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Novel erbia-yttria co-doped zirconia fluorescent thermal history sensor

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Abstract

Thermochromic pigments are commonly used for off-line temperature mapping on components from systems operating at a temperature higher than 1073 K. However, their temperature resolution is often limited by the discrete number of color transitions they offer. This paper investigates the potential of erbia-yttria co-doped zirconia as a fluorescent thermal history sensor alternative to thermochromic pigments. Samples of yttria-stabilized zirconia powder (YSZ, 8.3 mol% $\text{YO}_{1.5}$) doped with 1.5 mol% $\text{ErO}_{1.5}$ and synthesized by a sol-gel route are calcined for 15 minutes under isothermal conditions between 1173 and 1423 K. The effects of temperature on their crystal structure and room temperature fluorescence properties are then studied. Results show a steady increase of the crystallinity of the powders with temperature, causing a significant and permanent increase of the emission intensity and fluorescence lifetime which could be used to determine temperature with a calculated theoretical resolution lower than 1 K for intensity. The intensity ratio obtained using a temperature insensitive YSZ:Eu³⁺ reference phosphor is proposed as a more robust parameter regarding experimental conditions for determining thermal history. Finally, the possibilities for integrating this fluorescent marker into sol-gel deposited coatings for future practical thermal history sensing applications is also discussed.

Keywords: thermal history sensor, sol-gel, luminescence, temperature measurement

1. Introduction

The knowledge of the temperature of components subject to high-temperature environments is a key factor for the development and safe use of numerous systems such as boilers, chemical reactors, furnaces, engines and gas turbines, as it plays a critical role in many degradation mechanisms [1–4]. Temperature sensitive paints are the current mean for determining the thermal history of components in confined areas or in operating systems where the direct measurement of temperature over large surfaces is not possible or not convenient [5–8]. Following exposure to operating temperature for a known and short period of time under stationary conditions, these coatings allow to derive maps of temperature of exposure from the analysis, upon return at room temperature, of the thermoactivated color changes undergone by the embedded thermochromic pigments. However, the limited and discrete number of available color transitions distributed over wide

temperature ranges of a few hundreds of degrees Celsius often limits the temperature resolution of these paints to a few tens of degrees or more. The analysis of the color changes can also be difficult and highly dependent on operator and observation conditions. At last, many of the compositions of such paints include chemical substances falling under the restriction of international regulations such as REACH, further pushing the actors of the sector to develop alternative pigments.

One of the solutions under investigation over the last few years is the implementation of photoluminescent markers, whose luminescence emission properties are subjected to permanent evolutions with temperature and duration of exposure as a result of thermally activated microstructural or chemical changes occurring within the material [9–19]. With appropriate calibration, the changes observed in at least one luminescence characteristic can therefore be correlated back to the temperature to which the material has been subjected during a previous thermal event under controlled conditions.

Contrary to the discrete changes typical of thermochromic paints, the evolution with temperature of the luminescence parameters may be in most cases continuous over the range of interest [12–16], thus allowing to monitor continuous spectrum of temperature rather than temperature thresholds. Luminescence parameters can also be measured quantitatively with no ambiguity using standard spectroscopic equipments, leading to potentially much higher resolution than thermochromic paints in the determination of temperature.

Recently, the Institut Clément Ader, in collaboration with the CIRIMAT, has developed the synthesis by a sol-gel process of yttria-stabilized zirconia (YSZ) based phosphors functionalized with lanthanide ions such as Sm^{3+} , Eu^{3+} and Er^{3+} for future applications in temperature sensing within thermal barrier coatings for gas turbines [20–22]. These phosphors, not fully crystallized in the initial state, exhibit irreversible changes in their fluorescence properties after annealing in the range 1173–1373 K, suggesting they could be potentially used to record thermal history [20, 22].

This paper reports a study of the potential of an 8.25-YSZ:Er³⁺ phosphor powder (1.5 mol% ErO_{1.5}) synthesized by a sol-gel process as a fluorescent thermal history sensor in the temperature range 1173–1423 K. The effect of a 15 minute-long exposure between 1173 and 1423 K on its crystal structure and fluorescence characteristics (spectra, intensity and decay time) is investigated in order to determine its sensitivity to thermal history in this temperature range. The robustness of an intensity ratio approach, using a temperature insensitive YSZ:Eu³⁺ phosphor as a reference, and the possibilities for integrating these fluorescent markers into coatings are then investigated and discussed in order to evaluate the relevance of this marker as an alternative to current thermochromic paints for future off-line temperature mapping applications.

2. Fluorescent thermal history sensors: the choice of YSZ:Er³⁺

The concept of fluorescent thermal history sensor is mainly based on the occurrence of thermally activated microstructural and/or chemical changes within the sensing material, that produce a permanent evolution in at least one of its fluorescence characteristics, which in turn is sufficient to derive the heating temperature from a past thermal event. Many physico-chemical changes can indeed have a significant impact on the energy level distribution of activators and on the probability of occurrence of radiative electronic transitions, thus affecting spectral (emission wavelength and intensity) and temporal (decay time) luminescence properties. Currently, four main temperature-driven mechanisms have been identified which, providing that they are achieved under controlled conditions (temperature and exposure time), can serve to determine thermal history: (i) phase transformations [9, 10, 23], (ii) diffusion of activator or sensitizer/quencher species [12, 24], (iii) oxidation of activator ions [16–19] and

(iv) crystallization of the host matrix [11–15, 25, 26]. The latter mechanism is of particular interest in the case of inorganic phosphors produced by a sol-gel process or many similar soft chemistry routes, since an amorphous or incomplete crystallized state is generally obtained due to the relatively low temperatures involved during their fabrication, typically lower than 873–1073 K [27–29]. The temperature dependence of the crystallization rate constant, that follows an Arrhenius type relation [30], predicts a strong dependence on temperature of the microstructural state of the material which might thus pass on its fluorescence properties. Numerous studies carried out on the lanthanide activator ions Eu^{3+} (in Al₂O₃ [14], ZrO₂ [31], YAG [32], Y₂O₃ [13, 33, 34], YVO₄ [35], CaY₂Si₃O₁₂ [36]), Er^{3+} (in SiO₂ [25], GdAlO₃ [26], ZrO₂ [37], Gd₂TiO₇ [38]), Tb^{3+} (in Y₂O₃ [14, 34], Y₂SiO₅ [12]), Pr^{3+} (in PbO–Sb₂O₃–B₂O₃ [39]) and Ce^{3+} (in YAG [40]), have shown that their emission intensity at room temperature, initially nonexistent or weak, is greatly improved with the increase of crystallinity resulting from annealing at high temperature, typically in the range 1073–1573 K. This effect, that goes along with the sharpening of emission peaks and generally an increase of the luminescence lifetime [12–14, 32, 34, 38], results mainly from the coarsening of the average crystal size, the uniformization of the crystal field as well as the reduction of the number of crystal defects and residual organic groups acting as luminescence quenchers [12, 33, 34, 38].

Figure 1 shows the x-ray diffraction patterns and the fluorescence spectra of the YSZ:Er³⁺ phosphor powder produced by a sol-gel route at the Institut Clément Ader and CIRIMAT [21] in the initial state and after further annealing at 1373 K for 2 h.

In the initial state, the x-ray diffraction pattern in figure 1(a) is in good agreement with that of the metastable tetragonal phase t' of 10-YSZ (pattern #01-082-1241 from the ICSD database), although the relatively weak and broad peaks indicate an incomplete crystallization with small average crystal size. At this stage, the powder produces very weak fluorescence emission lines for wavelengths at 545 and 562 nm originating from the $^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$ transition of Er^{3+} under the excitation of a laser emitting at 532 nm (figure 1(b)). Annealing for 2 h at 1373 K leads to a significant increase of the diffraction peak intensity, to a reduction of the full width at half-maximum and to the deconvolution of the peak doublets at $2\theta = 35^\circ, 60^\circ, 74^\circ, 84^\circ$ as a result of the enhancement of crystallinity and crystal growth (figure 1(a)). It also generates an increase by a factor of more than 50 of the emission lines intensity (figure 1(b)). Finally, the decay time of the peak at 562 nm is observed to increase from ~16 to ~43 μs . Such pronounced behavior was not observed with Sm^{3+} , Eu^{3+} , Dy^{3+} and Tm^{3+} powders produced under the same conditions, for which the emission intensity increase after the enhancement of crystallinity was limited to a factor of 1.6 to 6 [22].

Therefore, the specific energy level distribution of the 4f orbitals of Er^{3+} ions [41], with no wide energy gap (figure 2),

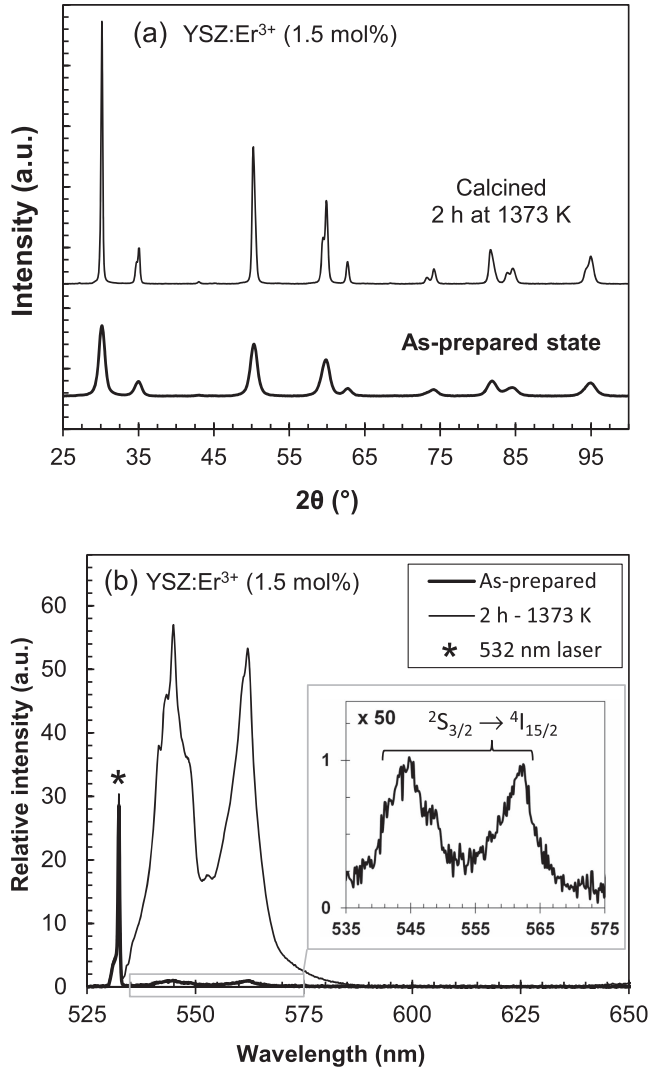


Figure 1. Comparison of (a) x-ray diffraction patterns and (b) fluorescence spectra of 1.5 mol% Er³⁺ doped YSZ powders synthesized by a sol-gel route in the as-prepared state (incomplete crystallization) and after 2 h annealing at 1373 K.

is assumed to be at the origin of the particularly strong sensitivity of their emissions to the environment in this sol-gel synthesized host matrix. The presence of numerous and relatively close energy levels between the emitting states ²H_{11/2} and ⁴S_{3/2} and the ground state ⁴I_{15/2} might indeed promote quenching in this YSZ matrix by the way of energy transfers with crystal defects having high phonon energy, or with residual organic groups such as hydroxyl HO⁻, carbonate CO₃⁻ and nitrate NO₃⁻ groups found in materials processed at temperature lower than 1073 K, to which Er³⁺ ions are particularly sensitive [38].

This work therefore proposes to investigate the extent of this temperature sensitivity in the context of short time heat treatments to assess for the thermal history sensing capability of this marker, which has the advantage to emit and to be readily excitable in the visible range as well as to be relatively easy to produce.

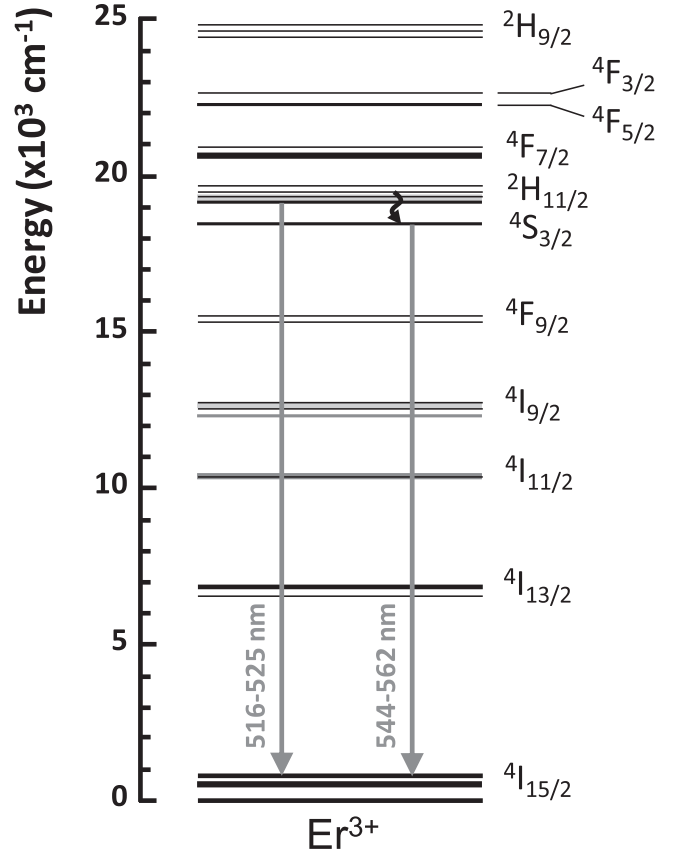


Figure 2. Energy diagram of the 4f configuration of Er³⁺ free ion illustrating its two main radiative electronic transitions in YSZ [41].

3. Methods

8.3-YSZ:Er³⁺ powder of composition (YO_{1.5})_{0.083}(ErO_{1.5})_{0.015}(ZrO₂)_{0.902} was synthesized by the sol-gel route developed in previous works [21, 22]. Samples of that powder were then calcined 15 minutes under isothermal conditions at different temperatures in the range 1173–1423 K with 50 K steps. This temperature range corresponds to the highest operating temperature range of YSZ-based thermal barrier coatings in gas turbines. The evolutions of the crystal structure and the fluorescence characteristics (emission intensity and decay time) of the samples were then investigated. Details about the preparation of the samples and their characterization can be found in the following paragraphs.

3.1. Sol-gel synthesis and preparation of YSZ:Er³⁺ phosphor powder

YSZ:Er³⁺ powder of composition (YO_{1.5})_{0.083}(ErO_{1.5})_{0.015}(ZrO₂)_{0.902} was synthesized by a sol-gel route using a protocol adapted from Lecomte *et al* [42, 43]. The sol was prepared by mixing zirconium (IV) propoxide (Zr(OPr)₄) (Sigma Aldrich) and appropriate contents of yttrium (III) and Er (III) nitrates (Sigma Aldrich) as precursors in a solution of 1-propanol (Sigma Aldrich) and ultrapure water. Acetylacetone (AcAc, Sigma Aldrich) was used as a complexing agent to control the kinetics of hydrolysis of the zirconium alkoxide [44]. The volume rates of

[AcAc/Zr(OPr)₄] and [H₂O/Zr(OPr)₄] were kept constant at 0.8 and 9.5 respectively. More details about the preparation of the mixtures from the reagents can be found in reference [42]. After 30 min of mechanical stirring, the solution was then maintained for one night (~16 h) at 323 K to accelerate the hydrolysis and condensation of the sol into a bright monolith gel free of precipitates. It was then dried above the supercritical point of 1-propanol ($T_c = 534$ K; $P_c = 5.1$ MPa) in a stainless steel autoclave (Paar Instrument 4621). Temperature was kept constant at 543 K and pressure at 9 MPa for 1 h. Afterwards the solvent was slowly released under isothermal conditions. The resulting brittle monolithic aerogel was calcined for 2 h at 973 K in order to remove the remaining traces of solvents, and ball milled at 250 rpm for 1 h in an agate mortar. The product obtained at this stage, that will be referred to as the *as-prepared state*, is a white powder consisting in 0.2–50 μm particles of partially crystallized YSZ with low levels of fluorescence emission intensity (see figure 1(b)).

3.2. Heat treatments

The sensitivity of the phosphor to the exposure at high temperature was investigated by heat treating samples of the as-prepared powder at different temperatures in the range 1173–1423 K with 50 K steps. This temperature range corresponds to the highest operating temperature range of YSZ-based thermal barrier coatings in gas turbines. $6 \times 3.5 \text{ mm}^2$ porcelain crucibles were filled with 1 g of powder and calcined in air under isothermal conditions for 15 minutes in a box furnace (Nabertherm), followed by quenching down to room temperature in open air. This duration of exposure was selected in order to produce significant evolution of the fluorescence properties over the whole temperature range of interest, while remaining close to the exposure times typical of temperature sensing procedures involving thermochromic paints (3–5 min) [5, 6, 8]. For each treatment, the temperature of exposure was monitored using two S-type thermocouples located above and as close as possible to the sample. Some samples of the as-prepared powder were also annealed for 2 h at 1373 K in air to ensure the complete crystallization of the material into the metastable tetragonal phase of YSZ.

3.3. Characterization

Structural analysis of YSZ powder samples were performed by x-rays diffraction measurements (XRD). XRD patterns were collected by scanning the angular range from 25° to 100° with a PANalytical X'Pert diffractometer equipped with a X'Celerator linear detector using Cu K α (λ (CuK α_1) = 1.5406 Å) as the x-ray source.

The fluorescence studies were performed using a frequency doubled, continuous wave diode-pumped Nd:YAG laser irradiating at 532 nm with maximum power of 1.1 W (CNI laser) as an excitation source. For emission spectra and intensities, the fluorescence signal of each sample was recorded at room temperature over the 400–700 nm range with an Ocean Optics USB2000+ spectrometer equipped

with a 1000 μm optical fiber. A high-pass filter with a cutoff wavelength of 535 nm was used to attenuate the contribution of the laser line at 532 nm.

For the determination of the fluorescence lifetime of the Er³⁺ ions emission line at 562 nm, an acousto-optic modulator operating in digital mode (1250C model, Isomet) and a diaphragm were used to generate ~20 μs long excitation pulses on the sample from the continuous laser beam. The fluorescence intensity decay after each pulse was collected using a Si photomultiplier (SensL) equipped with a narrow band-pass interference filter centered on 560 nm (FWHM ~10 nm). The signal was then digitized and averaged over 128 pulses with a semi-numerical oscilloscope (Fluke) computer-monitored. After subtraction of the constant offset term, each recorded decay $I(t)$ was fitted using the nonlinear least square Levenberg–Marquardt algorithm [45, 46] to a mono-exponential model of the form given in equation (1):

$$I(T) = I_0 e^{-\frac{t}{\tau}}, \quad (1)$$

where I_0 is a pre-exponential constant and τ is the decay time. For each exponential decay fit, an iteratively adapted fitting window was used (sets between $t = 0$ and $t = 10\tau$), according to the method proposed by Brübach *et al* in reference [47]. All measurements were performed at room temperature.

4. Results

4.1. Effects of temperature of exposure on the crystal structure of YSZ

The effects of temperature after 15 minutes long heat treatments on the crystallinity of the YSZ:Er³⁺ powders were investigated by XRD. Figure 3 shows the evolution of the diffraction patterns of samples exposed to various temperatures in the range 1173–1423 K. They remain in all cases typical of the metastable tetragonal phase of YSZ. However, as discussed in section 2 (see figure 2), relatively broad and weak diffraction peaks are observed in the diffraction pattern of the as-prepared powder, indicating that the sample remains in a partially crystallized state due to the relatively low temperature involved in its fabrication (973 K for 2 h). It can be noticed that these peaks steadily become sharper and more intense as the temperature of exposure increases up to 1423 K, as the result of the enhancement of crystallinity and the growth of YSZ crystals [31, 33, 37].

The effects of the heat treatments on crystallinity are already clearly marked at 1173 K, at which a 42% intensity increase and a 20% FWHM reduction is for instance observed on the main (1 0 1)_{*c*} diffraction peak at $2\theta = 30^\circ$. In addition, the comparison of peaks intensity and FWHM with those of the assumed final state obtained after 2 h of annealing at 1373 K shows that the crystallization process is not completed after 15 minutes at 1423 K. Indeed, the maximum intensity of the strongest peak at $2\theta = 30^\circ$ corresponding to the reflection by the family of atomic planes (1 0 1)_{*c*}, taken as an arbitrary

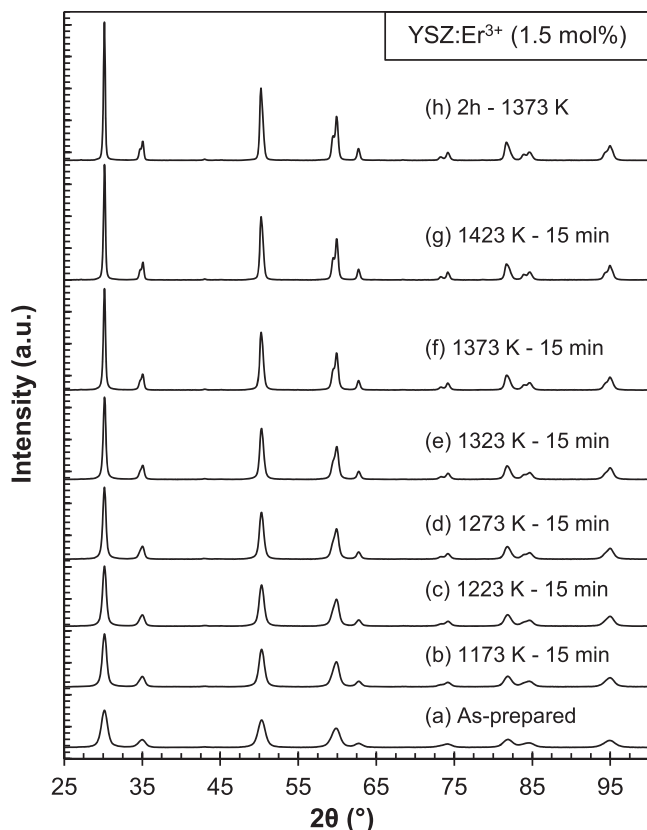


Figure 3. X-ray diffraction patterns of 1.5 mol% YSZ:Er³⁺ aerogel powder samples (a) as-prepared and calcined (b)–(g) for 15 min at various temperatures between 1173 and 1423 K, (h) 2 h at 1373 K for 2 h, showing increase of crystallinity with temperature.

indicator of the degree of crystallinity, represents only 83% of the intensity observed in the final state.

Therefore, 15 min long heat treatments induce a significant and permanent enhancement of the crystallinity over the whole range 1173–1423 K, whose extent appears to be a function of the temperature of exposure. It should also be noted that isothermal heat treatments at a fixed temperature of 1423 K and for various exposure times ranging between 5 and 30 min revealed that more than 80% of the crystallinity changes occurred during the first 5 min of the treatment, after which the evolution with time was much slower.

4.2. Effects of temperature of exposure on the fluorescence properties

4.2.1. Fluorescence intensity. Figure 4(a) illustrates the evolution of the relative fluorescence emission spectra of YSZ:Er³⁺ samples with annealing temperature under the excitation of a green laser emitting at 532 nm. It can be observed that the relative intensity of the two main emission peaks at 545 and 562 nm increases consistently with increasing temperature, up to a factor of 43–48 after 15 min at 1423 K. Calibration curves relating the integrated intensity of both above mentioned peaks to temperature are plotted in figure 4(b).

The relative integrated intensity for both emission lines, conveying the amplification of the initial intensity observed

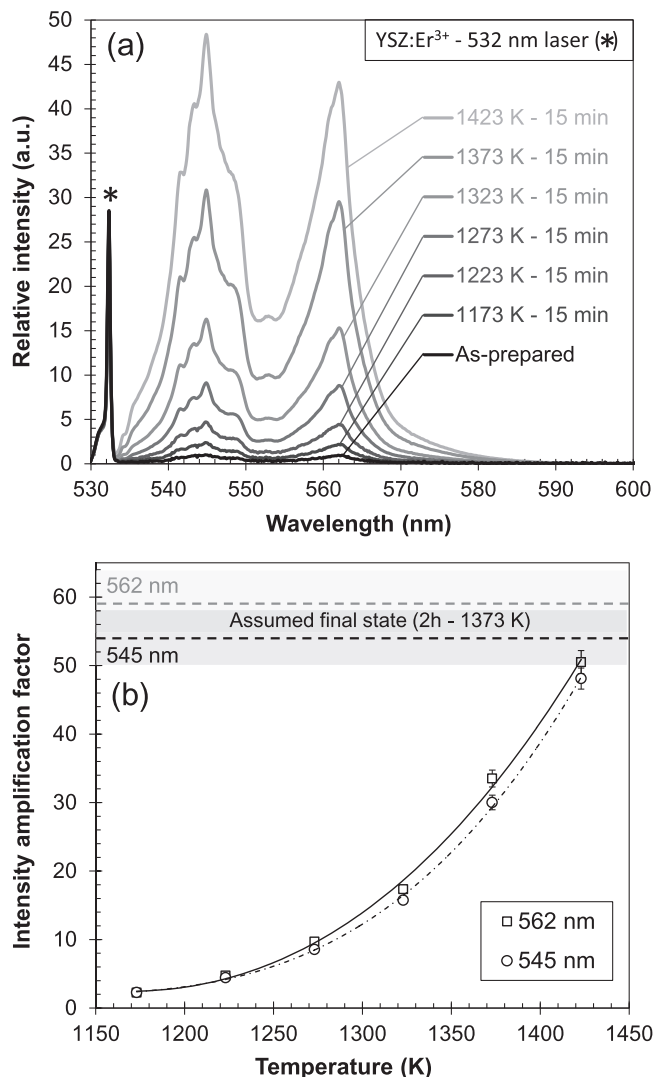


Figure 4. (a) Fluorescence emission spectra of YSZ:Er³⁺ calcined for 15 min at different temperatures and (b) evolution of relative emission peak intensity of the 545 and 562 nm transition lines with temperature. Intensities are normalized with the maximum intensity of the as-prepared powder. Plotted lines in (b) correspond to the fit of the experimental data points with a 3rd order polynomial.

with the as-prepared sample, increases steadily from a factor of ~2 to ~43–48 between 1173 and 1423 K, confirming the significant temperature sensitivity of this fluorescence parameter across this whole range. The sensitivity is in addition observed to increase with temperature and is significantly greater in the 1373–1423 K range. Fits of the experimental data points with an empirical 3rd degree polynomial, plotted on figure 4(b) as solid lines, give a fair approximation of the observed monotonic increase of the intensity with temperature.

4.2.2. Fluorescence lifetime. Figure 5 shows the evolution of the fluorescence lifetime of the 562 nm emission line of Er³⁺, measured at room temperature, with temperature of exposure.

It exhibits an almost linear increase from 28 to 44 μ s between 1173 and 1373 K. The fluorescence lifetime of all calcined samples is also significantly higher than for the as-

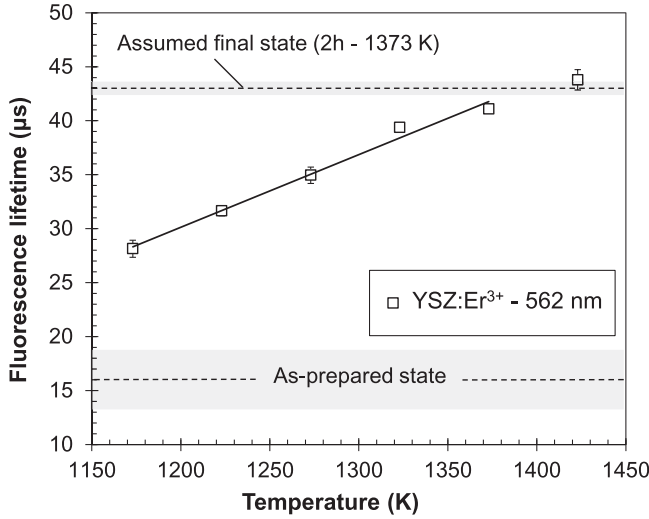


Figure 5. Temperature dependency of the YSZ:Er³⁺ fluorescence lifetime from the emission line at 562 nm for 15 minute-long heat treatments.

prepared sample ($\sim 16 \mu\text{s}$). However, the decay time obtained with the sample treated to 1423 K is identical to that of the assumed fully crystallized sample ($\sim 43 \mu\text{s}$). These results suggest that the sensitivity to temperature of the YSZ:Er³⁺ powder, for 15 min long heat treatments, ranges from a maximum temperature between 1423 and 1373 K to at least 1173 K, and might extend to temperature lower than 1173 K. The fluorescence decay time therefore appears to be a suitable complementary indicator of thermal history, which shows the advantage over absolute intensity to be relatively insensitive to experimental measurement conditions [48, 49].

4.3. Intensity ratio method

As a credible alternative to thermochromic paints, it is desirable that fluorescent thermal history sensors can allow rapid evaluation of temperature fields over large areas of components under investigation. Although decay time is intrinsically independent from most of experimental conditions, lifetime mapping with fast decaying phosphors such as YSZ:Er³⁺ ($\tau < 50 \mu\text{s}$) is in practice significantly trickier than intensity mapping. Intensity maps can indeed be easily recorded using experimental setups similar to those commonly used with temperature sensitive paints, involving continuous light sources and standard CCD cameras. Furthermore, intensity based methods would allow to take advantage of the particularly strong temperature sensitivity of the main emission lines of YSZ:Er³⁺ between 1173 and 1423 K (figure 4(b)).

In order to overcome the dependency of absolute intensities over experimental factors, mostly related to the excitation source, the sample and the detection system, a typical intensity ratio approach [48, 49] is proposed. A temperature sensitive parameter I_r is defined as the ratio of the temperature sensitive emission line of the YSZ:Er³⁺ marker at 562 nm ($I_{\text{Er}/562}(T)$) and the temperature insensitive emission line at 606 nm of an annealed YSZ:Eu³⁺ reference phosphor powder in a fully crystallized state ($I_{\text{Eu}/606}$), according to the

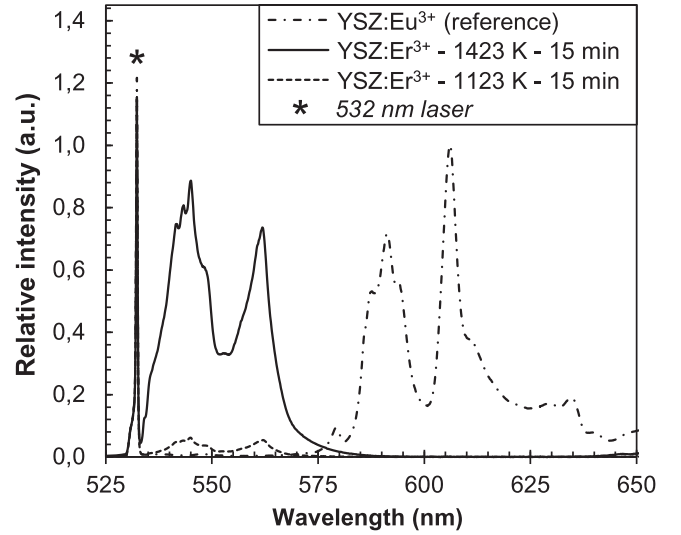


Figure 6. Comparison of the visible emission spectra of YSZ:Er³⁺ (temperature sensitive phosphor) and YSZ:Eu³⁺ (used as a reference) treated at various temperatures.

expression given in equation (2):

$$I_r(T) = \frac{I_{\text{Er}/562}(T)}{I_{\text{Eu}/606}}. \quad (2)$$

This latter phosphor was chosen amongst the phosphors available at the time in order to illustrate the concept of a composite fluorescent thermal history sensor. Easily synthesized through the sol-gel route described in the experimental section and annealed for 2 h at 1373 K to ensure complete crystallization, it exhibits strong emissions lines at 589 and 606 nm well distinct from those of Er³⁺ ions (figure 6). It also combines the advantages of being chemically compatible with YSZ:Er³⁺ and having an excitation band covering the emission wavelength of the 532 nm laser source [50], that can therefore be used to excite both phosphors simultaneously. High temperature stability tests carried out on sintered disks of YSZ:Eu³⁺ aerogel powders also showed that, once full crystallization is achieved following 2 h of exposure at 1373 K, no significant variation in the absolute intensity of its two main emissions is observed even after 500 h at 1423 K [22]. It is therefore expected that no further changes will occur within the powder that would affect the emitted intensity after a 15 minutes heat treatment even at 1423 K, except for some very limited coarsening of the average crystallite size with no significant impact on intensity for such a short duration. The I_r parameter should thus provide a robust indicator of the thermal history in the range 1173–1423 K, i.e. it should be insensitive to small variations in most of the measurement conditions such as distance of observation, viewing angle and inhomogeneities of the illumination zone.

In order to validate the relevance of the I_r parameter, a preliminary study was carried out on the YSZ:Er³⁺ powders previously investigated and a unique YSZ:Eu³⁺ powder sample (calcined for 2 h at 1373 K to ensure full crystallization), whose role was to reproduce the ideal behavior of the reference. Due to the limited quantity of reference powder

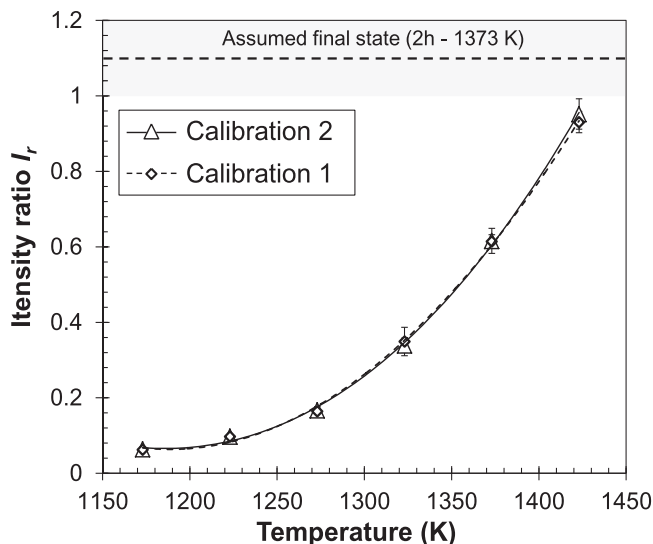


Figure 7. Evolution with temperature of the intensity ratio I_r , calculated using equation (2) for two different experimental setups having slightly different observation distances and volumes of powder probed.

available at the time, reference samples could indeed not be subjected to the same 15 min heat treatment as YSZ:Er^{3+} samples. The ratio was thus built, for each data point, from the signal recorded during measurements immediately consecutive on the corresponding YSZ:Er^{3+} sample and a unique reference respectively. These conditions remain nevertheless consistent with the expected temperature insensitive behavior of the ideal reference. For the fluorescence measurements, a 10 nm wide integrating window centered on the maximum of each peak was used in order to obtain a detection window similar to that considered when using a CCD camera equipped with a typical narrow band-pass filter (~ 10 nm FWHM).

Figure 7 shows the evolution of I_r with temperature for two different measurement configurations slightly differing by the volume of powder probed (laser beam diameter of (1) 1 mm and (2) 1.2 mm) and by the distance of observation (distance between the optical fiber and the sample of (1) 20 cm and (2) 25 cm).

Both calibration curves exhibit a steady increase with temperature well approximated by a third degree polynomial as observed with absolute intensities (see figure 4(b)). More importantly, no significant differences are observed between the two curves in the whole range 1173–1423 K. Although the approach only remains an approximation of the concept of a mixed fluorescent $\text{YSZ:Er}^{3+}/\text{YSZ:Eu}^{3+}$ marker, these results are promising in terms of the robustness of the I_r parameter for small variations in the observation distance and the number of luminescent centers excited.

4.4. Thermal stability of the temperature sensing capabilities

Previous results have demonstrated the strong sensitivity to temperature of exposure of YSZ:Er^{3+} and composite $\text{YSZ:Er}^{3+}/\text{YSZ:Eu}^{3+}$ phosphors in the range 1173–1423 K. The practical application of these markers however requires that they can be successfully integrated into coatings deposited at

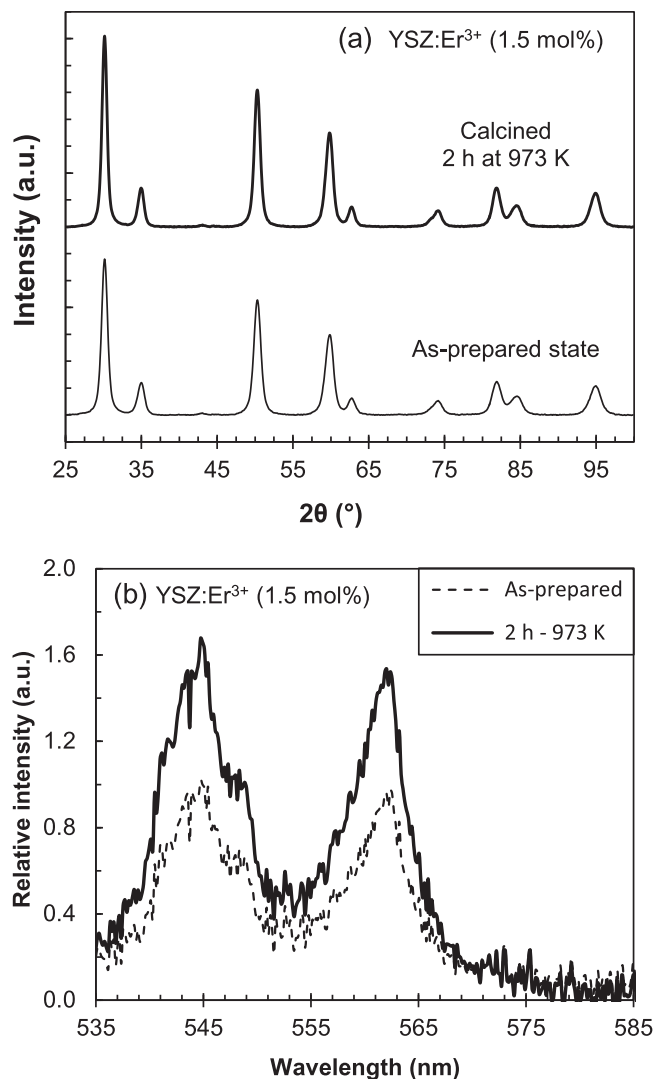


Figure 8. Comparison of (a) x-ray diffraction patterns and (b) fluorescence spectra of YSZ:Er^{3+} powder in the as-prepared state and calcined 2 h at 973 K with heating and cooling speeds of 100 K h^{-1} .

the surface of tested components. These coatings must sustain potentially harsh conditions during the short testing times, including high temperature, thermal shocks and mechanical stress for moving parts in engines. Numerous organic and inorganic binders exist for the integration of luminescent markers for high temperature applications [51], however thermal history sensors such as YSZ:Er^{3+} require that processing temperature and duration of exposure of the coating do not prematurely activate the temperature-driven mechanisms at the origin of its temperature sensing capability. The thermal stability at moderate temperature of the temperature sensitivity of the sensor was thus evaluated. Figure 8 illustrates the x-ray diffraction patterns and the emission spectra of YSZ:Er^{3+} powders in the as-prepared state and after an additional heat treatment for 2 h at 973 K with heating and cooling speeds of 100 K h^{-1} .

The latter treatment does not induce significant enhancement of the crystallinity (figure 8(a)) or emission intensity increase ($\sim 60\%$) as compared to the changes

observed previously in figures 3 and 4(b) after 15 minutes between 1173 and 1423 K (~100%–4500% intensity increase). It is therefore concluded that the temperature sensing capabilities remain unaffected by heat treatments up to 973 K for 2 h.

5. Discussion

Figures 4 and 5 show that both emission intensity and decay time of the YSZ:Er³⁺ phosphor synthesized by a sol-gel route exhibit a strong dependency on the temperature of exposure in the range 1173–1373 K/1423 K for 15 minute-long heat treatments, that would be suitable for recording the temperature of past thermal events achieved under controlled conditions. As discussed in section 2, these permanent evolutions are assumed to be related to temperature-driven changes within the host matrix induced by the enhancement of the crystallinity, whose extent is a function of the temperature of exposure in that range (figure 3). These changes most likely include the uniformization of the crystal field, the removal of quenching sites such as crystal defects and residual organic groups to which Er³⁺ ions are particularly sensitive [38], as well as the growth of YSZ crystallites [12, 33, 34, 52].

The fact that the fluorescence lifetime of the emission at 562 nm already has converged to its final state value after 15 minutes at 1373 K (figure 5) while intensity and crystallinity continue to improve up to 1423 K (figures 3 and 4(a)) indicates that the conditions to reach the converged decay time are met before the assumed complete crystallization of the material has occurred. This is probably due to final stages likely being dominated by YSZ crystallites growth which affect absorption of excitation energy and not radiative transition probabilities, thus having little impact on decay time. The relative independence of the decay time on the number of active Er³⁺ ions (i.e. on the energy absorbed by crystallites and on the total volume of material where the conditions for the Er³⁺ ions to emit are reunited) is therefore assumed to be the reason behind the difference in behavior between the evolution of the decay time (moderate linear increase by a factor of 2.7 up to 1373 K, figure 4(b)) and the evolution of the emission intensity (strong increase, following a 3rd degree polynomial, by a factor of 50 up to at least 1423 K, figure 5).

The fluorescence characteristics of the two main emission peaks of YSZ:Er³⁺ offer wide measuring and dynamic ranges between 1173 and 1423 K and 1173–1373 K in which intensity and fluorescence lifetime increase monotonously (figures 4 and 5). The empirical models determined with the fits of the experimental data points (a 3rd order polynomial for emission intensity, figure 4(b), and a linear model for decay time, figure 5) were used to calculate noise-equivalent temperatures (NET) as a first estimation of the theoretical resolution of the marker, using the expression given in equation (3):

$$\text{NET} = \frac{\sigma_i}{S_{T_i}}. \quad (3)$$

Parameters S_{T_i} correspond to the sensitivity of the marker in a small interval around the temperature T_i , straightforwardly derived from the empirical expressions obtained with the fit of the experimental data points. Parameters σ_i are the expanded uncertainties of the emission intensity or decay time measurement at T_i . For each sample, it corresponds to the standard deviation obtained from 30 repetitions of single measurements for a unique position of the sample weighted by a coverage factor of 2. Under the present experimental conditions, the NET values obtained are comprised between 0.7 and 0.3 K in the range 1223–1423 K for intensity measurements, and between 6 and 2 K in the range 1173–1423 K for decay time measurements, suggesting that the theoretical resolution of the marker would be much greater than the resolution of most thermochromic pigments in that temperature range, typically limited to one to four color transitions [6–8]. It should be noted though that, at this stage, these NET values remain a first approximation of the resolution based on a limited number of experimental data points. Nevertheless, these results confirm the relevance of YSZ:Er³⁺ fluorescence emission and decay time as thermal history indicators suitable for high temperature sensing. The high sensitivity of emission intensity at temperatures higher than 1273 K (increase of intensity per degree increment corresponding to 32% to that of the as-prepared material around 1373 K, figure 4(b)) is especially interesting for precise temperature sensing on critical components applied to the design of high temperature systems operating in that range, such as gas turbines.

It can be seen from figure 5 that the sensitivity range to temperature of the lifetime decay for 15 minutes long heat treatment ranges from a maximum temperature between 1423 and 1373 K to at least 1173 K, and might extend to temperature lower than 1173 K (figure 5). The fluorescence decay time, which shows the advantage over absolute intensity to be relatively insensitive to experimental measurement conditions [48, 49] therefore appears to be a suitable complementary indicator of thermal history for OD measurements, but could also improve or extend the overall thermal history sensitivity range of YSZ:Er³⁺ if combined with intensity measurements. More generally, it is expected that the whole temperature sensitivity range could be shifted to higher or lower temperature to adjust to the targeted desired application by increasing or reducing the exposure time respectively, in relation with the time dependency of the extent of crystallinity enhancement under isothermal conditions [12, 53, 54]. Reduction of the exposure time to five minutes or less, in order to keep up with actual testing procedures using thermochromic paints, is assumed to be possible and would shift the temperature sensitive range of the marker to higher temperatures, with, as a consequence, readouts more sensitive to time related errors. Investigation of the effect of the exposure time showed indeed that, as for crystallinity changes, most of the intensity increase is achieved within the first five minutes of exposure at high temperature (~75% of the final intensity is reached after 5 minutes for a 15 minutes heat treatment at 1423 K).

The robustness of the ratio I_r to small variations in the observation conditions (figure 7), suggest that this intensity ratio method, using a compatible temperature insensitive

reference phosphor such as fully crystallized YSZ:Eu³⁺, is a promising approach to take advantage of the strong temperature sensitivity of the emission intensity for off-line temperature mapping applications. Foreseen main limiting factors regarding the robustness of the I_r ratio that require further investigation include in particular the possible differences in the optical properties of the host YSZ matrix at the different wavelength investigated (562 nm for Er³⁺ and 606 nm for Eu³⁺), and the effective thermal stability of the YSZ:Eu³⁺ reference when blended with the YSZ:Er³⁺ marker and calcined for 15 min in the range 1173–1423 K. Other more convenient references could also be explored, such as Al₂O₃:Cr³⁺. Cr³⁺ ions indeed produce very strong fluorescence and have a broader excitation peak than lanthanide ions. Therefore, only a small amount of reference powder would need to be mixed in, and an excitation wavelength at 514 nm could be used, which would excite stronger fluorescence from Er³⁺ ions than at 532 nm [22].

Finally, the investigation of the thermal stability of the YSZ:Er³⁺ as-prepared powder (figure 8) has shown that the material retains almost all of its temperature recording capabilities after a 2 h long heat treatment at 973 K. Compatible binders with processing temperature up to 973 K for 2 h therefore appear to be appropriate candidates for the integration of the investigated YSZ:Er³⁺ and YSZ:Eu³⁺ phosphors in temperature sensitive coatings. Both the sensing YSZ:Er³⁺ powder and the annealed YSZ:Eu³⁺ powder (or any other relevant reference) would be mixed into a transparent high temperature binder and applied on the parts with an air brush, as for typical fluorescent coatings used for phosphor thermometry [51]. The satisfactory thermo-mechanical properties of YSZ coatings, widely used in thermal barrier coating applications up to 1473 K [55, 56], also offer the possibility to directly deposit the phosphors as coatings or using a YSZ host matrix, similarly to what is done for fluorescent thermal barrier sensor coatings [21, 50, 57]. The chemical compatibility of the marker with the reference and the matrix could indeed facilitate the sintering of the coating as a morphologically uniform single layer (see sol-gel TBC sensors in reference [21]) at relatively low temperature (<973 K). Therefore, low temperature deposition processes of YSZ coatings, such as the dip-coating process developed by the CIRIMAT in collaboration with the ICA [20, 21] and other soft chemistry processes followed by sintering at a temperature lower than 973 K are expected to be relevant and promising solutions for the deposition of YSZ:Er³⁺/YSZ:Eu³⁺ thermal history sensor coatings. A sol-gel method involving air-spraying at room temperature of a sol loaded with particles followed by a heat treatment at a temperature below 700 °C is currently under investigation. Such coatings could be especially interesting for being used on component already coated with YSZ-based thermal barrier coatings in gas turbines, both for a potentially enhanced adherence and for their temperature sensing range (1223–1423 K) covering typical operating range of such parts [55, 56]. Feasibility of sol-gel deposition of YSZ:Er³⁺/YSZ:Eu³⁺ coatings and associated mechanical adherence will be investigated in future a work.

6. Conclusion

The knowledge of temperature to which components are subjected in service is essential to anticipate and prevent failure issues during the design phase of many high temperature systems [1–4]. This paper discussed the concept of fluorescent thermal history sensor as an alternative to thermochromic paints, and confirmed the potential as such of Er³⁺-doped YSZ powders produced by a sol-gel route, whose fluorescence properties appear to be a function of the crystallinity of the YSZ host matrix. The continuous, temperature-driven enhancement of the crystallinity of YSZ:Er³⁺ powder samples observed after 15 minutes long heat treatments between 1173 K and 1423 K induces indeed substantial and consistent increases of their green emission intensity and fluorescence decay time in the ranges 1223 K–1423 K and 1173 K–1373 K respectively. These strong temperature dependencies are assumed to be suitable for off-line temperature sensing with an estimated theoretical resolution for those temperatures ranging between 0.3 and 1 K and between 6 and 2 K for intensity and decay time respectively, surpassing the resolution of actual thermochromic paints in that temperature range. It is also assumed possible to adjust the sensitivity ranges through exposure duration changes. Preliminary investigations confirmed the potential of an intensity ratio approach, using a red emitting, temperature insensitive YSZ:Eu³⁺ reference, as a robust method regarding small variations in experimental measurement conditions for future applications in off-line temperature mapping. Finally, the thermal stability of the YSZ:Er³⁺ marker up to 973 K for a short period of time (2 h) is promising for its integration within coatings for practical thermal history sensing applications, such as sol-gel deposited YSZ-based coatings [20, 21] that will be investigated in a future work.

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References

- [1] Viswanathan R 1989 *Damage Mechanisms and Life Assessment of High Temperature Components* (Materials Park, OH: ASM International)
- [2] Kurz R and Brun K 2000 Degradation in gas turbine systems *J. Eng. Gas Turbines Power* **123** 70–7
- [3] Hihara L H, Adler R P I and Latanision R M 2013 *Environmental Degradation of Advanced and Traditional Engineering Materials* (Boca Raton, FL: CRC Press)
- [4] Bar-Cohen Y 2014 *High Temperature Materials and Mechanisms* (Boca Raton, FL: CRC Press)
- [5] Chamberlain J R 1991 Temperature indicating paint and method of preparing a specimen with the same *US Patent* 5008136A

- [6] Hodgkinson E and Watson H 2004 Temperature indicating paint *US Patent* 6673271 B2
- [7] Watson H M L 2004 Temperature indicating paint *US Patent* 6682665 B2
- [8] Hughes M 2011 Temperature indicating paint *US Patent* 8012375 B2
- [9] Salek G, Demourgues A, Jubera V, Garcia A and Gaudon M 2015 Mn²⁺ doped Zn₃(PO₄)₂ phosphors: irreversible thermochromic materials useful as thermal sensors *Opt. Mater.* **47** 323–7
- [10] Choy K-L, Heyes A L and Feist J P 2009 Thermal barrier coating with thermoluminescent indicator material embedded therein *US Patent* 7510776 B2
- [11] Feist J P, Nicholls J R and Heyes A L 2011 Determining thermal history of components *US Patent* 20110069735 A1
- [12] Rabhiou A, Feist J, Kempf A, Skinner S and Heyes A 2011 Phosphorescent thermal history sensors *Sensors Actuators* **169** 18–26
- [13] Heyes A L, Rabhiou A, Feist J P and Kempf A 2013 Thermal history sensing with thermographic phosphors *AIP Conf. Proc., Temperature: Its Measurement And Control In Science And Industry, Volume 8: Proc. 9th Int. Temperature Symp.* (New York: AIP) pp 891–6
- [14] Stenders D, Karadagli I, Pflichtsch C and Atakan B 2014 Sol-gel deposited thermographic phosphors as possible thermal history coatings *IET ISA 60th Int. Instrumentation Symp.* 2014 pp 1–6
- [15] Feist J P, Karmakar Biswas S, Pilgrim C, Sollazzo P Y and Berthier S 2015 Off-line temperature profiling utilizing phosphorescent thermal history paints and coatings *J. Turbomach.* **137** 101003
- [16] Yáñez-González Á, Ruiz-Trejo E, van Wachem B, Skinner S, Beyrau F and Heyes A 2015 A detailed characterization of BaMgAl₁₀O₁₇:Eu phosphor as a thermal history sensor for harsh environments *Sensors Actuators* **234** 339–45
- [17] Cornu L, Duttine M, Gaudon M and Jubera V 2014 Luminescence switch of Mn-Doped ZnAl₂O₄ powder with temperature *J. Mater. Chem. C* **2** 9512–22
- [18] Rabhiou A, Kempf A and Heyes A 2013 Oxidation of divalent rare earth phosphors for thermal history sensing *Sensors Actuators B* **177** 124–30
- [19] González Á Y, Skinner S, Beyrau F and Heyes A L 2015 Reusable thermal history sensing via oxidation of a divalent rare earth ion-based phosphor synthesized by the sol-gel process *Heat Transfer Eng.* **36** 1275–81
- [20] Pin L, Pilgrim C, Feist J, Le Maout Y, Ansart F and Lours P 2013 Characterisation of thermal barrier sensor coatings synthesised by sol-gel route *Sensors Actuators* **199** 289–96
- [21] Copin E, Sentenac T, Le Maout Y, Bias F, Ansart F, Vidal V and Lours P 2014 Feasibility of luminescent multilayer sol-gel thermal barrier coating manufacturing for future applications in through-thickness temperature gradient sensing *Surf. Coat. Technol.* **260** 90–6
- [22] Copin E 2015 Fonctionnalisation de barrières thermiques aéronautiques YSZ issues de la voie sol-gel : mesure de température et diagnostic de l'endommagement par fluorescence *PhD Thesis* École Nationale Supérieure des Mines d'Albi-Carmaux
- [23] Gentili D, Durso M, Bettini C, Manet I, Gazzano M, Capelli R, Muccini M, Melucci M and Cavallini M 2013 A time-temperature integrator based on fluorescent and polymorphic compounds *Sci. Rep.* **3** 2581
- [24] Lan Y, Wang H, Skula N, Chen X, Lu Y, Chen G and Ren Z 2015 Determination of thermal history by photoluminescence of core-shelled quantum dots going through heating events *Part. Part. Syst. Charact.* **32** 65–71
- [25] Cattaruzza E, Back M, Battaglin G and Trave E 2014 Unexpected behavior of the 1.54 μ m luminescence in Er-doped silica films *J. Non-Cryst. Solids* **401** 186–90
- [26] Deng T, Yan S and Hu J 2015 Effect of calcination temperature on Up-conversion photoluminescence of the GdAlO₃:Er³⁺, Yb³⁺ phosphor *ECS J. Solid State Sci. Technol.* **4** R48–53
- [27] Ramanathan S, Soni N C and Prasad R 1993 Crystallization behaviour of zirconia 10 mol% yttria by hot-stage x-ray diffraction *J. Mater. Sci. Lett.* **12** 122–4
- [28] Málek J, Matsuda S, Watanabe A, Ikegami T and Mitsuhashi T 1995 Crystallization kinetics of zirconia-yttria gels *Thermochim. Acta* **267** 181–94
- [29] Rashad M M and Baioumy H M 2008 Effect of thermal treatment on the crystal structure and morphology of zirconia nanopowders produced by three different routes *J. Mater. Process. Technol.* **195** 178–85
- [30] Málek J 2000 Kinetic analysis of crystallization processes in amorphous materials *Thermochim. Acta* **355** 239–53
- [31] Quan Z W, Wang L S and Lin J 2005 Synthesis and characterization of spherical ZrO₂:Eu³⁺ phosphors by spray pyrolysis process *Mater. Res. Bull.* **40** 810–20
- [32] Zhou Y H, Lin J, Wang S B and Zhang H J 2002 Preparation of Y₃Al₅O₁₂:Eu phosphors by citric-gel method and their luminescent properties *Opt. Mater.* **20** 13–20
- [33] Zhang J, Tang Z, Zhang Z, Fu W, Wang J and Lin Y 2002 Synthesis of nanometer Y₂O₃:Eu phosphor and its luminescence property *Mater. Sci. Eng. A* **334** 246–9
- [34] Erdei S, Roy R, Harshe G, Juwhari H, Agrawal D, Ainger F W and White W B 1995 The effect of powder preparation processes on the luminescent properties of yttrium oxide based phosphor materials *Mater. Res. Bull.* **30** 745–53
- [35] Zhou Y H and Lin J 2005 Morphology control and luminescence properties of YVO₄:Eu phosphors prepared by spray pyrolysis *Opt. Mater.* **27** 1426–32
- [36] Bandi V R, Nien Y-T, Lu T-H and Chen I-G 2009 Effect of calcination temperature and concentration on luminescence properties of novel Ca₃Y₂Si₃O₁₂:Eu phosphors *J. Am. Ceram. Soc.* **92** 2953–6
- [37] Martínez-Hernández A, Guzmán-Mendoza J, Rivera-Montalvo T, Sánchez-Guzmán D, Guzmán-Olguín J C, García-Hipólito M and Falcony C 2014 Synthesis and cathodoluminescence characterization of ZrO₂:Er³⁺ films *J. Lumin.* **153** 140–3
- [38] Ting C-C, Chien Y-C and Sung W-F 2013 Erbium doping effects on the structural and infrared luminescence properties of Gd₂Ti₂O₇ nanocrystals *ECS J. Solid State Sci. Technol.* **2** R105–10
- [39] Satyanarayana T, Brik M G, Venkatramiah N, Kityk I V, Plucinski K J, Ravikumar V and Veeraiah N 2010 Influence of crystallization on the luminescence characteristics of Pr³⁺ ions in PbO–Sb₂O₃–B₂O₃ glass system *J. Am. Ceram. Soc.* **93** 2004–11
- [40] Liu C, Yu R, Xu Z, Cai J, Yan X and Luo X 2007 Crystallization, morphology and luminescent properties of YAG:Ce³⁺ phosphor powder prepared by polyacrylamide gel method *Trans. Nonferrous Met. Soc. China* **17** 1093–9
- [41] Venkatramu V *et al* 2012 Synthesis, structure and luminescence of Er³⁺-doped Y₃Ga₅O₁₂ nano-garnets *J. Mater. Chem.* **22** 13788–99
- [42] Fenech J, Viazzi C, Bonino J-P, Ansart F and Barnabé A 2009 Morphology and structure of YSZ powders: comparison between xerogel and aerogel *Ceram. Int.* **35** 3427–33
- [43] Lecomte A, Blanchard F, Dager A, Silva M C and Guinebrière R 1998 Synthesis and sintering of zirconium oxide aerogel *J. Non-Cryst. Solids* **225** 120–4
- [44] Shane M and McCartney M L 1990 Sol-gel synthesis of zirconia barrier coatings *J. Mater. Sci.* **25** 1537–44
- [45] Levenberg K 1944 A method for the solution of certain non-linear problems in least squares *Q. Appl. Math.* **2** 164–8

- [46] Marquardt D 1963 An algorithm for least-squares estimation of nonlinear parameters *J. Soc. Ind. Appl. Math.* **11** 431–41
- [47] Brübach J, Janicka J and Dreizler A 2009 An algorithm for the characterisation of multi-exponential decay curves *Opt. Lasers Eng.* **47** 75–9
- [48] Allison S W and Gillies G T 1997 Remote thermometry with thermographic phosphors: instrumentation and applications *Rev. Sci. Instrum.* **68** 2615–50
- [49] Khalid A H and Kontis K 2008 Thermographic phosphors for high temperature measurements: principles, current state of the art and recent applications *Sensors* **8** 5673–744
- [50] Gentleman M M and Clarke D R 2004 Concepts for luminescence sensing of thermal barrier coatings *Surf. Coat. Technol.* **188** 93–100
- [51] Bruebach J, Pflitsch C, Dreizler A and Atakan B 2013 On surface temperature measurements with thermographic phosphors: a review *Prog. Energy Combust. Sci.* **39** 37–60
- [52] Silver J, Martinez-Rubio M I, Ireland T G, Fern G R and Withnall R 2001 The effect of particle morphology and crystallite size on the upconversion luminescence properties of erbium and ytterbium Co-doped yttrium oxide phosphors *J. Phys. Chem. B* **105** 948–53
- [53] Avrami M 1939 Kinetics of phase change: I. General theory *J. Chem. Phys.* **7** 1103–12
- [54] Johnson W A and Mehl R F 1939 Reaction kinetics in processes of nucleation and growth *Trans. Am. Inst. Min. Metall. Eng.* **135** 416–42
- [55] Reed R C 2006 *The Superalloys: Fundamentals and Applications* (Cambridge: Cambridge University Press)
- [56] Bose S 2007 Thermal barrier coatings (TBCs) *High Temperature Coatings* (Burlington: Butterworth-Heinemann) ch 7 pp 155–232
- [57] Nicholls J R, Wellman R G, Steenbakker R and Feist J 2010 Self diagnostic EB-PVD thermal barrier coatings *Adv. Sci. Technol.* **72** 65–74