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Article

## Compact Modeling Solutions for OxRAM memories

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**Abstract:** Emerging non-volatile memories based on resistive switching mechanisms pull intense R&D efforts from both academia and industry. Oxide-based Resistive Random Acces Memories (namely OxRAM) gather noteworthy performances, such as fast write/read speed, low power and high endurance outperforming therefore conventional Flash memories. To fully explore new design concepts such as distributed memory in logic, OxRAM compact models have to be developed and implemented into electrical simulators to assess performances at a circuit level. In this paper, we present an compact models of the bipolar OxRAM memory based on physical phenomenons. This model was implemented in electrical simulators for single device up to circuit level.

**Keywords:** Compact modeling; OxRAM; design

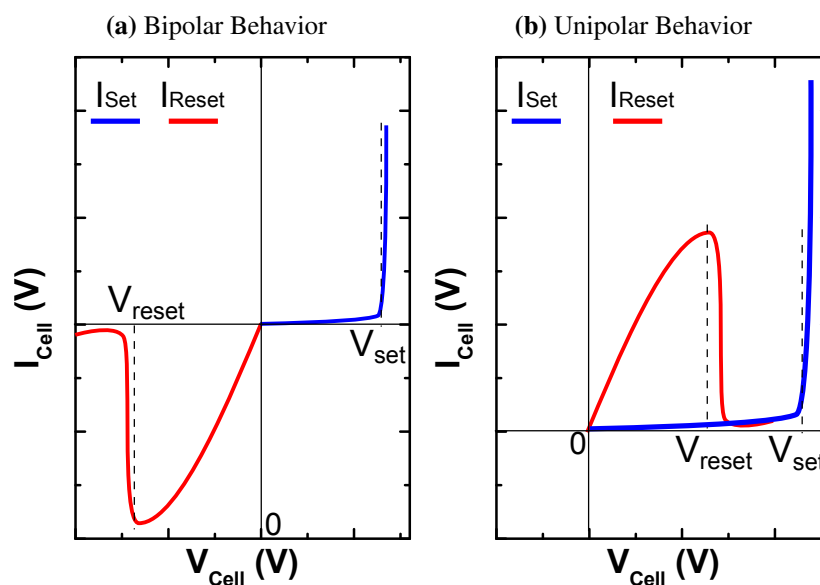
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### 1. Introduction

Memory devices based on resistive switching materials are currently pointed out as promising candidates to replace conventional non-volatile memory devices based on charge-storage beyond 2nm-technological nodes [1–3]. Indeed, as compared to conventional floating gate technologies, Resistive RAMs (so-called RRAM) gather fast write/read operations, low power consumption, CMOS voltage compatibility and high endurance. Moreover the resistive memory element consists of a simple Metal/Insulator/Metal (MIM) stack. In this way, one of the major advantages of resistive switching

8 memories is their capability, whatever the underlying physics is, to be integrated in the back-end-of-line  
 9 enabling NVM solutions to be distributed over CMOS logic. Relying on different based on different  
 10 physical mechanisms, various RRAM technologies are now categorized in the ITRS. The Redox Memory  
 11 category, covered in this study, includes Conductive Bridge RAM (CBRAM) [4] and Oxide Resistive  
 12 RAM (OxRAM) [5] both of whom exhibit a bipolar behavior, (i.e. switching relying on voltage polarity)  
 13 (cf Figure 1a). Conversely, RRAM technologies referred as Thermo-Chemical Memories (TCM)[6], or  
 14 fuse-antifuse memories, are mostly based on nickel oxide (NiO) and exhibit a unipolar behavior (i.e.  
 15 switching relying on voltage amplitude) as show in Figure 1b.

**Figure 1.** Typical I–V characteristic of Resistive memories.



16 For the OxRAM memory elements addressed in this paper, the MIM structure is generally composed  
 17 of metallic electrodes sandwiching an active layer, usually an oxygen-deficient oxide. A large number of  
 18 resistive switching oxides, like  $\text{HfO}_2$ ,  $\text{Ta}_2\text{O}_5$ , NiO,  $\text{TiO}_2$  or  $\text{Cu}_2\text{O}$ , are reported in the literature [7–10].  
 19 The Valency Change Mechanism (VCM) occurs in specific transition metal oxides and is triggered by a  
 20 migration of anions, such as oxygen vacancies.

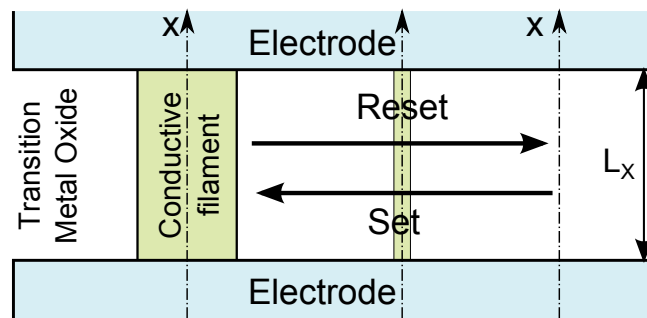
21 After an initial electroforming step, the memory element may be reversibly switched between a  
 22 High Resistance State (HRS) and a Low Resistance State (LRS). The electroforming stage corresponds  
 23 to a voltage-induced resistance switching from an initial very high resistance state (virgin state) to a  
 24 conductive state. In the case of bipolar switching, bipolar voltage sweeps are required to switch the  
 25 memory element (Figure 1a). Resistive switching in an OxRAM element corresponds to an abrupt  
 26 change between a HRS ( $R_{HRS}$ ) and a LRS ( $R_{LRS}$ ). This resistance change is achieved by applying  
 27 specific voltage to the structure (i.e.  $V_{Set}$  and  $V_{Reset}$ ). Generally, the electroforming voltage is superior  
 28 to these voltages. However several groups have demonstrated forming-free structures by adjusting the  
 29 stoichiometry of the active layer [11–13].

30 In this paper, an compact models of the bipolar OxRAM memory is presented. This model was  
 31 implemented in electrical simulators for single device up to circuit level.

32 **2. Compact model for OxRAM cells**

33 Even if OxRAM technology is still in its *infancy*, it is broadly accepted that the field-assisted motion  
 34 of oxygen vacancies governs the bipolar resistance switching [14]. The proposed OxRAM modeling  
 35 approach [15–17] relies on electric field-induced creation/destruction of Conductive Filament (CF)  
 36 within the switching layer. The model is based on a single master equation in which both *set* and  
 37 *reset* operations are accounted simultaneously and control the radius of the conduction pathway ( $r_{CF}$ ).  
 38 Figure 2 depicts the proposed model for the switchable MIM structure.

**Figure 2.** Formed and dissolved conductive filament resulting from *set* and *reset* operations respectively in MIM structure.



The *set* (*resp. reset*) process can be described by an electrochemical kinetic equation relying on the Butler-Volmer equation [18]. In the LRS, where conduction is controlled by the CF, charge transport is assumed to be ohmic accordingly to previous reports in the literature [19,20]. On the contrary, HRS is dominated by leakage current within the oxide layer where. To take into account a lot of trap assisted current (Poole-Frenkel, Schottky emission, Space Charge Limited Current (SCLC)), a power law between the cell current and the applied bias has been consider. The model assumes an uniform CF radius and electric field in the cell where temperature elevation (triggered by Joule effect) may accelerate redox reaction rates. In this way the local temperature of the filament is given by [15]:

$$T = T_{amb} + \frac{V_{Cell}^2}{8 \cdot k_{th}} \cdot \left( \frac{r_{CF}^2}{r_{CFmax}^2} \cdot (\sigma_{CF} - \sigma_{OX}) + \sigma_{OX} \right) \quad (1)$$

39 where  $T_{amb}$  is the ambient temperature, where  $V_{Cell}$  is the voltage applied between the top and the  
 40 bottom electrodes,  $k_{th}$  is the thermal conductivity and  $\sigma_{CF}$  (*resp.*  $\sigma_{OX}$ ) is the electrical conductivity of  
 41 the conductive filament (*resp.* oxide)

The *Set* operation is modeled based on the Butler-Volmer equation through the electrochemical reduction rate ( $\tau_{Red}$ ):

$$\frac{1}{\tau_{Red}} = A_{RedOx} \cdot e^{-\frac{E_a - q \cdot \alpha_{Red} \cdot V_{Cell}}{k_b \cdot T}} \quad (2)$$

42 where  $k_b$  is the Boltzmann constant.

Similarly, *reset* concerns the local dissolution of the CF and accounted by the oxidation rate ( $\tau_{Ox}$ ):

$$\frac{1}{\tau_{Ox}} = A_{RedOx} \cdot e^{-\frac{E_a + q \cdot \alpha_{Ox} \cdot V_{Cell}}{k_b \cdot T}} \quad (3)$$

where  $E_a$  is the activation energy,  $\alpha_{Red}$  and  $\alpha_{Ox}$  are the transfer coefficient (ranging between 0 and 1),  $A_{RedOx}$  is the nominal redox rate. Hence, the growth/dissolution of the filament results from the inter-play between both redox reaction velocity through the following master equation:

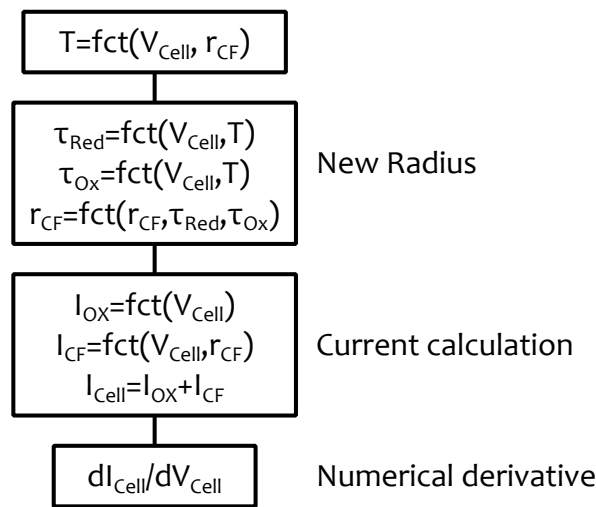
$$\frac{dr_{CF}}{dt} = \frac{r_{CFmax} - r_{CF}}{\tau_{Red}} - \frac{r_{CF}}{\tau_{Ox}} \quad (4)$$

where the local CF radius ( $r_{CF}$ ) is comprised between zero and a maximal value ( $r_{CFmax}$ ). To allow implementation into electrical simulation tools, a discrete writing is required. If the time step is sufficiently small,  $\tau_{Red}$  et  $\tau_{Ox}$  may be assumed as constant. The discrete form of Eq. 4 is then given by Eq. 5. Solving the differential equation 4 step by step allows a better convergence of simulation tools.

$$r_{CF_{i+1}} = \left( r_{CF_i} - \frac{\tau_{eq}}{\tau_{Red}} \right) \cdot e^{\frac{-\Delta t}{\tau_{eq}}} + \frac{\tau_{eq}}{\tau_{Red}} \quad (5)$$

43 where  $\tau_{eq} = \frac{\tau_{Red} \cdot \tau_{Ox}}{\tau_{Red} + \tau_{Ox}}$

**Figure 3.** Program flowchart employed for numerical simulation of OxRAM memory devices.



Finally, the total current in the OxRAM includes two components, *i.e.* one is related to the conductive species ( $I_{CF}$ ) [15] the other concerns conduction through the oxide ( $I_{OX}$ ):

$$I_{CF} = \frac{V_{Cell}}{L_x} \cdot (r_{CF}^2 \cdot \pi \cdot (\sigma_{CF} - \sigma_{OX}) + r_{CFmax}^2 \cdot \pi \cdot \sigma_{OX}) \quad (6)$$

$$I_{OX} = A_{HRS} \cdot S_{Cell} \cdot \left( \frac{V_{Cell}}{L_x} \right)^{\alpha_{HRS}} \quad (7)$$

where  $L_x$  is the oxide thickness and  $S_{Cell}$  is the total area of the device. Finally, the total current flowing through the cell is:

$$I_{Cell} = I_{OX} + I_{CF} \quad (8)$$

44 These equations were then implemented within an ELDO compact model following the flowchart  
 45 given in figure 3. At each call of the OxRAM instance during a transient simulation, the previous state

of the filament as well as the applied voltage are provided to the model in order to take into account the memory effect. The new filament state and the current are then computed as function of these inputs and the time step.

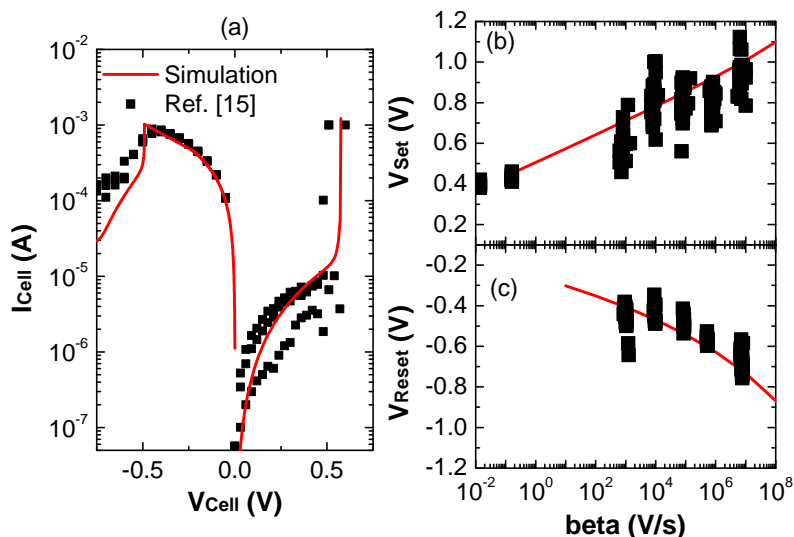
### 3. Model validation

The compact model was calibrated on recent electrical data measured on HfO<sub>2</sub>-based OxRAM devices [21]. To validate the proposed theoretical approach, the model was confronted to quasi-static and dynamic experimental data extracted from the literature. Figure 4(a) shows quasi-static *set* and *reset*  $I(V)$  characteristics measured on HfO<sub>2</sub>-based memory elements. In this study, the memory elements consisted in a Ti/HfO<sub>2</sub>/TiN stack with a hafnium oxide thickness of 10 nm. The description of the cell manufacture is presented in [21]. Using the *set* parameters given in Table 1, the present model shows an excellent agreement with experimental data for both *set* and *reset* operations.

**Table 1.** Physical parameters used for Bipolar simulations

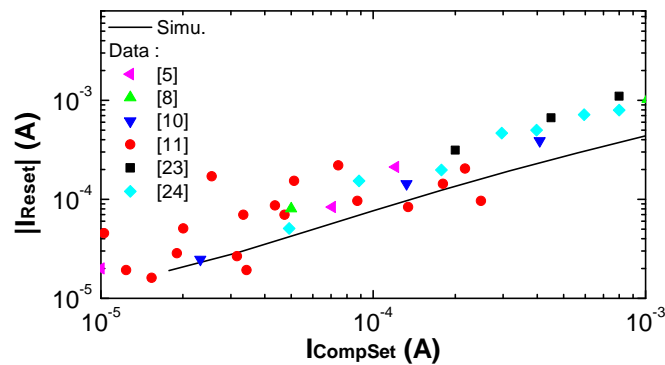
|                                                             |                                                    |
|-------------------------------------------------------------|----------------------------------------------------|
| $r_{CF_{max}} = 20 \text{ nm}$                              | $L_x = 10 \text{ nm}$                              |
| $S_{cell} = 1 \mu\text{m} \times 1 \mu\text{m}$             | $T_{amb} = 300 \text{ K}$                          |
| $A_{RedOx} = 1 \times 10^9 \text{ s}^{-1}$                  | $E_a = .95 \text{ eV}$                             |
| $\alpha_{Red} = 0.85$                                       | $\alpha_{Ox} = 0.85$                               |
| $A_{HRS} = 5 \times 10^{-9} \text{ A}/(\text{V}^2)$         | $\alpha_{HRS} = 2$                                 |
| $\sigma_{Ox} = 0.1 \text{ m} \cdot \text{S}$                | $K_{th} = 0.8 \text{ W}/(\text{K} \cdot \text{m})$ |
| $\sigma_{CF_0} = 12.5 \times 10^5 \text{ m} \cdot \text{S}$ |                                                    |

**Figure 4.** a) Experimental  $I(V)$  (■) and b) *set* and *reset* voltage as a function of the programming ramp speed measured on a HfO<sub>2</sub>-based memory structures presented in [21] and corresponding simulation results (—).



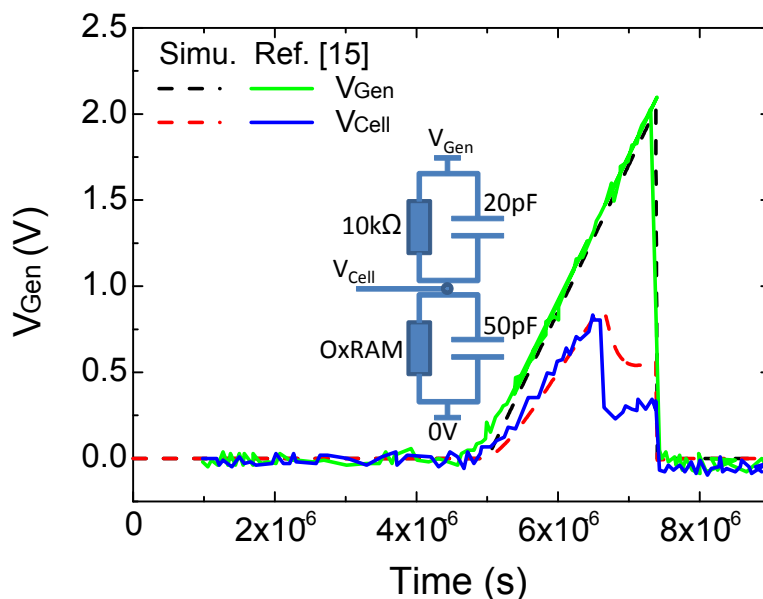
57 Figure 4(b) shows that the proposed model also satisfactorily catches the impact of the experimental  
 58 increase of  $V_{Set}$  and  $V_{Reset}$  with the programming ramp speed. Moreover the effect of the *set* current  
 59 limitation on the *reset* current is also taken into account by this compact model (Figure5). It is interesting  
 60 to note that this behavior appears for unipolar and bipolar memory [11]. But in our study, only the bipolar  
 61 structures will be studied.

**Figure 5.** Maximum current during the *reset* operation ( $I_{Reset}$ ) as a function of the maximum current during the preceding *set* operation ( $I_{CompSet}$ ). Experimental data were extracted from Ref. [5,8,10,22–24].



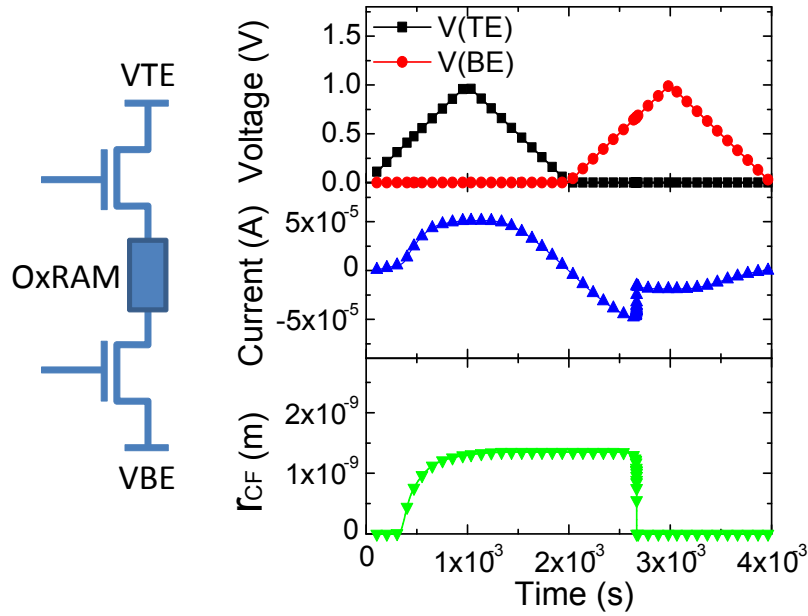
62 Figure 6 illustrates the transient current response of the cell when a voltage ramp is applied to the cell  
 63 [Figure 4(b)]. A significant cell voltage discontinuity is observed during the *set* operation. This behavior  
 64 highlights the self acceleration of *set* mechanism. Indeed, when the applied voltage is below the *set*  
 65 voltage, the resistance continuously decreases. Let us mention that our model, that already includes  
 66 a thermal activation of *set* operation, should be able to take into account this effect once the parasitic  
 67 capacitances originating from the measurement setup are provided.

**Figure 6.** Dynamic measurement of OxRAM (HfO<sub>2</sub>-based memory) and corresponding simulation results.



68 To fully validate the compact model and its integration into the electrical simulator, Figure 7 gives an  
 69 example of bipolar OxRAM cells simulated at a circuit level, *i.e.* surrounded by MOS transistors.

**Figure 7.** electrical simulation of 2T-1R OxRAM structure



70 These models have been successfully used to simulate new MOS-RRAM cells like a NVM flip-flop  
 71 [25], Non-Volatile SRAM[26] and OxRAM memory array [27].



72 **4. Model application**

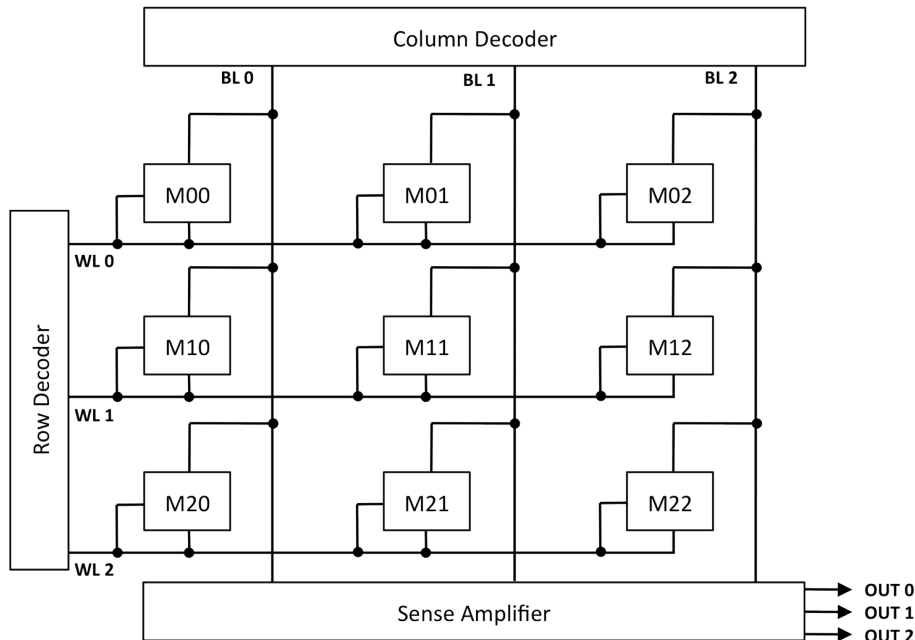
73 *4.1. OxRAM reliability evaluation versus OxRAM variability [16]*

74 In this section, an investigation in the impact of OxRAM variability on the memory array  
 75 performances is proposed [16]. Indeed, variability in advanced IC designs has emerged as a roadblock  
 76 and significant efforts of process and design engineers are required to decrease its impact.

77 Since the cell variability is calibrated on silicon using the previous OxRAM model, only the  
 78 realistically possible variations are reported in this study. A large number of Monte Carlo simulations are  
 79 performed to provide the statistics needed to characterize variability. Cell variations are introduced and  
 80 simulated sequentially using an electrical simulator. The goal is to track an important shift of reliability  
 81 parameters.

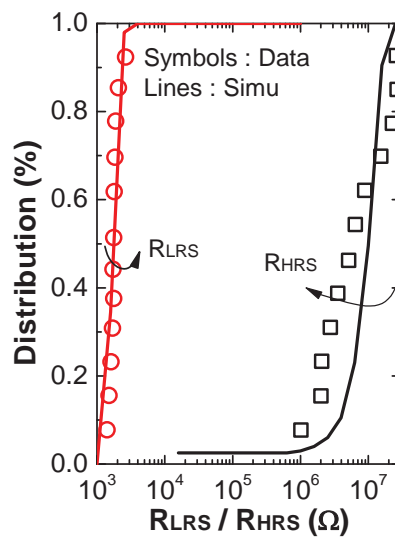
82 Figure 8 presents the elementary array used for simulation: it is constituted, a row decoder, a column  
 83 decoder and a sense amplifier for the read operation. Memory array cells are first placed in an erase state.  
 84 Then, the memory array programming is done in 2 cycles. First, all memory cells are set (logical '1'),  
 85 then the memory array is reset (logical '0'). Logical failures can be detected at the output of the sense  
 86 amplifier during the read operation after *set/reset*.

**Figure 8.** 3 × 3 OxRAM memory array.



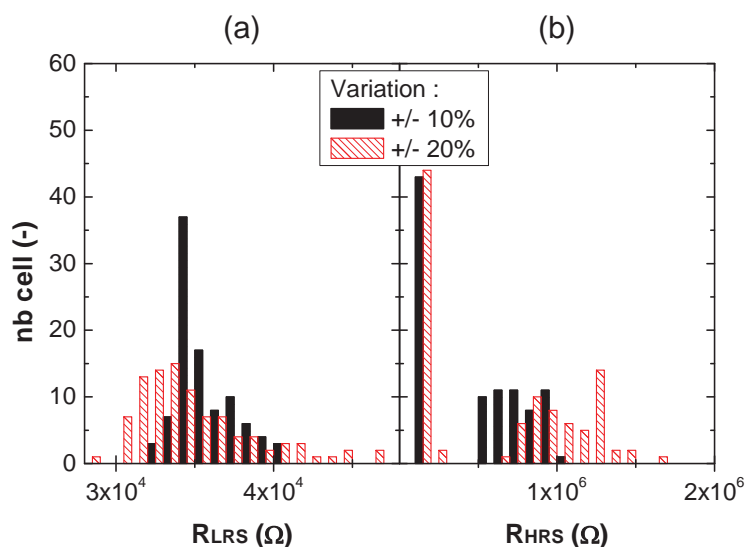
87 The best way to monitor the impact of variability on OxRAM electrical parameters is to plot the  
 88 OxRAM hysteresis in transient mode (i.e. cell current evolution versus cell voltage difference during a  
 89 Write/Erase cycle). Figure 9 shows the impact of the memory array cell variability (9 cells) on the circuit  
 90 hysteresis.

**Figure 9.** Variability impact on  $I - V$  hysteresis the memory array



91 It appears clearly that  $R_{LRS}$  distribution can be severely impacted by cell variability.  $V_{Reset}$  and  $V_{Set}$   
 92 parameters can also suffer from cell variability but in a lesser extent. Notice that  $R_{LRS}$  and  $R_{HRS}$  are  
 93 extracted @0.5V (read conditions). And  $V_{Set}$  and  $V_{Reset}$  are extracted @40 $\mu$ A, at a circuit level.  
 94  $R_{LRS}$  and  $R_{HRS}$  distributions are plotted in Figure 10. Results are presented for cell variability included  
 95 in the range  $\pm 10\%$  of the median value of the considered card model parameters (solid bar). Results are  
 96 also provided for cell variability included in the range  $\pm 20\%$  (dashed bar).

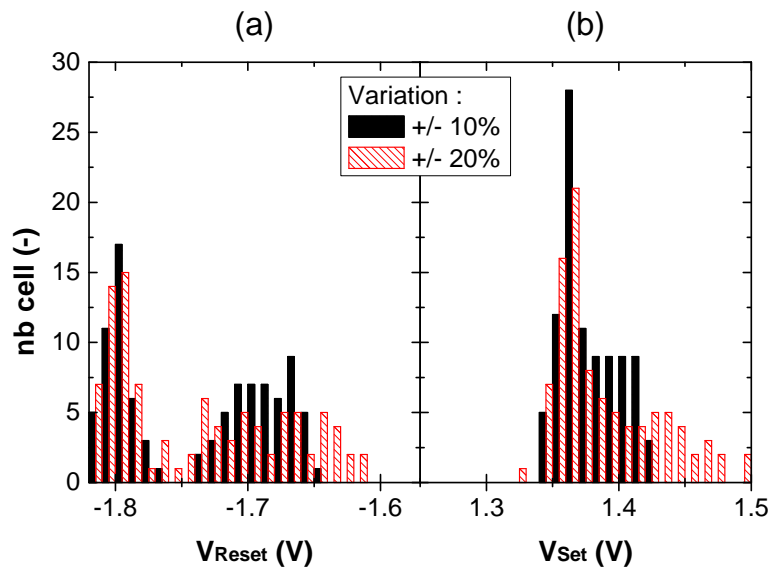
**Figure 10.**  $R_{LRS}$  and  $R_{HRS}$  distributions versus cell variability



97 At 10%, a spreading of  $R_{LRS}$  and  $R_{HRS}$  parameters is observed. The spreading increases significantly  
 98 at 20%. Although these values are related to a specific OxRAM technology, a good feedback can be  
 99 provided to designers to optimize the sensing circuitry according to the level of controllability of the  
 100 fabrication process.

101  $V_{Reset}$  and  $V_{Set}$  distributions are plotted in Figure 11a-b. Here again, the initial spreading (solid bar)  
 102 increases (dashed bar) according to the variability increase. These results are of the prime importance as  
 103 this study predicts an increase of  $V_{Set}$  to the value of 1.5 V. Which mean that the programming signals  
 104 provided to the cell needs to reach at least 1.5 V for cell to be programmed properly.

**Figure 11.**  $V_{Set}$  and  $V_{Reset}$  distributions versus cell variability

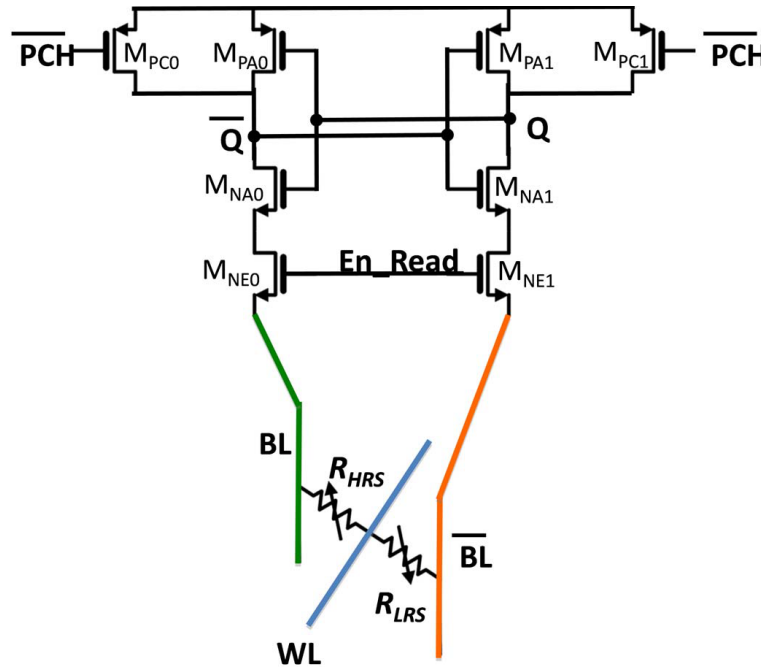


105 **4.2. SRAM Sense amplifier [26]**

106 The read operation of data stored in cross-point resistive switching memory is currently one of the  
 107 major challenges to develop this approach. Indeed, sneak path or destructive read with complementary  
 108 resistive switching element are a strong limit to develop this type of architecture. Moreover, the  
 109 resistance ratio ( $R_{HRS}/R_{LRS}$ ) and the process variations have to be considered when designing a sense  
 110 solution. A sense amplifier performing with high reliability is then required. Figure 12 shows a  
 111 pre-charge based sense amplifier, which has demonstrated the best tolerance to different sources of  
 112 variation, while keeping high speed and low power. In this sense amplifier, the read operation is  
 113 performed in two phases:

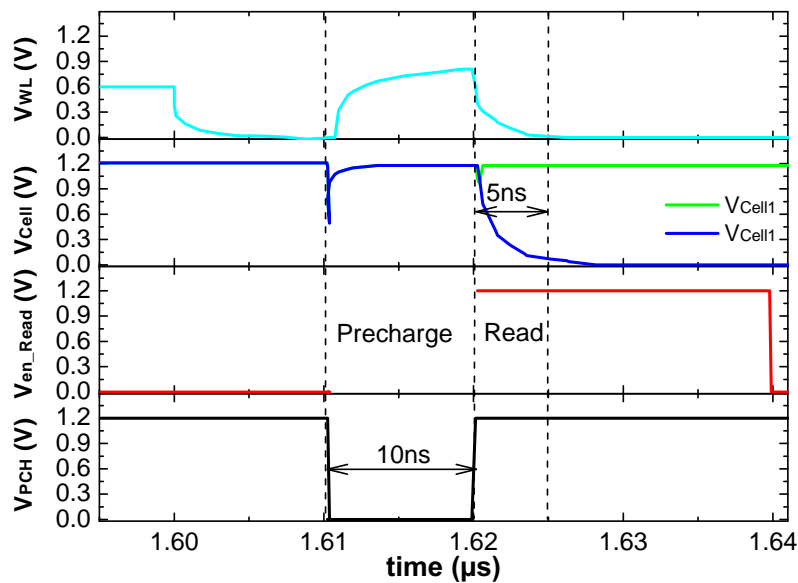
- 114 • 1st Phase: The sense amplifier is first connected to the bit-line of the selected word with SEN set  
 115 to '1' and the circuit is pre-charged with PCH equals '0'.
- 116 • 2nd Phase: The data stored in the 2R cell can be evaluated to logic level at the output Q as PCH is  
 117 changed to '1' and WL is pulled down to '0'.

**Figure 12.** Pre-Charged Sense Amplifier for data sensing. It consists of a pre-charge sub-circuit (MPC0, MPC1), a pair of inverters (MNA0-1, MPA0-1), which act as an amplifier.[26, 27]



118 The Figure 13 validates the ability of the architecture to successfully read in parallel a full word.  
 119 The model presented before has allowed us to assess the robustness of sense towards the variability  
 120 of OxRAM or CMOS transistors [26]; and the validated complete crossbar architecture based on 2R  
 121 complementary [27].

**Figure 13.** Simulation results with read phase of a selected cell [26,27]



## 122 5. Conclusion

123 In conclusion, this paper deals with a compact model well suited for describing simultaneously *set*  
124 and *reset* operations in bipolar resistive switching memories based on HfO<sub>2</sub>-based memory device. By  
125 gathering local electrochemical reactions and a thermal mechanism in a single master equation, the  
126 model enables accounting for both creation and destruction of conductive filaments. The simulation  
127 results satisfactorily match quasi-static and dynamic experimental data published in literature on resistive  
128 switching devices. Beside, the model was implemented into circuit simulators. It has been successfully  
129 used in many circuit and enabled predicting relevant trends required for designing innovative memory  
130 matrix architectures or proposing distributed memories solutions.

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