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Strontium-doped organic-inorganic hybrids towards three-dimensional scaffolds for osteogenic cells

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ABSTRACT

Biomimetic organic–inorganic hybrid bioscaffolds are developed to complement or replace damaged fragments in bone tissue surgery. The aim of this work was to develop a simple and fast method to prepare composite material for bone engineering, avoiding time consuming and complex methodologies. The resulting materials (also called in this work as hybrid composites or hybrid scaffolds) have a three-dimensional

macroporous polymer-like network derived from triethoxyvinylsilane (TEVS) and 2-hydroxyethylmethacrylate (HEMA) monomers, with incorporated calcium, strontium, and phosphate ions. The materials were fully characterized using FT-IR, biomineralization studies, scanning electron microscopy (SEM), energy dispersive X-ray spectroscopy, scratch tests, Young's modulus and compressive strength tests, and gas physisorption. We report a comprehensive study on the *in vitro* effect of novel strontium doped materials on human bone cells. *In vitro* investigations were conducted using a normal human osteoblast cell line that mimics the cellular events of the *in vivo* intramembranous bone formation process. The materials do not have a negative impact on the survival of the normal human osteoblasts; moreover, materials doped with strontium show that not only are cells able to survive, but they also attach to and grow on a bioscaffolds surface. For this reason, they may be used in future *in vivo* experiments.

1. Introduction

Biomimetic materials with structures that have hierarchical interconnected pores and a network-like nature, have been intensively studied in recent years, because they form excellent scaffolds for biomedical applications, including controlled drug release, sophisticated substitutes for tissue engineering, bioresorbable structures, etc. Such scaffolds are used as tissue templates because they can simultaneously form an ideal structure that is necessary to ensure cell growth and viability [1]. Inside the body, such materials induce a complex series of physicochemical reactions that lead to the formation of an interfacial bone-like apatite layer [2, 3]. The dissolution of biomaterial devoted to bone replacement induces ion exchanges with biological fluids (or their mimics) enabling the formation of a mineral phase(s), direct interfacial binding between bioactive material and the bone; it should be emphasized that a number of review papers described this issue [1, 2]. Here, both textural properties and chemical composition must be chosen in such a way that the material might be used as an efficient implant, creating a strong interfacial bond with host tissues, and which is extremely important, stimulating bone cell adhesiveness, differentiation, and proliferation [4].

Strontium has been used as a medicinal additive for more than a hundred years and its use as a dietary supplement is reported as the way to maintain strong bones [5-7]. Since the development of strontium ranelate, which is a drug that reduces the incidence of fractures in patients with osteoporosis, there has been an increasing awareness of the biological role of strontium – especially its role in bone and in interactions with calcium ions [8]. Furthermore, in the literature there are several examples of various

biomaterials, including scaffolds, containing Sr^{2+} ions [9-11]. Among the trace metals present in human bone, strontium is mainly, but not solely, correlated with bone compression strength [12]. As an alkaline earth that is closely chemically related to calcium, strontium affects bone formation and resorption through direct and indirect effects on bone cells or bone minerals. It also has an anabolic activity that is very important from a bone balance point of view, as can be seen by comparing healthy and osteopenic skeletons [5, 13].

There has been a greater interest in the biological role of strontium has arisen since the discovery of its beneficial effect on bone, involving the reduction of incidence of fractures in osteoporotic patients [14,15]. Osteoporosis is a disease that causes thinning of the bones with a reduction in bone mass due to depletion of calcium and bone protein. Hence, the amount of bone tissue in a certain volume of bone dramatically decreases. It has been shown that strontium, as a bone-seeking agent, inhibits bone resorption, decreases osteoclast activity, and inhibits bone-resorbing activity. In addition, strontium salts can increase bone mass without causing detectable bone mineralization defects [16-20]. The mechanism of the strontium action at the bone surface is not fully known, but some mechanisms are postulated. For instance, calcium receptors in some types of cell can be activated via the presence of Sr^{2+} cations, resulting in the activation of secondary messenger molecules used in signal transduction in biological cells [16].

In this study, we investigated the *in vitro* effect of strontium-doped organic–inorganic scaffolds manufactured using the sol-gel process on biological responses associated with osteoblast viability. In this manner, we examined human bone cell behavior on three-dimensional polymer-like networks based on HEMA and TEVS, with incorporated

Ca²⁺ and PO₄³⁻ ions and, additionally, doped with strontium. Selected samples were also covered with synthetic bone-like hydroxyapatite in order to compare their *in vitro* behavior to new systems reported in this paper. To prepare mentioned above materials we used well-known monomers, e.g. HEMA which is safe for human and is already used in medicine [21,22]. We report on a simple and fast method to obtain interconnected porous scaffolds for bone engineering with mechanical properties similar to those in human bones. In addition, a desirable feature of the obtained formulations based on creating a strong interfacial bond with host cells and tissues was studied on a normal human osteoblast cell line (NHOst), which mimics the cellular events of the *in vivo* intramembranous bone formation process. According to our best knowledge, such formulations possessing this sort of matrix with mentioned above dopants have not been studied so far, both taking into consideration their chemical and mechanical properties, and also *in vitro* studies.

2. Materials and methods

2.1. Materials for synthesis

TEVS, HEMA, triethylphosphate ((C₂H₅O)₃PO, used as a source of PO₄³⁻ anions), and tert-butyl hydroperoxide (~5.5 M in decane) were purchased from Aldrich, Germany, and used without further purification. Calcium dichloride dihydrate (CaCl₂·2H₂O, used as a source of Ca²⁺ cations), and strontium dichloride hexahydrate (SrCl₂·6H₂O, a source of Sr²⁺ cations) were purchased from Sigma-Aldrich, Germany. An aqueous solution of 0.01 M hydrochloric acid was prepared from 37% HCl (Chempur, Poland). Acetic acid

(80%), 25% aqueous ammonia solution, ammonium carbonate ((NH₄)₂CO₃), and sodium dihydrogen phosphate dihydrate (NaH₂PO₄·2H₂O) were purchased from POCh, Poland.

2.2. Material characterization

Microanalyses were conducted using an elemental analyzer 2400 CHNS Vario EL III (Elementar). The concentration of metal ions was determined by the inductively coupled plasma technique, using an ARL model 3410 sequential spectrometer (Fisons Instruments). Infrared spectra were recorded on a Bruker 66/s FT-IR spectrophotometer in KBr pellets. All materials were characterized by recording X-ray powder diffraction patterns using a Bruker D8 ADVANCE diffractometer equipped with a copper lamp ($\lambda_{\text{CuK}\alpha} = 1.5418 \text{ \AA}$). Standard measurements were performed for $2\theta = 10\text{--}90^\circ$, in increments of 0.016° with a counting time of 0.3 s. Scaffold morphology was tested by high-resolution transmission electron microscopy, using an FEI Tecnai G2 20 X-TWIN microscope operating at 200 kV and providing 0.25 nm resolution, and equipped with an energy dispersive X-ray spectroscopy microanalyzer, and SEM, using a Hitachi S-3400N microscope equipped with an EDS ThermoNoran System SIX. pH values were measured using an ELMETRON pH meter CP-401. Viscosity of prepared sols was measured using RVDV-I Prime rotary viscometer (Brookfield Engineering Laboratories) equipped with RV spindle set. Scratch tests were performed using a Wolff–Wilborn pencil hardness tester. The density of the resulting materials was measured using a gas pycnometer on an Ultrapyc 1200 e (Quantachrome Instruments). Mechanical properties, including Young's modulus and compressive strength, were studied using a TA-XT texture analyzer (TA Plus, Lloyd Instruments) equipped with mechanical grips and

ElectroForce 5100 BioDynamic Test Instrument from BOSE company. Nitrogen porosimetry was performed at 77 K on a Micromeritics ASAP 2020M system. The specific surface area (S_{BET}) was calculated by the Brunauer–Emmett–Teller method. The mesopore size distribution was determined using the Barrett–Joyner–Halenda method, applied on the desorption branch.

2.3. Preparation of the organic–inorganic hybrid materials

Organic–inorganic hybrid materials, labeled 1HCP, 1HCP_H, 1HCPP, 1HCPS, 1HCPS_H, and 1HCPSP, were prepared using the sol-gel process. There are two types of sample. The first is composed of TEVS and HEMA, and is doped with Ca^{2+} and PO_4^{3-} ions (1HCP-type materials). The second group of materials is additionally enhanced with strontium cations (1HCPS-type materials). Some samples in each group were biomineralized in biomimetic fluids (1HCPP and 1HCPSP) or covered with synthetic bone-like hydroxyapatite (1HCP_H and 1HCPS_H). For detailed labels, compositions, and comments, see Table 1.

In a typical synthesis, the mixture of TEVS/HEMA was dissolved in anhydrous ethanol mixed with an aqueous solution of 0.01 M HCl. Hydrochloric acid was used as a catalyst for the hydrolysis and condensation reactions. The mixture was polymerized by heat treatment at 70 °C in the presence of tert-butyl hydroperoxide, which is commonly used as an initiator in radical polymerization. Next, the obtained solution was vigorously stirred for 5 h and then cooled to room temperature. To obtain doped materials, the resulting composite solution was mixed with $\text{CaCl}_2 \cdot 2\text{H}_2\text{O}$, $\text{SrCl}_2 \cdot 6\text{H}_2\text{O}$ and $(\text{C}_2\text{H}_5\text{O})_3\text{PO}$,

leaving an hour interval between each addition, and the resulting mixture was left overnight, with stirring.

2.4. Preparation of porous scaffolds

Appropriate sugar templates were prepared from commercially available granulated sugar mixed with ammonium carbonate (3–5 wt%), moistened with evaporated copolymer solution, and closely packed in cubic plastic molds (4 x 4 cm). The sugar was left in the molds at 65 °C until the gelation process occurred. The sugar template was then washed out using warm distilled water (40–50 °C) until the filtrate had no traces of sugar. The presence of sugar was examined using the Molisch (α -naphthol) test. The chemical basis of this reaction is the production of furfural derivatives on heating carbohydrates with naphthols to give colored compounds. A reddish violet appears in the presence of a very small amounts of carbohydrates. The complete elimination of ammonia produced during decomposition of carbonate salt was examined using elemental analysis.

2.5. The Molisch test

In a standard procedure, 0.5 mL of the filtrate was mixed with 5 mL of cold 75% H_2SO_4 . Three drops of a 3% α -naphthol solution in ethanol were added to the acid mixture. A yellow color is produced by the addition of the naphthol. After this, the mixture was warmed up on a water bath at 80 °C. Depending on the amount of carbohydrates, a red to blue-violet color appears throughout the whole mixture. In the absence of carbohydrates, the examined solution remains yellow.

2.6. Biomineralization assay

The biomineralization process was performed by soaking the hybrid samples in Dulbecco's modified Eagle's medium (DMEM), maintaining the solution at 37 °C. The process was conducted using a static protocol; this means that the DMEM was introduced once, without exchanging the biological fluid in the container. Samples after exposition in DMEM were washed out using distilled water (three washes with 5 mL), to remove any water-soluble inorganic salts, and then using acetone (three washes with 5 mL) and dried under compressed air. The newly formed inorganic layer was characterized using FT-IR spectroscopy, X-ray powder diffraction, energy dispersive X-ray spectroscopy, and elemental analysis. The infrared spectra and elemental analyses were measured on approximately 1 mg of material directly scraped from the samples' surfaces. The X-ray powder diffraction and energy dispersive X-ray spectroscopy measurements were carried out on the surfaces of the samples. The ion concentration in the biological medium was measured for different soaking times (0.5, 1, 6, and 12 h, and 1, 2, 3, 4, and 7 days), using inductively coupled plasma analysis. Additionally, X-ray powder diffraction for all samples was performed after 7–9 weeks of soaking in DMEM solution, to confirm the nature of the mineral phase that crystallized or precipitated on the materials' surfaces.

2.7. Preparation of samples covered in synthetic hydroxyapatite

First, 16.17 g (109.99 mmol) of $\text{CaCl}_2 \cdot 2\text{H}_2\text{O}$ and 10.30 g (66.02 mmol) of $\text{NaH}_2\text{PO}_4 \cdot 2\text{H}_2\text{O}$ were added to 100 mL of distilled water and dissolved by stirring at room temperature. Next, the resulting clear solution was poured into a measuring flask

and 35.08 mL of 80% acetic acid was added and the volume made up to 1 L with distilled water. The hybrid scaffolds were evaporated under vacuum to remove the air from the pores. Then, the evaporated samples were placed in a beaker (volume, 35 mL, height, 50 mm), and flooded with the prepared solution. To a second beaker (the same size), 4 mL of 25% $\text{NH}_{3(\text{aq})}$ was added. Both beakers were covered with punched Parafilm, to reduce the diffusion of ammonia, and placed in a sealed chamber (volume, 1 L). After 48 h, the scaffolds were removed from the solution, washed out using distilled water (three washes of 15 mL) and dried under vacuum at 60 °C. The composition of the synthetic hydroxyapatite was confirmed using FT-IR spectroscopy and X-ray powder diffraction.

2.8. Young's modulus and compressive strength

The Young's modulus and compressive strength were determined for monoliths and scaffolds for both 1HCP and 1HCPS systems. These preliminary studies were carried out to determine the materials' utility as potential implants in bone tissue engineering. The collected data were compared with results obtained for cortical [23-26] and cancellous bone [27-30] in the case of monoliths and scaffolds, respectively. The mechanical tests were performed with ElectroForce 5100 BioDynamic Test Instrument from BOSE company and using a TA-XT texture analyzer (TA Plus, Lloyd Instruments) equipped with mechanical grips. Test instruments provide accurate characterization of biomaterials and biological specimens. They can be used for the evaluation of a variety of specimens, including biomaterials, cellular and cell-seeded scaffolds, native tissue

samples and tissue-engineered constructs. The obtained data were average value of measured factor \pm standard deviation. All samples were measured at least 7 times.

2.9. Contact angle and surface free energy measurements using the Owens–Wendt method

The surface free energy (γ) and its dispersive (γ^d) and polar (γ^p) components were determined for the obtained materials using contact angle measurements and the Owens–Wendt method [31, 32]. The aqueous contact angle analysis was carried out using sessile drop method and a DataPhysics OCA20 instrument to determine the wettability of the resulting coatings. The averages of at least five readings taken at different locations on each sample surface were calculated.

2.10. Scratch tests

Scratch tests were performed on a glass surface covered by a thin film of resulting materials. A pencil hardness test was carried out according to Chinese standard GB/T6739-2006.

2.11. Human osteoblast cell culture

Cells from NHOst were cultured in osteoblast growth basal medium supplemented with 10% fetal bovine serum, 0.1% ascorbic acid, and 0.1% gentamycin. The cell line and supplements were obtained from Lonza, Belgium. The adherent cells were cultured in 75 cm² flasks (Becton Dickinson, USA). The cells were maintained at 37 °C in a fully

humidified atmosphere at 5% CO₂ in air (Hera Cell Heraeus, Germany). The cell line populations used in this experiments were taken between passages 5 and 6.

2.12. Cytotoxicity evaluation of hybrid biocomposites on the NHOst cell line

In this studies both monoliths and foam-like samples were considered. Cylindrical specimens from each type of material with a 'diameter x height' dimensions: 6 mm x 2 mm and 12 mm x 10 mm for monoliths and scaffolds, respectively were investigated. In the SRB assay 1 sample (monolith or scaffold) per well of plate was added. Cells in the exponential growth phase were seeded at 0.2×10^6 cells/well on 24-well plates (Becton Dickinson, USA), and then incubated for 24 h. On the second day, after fixing the cells, the culture medium was refreshed and the following biocomposite samples were placed on the cells: 1HCP, 1HCP_H, 1HCPP, 1HCPS, 1HCPS_H, and 1HCPSP. The plates were incubated for 72 h. After this time, a sulforhodamine B assay (SRB assay) (Sigma, USA) was performed [33], to measure cell survival, after prior fixation of cells with trichloroacetic acid (Sigma, USA) [34]. Absorbance at a wavelength of 562 nm was measured on a plate reader (Tecan Infinite M200 Pro, Austria). Three independent repetitions of each test were performed.

2.13. Preparation of NHOst cell line for SEM

Prior to seeding the cells on the hybrid biocomposites, samples of 1HCPS were incubated for 24 h in osteoblast growth basal medium (Lonza, Belgium). The next day, cells at the exponential growth phase were seeded at 10^5 cells per sample and placed in a 12-well plate (Becton Dickinson, USA). This seeding procedure was repeated twice

more, on days 9 and 16 of incubation. Throughout the whole experiment, the medium was changed every 2–3 days. After 20 days, the cells were fixed for SEM. The samples were washed with phosphate buffered saline to remove unfixed cells, and the scaffolds were fixed in 2.5% glutaraldehyde (Serva, Germany) prepared on a 0.2 M cacodyl buffer (Serva, Germany) for 1 h at 4 °C. The samples were then rinsed three times in cacodyl buffer for 5 min, refixed with 1% osmium tetroxide (Serva, Germany) for 1 h at 4 °C, and finally rinsed three times with 0.2 M cacodyl buffer for 5 minutes. The samples were then dehydrated through a series of graded alcohol solutions (Chempur, Poland) with incubation for 15 minutes in each solution (50, 70, 80, and 96%). After three 15-minute incubation periods in absolute alcohol (Chempur, Poland), the samples were transferred to acetone (Chempur, Poland).

2.14. Statistical analysis

All results were obtained by performing three independent replicates of each experiment. The results are expressed as means \pm standard deviations. Statistical analysis was carried out using Dunnett's multiple comparison test. Values of $P < 0.05$ were considered statistically significant.

3. Results and discussion

3.1. Synthesis of organic-inorganic hybrid materials

Organic–inorganic hybrid materials were synthesized using the sol-gel process. The polymer network constitutes a mixture of TEVS and HEMA in a 1:4 molar ratio. The presence of specific groups of the synthesized composites were determined from FT-IR

spectroscopy. In the spectra, the characteristic absorption bands of siloxane Si-O-Si moiety are observed at ~ 1040 , 800 and 470 cm^{-1} . The peaks at ~ 1490 , 1415 and 870 cm^{-1} are assigned to the C-O vibration bands. It should be noted that for samples containing phosphate ions their bands are not present in the spectra, because they are overlapped by various organic groups. In turn, peak at $\sim 3420\text{ cm}^{-1}$ is matched to hydroxyl group. Such substrates were chosen for this specific purpose. Not only are the compounds able to form linear chain copolymers and cross-linking connections spontaneously, but they also create a specific polymer network that strongly affects the scaffold's strength. In TEVS, (i) EtO⁻ ions allow hydrolysis to form silanol groups, (ii) silanols allow condensation/polycondensation to form siloxane moieties, and additionally promote ions' exchange and hence bone-like apatite formation, and (iv) vinyl groups (also present in HEMA) allow polymerization [35]. In turn, HEMA is (i) biocompatible and already used in medicine; has wide applications such as contact or intraocular lenses, soft tissue replacement and vascular prostheses, and (ii) increases the immobilization of cells, bioactive molecules (e.g. enzymes) or pharmacological compounds [36]. The formation of various covalent bonds between organic chain fragments is possible because of reactive functional groups, such as vinyl ($-\text{CH}=\text{CH}_2$) and hydroxyl ($-\text{OH}$), which are present in the ramified matrix. In the TEVS/HEMA composite structure, the condensed silanol ($\equiv\text{Si-OH}$) groups easily form siloxane ($-\text{Si-O-Si-}$) moieties that have an influence on the formation of strong connections between polymer chains (Fig. 1).

Moreover, silanols are able to promote the exchange of ions between the composite and the outer environment. As a consequence, the formation of various inorganic salts, such as phosphates and carbonates, is possible [35]. It is worth noting that the

crystallization of such salts initiates bone growth and enables the treated bone fragment to recover. This also has a positive impact on cellular behavior and activity [37].

During syntheses, inorganic salts such as calcium and strontium chlorides and triethylphosphate were added. These salts constitute a source of calcium Ca^{2+} , strontium Sr^{2+} , and phosphate PO_4^{3-} ions, respectively. As emphasized in Introduction paragraph, calcium and phosphorous salts are the key factors for bone tissue formation, while strontium salts can effectively decrease bone resorption and simultaneously increase bone formation, raising the possibility for potential usage in osteoporosis treatment strategies. It is worth noting that these materials possess a sufficient shelf life (in the liquid state before curing), of about 6 months. During this time viscosity equaled ca. 950 cP and after 6 months viscosity rapidly grew up and sol transformed to transparent gel; this is important data for easy and wide distribution. Moreover, the developed process is simple, and does not require excessive costs or complex equipment. Homogeneous and transparent gels were obtained for all materials synthesized (Fig. 2).

3.2. Structure of hybrid scaffolds

Cancellous bone is a highly porous structure with a wide distribution of pore sizes and wall thicknesses [27-30]. The specific three-dimensional scaffold with a highly interconnected macroporous network and tunable mechanical properties is highly desirable in various medical applications [38]. The porosity and pore size of bioscaffolds play a crucial role in bone formation [39]. For pore sizes larger than 50 μm , tissue mineralization is observed; for bigger pores (>100 μm) this process extends to a depth

of 1000 μm and there is a good chance that hard tissue can be firmly rebuilt. Moreover, the roughness of the scaffold's surface is indispensable for efficient cells' attachment, both *in vitro* and *in vivo*. It strongly affects cellular morphology, proliferation, and phenotype expression [40].

In this study, the resulting organic–inorganic hybrid materials constitute a perfect platform in the formation of biomimicking scaffolds for hard tissue engineering. Porous structures were created using sugar templates. This easily-leachable and biocompatible porogen enables the formation of macropores of 150–350 μm in size that guarantee natural tissue ingrowth, mass transport, osteoinductive agent attachment, and ion exchange [41] (Fig. 3).

The total porous fraction for pure 1HCPP and 1HCPSP was approximately 90%, corresponding to an apparent density of $\sim 0.25 \pm 0.01 \text{ g cm}^{-3}$. In turn, the skeleton densities of the scaffolds were $2.488 \pm 0.009 \text{ g cm}^{-3}$ and $2.479 \pm 0.008 \text{ g cm}^{-3}$ for 1HCPP and 1HCPSP, respectively.

In addition to macroporosity, mesoporosity studies for both systems were also performed. Here, the Brunauer–Emmett–Teller surface area, average and modal pore diameters, and total pore volume were calculated. Surface areas were measured using the Brunauer–Emmett–Teller models for relative pressures of $0.05 < p/p_0 < 0.30$. The surface areas were of the order of $21 \text{ m}^2 \cdot \text{g}^{-1}$ for both 1HCPP and 1HCPSP. By comparison with, e.g., macroporous bioactive glasses [42], these surface areas are smaller, probably arising from the deeper packing inside the network. Surface area also affects nanopore sizes. Analysis of nanopore size distribution was based on the Horvath–Kawazoe model. Nanopores were present in a wide range of sizes from 1.6 to

2.1 nm for both materials. It should be noted, as a general comment, that *in vitro* lower porosity effectively stimulates osteogenesis by suppressing cell proliferation and forcing cell aggregation. Conversely, such a trend results in diminished mechanical properties, thereby setting an upper functional limit for pore size and porosity [40].

The roughness of the scaffolds' surface was achieved via the addition of ammonia carbonate to sugar porogen. During decomposition of carbonate salt at about 60 °C, gaseous carbon dioxide and ammonia are spontaneously liberated. Such vapor products play a dual role. While CO₂ serves as an agent that additionally opens non-interconnected macropores, the released NH₃ provides roughness of the surface. This is clearly shown in Fig. 4.

3.3. *Biom mineralization*

The resulting organic–inorganic hybrids were soaked in biological medium (DMEM, the composition of which is almost equal to body fluids,) to provoke or induce crystallization of inorganic salts on their surfaces. The concentrations of calcium, silicon, phosphate and strontium ions in biological fluid were measured using inductively coupled plasma optical emission spectroscopy before (“pure” DMEM) and after (1HCPP and 1HCPSP) exposure after different time periods: – 0.5, 1, 6, and 12 h, and 1, 2, 3, 4, and 7 days. Initially, the concentrations of specific ions in DMEM were: Ca²⁺, 67.97 mg L⁻¹; Si⁴⁺, 0.11 mg L⁻¹; PO₄³⁻, 30.94 mg L⁻¹; Sr²⁺, 0.00 mg L⁻¹. In this study, biological medium was used because it additionally contains amino acids and vitamins and provides better biomimetic conditions. However, owing to the presence of various proteins in this

solution, lower rates for the material's dissolution and a subsequent delay in surface layer formation are expected [43].

The dissolution experiments revealed that Ca^{2+} cations were released from the material in a different way for each of the 1HCPP and 1HCPSP systems (Fig. 5). In the 1HCPP system, the concentration of Ca^{2+} decreased in the first 15 h. After this time, it returned to the initial value and remained constant for approximately 40 h. Next, the quantity of Ca^{2+} dramatically decreased to $\sim 35\text{--}40 \text{ mg L}^{-1}$ to achieve equilibrium. By contrast, in the 1HCPSP system, the concentration of calcium ions in the DMEM changed continuously in the first 3 days. The 1HCPSP system quickly released Ca^{2+} ions during the first 10 h of immersion, reaching $\sim 100 \text{ mg L}^{-1}$ and then remained constant.

Similar changes were observed with the concentration of silicon ions. Firstly, for 1HCPP system, the concentration of silicon strongly varied and then reached $\sim 70\text{--}75 \text{ mg L}^{-1}$ after 75 h. The 1HCPSP system released Si^{4+} ions more quickly; during the first 50 h, a sharp increase in their concentration is observed. After this time, an equilibrium state of $\sim 25 \text{ mg L}^{-1}$ was established.

The concentration of phosphate anions also fluctuated for the 1HCPP system. A state of equilibrium was achieved after 70 h and remained constant at $\sim 47 \text{ mg L}^{-1}$. In the second material, 1HCPSP, after a rapid rise, the concentration of phosphate anions achieved equilibrium after 50 h of soaking and remained within a limit of 75 mg L^{-1} .

The smallest changes occurred in the case of strontium ions. The quantity of Sr^{2+} released by 1HCPSP slightly increased as a function of time and equaled $\sim 32 \text{ mg L}^{-1}$ after 170 h.

The observed ion exchange suggests the formation of a novel layer of inorganic salts containing calcium, strontium, and phosphorous ions. The concentrations of these ions depend on the initial concentrations in the scaffolds. These values clearly correspond to the mechanism described by Hench et al. [44], which suggests that there is precipitation of calcium phosphates and their derivatives at the interface of the biomaterial and the biomimetic solution. It is worth noting that silicon ions are leached from the material from the beginning of immersion in the biological medium. This phenomenon is widely considered a positive aspect of silicon-based biomaterials because silicon is treated as a substance that provides essential nourishment for the growth and maintenance of healthy bone. It is a helpful agent in building and maintaining hard tissue in different ways, including bone mineralization, reducing the number of osteoclasts that cause destruction of healthy bone, and increasing the amount of osteoblasts that regenerate bone [45]. Furthermore, in the case of 1HCPP, the concentration of PO_4^{3-} ions in DMEM increases over time. This may suggest a gradual replacement of phosphate groups with other anions, such as carbonates. In the case of 1HCPSP, this situation is reversed, which suggests that it is more likely that phosphate salts are crystallizing on the material's surface. This was confirmed using FT-IR spectroscopy. These analyses also suggest that ion exchange in the case of strontium-doped material 1HCPSP has an effect, since an equilibrium in the concentrations of selected ions is achieved more quickly and fluctuations at the beginning of soaking are not as spectacular as those noticed for 1HCPP.

During biomineralization, changes in pH were also observed (Fig. 6). In general, pH for 1HCPP is slightly higher than for 1HCPSP. This closely corresponds with the amount

of PO_4^{3-} anions released into the solution. In the case of 1HCPP, the PO_4^{3-} concentration is higher. Phosphates undergo hydrolysis and this leads to a decrease in H^+ concentration and increasing pH values. This may also confirm that, in the case of 1HCPSP, the surface is more favorable to phosphate salts than carbonates. This phenomenon is also confirmed by X-ray powder diffraction analysis of the precipitate scraped away from the scaffolds after biomineralization, e.g. diffraction peaks assigned to a bone-like apatite are observed at about 31.9 and 46.6° .

3.4. Synthetic hydroxyapatite

Owing to its chemical similarity to the mineral component of the bone [38], hydroxyapatite is considered one of the most important materials in biomedical applications. Synthetic hydroxyapatite ($\text{Ca}_{10}(\text{PO}_4)_6(\text{OH})_2$) is a material that has been successfully used as a filler of small bone defects and coating material to improve the biocompatibility of various implants. The particular interest in hydroxyapatite is for use as a bone substitute in dental and orthopedic implants, but because of its poor mechanical properties it cannot be applied to highly loaded components. In such cases, hydroxyapatite can only be used as a coating to improve materials properties [46].

In this study, synthetic hydroxyapatite was used to cover the synthesized scaffolds. As shown in Fig. 7, synthetic hydroxyapatite ideally covers the scaffold's surface. It is worth noting that the macropores are not closed or blocked, so do not preventing mass transport inside the network.

3.5. Properties of the organic–inorganic hybrid system

These studies are preliminary in nature, to collect data concerning the properties of the resulting materials. Depending on the properties to be determined, samples were examined in the form of thin films, monoliths, or scaffolds. Only pure samples were used in these studies. No samples after biomineralization or coating with synthetic hydroxyapatite were analyzed.

Compressive strength and Young's modulus were studied for both 1HCPP and 1HCPS. Studies were performed for two different structures, monoliths and three-dimensional scaffolds. To obtain monoliths, composites were collected into polypropylene molds directly after synthesis and cured at 65 °C for 12 h. Scaffolds were obtained as described in Experimental Section. The results are shown in Table 2 and Fig. 8.

The mechanical properties of monoliths are similar to those observed for cortical bone, whereas scaffolds seems to be perfect candidates for sophisticated cancellous bone replacements [24, 38, 47]. The values of Young's modulus and compressive strength are better than those observed for tricalcium phosphate and its composites with 2-hydroxyethylmethacrylate, 3-methacryloxypropyltrimethoxysilane, and methacrylate [48-53]. It should be noted the target mechanical properties which have been explicitly stated in the literature, span several different approaches which have been taken with regard to the design for mechanical properties. Some of them have stated that bone scaffold properties should match those of natural bone [54]. Moreover, such scaffold must be sufficient and not collapse during handling and patient's normal activity [55].

The results shown in Table 3 demonstrate that the presence of Sr^{2+} -dopant in the organic–inorganic hybrid composition had some influence on the wettability of the resulting coatings. The systems with Sr^{2+} ions exhibited smaller contact angles than the systems without. It should be emphasized that in the case of cells and proteins, an adhesion contact angle of $\sim 65^\circ$ distinguishes between hydrophobic and hydrophilic surfaces [56]. Here, additional factors, such as roughness and composition of the resulting materials may have a dominant influence on cell behavior.

The synthesized hybrids can also be used as coatings on various bioinert materials, such as metallic or ceramic implants. With this use in mind, the scratch test of the resulting coatings was evaluated to determine their resistance to scratching [57–59]. We employed a calibrated set of drawing pencils ranging from 9B, as the softest grading, through B, HB, F and H, to the hardest grading, 9H. Surprisingly, the hardness of 1HCP and 1HCPS differs drastically. The surface preparation and coating process were the same in both cases. 1HCP seems much softer material than 1HCPS; their values in the pencil scratch hardness tests were '2B' and '8H', respectively. Studies were conducted with ten independent repeats and the results were recurrent. Unfortunately, at this stage of studies it is hard to irrefutably clarify the observed values or draw any appropriate conclusions.

3.6. Evaluation of cytotoxicity of hybrid biocomposites on human osteoblast cell line

The survival of NHOst cells was examined using the sulforhodamine B assay. NHOst cells were incubated for 72 h in the presence of the test scaffolds. The control consisted of NHOst cells without test materials.

Percentage survival was calculated using the formula:

$$\text{Survival [\%]} = \left[\frac{A_s - A_m}{A_c - A_m} \times 100 \right]$$

where A_s , A_m , and A_c are the absorbances of the experimental samples, the control medium, and the control cells, respectively.

This study evaluated the effect of six scaffolds of 1HCP (after 7 days of incubation in DMEM), 1HCP_H, 1HCPP, 1HCPS (after 7 days of incubation in DMEM), 1HCPS_H, and 1HCPSP on the survival of the NHOst cell line during 72 h of incubation (Fig. 9 and Table 4). Our results indicate that the 1HCP and 1HCPS scaffolds do not show significant adverse effects on the survival of NHOst cells. Conversely, the 1HCP_H and 1HCPS_H materials significantly reduced the rate of NHOst cell survival (*P < 0.05); the survival rates in these cases were 65 and 45%, respectively. The samples of 1HCPP and 1HCPSP that were prepared without the biomineralization step also show an inhibitory effect on NHOst cells, with a reduction of about 20% in the survival rate compared with that of the control; however, these results are not statistically significant.

Scaffold 1HCPS were selected for the morphological studies using SEM, because NHOst cells in the presence of 1HCPS scaffold demonstrates high survival rate. In the Figure 10, SEM images illustrate the cell morphology on the resulting material.

Prior to SEM, the cells were grown on hybrid scaffolds for 20 days, following three prior inoculations of cells in the test materials. SEM after 20 days of culturing the NHOst cells on 1HCPS sample confirmed that these cells adhered to the scaffold's surface (Fig. 10), which is not toxic to NHOst cells.

The sulforhodamine B assay results indicated that the 1HCP and 1HCPS scaffolds do not have a negative impact on the survival of the NHOst cell line. For this reason, they may be used in future experiments, as they do not have a toxic effect on these cells. It should be emphasized that SEM of the 1HCPS scaffold shows that not only are the cells able to survive, but they are also able to attach to and grow on this material.

4. Conclusion

In this work, new organic-inorganic hybrid scaffolds derived from TEVS and HEMA doped with Ca^{2+} , Sr^{2+} , and PO_4^{3-} ions have been successfully developed and comprehensive characterization of the hybrid scaffolds for regenerative medicine was performed. Obtained materials possess a relatively high compressive strength, as well as high porosity (the total porous fraction for pure 1HCPP and 1HCPS is approx. 90%) and interconnectivity (open macropores of 150-350 μm in size). The mechanical properties of monoliths are similar to those observed for cortical bone (Young's modulus: 21.3 ± 2.4 and 21.7 ± 2.3 GPa, and compressive strength: 201 ± 5 and 199 ± 5 MPa for 1HCPP and 1HCPS, respectively), whereas scaffolds constitute a candidates for cancellous bone replacements (Young's modulus: 0.31 ± 0.05 and 0.33 ± 0.04 GPa, and compressive strength: 2.97 ± 0.12 and 2.22 ± 0.08 MPa for 1HCPP and 1HCPS, respectively). The values of Young's modulus and compressive strength are better than those observed for tricalcium phosphate and its composites with 2-hydroxyethylmethacrylate, 3-methacryloxypropyltrimethoxysilane, and methacrylate. These materials possess a sufficient shelf life (~6 months), in the liquid form before curing, which is important for an easy and wide distribution. The incorporation of

Sr²⁺ ions into the hybrid scaffolds was non-cytotoxic to the osteoblast cells. Moreover, obtained materials create a favourable rough topography which is friendly for cellular attachment and growth on the material surface. For this reason, they may be used in future *in vivo* experiments. Moreover, it should be noted that formulations possessing this sort of matrix with mentioned above dopants have not been studied so far, both taking into consideration their chemical and mechanical properties, and also *in vitro* studies.

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Figures and Tables

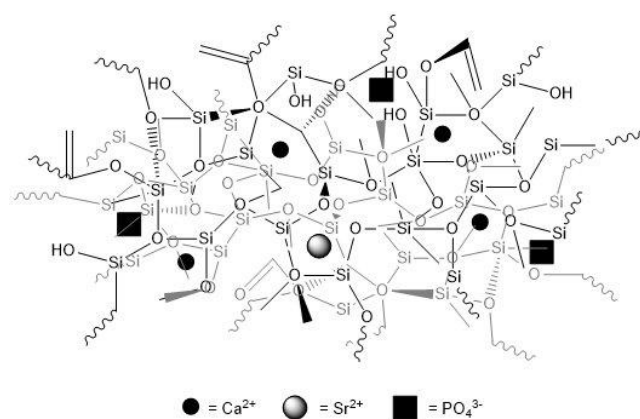


Figure 1. Ramified structure of the organic-inorganic hybrid network.



Figure 2. Homogeneous and transparent gels of resulting materials.

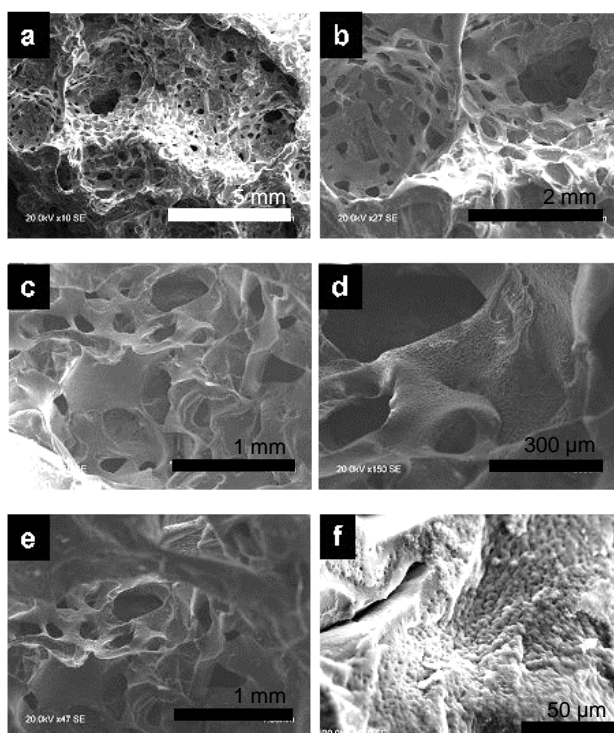


Figure 3. Resulting hybrid networks: (a–c) 1HCPP, (d–f) 1HCPSP.

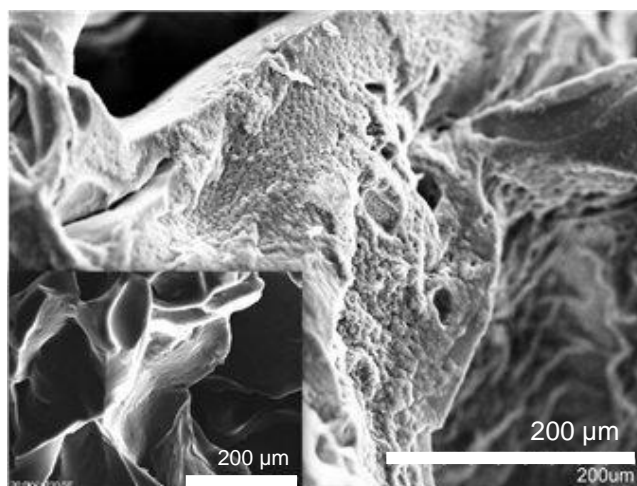


Figure 4. Addition of ammonia carbonate to sugar porogen forms rough scaffold surface. Inset: surface without the addition $(\text{NH}_4)_2\text{CO}_3$.

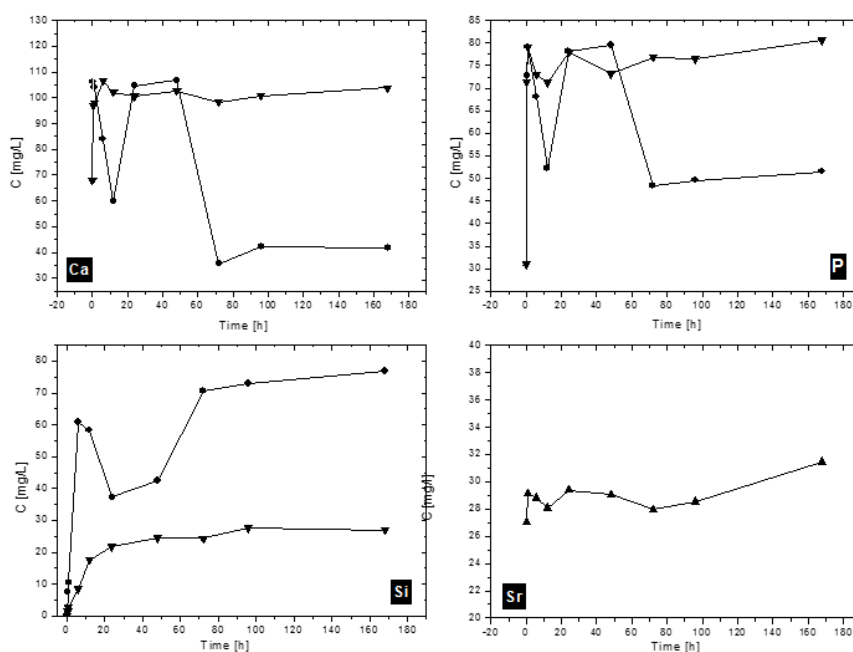


Figure 5. Inductively coupled plasma analysis of the evolution of Ca^{2+} , Si^{4+} , PO_4^{3-} and Sr^{2+} ions (mg L^{-1}) in DMEM. Dots, 1HCPP; triangles, 1HCPSP.

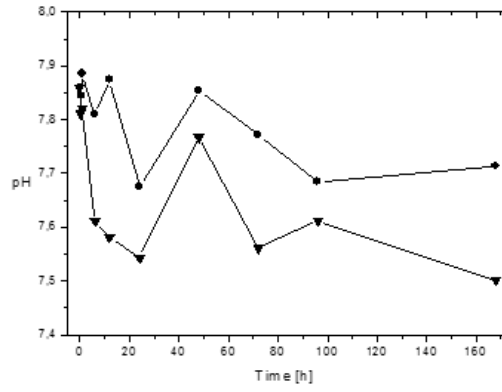


Figure 6. Changes in pH in DMEM during biomineralization. Dots, 1HCPP; triangles, 1HCPSP.

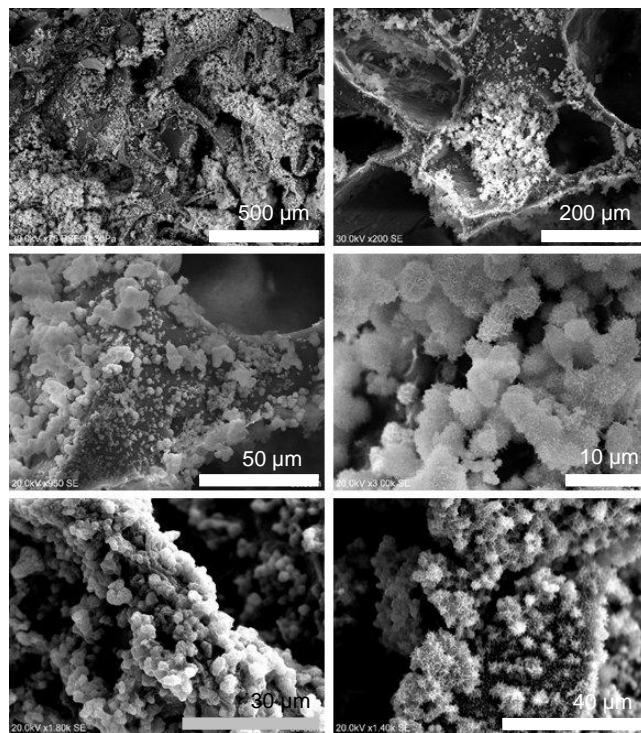


Figure 7. Scaffolds “decorated” by synthetic hydroxyapatite.

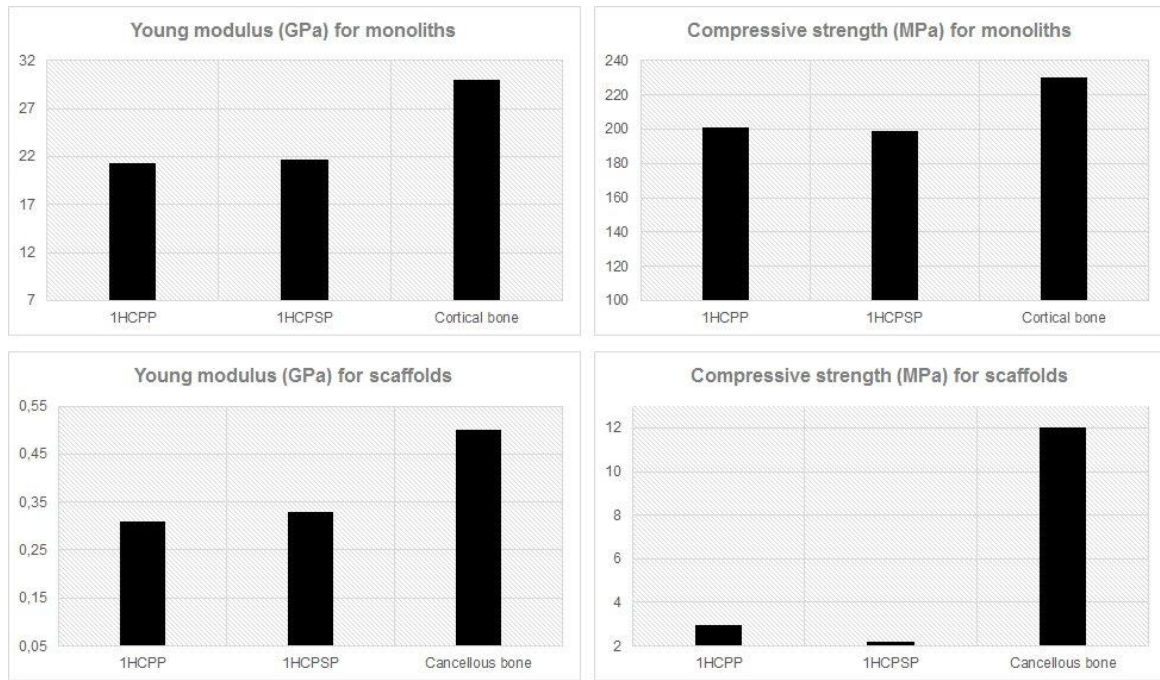


Figure 8. Mechanical properties of organic–inorganic hybrids, compared with cortical and cancellous bone.

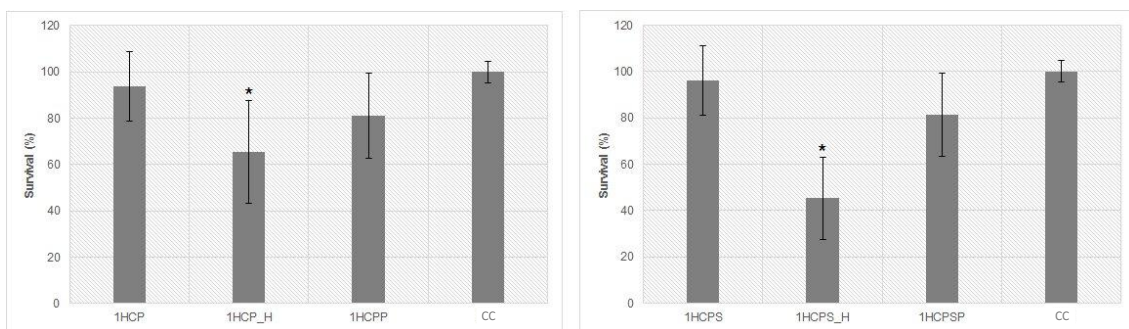


Figure 9. NHOst cell survival in the presence of organic–inorganic hybrid scaffolds after 72 h of incubation. The control consisted of NHOst cells without test scaffolds. *P < 0.05 vs. control cells (CC). Results for (left) 1HCP, 1HCP_H, and 1HCPP; (right) 1HCPS, 1HCPS_H, and 1HCPSP.

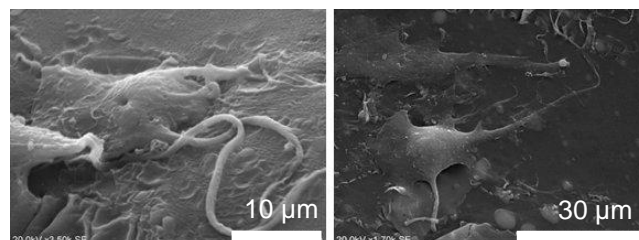


Figure 10. NHOst cells on a scaffold of 1HCPS after 20 days of culture. NHOst cells can be seen attached to the surface with numerous projections.

Table 1. Compositions of the organic-inorganic hybrid materials.

Samples	TEVS : HEMA ratio	Ca ²⁺ (%)	PO ₄ ³⁻ (%)	Sr ²⁺ (%)	Comments
1HCP	1 : 4	15.0	12.5	0.00	biomineralized in the biological fluid
1HCP_H					covered by the synthetic HAp
1HCPP					pure, without biomineralization
1HCPS		15.0	12.5	10.0	biomineralized in the biological fluid
1HCPS_H					covered by the synthetic HAp
1HCPSP					pure, without biomineralization

Table 2. Mechanical properties of organic–inorganic hybrids and some examples of selected systems.

Materials	Young's modulus (GPa)	Compressive strength (MPa)
1HCPP	21.3±2.4* / 0.31±0.05**	201±5* / 2.97±0.12**
1HCPSP	21.7±2.3* / 0.33±0.04**	199±5* / 2.22±0.08**
TCP [48,49]	0.025±0.003	0.59±0.16
TCP/HEMA/MPS [48,49]	1.21±0.33	57.4±27.5
TCP/MMA/MPS [48,49]	2.48±0.15	94.2±6.6
MMA/MPS [48-50]	2.03±0.09	110±2
Cortical bone [48-53]	7.00-30.0	100-230
Cancellous bone [48-53]	0.5-0.05	2-12

HEMA, 2-hydroxyethylmethacrylate; MMA, methacrylate; MPS, 3-methacryloxypropyltrimethoxysilane; TCP, tricalcium phosphate porous body; *monoliths; **scaffolds.

Table 3. Contact angle and surface free energy (Owens–Wendt method) for organic–inorganic hybrids.

materials	contact angle (°)		γ^d	γ^p	SFE
	H ₂ O	CH ₂ I ₂			
1THCP	69.5400	35.7700	0.1273	90.3916	90.5189
1THCPS	63.2100	34.8600	5.2626	69.4572	74.7198

Table 4. NHOst cell survival in the presence of 1HCP, 1HCP_H, 1HCPP, 1HCPS, 1HCPS_H, and 1HCPSP scaffolds after 72 h of incubation.

	1HCP	1HCP_H	1HCPP	1HCPS	1HCPS_H	1HCPSP	CC
Mean (%)	93.68	65.52	81.12	96.10	45.29	81.43	100.00
SD (%)	14.90	22.25	18.51	15.03	17.72	17.97	4.62