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## Fast solution to determine the elastic behavior of 3D reconstructed biopolymer cellular structures

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## Goal

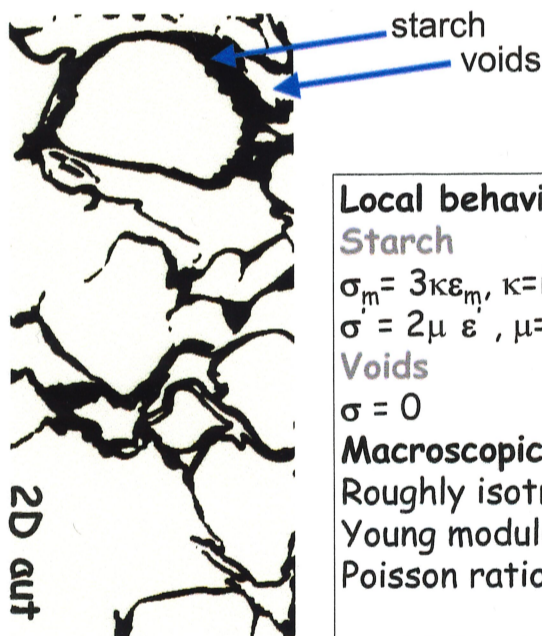
Numerical homogenization vs. experimental results

Investigate the accuracy of the FFT numerical method for predicting the mechanical behavior of complex microstructures, using comparison with experimental data.

## Microstructures

Choice of a two-phases (starch+70-90% of voids) biopolymer cellular material most "amenable" to numerical simulations:

- linear elastic regime,
- no imperfect interface effects,
- starch is quasi-incompressible (Guessasma 2009, Babin et al 2007),
- local behavior depends on only 1 elastic modulus, determined experimentally,
- choice of several sets of microstructures with highly-different macroscopic properties, at various density (volume fraction of starch).



**Local behavior**  
**Starch**  
 $\sigma_m = 3\kappa\varepsilon_m$ ,  $\kappa = \text{infinity}$   
 $\sigma = 2\mu\varepsilon$ ,  $\mu = \text{shear modulus}$   
**Void**  
 $\sigma = 0$   
**Macroscopic behavior**  
 Roughly isotropic:  
 Young modulus E  
 Poisson ratio  $\nu$

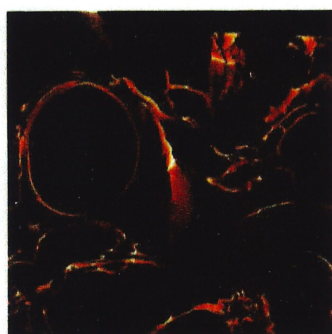
## FFT numerical method

Fixed-point algorithm based on the Lippmann-Schwinger equations (Moulinec and Suquet, 1994). Directly applied to **microstructure images** (no meshing). Local behavior and fields admissibility taken into account in the real space and Fourier domain respectively.

$$\varepsilon_{ij}(\mathbf{x}) = \varepsilon_{ij}^0 + \int d^2y G_{ij,kl}^0(\mathbf{x}-\mathbf{y}) \tau_{kl}(\mathbf{y})$$

$$\tau_{ij}(\mathbf{x}) \equiv \sigma_{ij}(\mathbf{x}) - L_{ij,kl}^{(0)} \varepsilon_{kl}(\mathbf{x})$$

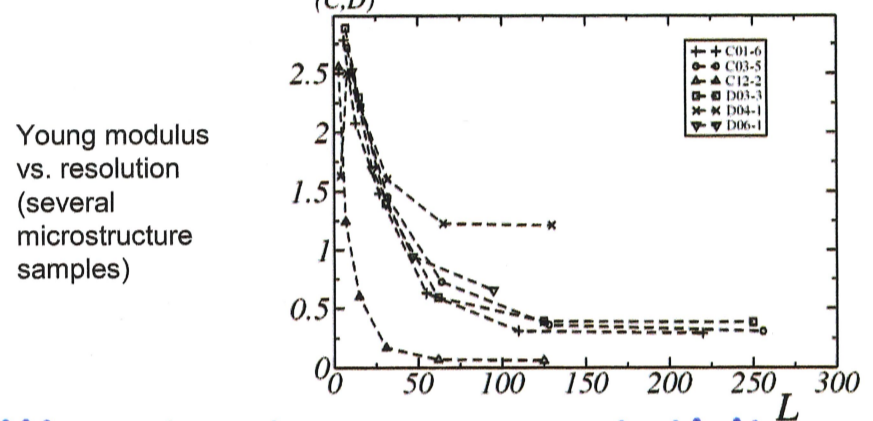
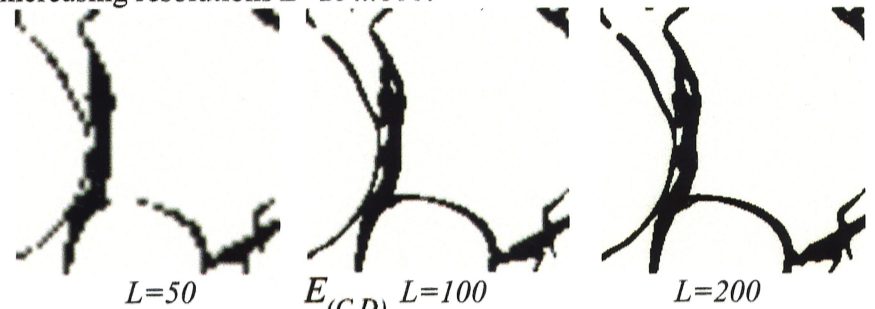
Use of the improved "Augmented Lagrangian method" (Michel et al, 2001) to treat **infinitely-contrasted media**. **Periodic boundary conditions**, better suited to numerical homogenization in terms of R.V.E. (Kanit et al, 2003).



Strongly localized mean strain field, induced by foam-type microstructure

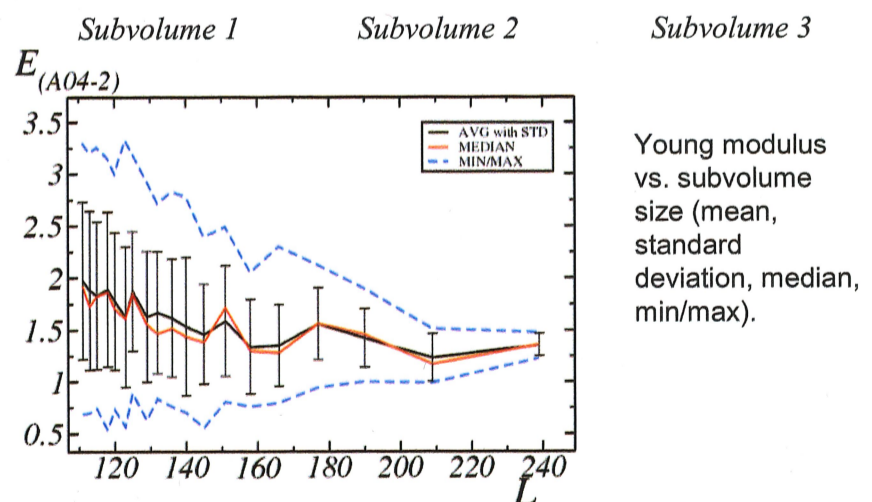
## Effect of image resolution

Effect of resolution on the macroscopic elastic behaviour as computed with the FFT method. Same samples discretized over  $L^3$  voxels with increasing resolutions  $L=25 \dots 800$ .



## Microstructure representativity

Study of size effect: elastic properties of random subvolumes of each sample at various sizes  $L^3$ .



## Macroscopic behavior and conclusion

Differing elastic properties of sets A and B, due to distinct microstructures, recovered by FFT numerical estimates.

For linear elasticity, numerical estimates globally more robust and reliable; a few aberrant experimental points taken out.

