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# Density of States Broadening in CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub> Hybrid Perovskites Understood from ab initio Molecular Dynamics Simulations

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7 **Density of States Broadening in  $\text{CH}_3\text{NH}_3\text{PbI}_3$**   
8 **Hybrid Perovskites Understood from ab initio**  
9 **Molecular Dynamics Simulations**  
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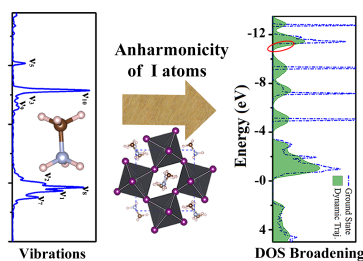
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## Abstract

Lead-halide perovskites are the promising materials in optoelectronic devices for their unique properties including direct band gap, strong light absorption, high carrier mobility and low fabrication cost. Here, by using ab initio molecular dynamics and electronic structure calculations, we report a systematic study on the broadening density of states (DOS) deep in the valence bands that has been experimentally observed but absent in static calculations. We quantify the broadening DOS reduction from cubic phase to lower temperature tetragonal and orthorhombic phases and attribute observed effects to the molecular vibrations and the anharmonicity of iodine atoms motion. Specifically, the MA cations' vibrations are strongly linked to the moderate C–N stretch and CH<sub>3</sub> bend, as well as the strong CH<sub>3</sub> and NH<sub>3</sub><sup>+</sup> stretches. These results present a theoretical perspective on the structural dynamics in lead-halide perovskites which may be valuable for future studies toward desired functionalities in perovskite-based optoelectronic devices.

## Graphical TOC Entry



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Solution-processed lead-halide perovskites have emerged as a promising optoelectronic technology due to many unique advantages including direct band gap, strong light absorption, high carrier mobility, and low fabrication cost.<sup>1</sup> For example, the photovoltaic efficiency has jumped spectacularly up to 15–22.1% for methyl ammonium lead iodine (MAPbI<sub>3</sub>, MA = CH<sub>3</sub>NH<sub>3</sub>) materials.<sup>2–4</sup> The underlying mechanisms behind these attractive photoelectronic properties have been widely explored in terms of defects,<sup>5</sup> ion migration,<sup>6–11</sup> polaronic effects (including both small<sup>12,13</sup> and large polarons),<sup>14,15</sup> ferroelectric order,<sup>16–18</sup> structural and electronic dimensionality,<sup>19</sup> etc. Studies of the relationship between the organic cations and device efficiency are rare and are mainly focused on the orientation of the MA and the induced distortion of the PbI<sub>6</sub> octahedral cage as well as direct-indirect band gap transition.<sup>20</sup> Moreover, the spontaneous generation of free carriers from exciton screening originating from the collective orientational motion of the organic cations has been experimentally confirmed.<sup>21</sup> Upon change of the organic cation, the carrier self-trapping phenomena,<sup>12</sup> interband transition properties, and optical absorption strength<sup>22</sup> are significantly altered. Additionally, due to thermal effects, MA is associated with various molecular vibrational modes and motionally disordered orientations,<sup>23</sup> which play an important role in the carrier mobility.<sup>24</sup> The molecular vibrational modes have already been measured by means of Infrared (IR)<sup>25</sup> and Raman<sup>26</sup> spectroscopies. Although MA electronic states are not the main contributors around the Fermi level, we expect their vibrations to partly affect material's functionality via enhancing the electronic dimensionality and band transport in devices.<sup>19</sup>

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Current theoretical calculations do not match the experimentally measured density of states (DOS) deep within the valence band (VB) of MAPbI<sub>3</sub> materials. Calculated DOS has sharp features within the valence band that are in stark contrast to the "flattened" nature of the experimentally measured DOS.<sup>27,28</sup> Literature has previously stated that due to a higher photo-ionization cross-section, only the Pb and I ions interact with high energy sources,<sup>27</sup> making the MA cation a spectator deep within the VB. If this is the case, electronic state

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3 separation due to spin orbit coupling (SOC), mainly associated with Pb and I,<sup>29,30</sup> could be  
4 the main factor in DOS broadening. Lack of direct experimental signatures from MA may be  
5 attributed to the low photoionization cross sections of light elements, consequently signals of  
6 C, N and H atoms are difficult to detect at high photon energies in hard X-ray photoelectron  
7 spectroscopy (HAXPES).<sup>31</sup> Computed valence spectra provide a qualitative agreement with  
8 experiment if the DOS peaks are broadened with a 0.3–0.5 eV wide Gaussian function.<sup>32,33</sup>  
9 Such an *ad hoc* way to account for the effect of molecular motion does not provide insights  
10 into the contributing atomic species. However, another hypothesis explaining the difference  
11 between the calculated and measured DOS is that the experimental spectra reflect only  
12 the average DOS resulting from a convolution of molecular motion and time resolution of  
13 spectral equipment. This is expected to recreate experimental observations deep within the  
14 VB and thus clarify the good X-ray detection ability of the material.<sup>34–36</sup> Moreover, the much  
15 more flattened and broadened characters of VB dispersion determined by angle resolved  
16 photoemission experiments further provides evidence for the feasibility of broadening the  
17 DOS by vibrations of MA.<sup>37</sup> The only way to get a detailed atomistic information on the  
18 processes taking place and to quantify the effects of molecular vibrational motions, inorganic  
19 lattice anharmonicity, and SOC is to perform molecular dynamics simulations of MA and  
20 Cs perovskite systems in different thermodynamic ensembles.  
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39 To address the discrepancies between theory and experiment, we consider factors pos-  
40 sibly influencing the theoretical DOS. Contributing physical properties, such as SOC and  
41 temperature induced distortions (octahedral tilting),<sup>38–41</sup> may play an important role in the  
42 shape and broadening of the DOS. SOC predominantly affects the orbitals of heavy atoms in  
43 systems. Previous theoretical calculations stated that the VB just below the Fermi level is  
44 derived from Pb-6s and I-5p states and the conduction band minimum (CBM) is dominated  
45 mostly by Pb-6p states, suggesting that SOC mainly influences the CBM in all temperature  
46 phases.<sup>29,30,42</sup> The distortion of the PbI<sub>6</sub> octahedral cage induces a direct-indirect band gap  
47 transition<sup>43</sup> and modifies the electronic states around the Fermi level.<sup>22</sup> Given that the en-  
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3 energy difference between DOS peaks deep within the VB are on the order of up to 0.5 eV, a  
4 substantial change in either atomic orientation or electronic distribution must occur in order  
5 to justify the wide range of absorption energies experimentally measured.<sup>34–36</sup> Identification  
6 of the atomic species participating in the state energy fluctuations can possibly answer if the  
7 broadened DOS primarily originates from molecular motion or re-distribution of electronic  
8 states. Addressing this issue successfully will enrich our understanding of photo-physics in  
9 hybrid perovskites and facilitate the experimental search for realistic optoelectronic applica-  
10 tions. In this letter, we systematically investigate that the effects of SOC and vibrational mo-  
11 tion of the organic cations in MAPbI<sub>3</sub> via first-principles calculations as outlined in Methods  
12 in the Supporting Information (SI). Our simulations determine the DOS broadening within  
13 the VB in cubic, tetragonal and orthorhombic phases at 400, 300K and 100K temperatures,  
14 respectively. This phenomenon stems from both the molecular vibrations in the cations  
15 and the anharmonicity in the motion of the iodine atoms. The MA cations' vibrations are  
16 associated with the moderate C–N stretch and CH<sub>3</sub> bend, as well as the strong CH<sub>3</sub> and  
17 NH<sub>3</sub><sup>+</sup> stretches. Knowledge of the specific atomistic motions and interactions that dominate  
18 structural and electronic properties is helpful when substituting the organic cation in order  
19 to tune materials properties.  
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37 Electronic structure calculations for the optimized structures of MAPbI<sub>3</sub> (Figure 1) pro-  
38 vide a starting point for pinpointing the cause of broadened DOS captured by experimental  
39 work. Figure 2 presents the DOS plots with and without SOC for the cubic phase of MAPbI<sub>3</sub>.  
40 These are compared to those of pure inorganic CsPbI<sub>3</sub> models taking the same lattice pa-  
41 rameters as the corresponding MAPbI<sub>3</sub> does. We notice that addition of SOC leads to a shift  
42 in energy near the CB edge producing a reduced band gap, which agrees with previously  
43 published electronic structure calculations.<sup>29</sup> The effect of SOC is not localized to specific  
44 momentum values but is known to shift band energies across all sampled k-points in the sys-  
45 tem. The SOC energy gaps are 0.67 and 0.63 eV for cubic MAPbI<sub>3</sub> and the corresponding  
46 CsPbI<sub>3</sub> model, respectively. These values are significantly corrected by about 0.99 and 0.94  
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3 eV compared to results obtained without SOC. Similar corrections of 0.9 and 1.0 eV are also  
4 observed for tetragonal and orthorhombic phases, respectively (Figure S1 and S2), in agree-  
5 ment with previous results.<sup>30</sup> When focusing on electronic states deeper in the VB, one can  
6 see a distinction in how SOC affects the Cs and MA systems. For the CsPbI<sub>3</sub> system, there  
7 exists a Cs peak located near  $-8.5$  eV with a SOC splitting amounting to 1.53 eV from the  
8 Cs p orbitals (Figure 2a and c). From atomic energy level tables,<sup>44</sup> the +1 oxidation state  
9 of a Cs cation has a SOC splitting of about 1.22 eV between the p orbitals of total angular  
10 momentum  $J = 1/2$  and  $J = 3/2$ , which is about 0.31 eV lower than the calculated splitting.  
11 In contrast, the energy structure deep within the VB for the MAPbI<sub>3</sub> system, is relatively  
12 unchanged with the addition of SOC (Figure 2b), rationalized by a negligible SOC effect  
13 from light-elements-constituted MA molecules. Additionally, SOC leads to the subsequent  
14 splitting of Pb-5d orbitals at about  $-17$  eV into two individual peaks (Figure S3), displaying  
15 a SOC splitting of 2.58 eV between the 5d orbitals of total angular momentum  $J = 3/2$  and  
16  $J = 5/2$ , in nice agreement with 2.65 eV as reported in atomic energy level tables.<sup>44</sup> This  
17 splitting has a minor impact on the electronic structure due to its extremely deep position.  
18 In the energy range from  $-12.5$  to  $-4$  eV, there are only a few electronic states associated  
19 with Pb or I atoms (Figure 2d). Moreover, SOC does not produce any sizable enlargement  
20 of the total DOS in this energy range down to  $-12.5$  eV (Figure 2b).

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Next, we consider the effect of molecular motion as a possible cause of a broadened DOS within the VB. To this end, we run a series of MD trajectories for Cs and MA-based perovskites adopting NVT thermostat condition. These simulations were performed at temperatures of 400, 300 and 100 K for the cubic, tetragonal and orthorhombic phases, respectively, on different sizes of periodic cells (see Methods in SI). Since SOC does very little to modify the electronic DOS in the energy range of interest, we neglect SOC in the calculations of MD. The radial pair distribution functions (RDF's) between the different atoms provides a clear picture of the magnitude of the vibrations of the different atoms around their equilibrium positions. As shown in Figures S4-7, the RDF's for the inorganic PbI<sub>3</sub> lattice are

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3 typical for a solid with clearly defined peaks broadened around their optimized positions.  
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5 However, owing to higher mobility, the situation is different for RDF's involving atoms of  
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7 the organic cation. There is much more variability for N–N pairs in MAPbI<sub>3</sub> than Cs–Cs  
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9 pairs in CsPbI<sub>3</sub>, especially for the high temperature cubic phase. This effect also appreciable  
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11 in the other two phases, overall being in a good agreement with previous studies.<sup>45–47</sup>  
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13 It is expected that the high frequency motions and vibrations of the organic cations could  
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15 broaden the DOS, especially deep in the VB. We next identify the atomic species participat-  
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17 ing in the state energy fluctuations so that we can investigate whether the broadened DOS  
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19 is based on molecular motion or not. While it has been established that SOC has very little  
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21 effect on the molecular motion and lattice parameters,<sup>48</sup> in order to test the effect of SOC  
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23 on the electronic structure we calculated the time-averaged DOS over 100 configurations  
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25 with SOC along the 15 ps NVT MD trajectories on the 2 × 2 × 2 cubic MAPbI<sub>3</sub> cell. As  
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27 shown in Figure S8, we can find that SOC does not lead to any sizable DOS broadening  
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29 in this energy range related to the electronic states of PbI<sub>6</sub> octahedrons. Consequently, we  
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31 exclude the SOC effect in forthcoming calculations of massive DOS dynamics, for which  
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33 the final 400 geometries along each MD trajectory were extracted. Figure 3a demonstrates  
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35 three broadened DOS peaks (denoted as peak I, II, III) existing deep inside the VB for cubic  
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37 MAPbI<sub>3</sub>, located around –5.0, –9.2 and –12.5 eV, respectively. Similar peak broadening  
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39 is also found in the other two lower temperature phases of MAPbI<sub>3</sub>, differing only by a  
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41 narrower half-width of three main broadening peaks (Figure 3b and c). This is due to a re-  
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43 duction in disordered states<sup>23,45–47</sup> and weakened vibrational motion of the MA cations from  
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45 temperature effects. Indeed, for CsPbI<sub>3</sub> (Figure S9) the time averaged DOS over the entire  
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47 energy range is almost the same as the ground state of CsPbI<sub>3</sub>, i.e. without significantly  
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49 broadened DOS peaks as found in the hybrid MAPbI<sub>3</sub>. The lack of DOS broadening in the  
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51 Cs system is thus attributed to the absence of MA vibrational modes. As the electronic di-  
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53 mensionality is increased, the electron mobility is enhanced. Although these electrons do not  
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55 contribute directly to important electronic transitions in devices, they increase the electronic  
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3 dimensionality deep within the VB.<sup>19</sup>  
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5 Note that since our MD calculations are based on a volume-fixed ensemble to match  
6 the experimental lattice of MAPbI<sub>3</sub> in each phase, the fluctuations in the system may not  
7 be strictly gauged. As reported in ref.,<sup>46</sup> the MA cations may rotate in NPT conditions  
8 that allow volume fluctuations. As soon as the volume restriction is relaxed in the MD  
9 simulations, the large distribution of PbI<sub>6</sub> octahedral tilting accompanied by an increase in  
10 the anharmonicity in the motion of the iodine atoms will occur, which has been observed  
11 in CsPbI<sub>3</sub>, MAPbI<sub>3</sub> and FAPbI<sub>3</sub> crystals.<sup>46,47</sup> It is expected that such anharmonicity in the  
12 iodine atom motion could influence the behaviors of DOS broadening. To confirm this point,  
13 MD calculations based on the NPT ensemble were carried out on cubic phase MAPbI<sub>3</sub> at  
14 450 K, which are based on a 4 × 4 × 4 supercell by replicating the cubic unit cell four  
15 times in each Cartesian direction (see ref.<sup>47</sup> for details). As demonstrated in Figure 3d there  
16 exists an obvious broadening of the electronic states up to 0.5 eV at around −11.5 eV. This  
17 confirms that the anharmonicity in the motion of the iodine atoms does contribute to DOS  
18 broadening in specific energy regions. Reversely and noteworthy, NVT calculations are not  
19 able to reveal any broadening related to the inorganic framework (vide supra). Although the  
20 anharmonicity of the I atoms does broaden the DOS, it is a smaller contribution compared  
21 to the effect of MA vibrations. Thus, we analyze next the contribution from MA vibrations  
22 to the broadening of the DOS.  
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41 We have calculated the intra-molecular vibrational modes of the free MA cation. There  
42 are 18 molecular vibrational modes, associated with the C<sub>3v</sub> symmetry of the structure,  
43 dividing the internal modes into one A<sub>2</sub>, five symmetric A<sub>1</sub> and twelve two-fold degenerate  
44 asymmetric E species.<sup>49,50</sup> The calculated vibrational frequencies of the isolated MA cation  
45 at its optimized configuration are listed in Table 1 and are visualized in Figures S10–S11.  
46 The six types of vibrational modes in ascending order are C–N stretch (281, 871 cm<sup>−1</sup>),  
47 CH<sub>3</sub>–NH<sub>3</sub><sup>+</sup> rock (909, 1256 cm<sup>−1</sup>), CH<sub>3</sub> bend (1437, 1473 cm<sup>−1</sup>), NH<sub>3</sub><sup>+</sup> bend (1545, 1679  
48 cm<sup>−1</sup>), CH<sub>3</sub> stretch (2963, 3064 cm<sup>−1</sup>), and NH<sub>3</sub><sup>+</sup> stretch (3260, 3341 cm<sup>−1</sup>), being in a good  
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3 agreement with previous results.<sup>26,49–53</sup>  
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5 Switching to MAPbI<sub>3</sub> periodic systems, each MAPbI<sub>3</sub> phase manifests similar vibrational  
6 properties. The corresponding vibrational spectrum is mainly divided into three energetic  
7 regions: (1) the low frequency band from 0–150 cm<sup>-1</sup>; (2) a mid frequency range from  
8 280–1700 cm<sup>-1</sup>, and (3) a high-frequency window from 2900–3300 cm<sup>-1</sup>.<sup>25,26</sup> The low-  
9 frequency modes are related entirely to the vibrations of the PbI<sub>6</sub> octahedra, while the mid-  
10 and high-frequency modes are linked to the vibrations of MA cations. We computed the  
11 Fourier transforms of the state energies in the region of interest, deep within the VB, in order  
12 to determine which molecular vibrational modes strongly couple to the electronic degrees of  
13 freedom (see Methods in SI). The resulting molecular vibrational spectra for the three crystal  
14 phases are shown in Figures 4 and S11–12 for peak I, II and III, respectively, which are in a  
15 good agreement with previous experimental values (Table 1).<sup>50,53</sup> In cubic MAPbI<sub>3</sub>, the main  
16 vibrations contributing to peaks I and III originate from the CH<sub>3</sub>/NH<sub>3</sub><sup>+</sup> bend (about 1400  
17 cm<sup>-1</sup>) and CH<sub>3</sub> stretch a about 3000 cm<sup>-1</sup> (Figure S12–13). The predominant vibrations  
18 contributing to peak II are mainly from CH<sub>3</sub> bend (about 1400 cm<sup>-1</sup>), NH<sub>3</sub><sup>+</sup> stretch (about  
19 3100 cm<sup>-1</sup>), as well as small amounts of C–N stretch (about 950 cm<sup>-1</sup>) and NH<sub>3</sub><sup>+</sup> bend (about  
20 1600 cm<sup>-1</sup>) (Figure 4). Similar results are obtained for the tetragonal and orthorhombic  
21 phases, except that the intensities are more attenuated due to the weakened vibrational  
22 amplitudes as temperature decreases (Figures S11–12). Exceptions are the broadened peaks  
23 I and II which couple to a slightly enhanced vibration along the C–N axis as compared to  
24 that in the cubic phase. Moreover, the CH<sub>3</sub> bends for peak I and III in the orthorhombic  
25 phase are almost non-existent due to the temperature effect. The two CH<sub>3</sub> stretches (2963,  
26 3064 cm<sup>-1</sup>) for peaks I and III become distinct from each other. Altogether, we mainly  
27 attribute widening of DOS deep inside VB in the three MAPbI<sub>3</sub> phases to mostly asymmetric  
28 CH<sub>3</sub>/NH<sub>3</sub><sup>+</sup> stretches, moderate CH<sub>3</sub> bend, and C–N stretch. Thus the observed coupling is  
29 exclusive to mid-frequency and high-frequency vibrational modes.  
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55 In summary, we computationally study the origin of the density of states broadening  
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3 experimentally reported in the hybrid organic-inorganic perovskite MAPbI<sub>3</sub>, using CsPbI<sub>3</sub> as  
4 a reference material. We show that spin-orbit coupling mainly modifies the density of states  
5 close to the Fermi level and does not contribute to its broadening deep within the valence  
6 band. From ab initio molecular dynamics simulations and electronic structure calculations,  
7 we further analyze the contribution of thermal vibrations to three phases of MAPbI<sub>3</sub>. As  
8 expected, upon reducing the temperature, the broadening gradually reduces when going from  
9 the cubic to tetragonal and to orthorhombic phases, consistent with experimental findings.  
10 The density of states broadening is ascribed to two different contributions. For states about  
11 12.5 eV below the top of the valence band, this broadening can be traced back to the  
12 anharmonicity in the motion of the iodine atoms. In addition, vibrational modes of the  
13 MA cation provide important additional contributions to the density of states broadening in a  
14 wide energy range. Contrastingly, SOC does not produce any sizable broadening of the total  
15 DOS. Specifically, MA cations' vibrations mainly stem from the C–N stretch (about 950  
16 cm<sup>-1</sup>), CH<sub>3</sub> bend (about 1400–1600 cm<sup>-1</sup>) and CH<sub>3</sub> and NH<sub>3</sub><sup>+</sup> stretches (from 2950–3300  
17 cm<sup>-1</sup>). These results present a detailed theoretical perspective on the structural dynamics  
18 in hybrid perovskites, rationalize experimental observations, and may guide future studies  
19 toward desired functionalities in optoelectronic applications.  
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## 39 Supporting Information Available

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42 The following files are available free of charge. Methods, Ground state DOS for tetragonal  
43 and orthorhombic CsPbI<sub>3</sub> and MAPbI<sub>3</sub> phases, partial DOS of cubic MAPbI<sub>3</sub> with Pb-d  
44 states included, radial distribution functions in tetragonal and orthorhombic phases, time  
45 averaged DOS for CsPbI<sub>3</sub> and for MAPbI<sub>3</sub> with SOC, calculated vibrational eigenvectors of  
46 isolated MA cation and Fourier transforms of the electronic state I and III in three MAPbI<sub>3</sub>  
47 phases.  
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### Notes

The authors declare no competing financial interest.

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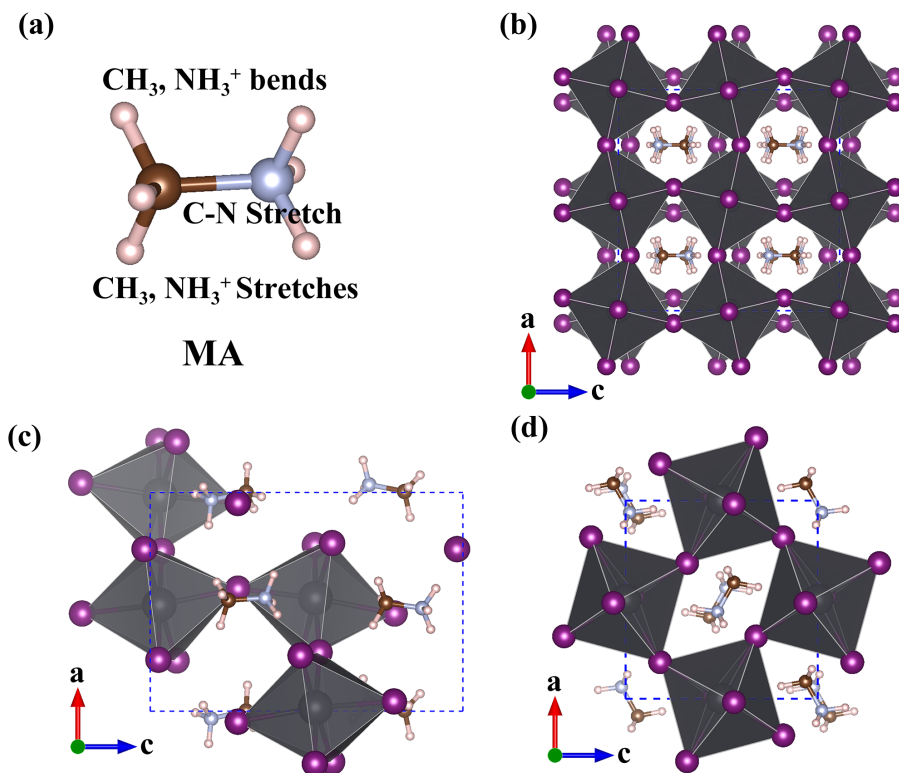


Figure 1: (a): Schematic diagram of the MA molecule. Three main vibrational modes of MA molecule's vibration are marked. (b) MAPbI<sub>3</sub> crystal structures for (b) cubic, (c) tetragonal and (d) orthorhombic phases.

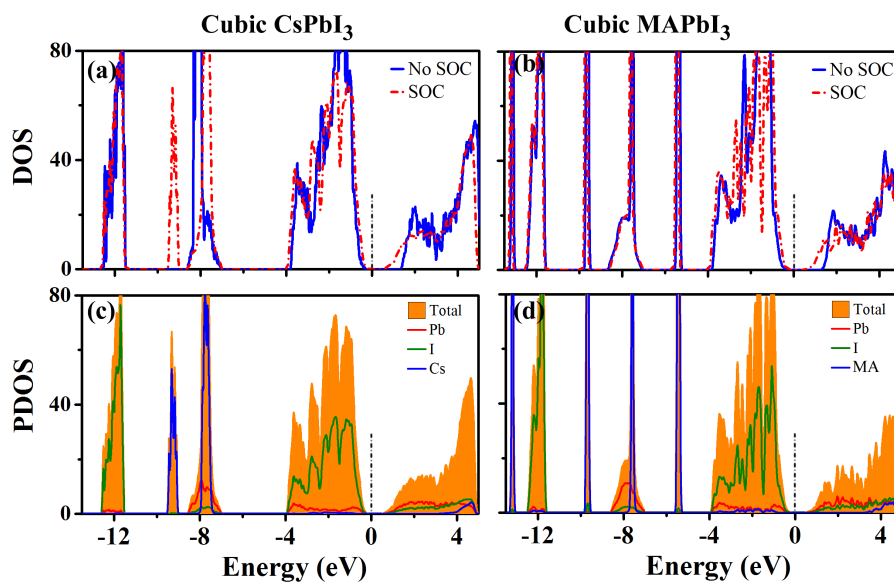


Figure 2: Ground state electronic structure corresponding to the cubic CsPbI<sub>3</sub> (a, c) and optimized MAPbI<sub>3</sub> (b, d) models. The top panels compare the calculated density of states (DOS) with and without spin-orbit coupling (SOC) as dashed red and solid blue lines, respectively. The bottom panels provide the SOC partial density of states (PDOS) with Pb (red), I (green), and the Cs or MA cations (blue), respectively. The Fermi energy is set to zero.

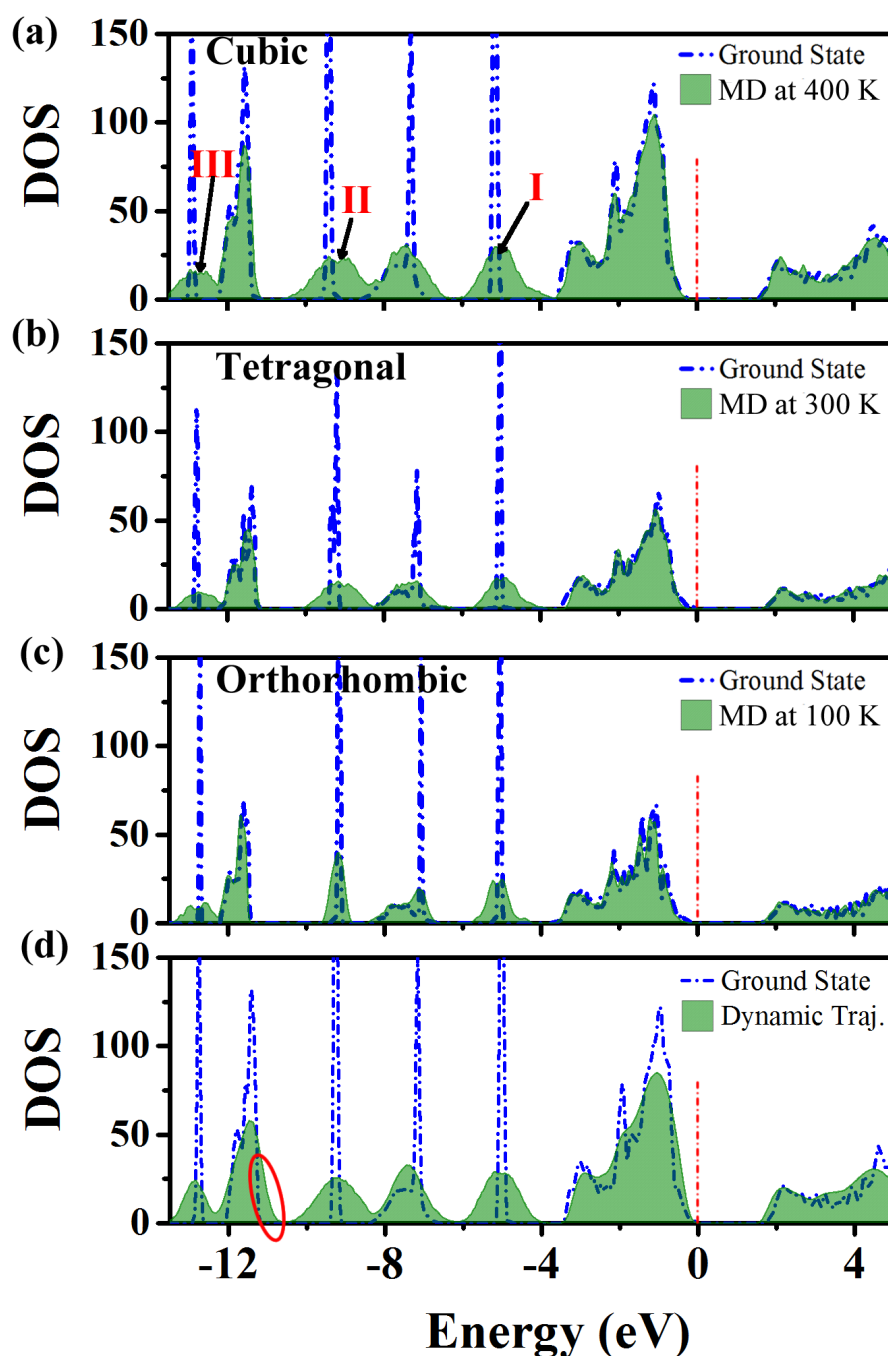


Figure 3: Time averaged DOS of (a) cubic, (b) tetragonal and (c) orthorhombic MAPbI<sub>3</sub> along MD trajectories under NVT conditions, overlaid with the ground state DOS. The Fermi levels are set to 0. The broadened DOS peaks from MA cations are indicated by Roman numbers of I, II and III in panel (a). (d) Time averaged DOS for cubic MAPbI<sub>3</sub> along MD trajectories performed under NPT conditions and  $4 \times 4 \times 4$  supercell, overlaid with the static DOS. The red circle indicates the contribution from the anharmonicity of I atoms' motion.

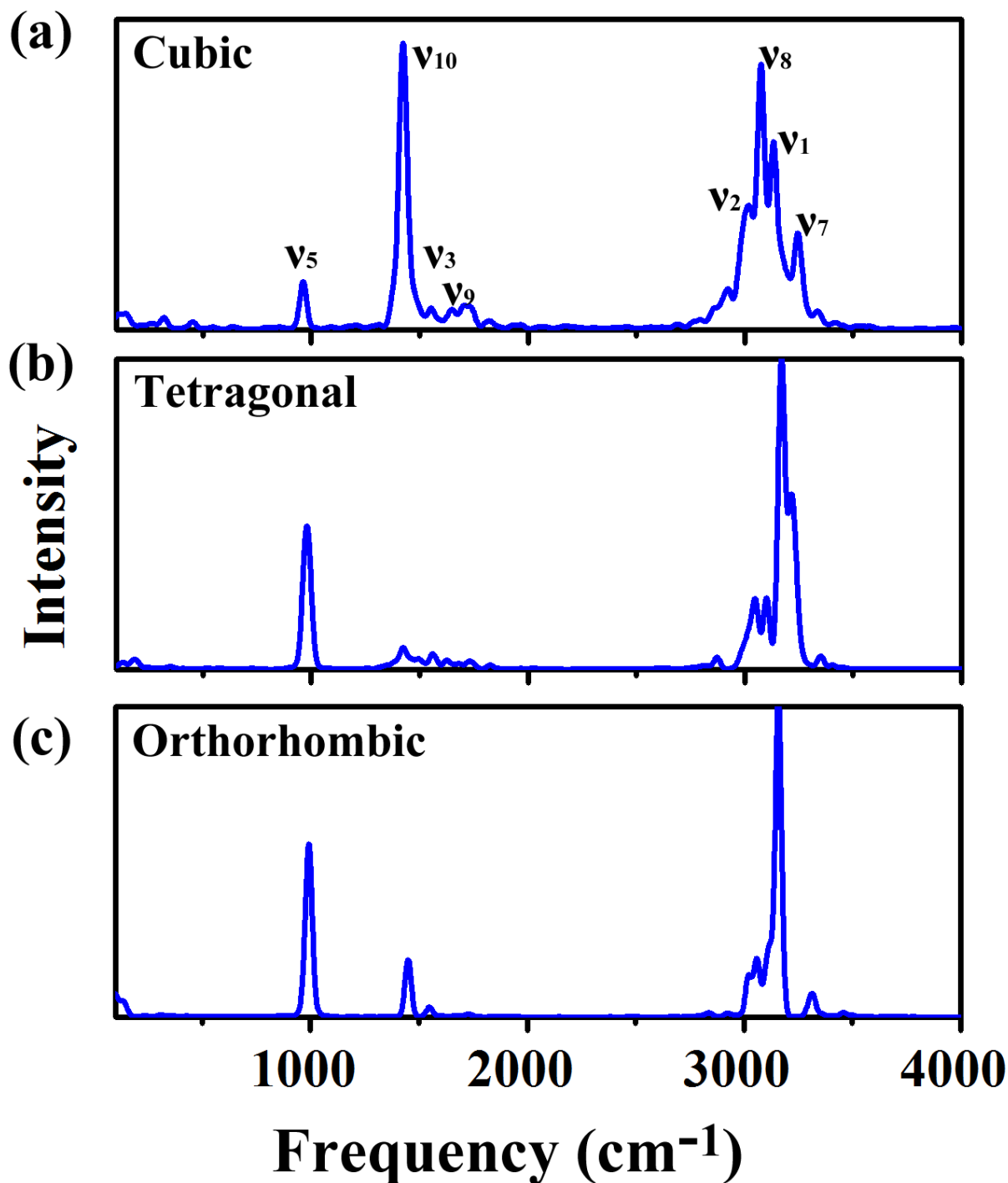


Figure 4: Fourier transforms of the state energies of the electronic states relevant to peak II (Figure 3a) in three phases of MAPbI<sub>3</sub>: (a) cubic, (b) tetragonal, (c) orthorhombic. The energy fluctuations are due to molecular vibrational modes of the MA cations. Peak assignments for modes in panel (a) are suggestions only.



**Table 1: Fundamental Vibrations of the Isotopic MA cation Calculated by B3LYP and MP2 Methods and, Experimental Measured Vibrational Modes for MAPbI<sub>3</sub> from references<sup>50,53</sup> (cm<sup>-1</sup>).**

Mode	Species	MA		MAPbI <sub>3</sub>	
		this work	MP2 <sup>53</sup>	Ref <sup>50</sup>	Ref <sup>53</sup>
$\nu_{12}$ , CH <sub>3</sub> -NH <sub>3</sub> <sup>+</sup> rock	<i>E</i>	909	1032	903.4	911
$\nu_5$ , C-N stretch	<i>A</i> <sub>1</sub>	871	982	990	960
$\nu_{11}$ , CH <sub>3</sub> -NH <sub>3</sub> <sup>+</sup> rock	<i>E</i>	1256	1409	1248.9	1250
$\nu_4$ , sym. CH <sub>3</sub> bend	<i>A</i> <sub>1</sub>	1437	1597	1415	1385
$\nu_{10}$ , asym. CH <sub>3</sub> bend	<i>E</i>	1473	1620	1452.3	1423
$\nu_3$ , sym. NH <sub>3</sub> <sup>+</sup> bend	<i>A</i> <sub>1</sub>	1545	1691	1469.9	1469
$\nu_9$ , asym. NH <sub>3</sub> <sup>+</sup> bend	<i>E</i>	1679	1809	1581	1577
$\nu_6$ , torsion	<i>A</i> <sub>2</sub>	281			
$\nu_2$ , sym. CH <sub>3</sub> stretch	<i>A</i> <sub>1</sub>	2963	3164	2896	2916
$\nu_8$ , asym. CH <sub>3</sub> stretch	<i>E</i>	3064	3263	2974.8	2958
$\nu_1$ , sym. NH <sub>3</sub> <sup>+</sup> stretch	<i>A</i> <sub>1</sub>	3260	3446	3188	3132
$\nu_7$ , asym. NH <sub>3</sub> <sup>+</sup> stretch	<i>E</i>	3341	3519	3225.6	3179

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