## **Supplementary information**

## Crystal structure and lithium diffusion pathways of a potential positive electrode material for Lithium-ion batteries: Li<sub>2</sub>V<sup>III</sup>(H<sub>0.5</sub>PO<sub>4</sub>)<sub>2</sub>

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Data collection							
Diffractometer	Bruker D8 Venture						
Radiation type	Μο <i>Κ</i> α (λ=0.71073 Å)						
Temperature (K)	293						
Crystal size (mm)	$0.14 \times 0.09 \times 0.07$						
Absorption correction	Numerical (SADABS; Bruker, 2013)						
$\mu (mm^{-1})$	2.35						
Data collection	APEX3 (BRUKER)						
No. of measured,	65122,						
independent and	3250,						
observed $[I > 2\sigma(I)]$ reflections	2702						
$(\sin \theta/\lambda)_{max} (\text{\AA}^{-1})$	1.116						
Refinement							
Cell refinement software	SAINT (BRUKER)						
R <sub>int</sub>	0.059						
Data reduction software	SAINT (BRUKER)						
Structure solving software	SHELXS97						
Structure solving method	Direct method						
Structure refinement software	SHELXL (Sheldrick, 2014)						
$R[F^2 > 2\sigma(F^2)]$	0.029						
$wR(F^2)$	0.067						
S	1.10						
No. of reflections	3250						
No. of parameters	62						
H-atom treatment	No H-atom parameter refined						
$\Delta \rho_{max}$ , $\Delta \rho_{min}$ (e Å <sup>-3</sup> )	1.04, -0.73						
Molecular graphic	VESTA						
Formatting CIF file	CHECKCIF						

Table S2: Anisotropic displacement parameters (Å<sup>2</sup>)

	U11	U22	U33	U12	U13	U23
V(1)	0.00441(5)	0.00463(5)	0.00461(5)	0.00026(4)	0.00148(4)	-0.00039(4)
P(1)	0.00400(6)	0.00539(6)	0.00505(6)	-0.00069(4)	0.00164(4)	-0.00055(4)
O(1)	0.0043(2)	0.0072(2)	0.0099(2)	-0.0016(1)	0.0006(1)	0.0001(1)
O(2)	0.0095(2)	0.0073(2)	0.0095(2)	-0.0021(1)	0.0023(2)	-0.0035(1)
O(3)	0.0082(2)	0.0102(2)	0.0091(2)	-0.0018(2)	0.0050(2)	0.0011(2)
O(4)	0.0085(2)	0.0150(2)	0.0067(1)	0.0033(1)	0.0024(1)	-0.0020(2)
Li(1)	0.0138(7)	0.0139(7)	0.0145(7)	-0.0032(6)	0.0030(6)	0.0019(6)

**Table S3:** Bond lengths (below the diagonal) and angles (above the diagonal) in coordination polyhedra of V<sup>3+</sup>, Li<sup>+</sup> and P<sup>5+</sup> in Li<sub>2</sub>V<sup>III</sup>(H<sub>0.5</sub>PO<sub>4</sub>)<sub>2</sub>. The polyhedral distortion is calculated as  $\Delta = \frac{1}{N} \sum_{i=0}^{N} \frac{(di - \langle d \rangle)^2}{\langle d \rangle^2}$  with N the number of bond distances considered, d<sub>i</sub> the distance between the atom i and the central atom of the polyhedron and  $\langle d \rangle$  the average of the distances considered.

VO <sub>6</sub>									
	C	D(1)	O(1')	O(2)	O(2')	O(3)	O(3')		
O(1)	2.0	18(2)	180	88.9(2)	91.1(2)	93.4(2)	86.6(2)		
O(1')	4.0	37(2) 2	2.018(2)	91.1(2)	88.9(2)	86.6(2)	93.4(2)		
O(2)	2.8	35(2) 2	2.891(2)	2.031(2)	180	84.0(2)	96.0(2)		
O(2')	2.8	91(2) 2	2.835(2)	4.062(2)	2.031(2)	96.0(2)	84.0(2)		
O(3)	2.9	05(3) 2	2.738(3)	2.679(3)	2.977(2)	1.974(2)	180		
O(3')	2.7	38(3) 2	2.905(3)	2.977(2)	2.679(3)	3.948(2)	1.974(2)		
$\Delta = 1.11 \times 10^{-4}$									
HPO <sub>4</sub>									
	O(1)	0(2	2)	O(3)	0(	4)	H(1)	H(1')	
O(1)	1.533(2)	111.5	5(2)	109.8(2)	111.	6(2)	93.1(2)	84.7(2)	
O(2)	2.503(3)	1.523	3(2)	110.1(2)	108.	0(3) 1	.09.7(3)	121.1(3	
O(3)	2.505(3)	2.503	3(3)	1.530(2)	105.	7(2) 1	24.1(3)	116.6(2)	
O(4)	2.547(2)	2.483	3(2)	2.451(2)	1.54	6(2) 1	28.9(4)	124.4(3)	
H(1)	2.856(2)	3.133	3(2)	3.423(2)	1.01	<b>8(2)</b> 2	2.329(1)	14.8(1)	
H(1')	2.971(2)	3.720	D(2)	3.639(2)	1.46	6(2)	0.58	2.692(1)	
$\Delta = 2.96 \times 10^{-5} (PO_{a})$									
LiO,									
	O(1	.)	0(2	.)	O(3)		O(4)		
O(1)	1.952	2(6)	113.3	(4)	119.1(4)		125.1(5)		
O(2)	3.344	(2)	2.052	(4)	83.3(3)		91.0(3)		
O(3)	3.390	)(3)	2.679	(3)	1.978(6)	:	111.9(4)		
O(4)	3.478	8(2)	2.869	(3)	3.272(2)	:	1.968(5)		
$\Delta = 3.69 \times 10^{-4}$									



Figure S1: Rietveld refinement of structure from X-ray powder diffraction. The theoretical Bragg contributions for  $Li_2V^{III}(H_{0.5}PO_4)_2$  are depicted as red markers and those of  $LiVPO_4OH$  as blue ones.



Figure S2: The residual electronic density located between two "free" oxygens (a) and the two models proposed for hydrogen position are illustrated in (b) and (c)

The distribution of residual electronic density between the O(4) and O(4') is peanutshaped (Fig S2a). Two models can therefore be proposed:

i) H atoms are located on a fully occupied 2*d* (½, ½, 0) Wyckoff site (on the inversion center of the peanut-shape residual electronic density) leading to two symmetrically equivalent O-H distances ( $d_{O(4)-H(1)} = d_{O(4')-H(1)} = 1.227(2)$  Å and a O(4)-H-O(4') angle of 180°) (Fig S2b),

ii) H atoms are located on a 4*e* position half occupied site generating two different O-H distances ( $d_{O(4)-H(1)} = 1.018(2)$  Å and  $d_{O(4)-H(1')} = 1.466(2)$  Å and an O(4)-H-O(4') angle of 153.4(3)°) (Fig S2c). If the hydrogen atom occupies a 2*d* Wyckoff position a three-centers fourelectrons bond has to be considered and in the case of 4*e* position a more common hydroxyl group involved in a hydrogen bond would be formed.

As it's very difficult to precisely localize H atoms in inorganic materials we weren't able to refine the fractional coordinates of this hydrogen, subsequently at the center of each lobes of the peanut-shaped residual electronic density (i.e. 0.45, 0.51, 0.01). This position is very close to the one found by Jin-Xiao Mi *et al.*<sup>1</sup> for the iron phase (i.e. 0.45, 0.52, 0.02). Based on this result, and even if the non-peculiar position of hydrogen can't be precisely determined, the split model was chosen.



**Figure S3:** In (a) Projection along [100] of Lithium paths obtained from Bond Valence Energy Landscape (BVEL) with an energy of 1.6 eV above minimum for  $Li_2Fe(SO_4)_2$  (right) and  $Li_2V^{III}(H_{0.5}PO_4)_2$  (left) and in (b) enlargement of the path corresponding to the shorter Li-Li distance (along [133]) showing that a hopping between two Li sites is possible in  $Li_2Fe(SO_4)_2$ whereas it isn't allowed in  $Li_2V^{III}(H_{0.5}PO_4)_2$ .

## References:

 Mi, J.-X. et al. Synthesis, Magnetism, and Crystal Structure of Li2Fe[(PO4)(HPO4)] and its Hydrogen Position Refinement. Zeitschrift für Anorg. und Allg. Chemie 2004, 630, 1632–1636.