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Soumaya Elama Bouzit, Brahim Boualy, Larbi El Firdoussi, Ahmad Mehdi, Abdelkader Outzourhit, et al.. Fast room temperature solution-phase approach for selective synthesis of nanostructured Cu(OH)2, Cu2O and CuO. International Research Journal of Pure and Applied Chemistry, 2015, 8 (3), pp.157-164. 10.9734/IRJPAC/2015/17920. hal-01206431

HAL Id: hal-01206431 https://hal.science/hal-01206431

Submitted on 2 Mar 2022

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International Research Journal of Pure & Applied Chemistry 8(3): 157-164, 2015, Article no.IRJPAC.2015.081 ISSN: 2231-3443

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Fast Room Temperature Solution-phase Approach for Selective Synthesis of Nanostructured Cu(OH)₂, Cu₂O and CuO

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Authors' contributions

This work was carried out in collaboration between all authors. All authors read and approved the final manuscript.

Article Information

DOI: 10.9734/IRJPAC/2015/17920 <u>Editor(s):</u> (1) Luigi Campanella, Research Centre on Environment and Cultural Heritage Chemistry, University of Rome "La Sapienza", Piazzale Aldo Moro, 5, 00185 Rome, Italy. <u>Reviewers:</u> (1) Anonymous, Cady Ayyed University, Morocco. (2) L. Z. Pei, Anhui University of Technology, China. Complete Peer review History: <u>http://www.sciencedomain.org/review-history.php?iid=1051&id=7&aid=9384</u>

Original Research Article

Received 30th March 2015 Accepted 28th April 2015 Published 23rd May 2015

ABSTRACT

A simple room-temperature solution-phase and efficient method for the selective preparation of nanostructured copper hydroxide $(Cu(OH)_2)$, copper oxide (II) CuO or Cuprous oxide (Cu_2O) from a Cu^{2+} solution was developed. Mono-disperse cubic Cu_2O were obtained by sodium hydroxide addition to this solution in the presence of ethylene glycol (EG) while monoclinic CuO nanoparticles were obtained in the presence poly-ethylene glycol (PEG) as dispersant agent. By adjusting the amounts of EG, octahedron and truncated octahedron aggregates of copper oxide (I) nanoparticles (Cu_2O) can be formed, whereas in case of PEG, $Cu(OH)_2$ nanowires, CuO sheetshaped nanoparticles () can be obtained. These observations were confirmed by X-ray diffraction, infrared spectroscopy and scanning electron microscopy.

It appears that the EG and PEG act as size-controlling and growth-directing agents in addition to their role as a dispersant.

Keywords: Nanostructured Cu₂O; CuO; ethylene glycol; polyethylene glycol; solution-phase; room temperature.

1. INTRODUCTION

Semiconductor nanoparticles have attracted a considerable attention due to their fundamental importance and potential future applications. Among them, copper(I) oxide (Cu₂O) is a p-type semiconductor with a direct band gap of 2.14 eV, which makes it promising as a low-cost material in applications such as catalysis and photovoltaic devices [1]. Copper(II) oxide (CuO), which is also a p-type semiconductor [2], with an indirect band gap of 1.2 to 1.5 eV, is widely used as a heterogeneous catalyst because of its high activity and selectivity in oxidation and reduction reactions [3]. This material was also used in Liion batteries as an active electrode material [4], in field emission [FE] emitters, gas sensors [5], and solar cells [6].

In recent years, several efforts were devoted to synthesize couprous oxide (Cu₂O) and cupric oxide (CuO) nanoparticles with different morphologies using several methodologies such as thermal oxidation [7], aqueous solution [8], hydrolysis [9], template-based sol-gel [10], micelle reverse technique [11] and electrochemical methods [12]. However, and in order to control the morphology of nanostructured copper oxides, these oxides were often prepared by high-temperature approaches [13] such as hydrothermal [14], solvothermal [15] and microwave-hydrothermal methods [16]. Therefore, the development of cost-effective synthesis methods which require less energy and material still remains a challenge in this field. It is known that the size of the nanoparticles can be controlled easily through the use of surfactants in the reaction mixture [17]. Surfactants with long hydrocarbon chains and hydrophobic ends are important in manipulating particle sizes [18]. Various surfactants can alter the shape, size and other surface properties of the nanoparticles to different extents depending upon their molecular structure i.e. the nature of the head group, the length of the hydrophobic tail and the type of counter ions. The termination of the nanoparticle growth is controlled by the diffusion and the

attachment rates of surfactants on the nanoparticle surface. Thus, surfactant supported stabilization of the nanoparticles in aqueous solution has been demonstrated to be one of the most effective nanoparticle synthesis method [19-20].

In the present work, we report a simple roomtemperature solution-phase route for the preparation, of mono-disperse cubic Cu₂O and monoclinic CuO nanoparticles in the presence of ethylene glycol (EG) and poly-ethylene glycol (PEG) respectively as size controlling and growth directing agent. The influence of the dispersant agent's concentration on the structure and morphology of the obtained Cu₂O and CuO nanoparticles was investigated. The prepared nanoparticles were characterized by X-ray diffraction measurements (XRD). Fourier Transform Infrared Spectroscopy (FT-IR). scanning electron microscopy (SEM) and Nitrogen adsorption (BET).

2. EXPERIMENTALS DETAILS

2.1 Materials

The chemicals used in the present work are copper fine powder 63μ m (99.5% purity Aldrich), Hydrochloric acid (37%, Aldrich), Hydrogen peroxide (30%, Aldrich), ethylene glycol (anhydrous, 99.8%, Aldrich), polyethylene glycol (Mn 6000) (99.9% purity, Sigma Aldrich), Sodium hydroxide (anhydrous, ≥98% purity, Sigma Aldrich), Ethanol (absolute, ≥98% purity, Aldrich) and deionized water.

2.2 Method

2.2.1 Synthesis of Cu₂O

0.5 g of Copper fine powder was first dissolved in 10 mL of 3.6 M HCl. A chosen volume (10 mL, 20 mL and 30 mL) of ethylene glycol (EG) was then added to the as-prepared solution depending on the desired EG concentration. The mixture was vigorously stirred for 20 min at room temperature and 2 mL of H_2O_2 (30%) was added drop-wise for 20 to 30 minutes to give a light blue colored solution. Then, 2.5 g of NaOH dissolved in deionized water (30 mL) was added drop-wise under constant stirring for 20 min, producing an intense brown colored precipitate. The mixture is filtered, washed several times with deionized water and ethanol and finally dried for 12 hours at 60°C. The formation of Cu₂O nanoparticles is confirmed by the reddish color of the product.

2.2.2 Synthesis of CuO

Copper powder was dissolved in hydrochloric acid as described above. In this case, polyehtylene glycol (PEG) aqueous solution was used as dispersant agent. Upon addition of NaOH, following the procedure described above, a brown-black colored precipitate was obtained within 10 to 15 minutes. The precipitate was then centrifuged, filtered, washed several times with ethanol and deionized water and dried for 12 hours at 60°C. The CuO nanostructures were obtained as black colored powder.

2.3 Characterization

The porous nature of the obtained samples was investigated using physical adsorption of nitrogen at liquid-nitrogen temperature (-196°C) with an automatic volumetric sorption analvzer (Quantachrome instruments). Prior to the measurement, the samples were degassed at 120°C for 12 h under vacuum. The specific surface areas were determined according to the Brunauer-Emmett-Teller (BET) method. Phases and structural characterizations the of synthesized powders were characterized by Xray powder diffraction (XRD) using a "Philips X'pert MPD " diffractometer operating at 40 kV and 30 mA, equipped with a Cu anticathode of wavelength (λ = 1.5406 Å). The collected nanoparticles samples were also analyzed by a scanning electron microscope (SEM) type "JEOL JSM – 5500 » operated at 10 kV. Chemical prepared information on bonding the nanostructures was studied with Fourier Transform Infrared Spectroscopy (FT-IR).

3. RESULTS AND DISCUSSION

 Cu_2O and CuO nanoparticles were prepared at room temperature from copper powder dissolved in hydrochloric acid in the presence of a chosen volume of EG (noted Cu_2O -10 mL, Cu_2O -20 mL and Cu₂O-30 mL) or a given volume of 0.16M PEG solution (noted CuO-20 mL and CuO-30 mL). The synthesis was conducted in the presence of sodium hydroxide solution and H_2O_2 as oxidant. The role of each dispersant on the synthesis of both types of nanoparticles have been reported in this work. In case of ethylene glycol as a dispersant agent, the X-ray diffraction patterns of the prepared samples show only peaks that are characteristic of Cu₂O oxide (Fig. 1a) with no additional impurity peaks or phases. The four diffraction are assigned to the (110), (111), (200) and (220), are characteristics of the cubic cuprous oxide as given by JCPDS 78-2076 data software. The formation of Cu₂O was also be confirmed by the reddish color of the product. It could be seen that the XRD patterns for all the three samples are the same except for the variation in the intensity and the width of some of the peaks.

Table 1 shows the values of crystallite size calculated by the Scherer's formula based on the major peak located at 36.40 degree. It was found that the XRD crystallite size was in the range of 17–24 nm for Cu₂O nanoparticles. The increase of the EG volume in the experiments promotes the formation of smaller crystallites and narrower size distribution of Cu₂O nanoparticles.

Fig. 1b shows the XRD patterns of the samples obtained in the presence of selected volumes of 0.16M PEG solution (10 mL, 20 mL and 30 mL). For the first sample obtained by adding 10 mL of PEG solution, all the observed diffraction peaks correspond to the lattice planes of Cu(OH)₂ with the orthorhombic symmetry (JCPDS No. 35-0505). This is also confirmed by the characteristic blue color of Cu(OH)₂. When the PEG solution volume was increased to 20 mL and 30 mL, the XRD patterns of the obtained samples (CuO-20mL PEG and CuO-30mL PEG) are attributed to the monoclinic CuO phase (JCPDS No. 48-1548). In addition, the crystallite size of CuO becomes smaller with increasing PEG solution volume (See Table 1).

The morphology of the nanostructured Cu_2O and CuO samples prepared by the chemical precipitation method was characterized by SEM. As shown in Fig. 2a, mono-disperse and regular octahedral Cu2O particles are obtained after adding 10 mL of EG. By increasing the EG volume to 20 mL, the Cu2O crystal morphology is changed from nano-octahedron



Fig. 1. XRD patterns of polycrystalline Cu₂O and CuO nanoparticles prepared with (a) EG as dispersant and (b) PEG as dispersant

Table 1. Average crystallite size of Cu₂O and CuO nanoparticles calculated by the Scherer's formula

Nanoparticles	2 0 (°)	Crystals orientation	B(°)	The crystallites size D(nm)
Cu ₂ O-10 mL EG	36,41	(111)	0.3541	23.61
Cu ₂ O-20 mL EG	36,40	(111)	0.3936	21
Cu ₂ O-30 mL EG	36,44	(111)	0.4723	17.71
CuO-20 mL PEG	35,35	(002)	0.3085	52.97
	38,47	(111)	0.3357	35.53
CuO-30 mL PEG	35,49	(002)	0.3097	21
	38,70	(111)	0.3377	24.45

to nano- truncated octahedron (See Fig. 2b). However, when increasing the volume of EG to 30 mL, some spherical nanoparticles were observed in addition to nano-truncated octahedrons (Fig. 2c). On the basis of the above results, it is clear that the shape and structure of the final particles, which evolved from nanooctahedron to nano-truncated octahedron and to nano-spheres, change with increasing the EG volume. These results suggest that these properties can be controlled by adjusting the EG amounts in the reaction mixture. This is in good agreement with the results of Li et al. [21] who reported that the shape of Cu₂O nano-particles evolves from cubic to octahedral and spherical shape when synthesized in hydrophilic ionic liquids. In this case, preferential adsorption was evoked as the mechanism for the morphology change [21]. Sun and his co-workers [22] attributed the observed changes in the morphology of Cu₂O nanoparticles to selective adsorption of ethanol molecules on different positions of Cu₂O octahedral particles.

Similarly, in the case of CuO nano-partciles, and as shown in Fig. 3a, the formed $Cu(OH)_2$ exhibits

long uniform nanowire shape. By increasing the PEG aqueous solution volume to 20 mL and 30 mL, the morphology of the produced CuO changed from nanowires to nanosheets (Figs. 3b and 3c).

While the changes of morphology and shape can be explained by the preferential adsorption of the hydrophilic parts of the dispersant on the surfaces of the growing nanoparticles, the changes in the stoichiometry may be accounted for by the reducing strength of the dispersant. It appears that EG is an effective reducing agent than PEG since under alkali condition (addition of NaOH), Cu₂O is formed from Cu(OH)₂ in the presence of EG, while CuO is formed form Cu(OH)₂ In the presence PEG.

The samples were further examined and compared by FT-IR analysis (Fig. 4). The single phase and crystalline nature of the prepared cuprous oxide Cu_2O-V_{EG} (V_{EG} = 10mL, 20mL or 30 mL) is further confirmed by the characteristic band at 634 cm⁻¹ (Fig. 4a), which represents the optically active lattice vibration mode of Cu_2O [23]. The observed broad band at 3437 cm⁻¹ is

attributed to the symmetrical stretching vibration v(OH) of hydrogen bonded OH [24] and shows clearly the presence of water molecules. A band in the region of 1640 cm⁻¹ is related to the bending vibration for H₂O [25].

The FT-IR spectra of $Cu(OH)_2$ and CuO are shown in Fig. 4b. In the case of CuO, there are three absorption bands at 470 cm⁻¹, 600 cm⁻¹ and a strong one at 496 cm⁻¹ which is attributed to

the stretching mode of the Cu–O bond [26]. The broad peaks located at 1423 cm⁻¹ can be attributed to the vC–O vibration mode. The two bands at 3437 cm⁻¹ and 1640 cm⁻¹ are the characteristic peaks of the O–H stretching and H₂O bending modes respectively. By increasing the volume of PEG aqueous solution from 20 mL to 30 mL, all absorption bands between 1000 cm⁻¹ and 2000 cm⁻¹ disappear.



Fig. 2. SEM images of Cu₂O-10EG (a) and Cu₂O-20EG (b) nanoparticles



Fig. 3. SEM images of CuO-10PEG (a), CuO-20PEG (b) and CuO-30PEG (c) nanoparticles



Fig. 4. FTIR spectra of Cu₂O and CuO nanoparticles prepared with (a) EG dispersant and (b) PEG dispersant

The comparison of the FT-IR spectra of Cu(OH)₂ and CuO indicates clearly that the band centered at 3437 cm⁻¹ for CuO, shifts and splits into two bands at 3570 and 3310 cm⁻¹ in the case of Cu(OH)₂. These broad peaks are attributed to the stretching modes (vOH) of hydrogen-bonded hydroxyl groups. The band at 3310 cm⁻¹ is usually attributed to bound OH, while the band at 3570 cm⁻¹ is assigned to free hydroxyl groups [27]. The broad and overlapping band in the region of 1650 cm⁻¹ is related to the bending vibration of H₂O [25]. The peak at 700 cm⁻ corresponds to the bending vibration of hydrogen-bonded OH groups ($\delta O-H$) and the strong absorption at 470 cm⁻¹ is due to Cu–O–H bending (δ Cu-O-H) and Cu-O stretching vibrations (vCu-O).

The textural characteristics of all samples were obtained by N₂ adsorption–desorption isotherms measurements. The pore volume and the specific surface area were calculated by BET method from the adsorption branch. The BET surface area and pore volume of CuO nanosheets are in the range of 30-37 m²/g and 0.04-0.06 cm³/g respectively, which are slightly larger than those of Cu(OH)₂ nanowires (22 m²/g and 0.04 cm³/g). Comparatively, the Cu₂O shows low BET surface area (10-12 m²/g) and small pore volume (0.02 cm³/g).

4. CONCLUSION

This work has clearly shown that the structure as well as the morphology and the shape of copper oxide nanostructures can be controlled through an appropriate choice of a dispersant. X-ray diffraction and SEM analysis showed that Cu_2O octahedron-shaped nanoparticles, Cu_2O truncated octahedron nanoparticles when EG was used as a dispersant agent, while $Cu(OH)_2$ nanowires and CuO Nano sheets are obtained when PEG was used for this purpose. These suggest that these two dispersing agents act also as growth directing agents.

This study may provide a new simple and fast solution-phase method to synthesize nanomaterials with high purity, great stability and variety of shapes without the need for specialized instruments and extreme conditions. The rate of particle aggregation is a major factor that controls the morphology and crystallinity of the final product.

COMPETING INTERESTS

Authors have declared that no competing interests exist.

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