



HAL
open science

Nitrogen-related intermediate band in P-rich GaN_xPyAs_{1-x-y} alloys

K. Zelazna, M. Gladysiewicz, M. Polak, Samy Almosni, Antoine Létoublon,
Olivier Durand, Charles Cornet, W. Walukiewicz, R. Kudrawiec

► **To cite this version:**

K. Zelazna, M. Gladysiewicz, M. Polak, Samy Almosni, Antoine Létoublon, et al.. Nitrogen-related intermediate band in P-rich GaN_xPyAs_{1-x-y} alloys. Scientific Reports, 2017, 7 (1), pp.15703. 10.1038/s41598-017-15933-1 . hal-01715130

HAL Id: hal-01715130

<https://hal.science/hal-01715130>

Submitted on 27 Jun 2018

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.



Distributed under a Creative Commons Attribution 4.0 International License

SCIENTIFIC REPORTS

OPEN

Nitrogen-related intermediate band in P-rich $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ alloys

K. Zelazna¹, M. Gladysiewicz¹, M. P. Polak¹, S. Almosni², A. Létoublon², C. Cornet², O. Durand², W. Walukiewicz³ & R. Kudrawiec¹

The electronic band structure of phosphorus-rich $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ alloys ($x \sim 0.025$ and $y \geq 0.6$) is studied experimentally using optical absorption, photomodulated transmission, contactless electroreflectance, and photoluminescence. It is shown that incorporation of a few percent of N atoms has a drastic effect on the electronic structure of the alloys. The change of the electronic band structure is very well described by the band anticrossing (BAC) model in which localized nitrogen states interact with the extended states of the conduction band of GaAsP host. The BAC interaction results in the formation of a narrow intermediate band (E_- band in BAC model) with the minimum at the Γ point of the Brillouin zone resulting in a change of the nature of the fundamental band gap from indirect to direct. The splitting of the conduction band by the BAC interaction is further confirmed by a direct observation of the optical transitions to the E_+ band using contactless electroreflectance spectroscopy.

The concept of intermediate band solar cells (IBSCs) has been originally proposed by Wolf¹. Later Luque and Martí² have shown that an IBSC with properly located bands could attain very high solar power conversion efficiencies. The elegant simplicity of the IBSC concept has motivated researchers to search for novel materials and materials structures with an intermediate band that would satisfy strict requirements regarding photon absorption and charge collection^{3–5}. In general, the intermediate band materials/structures can be divided into three main groups: nanostructures, such as quantum dots (QDs)^{4–7}; semiconductor bulk materials containing a high density of deep-level impurities^{4,5,8}; and highly mismatched alloys (HMAs)^{9–11}. An important example of intermediate band HMA is the GaNPs alloy in which P to As ratio can be tuned at will to change the band gap and modify the respective offsets between the conduction band edge and the localized N level energy making this alloy one of the most promising materials for IBSC applications^{11–14}. The possibility of growing GaNPs on Si substrates is a very important advantage of this alloy as it makes it feasible to co-integrate IBSC, multi-junctions solar cells¹⁵ or laser emitters with Si technology. For these reasons As-rich GaNPs alloys and quantum wells with a few percent of nitrogen atoms have been investigated very intensively in recent years^{16,17}. An electrically pumped laser on Si substrate with the active region containing GaNPs quantum wells has been demonstrated by Liebich *et al.*¹⁸. Recently $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ alloy with $y \sim 0.4$ and a few percent of nitrogen has been identified and studied as the optimal material for the IBSC applications^{11,13,19}. Much less research has been done on the $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ alloys with P concentration larger than $y \geq 0.6$ ^{20–22}. Especially, although this composition range is interesting from the viewpoint of the intermediate band formation and the lattice matching with Si substrates in a multi-junction approach, complementary experimental and theoretical investigations of the whole conduction band properties was not proposed yet²³.

As seen in Fig. 1(a) the energy level of single nitrogen atoms is located below the conduction band of $\text{GaP}_y\text{As}_{1-y}$ host for the $y > 0.4$ compositions of the unstrained alloy. According to the BAC model the interaction of the localized N level with the states of the GaAs host leads to formation of E_- and E_+ bands in GaNPs alloy¹¹, see mathematical formulas for E_- and E_+ bands in the Methods section. Figure 1(b) and (c) show the dispersion relations for E_+ and E_- bands in $\text{GaN}_{0.025}\text{P}_{0.9}\text{As}_{0.075}$ and $\text{GaN}_{0.025}\text{P}_{0.6}\text{As}_{0.375}$ calculated according to the model described in the Methods section. As is seen in this case the E_- band forms a narrow intermediate band (IB), that is well separated from the E_+ band i.e., the upper conduction band (CB). These assertions of the BAC model have been never fully confirmed. Specifically, the optical transitions from the valence band (VB) edge to the E_+ band were not observed yet in P-rich GaNPs alloys. In this work, we have applied several optical spectroscopy methods including contactless electroreflectance (CER), photomodulated transmission (PT), optical absorption, and

¹Faculty of Fundamental Problems of Technology, Wrocław University of Science and Technology, Wybrzeże Wyspińskiego 27, 50-370, Wrocław, Poland. ²UMR FOTON, CNRS, INSA-Rennes, F-35708, Rennes, France. ³Materials Sciences Division, Lawrence Berkeley National Laboratory, Berkeley, California, 94720, USA. Correspondence and requests for materials should be addressed to R.K. (email: robert.kudrawiec@pwr.edu.pl)

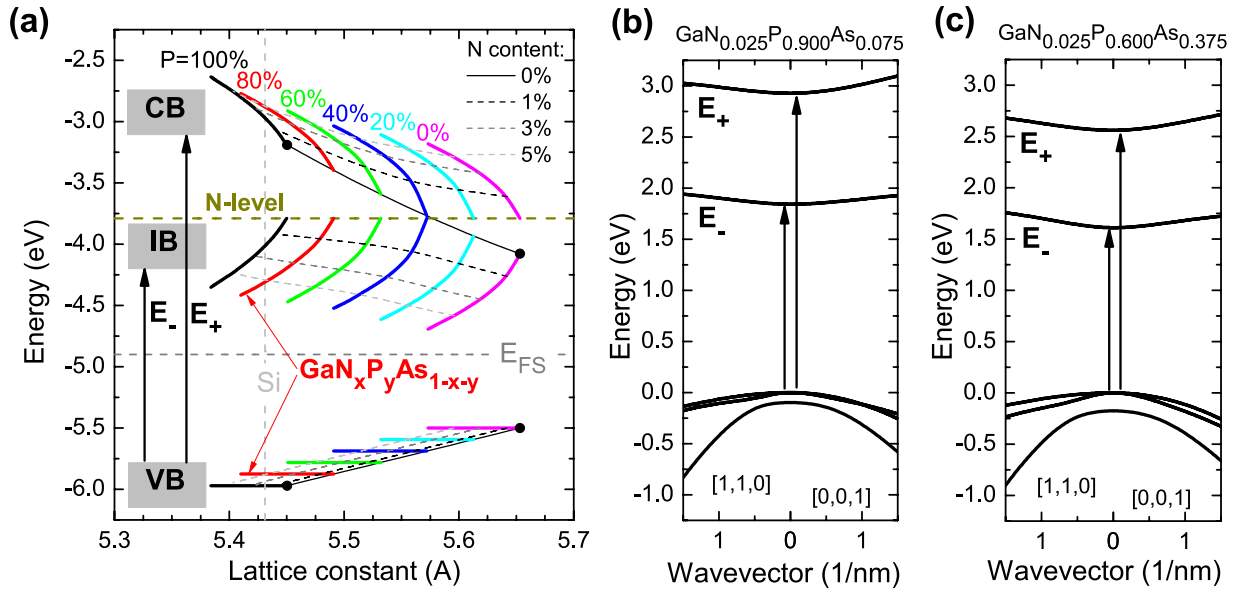


Figure 1. (a) Energies of VB, IB (E_-), and CB (E_+) band edges versus the lattice constant in unstrained bulk $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ with various P concentrations (thick color lines). Dashed lines correspond to $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ with the same nitrogen concentration: $x = 0.01$ – dark grey lines; $x = 0.03$ – grey lines; $x = 0.05$ – light grey lines. Calculated energies of the E_+ and E_- bands relative to the valence band edge for $\text{GaN}_{0.025}\text{P}_{0.9}\text{As}_{0.175}$ (b) and for $\text{GaN}_{0.025}\text{P}_{0.6}\text{As}_{0.375}$ (c). These calculations have been performed according to the BAC model described in the Methods section and ref.¹¹.

photoluminescence (PL) to study the electronic band structure and the changing character of bandgap in P-rich GaNPAs alloys. We show that all the experimental results can be consistently explained by the BAC model and confirm that GaNPAs alloys could be utilized for IBSC applications.

Results and Discussion

In order to delineate the effects of the GaPAs host matrix composition from the effects of nitrogen we have grown two sets of samples. The first set consists of $\text{GaN}_x\text{P}_{1-x}$ layers with various nitrogen concentrations ($x = 0.005, 0.013, 0.015, 0.023, \text{ and } 0.025$), and the second set are $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ layers with $x = 0.025$ and various P concentrations ($y = 0.6, 0.7, 0.9, 0.95$). All the samples were grown on GaP substrate by molecular beam epitaxy (MBE). Relevant details of the growth are described in the Methods section.

Optical properties of GaNP alloy. First, we present the results of the measurements of optical properties on GaNP samples. It is worth noting that although GaNP alloy has been studied quite intensively for last few years^{24–29} there is only a limited number of papers reporting experimental observation of optical transitions to the E_+ subband^{26,27}, and no experimental evidence of the spin-orbit splitting of the valence band of the alloy.

Figure 2(a) shows photomodulated transmission (blue line), photoluminescence (red line) and absorption (green line) spectra (i.e., a square of the absorption coefficient obtained from transmittance and reflectance measurements) for GaNP layers with various N concentrations. The strong absorption, observed for the photon energies above ~ 2.2 eV, comes from the indirect gap of the thick GaP buffer/substrate. Additional absorption edge is clearly observed at lower energies. The absorption edge shifts to longer wavelengths with increasing nitrogen concentration. The linear slope of the α^2 plot indicates that the low energy absorption edge originates from direct optical transitions. The direct character of this bandgap is further confirmed by PL and PT measurements.

Low temperature PL spectra presented in Fig. 2(a) clearly show quite significant Stokes shift with the PL peak energy falling significantly below the absorption edge energy. The shift can be attributed to large random fluctuations of the direct bandgap that lead to exciton localization and a shift of the PL peak to lower energy¹⁹. Additionally, as shown in Fig. 2(a) clear observation of the optical transitions with PT provides further support for the direct nature of the optical transitions^{9–11,30,31}. In PT spectra two resonances can be resolved. These resonances are related to optical transitions between the VB edge and the E_- band (transition labeled as E_-) and the spin-orbit split-off band and the E_- band (transition labeled as $E_- + \Delta_{\text{SO}}$). In order to determine energies and broadening of these transitions, PT resonances are fitted by Aspnes' formula³²

$$\frac{\Delta T}{T}(E) = \text{Re} \left[C e^{i\vartheta} (E - E_j + i\Gamma)^{-m} \right] + f(E), \quad (1)$$

where $\frac{\Delta T}{T}(E)$ is the energy dependent PT signal, C and ϑ are the amplitude and phase of the resonance, and E_j and Γ are the energy and the broadening parameter of the optical transition, respectively, m depends on the type of optical transition and is assumed to be $m = 2.5$ in this case. $f(E)$ is a parabolic function which simulates the background sign related to Fabry-Perot oscillation^{33,34} or a wing of GaP-related signal. The fitted curves are shown as

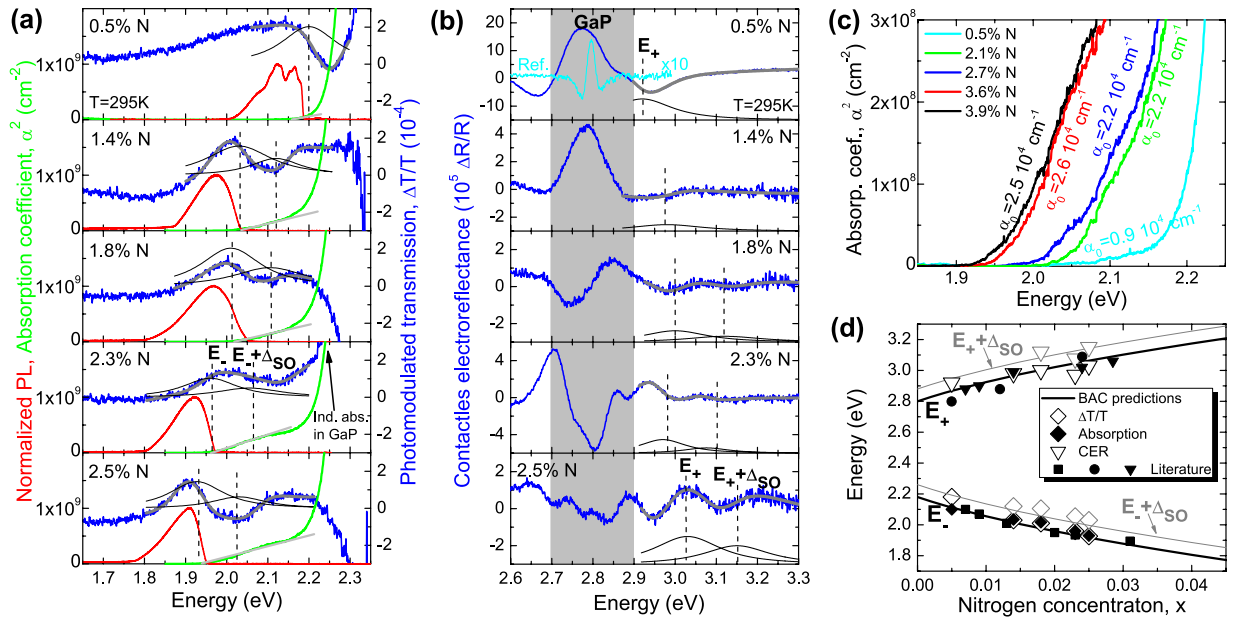


Figure 2. (a) Absorption (green lines) and photomodulated transmission (blue lines) spectra of $\text{GaN}_x\text{P}_{1-x}$ layers with various N concentrations measured in the vicinity of E_- and $E_- + \Delta_{\text{SO}}$ transitions together with low temperature photoluminescence spectra (red line). (b) Contactless electroreflectance spectra of GaNP layers (blue lines) measured in the vicinity of E_+ transition. The fitting curves are shown by thick grey lines. Modulus of individual resonances are shown by thin solid black lines. Since GaNP layers are tensile strained the fundamental transition is between the light-hole band and the E_- band. In our fitting the strain induced splitting between light- and heavy-hole subbands is neglected and a single resonance is used to simulate the E_- and the E_+ transition. (c) Absorption curve in the vicinity of absorption edge used to determine the absorption constant α_0 . (d) Comparison of energies of E_- , $E_- + \Delta_{\text{SO}}$, E_+ , and $E_- + \Delta_{\text{SO}}$ transitions obtained from BAC model for GaNP with various N concentrations (solid lines) with experimental data (points) obtained in this work and taken from the literature^{26,27}.

thick grey lines in Fig. 2(a) together with the moduli of PT resonances, which are shown as solid black lines. The moduli of PT resonances (ρ) are obtained according to Eq.(2) with parameters taken from the fit.

$$\Delta\rho(E) = \frac{|C|}{\left[(E - E_j)^2 + \Gamma^2\right]^{\frac{m}{2}}} \quad (2)$$

Comparing the energy of E_- transition, which is a direct optical transition in PT, with the energy of absorption edge determined from the α^2 plot it has been found that they are the same within the range of experimental uncertainties. It confirms the direct gap character of bandgap observed in absorption for this alloy.

The photomodulated transmission cannot be used to observe the higher energy E_+ transitions in GaNP because of a strong absorption of the GaP substrate. Therefore, the CER has been applied to measure this transition. In this case, besides the E_+ transition in GaNP, very strong resonances related to E_0 and $E_0 + \Delta_{\text{SO}}$ transitions in GaP buffer layer are also observed. The observation of these resonances indicates a weak light absorption in this spectral range in GaNP layer which confirms intermediate character of E_- band in GaNP whose density of states is determined by the N content.

An interesting feature of the CER spectra shown in Fig. 2(b) is the large broadening of GaP-related resonances measured for samples with the top GaNP layer. This broadening is few times larger than the broadening of the same transition in the reference GaP epilayer, see CER spectrum for the reference sample plotted by light blue line in Fig. 2(b). In general an increase of broadening of CER resonance can be attributed to the alloying related bandgap fluctuations and/or a strong built-in electric field. Since the GaP buffer is a binary compound, the large broadening of GaP-related resonance in the studied samples can be only due to a strong band bending at the GaNP/GaP interface.

As shown previously^{35,36}, the Fermi level in as-grown GaInNAs(Sb) tends to be pinned at the Fermi stabilization energy (E_{FS}) because of native defects in this material. The same phenomenon is expected for GaNP(As) alloys. In this case the E_{FS} is located in the middle of the bandgap as shown in Fig. 1. On the other hand the Fermi level in n -type GaP is located near the conduction band. This leads to formation of a strong band bending at the GaNP(As)/GaP interface (i.e., an inhomogeneous built-in electric field in these layers). The CER signal originates from modulation of the built-in electric field. In the case of spatially homogeneous field this leads to a CER resonance followed by a Franz-Keldysh oscillation (FKO)^{37–39}. However, the FKO can be damped and not observed for an inhomogeneous built-in electric field. In such case presence of a strong inhomogeneous electric field manifests itself in a large broadening of CER resonance as is observed for GaNP samples.

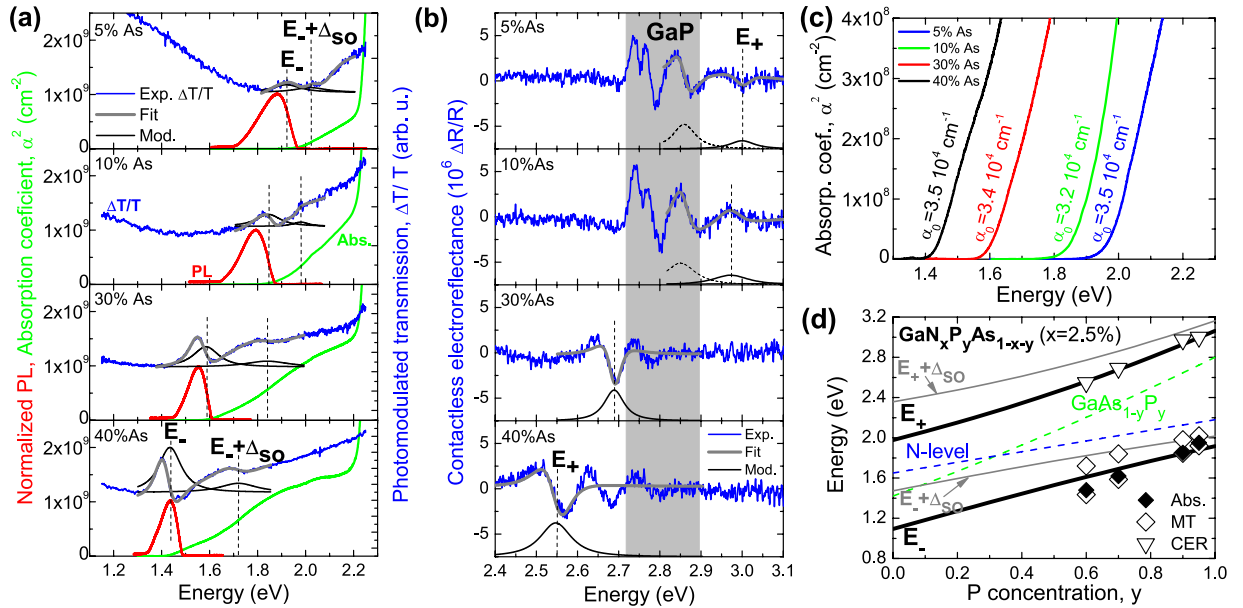


Figure 3. (a) Absorption (green lines) and photomodulated transmission (blue lines) spectra of $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ layers with $x = 0.025$ and various P concentrations measured in the vicinity of E_- and $E_- + \Delta_{\text{SO}}$ transitions together with low temperature photoluminescence spectra (red line). (b) Contactless electroreflectance spectra of GaNPAs layers (blue lines) measured in the vicinity of the E_+ transition. The fitting curves are shown by thick grey lines. Modulus of individual resonances are shown by thin solid black lines. Due to the compressive strain in the studied GaNPAs layers a splitting between light- and heavy-hole subbands is present but this splitting is neglected and a single resonance is used to simulate the E_- and the E_+ transition. This resonance is attributed to the heavy-hole subband as the dominant contributor in this case. (c) Absorption curve in the vicinity of absorption edge used to determine the absorption constant α_0 . (d) Comparison of energies of E_- , $E_- + \Delta_{\text{SO}}$, E_+ , and $E_+ + \Delta_{\text{SO}}$ transitions obtained from BAC model for $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ with $x = 0.025$ and various P concentrations (solid lines) with experimental data (points).

The CER resonances related to E_+ and $E_+ + \Delta_{\text{SO}}$ transitions in GaNP are expected for energies higher than 2.8 eV, i.e. above the direct gap of pure GaP. Therefore it is possible to observe these transitions separately from the transitions originating from the GaP epilayers/substrates. Fits of the experimental curves fitted with Aspnes' formula³² given by Eq. (2) are shown as thick grey lines in Fig. 2(b) together with the moduli of CER resonances, represented by solid black lines.

Figure 2(c) presents the square of the absorption coefficient in the vicinity of absorption edge. These results allow to determine the absorption edge and the absorption constant α_0 . Values of α_0 determined for GaNP layers with different N concentrations are given in Fig. 2(c) and discussed in details in the next part of this paper.

Figure 2(d) compares the energies of E_- , $E_- + \Delta_{\text{SO}}$, E_+ , and $E_+ + \Delta_{\text{SO}}$ transitions calculated using the BAC model with the energies of direct optical transitions determined from PT (open diamond points) and CER (open triangle points) measurements as well as the absorption edge energy determined from transmission and reflectance measurements (solid diamond points). In addition, energies of E_- and E_+ transitions taken from the literature^{26,27} are shown in this figure (other solid points). Taking into account the experimental uncertainties the found energies of E_- , $E_- + \Delta_{\text{SO}}$, E_+ , and $E_+ + \Delta_{\text{SO}}$ transitions are in a good agreement with BAC predictions.

Optical properties of GaNPAs films. The results of previous section show that the BAC model well describes the N dependence of the electronic band structure of GaNP alloy. The current section discusses the effect of BAC interaction in GaNPAs. It is shown in Fig. 1(a) that the energy separation between the nitrogen level and the conduction band of the GaPAs host varies with As concentration. Figure 3(a) shows room temperature absorption (green line), photomodulated transmission (blue line), and low temperature photoluminescence (red line) spectra for $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ layers with $x \sim 0.025$ and various P concentrations ($y = 0.6, 0.7, 0.9$, and 0.95) in the vicinity of E_- and $E_- + \Delta_{\text{SO}}$ transitions. The PT spectra are fitted with Aspnes' formula. The moduli of PT resonances are plotted as thin black lines in Fig. 3(a). Similar as in the case of GaNP layers two resonances are observed in PT spectra. They are attributed to optical transitions between the VB and the E_- band (transition labeled as E_-) and between spin-orbit split-off VB and the E_- band (transition labeled as $E_- + \Delta_{\text{SO}}$).

The results presented in Fig. 3 show that similarly as in the case GaNP the P-rich GaNPAs alloy exhibits a direct band gap. This is evidenced by well resolved PT resonances as well as the fundamental absorption edge with linear dependence of α^2 on the photon energy. This conclusion is further supported by a strong PL observed for these samples, see Fig. 3(a).

The results of CER measurements of GaNPAs films are presented in Fig. 3(b). Notably the E_+ transition energy in $\text{GaN}_x\text{P}_y\text{As}_{1-x-y}$ shifts from about 3 eV for $y = 0.95$ to 2.55 eV for $y = 0.6$. CER spectra for these samples in the vicinity of E_+ transitions are shown in Fig. 3(b). As seen in Fig. 3(b) the optical transitions associated with GaP

about $0.2\text{--}0.4 \times 10^5 \text{ cm}^{-1}$ for VB \rightarrow IB transition (E_- transition). This indicates that the GaNPs thin films have the optical absorption for VB to IB transitions strong enough to be used in thin film IB solar cells.

It is important to note that the IB \rightarrow CB transition is also interesting to explore from the viewpoint of application of GaNP(As) alloys in IB solar cells. But for this purpose *n*-type samples are needed since IB should be partially occupied by electrons in order to obtain an absorption between the IB and the CB. The studied samples are nominally undoped with the unintentional electron concentration below 10^{17} cm^{-3} and thereby they are not useful for this study. Therefore a further studies of *n*-type GaNP(As) will be very interesting in this case.

Summary

It is shown that incorporation of a few percent of nitrogen into P-rich GaPAs leads to formation of intermediate band and the change of the nature of the fundamental band gap from indirect to direct. The direct band gap in GaNPs alloys has been confirmed by absorption and PL measurements. In addition direct optical transitions between the valence band and the upper conduction band (E_+ transition) have been clearly identified in optical CER spectra. The experimentally observed N-induced modification of the electronic band structure are well accounted for by the BAC model. The observed formation of the isolated intermediate band offers a potential of using P-rich GaNPs alloys for intermediate band solar cells.

Methods

BAC model. According to the BAC model in GaNP and GaNPs¹¹ the interaction of N-related states with a conduction band minimum of GaPAs host is modeled using perturbation theory by following Hamiltonian:

$$H_{BAC} = \begin{pmatrix} E_M(k) & C_{NM}\sqrt{x} \\ C_{NM}\sqrt{x} & E_N \end{pmatrix} \quad (4)$$

where x is the mole fraction of substitutional N atoms and C_{NM} is a constant, which describes the interaction between the nitrogen level and the conduction band. This constant depends on the semiconductor matrix and can be determined experimentally^{9–11}. $E_M(k)$ is the energy dispersion of the lowest conduction band of the III-V host, which can be calculated using *kp* method¹¹, and E_N is the energy of N-related states, all referenced to the top of the valence band of the III-V semiconductor host.

The interaction of dispersionless N-related states with the conduction band states leads two highly non-parabolic subbands, $E_-(k)$ and $E_+(k)$, which are given by Eq. (5):

$$E_{\pm}(k) = \frac{1}{2}[E_N + E_M(k) \pm \sqrt{[E_N - E_M(k)]^2 + 4C_{NM}^2 x}] \quad (5)$$

Sample growth. 100 nm-thick GaNP and GaNPs films were grown on GaP(001) substrates by Molecular Beam Epitaxy (MBE) using a Riber Compact 21 solid source MBE system⁴⁰. For both sets of samples, growth temperature was 450 °C (as measured by an optical pyrometer). Nitrogen has been incorporated using a valved, RF plasma source with a nominal 0.5 sccm N_2 flow and a RF power of 400 W⁴¹. A V/III beam equivalent pressure ratio was set equal to or greater than 10 in order to control the incorporation of nitrogen^{42,43}. In GaNPs samples, the As flux has been varied while maintaining the P flux constant to vary the As/P ratio and to change the As content in the sample: thus leading to P nominal compositions of 95%, 90%, 70% and 60% respectively, and nominal N content of 2.5% (which is assumed to increase from one sample to the other within 0.5%: with the roughness induced by the strain relaxation)⁴³.

Optical measurements. Transmission (T) and reflection (R) spectra have been measured to determine absorption spectra. T and R measurements have been performed on a single grating 0.55-m focal-length monochromator. The optical signal has been detected by an Si *pin* photodiode using a lock-in amplifier. The absorption coefficient (α) has been calculated according to the textbook formula $\alpha(E) = -\frac{1}{d} \ln\left(\frac{T}{(1-R)^2}\right)$, where d is the thickness of the GaNP(As) film. PT and CER measurements have been performed on the same monochromator. In PT measurements the band bending inside the samples was modulated by a beam from a laser emitting at 532 nm. This beam has been modulated by a mechanical chopper at a frequency of 280 Hz. For CER measurements samples have been placed in a capacitor with the top electrode made from a copper-wire mesh which is semi-transparent for light. This electrode has been kept at a distance of ~ 0.5 mm from the sample surface while the sample itself was fixed to the bottom copper electrode by a silver paste. The distance between the sample surface and the top electrode has been ~ 0.5 mm. A maximum peak-to-peak alternating voltage of ~ 3.5 kV with the frequency of 285 Hz was applied. In both PT and CER measurements the sample has been illuminated by spectrum of white light from a halogen lamp (150 W) at near normal incidence⁴⁴. The light transmitted through (PT case) and reflected from (CER case) the sample has been dispersed through the monochromator and detected by an Si detector or a photomultiplier. Phase-sensitive detection of the PT and CER signal was performed using a lock-in amplifier. Other relevant details on CER measurements can be found in ref.⁴⁴. For PL measurements the samples have been excited by the 405 nm line of the semiconductor laser and PL signal has been measured on a very similar but independent set-up, i.e., a single grating 0.55-m focal-length monochromator combined with the liquid nitrogen cooled Si CCD detector.

References

- Wolf, M. Limitations and possibilities for improvements of photovoltaic solar energy converters. Part I: Considerations for earth's surface operation. *Proc. IRE* **48**, 1246–1263 (1960).
- Luque, A. & Martí, A. Increasing the Efficiency of Ideal Solar Cells by Photon Induced Transitions at Intermediate Levels. *Phys. Rev. Lett.* **78**, 5014 (1997).
- Luque, A. & Martí, A. The Intermediate Band Solar Cell: Progress Toward the Realization of an Attractive Concept. *Adv. Mater.* **22**, 160 (2010).
- Ramiro, I., Martí, A., Antolín, E. & Luque, A. Review of Experimental Results Related to the Operation of Intermediate Band Solar Cells. *IEEE J. Photovoltaics* **4**, 736 (2014).
- Okada, Y. *et al.* Intermediate band solar cells: Recent progress and future directions. *Appl. Phys. Rev.* **2**, 021302 (2015).
- Sogabe, T. *et al.* Intermediate-band dynamics of quantum dots solar cell in concentrator photovoltaic modules. *Scientific Reports* **4**, 4792 (2014).
- Liu, W. S., Wu, H. M., Tsao, F. H., Hsu, T. L. & Chyi, J. I. Improving the characteristics of intermediate-band solar cell devices using a vertically aligned InAs/GaAsSb quantum dot structure. *Solar Energy Materials and Solar Cells* **105**, 237 (2012).
- Luque, A., Martí, A., Antolín, E. & Tablero, C. Intermediate bands versus levels in non-radiative recombination. *Physica B* **382**, 320 (2006).
- Shan, W. *et al.* Band anticrossing in GaInNAs alloys. *Phys. Rev. Lett.* **82**, 1221 (1999).
- Welna, M., Kudrawiec, R., Nabetani, Y. & Walukiewicz, W. Band anticrossing in ZnOSe highly mismatched alloy. *Appl. Phys. Express* **7**, 071202 (2014).
- Kudrawiec, R. *et al.* Electronic Band Structure of GaN_xP_yAs_{1-x-y} Highly Mismatched Alloys: Suitability for Intermediate-Band Solar Cells. *Phys. Rev. Appl.* **1**, 034007 (2014).
- Kuang, Y. J. *et al.* GaNAsP: An intermediate band semiconductor grown by gas-source molecular beam epitaxy. *Appl. Phys. Lett.* **102**, 112105 (2013).
- Jussila, H., Kivisaari, P., Lemettinen, J., Tanaka, T. & Sopanen, M. Two-Photon Absorption in GaAs_{1-x-y}P_yN_x Intermediate-Band Solar Cells. *Phys. Rev. Applied* **3**, 054007 (2015).
- Ilahi, S. *et al.* Optical absorption and thermal conductivity of GaAsPN absorbers grown on GaP in view of their use in multijunction solar cells. *Solar Energy Materials and Solar Cells* **141**, 291 (2015).
- Yamane, K. *et al.* Growth of a lattice-matched GaAsPN p–i–n junction on a Si substrate for monolithic III–V/Si tandem solar cells. *Appl. Phys. Express* **10**, 075504 (2017).
- Kunert, B., Volz, K., Koch, J. & Stolz, W. W. Direct-band-gap Ga(NAsP)-material system pseudomorphically grown on GaP substrate. *Appl. Phys. Lett.* **88**, 182108 (2006).
- Kunert, B., Volz, K., Nemeth, I. & Stolz, W. Luminescence investigations of the GaP-based dilute nitride Ga(NAsP) material system. *J. Luminescence* **121**, 361 (2006).
- Liebich, S. *et al.* Laser operation of Ga(NAsP) lattice-matched to (001) silicon substrate. *Appl. Phys. Lett.* **99**, 071109 (2011).
- Baranowski, M. *et al.* Temperature evolution of carrier dynamics in GaN_xP_yAs_{1-y-x} alloys. *J. Appl. Phys.* **117**, 175702 (2015).
- Biwa, G., Yaguchi, H., Onabe, K. & Shiraki, Y. Photoluminescence and photoluminescence-excitation spectroscopy of GaPAsN/GaP lattice-matched multiple quantum well structures. *J. Cryst. Growth* **195**, 574 (1998).
- Robert, C. *et al.* Room temperature operation of GaAsP(N)/GaP(N) quantum well based light-emitting diodes: Effect of the incorporation of nitrogen. *Appl. Phys. Lett.* **98**, 251110 (2011).
- Babichev, A. V. *et al.* Ultra-wide electroluminescence spectrum of LED heterostructures based on GaPAsN semiconductor alloys. *Semiconductors* **48**, 501 (2014).
- Robert, C., Perrin, M., Cornet, C., Even, J. & Jancu, J. M. Atomistic calculations of Ga(NAsP)/GaP(N) quantum wells on silicon substrate: Band structure and optical gain. *Appl. Phys. Lett.* **100**, 111901 (2012).
- Xin, H. P., Tu, C. W., Zhang, Y. & Mascarenhas, A. Effects of nitrogen on the band structure of GaN_xP_{1-x} alloys. *Appl. Phys. Lett.* **76**, 1267 (2000).
- Kent, P. R. C. & Zunger, A. Theory of electronic structure evolution in GaAsN and GaPN alloys. *Phys. Rev. B* **64**, 115208 (2001).
- Wu, J. *et al.* Band anticrossing in GaP_{1-x}N_x alloys. *Phys. Rev. B* **65**, 241303(R) (2002).
- Buyanova, I. A. *et al.* Analysis of band anticrossing in GaN_xP_{1-x} alloys. *Phys. Rev. B* **70**, 085209 (2004).
- Gungerich, M. *et al.* Experimental and theoretical investigation of the conduction band edge of GaN_xP_{1-x}. *Phys. Rev. B* **74**, 241202 (2006).
- O'Reilly, E. P., Lindsay, A., Klar, P. J., Polimeni, A. & Capizzi, M. Trends in the electronic structure of dilute nitride alloys. *Semicond. Sci. Technol.* **24**, 033001 (2009).
- Kudrawiec, R. *et al.* Photoreflectance, photoluminescence, and microphotoluminescence study of optical transitions between delocalized and localized states in GaN_{0.02}As_{0.98}, Ga_{0.95}In_{0.05}N_{0.02}As_{0.98}, and GaN_{0.02}As_{0.90}Sb_{0.08} layers. *Phys. Rev. B* **88**, 125201 (2013).
- Dybala, F. *et al.* Pressure coefficients for direct optical transitions in MoS₂, MoSe₂, WS₂, and WSe₂ crystals and semiconductor to metal transitions. *Scientific Reports* **6**, 26663 (2016).
- Aspnes, D. E. Third-derivative modulation spectroscopy with low-field electroreflectance. *Surf. Sci.* **37**, 418 (1973).
- Kudrawiec, R. *et al.* Interference effects in electromodulation spectroscopy applied to GaAs-based structures: A comparison of photoreflectance and contactless electroreflectance. *Appl. Phys. Lett.* **86**, 091115 (2005).
- Welna, M. *et al.* Contactless electroreflectance spectroscopy of ZnO/ZnMgO quantum wells: Optical transitions and Fabry-Perot features. *Physica Status Solidi (a)* **212**, 780 (2015).
- Kudrawiec, R. *et al.* On the Fermi level pinning in as-grown GaInNAs(Sb)/GaAs quantum wells with indium content of 8–32%. *J. Appl. Phys.* **104**, 033526 (2008).
- Kudrawiec, R. & Misiewicz, J. Evidence for Fermi level shift in GaInAs/GaAs quantum well upon nitrogen incorporation. *Solid State Commun.* **150**, 227 (2010).
- Shen, H. & Dutta, M. Franz–Keldysh oscillations in modulation spectroscopy. *J. Appl. Phys.* **78**, 2151 (1995).
- Kudrawiec, R. *et al.* Contactless electroreflectance study of the Fermi level pinning on GaSb surface in n-type and p-type GaSb Van Hoof structures. *J. Appl. Phys.* **112**, 123513 (2012).
- Kudrawiec, R. *et al.* Contactless electroreflectance studies of surface potential barrier for N- and Ga-face epilayers grown by molecular beam epitaxy. *Appl. Phys. Lett.* **103**, 052107 (2013).
- Quinci, T. *et al.* Defects limitation in epitaxial GaP on bisterred Si surface using UHV-CVD-MBE growth cluster. *J. Cryst. Growth* **380**, 157 (2013).
- Almosni, S. *et al.* Evaluation of InGaPN and GaAsPN materials lattice-matched to Si for multi-junction solar cells. *J. Appl. Phys.* **113**, 123509 (2013).
- Kuyyalil, J. *et al.* Nitrogen–phosphorus competition in the molecular beam epitaxy of GaPN. *J. Cryst. Growth* **377**, 17 (2013).
- Cornet, C. *et al.* Preferential incorporation of substitutional nitrogen near the atomic step edges in diluted nitride alloys. *Appl. Phys. Lett.* **101**, 251906 (2012).
- Kudrawiec, R. J. & Misiewicz, J. Chapter 4: Optical Modulation Spectroscopy, pp. 95–124 in *Semiconductor Research -Experimental Techniques*, Springer Series in Materials Science Volume **150** (2012).

Acknowledgements

This work was performed within the grant of the National Science Centre Poland through grant HARMONIA 2013/10/M/ST3/00638. In addition, this work was supported by the Director, Office of Science, Office of Basic Energy Sciences, Materials Sciences and Engineering Division, of the U.S. Department of Energy under Contract No. DE-AC02-05CH11231. M. P. Polak acknowledges the support within the “Diamond Grant” no. DI2013 006143.

Author Contributions

K.Z. performed optical measurements; M.G. and M.P. performed calculations; S.A., A.L., Ch.C., and O.D. were responsible for the sample growth and their structural characterization; W.W. discussed results and revised manuscript; and R.K. supervised this project, analyzed results and wrote the manuscript. All authors contributed in the preparing this paper.

Additional Information

Competing Interests: The authors declare that they have no competing interests.

Publisher's note: Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2017