH 4.2 and pH 7. This result showed the formation of iron octahedra as the 235 main iron species. However, at pH 5, the coordination number was significantly lower 236 (4.9±1.0). This number was either related to iron octahedra considering the 15-25% 237 range error (2000; O'Day et al., 1994) or to the presence of a proportion of tetrahedral 238 coordinated iron like in ferrihydrite due to a Keggin-like motif (Maillot et al., 2011; 239 Michel et al., 2007).

240 The SSF of the second and third coordination spheres of all samples was performed 241 by successively testing Fe-Fe, Fe-Si and Fe-Al atomic pairs. R-factors of all three atomic 242 shell fits were systematically better than two atomic shell fits. Only three shell fits were 243 selected (Figure 4). Detailed steps in the SSF for all three phases are given in Figures C2. 244 For the pH 4.2 sample, the best solution was obtained with Fe in the 2<sup>nd</sup> atomic shell with  $N_{Fe2}$ =1.5±0.3 at  $R_{Fe-Fe2}$ =3.01±0.02 Å and Fe in the 3<sup>rd</sup> atomic shell with  $N_{Fe3}$ =1.9±0.4 245 at a distance of R<sub>Fe-Fe3</sub>=3.38±0.02 Å. The interatomic distances were in agreement with 246 247 Fe-Fe octahedra edge and double corners sharing. The total Fe-Fe coordination number 248 (N=3.4±20%) indicates a low level of polymerization (Table B1 and B2). For the pH 5 249 sample, two solutions had almost equal quality of fit (only differing in the path used for 250 the 3<sup>rd</sup> atomic shell). The first solution used Fe-Fe<sub>2</sub> edge sharing interaction at 251 R=3.01±0.02 Å with N<sub>Fe2</sub>=1.1±0.2 and Fe-Si (or Al) path at R=3.18±0.02 Å with 252 N<sub>Si</sub>=3.9±0.8. The Fe-Si (Al) distance corresponded to a single corner sharing between the 253 Fe octahedron and the Si (or Al) tetrahedron. The second solution only used Fe-Fe 254 octahedra edge sharing with Fe-Fe<sub>2</sub> at R=2.99 $\pm$ 0.02 Å with N<sub>Fe2</sub>=1.2 $\pm$ 0.2 and Fe-Fe<sub>3</sub> path 255 at R=3.13 $\pm$ 0.02 Å with N<sub>Fe3</sub>=0.6 $\pm$ 0.1. However, in both cases, the Fe-Fe coordination number (1.1-1.8±20%) was much smaller than for the pH 4.2 sample. For the pH 7 256 257 sample, the best fits showed 3 solutions. The fit with the Fe-Fe<sub>2</sub> edge sharing path at 258 R=3.07 $\pm$ 0.02 Å with N<sub>Fe2</sub>=3.4 $\pm$ 0.7 and the Fe-Fe<sub>3</sub> double corner sharing path at 259 R=3.36 $\pm$ 0.02 Å with N<sub>Fe3</sub>=0.5 $\pm$ 0.1 could be the most reliable solution, according to the Fe 260 dominant composition of the nanophases. Whatever the solution, at pH 7, the total Fe-Fe 261 coordination number (2.5-3±20%) was higher than at pH 5 but lower than at pH 4.2.

## 262 **4 Discussion**

# 4.1 Biotite dissolution rate and physical-chemical conditions of polymerization in batches

265 The proportions of Fe, Si, Al, K and Mg in the leachate solution (Table 1) differed from those in stoichiometry in biotite (Figure A1b). These proportions showed, as expected, 266 267 that the dissolution of biotite is incongruent (Kalinowski and Schweda (1996); Turpault and Trotignon (1994); Bonneville, et al., 2009; Bonneville et al. (2011)). However, the 268 269 calculated dissolution rates (Table A1) were slower than the values reported in the 270 literature (see Bonneville et al. (2011) for a review). A 10 kD filtration was applied in 271 our protocol. This  $\sim$ 2 nm cutoff allowed the truly dissolved components to be retreived 272 separately from the undissolved colloidal particules (Buffle and van Leeuwen, 1992; 273 McCarthy and Zachara, 1989). By comparison, the most frequently used filtration 274 protocols in other studies utilized filters with a cutoff size of 200 nm (Bonneville et al., 275 2011; Bray et al., 2015) to define the threshold for dissolved species. This difference 276 may explain the observed slower dissolution rates in our experiments. We thus obtained 277 a solution with elemental proportions that are representative of the weathering of a

mineral (Table 1). This composition is therefore more representative of soil solutions
than solutions produced by commercial metal salts as well as solutions that consider
only Fe and Al ions in the formation of nanophases (Kleber et al. (2015) and references
therein).

282 As revealed by TEM, the polymerization step enabled the formation of nanophases at 283 pH 4.2, 5 and 7. These values correspond to realistic pH values in soils (Schwertmann 284 and Murad, 1983; Yong and Phadungchewit, 1993). Moreover, experimental 285 polymerization involves rapid changes in pH. Such abrupt variations have also been 286 observed at very small spatial scales (Bonneville et al., 2011) on biotite surfaces. 287 Consequently, the batch approach used in our protocol, although greatly simplified 288 compared to a soil system, made it possible to reproduce physical-chemical conditions 289 that realistically mimic soil processes.

#### 290 4.2 Local structure of nanophases

#### **4.2.1** The presence of Al, Si, Mg and K in the system limits Fe polymerization

292 The coordination number (Ntot) is a direct proxy for the degree of polymerization. 293 The levels of polymerization of the nanophases synthesized in our experiments were 294 less than 3.4 whatever the pH (Figure 5). This value is lower than the number of Fe-Fe 295 bonds in polymers synthesized in pure Fe systems (Combes et al., 1990; Combes et al., 296 1989; Doelsch et al., 2000). The same difference in values was also observed in systems 297 in which Fe is hydrolyzed in presence of Si (Doelsch et al., 2003; Doelsch et al., 2000; 298 Doelsch et al., 2002; Doelsch et al., 2001; Pokrovski et al., 2003). In our case, the 299 presence of Al, Si, Mg and K in the solution clearly hinders the polymerization of Fe 300 regardless of pH.

#### 301 4.2.2 The pH controls the local structure of nanophases

At pH 4.2, the level of polymerization of the nanophases was the highest of our experiments (Figure 5) and Fe was the main element in the nanophases (Figure 2). Fe-Fe bonds with double corner interactions dominated the local structure. The number of edge interactions was lower than double corners interactions (Figure 5). These polymers are therefore structured in three dimensions, resembling the local structure of the polymers described in systems where Fe polymerizes in the presence of Si at a pH close to 3 (Doelsch et al., 2000; Pokrovski et al., 2003). The occurrence of Fe-O-Si bonds between Fe octahedra and Si-tetrahedra was not observed (Table 2), which is consistent
with the Fourier transformed infrared (FTIR) results obtained in conditions similar to
our systems (Si/Fe = 1, pH 3) (Doelsch et al., 2001).

At pH 7, Fe was still the main element (Figure 2) but its degree of polymerization was slightly lower than that at pH 4.2. However, unlike pH 4.2, the octahedra were mostly bounded by their edges (Figure 5), forming rather planar local structures. Again, the system we studied behaved in a quite similar way to the Fe polymerization systems in the presence of Si (Doelsch et al., 2000; Doelsch et al., 2001). Moreover, these authors showed by FTIR the existence of Fe-O-Si bonds at pH 7, in agreement with the third layer of our quantification by SSF (Table 2).

319 At pH 5, the level of polymerization of Fe was significantly lower than that of the 320 structures observed at pH 4.2 and 7 (Figure 5). The number of edge interactions was 321 slightly greater than the number of double corners. This local structure is therefore 322 quite close to a mixture of dimers and trimers. Quantification by SSF suggests the potential presence of Fe in tetrahedral form in the local structure as observed by 323 324 (Pokrovski et al., 2003), but without clearly demonstrating it. On the other hand, like at 325 pH 7, SSF analysis (Table 2) suggests the existence of Si-O-Fe bonds that correspond to 326 the FTIR results obtained at pH 5 for Si/Fe = 1 (Doelsch et al., 2001). Thus, unlike the 327 nanophases formed at pH 4.2 and 7, the composition of the nanophases formed at pH 5 328 is largely dominated by Si (Figure 2). Fe is present only as dimers and trimers, probably 329 isolated in an amorphous silica matrix. The polymerization of Fe at pH 5 was thus 330 counteracted by the polymerization of high amounts of amorphous Si. Thermodynamic 331 and/or kinetic mechanisms that control the variations in the proportions of Fe and Si in 332 the nanophases as a function of pH remain to be explained.

#### 333 4.3 Two stages of precipitation: growth and maturation

The individual nanophases had two particle size ranges (Figure 1) regardless of the pH considered. The differences in chemical composition as a function of size are better illustrated in the Fe, Si and Al ternary diagram in Figure 6. Regardless of pH, the chemical composition of the smaller particles was close to that of the initial leachate solution, with a slight enrichment in Fe. Two hypotheses can be proposed to highlight the chemical heterogeneity as function of the particle size. First, since no time series experiments were conducted the two particle types could have been formed from 341 different precursors or second, the difference may be due to a maturation/aging 342 mechanisms. In the case of the second hypothesis, the smallest particles could be the 343 first nanometric phases that polymerize during titration. They could derive from the 344 rapid growth of the initial nuclei. The chemical compositions of the larger particles 345 differed from those of the smaller particles at the same pH. They were enriched in Fe at 346 pH 4.2 and pH 7, and enriched in Si at pH 5. The difference in the composition of the 347 larger and smaller particles means that simple aggregation of the smaller particles 348 cannot explain the formation of the larger particles. Instead, we propose that these 349 particles could be formed after the titration step, when the pH is stable, by a stage of 350 maturation. Maturation (e.g. Ostwald ripening (Voorhees, 1985)) is a stage of 351 dissolution/precipitation: the most unstable small particles as a result of the large 352 surface area can be dissolved and re-precipitation occurs on the surface of other 353 growing particles. The composition and structure of the nanophases formed during the 354 hypothesized stage of maturation appeared to be strongly controlled by the pH 355 conditions.

356 The TEM results thus tend to indicate two polymerization steps in the formation of 357 nanophases: we propose a first initial growth, and second maturation. It is important to 358 note that data from EXAFS were unable to distinguish between the two stages of 359 polymerization. EXAFS information is still an average signal over all particle size ranges 360 within a given sample. However, the observed structural differences, in particular the 361 contribution of Si to the structure of the particles at pH 5 (Table 2 and Figure 5) tends to 362 show that the EXAFS signal of the polymerized phases was rather dominated by the 363 structure of the larger particles.

# 364 4.4 Synthetized nanophases are representative of nanophases formed in surface 365 environments

In order to test the representativeness of the products formed in our experiment as analogues to nanophases formed in surface environments, we compared the particles synthesized with two types of nanophases naturally formed: the altered layer formed at the surface of biotite and the nanophases formed at a distance from the initial altered surfaces.

The study of Bonneville et al. (2011) is particularly relevant to compare our results with a layer of biotite alteration, since these authors reproduced the chemical alteration 373 processes at mineral-mycorrhiza interface, representing the hotspot of the weathering 374 engine at work in soils. The composition of the nanophases formed in our pH 5 375 experiments are reported in Figure 7 together with their results averaged over the 20 376 nm thick altered-biotite zone. The comparison shows very similar compositions 377 between nanophases formed in batch at pH 5 and the products of the surface layer 378 alterated biotite. Moreover, these authors also measured pHs at the mineral-mycorrhiza 379 interfaces, varying from values below 4.6 up to 5.8. The nanophases formed in our pH 5 380 experiments can therefore be considered to be formed under rather similar conditions.

381 Even though the chemical compositions of nanophases pH 5 and biotite-altered layer 382 seemed similar, comparison of local atomic structures was more complex. In an earlier 383 study, Bonneville et al. (2009) identified the presence of vermiculite and ferrihydrite in 384 the altered biotite layer. These observations were consistent with those of Murakami et 385 al. (2003) who, using high resolution TEM, observed these gradual mineralogical 386 transformations in addition to the presence of chlorite. On the other hand, other studies 387 tended to show that the alteration layer formed on the mineral surface is amorphous. 388 The mechanism proposed is no longer a gradual transformation associated with the 389 preferential diffusion of selected ions, but a process of interfacial dissolution-390 reprecipitation (Hellmann et al., 2015; Hellmann et al., 2003; Hellmann et al., 2012). The 391 results obtained by Kawano and Tomita (2001) on an altered natural bytownite, which 392 showed alteration of the amorphous layer, also support the latter process. In the case of 393 an amorphous surface alteration layer, the nanophases synthesized by our protocol 394 could be good analogs of the surface processes of the minerals.

395 Concerning the second type of nanophases (neoformed nanophases in soils at a 396 distance from the altered mineral surface), the nanophases in our study may also be 397 analogs of natural secondary nanomineral phases. Only a few authors have studied the 398 composition and local atomic structure of such nanophases. Dai et al. (2014) observed 399 the formation of nanoparticle aggregates after experimental alteration of bentonite by 400 bacteria. SEM-EDX analyses showed that the nanoparticle aggregates were composed of 401 Si, Na, Ca, Mg, Al, Fe, K. Wen et al. (2014) also extracted nanophases in a <450 nm 402 fraction that exhibited Al, Si and Fe compositions and amorphous structures. These two 403 studies demonstrate that not only Al and Fe, as classically described in literature, but 404 also other elements such as Si, Na, K, Mg, and Ca may be involved in the formation 405 secondary nanophases in surface environments. It is therefore probable that the

406 nanophases that were synthesized in the present work are, at least in the case of the
407 larger particles, representative of some nanophases formed in soils, sediments or other
408 surface environments.

#### 409 **5** Conclusion

410 This laboratory work simulated the successive processes of biotite alteration and the 411 following neoformation of secondary phases from ions in solution. The nanophases 412 formed were amorphous phases that polymerized in two steps: (1) the initial growth 413 phase formed 2 to 15 nm particles whose chemical composition closely resembled that 414 of the alteration solution; (2) the maturation phase then resulted in 15 to 60 nm 415 particles whose compositions are strongly controlled by the pH conditions. At pH 4.2 416 and pH 7, the structure of the nanophases was dominated by Fe but its polymerization 417 was hindered by the presence of Al, Si, Mg and K. Conversely, at pH 5, the polymerization 418 of Fe was counteracted by the polymerization of high amounts of amorphous Si. 419 Surprisingly, Al appeared to play a minor role in polymerization, regardless of the pH. 420 However, thermodynamic and/or kinetic mechanisms that control the variations in the 421 proportions of Fe and Si in the nanophases as a function of pH remain to be explained. In 422 comparison with previous alteration studies, the laboratory-synthetized phases of our 423 study were estimated to be analogs of the nanophases formed in natural biotite-bearing 424 soils at varying distances from biotite surfaces.

These nanophases have a high potential for contaminant sorption, aggregation processes as well as for organic matter stabilization in soils. In the framework of the C budget issues and the 4 per 1000 initiative, it would be interesting to examine the role and the mechanisms of the interactions between these Fe, Si, Al, Mg, K nanophases and soil organic matter. Like for the Fe or Al phases reported in the literature, one could expect coprecipitation processes leading to-medium to long term stabilization of C in soils.

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#### 633 Figure caption

Figure 1: TEM micrographs of nanophases. For each pH, magnification increases from left to right. On 50
nm scale images, electronic diffraction is superimposed, revealing the amorphous structure of the
nanophases. SP: smaller particles; LP: larger particles.

Figure 2: TEM-EDX chemical analysis of polymerized particles at three pH expressed in stoichiometric atomic %. The box-and-whisker plots represent the median value, the upper and lower quartiles, as well as the min and max of the data. The chemical composition of the leachate solution (Leachate S.) is also shown for the sake of comparison (bold dashed line, with their respective element specific error values). A: chemical composition of the smaller nanophases; B: chemical composition of the larger nanophases. n: number of particles analyzed.

Figure 3: EXAFS (a) and RDF (b) plots. Arrows indicate locations of confirmed Fe-Fe signals. Spectra in
black represent references. E and DC stand for edge-sharing and double corner sharing interactions
respectively. Distances on the RDF are not corrected for phase shifts.

646 Figure 4: Shell-by-Shell fitting for samples pH=4.2, pH=5 and pH=7. Fit numbers refer to Table 2.

647 Figure 5: Changes in the coordination numbers for all Fe-Fe atomic pairs in the SSF analysis (Table 2) with

648 the lowest R factors, leading to minimum (min) and maximum values (max) for each type of interaction and

- for each pH. Total represents the addition of coordination numbers by edge and double corner interactions.
  Total coordination numbers from pure Fe system from the literature are also shown for the sake of
  comparison.
- Figure 6: Ternary diagrams of Fe, Si and Al atomic proportions in the nanophases formed at pH 4.2, 5 and
  7. The white star represents the initial composition of the leachate solution, the small black dots represent
  the composition of smaller nanophases and the larger grey dots the composition of larger nanophases.

655 Figure 7: Chemical compositions expressed as K:Si, Mg:Si, Fe:Si and Al:Si ratios measured by TEM-EDX.

Empty circles represent the chemical composition of the surface layer of an altered biotite (Bonneville et al.,
2011) averaged over its 20 nm thickness. Black circles represent the mean composition of the larger particles
formed in our pH 5 experiment. Error bars are standard deviations.

#### 659 **Table Caption**

660 Table 1: Chemical composition of the leachate solution provided by the dissolution experiment.

Table 2: SSF analysis of pH 4.2, 5 and 7 phases. The R window is the radial distance range over which the
fit is carried out. N, R and σ represent the coordination number, radial distance and disorder, respectively. E<sub>0</sub>
is the starting energy value for the fit and R-factor is a parameter for fit quality. Fe-Fe<sub>2</sub> corresponds to edge
sharing, Fe-Fe<sub>3</sub> corresponds to double corner sharing. Fit numbers refer to the graph presented in Figure 4

665 and Figure C1.

	R window Å	N ±20%	R (Å) ±0.02	σ (Å)	N ±20%	R (Å) ±0.02	σ (Å)	N ±20%	R (Å) ±0.02	σ (Å)	E <sub>o</sub> (eV)	R- factor	Fit No
рН 4.2	1.00-3.63		Fe-O		Fe-Si			Fe-Fe <sub>3</sub>				0.420	
		6.8	1.95	0.12	3.5	3.25	0.05	2.2	3.42	0.06	0,0	0,128	2
			Fe-O			Fe-Al			Fe-Fe₃		0.0	0.040	3
		6.6	1.94	0.12	3.5	3.02	0.04	1.6	3.40	0.03			
			Fe-O		Fe-Fe <sub>2</sub>			Fe-Fe₃			4.0	0 022	Л
		6.3	1.97	0.12	1.5	3.01	0.08	1.9	3.38	0.08	4.0	0.025	4
рН 5	1.00-3.09		Fe-O			$Fe-Fe_2$			Fe-Fe <sub>3</sub>		3.0	0.010	3
		4.7	1.96	0.09	1.2	2.99	0.07	0.6	3.13	0.05	3.0		
		Fe-O			Fe-Fe <sub>2</sub>			Fe-Si			3.0	0 006	Д
		4.9	1.96	0.09	1.1	3.01	0.09	3.9	3.18	0.09	5.0	0.000	-
			Fe-O		Fe-Fe <sub>2</sub>		Fe-Al			2.0	0.006	5	
		4.9	1.96	0.09	1.0	3.00	0.08	4.9	3.21	0.08	3.0	0.000	5
рН 7	1.00-3.24	Fe-O			Fe-Fe <sub>2</sub>			Fe-Fe <sub>3</sub>			4 0	0 011	5
		6.5	1.98	0.11	2.5	3.06	0.11	0.5	3.35	0.11	4.0	0.011	5
			Fe-O			Fe-Fe <sub>2</sub>			Fe-Al		4.0	0.011	6
		6.5	1.98	0.11	2.5	3.06	0.11	2.1	3.45	0.08			
			Fe-O			Fe-Fe <sub>2</sub>			Fe-Si		4.0	0.011	7
		6.5	1.98	0.11	2.6	3.06	0.11	1.6	3.42	0.08	4.0		







Figure





Appendix Click here to download Appendix: TamratGCA\_revised\_4 with corrections.docx Electronic Annex Click here to download Electronic Annex: Tamrat et al GCA Supp Inf revised V2.docx