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# Effect of heterovalent substitutions in yttrium chromite on the hyperfine interactions of <sup>119</sup>Sn<sup>4+</sup> studied by Mössbauer spectroscopy

Pavel B. Fabritchnyi <sup>a</sup>, Mikhail I. Afanasov <sup>a</sup>, Evgeny M. Mezhuev <sup>a</sup>, Alain Wattiaux <sup>b</sup>, Mathieu Duttine <sup>b</sup>, Christine Labrugère <sup>c</sup>

**Abstract :** In order to develop the <sup>119</sup>Sn Mössbauer spectroscopic probe technique to study magnetically ordered materials, three Ca-substituted yttrium chromites, *i.e.* Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub>, Y<sub>0.9</sub>Ca<sub>0.1</sub>Cr<sub>0.9</sub>Ti<sub>0.1</sub>O<sub>3</sub> and Y<sub>0.8</sub>Ca<sub>0.2</sub>Cr<sub>0.8</sub>Ti<sub>0.2</sub>O<sub>3</sub>, doped with 0.3 atom-% Sn<sup>4+</sup>, were for the first time investigated. <sup>119</sup>Sn Mössbauer spectra, recorded at 4.2 K, have allowed, through analysis of the magnetic hyperfine field values, probed by <sup>119</sup>Sn nuclei, to gain insight into the local magnetically active surrounding of different Sn<sup>4+</sup> ions. In all of these compounds, partial segregation of Sn<sup>4+</sup> ions is revealed. In the case of Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub>, neither highly oxidized Cr<sup>4+</sup> nor Cr<sup>6+</sup> species, expected to compensate for the Ca<sup>2+</sup> positive charge deficit, is found in the vicinity of the <sup>119</sup>Sn<sup>4+</sup> probe. In the case of both studied Ti-containing chromites, <sup>119</sup>Sn Mössbauer spectra have provided the original indirect evidence for the statistical distribution of Cr<sup>3+</sup> and Ti<sup>4+</sup> ions over the octahedral sites and permitted characterization of the occurring associates of Sn<sup>4+</sup>.

## 1. Introduction

Mössbauer spectroscopy of 119Sn diamagnetic dopant cations, introduced as resonant probes into a magnetically ordered compound, can provide particularly valuable information about their magnetically active surrounding [1]. This becomes possible due to the spin polarization of the <sup>119</sup>Sn electronic shell by neighboring cations giving rise to a Zeeman splitting of 119Sn energy levels. Analysis of the observed hyperfine pattern thus permits to characterize changes in the local surrounding of the probe and relate them to the studied process. As an example, we can mention our results recently reported in [2] for tin-doped perovskites RECrO<sub>3</sub> (RE=rear earth element or Y) exhibiting a G-type AF arrangement [3]. In these compounds, <sup>119</sup>Sn<sup>4+</sup> probe ions were located on the Cr3+ site and their spin polarization was created by six Cr3+ neighbors with parallel spins. In order to develop the 119Sn Mössbauer spectroscopic probe technique to study such perovskite-type antiferromagnetics, we have attempted to investigate the mechanisms of heterovalent substitution in an isostructural compound RE<sub>0.8</sub>Ca<sub>0.2</sub>CrO<sub>3</sub> containing charge-deficient Ca<sup>2+</sup> ions within the dodecahedral substructure. According to previous

studies, the Ca2+ charge deficit compensation, under oxidative conditions, can involve the stabilization of both  ${\rm Cr}^{4+}$  (3 ${\it d}^2$ ) and  $Cr^{6+}$  (3 $d^{0}$ ) ions [4,5], whereas, under reducing conditions, the instability of these high oxidation states of chromium is expected to result in the loss of oxygen along with the concomitant creation of oxygen ion vacancies ( $V_0$ ) [6]. <sup>119</sup>Sn doping and Mössbauer spectroscopy should thus allow to discern different charge compensation species in the neighborhood of the resonant probe and, consequently, gain insight into the local composition of the studied compound. Mössbauer spectrum of 119Sn4+:Y0.8Ca0.2CrO3 sample, containing 0.3 atom-% Sn<sup>4+</sup> substituted for Cr<sup>3+</sup> [7], have revealed the presence of an additional magnetically split component which was not observed for <sup>119</sup>Sn<sup>4+</sup>:YCrO<sub>3</sub> [8]. The relevant value of the magnetic hyperfine field, probed by <sup>119</sup>Sn at 4.2 K, was found to be consistent with the lack of one of the six Cr3+ ions in the nearest magnetic surrounding of the concerned 119Sn4+ ion. Because of the much higher content of Ca<sup>2+</sup> substituent ions, as compared to that of Sn<sup>4+</sup>, this new spectral component could not be explained by the specific charge compensation mechanism involving the formation of a Cr3+ vacancy in the vicinity of a 119Sn4+ probe. On the contrary, the lack of one of its magnetic neighbors, a priori, might reflect the formation of diamagnetic Cr<sup>6+</sup> ions to electrically compensate the Ca<sup>2+</sup>-charge deficit. Nevertheless, taking into account other results obtained in [7], it was finally concluded that this spectral component could not be assigned to a

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 $\rm Sn^{4+}$ – $\rm Cr^{6+}$  pair, but rather, to an associate of two  $\rm Sn^{4+}$  dopant ions. Interestingly, analysis of the  $^{119}\rm Sn$  Mössbauer spectra of  $\rm Sn:Y_{0.8}\rm Ca_{0.2}\rm CrO_3$  did not reveal the presence of  $\rm Cr^{4+}$  in the  $\rm Sn^{4+}$  ion vicinity although the formation of  $\rm Cr^{4+}$  in  $\rm Y_{1-x}\rm Ca_x\rm CrO_3$  perovskites has been evidenced by other techniques [4]. In the present work, to obtain more information about the local surrounding of the  $\rm ^{119}\rm Sn$  probe, we have investigated three Ca-substituted yttrium chromites with a different  $\rm Ca/Y$  ratio or containing different amounts of  $\rm Ti^{4+}$  introduced to prevent the formation of chromium in higher than +3 oxidation states.

#### 2. Experimental

#### 2.1. Synthesis method

Polycrystalline  $Y_{1-x}Ca_xCrO_3$  or  $Y_{1-x}Ca_xCr_{1-x}Ti_xO_3$  samples (with x=0.1 or 0.2), all of them containing 0.3 atom-%  $^{119}Sn^{4+}$  on Cr3+ substitution sites in the bulk crystallites, were synthesized using a simplified citric acid method previously applied to prepare tin-free Y<sub>1-x</sub>Ca<sub>x</sub>CrO<sub>3</sub> powder samples [6]. Accordingly, proper amounts of yttrium and calcium carbonates were dissolved in 1 M HNO<sub>3</sub> and the obtained solution was poured into the mixture of required volumes of chromium (III) nitrate solution and tin (IV) chloride solution, enriched up to 92% in the 119Sn Mössbauer isotope. To obtain Y<sub>1-x</sub>Ca<sub>x</sub>Cr<sub>1-x</sub>Ti<sub>x</sub>O<sub>3</sub>, the required volume of acidified titanium (IV) chloride solution was additionally poured into the reaction mixture. The concentration of tin was  $0.01 \text{ mol } L^{-1}$ and that of either main constituent metal 0.5 mol L<sup>-1</sup>. Finally, citric acid monohydrate (used at a rate of  $[C_6H_8O_7 \cdot H_2O]$ :[Cr] = 3:1) was dissolved in the mixture of solutions heated to 80 °C. The gel, formed upon evaporation of the resulting solution at 80 °C, was heated in air for 3 h at 150 °C. The obtained porous solid substance was then heated at a rate of ca. 3 °C/min up to 1050 °C and kept at this temperature for 12 h. This operating procedure has allowed 119Sn<sup>4+</sup> probe ions to be located, as it was the case in unsubstituted YCrO<sub>3</sub> [8], on Cr sites in the bulk of both  $Y_{1-x}Ca_xCrO_3$ and  $Y_{1-x}Ca_xCr_{1-x}Ti_xO_3$  microcrystals.

#### 2.2. Characterization techniques

XRD measurements were performed with a powder sample ARL X'TRA Thermo Scientific Diffractometer using Cu  $K_{\alpha}$  radiation (wavelength  $\lambda$ =1.5418 Å).

Mössbauer spectroscopic measurements were carried out using a conventional spectrometer of the electromechanical type, operating in constant acceleration mode, with a  $Ca^{119m}SnO_3$  source kept at room temperature. During the measurements the studied sample (absorber) was placed in an Oxford Instrument cryostat cooled down to 4.2 K. All spectra were analyzed by a least-squares fitting program. The isomer shift values quoted below are referred to a BaSnO<sub>3</sub> absorber at 295 K. The experimental value of the full width at half maximum, obtained with an optically thin (5.6 mg/cm²) BaSnO<sub>3</sub> absorber,  $\Gamma_{ref.}$ =0.83  $\pm$ 0.04 mm s<sup>-1</sup>, was assumed to characterize, in all studied chromites, a single absorption peak in the absence of any magnetic interaction.

X-ray photoelectron spectroscopy (XPS) measurements were performed with a K-Alpha X-Ray photoelectron spectrometer (ThermoFisher Scientific Inc.) using monochromatic Al  $K_{\alpha}$  radiation ( $h\nu\!=\!1486.6$  eV), spot size: 200  $\mu\text{m}$ , and charge compensation. Overview spectra were taken at a pass energy  $E_p\!=\!200$  eV and high resolution spectra at  $E_p\!=\!40$  eV. Powder samples, analyzed in the as-prepared state, were pressed onto indium foil. High resolution spectra were fitted using the AVANTAGE software provided by ThermoFisher Scientific Inc.

#### 3. Results and discussion

The X-ray diffraction patterns of all samples investigated in the present work were consistent with a rhombohedrally distorted perovskite structure (space group *Pbnm*) reported in [6] for the  $Y_{1-x}Ca_xCrO_3$  solid solutions. As an example, Fig. 1 shows an XRD pattern of a tin-doped  $Y_{0.9}Ca_{0.1}CrO_3$  sample.

#### 3.1. $Y_{0.9}Ca_{0.1}CrO_3$

As evidenced by earlier measurements of the magnetic susceptibility [4], the antiferromagnetic ordering of  ${\rm Cr^{3+}}$  moments in  ${\rm Y_{0.9}Ca_{0.1}CrO_3}$ , as well as in the unsubstituted YCrO<sub>3</sub>, occurs in the vicinity of  $T_N$ =140 K. This means that the values of the hyperfine field  $H(^{119}{\rm Sn})$ , determined from  $^{119}{\rm Sn}$  Mössbauer spectra recorded at 4.2 K, will correspond, in both compounds, to saturation values of the Cr-substructure magnetization.

A <sup>119</sup>Sn Mössbauer spectrum of the Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub> sample is composed of a number of magnetically split components (Fig. 2a). The symmetrical shape of the spectrum is consistent with the value of the isomer shift ( $\delta$ =0.12  $\pm$  0.05 mm s<sup>-1</sup>) and the apparent quadrupole shift  $\varepsilon_Q$  (close to zero) which are the same for all components (Table 1). These hyperfine parameters are in good agreement with the tetravalent state of the tin atoms and data previously reported for <sup>119</sup>Sn<sup>4+</sup> in bulk YCrO<sub>3</sub> [8]. Fitting of the spectrum shows that it can be represented by a superposition of the following components:

- (1) A major sextet (denoted hereafter as ms), corresponding to the hyperfine field  $H_{ms}$ =80  $\pm$  1 kOe, with a relative area  $A_{ms}$ =0.78:
- (2) A less intense but particularly intriguing sextet (denoted as xs), corresponding to  $H_{xs}$ =67 ± 3 kOe, with  $A_{xs}$ =0.14; and
- (3) Several very weak unresolved sextets (denoted as vws), corresponding to H<sub>vws</sub> ≤ 54 kOe, with a total relative area A<sub>vws</sub> lower than 0.10.

The spectrum of  $Y_{0.9}Ca_{0.1}CrO_3$  is quite similar to that recently reported in [7] for  $^{119}Sn^{4+}$ : $Y_{0.8}Ca_{0.2}CrO_3$ . In fact, in both cases the value of  $H_{ms}$  is equal within experimental error to the value of  $H_6$  =80 ±3 kOe which was previously assigned to  $Sn^{4+}$  on a regular  $Cr^{3+}$  site, with 6  $Cr^{3+}$  nearest magnetic neighbors, in the bulk of the unsubstituted YCrO<sub>3</sub> [8].

Moreover, xs, which was not observed in the spectrum of

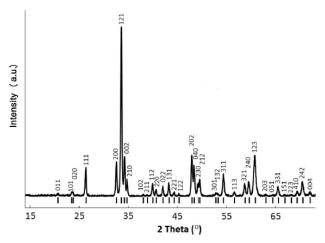


Fig. 1. XRD pattern of a tin-doped Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub> polycrystalline powder sample exhibiting a single-phase orthorhombic perovskite structure (space group *Pbnm* [6]).

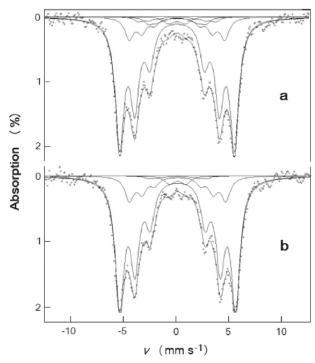


Fig. 2.  $^{119}Sn$  Mössbauer spectra of tin-doped  $Y_{0.9}Ca_{0.1}CrO_3$  recorded before (a) and after (b)  $H_2$  annealing the sample at 700  $^{\circ}C.$ 

**Table 1** Mössbauer spectroscopic parameters of the  $^{119}\text{Sn}^{4+}$  probe in the bulk of  $Y_{0.9}\text{Ca}_{0.1}\text{CrO}_3$  at 4.2 K.

		$\delta$ (mm s <sup>-1</sup> )	H (kOe)	$\varepsilon_{\rm Q}  ({\rm mm  s^{-1}})$	$\Gamma$ (mm s <sup>-1</sup> )	Α
As synthesized						
ms:		$0.12 \pm 0.02$	$80 \pm 1$	~0	$1.00 \pm 0.05$	0.78
xs:		$0.12 \pm 0.04$	$67 \pm 3$	-	1.00*	0.14
vws:	(1)	$0.12 \pm 0.05$	$54 \pm 5$	_	1.00*	0.03
	(2)	$0.12 \pm 0.05$	$40 \pm 5$	-	1.00*	0.03
	(3)	$0.12 \pm 0.05$	$13 \pm 5$	-	1.00*	0.02
ms:	bsequent	$H_2$ annealing a 0.12 $\pm$ 0.02 0.12 $+$ 0.04	at 700 °C 80 ± 1 67 + 3	~0	1.00 ± 0.05	0.80 0.15
xs:	(1)	$0.12 \pm 0.04$ 0.12 + 0.05	_	_	1.00*	0.00
vws:	(1) (2)	$0.12 \pm 0.05$ $0.12 \pm 0.05$	$54 \pm 5$ $40 \pm 5$	-	1.00*	0.00
	(3)	$0.12 \pm 0.05$	$13 \pm 5$	-	1.00*	0.03

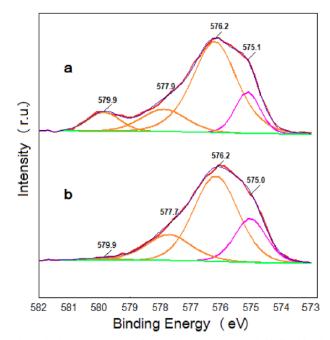
The quoted fixed (\*) values of  $\Gamma$ , somewhat higher than that obtained with our reference BaSnO<sub>3</sub> absorber ( $\Gamma_{ref}$ =0.83 mm s<sup>-1</sup>), are assumed to reflect the relatively weak fluctuations in H values probed by different <sup>119</sup>Sn<sup>4+</sup> ions with equal numbers of magnetic neighbors (formally considered as possessing "magnetically equivalent surroundings").

YCrO<sub>3</sub>, exhibits relative areas of comparable magnitude in  $Y_{0.9}Ca_{0.1}CrO_3$  and  $Y_{0.8}Ca_{0.2}CrO_3$  ( $A_{xs}$ =0.14 and 0.19 [7], respectively).

Finally, as to the above–mentioned  $\emph{vws}$ , their total contribution in  $Y_{0.9}Ca_{0.1}CrO_3$  is also close to that in either  $Y_{0.8}Ca_{0.2}CrO_3$  or  $YCrO_3$ .

Analysis of Fig. 2a thus leads to the following conclusion: whereas xs is undoubtedly the only spectral component which can be thought of as a Mössbauer spectroscopic signature of the Ca<sup>2+</sup> substitution for Y<sup>3+</sup>, no obvious relation seems to appear between the atomic ratio Ca/Y and the relative area  $A_{xs}$ .

XPS measurements on the considered  $Y_{0.9}Ca_{0.1}CrO_3$  powder have yielded, for yttrium and calcium, the binding energies  $E_b$ 



**Fig. 3.** Fitted  $Cr2p_{3/2}$  XP spectra of tin-doped  $Y_{0.9}Ca_{0.1}CrO_3$  recorded before (a) and after (b)  $H_2$  annealing the sample at 700 °C.  $Cr^{6+}$  species are represented by the component centered at  $E_b = 579.9$  eV.

(Y3d,  $E_b$ =156.8 eV and Ca2p,  $E_b$ =346.8 eV) which were typically observed in Y and Ca oxides [9]. Hence, no specific information about charge compensation mechanisms in Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub> could be extracted from XP spectra of these two constituent elements. By contrast, analysis of the composite Cr2 $p_{3/2}$  peak ( $E_b$ =572–582 eV), appearing in the XP spectrum of this material, predominantly composed of trivalent chromium, has revealed the concomitant presence of Cr<sup>6+</sup> (3 $d^0$ ) diamagnetic surface species characterized by a single spectral component with a binding energy of 579.9 eV (Fig. 3a). As to the anticipated presence of Cr<sup>4+</sup> (3 $d^2$ ) species, possessing, like Cr<sup>3+</sup> (3 $d^3$ ) ions, unpaired electrons (that give rise, in both cases, to a complex unresolved multiplet splitting of the Cr2 $p_{3/2}$  peak [10]), these species cannot be, for this reason, reliably detected by XPS in the considered compound.

Nevertheless, we could prove through complimentary <sup>119</sup>Sn Mössbauer spectroscopic measurements, that neither  $Cr^{6+}$  nor  $Cr^{4+}$  species were responsible for xs. In fact, this sextet was still present in the <sup>119</sup>Sn Mössbauer spectrum (Fig. 2b) recorded after  $H_2$  annealing the sample at 700 °C, this chemical treatment being known to easily transform  $Cr^{6+}$  (as well as  $Cr^{4+}$ ) into the most stable  $Cr^{3+}$ , as confirmed by the observed drastic decrease of the contribution at 579.9 eV in the  $Cr2p_{3/2}$  XP spectrum (Fig. 3b).

Accordingly, these results lead us to believe that the most plausible origin of  $\pmb{xs}$  is the formation of an associate involving two neighboring  $Sn^{4+}$  dopant ions, as it was earlier suggested in the case of tin-doped  $Y_{0.8}Ca_{0.2}CrO_3$  [7].

In this context, it is noteworthy that the  $^{119}$ Sn Mössbauer spectrum of the pristine  $Y_{0.9}$ Ca<sub>0.1</sub>CrO<sub>3</sub>, which was not annealed in the reducing atmosphere (*cf.* Fig. 2a), has nevertheless provided no evidence for Cr<sup>4+</sup> ions in the vicinity of the  $^{119}$ Sn probe, either. In order to account for this finding we assume that the hole hopping between Cr<sup>4+</sup> and Cr<sup>3+</sup>, responsible for the electrical properties of  $Y_{1-x}$ Ca<sub>x</sub>CrO<sub>3</sub> observed in the temperature range 300–550 K [4], could be completely stopped upon Mössbauer spectroscopic measurements at 4.2 K, the hole being frozen on a Cr<sup>3+</sup> ion, containing no Sn<sup>4+</sup> dopant in its neighborhood. Accordingly, this allows the highly charged Cr<sup>4+</sup> and Sn<sup>4+</sup> ions to avoid location in

close proximity to one another. In other words, at 4.2 K, the magnetically active surrounding of  $^{119}{\rm Sn^{4+}},$  incorporated into  ${\rm Y_{1-x}Ca_xCrO_3},$  involves chromium only in its trivalent state, as it was the case in YCrO\_3.

Indeed, quite a different situation is expected when a chemically stable  $M^{4+}$  dopant, e.g.  $\mathrm{Ti}^{4+}$ , was used to compensate the unbalance of charge created in the considered chromites by heterovalent  $\mathrm{Ca}^{2+}$  ions. In order to elucidate this point we have investigated the spectra of the  $^{119}\mathrm{Sn}^{4+}$  probe introduced in the bulk of two relevant compounds with different titanium substitution levels.

#### 3.2. The solid solution $Y_{1-x}Ca_xCr_{1-x}Ti_xO_3$

XRD measurements, performed for both Ti4+-containing materials discussed in the following two sections, proved that they contained only one phase consistent with the formation of the relevant solid solution exhibiting a distorted perovskite-type structure. Moreover, a slight shift of all the Bragg peaks to lower values of the diffraction angle was observed, in agreement with the larger ionic radius of Ti<sup>4+</sup> (0.605 Å) with respect to that of  ${\rm Cr^{4+}}$  (0.55 Å), which is believed to compensate for the charge deficit generated by  $Ca^{2+}$  in  $Y_{1-x}Ca_xCr_{1-x}Ti_xO_3$ . However, our <sup>119</sup>Sn Mössbauer spectroscopic measurements, performed with the tindoped samples of YCrO<sub>3</sub>, Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub>, Y<sub>0.9</sub>Ca<sub>0.1</sub>Cr<sub>0.9</sub>Ti<sub>0.1</sub>O<sub>3</sub>, and  $Y_{0.8}Ca_{0.2}Cr_{0.8}Ti_{0.2}O_3$ , have shown the value of  $H(^{119}Sn)$  for the regular Cr3+ site with 6 Cr3+ nearest magnetic neighbors, appearing clearly in the spectra all of these isostructural materials, to be the same within measurement error ( $H=80\pm3$  kOe). This means that the spin polarization of the <sup>119</sup>Sn<sup>4+</sup> ions is virtually unaffected by the crystallographic details which may therefore be neglected for following consideration.

#### 3.2.1. $Y_{0.9}Ca_{0.1}Cr_{0.9}Ti_{0.1}O_3$

XPS revealed the presence of  $Ti^{4+}$  ions through the Ti2p3 peak (binding energy  $E_b$ =458.4 eV) which is commonly observed for  $Ti^{4+}$ -based oxides [9]. The presence of  $Ti^{4+}$  ions did not suppress the formation of  $Cr^{6+}$  at the surface (Fig. 4a). The relative amount (ca. 6%) of these surface species, characterized by the 579.9 eV peak, remained comparable to that in the as-prepared  $Y_{0.9}Ca_{0.1}CrO_3$  (ca. 9%). This result was confirmed by further XPS characterization of a sample with a higher  $Ti^{4+}$  content (cf. Section 3.2.2). We assume that the hexavalent chromium has been essentially formed, in both cases, by the reaction between surface-located  $Cr^{3+}$  ions and  $O_2$  molecules upon air annealing the samples. Accordingly, the  $Cr^{6+}$  content could only be slightly influenced by the different  $Ca^{2+}$ -charge compensation mechanisms in  $Y_{0.9}Ca_{0.1}Cr_{0.9}Ti_{0.1}O_3$  and  $Y_{0.9}Ca_{0.1}CrO_3$ . As expected,  $H_2$  annealing the former chromite has also resulted in a drastic decrease in the  $Cr^{6+}$  spectral contribution (Fig. 4b).

As seen in Fig. 5, the <sup>119</sup>Sn Mössbauer spectrum of tin-doped Y<sub>0.9</sub>Ca<sub>0.1</sub>Cr<sub>0.9</sub>Ti<sub>0.1</sub>O<sub>3</sub> differs considerably from that of Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub> (cf. Fig. 2a). In fact, it is clearly indicative of both the sharp decrease in the number of Sn sites, characterized by  $H=81\pm1$  kOe, and the concomitant increase in the spectral contribution of all sextets related to lower values of H. This spectrum is thus qualitatively consistent with the expected fluctuation in the number of diamagnetic Ti<sup>4+</sup> ions in the neighborhood of various Sn<sup>4+</sup> ions distributed throughout the relevant solid solution. However, one can also see that most of the Sn4+ ions still exhibit the value of H=81 kOe which is close to those observed in the above three titanium-free chromites. Besides, the values of the isomer shift and the apparent quadrupole shift for all the 119Sn constituent sextets of the spectrum in Fig. 5 are found to be equal, within experimental error, to those for Sn<sup>4+</sup> in the bulk of the titaniumfree chromites (cf. Table 1).

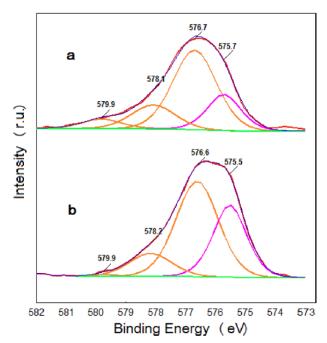


Fig. 4. Fitted  $Cr2p_{3/2}$  XP spectra of tin-doped  $Y_{0.9}Ca_{0.1}Cr_{0.9}Ti_{0.1}O_3$  recorded before (a) and after (b)  $H_2$  annealing the sample at 700 °C.  $Cr^{6+}$  species are represented by the component centered at  $E_b = 579.9$  eV.

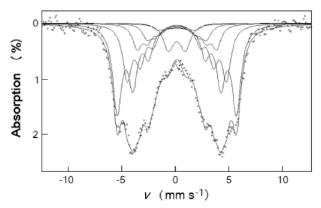


Fig. 5. <sup>119</sup>Sn Mössbauer spectrum of tin-doped  $Y_{0.9}Ca_{0.1}Cr_{0.9}Ti_{0.1}O_3$ . For all the constituent sextets the values of  $\delta$  and  $\varepsilon_Q$  are assumed to be equal to those observed in the case of tin-doped  $Y_{0.9}Ca_{0.1}CrO_3$  ( $\delta$ =0.12  $\pm$  0.05 mm s<sup>-1</sup> and  $\varepsilon_Q$   $\cong$  0 mm s<sup>-1</sup>).

In an attempt to interpret the spectrum in Fig. 5, we have applied the Gilleo's model of randomly incomplete magnetic linkages [11]. This model has been successfully used by Lyubutin and Vishnyakov [12] to account for the evolution of <sup>119</sup>Sn Mössbauer spectra of  $RE_{1-x}Ca_xFe_{1-x}Sn_xO_3$  perovskites with increasing x value. According to [12], in these compounds, containing  $Sn^{4+}$  ions statistically distributed over the octahedral B-sites, the probability P(k) to find k magnetic (Fe<sup>3+</sup>) cations among six nearest neighboring B-cations, forming the magnetically active surrounding of the reference B-cation, can be defined as

$$P(k) = C_6^k(x)^{6-k}(1-x)^k \tag{1}$$

where  $C_6^k = \frac{6!}{(6-k)!k!}$  is the binominal coefficient with k=0, 1, 2,...6.

*A priori*, the same relation can also be applied to the material, we are interest in,  $Y_{0.9}Ca_{0.1}Cr_{0.9}Ti_{0.1}O_3$  (x=0.1), assuming that the <sup>119</sup>Sn<sup>4+</sup> probe ions are statistically distributed over the sites of the octahedral substructure. In this case, the following statistical

**Table 2** Values of magnetic hyperfine fields  $H_k$  transferred to the  $^{119}\text{Sn}^{4+}$  probe, located in the bulk of  $Y_{0.9}\text{Ca}_{0.3}\text{Cr}_{0.9}\text{Ti}_{0.1}\text{O}_3$  and  $Y_{0.8}\text{Ca}_{0.2}\text{Cr}_{0.8}\text{Ti}_{0.2}\text{O}_3$  on Cr sites with different magnetically active surroundings, and experimental values  $A_{exp}(k)$  of the relevant spectral contributions at 4.2 K. Comparison of the statistical probabilities P(k) with the adjusted  $A^*(k)$  values allowing for the presence of the non-model spectral components.  $R_{model}$  is the ratio of the probabilities  $\frac{P^*(k)}{P^*(6)}$  obtained using the relevant P(k) values allowing for the formation of various  $\text{Sn}^{4+}\text{-Sn}^{4+}$  associates, as explained in the text.

k	$H_k$ (kOe)	<i>P</i> ( <i>k</i> )	$\frac{P(k)}{P(6)}$	$A_{exp}(k)$	A*(k)	$\frac{A^*(k)}{A^*(6)}$	$R_{model}$
Y <sub>0.9</sub> Ca <sub>0.1</sub> Cr <sub>0.9</sub> Ti <sub>0.1</sub> O <sub>3</sub>							
6	81	0.531	1	0.441	0.488	1	1
5	67.5	0.354	0.67	0.272	0.301	0.62	0.62
4	54	0.098	0.19	0.119	0.132	0.27	0.27
3	40.5	0.015	0.03	0.071	0.079	0.16	0.16
2	27	0.001	$2 \cdot 10^{-3}$	0.024	-	-	-
1	13.5	$5 \cdot 10^{-5}$	$1 \cdot 10^{-4}$	0.073	-	-	-
Y <sub>0,8</sub> Ca <sub>0.</sub>	<sub>2</sub> Cr <sub>0.8</sub> Ti <sub>0.2</sub> O <sub>3</sub>						
6	81	0.262	1	0.230	0.253	1	1
5	67.5	0.393	1.5	0.317	0.348	1.38	1.38
4	54	0.246	0.94	0.227	0.249	0.99	1.04
3	40.5	0.082	0.31	0.137	0.150	0.60	0.47
2	27	0.015	0.06	0.043	-	-	-
1	13.5	0.0015	$6 \cdot 10^{-3}$	0.046	-	-	-

probabilities, related to k=6, 5, 4, 3, 2, 1, and 0, are expected: 0.531: 0.354: 0.099: 0.015: 0.001: 0.00005: 0.000001, respectively.

Accordingly, for x=0.1, only the first four sites of  $Sn^{4+}$  can actually influence the <sup>119</sup>Sn spectrum. Besides, the experimental value of the hyperfine field, assigned to six neighboring  $Cr^{3+}$  (k=6),  $H_6$ =81 kOe, allows the partial contribution of each  $Cr^{3+}$  cation to the resulting spin polarization of a  $Sn^{4+}$  ion to be quantified (h=13.5 kOe) and, consequently, all other  $H_k$  values to be determined.

To analyze the <sup>119</sup>Sn spectrum of a tin-doped  $Y_{1-x}Ca_xCr_{1-x}Ti_x$   $O_3$ , we thus assume that the presence of  $Ti^{4+}$  ions can only give rise to the sextets characterized by the following values of H:

 $H_6$ =81,  $H_5$ =67.5,  $H_4$ =54,  $H_3$ =40.5,  $H_2$ =27,  $H_1$ =13.5, and  $H_0$ =0 kOe.

The parameters determined from fitting the spectrum in Fig. 5, are gathered in Table 2. One can see that our approach yields a rather satisfactory representation of the experimental spectrum shape. However, the apparent contributions  $A_{exp}(k)$  of the constituent sextets significantly differ from those expected from the P (k) values related to the statistical distribution of  $Ti^{4+}$ . So, we attempted to verify, keeping in mind the above tentative assignment of xs in 119Sn spectra of both titanium-free Ca-substituted chromites, whether this discrepancy could also be accounted for by the formation of various Sn-O-Sn associates in the structure still preserving the statistical distribution of Cr3+ and Ti4+ within the B-substructure. Considering the constituent sextets in the fitted spectrum (cf. Fig. 5 and Table 2), we conclude that the sextets related to the lowest values of  $H_k$  (corresponding to k < 3), that were already observed in the titanium free samples, ought to be omitted from further discussion. In fact, their spectral contributions are obviously too high as compared to those expected for x=0.1 referred to  $Y_{0.9}Ca_{0.1}Cr_{0.9}Ti_{0.1}O_3$  with statistically distributed  $Cr^{3+}$  and Ti<sup>4+</sup> ions. Accordingly, to analyze the relative contributions of the model-consistent sextets (with k=6, 5, 4, and 3), the adjusted values of the relative areas  $A^*(k)$ , allowing for the total contribution of non-model components present in the spectrum ( $A_{exp}$ =0.097), should be used (Table 2). So, as can be seen in Table 2, the value of

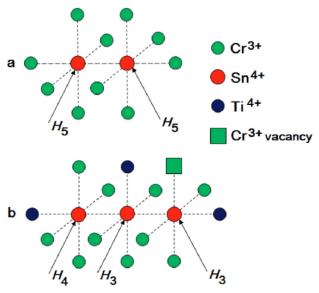


Fig. 6. Schematic representation of the Sn<sup>4+</sup> associates allowing <sup>119</sup>Sn Mössbauer spectra of  $Y_{0.9}Ca_{0.1}Cr_{0.9}Ti_{0.1}O_3$  and  $Y_{0.8}Ca_{0.2}Cr_{0.8}Ti_{0.2}O_3$  to be accounted for: simple associate involving two neighboring Sn<sup>4+</sup> (a); complex associate involving three Sn<sup>4+</sup> together with a Cr<sup>3+</sup> vacancy (b). Only relevant B-cations are shown.

 $A^*(6) = A_{exp}(6)/0.903 = 0.488$ , assumed to be that  $P^*(6)$  representing the actual probability to find, within this substituted chromite, a Sn<sup>4+</sup> ion having six Cr<sup>3+</sup> neighbors, is somewhat lower than the theoretical value P(6). Assuming that the observed difference  $P(6)-A^*(6)=0.043$  is due to the formation of an associate of two Sn<sup>4+</sup> ions, each initially having six Cr<sup>3+</sup> neighbors (Fig. 6a), it becomes possible to determine the increased spectral contribution of sites with k=5, probed by <sup>119</sup>Sn, *i.e.* P(5)+0.043=0.397. The latter value is higher than the adjusted experimental value  $A^*(5)=0.301$ , which we also assume to be that  $P^*(5)$  representing the actual probability to find a  $\mathrm{Sn^{4+}}$  ion with five  $\mathrm{Cr^{3+}}$  neighbors. Hence, the difference observed (0.397-0.301=0.096) is suggestive of the concomitant formation of a more complex associate, involving several neighboring Sn4+ ions. In this way we reach, in the next step, a fair agreement of the values of  $\frac{A^{*}(4)}{A^{*}(6)}$  and  $\frac{A^{*}(3)}{A^{*}(6)}$ , which seem at first glance surprisingly high, with those of  $\frac{p^*(4)}{p^*(6)}$  and  $\frac{p^*(3)}{p^*(6)}$ , respectively (cf. Table 2). Such an associate, consistent with the above model's values of H, is shown in Fig. 6b. As can be seen, it involves a  $Cr^{3+}$  vacancy ( $\square_{cr}$ ). The formation of  $\square_{cr}$ , caused by the charge compensation of the Sn4+ heterovalent dopant itself, has already been revealed by 119Sn Mössbauer spectra of tin-doped chromium sesquioxide [13]. By contrast, in both Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub> and Y<sub>0.8</sub>Ca<sub>0.2</sub>CrO<sub>3</sub> chromites Sn<sup>4+</sup> dopant ions were involved, together with Cr<sup>4+</sup> ions, in the charge compensation mechanism related to the  $Ca^{2+}$ -substitution for  $Y^{3+}$ . So, we can conclude that the spectrum of the  $^{119}\mbox{Sn}$  probe in  $\mbox{Y}_{0.9}\mbox{Ca}_{0.1}\mbox{Cr}_{0.9}\mbox{Ti}_{0.1}\mbox{O}_3$  is consistent with the statistical distribution of Cr<sup>3+</sup> and Ti<sup>4+</sup>, provided that partial segregation of Sn<sup>4+</sup> dopant ions in the B-substructure took place. Despites their very low content, such segregations are thus proven capable of changing significantly the spectrum of the 119Sn probe. Indeed, these results are in qualitative agreement with our above interpretation of the "intriguing" xs sextet in the Mössbauer spectrum of titanium-free Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub>.

## 3.2.2. Y<sub>0.8</sub>Ca<sub>0.2</sub>Cr<sub>0.8</sub>Ti<sub>0.2</sub>O<sub>3</sub>

This chromite, characterized by the different Cr/Ti ratio and, consequently, by the different probability distribution P(k), was used for further testing of the procedure described in Section 3.2.1.

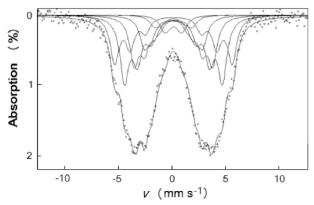


Fig. 7. 119Sn Mössbauer spectrum of tin-doped Y<sub>0.8</sub>Ca<sub>0.2</sub>Cr<sub>0.8</sub>Ti<sub>0.2</sub>O<sub>3</sub>.

In the present case, according to Eq. (1), the B-site with six Cr<sup>3+</sup> neighbors is no longer the predominant one and the main contribution to the 119Sn Mössbauer spectrum in Fig. 7 arises from the site with k=5. The results of analysis of this spectrum, gathered in Table 2, show that the procedure developed for interpreting the <sup>119</sup>Sn spectrum of the precedent chromite with x=0.1, remains valid in the case of  $Y_{0.8}$ Ca<sub>0.2</sub>Cr<sub>0.8</sub>Ti<sub>0.2</sub>O<sub>3</sub> (x=0.2), provided that partial segregation of the Sn<sup>4+</sup> dopant ions also took place. So, this furnished a further support for our interpretation of all other results reported in this paper.

#### 4. Conclusions

Low-temperature Mössbauer spectra of 119Sn4+ probe ions, located in the bulk of three Ca-substituted yttrium chromites, i.e.  $Y_{0.9}Ca_{0.1}CrO_3$ ,  $Y_{0.9}Ca_{0.1}Cr_{0.9}Ti_{0.1}O_3$  and  $Y_{0.8}Ca_{0.2}Cr_{0.8}Ti_{0.2}O_3$ , have pointed out the occurrence of partial segregation of Sn<sup>4+</sup> dopant ions in all of these compounds. In the case of Y<sub>0.9</sub>Ca<sub>0.1</sub>CrO<sub>3</sub>, neither highly oxidized Cr<sup>4+</sup> nor Cr<sup>6+</sup> species is found in the vicinity of the

bulk-located <sup>119</sup>Sn<sup>4+</sup> probe. Although Cr<sup>4+</sup> ions are distributed, like the 119Sn probe, over B-sites in the bulk of crystallites, they are thought to be frozen, at 4.2 K, outside the magnetically active surrounding of Sn<sup>4+</sup>, whereas Cr<sup>6+</sup> ions are essentially located on the surface of crystallites, thus too far from the 119Sn4+ probe. In the case of both studied Ti-containing chromites, 119Sn Mössbauer spectra have provided the original indirect evidence for the statistical distribution of Cr3+ and Ti4+ ions over octahedral sites and allowed us to characterize the detected associates of Sn<sup>4+</sup>.

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