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Multiscale identification of the random elasticity field at mesoscale of a heterogeneous microstructure using multiscale experimental observations \hat{z}

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Abstract

This paper deals with a multiscale statistical inverse method for performing the experimental identification of the elastic properties of materials at macroscale and at mesoscale within the framework of a heterogeneous microstructure which is modeled by a random elastic media. New methods are required for carrying out such multiscale identification using experimental measurements of the displacement fields carried out at macroscale and at mesoscale with only a single specimen submitted to a given external load at macroscale. In this paper, for a heterogeneous microstructure, a new identification method is presented and formulated within the framework of the three-dimensional linear elasticity. It permits the identification of the effective elasticity tensor at macroscale, and the identification of the tensorvalued random field, which models the apparent elasticity field at mesoscale. A validation is presented first with simulated experiments using a numerical model based on the hypothesis of 2D-plane stresses. Then, we present the results given by the proposed identification procedure for experimental measurements obtained by digital image correlation (DIC) on cortical bone

Key words: Multiscale identification, heterogeneous microstructure, random elasticity field, mesoscale, multiscale experiments

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1. Introduction

The inverse methods devoted to the experimental identification of the elastic properties of materials at the macroscale and/or mesoscale have been extensively studied. The experimental identification of microstructural morphology by image analysis began in the 1980s [35, 36, 37], and it has led to significant advances in the identification of mechanical properties. We refer the reader, for instance, to [16] for the identification of elastic moduli from displacement-force boundary measurements, to [38, 57, 58, 31, 30, 47, 8, 33, 45, 46, 48, 34] for works devoted to the displacement field measurements in micro and macromechanics using digital image correlation in 2D and 3D, including software developments, to [32, 11] for aspects concerning multiscale full-field measurements, to [10] for aspects concerning measurement errors in digital image correlation technique, to [22, 23, 9] for inverse problems in elasticity and for the identification of elastic parameters by using displacement field measurements, to [13, 12, 3, 41, 2, 4] for identification methods of mechanical parameters from full-field measurements in linear and nonlinear elasticity, and to [7, 24] for stochastic aspect of random media using a moving-window technique.

Concerning the identification of stochastic models, the methodologies for statistical inverse problems in finite and infinite dimensions are numerous and have given rise to numerous studies and publications. These methods make extensive use of the formulations and the tools of the functional analysis for the boundary value problems as well as those of probability theory, including mathematical statistics (finite and infinite dimensional cases). Concerning the mathematical statistics, one can refer to [40, 49] and to [15, 59, 55, 39] for the general principles on the statistical inverse problems. Early works on the statistical inverse identification of stochastic fields for random elastic media, using partial and limited experimental data, have primarily be devoted to the identification of the hyperparameters of *prior* stochastic models (such as the spatial correlation scales and the level of statistical fluctuations) [19, 20, 1, 18, 17, 29], and then, methodologies have recently been proposed for the identification of general stochastic representations of random fields in high stochastic dimension [52, 53, 44, 43]. Those probabilistic/statistical methods are able to solve the statistical inverse problems related to the identification of *prior* stochastic models for the apparent elastic fields at mesoscale. Nevertheless, such experimental identification, which is carried out

using measurements of the displacement fields at macroscale and mesoscale on a single specimen submitted to a given external load at macroscale, requires new methods for identifying the statistical mean value of the random apparent elasticity tensor at mesoscale and the other hyperparameters controlling its *prior* stochastic model as, for instance, the spatial correlation lengths and the hyperparameters allowing the statistical fluctuations of the stochastic field to be controlled.

The present paper is not devoted to the presentation of a new stochastic model for representing the random elasticity field of a heterogeneous microstructure at mesoscale. This work proposes a new methodology for identifying a *prior* stochastic model of such random elasticity field using multiscale experimental data performed on a single specimen. Taking into account the major difficulty of this challenging statistical inverse problem induced by the high stochastic dimension of the tensor-valued random field to be identified, a parameterized *prior* stochastic model can be used in order to obtain a well-posed statistical inverse problem. Taking into account the mathematical properties of the tensor-valued random elasticity field of the heterogeneous elastic microstructure at mesoscale, the model proposed in [50] has been used for validating the multiscale identification.

In this paper, an identification method and its validation are presented using multiscale measurements. A new multiscale statistical inverse method is introduced for a heterogeneous microstructure within the framework of the three-dimensional linear elasticity. This method permits both the identification of the effective elasticity tensor at macroscale and the identification of the tensor-valued random field that models the apparent elasticity field at mesoscale. The methodology to perform the experimental identification of both the mean part and the statistical fluctuations of a stochastic model of the elasticity field at mesoscale of a heterogeneous microstructure using experimental data for a single specimen, has not been proposed in the literature. It should be noted that the random elasticity field is not a real valued random field, but a tensor-valued random field for which the entries of the tensor cannot be identified separately (entry by entry) due to algebraic properties that relate the entries. If a digital image correlation method was only used at mesoscale, then the mean part of the stochastic model at mesoscale could not be identified. Conversely, if a digital image correlation method was only used at macroscale, then the statistical fluctuations of the stochastic model at mesoscale could not be identified. Since, by assumptions, only a single specimen is experimentally tested, a multiscale digital image correlation must simultaneously be

used at mesoscale and at macroscale for the specimen submitted to a given load. Such experimental procedure has specially been implemented and validated (see [42]). Therefore, the complete statistical information (mean and fluctuations) of the random elastic field at mesoscale must be transferred to the macroscale in order to use the macroscale measurements for identifying the mean part. Such a transfer of information is carried out by using a stochastic homogenization. These considerations have led the authors to propose an innovative and new multiscale experimental identification procedure.

In the presented work, a fundamental mechanical property is used: there exists a Representative Volume Element (RVE) that allows for a separation of the mesoscale with the macroscale. In the context of homogenization in micromechanics of heterogeneous materials, it is proved (see for instance [51] and references included) that, if such a RVE exists, then there is no statistical fluctuations at macroscale (with a very high level of probability), which means that there exists a scale separation. By definition of a RVE, it is necessary that the random elasticity field at mesoscale be ergodic and consequently, must be homogeneous (stationary). For the class of materials considered in this work (cortical bone) this hypothesis is experimentally verified. The statistical fluctuations of the cortical bone at macroscale are negligible, and therefore, there exist a RVE, and thus the hypothesis use (homogeneous random field at mesoscale) is reasonable. In addition, it should be noted that, if such a hypothesis was not verified, and taking into account that only one specimen is tested, then the statistical inverse problem would be an ill-posed problem, because the statistical averaging could not be done, due to the lack of the ergodic property.

It is assumed that the experimental measurements of the displacement field are available at macroscale and at mesoscale. The *prior* stochastic model is a non-Gaussian tensor-valued random field adapted to the algebraic properties of the 3D-elasticity field and to the corresponding stochastic elliptic boundary value problem. The hyperparameters of the *prior* stochastic model of the apparent elasticity random field at mesoscale, are its statistical mean value, its spatial correlation lengths and its level of statistical fluctuations. The identification of such a stochastic model at mesoscale requires the knowledge of the effective elasticity tensor at macroscale, as well as the measurements of the displacements field at the two scales, simultaneously, for one given specimen submitted to a given static external load (as explained before).

The theory is presented for the three-dimensional case. A numerical validation is presented first for the 2D plane stresses using simulated experiments. Then, we present the results given by the proposed identification procedure for experimental measurements obtained on cortical bone.

In all this paper, the framework of the three-dimensional linear elasticity is used. Consequently, and for instance, the terminology "strain tensor" will be used instead of "linearized strain tensor", etc.

2. Multiscale experimental configuration

The specimen (whose microstructure is complex and heterogeneous at microscale) occupies a bounded macroscopic domain Ω^{macro} in \mathbb{R}^3 (the three-dimensional Euclidean space). A surface force field, f^{macro} , is applied on a part Σ^{macro} of the boundary $\partial \Omega^{\text{macro}}$ of Ω^{macro} . The other part Γ^{macro} of $\partial \Omega^{\text{macro}}$ is fixed such that there is no rigid body displacement. At macroscale, the measured displacement field in Ω^{macro} is denoted by $\mathbf{u}_{\text{exp}}^{\text{macro}}$ and its associated tensor-valued strain field is denoted by $\varepsilon_{\exp}^{\rm macro}$.

Let Ω^{meso} be a subdomain of the specimen at mesoscale. It is assumed that Ω^{meso} is a representative elementary volume (REV). Let $\partial \Omega^{\text{meso}}$ be the boundary of Ω^{meso} . Let $\mathbf{u}_{\text{exp}}^{\text{meso}}$ be the experimental measurement of the \mathbb{R}^3 -valued displacement field in Ω^{meso} at mesoscale. The associated tensor-valued strain field is denoted by $\epsilon_{\text{exp}}^{\text{meso}}$. It is assumed that the experimental measurements of displacement field $\mathbf{u}_{\rm exp}^{\rm meso}$ are obtained only for one subdomain Ω^{meso} related to one specimen. The volume average at mesoscale, $\underline{\epsilon}_{\text{exp}}^{\text{meso}}$, of $\epsilon_{\text{exp}}^{\text{meso}}$ is introduced such that

$$
\underline{\varepsilon}_{\exp}^{\text{meso}} = \frac{1}{|\Omega^{\text{meso}}|} \int_{\Omega^{\text{meso}}} \varepsilon_{\exp}^{\text{meso}}(\mathbf{x}) \, d\mathbf{x}, \tag{1}
$$

in which **x** is the generic point in \mathbb{R}^3 and where dx is the volume element. The level of statistical fluctuations (around the volume average $\frac{R_{\text{exp}}}{R_{\text{exp}}}$) of $\epsilon_{\text{exp}}^{\text{meso}}$ is estimated by the quantity $\delta_{\text{exp}}^{\text{meso}}$ defined by

$$
\delta_{\exp}^{\text{meso}} = \frac{\sqrt{V_{\exp}^{\text{meso}}}}{\|\underline{\mathbf{\varepsilon}}_{\text{exp}}^{\text{meso}}\|_{F}},\tag{2}
$$

in which

$$
V_{\rm exp}^{\rm meso} = \frac{1}{|\Omega^{\rm meso}|} \int_{\Omega^{\rm meso}} ||\varepsilon_{\rm exp}^{\rm meso}(\mathbf{x}) - \varepsilon_{\rm exp}^{\rm meso}||^2 \, d\mathbf{x} \tag{3}
$$

and where $||T||_F$ is the Frobenius norm such that, for any second-order tensor $T = \{T_{ij}\}_{ij}$, one has

$$
||T||_F^2 = \sum_{i=1}^3 \sum_{j=1}^3 T_{ij}^2.
$$
 (4)

3. Multiscale statistical inverse problem

In a first subsection, numerical indicators used in the identification procedure are defined. In the second subsection, the identification procedure is define for the case of 2D plane stresses.

3.1. Construction of indicators for the identification procedure

At macroscale, a deterministic boundary value problem is introduced for a threedimensional linear elastic medium, which models the specimen in its experimental configuration (geometry, surface forces and Dirichlet conditions). At this macroscale, the constitutive equation involves a *prior* model for the fourth-order elasticity tensor $C^{\text{macro}}(\mathbf{a})$ that is parameterized by a vector \mathbf{a} . For a three-dimensional anisotropic elastic material, **a** represents the 21 constants of the fourth-order elasticity tensor. The boundary value problem is formulated in displacement and the solution is denoted by $\mathbf{u}^{\text{macro}}$ (deterministic \mathbb{R}^3 -valued displacement field at macroscale, which depends on **a**). The tensor-valued strain field associated with **u**^{macro} is denoted by $\varepsilon^{\text{macro}}$ and depends on **a**. Tensor $C^{\text{macro}}(\mathbf{a})$ is unknown and must experimentally be identified, which means that the parameter **a** must be identified using the measurements of the displacement field at macroscale. Consequently, a first numerical indicator $\mathcal{I}_1(a)$ is introduced in order to quantify the distance between $\varepsilon_{\text{exp}}^{\text{macro}}$ and $\varepsilon^{\text{macro}}(a)$. For a fixed value of parameter **a**, this indicator is defined by

$$
\mathcal{I}_1(\mathbf{a}) = |||\varepsilon_{\exp}^{\text{macro}} - \varepsilon^{\text{macro}}(\mathbf{a})|||^2, \qquad (5)
$$

in which

$$
|||\varepsilon_{\exp}^{\text{macro}} - \varepsilon^{\text{macro}}(\mathbf{a})|||^2 = \int_{\Omega^{\text{macro}}} ||\varepsilon_{\exp}^{\text{macro}}(\mathbf{x}) - \varepsilon^{\text{macro}}(\mathbf{x}; \mathbf{a})||_F^2 d\mathbf{x}.
$$
 (6)

At mesoscale, two additional numerical indicators, $\mathcal{I}_2(\mathbf{b})$ and $\mathcal{I}_3(\mathbf{a}, \mathbf{b})$, are constructed to identify the hyperparameter **b** involved in the *prior* tensor-valued random field, **C** meso(**b**), which models the apparent elasticity random field that is considered as the restriction to subdomain Ω^{meso} of a statistically homogeneous random field $\{C^{\text{meso}}(x; \mathbf{b}), x \in \mathbb{R}^3\}.$

Concerning the construction of the numerical indicator $\mathcal{I}_2(\mathbf{b})$, a stochastic boundary value problem is introduced for a three-dimensional linear elastic random medium occupying subdomain Ω^{meso} , and for which the apparent elasticity random field is **C** meso(**b**). This stochastic boundary value problem is formulated in displacement, and the solution is the random displacement field denoted by **U** meso (\mathbb{R}^3 -valued random field) with the Dirichlet condition $\mathbf{U}^{\text{meso}} = \mathbf{u}_{\text{exp}}^{\text{meso}}$ on boundary ∂Ω meso. The tensor-valued random strain field associated with **U** meso is denoted by ϵ^{meso} . For any given hyperparameter **b**, the numerical indicator $\mathcal{I}_2(\mathbf{b})$ is defined by

$$
\mathcal{I}_2(\mathbf{b}) = \int_{\Omega^{\text{meso}}} (\delta^{\text{meso}}(\mathbf{x}; \mathbf{b}) - \delta^{\text{meso}}_{\text{exp}})^2 d\mathbf{x}, \qquad (7)
$$

in which $\delta_{\text{exp}}^{\text{meso}}$ is defined by Eq. (2) and where $\delta^{\text{meso}}(x; \mathbf{b})$ is defined by

$$
\delta^{\text{meso}}(\mathbf{x}; \mathbf{b}) = \frac{\sqrt{V^{\text{meso}}(\mathbf{x}; \mathbf{b})}}{\|\underline{\varepsilon}^{\text{meso}}(\mathbf{b})\|_{F}},
$$
\n(8)

and where $\underline{\mathbf{s}}^{\text{meso}}(\mathbf{b})$ and $V^{\text{meso}}(\mathbf{x}; \mathbf{b})$ are such that

$$
\underline{\varepsilon}^{\text{meso}}(\mathbf{b}) = \frac{1}{|\Omega^{\text{meso}}|} \int_{\Omega^{\text{meso}}} \varepsilon^{\text{meso}}(\mathbf{x}; \mathbf{b}) \, d\mathbf{x}, \tag{9}
$$

$$
V^{\text{meso}}(\mathbf{x}; \mathbf{b}) = E\{\|\varepsilon^{\text{meso}}(\mathbf{x}; \mathbf{b}) - \underline{\varepsilon}^{\text{meso}}(\mathbf{b})\|_F^2\},\tag{10}
$$

in which E is the mathematical expectation. It should be noted that, for all \mathbf{b} , $\underline{\epsilon}^{\text{meso}}(\mathbf{b}) = \underline{\epsilon}^{\text{meso}}_{\text{exp}}$. Finally, the numerical indicator $\mathcal{I}_3(\mathbf{a}, \mathbf{b})$ quantifies the distance between the elasticity tensor $C^{\text{macro}}(\mathbf{a})$ used in the deterministic boundary value problem at macroscale, and the effective tensor $C^{\text{eff}}(b)$ calculated by homogenization of the mesoscale stochastic model in subdomain Ω^{meso} which is a REV. We then have

$$
\mathcal{I}_3(\mathbf{a}, \mathbf{b}) = ||C^{\text{macro}}(\mathbf{a}) - E\{C^{\text{eff}}(\mathbf{b})\}||_F^2.
$$
 (11)

The identification of parameters **a** and **b** that describe the stochastic model of the apparent elasticity random field $C^{meso}(\mathbf{b})$ at mesoscale is obtained by solving a multi-objective optimization problem for the three indicators $\mathcal{I}_1(\mathbf{a})$, $\mathcal{I}_2(\mathbf{b})$, and $\mathcal{I}_3(\mathbf{a}, \mathbf{b}).$

3.2. Identification procedure for 2D plane stresses

In this section, we present the particular case of 2D plane stresses, which will be the configuration of the experimental measurements presented in Section 5.

3.2.1. Prior stochastic model of the apparent elasticity random field for 2D plane stresses

In this subsection, a *prior* probabilistic model of the apparent elasticity random field is constructed for the case of 2D plane stresses. This *prior* probabilistic model will be used in the next sections (i) for the validation of the method in 2D plane stresses (Section 4), and (ii) for the application of the method with experimental measurements obtained through DIC method (Section 5).

At mesoscale, the *prior* stochastic model of the apparent elastic random field **C** meso is indexed by subdomain Ω^{meso} which is assumed to be a REV. We are interested in using a stochastic representation of **C** meso with a minimum of hyperparameters (dimension of vector **b**), which is adapted to the elliptic boundary value problem corresponding to the linear elastostatic problem. Parametric stochastic models have been proposed for real-valued stochastic fields [24, 6, 5, 21], and for non-Gaussian tensor-valued random fields in the framework of the heterogeneous anisotropic linear elasticity [50, 51, 56, 54, 14], with important enhancements to take into account the material symmetry and the existence of elasticity bounds [25, 26, 27, 28]. Hereinafter, the stochastic model for the apparent elastic tensorvalued random field C^{meso} is based on the model proposed in [50] for a heterogeneous anisotropic microstructure at the mesoscale.

In using the Voigt notation, for all **x** fixed in Ω meso, the random fourth-order elasticity tensor $C^{meso}(x)$ can be represented by a (6×6) real random matrix. The strain random vector is then denoted by $(\varepsilon_{11}, \varepsilon_{22}, 2 \varepsilon_{12}, \varepsilon_{33}, 2 \varepsilon_{23}, 2 \varepsilon_{13})$ and the associated random stress vector is denoted by $(\sigma_{11}, \sigma_{22}, \sigma_{12}, \sigma_{33}, \sigma_{23}, \sigma_{13})$. Such numbering of those random vectors (that is not usual) has been chosen for the sake of simplicity in 2D plane stresses, for which the (3×3) compliance random matrix $[\mathbf{S}^{2D}(\mathbf{x})]$ corresponds to the first (3×3) block of the (6×6) compliance random matrix $[\mathbf{S}^{\text{meso}}(\mathbf{x})] = [\mathbf{C}^{\text{meso}}(\mathbf{x})]^{-1}$.

The *prior* stochastic model of C^{meso} is then constructed in choosing $[S^{meso}]$ = $\{[\mathbf{S}^{\text{meso}}(\mathbf{x})], \mathbf{x} \in \Omega^{\text{meso}}\}$ in the set SFE⁺ (defined in [50, 51, 54]) of the non-Gaussian second-order stochastic fields with values in the set of all the positivedefinite symmetric (6×6) real matrices denoted by $M_6^+(\mathbb{R})$. For all **x** in Ω^{meso} , the mean value is a given matrix $[\underline{S}^{\text{meso}}] = E\{[\mathbf{s}^{\text{meso}}(\mathbf{x})]\}$ in $M_6^+(\mathbb{R})$, which is assumed independent of **x**. The $M_6^+(\mathbb{R})$ -valued random field $\{[\mathbf{s}^{meso}(\mathbf{x})], \mathbf{x} \in \Omega^{meso}\}\$ is parameterized by the symmetric matrix $[\underline{S}^{\text{meso}}]$, three spatial correlation lengths

denoted by ℓ_1 , ℓ_2 , ℓ_3 , and one positive dispersion coefficient δ that controls the level of statistical fluctuations. Finally, an explicit generator of independent realizations (sample paths) of such a random field is completely defined in [50].

For the case of 2D plane stresses, for all **x** in Ω^{meso} , the random matrix $[\mathbf{S}^{2D}(\mathbf{x})]$ is defined as the left upper (3×3) block matrix of the random matrix $[\mathbf{S}^{\text{meso}}(\mathbf{x})]$. Consequently, the $M_3^+(\mathbb{R})$ -valued random field $\{[\mathbf{s}^{2D}(\mathbf{x})], \mathbf{x} \in \Omega^{\text{meso}}\}$ is parameterized by the symmetric matrix $[\underline{S}^{2D}]$ in $\mathbb{M}_3^+(\mathbb{R})$ (left upper (3×3) block matrix of $[\underline{S}^{\text{meso}}]$), one spatial correlation length $\ell = \ell_1 = \ell_2$ and dispersion parameter δ . The *prior* model of the apparent elasticity random field $[\mathbf{C}^{2D}] = \{ [\mathbf{C}^{2D}(\mathbf{x})], \mathbf{x} \in \Omega^{\text{meso}} \}$ with values in $M_3^+(\mathbb{R})$ is thus constructed, for all **x** in Ω^{meso} , as

$$
\left[\mathbf{C}^{2D}(\mathbf{x})\right] = [\mathbf{S}^{2D}(\mathbf{x})]^{-1}.
$$
 (12)

Consequently, the hyperparameter **b** of the *prior* stochastic model of the apparent elasticity random field $[\mathbf{C}^{2D}(\mathbf{b})] = \{[\mathbf{C}^{2D}(\mathbf{x}; \mathbf{b})], \mathbf{x} \in \Omega^{\text{meso}}\}$ are $\mathbf{b} = (\delta, \ell, \mathbf{c})$ algebraically independent parameters spanning the symmetric matrix $[\underline{S}^{2D}]$).

3.2.2. Multi-objective optimization problem for 2D plane stresses

The identification of hyperparameter **b** is carried out in searching for the optimal values a^{macro} and **b**^{meso} which solve the following multi-objective minimization problem

$$
(\mathbf{a}^{\text{macro}}, \mathbf{b}^{\text{meso}}) = \underset{\mathbf{a} \in \mathcal{A}^{\text{macro}}, \mathbf{b} \in \mathcal{B}^{\text{meso}}}{\arg \min} \mathcal{I}(\mathbf{a}, \mathbf{b}) ,
$$
(13)

where A^{macro} and B^{meso} are the sets of the admissible values for **a** and **b**, and where the components $\mathcal{I}_1(\mathbf{a}), \mathcal{I}_2(\mathbf{b})$ and $\mathcal{I}_3(\mathbf{a}, \mathbf{b})$ of vector $\mathcal{I}(\mathbf{a}, \mathbf{b})$ are defined by Eqs. (5), (7) and (11). Concerning the multi-objective optimization, if the material at macroscale is assumed to be transverse isotropic and in 2D plane stresses, then the vector-valued parameter $\mathbf{a} = (E_T^{\text{macro}}, \nu_T^{\text{macro}})$ corresponds to the transverse Young modulus and the transverse Poisson coefficient at macroscale. The hyperparameter **b** will be defined later. The stochastic boundary value problem at mesoscale is solved in using the finite element method. The Monte Carlo numerical method is used to calculate the indicators $\mathcal{I}_2(\mathbf{b})$ and $\mathcal{I}_3(\mathbf{a}, \mathbf{b})$ with a number n_s of statistical independent realizations of the random elasticity field $\{[\mathbf{C}^{2D}(\mathbf{x})], \mathbf{x} \in \Omega^{\text{meso}}\}.$ The multi-objective optimization problem defined by Eq. (13) is solved by using a genetic algorithm and the Pareto front is iteratively constructed at each generation of the genetic algorithm. The initial value of parameter **a** has been set to **a**⁽⁰⁾ and corresponds to the solution of the following partial optimization problem:

 $\mathbf{a}^{(0)} = \arg \min \mathcal{I}_1(\mathbf{a})$ for $\mathbf{a} \in \mathcal{A}^{\text{macro}}$, which is solved with the simplex algorithm. The optimal value **b**^{meso} is then chosen as the point on the Pareto front that minimizes the distance between the Pareto front and the origin.

4. Validation of the method for 2D plane stresses

The validation is performed within the framework of the linear elasticity for 2D plane stresses. It should be noted that the two directions are observed when the displacement fields are measured at macroscale and at mesoscale with a camera.

4.1. Construction of a simulated "experimental" database

In order to validate the methodology, some "experimental" measurements are simulated for the macroscale and the mesoscale by using a computational model. The 2D domain, Ω^{macro} , is defined in the plane (Ox_1x_2) as a square whose dimension of the edge is $h = 10^{-2} m$. At mesoscale, the material is heterogeneous, anisotropic and linear elastic. A line force directed along $-x_2$, with an intensity of 5×10^{-2} N/m, is applied on the edge $x_2 = h$. The edge $x_2 = 0$ is fixed. A 2D plane stress state is assumed. At mesoscale, the 2D apparent elasticity field is constructed as a realization of the *prior* stochastic model of the random field $\left[C^{2D}(\mathbf{b}) \right]$ with $\ell = 1.25 \times 10^{-4}$ m, $\delta = 0.4$ and where the entries of $\left[\underline{S}^{2D}\right]$ are defined using $[S^{meso}]$ which corresponds to a transverse isotropic linear elastic medium,

$$
\begin{bmatrix}\n\frac{1}{E_T} & -\frac{\nu_T}{E_T} & 0 & -\frac{\nu_L}{E_L} & 0 & 0 \\
-\frac{\nu_T}{E_T} & \frac{1}{E_T} & 0 & 0 & 0 & 0 \\
0 & 0 & \frac{2(1+\nu_T)}{E_T} & 0 & 0 & 0 \\
-\frac{\nu_L}{E_L} & -\frac{\nu_L}{E_L} & 0 & \frac{1}{E_L} & 0 & 0 \\
0 & 0 & 0 & 0 & \frac{1}{G_L} & 0 \\
0 & 0 & 0 & 0 & 0 & \frac{1}{G_L}\n\end{bmatrix},
$$
\n(14)

with $E_L = 15.8 \times 10^9 Pa$, $E_T = 9.9 \times 10^9 Pa$, $G_L = 5.2 \times 10^9 Pa$, $\nu_L = 0.31$ and $\nu_T = 0.38$. Consequently, we have

$$
\left[\underline{S}^{2D}\right] = \begin{bmatrix} \frac{1}{E_T} & -\frac{\nu_T}{E_T} & 0\\ -\frac{\nu_T}{E_T} & \frac{1}{E_T} & 0\\ 0 & 0 & \frac{2(1+\nu_T)}{E_T} \end{bmatrix} . \tag{15}
$$

Thus, the vector-valued hyperparameter **b** is written as $\mathbf{b} = (\delta, \ell, E_T, \nu_T)$. At mesoscale, the realization of the apparent elasticity random field is simulated on the whole domain Ω^{macro} .

A computational model is constructed with the finite element method and a regular finite element mesh with one million quadrangle elements $(1, 000$ along x_1 and 1, 000 along x_2 , see Fig. 1, left). The strain field is numerically simulated in using a finite element interpolation in a regular grid of nodes with a mesoscale resolution on the whole domain Ω^{macro} (see Fig. 1, center). Measurements of the strain field $\varepsilon_{\rm exp}^{\rm macro}$ is simulated at macroscale in extracting the values of the displacement field in a regular grid of 10×10 nodes and in using a finite element interpolation (see Fig. 1, upper right). In addition, in the subdomain defined as a square with dimension 10^{-3} m (mesoscale), the measurements of the strain field $\varepsilon_{\text{exp}}^{\text{meso}}$ are simulated at mesoscale in extracting the values of the displacement field in a regular grid of 100×100 nodes and in using a finite element interpolation (see Fig. 1, lower right).

Figure 2 shows the values of $\{\varepsilon_{\exp}^{\text{macro}}\}_{22}$ for the simulated experimental strain field at macroscale with a resolution 10×10 . The square in black dashed line represents the considered mesoscale subdomain. Figure 3 shows the values of $\{\varepsilon_{\exp}^{\text{meso}}\}_2$ for the simulated experimental strain field at mesoscale with a resolution 100×100 .

4.2. Numerical results and validation

At macroscale, the *prior* model of the material is chosen as a transverse isotropic model. Consequently, for 2D plane stresses, the vector-valued parameter $a =$ $(E_T^{\text{macro}}, \nu_T^{\text{macro}})$ is made up of the transverse Young modulus and the transverse Poisson coefficient. The optimal value of $\mathbf{a} = (E_T^{\text{macro}}, \nu_T^{\text{macro}})$ is $\mathbf{a}^{\text{macro}} = (9.565 \times$ $10^9 Pa$, 0.3987). The finite element mesh of the subdomain at mesoscale is (i) cartesian with a constant size in directions x_1 and x_2 , and (ii) made up of $9.801 = 99 \times 99$ finite elements with 4 nodes. The total number of Gauss points associated with the finite element mesh is 39, 204. The convergence of the statistical estimators is reached for 500 independent realizations of the random elasticity field [**C** ²^D]. An initial population size of 50 is used for the solving the optimization problem defined by Eq. (13) with the genetic algorithm. Less than 100 generations has been enough for constructing the Pareto front which is iteratively constructed, at each generation of the genetic algorithm. The value of **a** macro is almost unchanged through the iterations when the multi-objective problem is solved. The computation has been performed with 50 cores with a CPU at 2GHz, and has required 120 hours of CPU time.

Figure 1: Description of the methodology for the construction of the simulated experimental measurements in using the finite element method at macroscale and at mesoscale: FE model of the specimen at macroscale with a mesoscale resolution (left); component $\{11\}$ of the strain field at macroscale with a mesoscale resolution (center); component {11} of the strain field at macroscale with a macroscale resolution (upper right); component $\{11\}$ of the strain field at mesoscale with a mesoscale resolution (lower right).

Table 1 shows the values of $\mathbf{b} = (\ell, \delta, E_T, \nu_T)$ for each point of the Pareto front displayed in Fig. 4. The optimal values correspond to the points 5, 6, 7, 8 and 9 where points 6 et 7 are close. The optimal value \mathbf{b}^{meso} is such that ℓ^{meso} = 9.66×10^{-5} m, $\delta^{\text{meso}} = 0.37$, $E_T^{\text{meso}} = 1.023 \times 10^{10}$ Pa, $\nu_T^{\text{meso}} = 0.376$. This result yields a validation of the proposed methodology since this identified optimal value **b** meso is very close to the value **b** that has been used to construct the simulated experimental database for which $\ell = 1.25 \times 10^{-4}$ m, $\delta = 0.4$, $E_T = 9.9 \times 10^9$ Pa, $\nu_T = 0.38$.

Figure 2: Component $\{\varepsilon_{\exp}^{\text{macro}}\}_{22}$ of the simulated experimental strain field at macroscale with a resolution 10×10 .

5. Application of the method with multiscale experimental measurements

In this section, results are presented for the method with real multiscale experimental measurements obtained by using a CCD camera coupled with a microscope and the digital image correlation (DIC) method for measuring the displacement fields on one sample of cortical bone at macroscale and at mesoscale. The measurements have been carried out at the *Laboratory of Solid Mechanics* of *Ecole Polytechnique*, and all the details concerning these multiscale experiments are presented in [42].

5.1. Multiscale experimental database

The DIC method allows the displacement field on an enlightened face of a sample under external loads to be measured for a given resolution of the camera and of the speckle pattern. The displacement field is obtained by comparing two images of the enlightened face, the first one when the sample is undeformed (image of reference) and the second image when the sample is deformed by the external loads (deformed image). Fig. 5 shows the images obtained for a sample of cortical bone. The comparison of the images is based on the correlation of the images under the assumption that the contrast is conserved locally in a vicinity of the points where the displacement is measured.

Figure 3: Component $\{\varepsilon_{\exp}^{\text{meso}}\}_{22}$ of the simulated experimental strain field at mesoscale with a resolution 100×100 .

The experimental configuration for one sample of cortical bone, which came from beef femur, is the same as for the example presented in section 4. The uniaxial load had to be limited to $9000 N$ in order to be in the elastic domain of the sample. A resolution of 10×10 points at macroscale, and a resolution of 100×100 points at mesoscale are used to identify the displacement fields at mesoscale and at macroscale. Fig. 6 and Fig. 7 display the identified displacement fields along directions x_1 and x_2 at macroscale. Fig. 8 and Fig. 9 display the identified displacement fields along directions x_1 and x_2 at mesoscale.

5.2. Numerical results

The optimal value of $\mathbf{a} = (E_T^{\text{macro}}, \nu_T^{\text{macro}})$ is $\mathbf{a}^{\text{macro}} = (6.74 \times 10^9 Pa, 0.32)$. The parameters of the method (finite element mesh size, number of independent realizations, initial population size for the genetic algorithm) are the same than in Section 4.2. Less than 100 generations has been enough for constructing the Pareto front which is iteratively constructed, at each generation of the genetic algorithm.

For each point of the Pareto front, displayed in Fig. 10, the values of $\mathbf{b} = (\ell, \delta, E_T, \nu_T)$ are summarized in Tab 2. The ninth point of the Pareto front minimizes the distance to the origin. Its optimal components are $\ell^{\text{meso}} = 5.06 \times 10^{-5} m$, $\delta^{\text{meso}} =$

Table 1: Optimization results of the genetic algorithm

k	$\mathcal{I}_2(\mathbf{b})$	$\mathcal{I}_3(\mathbf{a},\mathbf{b})$		δ	E_T	ν_T
	5.006529×10^{-9}	2.311672×10^{-1}	1.886667×10^{-4}	0.400000	1.023000×10^{10}	0.392667
2	5.006529×10^{-9}	9.477024×10^{-2}	2.500000×10^{-4}	0.400000	1.023000×10^{10}	0.392667
3	5.010827×10^{-9}	9.469903×10^{-2}	9.666667×10^{-5}	0.366667	1.023000×10^{10}	0.376200
4	5.132208×10^{-9}	9.201960×10^{-2}	1.273333×10^{-4}	0.383333	1.023000×10^{10}	0.392667
5.	5.240100×10^{-9}	3.467300×10^{-2}	9.666667×10^{-5}	0.366667	1.023000×10^{10}	0.359733
6	5.259407×10^{-9}	2.455275×10^{-2}	5.066667×10^{-5}	0.350000	8.943000×10^{9}	0.293867
	5.259407×10^{-9}	2.455275×10^{-2}	9.666667×10^{-5}	0.366667	1.023000×10^{10}	0.376200
8	5.386876×10^{-9}	2.064010×10^{-2}	5.066667×10^{-5}	0.350000	8.943000×10^{9}	0.310333
9	5.490529×10^{-9}	1.968774×10^{-2}	5.066667×10^{-5}	0.350000	1.237500×10^{10}	0.293867
10	6.57386×10^{-9}	1.962839×10^{-2}	2.193333×10^{-4}	0.400000	1.023000×10^{10}	0.392667
11	6.895467×10^{-9}	1.885624×10^{-2}	2.500000×10^{-4}	0.383333	1.023000×10^{10}	0.392667
12	7.254986×10^{-9}	1.759584×10^{-2}	2.500000×10^{-4}	0.333333	1.023000×10^{10}	0.392667
13	7.567184×10^{-9}	1.688894×10^{-2}	9.666667×10^{-5}	0.383333	1.023000×10^{10}	0.392667
14	7.996816×10^{-9}	1.623193×10^{-2}	2.000000×10^{-5}	0.350000	8.943000×10^{9}	0.310333
15	9.129340×10^{-9}	1.507042×10^{-2}	2.500000×10^{-4}	0.366667	1.023000×10^{10}	0.392667
16	9.368447×10^{-9}	1.333442×10^{-2}	1.273333×10^{-4}	0.266667	1.023000×10^{10}	0.392667

0.28, $E_T^{\text{meso}} = 6.96 \times 10^9 Pa$, $\nu_T^{\text{meso}} = 0.37$. This optimal solution yields a spatial correlation length equal to 5.06×10^{-5} m which is in agreement with the assumption introduced concerning the separation of the scales. This length is also of the same order of magnitude than the distance between adjacent lamellae or osteons in cortical bovine femur.

Table 2: Optimization in using the genetic algorithm with experimental data

k	$\mathcal{I}_2(\mathbf{b})$	$\mathcal{I}_3(\mathbf{a},\mathbf{b})$		δ	E_T	ν_T
	9.905566×10^{-4}	1.130759×10^{-1}	5.066667×10^{-5}	0.300000	6.964455×10^{9}	0.399180
2.	9.905566×10^{-4}	5.529077×10^{-2}	5.066667×10^{-5}	0.300000	6.964455×10^{9}	0.385342
3	9.905566×10^{-4}	5.529077×10^{-2}	5.066667×10^{-5}	0.300000	6.964455 \times 10 ⁹	0.399180
4	9.923024×10^{-4}	4.566584×10^{-2}	5.066667×10^{-5}	0.300000	6.964455 \times 10 ⁹	0.371504
5	9.942891×10^{-4}	3.721737×10^{-2}	5.066667×10^{-5}	0.300000	6.964455 \times 10 ⁹	0.371504
6	9.942891×10^{-4}	3.721737×10^{-2}	5.066667×10^{-5}	0.266667	6.964455 \times 10 ⁹	0.371504
	9.943808×10^{-4}	3.707802×10^{-2}	5.066667×10^{-5}	0.300000	6.380340 \times 10 ⁹	0.399180
8	9.947595×10^{-4}	3.698333×10^{-2}	1.426667×10^{-4}	0.383333	6.672398×10^9	0.316151
9	9.965090×10^{-4}	3.092503×10^{-2}	5.066667×10^{-5}	0.283333	6.964455×10^{9}	0.371504
10	9.989547×10^{-4}	2.826733×10^{-2}	2.040000×10^{-4}	0.416667	6.672398×10^{9}	0.343827
11	1.009211×10^{-3}	2.125592×10^{-2}	2.040000×10^{-4}	0.483333	6.672398×10^9	0.343827
12.	1.011463×10^{-3}	2.064572×10^{-2}	5.066667×10^{-5}	0.300000	6.964455 \times 10 ⁹	0.357666
13	1.019979×10^{-3}	1.925469×10^{-2}	5.066667×10^{-5}	0.300000	6.964455 \times 10 ⁹	0.343827
14	1.023436×10^{-3}	1.901348×10^{-2}	2.040000×10^{-4}	0.500000	6.672398×10^{9}	0.343827
15	1.024812×10^{-3}	2.645432×10^{-3}	2.040000×10^{-4}	0.433333	6.672398 \times 10 ⁹	0.343827

6. Conclusions

In the framework of the linear elasticity, a multiscale inverse statistical method has been presented for the identification of a stochastic model of the apparent elasticity random field at mesoscale for a heterogeneous microstructure using experimental

Figure 4: Pareto front for the numerical indicators $\mathcal{I}_2(\mathbf{b})$ and $\mathcal{I}_3(\mathbf{a}, \mathbf{b})$.

Figure 5: Comparison between a reference image (left) and a deformed image (right) at macroscale for a cubic cortical bovine bone sample.

measurements at macroscale and at mesoscale. A *prior* stochastic model depending of a hyperparameter has been proposed for the apparent elasticity random field at mesoscale in the case of 2D plane stresses. The identification procedure has been formulated as a multi-objective minimization problem with respect to the parameter of the elastic model at macroscale and the hyperparameter of the *prior* stochastic model at mesoscale. The optimal value of the parameter and the hyperparameter corresponds to the point that minimizes the distance of a Pareto

Figure 6: Component $\{u_{\text{exp}}^{\text{macro}}\}_1$ in direction x_1 (horizontal) for the experimental displacement at macroscale: in color (left) and in black and white (right).

Figure 7: Component $\{u_{\text{exp}}^{\text{macro}}\}_2$ in direction x_2 (vertical) for the experimental displacement at macroscale: in color (left) and in black and white (right).

Figure 8: Component $\{u_{\text{exp}}^{\text{meso}}\}_1$ in direction x_1 (horizontal) for the experimental displacement at mesoscale: in color (left) and in black and white (right).

front to the origin. The proposed statistical inverse method has been validated with a simulated experimental database and results have been presented in the

Figure 9: Component $\{u_{\text{exp}}^{\text{meso}}\}_2$ in direction x_2 (vertical) for the experimental displacement at mesoscale: in color (left) and in black and white (right).

case of experimental measurements obtained by the DIC method on one sample of cortical bone observed by a CCD camera at both macroscale and mesoscale.

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Figure 10: Pareto front for the two numerical indicators $\mathcal{I}_2(\mathbf{b})$ and $\mathcal{I}_3(\mathbf{a}, \mathbf{b})$ for the experimental identification.

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