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Gold nanoparticle self assembly moderated by a cholesteric liquid crystal

Joel S. Pendery,^{abc} Olivier Merchiers,^{de} Delphine Coursault,^{ab} Johan Grand,^f Habib Ayeb,^{ab} Romain Greget,^g Bertrand Donnio,^g Jean-Louis Gallani,^g Charles Rosenblatt,^c Nordin Felidj,^f Yves Borensztein,^{ab} Emmanuelle Lacaze * ^{ab}

We show that the study of gold nanoparticle self-assemblies induced by a liquid crystal matrix reveals the intimate distorted structure of the liquid crystal existing prior to nanoparticles incorporation. We also show how this intimate structure monitors the spacing between nanoparticles in the self-assemblies. We have created hybrid films of cholesteric liquid crystal (CLC) and gold nanoparticles, the CLC being deformed by competing anchorings at its two interfaces. Whereas previous results have evidenced formation of only slightly anisotropic clusters of large nanoparticles (diameter $20nm$), we now demonstrate for smaller nanoparticles (diameter $4.2nm$) formation of long needles of length larger than 50 nanoparticles and width smaller than 5 nanoparticles, on average oriented perpendicular to the anchoring direction. The difference between the two kinds of nanoparticle aggregations is interpreted by a modification of the balance between aggregation between nanoparticles and trapping by the defects, favoured by the disorder induced by the alkylthiol molecules grafted around the nanoparticles. This leads to a well-defined, anisotropic Localized Surface Plasmonic Resonance (LSPR) of the $4.2nm$ embedded nanoparticles. Interpretation of these optical properties using generalized Mie theory allows for a comparison between CLC/gold nanoparticles and the same nanoparticles trapped within smectic topological defects or deposited on the same substrate without liquid crystal. A smaller spacing between nanoparticles is demonstrated in the CLC system with an attraction between nanoparticles induced by the CLC matrix, related to the additional disorder associated with the nanoparticles presence. The experimental observations allow us to estimate the disordered size of the liquid crystal shell around the nanoparticles in the CLC to be of some nanometers. They also suggest that the CLC distorted by competing anchorings is characterized by the presence of arrays of defects with topological cores of width smaller than $5nm$ that act as efficient anisotropic traps for the nanoparticles.

1 Introduction

It is now established that a large number of well-defined hybrid systems composed of liquid crystals (LC) and nanoparticles can be prepared, either to vary the properties of the LC or those of the nanoparticles [3, 20]. On the other hand, it is also well-known that optical properties of

the nanoparticles can be tuned if controlled self-assemblies of nanoparticles can be prepared. This calls for a better understanding and control of these hybrid systems. For example, Quantum dots emission can be modified through well-controlled self-assemblies [36, 35, ?]. In addition, the localized surface plasmon resonance (LSPR) properties of metallic nanoparticles may be greatly varied by the coupling between nanoparticles [27]. In particular, formation of linear self-assemblies is highly desirable since it exhibits anisotropic optical properties which can be activated by external parameters, like light polarization [9], similarly to oriented nano-rods for example [7, 23].

In the case of nanoparticles dispersed in thermotropic LCs, two main mechanisms have been identified, which facilitate control of the nanoparticles self-assembly by the LC matrix. First a segregation process leads to a localization of the nanoparticles at the LC/air interface. For cholesteric LC films (CLCs), the obtained nanoparticle assemblies can mimic the well-known cholesteric fingerprints [4]. Second, trapping phenomena have been demonstrated for systems with topological defects, such as saturn ring defects surrounding micrometer sized colloids [34], blue

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phase disclinations [17, 13, 33, 34], screw disclinations of twist-grain-boundary phases [8] or linear patterns of smectic edge dislocations [9, 26]. In the presence of topological defects, nanoparticles are viewed as particulate impurities that distort the LC in their vicinity, or induce disorder. Located in topological defects, they take the place of a portion of the already disordered defect cores and allow for a decrease of the energy within the system [17]. When the defects are linear, like in smectic oily streaks dislocations, single chains of nanoparticles can be obtained exhibiting highly anisotropic LSPR [9]. It now becomes important to identify if new kinds of topological defects can be oriented. Two questions can thus be asked: First, are other kinds of topological defects appropriate for a controlled and oriented self-assembly of nanoparticles? Second, how the nature of the trapping sites controls the self-assembly structure, in particular the spacing between nanoparticles which, in return, controls the optical response of the hybrid system associated with gold nanoparticles?

The goal of the present paper is to study gold nanoparticle self-assembly based on the nanoparticles dispersion in a CLC matrix distorted by competing anchorings at the substrate/CLC and CLC/air interfaces. These competing anchorings are known to produce a 90° rotation of the cholesteric helix in the vicinity of the CLC/air interface [2, 6]. Below the area associated with the 90° rotation, the cholesteric helix is perpendicular to the substrate. Above this area, the helix is parallel to the CLC/air interface, its orientation being linked to the ratio e/p , with e the thickness and p the cholesteric pitch [2, 6]. A rotation of the helix parallel to the CLC/air interface can consequently be induced when the pitch is varied, for example by irradiation of cholesterics doped by photoactive chiral dopants [10, 6]. However the intimate structure of the area associated with 90° rotation of the cholesteric helix is not known. The electromagnetic coupling between gold nanoparticles of diameter $4.2nm$ trapped in the distorted CLC matrix has been evidenced through the measurements of a well-defined anisotropic LSPR. In a second step, the majority of the CLC matrix has been eliminated by evaporation, revealing nanoparticle self-assemblies with a needle shape. Analysis of LSPR data using calculations within the frame of generalized Mie theory allows to interpret both data, with and without CLC. They both appear to be consistent with the presence of nanoparticle needles. Whereas these hybrid systems lead to the formation of only slightly anisotropic aggregates for large nanoparticles [1] the balance between aggregation and trapping phenomena turns in favor of trapping for small nanoparticles. Moreover the comparison with LSPR data obtained with smectic topological defects or nanoparticle deposition on the substrate without liquid crystal evidences a shorter internanoparti-

cle distance in CLC matrix. Interpretation of these data suggests that the presence of CLC topological defects may induce the trapping of nanoparticles and the formation of needles, but within a defect core of typical diameter smaller than the nanoparticle size. The results demonstrate a direct relation between nanoparticles spacing and nature of liquid crystal traps. They demonstrate additionally that the study of nanoparticle self-assembly processes can reveal information about the intrinsic deformations of the LC film.

2 Materials

Polyvinyl alcohol (PVA) was spin coated onto cleaned glass slides, baked at $120^\circ C$ to polymerize, and manually rubbed unidirectionally with cotton to promote planar anchoring at the surface. A cholesteric liquid crystal (CLC) mixture was prepared by doping 2wt.-% of the chiral agent S-2011 (MLC-6247 from Merck) in the nematic liquid crystal E7 (Merck). This sets the helical pitch at $p = 5.5\mu m$. E7 is composed of the mesogens 5CB (4'-pentyl-4-cyanobiphenyl), 7CB (4-heptyl-4'-cyanobiphenyl), 8OCB (4-octyloxy-4'-cyanobiphenyl) and 5CT (4-cyano-4''-pentylterphenyl) and is in the nematic phase at room temperature with the phase diagram of K($-20^\circ C$)N($58^\circ C$)I. Here K, N and I stand for the crystalline, nematic and isotropic phase respectively. It has a high optical anisotropy $\Delta n = n_e - n_o = 0.23$, where $n_e = 1.75$ is the extraordinary index, parallel to the optical axis and $n_o = 1.52$ is the ordinary index, perpendicular to the optical axis.

The CLC mixture was spin coated onto the rubbed PVA-coated glass to give an average thickness of $\sim 12\mu m$. Inspection by optical microscopy showed a striped texture, of period Λ , of the cholesteric hybrid sample (Figure 1a). Here the PVA substrate promotes planar anchoring and the CLC helical axis, at the substrate and within the bulk, is perpendicular to the substrate. At distance around $p/2$ below the CLC/air interface, the helical axis undergoes a rotation to change from perpendicular to the substrate in order to better accommodate the director's preferred homeotropic boundary condition [2, 6] (Figure 1b). Stripes are observed by optical microscopy and are associated with the helical axis parallel to the CLC/air interface. They are perpendicular to the helix axis and of period slightly larger than two times the cholesteric pitch [2, 6].

Gold nanoparticles have been prepared following the "inverted Brust-Schiffrin" method, as described by Li et al. [21]. 1.4mL of a 0.2 mmol solution of $HAuCl_4/3H_2O$ in water are mixed with 20mL of a 0.60mmol solution of

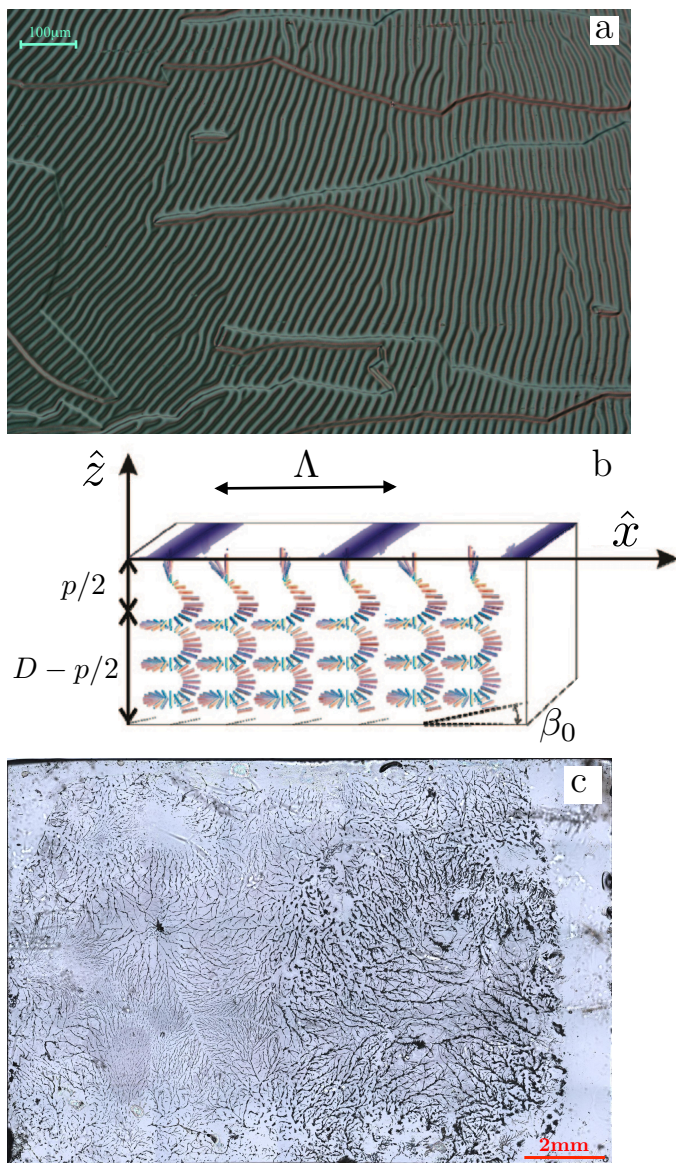


Figure 1 (a) A typical area of the CLC sample, which exhibits the expected striped texture of a hybrid aligned CLC, (b) The director field of the aligned CLC sample. The helical axis is along the \hat{z} axis, perpendicular to the planar aligning surface over a thickness $(D - p/2)$, p being the cholesteric pitch. It gives way to a rotation of the director within a thickness around $p/2$ from the CLC/air interface to allow the director to meet the homeotropic boundary condition at the interface[6], (c) map of the entire sample after evaporation of the liquid crystal.

TOAB in toluene (TOAB is trioctadecylammonium bromide), under vigorous stirring at room temperature. After 1 hour, 2mL of freshly prepared aqueous solution (3 mmol) of NaBH_4 are added. After 10 minutes, 2mL of a 0.6mmol solution of dodecanethiol in toluene are added. The reaction medium is left under stirring for 20 hours. The organic phase is kept, washed once with 500mL of a 2 percent sulfuric acid solution, and then three times with 500mL of pure water. The organic phase is then centrifuged to eliminate the larger aggregates. The supernatant is kept and evaporated under vacuum down to a volume of a few mL. It is poured into an excess of ethanol. A light centrifugation is performed (2 minutes at 2000rpm), the supernatant is discarded, the precipitate is dried under vacuum. Gold nanoparticles, 4.2nm in diameter (polydispersity 18 percent) were then suspended in toluene. A solution of concentration of $500\mu\text{g}/\text{mL}$ was used. Deposition of $10\mu\text{L}$ of the suspended nanoparticles solution was conducted on a hot CLC sample, at 55°C to preserve the striped texture but to expedite the evaporation of the solvent. After the toluene evaporated, the sample was removed from the heating stage and the CLC texture was observed by optical microscope to confirm the absence of any modification to the period of the stripes. This demonstrates that the CLC pitch was not affected. We can see the presence of nanoparticles by a change of colour of the striped areas larger than $500\mu\text{m} \times 500\mu\text{m}$. These were the only modifications to the striped textures we observed. This observation allowed for localized measurements of the LSPR of the hybrid system, CLC/nanoparticles.

Extinction spectra were measured with an optical microscope using a 50x objective lens, with a numerical aperture of 0.55. The sample was inverted such that the polarized light is incident on the CLC/air interface. The transmitted light was collected and then dispersed using a LOT Oriel MS260i spectro-photometer. A 2mm pinhole allowed an area of $100\mu\text{m} \times 100\mu\text{m}$ to be probed. To obtain only the extinction spectra of the nanoparticles, the spectrometer data of the CLC/nanoparticle hybrid sample for a given area and stripe orientation was referenced to a CLC sample with a similar thickness and stripe orientation. Data were taken for incident light polarizations parallel and perpendicular to the stripes.

3 Results

3.1 LSPR of nanoparticles/CLC hybrid films

The extinction spectra of Figure 2a were obtained on the CLC/gold nanoparticle film with stripes perpendicular to

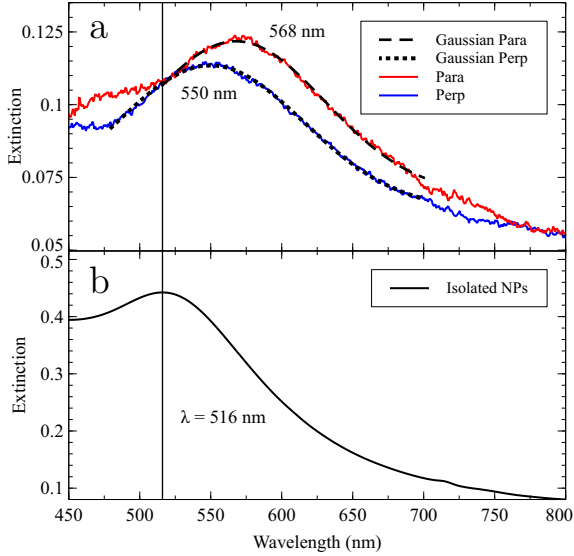


Figure 2 Localized surface plasmon resonance of (a) a typical area for the CLC/nanoparticle hybrid sample (stripes perpendicular to the anchoring direction) and (b) isolated gold nanoparticles suspended in toluene. Parallel (red) and perpendicular (blue) curves correspond to the incident light polarization orientation with respect to the stripe orientation. A Gaussian fit was performed to determine the band wavelength value.

the anchoring direction. The experimental data were fitted with a Gaussian curve. Spectra in Figure 2a display a plasmon resonance of $\lambda_{para} = 568nm$ when the light is polarized parallel to the stripes and $\lambda_{perp} = 550nm$ when the light is polarized perpendicular to the stripes. An extinction spectrum was measured for isolated gold nanoparticles in toluene (see Figure 2b) and the plasmon resonance peak was $516 \pm 1nm$. This gives a red-shift for the CLC/gold nanoparticles hybrid film of $52nm$ for light polarized parallel to the stripes and $34nm$ for light polarized perpendicular to the stripes, exhibiting an anisotropy in the LSPR with respect to the stripes orientation. However the LSPR anisotropy appears to vary from place to place. In fact, LSPR anisotropy is related to anchoring orientation. For stripes oriented parallel to the anchoring direction (5 of the 15 measured zones), the larger red-shift occurs in majority for polarization perpendicular to the stripes and thus for polarization perpendicular to the anchoring direction (4 of the 5 measured zones of stripes parallel to the anchoring direction). This is also true for stripes perpendicular to the anchoring direction. For these 10 measured zones, the red-shift is larger for polarization parallel to the stripes. If we consider only stripes perpendicular to the anchoring directions, we ob-

tain $\langle \lambda_{para} \rangle = 565 \pm 3nm$ and $\langle \lambda_{perp} \rangle = 555 \pm 5nm$

It is well known that the localized surface plasmon resonance of a metal nanoparticle is highly dependent upon the optical index of the surrounding media [14]. However, for the case of metal nanospheres in a liquid crystal, Park and Stroud [28] have shown that the LSPR is weakly dependent on the orientation of the electric field along the ordinary or the extraordinary direction. By using their model, we calculated an anisotropy of the LSPR of nanoparticles in CLC of only $2nm$, against a maximum of $18nm$ observed experimentally (Figure 2a). Moreover, the obtained red shifts for the nanoparticles in the CLC with respect to the LSPR of the isolated nanoparticles in toluene, whose index of refraction is $n_{tol} = 1.498$, is calculated to be of only 8 and $10nm$, along the ordinary and extraordinary directions, respectively. Such an effect therefore does not account for the large observed red-shifts. On the other hand, it is also known that the electromagnetic interaction of close particles can lead to large red-shifts (e.g. for polarisation along a chain of particle, or for two- or three-dimension aggregates). The observed red-shift, not explained by the index of refraction of the CLC, therefore indicates that aggregates of nanoparticles are likely present within the CLC matrix. As shown by the optical anisotropy, their shape and structure may be directly induced by the cholesteric matrix. To understand how the self-assembly process may occur, the aggregate's shape and structure have to be elucidated.

3.2 AFM and LSPR after CLC evaporation

To resolve the aggregate shape utilizing an atomic force microscope (AFM), the sample was placed in a vacuum oven to remove a significant portion of the liquid crystal. This evaporation was performed within a vacuum at 10^{-7} bar for two days and then heating the sample to 40° for an additional three days to expedite the evaporation process. The sample was removed from the vacuum and the surface was mapped using optical microscopy (see Figure 1c). Pictures of the entire surface were taken using a Leica DMRX optical microscope with an objective lens magnification of $5x$ and then stitched together (Figure 1c). Not all of the CLC was evaporated, which is evident by the presence of the dark, divergent lines on the map corresponding to thick CLC regions. This was expected since evaporation of large liquid crystal molecules is difficult. However, using this panoramic surface map, regions with only a thin CLC layer could be selected for AFM imaging (Figure 3c). In addition, these same areas were used for LSPR measurements (Figure 4).

Both regions with and without a plasmon resonance

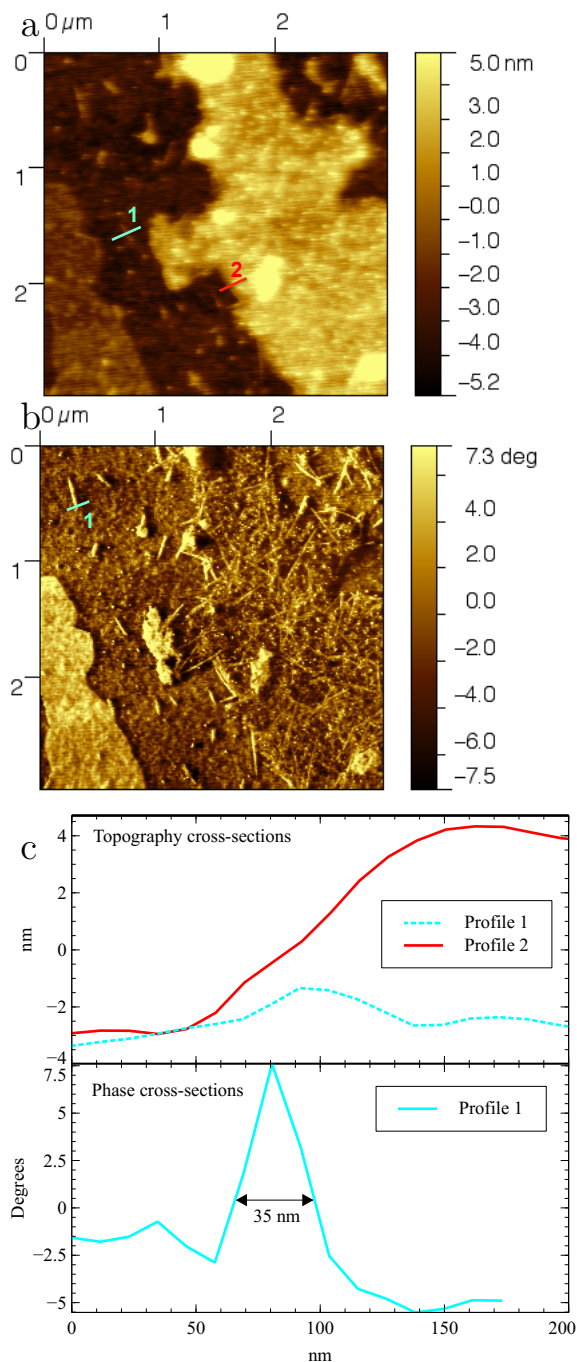


Figure 3 (a) AFM Topographical and (b) phase data for a region with a plasmon resonance peak. (c) Corresponding topographical cross sections according to the lines drawn on (a) and (b). A high number of anisotropic aggregates are visible in the phase data below a liquid crystal layer. The dimensions of a needle (width 35nm - (bottom of (c))) are determined using the phase data (b) as this allows for precise location of the needle's edges.

were probed using the AFM. Regions of $100\mu\text{m} \times 100\mu\text{m}$, of size similar to the one measured by spectro-photometry, were imaged and zooms were performed until individual aggregates could be resolved. No particles could be imaged in regions without a plasmon resonance, in agreement with having the number of nanoparticles below the threshold for LSPR measurement. In areas that exhibited a LSPR, either a homogeneous AFM topography and phase were present or areas with needle-like structures were observed (Figure 3a and 3b). The needles, absent in the regions where no LSPR is measured, are therefore the nanoparticle aggregates, as it will be discussed in detail below. If we concentrate on Figure 3a, two distinct regions are observed with needles: 1) the valley area on the left half of the image and 2) the large, elevated region on the right half of the image. In the topography, flat needle-like aggregates are clearly visible, but only in the lowest region (1). In contrast, in the phase data (Figure 3b), the needle shaped aggregates are visible in the lowest region but also appear, less clearly but in larger quantity, in the highest region (2). The length of the needles varies from 100 to 400nm while the width varies from approximately 30 to 40nm . Taking into account the curvature radius of the AFM tip, varying typically between 5 and 30nm [32], the needles width can not be precisely determined. It may range from 1 to 5 nanoparticles.

The topography of the needles investigated in the (1) valley (Figure 3c profile 1) shows a height of approximately $2 - 3\text{nm}$ for the needles. Since the diameter of a nanoparticle is 4.2nm , one must conclude that the nanoparticle needles are flat on the surface, one particle thick but still partially submerged in the CLC. In addition, around the nanoparticles, in the lowest (1) as well as in the elevated (2) region, the AFM phase image is similar (Figure 3b). The phase is correlated with the stiffness of a given region [25], and it suggests that both of these regions are covered in CLC. Accordingly, the AFM phase on top of the nanoparticles is higher since nanoparticles consist of harder material. This is evident in the lowest area (1) where the AFM tip directly interacts with the nanoparticles. This is also the case in the most elevated region (2), although less clearly, in agreement with needles covered by CLC, as shown by topography data, with no protuberant needles. These needles, after CLC evaporation, get close to the air/CLC interface and modify its stiffness. This allows us to observe the needles using AFM. These results finally reveal the presence of needles mostly covered by CLC, which may be responsible for the LSPR signal (Figure 4), measured over the same area as the AFM data (Figure 3).

The region at the bottom left of the image is believed to correspond to a demixing of E7 liquid crystal induced

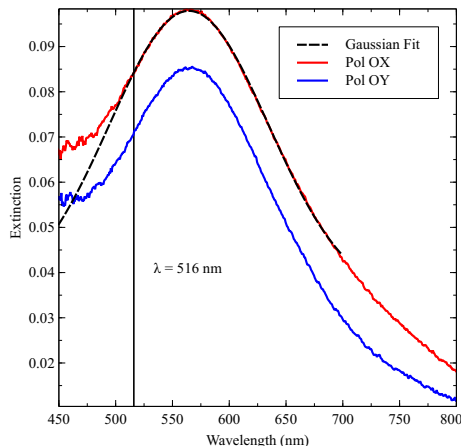


Figure 4 Extinction spectra from the selected region where AFM measurements were conducted. Parallel (red) and perpendicular (blue) curves correspond to the incident light polarization orientation with respect to anchoring orientation.

by the evaporation, leading to a crystallized hard area, as shown by phase image. It may correspond to pure 8OCB, pure 5CT or a mixture of the two, which are expected to be crystalline at ambient temperature.

Measurements of LSPR band performed on 7 areas of the evaporated sample mainly lead to similar LSPR band (Figure 4), parallel (x-axis $\langle \lambda \rangle = 564 \pm 1nm$) and perpendicular (y-axis $\langle \lambda \rangle = 566 \pm 1nm$) to the anchoring direction. The homogeneity of the LSPR signal suggests that the needles can be present but covered by a CLC film too thick to allow their observation with AFM. This is the case only when fully homogeneous topology and phase data are obtained with AFM. The absence of anisotropy, in contrast with the results obtained previous to evaporation, is in agreement with the observation by AFM of disordered needles. We will show in the following that formation of needles not only allows one to interpret the data after evaporation, but also allows one to interpret the optical properties in hybrid nanoparticles/CLC films, strongly suggesting that aggregation phenomenon is mostly independent of the evaporation process. The small size of the needles is consistent with the absence of any change of texture for the hybrid nanoparticle/CLC films observed by optical microscopy compared to pure CLC. A smaller disorder of the needles orientation may be present in the CLC/nanoparticle hybrid sample, the CLC evaporation process modifying mainly the orientation of the needles.

3.3 LSPR analysis

To interpret the LSPR data after removal of the cholesteric film, we associate the needles with the presence of

nanoparticle ribbons of disordered orientation. We have considered ribbons formed by hexagonally packed nanoparticles, characterized by the inter-nanoparticles edge-to-edge distance, the so-called spacing, s . To fit with the AFM data, we have considered long 2D ribbons only and calculated the related LSPR by the generalized Mie theory [24]. The optical index of the embedding medium has been taken equal to that of alkythiols, ($n=1.458$). Due to the small distance between the nanoparticles, there will be mainly alkythiol molecules between them. It is well-known that in nanoparticle chains, for a given nanoparticle spacing, the red-shift of the LSPR wavelength with respect to isolated nanoparticles, increases with the length of the chain until it saturates [16, 40, 30, 11, 38]. We expect similar behavior for the ribbons, and we have considered ribbons of length 45 nanoparticles, of the order of the length of the measured needles $45 \times 4.8nm = 216nm$. AFM data show widths of the order of $30 - 40nm$ (Figure 3c). Taking into account the convolution by the AFM tip of curvature radius around $10nm$ [32], the exact width values can not be obtained, only evidencing widths equal to or smaller than 5 nanoparticles. We note that the red-shift of ribbons saturates more slowly than chains and that the red-shift of a ribbon with a length of 45 nanoparticles is not completely saturated but very close to saturation. Ribbons are associated with a LSPR wavelength for polarization parallel to the ribbons larger than the one for polarization perpendicular to the ribbons (Figure 5). For ribbons of disordered orientation, the average LSPR is proportional to the sum of both extinctions and is shown in Figure 5 for a width equal to 3 nanoparticles ($s = 0.56nm$). The spacings between nanoparticles accounting for an average plasmon band at $565nm$, are summarized in table 1 for ribbons widths ranging from 1 to 5 nanoparticles. We can already conclude that s is smaller than $0.6nm$, which suggests a strong interdigitation/compression of the alkythiols between the nanoparticles.

The anisotropy of LSPR bands observed in the presence of cholesteric films is associated with ribbons that, on average, are oriented perpendicular to the anchoring direction. However, they are not strictly oriented perpendicular to the anchoring direction since a larger anisotropy than the maximum $18nm$ would be expected between the polarization parallel to the anchoring direction and the perpendicular polarization (Figure 5). If we consider the spectra of figure 2a, we can interpret the observed LSPR anisotropy of $18nm$, as a consequence of some disorientation of the ribbons, using the property that the final extinction can be approximated by a linear combination of extinction parallel and perpendicular to the ribbons long axis. We will now explain briefly the theoretical argument

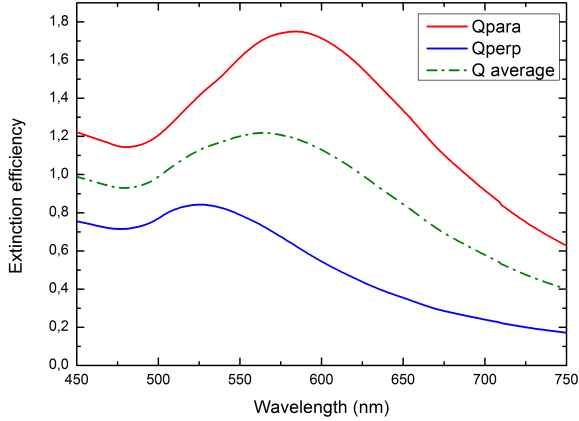


Figure 5 Extinction spectra calculated in generalized Mie theory in a thiol index for a ribbon of length 45 nanoparticles and width 3 nanoparticles ($s = 0.56nm$). In blue the polarization is perpendicular to the ribbon in red it is parallel. In green the average extinction is presented.

underlying this property.

The extinction efficiency (cross-section normalized by the geometrical section) for an arbitrary incident polarization can be written as[5]:

$$Q(\theta) = Q_{par} \cos^2 \theta + Q_{perp} \sin^2 \theta + Q_{par,perp} \sin \theta \cos \theta, \quad (1)$$

where θ is the angle between the ribbon's long axis and the incident polarization vector. Q_{par} and Q_{perp} are the extinction efficiencies for incident polarization parallel or perpendicular to the ribbon's axis respectively. The term $Q_{par,perp}$, is the contribution due to cross-polarization (off-diagonal terms in the scattering amplitude matrix are non-zero in general). We consider now the needles to be disoriented with respect to the incident polarization vector. If $n(\theta)$ is the distribution function of the angles, we can write the average extinction efficiency as

$$\langle Q \rangle = \int_0^\pi n(\theta) (Q_{par} \cos^2 \theta + Q_{perp} \sin^2 \theta + Q_{par,perp} \sin \theta \cos \theta) d\theta \quad (2)$$

For a uniform probability density function of the type $n(\theta) = 1/\pi$ (our hybrid films after evaporation), we get:

$$\langle Q \rangle = (Q_{par} + Q_{perp})/2. \quad (3)$$

In the case of a non-uniform distribution, the cross-term does not vanish in general. However, for the ribbons considered here, we found that the factor $Q_{par,perp}$ is negligible. This simplifies the analysis greatly, since now we can write the very simple equation:

$$\langle Q \rangle = Q_{par}C + Q_{perp}(1 - C), \quad (4)$$

where we used the fact that the distribution $n(\theta)$ is normalized to one. Hence $C = \int_0^\pi n(\theta) \cos^2 \theta d\theta$, Q_{par} and Q_{perp} being calculated. We can now use this equation to fit the experimental data and estimate the alignment of the ribbons ($C = 0.5$ being completely random and $C = 0.0$ or $C = 1.0$ being perfectly aligned).

We have considered 5 cases to account for the extinction spectra of figure 2a, each case being associated with a given ribbon width, ranging from 1 to 5 nanoparticles. The results are summarized in table 1. These results give a measure of the disorientation of the ribbons along the CLC stripes, around $C = 0.65$. They suggest that the areas where anisotropy is maximum (figure 2a) could display homogeneous ribbons width and inter-particle spacings. We can not establish their exact values. However, from Table 1 it appears that the spacing obtained for the initial cholesteric film (before evaporation) is very similar to the one after evaporation. Only a small spacing reduction after evaporation is evidenced, which on the other hand depends on the considered ribbon width. For chains or thin ribbons with 2 nanoparticles-width, a very small reduction is associated with the LSPR data. Furthermore, because we expect a mainly constant inter-particle spacing in the initial cholesteric film, the position dependent anisotropy can be interpreted in two ways: 1. for a constant value of the aggregate width, the disorder varies (varying C from place to place), or 2. for a nearly constant C value, the width varies together with a possibly varying local nanoparticle concentration. More precisely: to maintain the same anisotropy (for example the $\langle \Delta \lambda \rangle = 18nm$ value of figure 2a), C has to increase (better alignment) together with the ribbon width. This is expected since chains have a higher aspect ratio than wide ribbons and the former are thus intrinsically more anisotropic. This is why we can expect a majority of thin ribbons in the area of maximum anisotropy ($\Delta \lambda = 18nm$) with C values around 0.64. With the hypothesis of mainly constant C value, the observed average smaller anisotropy ($\langle \Delta \lambda \rangle = 10nm$) would be the consequence of an increasing number of wider ribbons when local nanoparticle concentrations increase.

Table 1 Results of the fit using eq. 2, for different widths.

Ribbon width (#NP)	C before evaporation	before evaporation s/nm	after evaporation s/nm
1	0.65	0.54	0.53
2	0.64	0.57	0.55
3	0.65	0.59	0.56
4	0.68	0.62	0.57
5	0.71	0.64	0.58

4 Discussion

4.1 Needles as the result of trapping in specific areas

In order to understand the origin of the needles, the first question concerns their localization at the CLC/air interface or within CLC bulk. We have demonstrated by AFM that, even after evaporation of the majority of the liquid crystal film, most of the needles are still covered by CLC molecules. This suggests that they are composed of nanoparticles which have penetrated into the CLC film, in contrast with segregation at the surface. This latter process occurs for micrometrical colloids at cholesteric/oil interface [29, 22]. For nanoparticles it occurs in cholesterics after long annealing [4] or for nanoparticles originally dispersed in bad solvents for liquid crystals, like water [1].

The second question directly concerns the origin of the needles. It is remarkable that the two kinds of nanoparticles, nanoparticles of diameter $20nm$ covered by PVP polymer, studied by Ayeb *et al.* [1] and the present nanoparticles of diameter $4.2nm$ with grafted dodecanethiol, both lead to formation of elongated aggregates of limited size, oriented on average by the CLC matrix. This is consistent with a minimum of energy associated with a trapping of nanoparticles in a favorable area. One can not exclude that kinetic issues also are involved and that elongated aggregates only correspond to metastable states. On the other hand, our results reveal a more anisometric shape for the aggregates formed from small nanoparticles. In order to understand this difference, let us recall first that in nematic 5CB in the absence of any localized deformation of the liquid crystal director, isotropic aggregates, possibly large, are formed from small nanoparticles coated with dodecanethiols [31], in contrast with large nanoparticles, of diameter larger than $30nm$ [?]. This shows a strong tendency for nanoparticles in the nematic film to aggregate, at least partially under the influence of Van der Waals forces. Comparison between nanoparticles size ($4.2nm$) and helix pitch ($p = 5.5\mu m$) shows that, in the CLC, in the absence of any specific distortions, the environment around the nanoparticles is locally a very weakly twisted nematic. We thus would expect formation of isotropic aggregates like in 5CB nematic films. However the observation of anisometric aggregates of limited size and on average oriented by the CLC matrix suggests that aggregation is balanced by a phenomenon of nanoparticle trapping. For large nanoparticles, cluster formation would occur prior to trapping, leading to formation of only slightly anisometric clusters. For small nanoparticles, we observe mainly long needles (with lengths of several hundreds of nanometers) with small width (not larger

than 5 nanoparticles). Such an anisotropy is not compatible with bulk aggregation. This suggests that trapping would preferentially occur in a specific anisotropic area prior to Van der Waals induced aggregation. Aggregation would occur within this anisotropic trapping area in a second step, the complete process being associated with solvent evaporation. It is known that distorted liquid crystal areas attract nanoparticles due to the reduction of elastic energy accompanying the replacement of a distorted volume by a nanoparticle. Nanoparticles are thus attracted by distortion gradients [39, 34]. In CLC it has been shown that even polymers can be attracted by distorted areas [?]. For nanoparticles this is expected to occur independently of the nature of the nanoparticles, but the attraction efficiency has been shown to depend on the anchoring around the nanoparticles [33, 34]. This may at least partially explain the better trapping observed for the small nanoparticles in CLC since they were covered by alkythiol, in contrast with the large nanoparticles covered by PVP polymer [1], the nanoparticle size also coming into play.

4.2 Trapping sites for the needles

It is now well-known that, once nanoparticles are attracted, stable sites of trapping can be at liquid crystal topological defects. This is due to the particularly large energetic advantage associated with the replacement of a defect core by a nanoparticle. Stable sites have been evidenced for topological defects localized on micrometer-size colloids in nematic films [34], for disclinations in blue phases [41, 13, 33], screw disclinations of twist-grain-boundary phases [8] as well as for smectic dislocations [9, 26]. To better understand the nature of the trapping sites in the present CLC, it may be useful to compare trapping within the CLC film and trapping within smectic dislocations for the same nanoparticles.

We thus have trapped our $4.2nm$ nanoparticles in linear patterns of smectic dislocations of 8CB [9], expected to be associated with a linear dislocation core of diameter several nanometers and composed of disordered molecules [19]. We have obtained chain-like aggregates along the dislocation patterns associated with a LSPR band at $553nm$ for light polarized parallel to the dislocations, whereas the LSPR for light polarized perpendicular to the dislocations remains close to that of isolated nanoparticles ($516nm$). Analysis with Generalized Mie theory gives an average distance of $0.98nm$ between nanoparticles, larger by more than 50 percent than the distance obtained in the CLC. Moreover, when a simple monolayer of nanoparticles is formed in the absence of liquid crystal by depositing a drop of nanoparticles in toluene on the rubbed PVA sub-

strate, we observe a LSPR band of $556nm$, independent of the light polarization. This band does not vary with the nanoparticles concentration. It consequently does not vary with the size of the domains of the monolayer. This implies that the monolayer is formed of 2D domains of coupled nanoparticles that can be considered as infinite for a simulation of the monolayer LSPR since their physical size can be considered as beyond the coupling saturation for LSPR. As shown by dipole approximation calculations performed for an infinite hexagonal network of nanoparticles in a medium with an optical index equal to that of alkylthiol, the inter-particle spacing is $s = 1nm$. This suggests that, in smectic defects, the spacing between nanoparticles is very similar to that in the monolayer network and may be mainly mediated by the same attractive Van der Waals interactions between nanoparticles, as well as steric repulsive interactions. This is not the case in CLC.

In CLC, compression of the nanoparticles spacing imposes a larger interdigitation between the thiols around the nanoparticles. This compression must be imposed by the surrounding CLC and confirms localization of the needles within the CLC film. This compression may arise from the disorder induced by the nanoparticles in the cholesteric sample. If the trapping area is not fully disordered, the presence of nanoparticles locally induces disorder in the liquid crystal, entailing an additional energetic cost that is absent in topological defects with large enough disordered cores. Consequently in CLC, the distance between two nanoparticles must be as close as possible to reduce the disordered volume around the nanoparticles in contrast with smectic disordered cores [9].

A simple calculation can be made to account for nanoparticles that are further attracted by the $0.4nm$ difference that is observed between nanoparticles spacing in nanoparticle monolayers without liquid crystal or in smectic defects vs. in CLC. If we calculate the Van der Waals interactions between two $4.2nm$ gold spheres in a thiol environment, for a spacing $0.98nm$, an energy of $1.5kT$ is obtained. For a spacing $0.6nm$, an energy of $4.5kT$ is obtained [15]. In the monolayer, Van der Waals attractions are balanced by steric interactions that do not allow for a reduced spacing. Due to the alkanethiol coating, allowing quasi-no LC to penetrate between the nanoparticles, we can consider that the Hamaker constant for the nanoparticle monolayer on PVA is similar to that for the nanoparticle needles in CLC, as only the host medium has changed. The CLC compression must be significantly larger than the gain due to Van der Waals interactions in order to successfully compete with steric repulsion associated with strong interdigitation and compression imposed by the thiol molecules between nanoparticles [15].

To obtain an increase of disorder energy of 50 times the attractive Van der Waals energy ($150kT$) when s decreases by $0.4nm$, we need a disordered volume of E7 around the nanoparticle equal to approximately $50nm^3$. This volume is calculated using the heat capacity of 8CB [37], which has been shown to be very close to 5CB, the major component of E7 [12]. This disordered volume ($50nm^3$) corresponds to a shell of disordered liquid crystal of the order of $3.5nm$ surrounding the nanoparticles. This liquid crystal disorder induced in the presence of the alkylthiols around nanoparticles has previously been suggested to explain the observed attraction of these nanoparticles by distorted liquid crystal areas [33, 9]. It is also in agreement with the better trapping observed for the $4.2nm$ nanoparticles with respect to the large ones covered by PVP.

Despite the observed stability of the hybrid systems, the disorder energy may cancel the energy advantage associated with trapping, unlike the case where nanoparticles are trapped in smectic dislocations. On the other hand, in the presence of a shell of disordered CLC around the nanoparticles, we expect spherical aggregations of the nanoparticles to be of smaller disorder energy than ribbons. Spherical aggregation occurs in nematic liquid crystals [31] but not in our distorted CLC. The large difference between the disorder energies of nanoparticles aggregated in spheres and nanoparticles aggregated in ribbons can be compensated by the advantage associated with the localization of the nanoparticles in topological defect cores. It can not be compensated by a localization of nanoparticles only in elastically distorted areas. Two interpretations are possible to explain such a result. First nanoparticles are trapped by topological cores. These cores must be significantly smaller than the nanoparticle size. Thus molecular disorder is induced around the nanoparticles and compression of the nanoparticles spacing occurs. Second, needles may not be trapped by disordered defect cores but by highly distorted cholesteric areas. This would correspond to a metastable state. For example, at a distance D from the center of curvature of a curved nematic, the replacement of the distorted liquid crystal constitutes an elastic advantage that can be roughly estimated to be $4KR^3/D^2$, where K is the bend modulus, of the order of $10^{-11}Jm^{-1}$, and R the particle radius. For particles of radius $2.1nm$, the elastic term is larger than kT (k is Boltzmann's constant) if D is smaller than $10nm$, only two times the nanoparticle diameter. We consequently do not expect a strong stability of trapping by elastically distorted areas for small nanoparticles. The observed stability of the hybrid systems, together with the observation that no modification in the CLC texture is observed by optical microscopy more likely suggests the presence of disordered defect cores as trapping zones.

It has been shown that dislocations at the CLC/air interface, associated with the connection between planar cholesteric helix and homeotropic anchoring, can trap micrometer-sized colloids [29, 22]. We do not believe that they are responsible for the present nanoparticles trapping since only one dislocation is expected per stripe, as shown in figure 1a. This would produce a small density of needles, not consistent with the intense well-defined LSPR usually observed, together with AFM observations of locally large numbers of needles. It thus appears natural to assume that trapping occurs in the area where the cholesteric helix rotates by 90° to become parallel to the CLC/air interface. The trapping area may be constituted by an array of linear disclinations, on average perpendicular to anchoring direction, of width smaller than $5nm$. To account for such a small width, they may correspond to the so-called τ dislocations, of singular nematic director [18]. Due to this small width, we expect formation of nanoparticle chains in a first step. The formation of chains may be followed in a second step by a localization of the nanoparticles around the chains in the elastically deformed zones around the defect cores to produce thicker ribbons (Figure 3b). This is in agreement with LSPR data which can be interpreted by a majority of thin ribbons, chains or ribbons of 2 nanoparticles width, for the more anisotropic zones (Figure 2a). The observation of an experimental extinction anisotropy of hybrid films associated with needles mainly perpendicular to the anchoring orientation for various stripe directions can be interpreted as the signature of an evolution of the 90° helix deformation with the film thickness. The stripe orientation with respect to the anchoring direction is indeed proportionnal to the thickness [2]. This would be true at least for the range of CLC film thickness used in the present study, around twice the cholesteric pitch.

5 Conclusion

We show that CLC films, deformed by competing anchoring at the interfaces, planar unidirectional versus homeotropic, are characterized by the presence of linear traps for nanoparticles. They may be either highly elastically distorted areas, or oriented linear defects with disordered cores of width smaller than a few nanometers, associated with the helix rotation in the vicinity of CLC/air interface, the stability of the hybrid system being in favour of the latter assumption. These traps lead to the formation of nanoparticle needles of width smaller than 5 nanoparticles, of length larger than 50 nanoparticles and characterized by an intense and well-defined anisotropic LSPR response. The spacing between nanoparticles in the needles appears to be about 50 percent smaller than for

the same nanoparticles deposited on the substrate without liquid crystal or trapped within smectic dislocation cores. This may be related to disorder induced by the nanoparticles within CLC elastically distorted areas or around the CLC defect cores. The corresponding disorder is estimated to be a disordered CLC shell around the nanoparticles of some nanometers. In the presence of topological defects, the main parameter controlling the nanoparticles spacing is the size of the defect disordered cores versus the nanoparticle diameter. These results not only gives a limit for the defect size in the CLC distorted films, but they illustrate the role of the defects nature on the crystallographic characteristics of the induced nanoparticle self-assembly. These hybrid films are finally characterized by nanoparticles strictly confined within a specific area, in the thickness of the CLC film. They open a route for a number of hybrid systems of anisotropic optical properties. They also open a path for new anisotropic electro-optical properties, with the possibility, for example, to locally heat within the CLC film trapping area, using the LSPR field excitation. This would allow for further control of the rotation of macroscopic objects under irradiation of molecular motors replacing the classical chiral dopants [10, 6]. We finally note that studying the LSPR of trapped nanoparticles gives new information on the nature of highly deformed areas in these CLC films.

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