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## Electron Paramagnetic Resonance (EPR) studies on the photo-thermo ionization process of photo-thermo-refractive glasses

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*Journal of*  
Non-Crystalline Solids

**Confirmation of Authorship**

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“Confirmation of Authorship” file.**

As corresponding author, I \_\_\_\_\_ Hellmut Eckert \_\_\_\_\_, hereby confirm on behalf of all authors that:

1. This manuscript has not been published, was not, and is not being submitted to any other journal. If presented at a conference, the conference is identified. If published in conference proceedings, substantial justification for re-publication must be presented.
2. All necessary permissions for publication were secured prior to submission of the manuscript.
3. All authors listed have made a significant contribution to the research reported and have read and approved the submitted manuscript, and furthermore, all those who made substantive contributions to this work have been included in the author list.

**NOC-D-16-00458**

**Response to Referee Comments**

**Referee Comment:**

Reviewer #2: The manuscript reports an interesting study on the EPR properties of sodium alumino silicate glasses. The results are clearly described and support the photoionization processes involving  $Ce^{3+}$ ,  $Sb^{5+}$  and  $Sb^{4+}$  ions. The papers is well-written but for small typos (ref, 28 Hosono instead of Hosone). I just recommend the authors to provide the errors when they repport the measured A and g values. After that, I support its publication

**Our response:**

We thank the reviewer for his positive assessment. The paper has been checked again for typos and all of the ones identified have been removed. Experimental error estimates of A and g values have been added to Table 3.

**Referee Comment:**

Reviewer #4: This is an interesting contribution to the science of PTR glasses. The experimental design is simple and appropriate, leading to fairly clear-cut conclusions. Frankly it's pretty surprising that this study hasn't been before on this system, given the large amount effort expended on identifying the mechanism of formation of the photo-active elements. The paper should have broad appeal in JNCS, and I have only minor comments:

**Our response:**

We thank the reviewer for his positive assessment. We were equally surprised that no EPR study of this kind had been done before on PRT glasses and hope that the paper will have a good impact.

**Referee Comment:**

1) In the future, please number the pages as it makes it easier to review. Also, proof read carefully for punctuation in particular.

**Our response:**

We have numbered the pages and proofread the manuscript again, consulting the rules on punctuation.

**Referee Comment:**

2) Ref. 8 doesn't really have much to do with NMR or X-ray diffraction. Probably should be changed.

**Our Response:**

We thank the referee for bringing this to our attention. We have inserted the correct reference to the diffraction and solid state NMR experiments.

**Referee Comment:**

3) Table 1: specify in the caption what "no dopants" means (in other words, what is the composition of B1 and B2?)

**Our Response:**

We have made the change suggested by the referee.

**Referee Comment:**

4) Experimental: I wasn't aware of a 325 nm He-Ne laser, do you maybe mean He-Cd?

**Our Response:**

We thank the referee pointing out this error. Of course we meant the He-Cd laser.

**Referee Comment:**

5) Table 2: Caption--do you mean "PTR" (not "PRT") and also does this composition match sample B5/B6 (or another sample used here)? Also, I don't see Table 2 referenced in the text anywhere.

**Our response:**

The caption has been fixed (PRT has been changed into PTR), and the compositions are the same ones used throughout the paper. A reference to Table 2 has been added to the text.

**Referee Comment.**

6) Just after Eq 1, sentence seems to end with a comma rather than a period.

**Our response:**

We have changed the punctuation.

**Referee Comment:**

7) Please explain more fully what it means to use "strain parameters" to represent disorder.

**Our response:**

The meaning of the strain parameter has been explained in the text in more detail.

**Referee Comment:**

8) The assignment of the multiplets to Sb<sup>4+</sup> is convincing, and the mechanism proposed (reduction of Sb<sup>5+</sup>) seems reasonable. It is therefore very important to provide convincing evidence or at least argument as why Sb<sup>5+</sup> would be present in the first place. Can the authors provide more than just a reference that "Sb<sup>5+</sup> often appears in silicate glasses"?

**Our response:**

We have amended the discussion and added more references in view of the referee comment. Sb<sup>5+</sup> is generally expected in glasses because of the redox equilibrium  $\text{Sb}^{3+} + \text{O}_2 \leftrightarrow \text{Sb}^{5+}$  under melting conditions, which has been studied previously in detail. References 29-31, reporting on these equilibria have been added. In particular, reference 29 combines the electrochemical measurements with the direct Mössbauer spectroscopic evidence.

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32816, USA

**Abstract**

Photo-thermo-refractive (PTR) glass is an optically transparent photosensitive sodium alumino silicate glass, containing NaF and KBr additives, along with cerium, silver, tin and antimony oxide dopants. UV-exposed regions of this glass produce NaF nanocrystals upon heating, giving rise to a permanent, localized refractive index change. In this article we examine the initial stages of this crystallization process by continuous-wave and pulsed X-band electron paramagnetic resonance (EPR) spectroscopy. UV exposure of PTR glass produces unpaired electrons whose EPR spectrum is characterized by pronounced peak splitting arising from nuclear magnetic hyperfine interactions with spin-5/2 and spin-7/2 nuclei suggesting close proximity of the unpaired electrons with  $^{121}\text{Sb}$  and  $^{123}\text{Sb}$  nuclei. These results indicate that the  $\text{Sb}_2\text{O}_3$  dopant plays a key role in the initial stages of the crystallization mechanism. Upon thermal annealing, leading to the crystallization of NaF, these species disappear, indicating their transient nature. A number of other unpaired electron species identified in the dopant free matrix appear to be unrelated to the crystallization process. These results clearly challenge the classical mechanism proposed decades ago to explain the complex crystallization process of PTR glass. Together with more recent results from optical spectroscopy they support the Nikoronov model involving (1) photoionization of  $\text{Ce}^{3+}$ , (2) transfer of this electron to  $\text{Sb}^{5+}$  species to create a  $\text{Sb}^{4+}$  species, (3) upon annealing electron transfer from  $\text{Sb}^{4+}$  to  $\text{Ag}^+$  ions, producing silver atoms, (4) coalescence of these species into Ag clusters, which (5) serve as nucleation catalysts for NaF nanocrystals.

**NOC-D-16-00458 – Highlights**

First EPR study of PTR glasses

Proof of  $\text{Sb}^{4+}$  formation upon photoionization

Confirmation of the Nikonorov model

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## **Abstract**

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disappear, indicating their transient nature. A number of other unpaired electron species identified in the dopant free matrix appear to be unrelated to the crystallization process. These results clearly challenge the classical mechanism proposed decades ago to explain the complex crystallization process of PTR glass. Together with more recent results from optical spectroscopy they support the Nikoronov model involving (1) photoionization of  $Ce^{3+}$ , (2) transfer of this electron to  $Sb^{5+}$  species to create a  $Sb^{4+}$  species, (3) upon annealing electron transfer from  $Sb^{4+}$  to  $Ag^+$  ions, producing silver atoms, (4) coalescence of these species into Ag clusters, which (5) serve as nucleation catalysts for NaF nanocrystals.

**Keywords:** photo-thermo-refractive, glass, EPR spectroscopy, crystallization, PTR

## Introduction

Photo-thermo refractive (PTR) glass is a photosensitive silicate glass whose optical properties can be modified in a controlled fashion by UV-exposure and subsequent annealing near the glass transition temperature,  $T_g$ . The chemical composition of a typical PTR glass is 15  $Na_2O$  - 4  $Al_2O_3$  - 70  $SiO_2$  - 5  $NaF$  - 5  $ZnO$  - 1  $KBr$ , with trace amounts ( $\sim 0.01$  mol%) of the dopants Ce, Ag, Sb and Sn). By exposing glass samples to UV irradiation followed by adequate heat treatment, crystallization of NaF can be induced. Spatially patterned irradiation schemes afford glass-ceramics with periodically modulated indices of refraction, which are suitable for various photonic applications (generation of holograms, spectral and spatial filtering, and production of Bragg gratings) [1-4]. Devices of this kind are currently commercialized by a number of companies. The refractive index variations have been commonly postulated to arise from the formation of NaF nanocrystals in the UV-exposed areas of the sample upon annealing whose presence in the glass-ceramic has been detected by X-ray powder diffraction and solid state NMR [5]. For this reason many experimental studies have been devoted to elucidating the mechanism of NaF crystallization, dominantly using classical (macroscopic) examination methods [1-9]. However, it must be emphasized that there is – as of yet – only weak, indirect evidence linking the NaF crystals to the physical phenomenon of the refractive index variation. It is well-known that  $Ce^{3+}$  ions in glasses can be photo-ionized by UV irradiation and that the released electron can be trapped by an intrinsic defect or by an ion of a polyvalent element [10]. Initially it had been postulated that the main trapping mechanism is expected to be the formation of atomic Ag species, according to the reaction scheme [6-8]:



Upon heating near the glass transition temperature, these Ag species are considered to form silver containing nanoparticles that serve as catalysts for the nucleation of NaF nanocrystals. The formation of these nanoparticles has been confirmed by a plasmon resonance broadband absorption centered at about 465 nm [11]. However, the special selective crystallization is known to proceed only in the presence the Sb and Sn dopants and the KBr additive, whose role in the crystallization mechanism is not at all clear.

To obtain more specific structural information about the role of these dopants at the atomic level, suitable spectroscopic techniques are required. A wide range of optical spectroscopic studies have been carried out recently. For example Anne et al. studied the structure of cerium ions in PTR glass matrix [12]. They showed that even if the two absorption bands of  $\text{Ce}^{3+}$  and  $\text{Ce}^{4+}$  species are overlapped in the UV spectral region, their spectra can be decorrelated and decomposed into individual Gaussian bands (two for each type of ion). This study was then completed by Efimov et al. who studied the structure of the intrinsic absorption tail of glass matrix and impurities in UV [13] but also proposed a more complex and accurate model for the cerium absorption in UV range based on 3 elementary Gaussian bands for each ion [14]. Finally, additional optical spectroscopic work on silver ion exchanged PTR glass corroborated the suggestion that the photo-thermo-induced process mechanisms are more complex than that shown by equation (1) and that, similarly to polychromatic glasses [2], additional ions, such as antimony, may be involved in these photo-thermo-induced processes [15]. However, due to the overlap of the absorption bands of the active species (cerium, silver, antimony, glass matrix, impurities such as iron, color centers), the analysis of optical spectroscopic curves is a very challenging task and the results are subject to a large uncertainty.

Despite the fact that several of the postulated species involved in the proposed reaction mechanisms ( $\text{Ce}^{3+}$ , atomic silver and small  $\text{Ag}_n^{(n-1)+}$  clusters,  $\text{Sb}^{5+}$  species trapping an electron) carry unpaired electrons, to the present date no studies by electron paramagnetic resonance (EPR) have been carried out on this system. In this contribution we present, for the first time, results from continuous-wave and pulsed EPR spectroscopic measurements in order to characterize the mechanisms of photo-thermo-induced ionization that precede the crystallization process and the appearance of desired localized refractive index change. By varying their dopant compositions we have identified the various types of paramagnetic species formed during this process and characterized them according to their spin Hamiltonian parameters. Furthermore the observation (or absence) of nuclear magnetic

hyperfine couplings in the EPR spectra can provide evidence for (or against) the involvement of the various different dopants in the ionization and nucleation process. Our results offer new evidence for the key role of the antimony oxide dopant in PTR glasses.

## Experimental

**Sample Preparation and Characterization.** Table I summarizes the samples investigated in the present study. Samples were prepared according to previously published protocols [4]. The powdered starting materials  $\text{Na}_2\text{CO}_3$ ,  $\text{Al}_2\text{O}_3$ ,  $\text{SiO}_2$ ,  $\text{NaF}$ ,  $\text{ZnO}$ ,  $\text{KBr}$ ,  $\text{Ce}_2\text{O}_3$ ,  $\text{Ag}_2\text{O}$ ,  $\text{Sb}_2\text{O}_3$ , and  $\text{SnO}_2$  were melted at  $1450\text{ }^\circ\text{C}$  for 120min in a platinum crucible (typical batch size 500 g), and subsequently cooled to room temperature at a rate of  $0.1\text{K/s}$ . Light exposure was done using either a Xenon lamp or a He-Cd laser (wavelength 325 nm) under the conditions specified in Table 1. Table 2 summarizes the glass composition of samples B5 and B6, according to chemical analysis. Samples B1 and B2 have the same base compositions, but were prepared without the dopants  $\text{Ce}_2\text{O}_3$ ,  $\text{Ag}_2\text{O}$ ,  $\text{Sb}_2\text{O}_3$ , and  $\text{SnO}_2$  while in the samples B3 and B4  $\text{Ag}_2\text{O}$  was the sole dopant species present. X-ray powder diffraction confirmed that all the untreated samples were amorphous, as expected. Following this preparation step, some samples were annealed at  $450\text{ }^\circ\text{C}$  for 2 h, to induce nucleation of silver clusters and NaF.

**Table 1.** Compositions and Treatment of the samples under study.

| Sample | Dopant Composition   | Light Exposure                                |
|--------|--|---|
| B1     | none   | none  |
| B2     | none   | 20 min, Xe lamp [16],                         |
| B3     | $\text{Ag}_2\text{O}$  | none  |
| B4     | $\text{Ag}_2\text{O}$  | 20 min, Xe lamp [16],                         |
| B5     | $\text{Ce}_2\text{O}_3$ , $\text{Ag}_2\text{O}$ , $\text{Sb}_2\text{O}_3$ , $\text{SnO}_2$ | none  |
| B6     | $\text{Ce}_2\text{O}_3$ , $\text{Ag}_2\text{O}$ , $\text{Sb}_2\text{O}_3$ , $\text{SnO}_2$ | He-Cd laser (325 nm), $10\text{ J/cm}^2$ [17] |

**Table 2:** Precise chemical composition of the PTR glasses B5 and B6, according to chemical analysis.

| N PTR glass composition        | Wt.%  | Mol.%  |
|--------------------------------|-------|--------|
| SiO <sub>2</sub>               | 68.5  | 71.127 |
| Al <sub>2</sub> O <sub>3</sub> | 4.8   | 2.924  |
| Na <sub>2</sub> O              | 12.9  | 12.987 |
| ZnO                            | 6.6   | 5.100  |
| NaF                            | 4.3   | 6.362  |
| KBr                            | 2.8   | 1.457  |
| Ag <sub>2</sub> O              | 0.02  | 0.0062 |
| CeO <sub>2</sub>               | 0.02  | 0.0072 |
| Sb <sub>2</sub> O <sub>3</sub> | 0.10  | 0.021  |
| SnO <sub>2</sub>               | 0.02  | 0.0082 |
| Na(Carb)                       | 0.495 | Mol.   |
| Na(Nitr)                       | 0.505 | weight |
| Total:                         | 100.0 | 62.4   |

**EPR Spectroscopy.** Continuous wave and pulsed EPR experiments were carried out in an E-580 BRUKER ELEXIS X-band EPR spectrometer operating at a microwave frequency around 9.5 GHz. The temperature was controlled by a continuous flow liquid helium cryostats ESR-900 (cw) and CF-935 (pulsed) and PID controller from Oxford Instruments ITC503. CW measurements were performed at 30 K and pulsed experiments at 10 K. All samples were broken into small pieces to fit the bottom of a 3 mm inner diameter quartz tube used as a sample holder. The echo-detected field-sweep absorption spectra, EDFS, were recorded using a two-pulse echo sequence  $(\pi/2)-\tau-(\pi)-\tau$ -echo with a typical  $\pi/2$  pulse width of 8 ns and repetition time of 2 ms. The pulse spacing,  $\tau$ , was set long enough to suppress modulation effects and the integrated echo intensities were measured as a function of the magnetic field strength over the range of 0.1 – 12.1 kG. Spectral simulations were done by the function “pepper” of the software package EasySpin® implemented in MATLAB (MathWorks, Inc) [18] The simulations consider the static spin Hamiltonian for electron-nucleus spin pairs in the solid state, given by

$$\mathcal{H} = \beta_e \mathbf{H} \cdot \mathbf{g} \cdot \mathbf{S} + \sum_j \mathbf{S} \cdot \mathbf{A} \cdot \mathbf{I}_j, \quad (1)$$

where the summation extends over the nuclear species considered in the simulation. In this expression the symmetric tensor  $\mathbf{g}$ , with its principal values  $g_{xx}$ ,  $g_{yy}$  and  $g_{zz}$ , describes the anisotropic interaction of the electron spins  $\mathbf{S}$  with the external magnetic field  $\mathbf{H}$ , while the tensor  $\mathbf{A}$  describes the anisotropic magnetic hyperfine interaction between the electron spins and the nuclear spins  $\mathbf{I}$  present. In the case of glassy systems we find a distribution the principal values for  $g$  and  $A$ , therefore, strain parameters were used to represent the disorder found in glasses. These strain parameters characterize Gaussian distribution functions of the  $g$ - and the  $A$ - parameters, resulting in field-dependent ( $g$ -strain) and field-independent ( $A$ -strain) line-broadening effects in the spectra [18]. Euler angles describing the orientation of  $g$ - and  $A$ -tensors relative to the molecular frame are considered all zero (coincident tensors).

## Results and Discussion

Figures 1 and 2 give an overview of the experimental cw- and EDFS-EPR spectra of the samples under study. Comparing the two Figures, the complementary character of the two signal detection modes becomes very evident: while the continuous-wave method is the superior one for recording and resolving the narrower spectral features, the detection of very wide spectral patterns is only possible using the EDFS method. For example, the broad EPR lineshape attributed to the  $\text{Ce}^{3+}$  ions present in samples B5 and B6 can only be detected by the EDFS method. The lineshape of this broad signal arises from the  $^2\text{F}_{5/2}$  ground state of the  $\text{Ce}^{3+}$  ions, and is probably influenced by both an anisotropic zero-field splitting compounded by an anisotropic  $g$ -tensor. Broad and poorly resolved lineshapes of this kind have been observed for various  $\text{Ce}^{3+}$  doped glasses and tentative analyses in terms of spin Hamiltonian parameters have been discussed in the literature, even though considerable uncertainty remains regarding the correct theoretical description [19-23]. We refrain from explicit simulation attempts of this component, which remained unchanged after UV irradiation and subsequent thermal annealing steps.

No EPR signals are obtained on the unexposed glass matrix (sample B1) and the unexposed sample doped only with Ag (sample B3) in either experiment. These results clearly indicate that UV exposure is crucial for the formation of unpaired electron species. Therefore, we attribute mechanistic significance with relevance to the PTR process to only those paramagnetic species that are newly formed under the stated photo-treatment conditions when all of the dopants are present. Nevertheless, some paramagnetic species are also generated

upon the UV-exposure of the dopant-free glass matrix itself (sample B2) and in the one exclusively doped with Ag (sample B4). The spectra obtained on these two samples are essentially identical, containing a very sharp singlet near  $g = 2.009$  (“component 1”, and a complex peak pattern which may reflect an unpaired electron with an anisotropic  $g$ -tensor and an anisotropic magnetic hyperfine interaction with a spin  $> 1/2$  nucleus. The sharp feature can be assigned with an oxygen defect species as has been frequently observed in irradiate silicate glasses [19]. For the remaining complex peak pattern, two possible simulations are shown in Figure 3, however, it must be pointed out that there are many alternate ways of analysis. The peak manifold associated with this species may reflect hyperfine coupling of the unpaired electron with either a spin-3/2 species ( $^{23}\text{Na}$  or  $^{79/81}\text{Br}$ ) or a spin-5/2 species ( $^{27}\text{Al}$ ), or even both. As the spectra observed in samples B2 and B4 are essentially identical, we conclude that they reflect the same kind of unpaired electron species. Since both samples only differ with respect to the presence (absence) of the Ag dopant, we can exclude the possibility that these spectral features originate from any unpaired electrons associated with silver atoms or clusters. While the nature of these unpaired electron species cannot be clarified at the present stage, it is important to note that these species are not observed in the spectrum of the UV-exposed fully doped sample. We thus conclude that they are of little relevance with respect to the mechanistic details of the photothermal refractive glasses.

Figure 4 shows the simulation of the narrow features observed in the spectrum of the fully doped unexposed sample, B5, revealing two singlets near  $g = 1.9980$  and  $1.9038$  (“components 2 and 3”, respectively). The former one represents an unpaired electron with a small  $g$ -anisotropy, while the latter one can be described by an isotropic  $g$ -tensor. The same species are present in the UV-exposed photothermo refractive glass, sample B6, in which the intensity of component 2 is greatly enhanced. The absence of nuclear hyperfine splittings suggests that these signals arise from unpaired electrons located on oxygen atoms as reported in silica-based glasses [19, 24, 25]. The most striking feature of sample B6, however, is a complex multiplet centered near  $g = 1.918$ . This species is characterized by strong magnetic hyperfine splitting revealing the interaction with the two nuclear isotopes  $^{121}\text{Sb}$  (spin-5/2) and  $^{123}\text{Sb}$  (spin-7/2). The spectrum shows very close resemblance to that observed by Schreurs and Davis in UV-exposed alkali silicate glasses containing small amounts of  $\text{Sb}_2\text{O}_3$  and  $\text{Ce}^{3+}$  dopants [26]. The formation of  $\text{Sb}^{4+}$  in glasses upon UV or gamma-irradiation has also been briefly reported by other authors [27, 28]. Figure 5 shows the simulation of our experimental spectra, assuming an isotropic hyperfine interaction for simplicity. The constants  $A_{\text{iso}} = 1427$  MHz ( $^{121}\text{Sb}$ ) and  $772.7$  MHz ( $^{123}\text{Sb}$ ) are very close to those observed in reference [26]. For

optimal fitting of these spectra, a Gaussian distribution of magnetic hyperfine couplings must be assumed, reflected in the  $A$ -strain parameter (full width at half height of the distribution). Close inspection of Figures 1 and 2 indicates further that the unexposed sample B5 contains some  $\text{Sb}^{4+}$  albeit with much reduced intensity. We attribute this observation to photo-activation by the UV component present in the spectrum of ambient laboratory lighting. The results of the present study indicate the important role of the antimony dopant species in the photonic activation process. The  $\text{Sb}^{4+}$  species can be formed if the electron generated by the photoionization of  $\text{Ce}^{3+}$  is trapped by a  $\text{Sb}^{5+}$  ion in the glass. The presence of  $\text{Sb}^{5+}$  ions in antimony containing silicate glass melts is expected based on detailed electrochemical studies characterizing the  $\text{Sb}^{3+}/\text{Sb}^{5+}$  redox equilibrium [29-31]. Direct evidence for  $\text{Sb}^{5+}$  in silicate glasses has also been given by Mössbauer spectroscopy [29, 32].

In contrast none of the EPR spectra shown here give any evidence for the presence of unpaired electrons associated with silver atoms or  $\text{Sn}^{3+}$  species. Unpaired electrons associated with either  $\text{Ag}^0$  or  $\text{Ag}^{2+}$  can be identified in the EPR spectra by the doublets resulting from the hyperfine interaction with the  $^{107}\text{Ag}$  (nuclear spin  $I = 1/2$ , 51.8% abundant) and  $^{109}\text{Ag}$  ( $I = 1/2$ , 48.2% abundant) nuclei. As documented in the literature, large nuclear hyperfine splitting is generally observed for atomic Ag [33, 34] and small  $\text{Ag}_n^{m+}$  ( $m < n$ ) clusters [35, 36]. In addition, any EPR signals belonging to  $\text{Sn}^{3+}$  ions can be identified by satellite lines arising from nuclear hyperfine coupling with the spin-1/2 nuclear isotopes  $^{117}\text{Sn}$  and  $^{119}\text{Sn}$  (natural abundances of 7.6% and 8.6%) which have been observed for  $\text{Sn}^{3+}$  ions in inorganic matrices [37]. As none of such features are observed in the EPR spectra of our glasses the results of the present study offer no evidence for the direct involvement of Ag atoms or  $\text{Sn}^{3+}$  ions in the formation of unpaired electron species.

Based on the evidence presented here, we conclude that the electron generated by the photoionization of  $\text{Ce}^{3+}$  in the first step is not trapped by an  $\text{Ag}^+$  ion but rather by a  $\text{Sb}^{5+}$  ion, resulting in the characteristic EPR signal of  $\text{Sb}^{4+}$ . The species remains rather transient, however, and disappears during the further annealing stages, which ultimately result in NaF crystallization. This is shown in Figure 6, which reveals that all the photo-generated paramagnetic species ( $\text{Sb}^{4+}$ , along with the other two sharp features near  $g = 1.988$  and  $g = 1.9038$ ) disappear upon thermal annealing. We may speculate that the electron is transferred further to Ag at this particular stage, resulting in the formation of larger silver clusters that do not give rise to a distinctive EPR lineshape. Together with the recent results from optical spectroscopy our results thus support the Nikoronov model [38, 39] involving (1) photoionization of  $\text{Ce}^{3+}$ , (2) transfer of this electron to  $\text{Sb}^{5+}$  species to create a  $\text{Sb}^{4+}$  species,

(3) upon annealing electron transfer from  $\text{Sb}^{4+}$  to  $\text{Ag}^+$  ions, producing silver atoms, (4) coalescence of these species into Ag clusters, which (5) serve as nucleation catalysts for NaF nanocrystals.

## Conclusions

Continuous-wave and pulsed electron spin echo resonance spectra obtained on UV-irradiated photo-thermal refractive glasses reveal a number of EPR active unpaired electron species, whose interaction parameters have been determined with the help of lineshape simulation routines. For detecting and characterizing the entire inventory of unpaired electrons generated during this process the combined use of continuous-wave and echo detected field sweep experiments turns out to be essential. The formation of  $\text{Sb}^{4+}$  species, which is characterized by a strong hyperfine interaction with the  $^{121}\text{Sb}$  and  $^{123}\text{Sb}$  nuclear isotopes is suggested to be a key step in the photo-activation process. Our results evidence that subsequent to the photoionization process of the  $\text{Ce}^{3+}$  ions, the electron is trapped by a nearby  $\text{Sb}^{5+}$  ion, resulting in the  $\text{Sb}^{4+}$  species. These findings corroborate the mechanism recently proposed by Nikodorov's team. There is no direct EPR evidence for the formation of  $\text{Ag}^0$  or small  $\text{Ag}_n^{m+}$  clusters upon photo-activation. The presence of various other paramagnetic species with  $g$ -values between 1.9 and 2.1 appears to be coincidental and most likely unrelated to the photothermal activation mechanism.

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## References

- [1] L. Glebov, Proc. SPIE 6545 (2007) 654507.
- [2] S. D. Stookey, D. H. Beall, J. E. Pierson, J. Appl. Phys. 49 (1978), 5114.
- [3] O. M. Efimov, L. B. Glebov, L. N. Glebova, K. C. Richardson, V. I. Smirnov, Applied Optics 38 (1999), 619.

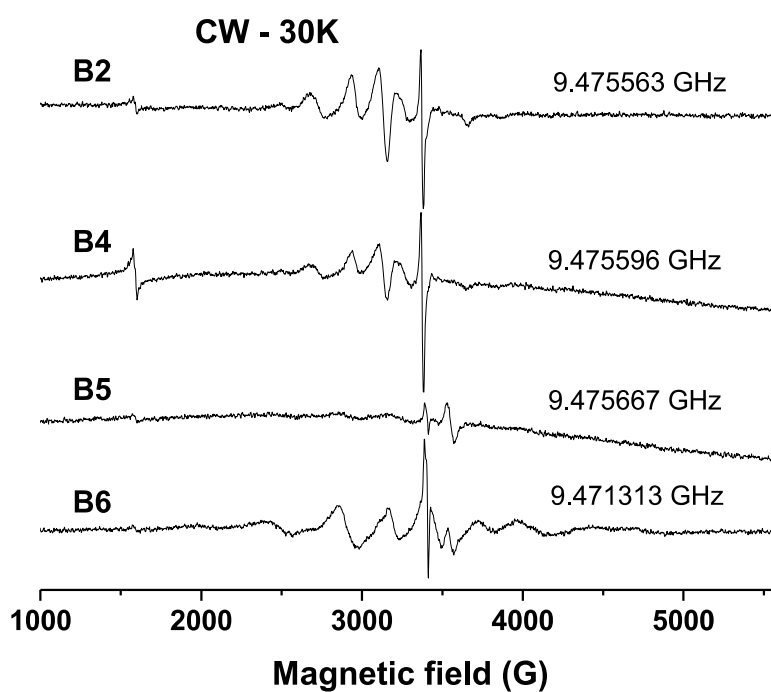


- [4] O. M. Efimov, L. B. Glebov, V. I. Smirnov, L. Glebova, U.S: Patent Nr. 6,586,141 (2003).
- [5] J. W. Zwanziger, U. Werner-Zwanziger, E. D. Zanotto, E. Rotari, L. N. Glebova, L. B. Glebov, J. F. Schneider *J. Appl. Phys.* 99 (2006), 083511.
- [6] L. B. Glebov, *Opt. Mater.* 25 (2004), 413
- [7] J. Lumeau, L. Glebova, V. Golubkov, E. D. Zanotto, L. B. Glebov, *Opt. Mater.* 32 (2009), 139.
- [8] J. Lumeau, L. Glebova, L. Glebov, *J. Non-Cryst. Solids* 354 (2008), 425
- [9] L. Glebova, J. Lumeau, M. Klimov, E. D. Zanotto, L. B. Glebov, *J. Non-Cryst. Solids* 354 (2008), 456.
- [10] J. S. Stroud, *J. Chem. Phys.* 35 (1961), 844
- [11] J. Lumeau, L. Glebova, L. Glebov, *Opt. Mater.* 36, 621–627 (2014)]
- [12] M. Anne, J. Lumeau, L. Glebova and L.B. Glebov, *J. Non-Cryst, Solids* 356 (2010), 2337.]
- [13] O. M. Efimov, A. I. Ignatiev, N. V. Nikonorov, E. S. Postnikov, *J. Non-Cryst. Solids* 357 (2011), 3500.
- [14] O. M. Efimov, A. I. Ignatiev, N. V. Nikonorov, E. S. Postnikov, *J. Non-Cryst. Solids* 361 (2013), 26-37
- [15]. E M. Sgibnev, A. I. Ignatiev, N. V. Nikonorov, A, M. Efimov, E. S. Postnikov, *J. Non-Cryst. Solids* 378 (2013) 213.
- [16] K. Chamma, J. Lumeau, L. Glebova, and L. B. Glebov, *J. Non Cryst. Solids* 356 (2010), 2363.
- [17] O. M. Efimov, L. B. Glebov, H. P. Andre, *Appl. Opt.* 41 (2002), 1864.
- [18] S. Stoll, A. Schweiger, EasySpin, a Comprehensive Software Package for Spectral Simulation and Analysis in {EPR}. *J. Magn. Reson.* 178 (2006) 42.
- [19] D. L. Griscom, *J. Non-Cryst. Solids* 40 (1980), 211.
- [20] E. Malchukova, B. Boizot, G. Petite, and D. Ghaleb, *Phys. Status Solidi (c)* 4 (2007), 1280.
- [21] C. Carnevali, M. Mattoni, F. Morazzoni, R. Scotti, M. Casu, A. Musinu, R. Krsmanovic, S. Polizzi, A. Speghini, M. Bettinelli, *J. Am. Chem. Soc.* 127 (2005), 14681.
- [22] A. Bishay, C. Quadros and A. Piccini, *Phys. Chem. Glasses* 15 (1974) 109.
- [23] M. Chiesa, K. Mattsson, S. Taccheo, T. Robin, L. Lablonde, D. Mechin, D. Milanese. *J. Non-Crystalline Solids* 403 (2014) 97.
- [24] D. Möncke, D. Ehrt, *Opt. Mater.* 25 (2004), 425
- [25] D. L. Griscom, *J. Non-Cryst. Solids* 357 (2011), 1945

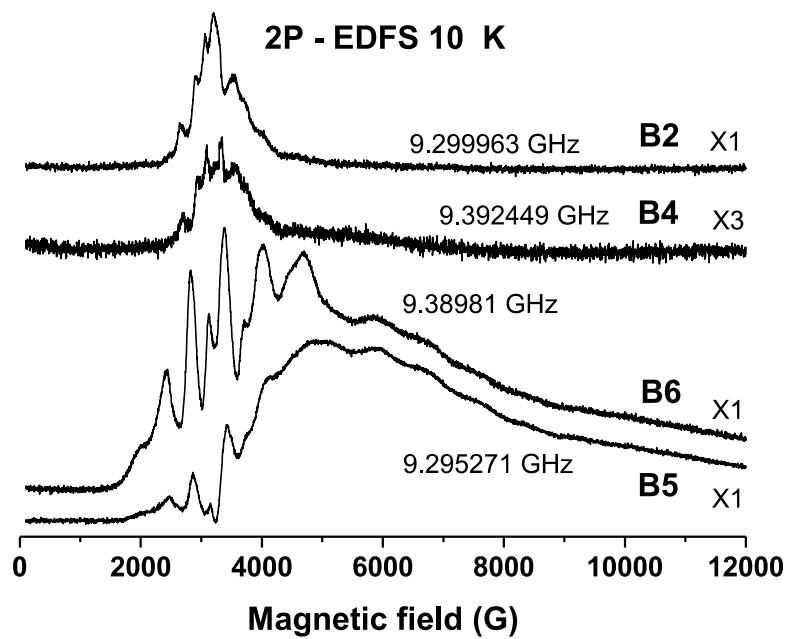
- [26] J. W. H. Schreurs, D. H. Davis, *J. Chem. Phys.* 71 (1979), 557.
- [27] A. I. Aleksandrov, N. N. Bubnov, A. I. Prokof'ev, *Appl. Magn. Reson.* 9 (1995), 251.
- [28] D. Möncke, D. Ehrt, *J. Non-Cryst. Solids* 345/346 (2004), 319.
- [29] B. Stahlberg, B. D. Mosel, W. Müller-Warmuth, F. G. K. Baucke, *Glastechn. Ber. – Glass Sci. Technol.* 61 (1988), 335
- [30] C. Rüssel, G. Gönna, *J. Non-Cryst. Solids* 260 (1999), 147
- [31] K. D. Kim, S. H. Lee, *Ceramics-Silikáty* 54 (2010), 14.
- [32] W. Müller-Warmuth, H. Eckert, *Phys. Rep.* 88 (1982), 91, and references therein.
- [33] H. Hosono, J. Nishii, H. Kawazoe, T. Kanazawa, K. Ametani, *J. Phys. Chem.* 84 (1980), 2316.
- [34] E.J. Friebele, D.L. Griscom in *Treatise on Materials Science and Technology*, vol. 17, Glass II. M. Tomozawa, R.H. Doremus (editors). Academic Press 1979.
- [35] N. I. Mel'nikov, D. P. Peregood, R. A. Zhitnikov, *J. Non-Cryst. Solids* 16 (1974), 195.
- [36] A. T. Brant, I. E. Halliburton, N. C. Giles, S. A. Basun, A. . Grabar, D. R. Evans, *J. Phys. Cond. Matter* 25 (2013), 205501.
- [37] B. Xu, L. Kevan, *J. Phys. Chem.* 95 (1991), 1147.
- [38] S. A. Ivanov, N. V. Nikonorov, A. I. Ignatiev, *Proc. 3<sup>rd</sup> Int. Conf. Photonics, Optics and Laser Technology*, 2015, 78.
- [39] S. A. Ivanov, A. I. Ignatiev, N. V. Nikonorov, in *Holography, Advances and Modern Trends IV*, M. Hrabovsky, J. T. Sheridan, A. Fimia, eds, *Proc. SPIE* 9508, 95080E (2015), 1.

**Table 3:** EPR simulation parameters extracted from the spectra. Electron spin is 1/2 in all cases.  $Lwpp$  is the line width for isotropic broadening used for convolution of the CW-EPR spectrum. Peak-to-peak (pp) refers to the horizontal distance between the maximum and the minimum of a first-derivative lineshape. The first and second entry of  $lwpp$  specifies Gaussian and Lorentzian broadening, respectively.  $A$ - and  $g$ -strain specifies the width of the Gaussian distribution of  $A$ - and  $g$ -values. For components 1, 2, 3 and  $^{121/123}\text{Sb}$  the uncertainty of  $g$  is  $\pm 0.0005$ ; in the other cases it is  $\pm 0.01$ .

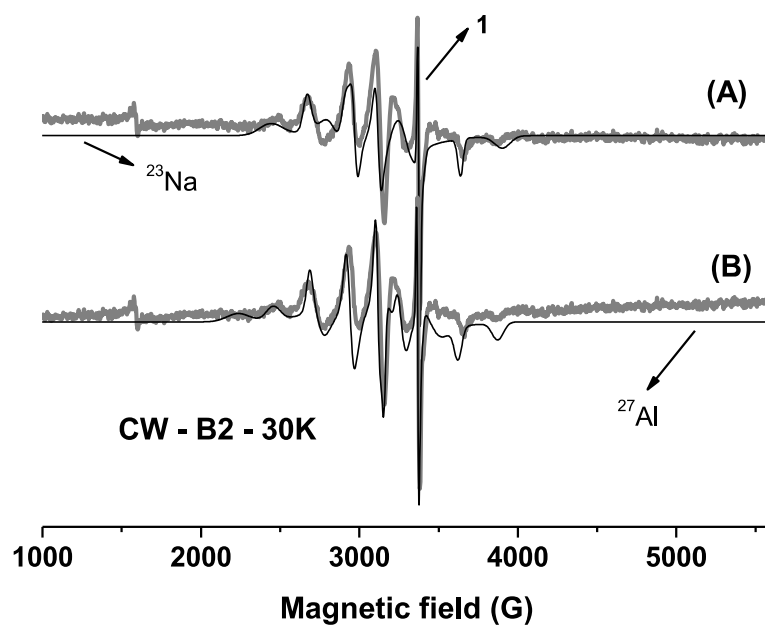
| Sample                | Spectral Component                         | $g$    | $g$ -strain<br>$\pm 5\%$                              | $A$<br>(MHz)<br>$\pm 2\%$                          | $A$ -strain<br>(MHz)<br>$\pm 5\%$ | $Lwpp$<br>(MHz)<br>$\pm 1\%$ |
|-----------------------|--|--------|---|--|-----------------------------------|------------------------------|
| B2, B4                | 1  | 2.0053 | -   | -  | -                                 | 1.5, 0                       |
|                       | $^{23}\text{Na}$ or<br>$^{79/81}\text{Br}$ | 2.41   | -   | 820  | 380                               | 1.0, 0                       |
|                       |  | 2.20   | -   | 450  | 220                               |                              |
|                       |  | 1.91   | -   | 660  | 200                               |                              |
| $^{27}\text{Al}$      | 2.39                                       | -      | 794   | 220  | 2.5, 0                            |                              |
|                       | 2.12                                       | -      | 0   | 0  |                                   |                              |
|                       | 2.04                                       | -      | 89  | 650  |                                   |                              |
| B5                    | 3  | 1.9063 | -   | -  | -                                 | 4.0, 0.6                     |
|                       | 2  | 1.9969 | 0.003   | -  | -                                 | 0.2, 0.3                     |
|                       |  | 1.9869 | 0.009   | -  | -                                 |                              |
|                       |  | 1.9835 | 0.004   | -  | -                                 |                              |
| B6                    | 3  | 1.9062 | -   | -  | -                                 | 0.7, 5.0                     |
|                       | 2  | 1.9972 | 0.004   | -  | -                                 | 0.2, 0.3                     |
|                       |  | 1.9851 | 0.008   | -  | -                                 |                              |
|                       |  | 1.9836 | 0.004   | -  | -                                 |                              |
| $^{121/123}\text{Sb}$ | 1.9169                                     | -      | $^{121}\text{Sb}$ : 1427<br>$^{123}\text{Sb}$ : 772.7 | $^{121}\text{Sb}$ : 360<br>$^{123}\text{Sb}$ : 195 | 0, 4.5                            |                              |



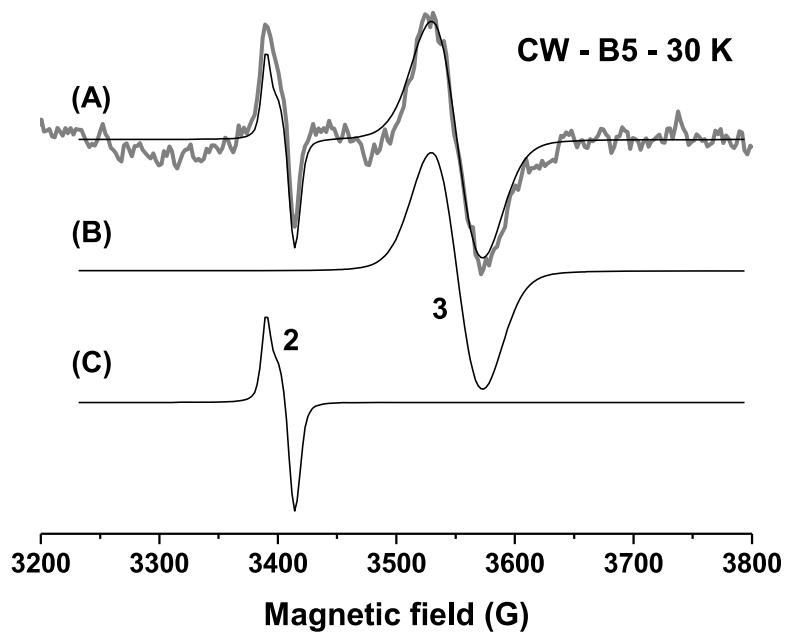
**Figure 1:** Continuous wave X-band EPR spectra obtained at 30 K on samples B2, B4, B5, and B6. The sharp signal at 1593 G is attributed to Fe<sup>3+</sup> impurities. Microwave frequencies are indicated.



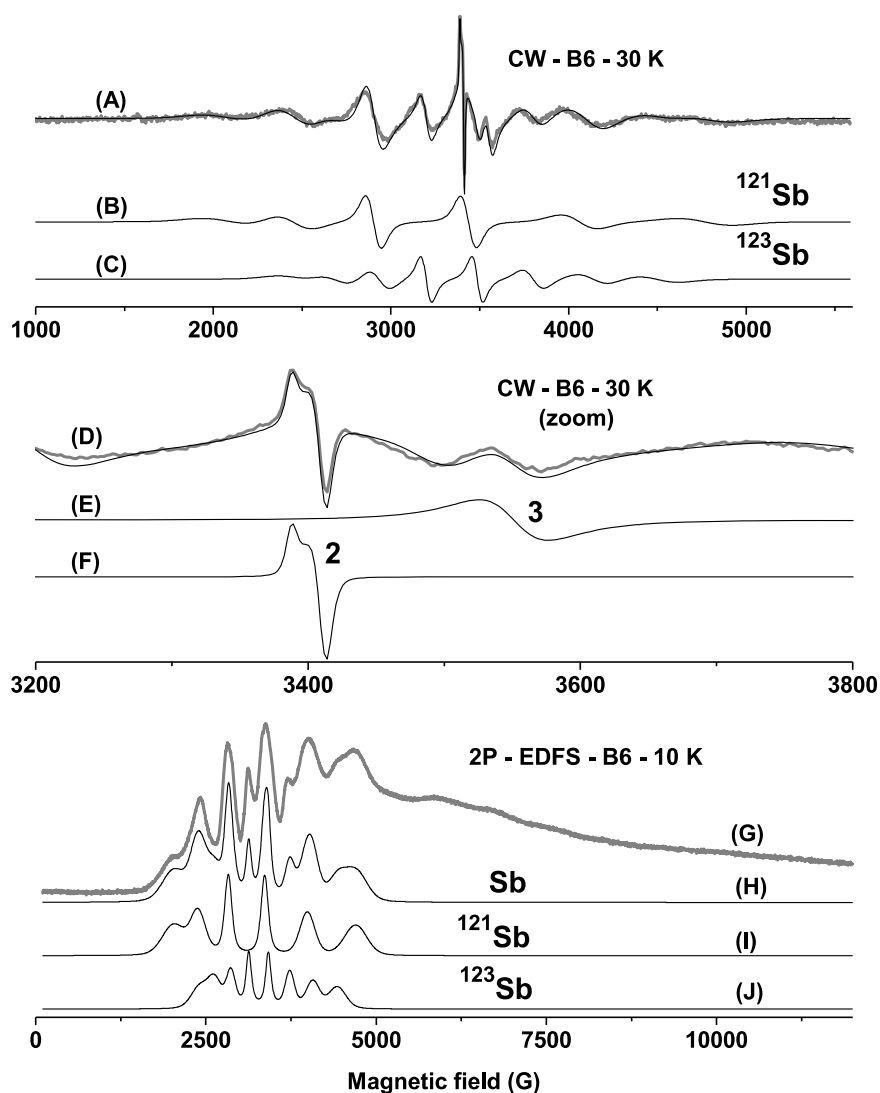
**Figure 2:** Echo detected field sweep X-band EPR spectra obtained at 10 K on samples B2, B4, B5, and B6. Microwave frequencies are indicated.



**Figure 3:** Experimental cw spectrum of sample B2 (gray traces) and simulations (black traces). (A) Simulation with isotropic component 1 and hyperfine coupling to one spin-3/2 nucleus ( $^{23}\text{Na}$  or  $^{79/81}\text{Br}$ ). (B) Simulation with isotropic component 1 and hyperfine coupling to one spin-5/2 nucleus ( $^{27}\text{Al}$ ). The sharp signal at 1593 G is attributed to  $\text{Fe}^{3+}$  impurities. Microwave frequency: 9.475563 GHz.

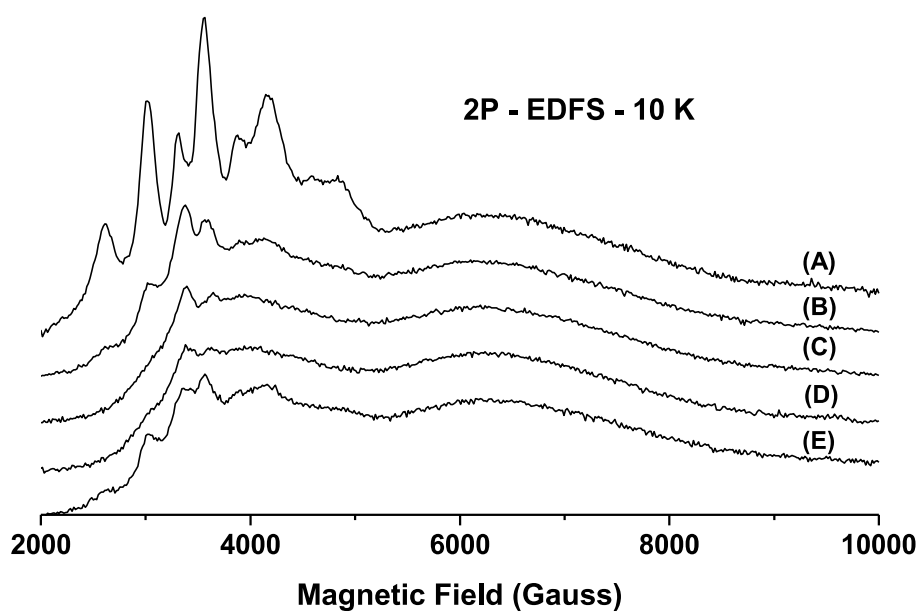


**Figure 4:** Experimental cw spectrum of sample B5 (gray trace) and simulations (black traces). (A) Simulation with both anisotropic and isotropic components 2 and 3, respectively. (B-C) Isolated traces of simulated components 2 and 3. Microwave frequency: 9.475563 GHz.



**Figure 5:** Experimental cw and EDFS spectra of sample B6 (gray traces) and simulations (black traces). (A) Experimental and simulated cw-EPR spectra. (B)  $^{121}\text{Sb}$  isotope component of simulated spectrum. (C)  $^{123}\text{Sb}$  isotope component of simulated cw-EPR spectrum. (D-F) Zoom at the central part of the cw-EPR spectrum where components 2 and 3 can be visualized. (G) Experimental EDFS spectrum. (H-I) Simulations of the spectral component assigned to  $\text{Sb}^{4+}$  ions. Microwave frequency: 9.471313 GHz.





**Figure 6:** Effect of thermal annealing upon the EDFS spectrum of PTR glass: From top to bottom: (A) sample B6, no annealing, (B) sample B6, annealed at 450 °C, 1h, and at 520 °C, 2h; (C) sample B6, annealed at 450 °C, 1h, (D) sample B5, annealed at 450 °C, 1h, (E) sample B5, no annealing.