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Porous Mn-doped ZnO nanoparticles for enhanced solar and visible light photocatalysis



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ABSTRACT

Porous Mn-doped ZnO (ZnO:Mn) nanoparticles with an average diameter of ca. 21 nm were prepared by a simple and cheap solvothermal process involving no templates, post-synthetic annealing or etching. The particles produced were characterized by XRD, Raman spectroscopy, SEM, TEM, XPS, diffuse reflectance spectroscopy and BET surface area measurements and the effects of Mn²⁺-doping on the structural, optical and photocatalytic properties of ZnO particles were investigated. The particles doped with 3 mol% Mn²⁺ were found to exhibit the highest catalytic activity toward the photodegradation of the Orange II dye under solar light irradiation. Our results demonstrate that Mn²⁺-doping shifts the optical absorption to the visible region, increases the specific surface area of the photocatalyst and reduces the recombination of electron-hole pairs. The influence of various operational parameters (amount of catalyst, concentration of dye and pH) on the photocatalyst is stable and can be easily recycled up to ten times without any significant decrease in photocatalytic activity.

1. Introduction

The photocatalytic degradation of organic pollutants like dyes or pesticides from water using semiconductor materials has recently attracted a lot of attention [1–3]. Among photocatalysts, zinc oxide ZnO is a promising material either for the generation of hydrogen or for decomposition of organics owing to its large exciton binding energy (60 meV) at room temperature, high photosensitivity, low cost, high chemical stability and weak toxicity [4–7]. However, the wide bandgap of ZnO (3.37 eV) limits its response to the UV light only, leaving 95–97% energy of the whole solar spectrum unusable. Moreover, due to the very fast recombination of photogenerated electron (e^-)/hole (h^+) pairs, surface reactions that generate the reactive oxygen species (ROS) like hydroxyl •OH or superoxide $O_2^{\bullet-}$ radicals and H_2O_2 responsible for the catalytic photodegradation cannot optimally occur.

In recent years, a variety of effective approaches like semiconductor coupling [6,8,9] or doping [10–13] have been developed to modify ZnO in order to improve its photoresponse capability. Doping, which consists in the intentional incorporation of impurities into the host lattices, is the most commonly used method to tune the optical and chemical

* Corresponding author. *E-mail address*: raphael.schneider@univ-lorraine.fr (R. Schneider). properties of nanomaterials [14,15]. Because Mn²⁺ doping generates new energy states within the bandgap of ZnO [16–22], Mn-doped ZnO (ZnO:Mn) particles were recently found to be efficient for solar or visible-light driven photocatalysis for two reasons. First, because this doping enhances the absorption in the visible region. Second, because the new energy states act as intermediate steps for e⁻ in their transitions between the valence and the conduction bands and thus efficiently promote the separation of photogenerated e⁻ and h⁺. All studies clearly demonstrated that ZnO:Mn particles exhibit a higher photocatalytic activity than ZnO under UV or visible light irradiation and that the morphology, the size, the defect concentrations and the level of doping play crucial roles on the photodegradation efficiency of various organic dyes like methyl orange or methylene blue [16–22].

Porous ZnO materials have also gained much attention recently owing to their superior photocatalytic properties compared to ZnO particles [23–27]. Since the contaminant molecules need to be adsorbed on the photocatalyst surface for the redox reactions to occur, a high effective surface area associated to the high diffusivity of contaminants will give enhanced photocatalytic activity [28]. Up to now, several methods such as hydrolysis of Zn(OAc)₂ followed by high temperature annealing [29,30], hydrothermal synthesis followed by etching [31], amino acidassisted synthesis [32] or hydrolysis of Zn(OAc)₂ in the presence of templates (starch, polystyrene, gelatin, ...) followed by calcination [33–37] have been developed to obtain porous ZnO particles. However, most of these preparation methods are often faced with problems such as high temperature or tedious procedures. In this paper, we report a facile method for the preparation of porous Mn-doped ZnO particles via a solvothermal process. The high photocatalytic activity of ZnO:Mn particles for the degradation of Orange II under solar light irradiation was further demonstrated.

2. Experimental section

2.1. Materials

 $Zn(OAc)_2 \cdot 2H_2O$ (>98%, Sigma), Mn(OAc)_2 \cdot 4H_2O (99.99%, Sigma), Orange II sodium salt (>85%, Sigma), sodium hydroxide (>97%, Sigma), and anhydrous ethanol were used as received without further purification. All solutions were prepared using Milli-Q water (18.2 M $\Omega \cdot$ cm, Millipore) as solvent.

2.2. Preparation of ZnO and Mn-doped ZnO particles

ZnO particles were synthesized by a solvothermal method based on the hydrolysis of $Zn(OAc)_2$. Typically, in a three-necked flask equipped with a condenser and a dropping funnel, $Zn(OAc)_2 \cdot 2 H_2O$ (511 mg, 2.33 mmol) was dissolved in 35 mL ethanol. To this solution, NaOH (96 mg, 2.33 mmol) in 35 mL ethanol was added dropwise and the mixture stirred for 30 min at room temperature. Then, the solution was transferred into a 140 mL Teflon-sealed autoclave and was heated at 160 °C for 24 h. After cooling to room temperature, the ZnO particles were collected by centrifugation, washed three times with water, one time with ethanol, and dried at 70 °C overnight.

Mn-doped ZnO particles were prepared using a similar synthetic procedure. For the particles doped with 3% Mn, $Zn(OAc)_2 \cdot 2 H_2O$ (496 mg, 2.259 mmol) and $Mn(OAc)_2 \cdot 4H_2O$ (17 mg, 0.069 mmol) were used. The purification and drying procedures are similar to those previously described for ZnO particles.

2.3. Photocatalytic degradation of Orange II

The photocatalytic activity was evaluated by the degradation of an aqueous solution of Orange II (10 mg/L) at room temperature under solar light irradiation. In a typical experiment, the ZnO:Mn nanoparticles (60 mg) were dispersed in 30 mL Orange II aqueous solution (10 mg/L) and the suspension was magnetically stirred under ambient conditions for 60 min in the dark to reach an adsorption-desorption equilibrium. Under stirring, the suspension was exposed to simulated solar light irradiation produced by Sylvania LuxLine FHO T5 neon tubes. The light intensity was estimated to be 5.5 mW/cm². At certain time intervals, 1 mL of the suspension was extracted and centrifuged (15,000 rpm for 2 min) to remove the photocatalyst. The degradation process was monitored by measuring the absorption of Orange II at 485 nm using a UV-visible absorption spectrometer. The visible light irradiation was carried out using the neon tubes previously described and a polycarbonate film was used as the UV cutoff filter.

2.4. Quantification of hydroxyl radical production. DST assays

The production of •OH radicals by ZnO and ZnO:Mn particles was estimated using disodium terephthalate (DST), which turns into fluorescent 2-hydroxyterephthalate, 2-OH-DST ($\lambda_{em} = 428$ nm) upon reaction with •OH radicals [38–40]. Briefly, 5 mg of ZnO or ZnO:Mn particles were dispersed by magnetic stirring in 100 mL water. Next, 1 mL of this dispersion was mixed with 1 mL of DST (0.1 M in water) before being irradiated with a Hg–Xe lamp (light intensity = 200 mW \cdot cm⁻²) for various times. The mixture was then treated with 1 mL of 1 M NaOH and incubated for 50 min at room temperature before filtration on 0.2 µm polyvinylidene fluoride Acrodisc Syringe Filter. The

photoluminescence spectra were recorded to estimate 2-OH-DST formed ($\lambda_{ex} = 300$ nm). Control samples were (i) DST irradiated in the absence of particles and (ii) DST and particles but without light activation. PL intensities measured after these control experiments were subtracted from those measured when the particles were irradiated in the presence of DST.

2.5. Characterization

Transmission electron microscopy (TEM) images were taken by placing a drop of the particles dispersed in methanol onto a carbon film-supported copper grid. Samples were studied using a Philips CM200 instrument operating at 200 kV. Scanning electron microscopy (SEM) pictures were prepared using JEOL Scanning Electron Microscope JSM-6490 LV. The X-ray powder diffraction (XRD) diagrams of all samples were measured using Panalytical X'Pert Pro MPD diffractometer using Cu K α radiation. The X-ray powder diffraction data were collected from an X'Pert MPD diffractometer (PANalytical AXS) with a goniometer radius 240 mm, fixed divergence slit module $(1/2^{\circ})$ divergence slit, 0.04 rd Sollers slits) and an X'Celerator as a detector. The powder samples were placed on a silicon zero-background sample holder and the XRD patterns were recorded at room temperature using Cu K_{α} radiation ($\lambda = 0.15418$ nm). X-ray photoelectron spectroscopy (XPS) analyses were performed on a Gammadata Scienta (Uppsala, Sweden) SES 200-2 spectrometer under ultra-high vacuum ($P < 10^{-9}$ mbar). The measurements were performed at normal incidence (the sample plane is perpendicular to the emission angle). The spectrometer resolution at the Fermi level is about 0.4 eV. The depth analyzed extends up to about 8 nm. The monochromatized AlK α source (1486.6 eV) was operated at a power of 420 W (30 mA and 14 kV) and the spectra were acquired at a take-off angle of 90° (angle between the sample surface and photoemission direction). During acquisition, the pass energy was set to 500 eV for wide scans and to 100 eV for high-resolution spectra. CASAXPS software (Casa Software Ltd, Teignmouth, UK, www. casaxps.com) was used for all peak fitting procedures and the areas of each component were modified according to classical Scofield sensitivity factors.

The textural properties of the materials were investigated with a Micromeritics 3Flex Surface Characterization Analyzer instrument using liquid nitrogen (-196 °C). Prior to the analyses, the samples were out-gassed overnight under primary vacuum at 40 °C on the ports of the Micromeritics VacPrep 061 degasser followed by 4 h out-gassing under high vacuum on the analyse ports. The resulting isotherms were analysed using the BET (Brunauer-Emmett-Teller) method.

All the optical measurements were performed at room temperature $(20 \pm 1 \,^{\circ}\text{C})$ under ambient conditions. Absorption spectra of liquid samples were recorded on a Thermo Scientific Evolution 220 UV–visible spectrophotometer. The diffuse reflectance absorption spectra (DRS) were recorded on a Shimadzu 2600 UV–visible spectrophotometer. BaSO₄ powder was used as a standard for baseline measurements and spectra were recorded in a range of 250–1400 nm. Raman spectra were recorded using a Xplora spectrometer from Horiba Scientific with 532 nm wavelength incident YAG laser light.

3. Results and discussion

3.1. Synthesis and characterization of photocatalysts

Fig. 1 shows the powder XRD patterns of ZnO particles when varying the molar dopant percentage in Mn^{2+} from 0 to 7. All diffraction peaks could be indexed to the standard hexagonal wurtzite structure of ZnO (space group $P6_3mc$, JCPDS No 36-1451, a = 3.250 Å, c = 5.207 Å). The diffraction peaks were sharp, indicating the high crystallinity of the materials produced. Although the ionic radius of Mn^{2+} (0.80 Å) is larger than that of Zn^{2+} (0.74 Å), no significant distortion in the lattice structure was observed. No impurity phase attributed to manganese



Fig. 1. XRD patterns of ZnO and Mn-doped ZnO particles.

oxides could be detected, indicating that ${\rm Mn}^{2+}$ was completely doped into the ZnO crystal lattice.

The wurtzite crystal structure of ZnO particles was further confirmed by Raman spectroscopy (Fig. 2). The peaks located at 330, 378, 410 and 436 cm⁻¹ can be assigned to E_2 , A_{170} , E_1 , and E_2 (*high*) vibration modes of ZnO with $P6_3mc$ symmetry, respectively [41]. Well-resolved peaks originating from multiphonon processes could also be observed at 206 and between 1030 and 1200 cm⁻¹. XRD and Raman analyses demonstrate that ZnO:Mn particles produced are composed of hexagonal ZnO with good crystal quality.

Fig. S1 shows SEM images of ZnO nanoparticles and of the ZnO sample doped with 3% Mn²⁺. Both samples assemble into microparticles of globular morphology that are more homogeneous in size for the doped nanoparticles. Numerous pores can be seen in the ZnO and ZnO:Mn materials which should be beneficial to increase the contact area for reactant diffusivity and for enhanced photocatalytic activity [42]. The energy dispersive X-ray (EDX) spectrum of the ZnO:Mn (3%) sample only show peaks corresponding to Zn, O, and Mn and no trace amount of impurities could be seen in the detection limit of the EDX analysis (Fig. S2). Quantitative analysis shows that the concentration of Mn^{2+} in the 3% atomic Mn-doped ZnO particles is 2.2%, indicating that the amount of Mn incorporated into the ZnO lattice is slightly lower than the nominal amount of Mn used during the synthesis. To obtain more details about the structures of these materials, TEM and HR-TEM experiments were conducted (Fig. 3). The ZnO sample is mainly composed of spherical/ellipsoidal nanoparticles with an average diameter of 28 nm (Fig. 3a). A few particles with hexagonal morphology can also be observed. A decrease in nanoparticles size to ca. 21 nm was observed for



Fig. 2. Room temperature Raman spectra of ZnO and Mn-doped ZnO nanoparticles.



Fig. 3. TEM images of (a) ZnO and (b) ZnO:Mn (3%) nanoparticles (the inserts are the SAED patterns). (c) HR-TEM micrograph of ZnO:Mn (3%) particles.

ZnO:Mn (3%) particles (Fig. 3b). This decrease may be attributed to the growth inhibition of the nanocrystals originating from the presence of dopant ions in the reaction medium [43,44]. The smaller size of ZnO:Mn particles should be beneficial for photocatalytic activity. The selected area electron diffraction (SAED) patterns (insets of Fig. 3a-b) show a set of concentric reflexes, indicating the hexagonal structure of the ZnO particles prepared. This was further confirmed by a HR-TEM image of ZnO:Mn particles which shows clear lattice fringes with d-spacing of 0.55 nm, value consistent with the distance between two (001) crystal planes of hexagonal ZnO (Fig. 3c).

Further characterization studies using XPS provided evidences for the surface states and composition and the results obtained with ZnO doped with 3% Mn^{2+} are shown in Fig. 4. Fig. 4a shows the survey scan of the sample where the presence of Zn, O and Mn is evident. The spectrum also shows the signal of Na which originates from NaOH used for the synthesis. The fine XPS spectra show that the Zn 2p_{3/2} signal appears at 1021.46 eV, which corresponds to Zn–O bonds in the ZnO lattice (Fig. 4b). For Mn 2p_{3/2}, the energy of the signal located at 640.54 eV is close to Mn^{2+} in MnO (Mn 2p_{3/2}: 640.7 eV). The signals observed at 641.51 and 642.44 eV could be assigned to Mn_2O_3 and MnO_2 , respectively, indicating the presence of Mn oxides associated Mn-doped ZnO and/or the incorporation of Mn ions in the ZnO lattice in the +3 or +4 oxidation states. The latter hypothesis is supported by recent reports [45,46]. Finally,



Fig. 4. (a) XPS survey scan for ZnO:Mn (3%) particles. (b-d) High resolution peaks for Zn 2p, Mn 2p, and O 1s, respectively.

Fig. 4d shows the high resolution signal for the O 1s orbital. The broad peak is due to the surface-adsorbed OH^- ions on ZnO particles. The high intensity peak located at the lower binding energy side is associated to Zn–O or Mn–O bonds.

The optical properties of undoped and Mn-doped ZnO samples were further analyzed by UV–visible absorption and diffuse reflectance spectroscopy (DRS). ZnO shows a sharp near band edge absorption at ca. 380 nm due to its bandgap of 3.3 eV. After Mn²⁺ doping, the absorption markedly increases in the visible region (Fig. S3). ZnO particles exhibit the highest reflectance throughout the whole visible region while the reflectance decreases with the increase of Mn²⁺ doping in the visible region (Fig. 5), indicating that ZnO:Mn particles



Fig. 5. Room temperature UV–visible diffuse reflectance spectra of ZnO and ${\rm Mn}^{2+}\mbox{-doped}$ ZnO particles.

should be better photocatalysts under solar and visible light irradiation. The bandgap values were determined from DRS spectra by converting the absolute reflection to the Kubelka-Munk function (F_{KM}) and were found to gradually decrease from 3.06 for ZnO to 2.83 eV for ZnO:Mn (7%) (Fig. S4). This decrease originates from the formation of various optically active sub-levels through the bandgap of ZnO, resulting in a decrease in its optical bandgap. Finally, no fluorescence signal was observed for all Mn-doped samples after excitation at 300 nm. This indicates that the electron-hole recombination is completely inhibited through the Mn²⁺ doping and the presence of a large number of carrier traps within the nanoparticles.

3.2. Photocatalytic activity

The photocatalytic performances of ZnO:Mn particles have been evaluated for the degradation of Orange II (10 mg/L) and compared with ZnO particles under solar light irradiation (5.5 mW/cm²). In preliminary experiments conducted in the absence of the photocatalyst, no significant photolysis of the dye was observed during 6 h of irradiation. Once the catalyst added, the photodegradation was monitored by the time dependent UV-visible spectral changes of Orange II in the presence of ZnO and ZnO:Mn catalysts (Fig. 6). Using the ZnO:Mn (3%) catalyst, the characteristic peak at 485 nm gradually decreases with irradiation time and disappears completely after 210 min, suggesting the complete photodegradation (Fig. 6b). All other photocatalysts exhibited a weaker activity. The results described in Fig. 6a also demonstrate that when the doping in Mn²⁺ is higher that the optimal value (3%), Mn²⁺ ions probably act as electron and holes recombination centers and hence decrease the photocatalytic activity.

We also evaluated the photodegradation of Orange II under visible irradiation using a light intensity of 15 mW/cm² (Fig. S5). Results obtained showed that the mineralization of the dye could be achieved



Fig. 6. (a) Influence of the Mn doping of ZnO nanoparticles for the degradation of Orange II in aqueous solution (C is the Orange II concentration at time t, and C_0 is the concentration of the dye at t = 0; Volume of solution, 30 mL; Mass of photocatalyst, 60 mg; Orange II concentration, 10 mg/L). (b) Variation of Orange II concentration as a function of irradiation time using the ZnO:Mn (3%) photocatalyst.

(ca. 75% of degradation after 240 min) but required a longer time than under solar irradiation (near complete photodegradation after 210 min illumination).

3.3. Effect of catalyst dosage, of Orange II concentration and of pH

The effect of the catalyst amount on the photodegradation of Orange II under simulated solar light was studied by varying the ZnO:Mn (3%) catalyst concentration from 0.5 to 3.0 g/L (Fig. 7a). From 0.5 to 2.0 g/L of catalyst, the photodegradation rates increase with the catalyst amount. The initial rate constants given by the slope at the initial time of irradiation are 0.004, 0.005, 0.007, 0.017 and 0.006 min⁻¹ for 0.5, 1.0, 1.5, 2.0 g/L and 3 g/L of catalyst, respectively. Since the adsorption phase is not markedly influenced by the amount of catalyst used, the increase of the photodegradation rate when increasing the amount of catalyst probably originates from the increase of the surface illuminated during photocatalytic experiments. A decrease of the reaction rate was observed when using 3 g/L of the catalyst. This probably originates from light scattering by the ZnO:Mn particles, as previously observed for ZnO materials [47].

The effect of the initial Orange II concentration (5, 10, or 20 mg/L) on the photocatalytic efficiency was also investigated (Fig. 7b). As can be seen, the photodecomposition rate was found to decrease with the increase of the dye concentration (k = 0.025, 0.024, and 0.003 min⁻¹ for Orange II concentrations of 5, 10 and 20 mg/L, respectively). Since no marked differences were observed during the adsorption phase, the decrease in light penetration by Orange II molecules (filter effect) is probably the cause of this phenomenon.



Fig. 7. Influence of (a) the amount of catalyst, (b) the dye concentration and (c) the pH of the Orange II solution on the photocatalytic activity of ZnO:Mn (3%) particles.

The adsorption of the dye at the surface of the catalyst is well-known to depend on the pH of the solution used [48]. We varied the pH of the Orange II solution from 4.5 to 9.0 before the adsorption phase and the results are shown in Fig. 7c. No significant differences were observed on the adsorption of the dye. Results obtained show that the optimum pH for Orange II photodegradation is 9.0 but the photocatalytic activity is not markedly altered at pH 6.5 and 4.5. The later results are of interest since ZnO is known to exhibit a lower catalytic efficiency at acidic pH due to its slow dissolution at such pH values. The initial pseudo-first order rate constants of the ZnO:Mn photocatalyst at pH 4.5, 6.5, and 9.0 are 0.014, 0.017, and 0.019 min⁻¹, respectively. We attribute the high catalytic activity at pH = 9 to the excess of OH⁻ anions that facilitate the photogeneration of •OH, which are well-established as the primary oxidizing species during photocatalytic experiments.

3.4. Mechanism

It is well-known that the photocatalytic efficiency of semiconductor oxides like ZnO is mainly governed by surface area, defects and surface hydroxyl groups [4–7]. To gain some informations on the mechanism of the photocatalytic degradation, we first measured the specific surface areas and the microporosity of ZnO and of the 3% Mn²⁺-doped ZnO sample by nitrogen sorption performed at 77 K (Fig. 8). For both materials, the N₂-adsorption-desorption isotherms are of type II, according to the Brunauer-Deming-Deming-Teller (BDDT) classification [49]. A H3-type hysteresis loop can be observed at high relative pressures, characteristic of an adsorbent composed of aggregates, having a non-rigid texture, and indicating the existence of an undefined mesoporosity, which is in good agreement with SEM observations (Fig. S1). ZnO:Mn particles were found to exhibit a higher specific surface area than ZnO (41.46 and 54.03 m²/g for ZnO and ZnO:Mn, respectively). The pore size distributions were determined using the Barrett-Joyner-Halenda (BJH) method and are shown in Fig. S6. ZnO:Mn particles have a smaller pore size (23.2 nm) than ZnO particles (29.1 nm). These data suggest that the high specific surface area of ZnO:Mn particles may explain the enhancement of the solar-light driven photocatalytic oxidation of Orange II.

For the mineralization of organic dyes by photocatalysis, •OH radicals are generally considered to have a prominent role as the active species in photodegradation. We therefore compared the production of •OH radicals upon irradiation of ZnO and ZnO:Mn particles using a fluorescence technique. Hydroxyl radicals are well-known to react with disodium terephthalate (DST) to generate 2-OH-DST which strongly emits fluorescence centered at 428 nm upon excitation at 312 nm (Fig. 9a) [6,38–40]. As can be seen on Fig. 9b, a gradual increase



Fig. 8. N_2 adsorption-desorption isotherms at 77 K of ZnO and 3% $Mn^{2+}\mbox{-doped}$ ZnO particles.



Fig. 9. (a) 2-OH-DST production upon irradiation of DST in the presence of the ZnO photocatalyst. (b) Variations of the PL intensity at 428 nm of 2-OH-DST upon irradiation of DST with ZnO and ZnO:Mn (3%) nanoparticles. Results are the average of three experiments.

in fluorescence intensity at 428 nm was observed with increasing the irradiation time. The production of •OH radicals was markedly enhanced when irradiating ZnO:Mn particles. For example, a ca. 3.7-fold increased formation of •OH radicals was observed for ZnO:Mn after 15 min of irradiation compared to the experiment conducted with ZnO (Fig. 9b). Finally, it is worth to mention that 2-OH-DST was not produced in control experiments performed without light irradiation thus confirming a photo-induced mechanism.

The key role played by •OH radicals during the photocatalytic degradation of Orange II was further confirmed with the use of *tert*-butyl alcohol (*t*-BuOH), a chemical scavenger of these radicals [50]. A strong decrease of the photocatalytic degradation rate was observed in the presence of *t*-BuOH (Fig. S7). The previous results demonstrate that the higher specific surface of ZnO:Mn particles associated to the ability of Mn^{2+} ions to reduce charge recombinations and thus increase •OH radicals production are beneficial for enhancing the photocatalytic activity.



Fig. 10. Schematic illustration of the photocatalytic mechanism of ZnO:Mn nanoparticles under solar light irradiation.



Fig. 11. Plot of photodegradation percent of Orange II using the ZnO:Mn catalyst at optimized conditions versus cycle number.

The following mechanism can be proposed for the solar light-driven photodegradation of Orange II using ZnO:Mn particles (Fig. 10). Under solar light irradiation, the photogenerated e⁻ can be transferred either to the conduction band (CB) of ZnO or to Mn^{2+} energy levels localized within the bandgap of ZnO. Note that electrons promoted to the CB of ZnO can also be trapped by the Mn^{2+} dopant sites. This process inhibits e^-/h^+ recombinations and allows an enhanced production of ROS. Next, e^- react with adsorbed molecular oxygen to yield $O_2 \cdot -$ radicals. Meanwhile, h^+ react with surface-bound water to produce •OH radicals. $O_2 \cdot -$ and •OH free radicals, and the reactive species like $HO_2 \cdot$ and H_2O_2 obtained after association with H^+ , oxidize Orange II into carbon dioxide and water.

3.5. Stability

The stability and the reusability of the ZnO:Mn photocatalyst are important parameters in practical applications and were evaluated in ten successive cycles for the degradation of Orange II (10 mg/L) under solar light irradiation (5 mW/cm²). After each run, the photocatalyst was recovered by centrifugation (4000 rpm for 15 min) and redispersed in an Orange II solution without any washing or drying. As shown in Fig. 11, after ten cycles, the photocatalytic efficiency decreased only slightly compared to the as-synthesized ZnO:Mn particles, thus indicating that the photocatalyst exhibits high stability.

4. Conclusions

In summary, we successfully developed a new synthetic route giving access to porous ZnO and Mn-doped ZnO particles and which does not require post-synthetic annealing, etching or the use of templates. The influence of the Mn²⁺ dopant percentage on the structural, optical and photocatalytic properties of ZnO:Mn particles was investigated. ZnO:Mn (3%) exhibit the highest photocatalytic activity for the degradation of Orange II under solar light irradiation. This material exhibits also a higher specific surface compared to pure ZnO. This high specific surface associated to the ability of the Mn²⁺ dopant to act as an electron trap and to decrease the photogenerated electron/hole pair recombination results in an increase of the •OH radicals by the ZnO:Mn (3%) sample and thus of its photocatalytic performances. Moreover, the ZnO:Mn catalyst can be reused up to ten times with at least 85% of activity. The synthetic method developed gives an easy and fast access to porous Mn-doped ZnO particles displaying high interest for new technological applications and especially for environmental remediation.

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Appendix A. Supplementary data

Supplementary data to this article can be found online at http://dx. doi.org/10.1016/j.matdes.2016.04.015.

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