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Modelling, identification and application of phenomenological constitutive laws over a large strain rate and temperature range

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Abstract

A review of the different phenomenological thermo-viscoplastic constitutive models often applied to forging and machining processes is presented. Several of the most common models have been identified using a large experimental database (Hor et al. (2013)). The latter consists of the tests were done in compression on cylindrical shaped specimens and in shear using hat-shaped specimens. The comparison between these different models is shown that the group of decoupled empirical constitutive models (e.g. the Johnson and Cook (1983) model), despite their simple identification procedures, are relatively limited, especially over a large range of strain rates and temperatures. Recent studies have led to the proposal of coupled empirical models. Three models in this class have also been studied. The Lurdos (2008) model shows the best accuracy but requires a large experimental database to identify its high number of parameters. After this comparison, a constitutive equation is proposed by modifying the TANH model (Calamaz et al. (2010)). Coupling between the effects of strain rate and temperature is introduced. This model is easier to identify and does not require knowledge of the saturation stress. Compared to other models, it better reproduces the experimental results especially in the semi-hot and hot domains.

In order to study real machining conditions, an orthogonal cutting tests is considered. The comparison between experimental test results and numerical simulations conducted using the previously identified constitutive models shows that the decoupled empirical models are not capable of reproducing the experimental observations. However, the coupled constitutive models, that take into account softening, improve the accuracy of these simulations.

Keywords:

rheology, large strain rate and temperature range, 42CrMo4, 100Cr6, identification, numerical

1. Introduction

In order to correctly study rheology in forging and machining operations a wide range of strain rates (between 10^{-2} et $10^5 s^{-1}$) and temperatures (up to the melting temperature) must be taken into account (Hor et al. (2013)). Over this large range of rheological variables, the investigated steels are subject to phenomena accompanying deformation such as strain hardening, softening and phase changes (Hor et al. (2013)). These phenomena affect the overall material behaviour and result in a change in the shape of the stress-strain curves. To model the thermo-viscoplastic behaviour, many constitutive models exist. Most were developed using the assumption of isotropic behaviour. The general form of these models can be written as:

$$\sigma = \sigma(\dot{\epsilon}, T, \alpha_i) \quad \text{and} \quad \dot{\alpha}_i = \alpha_i(\sigma, \dot{\epsilon}, T, \alpha_j) \quad (1)$$

The stress and strain rate tensors are usually replaced by the von Mises equivalent stress and strain rate (given by σ and $\dot{\epsilon}$ respectively). One or more internal variables (noted α_i) are used to take into account more accurately the deformation history and the physical parameters of the studied material (deformation, dislocation density, grain size, etc.). In general, it is the evolution laws $\frac{d\alpha_i}{dt}$ of the internal variables that are defined.

Stress-strain constitutive models fall into three categories: empirical models, semi-empirical models and models incorporating physical phenomena. The latter are the most evolved, they offer solutions that are based on the physics of deformation processes, but are often more complex to use. Internal variables represent the deformation history and/or characteristics sizes of the physical state of the material (i.e. grain size, dislocation density, etc.). The Kocks and Mecking model (Kocks (1976)) and the Bergström et al. model (Bergstrom and Roberts (1971)) can be cited as examples of this class of models, describing the evolution of the density of mobile (or all) dislocations to account for the hardening and restoration of the material. Semi-empirical models

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can be defined as having a simple formulation while incorporating the minimum physical phenomenon. Although, they do not contain internal variables. They have an additive form which regroups an athermal part and a thermal part making reference to thermally activated phenomena. Include in this category are the Klepaczko model (Klepaczko (1987)) and the Zerilli and Armstrong model (Zerilli and Armstrong (1987)). These two classes of constitutive models are used primarily by metallurgists to reflect the evolution of certain phenomena during manufacturing processes (recrystallization, twinning, etc.), but rarely to simulate the thermo-mechanical behaviour of materials during these processes. Unlike empirical models, which generally have a simple formulation. Only a few tests are required to identify their coefficients. Moreover, their mathematical form is chosen so as to facilitate their implementation in numerical software. This allows calculations to be undertaken at low computational and experimental cost. These two benefits explain why empirical models are very often used. These models have undergone several developments in recent years with advances in numerical and experimental methods.

In this work, the predictive power of these different empirical models is tested in terms of machining and forging operations while outlining the experimental and numerical difficulties encountered in their identification. This is done by using a large experimental stress-strain database (Figure 1) of three steels: the 42CrMo4 steel with two different microstructures, ferrite-perlitic and bainitic (referred to as 42CrMo4-FP and 42CrMo4-BA respectively) and the 100Cr6 steel (referred to as 100Cr6-FP). This experimental database is widely detailed in Hor et al. (2013). Then, a coupled empirical model with a better cost/accuracy ratio is proposed. Finally, the value of using more accurate models will be discussed through the numerical simulations of orthogonal cutting.

2. Experimental database

In this section, the material behaviour will be briefly analyzed for both loading cases (compression and shear) in terms of strain rate and temperature sensitivity on the flow stress. This analysis only focuses on 42CrMo4-BA steel, a more detailed study is provided for the three studied steels in Hor et al. (2013). The results for the compression and shear tests for the 42CrMo4-BA steel are given in Figure 2. The experimental procedure and the determination of the different variables (stress, shear stress, plastic strain and plastic shear strain) are described in Hor et al. (2013).

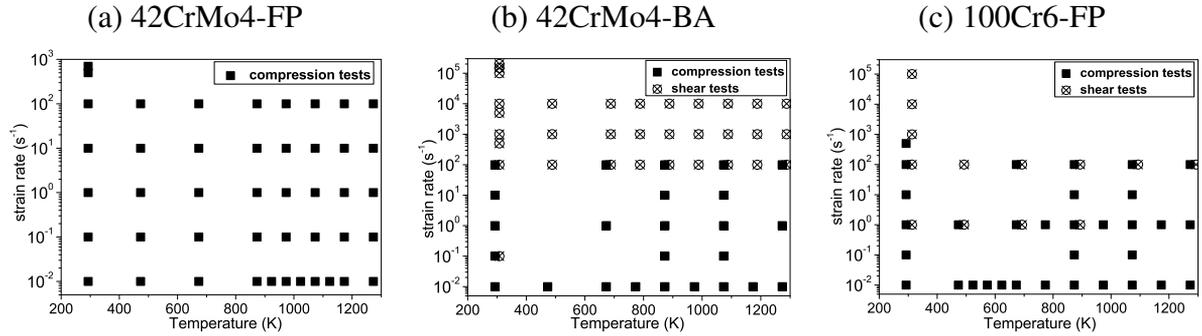


Figure 1: Thermo-mechanical testing conditions of the three steels investigated.

From Figure 2-a, it can be seen that, for a given strain rate (10^{-2} s^{-1}), a temperature rise moved the stress-plastic strain curves towards the bottom. A radical change could be observed in the shape of these curves. Their slopes can be both positive, indicating work hardening, or negative indicating softening. With respect to this competition between work hardening and softening, three deformation domains can be distinguished, as a function of the temperature. These observations remain valid for the shear loading mode (Figure 2-b). Although the attained shear levels were very high (up to 6 in von Mises equivalent strain), a saturation stress was observed only for some temperature and equivalent strain rate conditions.

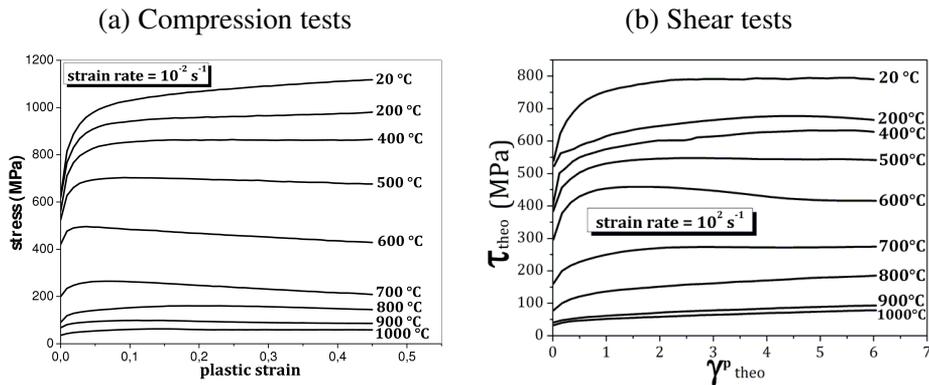


Figure 2: Evolution of (a) the compressive stress as a function of the plastic strain at a strain rate of 10^{-2} s^{-1} and (b) the shear stress as a function of the plastic shear strain at an equivalent strain rate of 10^2 s^{-1} for the 42CrMo4-BA steel.

2.1. Cold deformation domain ($T/T_f < 0.3$)

In this domain, work hardening is the phenomenon observed for low strain rates (up to 1 s^{-1}). Increasing the strain rate led to the appearance of softening (Figure 3-a1). This phenomenon can appear after a change in the deformation mechanism caused by the speed of the deformation process. Besides, during dynamic tests ($\dot{\epsilon} \geq 1 \text{ s}^{-1}$), the plastic power was converted into heat which could not be evacuated in this range of strain rate. This phenomenon will be called **self-heating**.

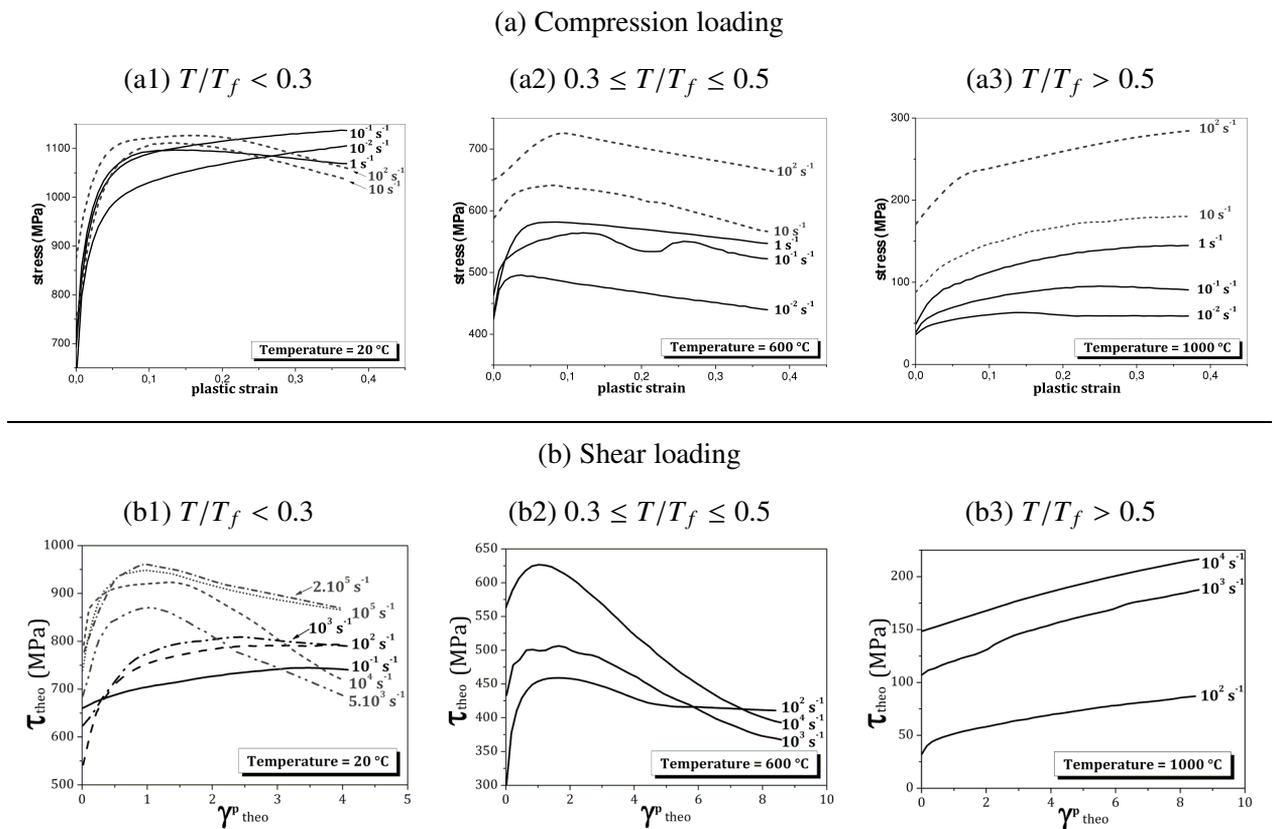


Figure 3: Evolution of (a) the compressive stress as a function of the plastic strain and (b) the shear stress as a function of the plastic shear strain, at 20°C , 600°C and 1000°C , for different equivalent strain rates for the 42CrMo4-BA steel.

For the shear tests, the stress increased with the equivalent strain rate. For equivalent strain rates between 10^{-1} and 10^3 s^{-1} , the increase was modest. Above 10^3 s^{-1} , the rate sensitivity became important and the shape of the curves was also changed (see Figure 3-b1). From an equivalent strain rate of 10^3 s^{-1} , the stresses reached a local maximum and then decreased gradually. Damage

was not observed in this case. This softening was also due to self-heating.

To evaluate the level of self-heating, the evolution of temperature was measured at the surface of the specimen subjected to compression loading at an initial temperature of 20°C and at a strain rate of $\dot{\varepsilon} = 10^2 \text{ s}^{-1}$. The evolutions of temperature and strain as function of test time are plotted in Figure 4. An increase in the surface temperature of approximately 120°C was measured during the test, and it is very likely that the temperature in the centre of the specimen was higher. This is reinforced by the observed time difference between the strain evolution and the appearance of a high surface temperature. A non-negligible time period was necessary for the surface temperature to increase. This phenomenon caused the flow stress to decrease at the end of the deformation process. In the literature, constitutive models are typically based on isothermal flow curves. Consequently, adiabatic flow curves should be corrected to isothermal flow curves (Hor et al. (2013)). Self-heating is therefore a phenomenon that provokes softening and reduces the level of work hardening for steel materials.

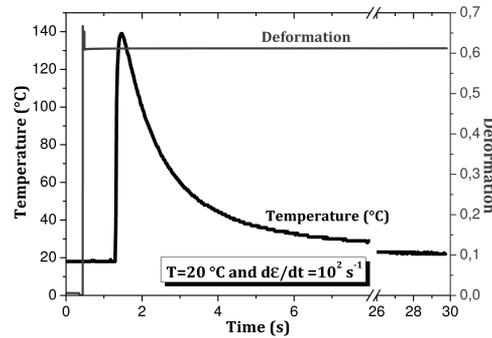


Figure 4: Experimental measurement of the temperature increase during compression deformation of the 42CrMo4-BA steel at $T = 20^{\circ}\text{C}$ and $\dot{\varepsilon} = 10^2 \text{ s}^{-1}$ (Hor et al. (2013)).

Two axisymmetric finite element calculations corresponding to the compression and shear loading conditions were carried out. The models assumed adiabatic flow where the coefficient of the plastic power converted into heat (Taylor-Quinney factor) was 0.95. Abaqus Explicit finite element code was used in conjunction with a Johnson-Cook constitutive law (Johnson and Cook (1983)). The material parameters used were identified from the compression tests (see Table 7). Frictional effects were neglected in both cases. The initial temperature was equal to 20°C (293 K).

Two speeds (10^{-2} ms^{-1} and 1 ms^{-1}) were applied on the top face of the specimen in the (-Y) direction. These two speeds respectively corresponded to an average strain rate of 1 s^{-1} and 10^2 s^{-1} in compression and 10^2 s^{-1} and 10^4 s^{-1} in shear. The bottom face was fixed. For the compression simulation ($T = 20^\circ\text{C}$, $\dot{\epsilon} = 10^2 \text{ s}^{-1}$), the surface temperature (at the node corresponding to the thermocouples location) reached 135°C , to compare with 140°C obtained experimentally in the same conditions (Figure 4). The temperature at the center of the specimen reached 200°C (473 K) for a speed of 1 ms^{-1} or 110°C (383 K) for a speed of 10^{-2} ms^{-1} . In the case of the shear simulation, the self-heating temperature was higher than the temperature reached in compression ($300^\circ\text{C}/573 \text{ K}$) and more localized (Figure 5-b). This localization was more pronounced for the highest speed (1 ms^{-1}). The temperature numerically calculated was lower than the temperature causing adiabatic shear bands (Meyers et al. (1994)). These bands were not experimentally observed.

In Hor et al. (2013), comparison was done between isothermal, adiabatic (present simulation) and experimental (data in Figure 3-a1) flow curves. The softening observed experimentally was reproduced only in adiabatic flow simulation. However, the relative differences between the two simulated flow stresses remained low (below 7%). These differences will be taken into account at the step of identification.

2.2. Semi-hot deformation domain ($0.3 \leq T/T_f \leq 0.5$)

The phenomenon which dominated in this domain for all strain rates is softening (see Figure 3-a2). The strain rate did not have an influence on the amount of softening but rather on the stress level. This structural softening was not provoked by self-heating but rather by a change in the deformation mechanisms. The dislocation mobility was perturbed by thermally activated phenomena such as dynamic recrystallisation (Hor et al. (2013)). These phenomena appeared even more clearly in the 42CrMO4-BA steel where oscillations in the stress-strain curve were observed ($T = 600^\circ\text{C}$ and $\dot{\epsilon} = 10^{-1} \text{ s}^{-1}$). Finally, with respect to the cold deformation domain, the flow stress was more sensitive to the strain rate.

For the shear tests, for all equivalent strain rates, the shear stress-strain curves were displayed by a local maximum followed by a decrease in the flow stress towards a saturation stress (see Figure 3-b2). The rate of this softening, increased with increasing equivalent strain rate. This was

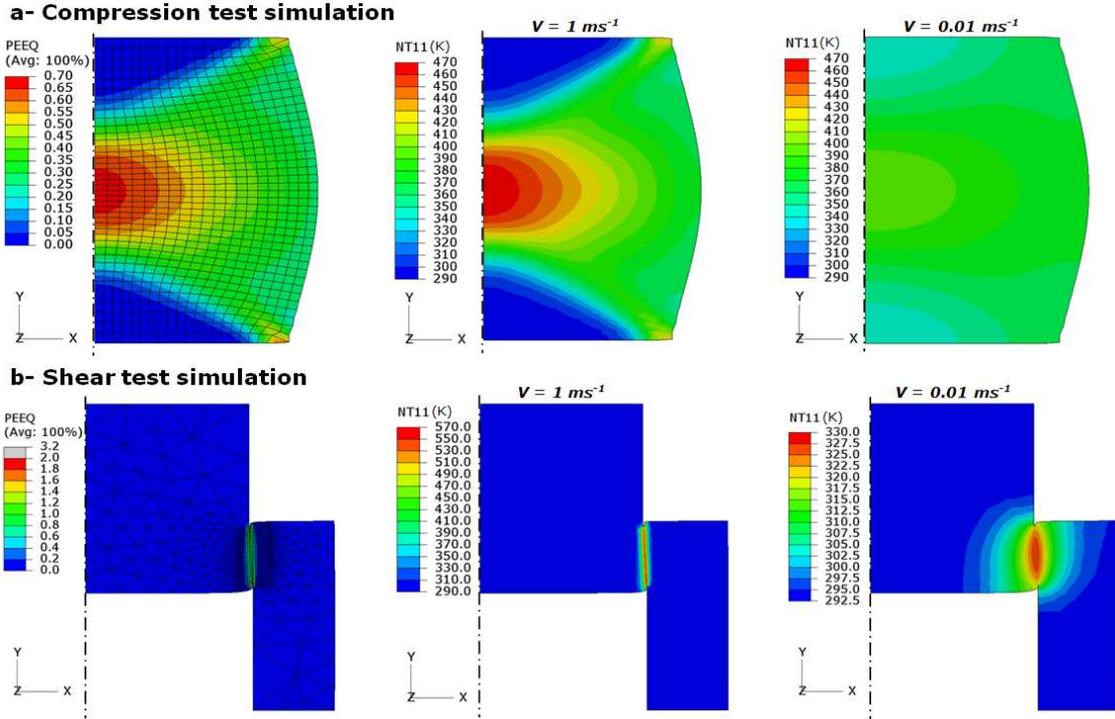


Figure 5: Self-heating temperature fields obtained by numerical simulation of the (a) compression and (b) shear tests for two speeds (1 and 10^{-2} ms^{-1}).

attributed to the effect of the very localised self-heating in the shear zone of the specimen. This phenomenon was not observed in compression, not only because the strain and the strain rates were lower, but also because the strain field in compression test was more homogeneous and the self-heating was less intense and less localized.

2.3. Hot deformation domain ($T/T_f > 0.5$)

The austenization temperature was between 800 and 900°C . At these temperatures, the material started to transform from a *body centred cubic* (BCC) structure to a *face centred cubic* (FCC) structure (Hor et al. (2013)). In this temperature interval, the shape of the stress-strain curves changed completely for the compression and shear loading cases (see Figures 3-a3 and 3-b3). The stress was more sensitive to the strain rate than in the cold and semi-hot deformation domains. Like in the cold domain, the strain rate had an influence on the competition between work hardening and softening. The effect is opposite to the one observed in the first temperature interval, that

is, the higher the strain rate, the lower the level of softening and the higher the work hardening. Also, softening that occurred at lower strain rates was different to that observed in the cold and semi-hot deformation domains.

2.4. Syntheses

The investigation of the material behaviour of the 42CrMo4-BA steel over a large range of strain rates and temperatures, for two loading modes (i.e. compression and shear), highlighted the existence of three deformation domains (cold, semi-hot and hot deformation) that can be defined by certain temperature ranges. The shape of the stress-strain curves and the evolution of the flow stress as a function of the temperature were very different in each domain. As well as the influence on the flow stress, the strain rate played a role on the phenomena accompanying plastic deformation such as work hardening and softening. The comparison between the behaviour of the three steels revealed much similarity (see Hor et al. (2013)), which was encouraging from a modelling point of view: the development of a general constitutive model which can be identified over such a large range of rheological state variables appeared to be possible. The constitutive model should be able to reproduce the different configurations shown in Figure 6. For this purpose, two choices are possible :

- The first choice consists in identifying the appropriate constitutive law in each deformation domain. Three different models should be determined, which requires sufficient experimental data in each domain. This would be very costly in terms of experimental work. Besides, the numerical implementation of three models to capture the constitutive law of one single material would be difficult. This strategy will not be selected due to these disadvantages.
- A second choice would be to identify a single constitutive model for the three deformation domains even with a large number of parameters. This would require finding the most comprehensive model whose formulation can reproduce the different configurations shown in Figure 6.

It is this second choice that will be discussed in the following.

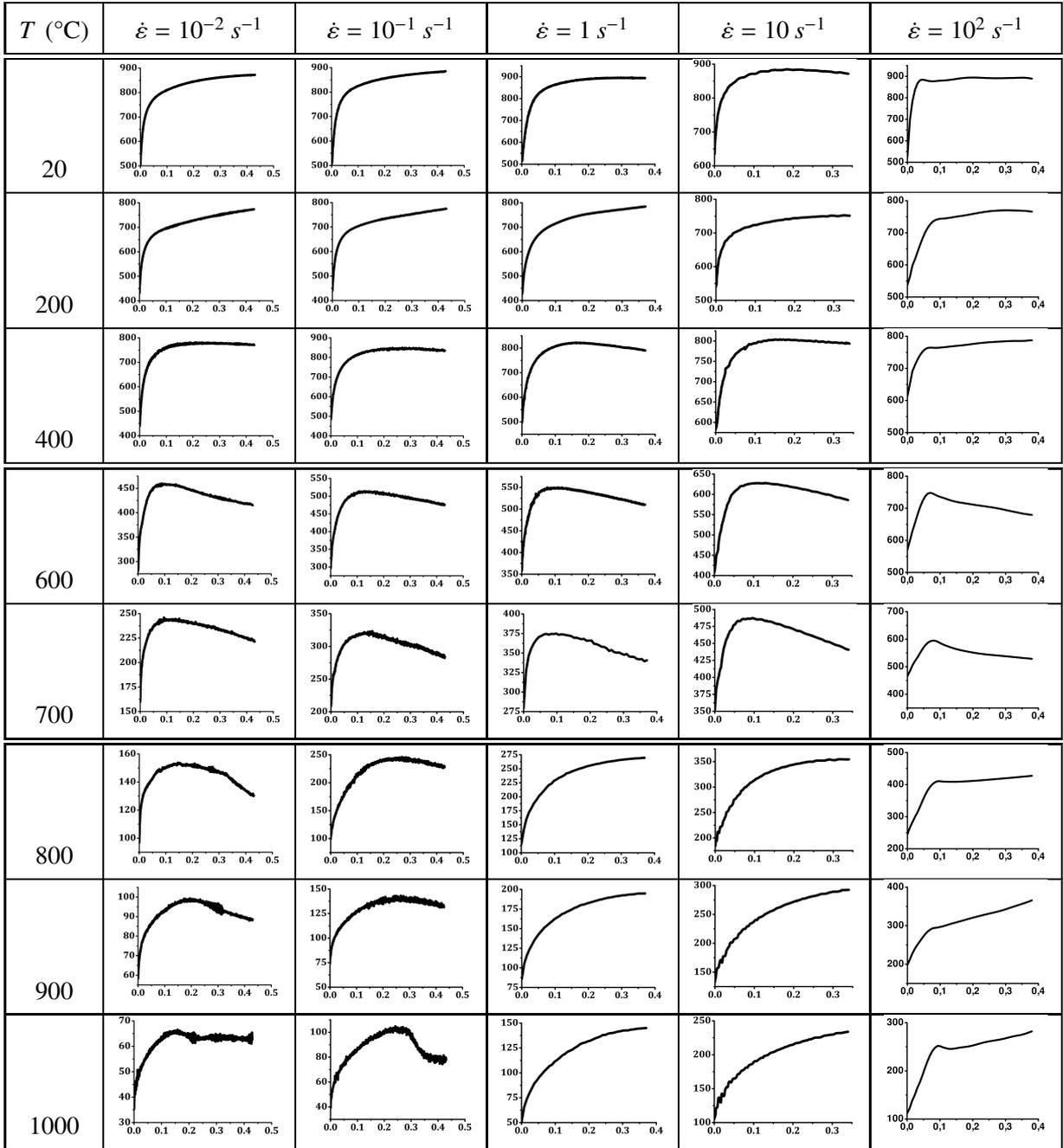


Figure 6: Various shapes of stress-strain curves of the 42CrMo4-FP steel as a function of thermomechanical conditions.

3. Phenomenological thermo-viscoplastic modelling

This paragraph lists the most commonly used phenomenological (or empirical) models. The following general notation will be used: σ_0 refers to the yield strength (or the stress corresponding

to zero plastic deformation), σ_s to the saturation stress (when the strain becomes very large). The terms ε and $\dot{\varepsilon}$ respectively denote the plastic strain and the plastic strain rate. Finally, the various model parameters will be presented as follows: the notations n , m , p , ... will be used as exponents, and the coefficients will be noted as A , B , C ,

3.1. Decoupled models

In this class, the quasi-static constitutive laws (Table 1) are usually multiplied by factors empirically describing the influence of the temperature and the strain rate. A single mathematical formula then describes the isotropic hardening and the effects of strain rate and temperature. The general form of this class of constitutive equations is:

$$\sigma = \sigma_\varepsilon(\varepsilon) \sigma_{\dot{\varepsilon}}(\dot{\varepsilon}) \sigma_T(T) \quad (2)$$

where σ_ε is a function chosen from the expressions in Table 1, $\sigma_{\dot{\varepsilon}}$ is the factor describing the sensitivity to the strain rate, and σ_T is the factor describing the influence of the temperature. Different proposed expressions for the functions $\sigma_{\dot{\varepsilon}}$ and σ_T are listed in Table 2.

Reference	Formulation	Parameters
Ludwik (1909)	$\sigma = \sigma_0 + B \varepsilon^n$	σ_0, B, n
Hollomon (1945)	$\sigma = \sigma_0 \varepsilon^n$	σ_0, n
Prager (1955)	$\sigma = \sigma_0 + \tanh(E\varepsilon/\sigma_0)$	σ_0, E
Voce (1948)	$\sigma = \sigma_s + (\sigma_0 - \sigma_s) \exp(-\varepsilon/\varepsilon_0)$	$\sigma_0, \sigma_s, \varepsilon_0$
Swift (1952)	$\sigma = \sigma_0 (\varepsilon + \varepsilon_0)^n$	$\sigma_0, \varepsilon_0, n$

Table 1: Different formulations for the quasi-static stress-strain constitutive laws.

A stress-strain constitutive equation can be constructed by combining the factors most adapted to the problem. These models are referred to as decoupled constitutive equations because there is no coupling between the effects of hardening, temperature sensitivity and strain rate sensitivity. Among the models most used to simulate manufacturing processes in industry and for research purposes: Sellars (1985) model, Picart (1986) model and Johnson and Cook (1983) model can

$\sigma_{\dot{\epsilon}}$ function		σ_T function	
Power law	$\dot{\epsilon}^m$	Power law	T^{-m}
Cowper and Symonds (1957)	$1 + (\dot{\epsilon}/\dot{\epsilon}_0)^m$	Johnson and Cook (1983)	$1 - \left(\frac{T-T_0}{T_m-T_0}\right)^m$
Ludwik (1909)	$\log(\dot{\epsilon}/\dot{\epsilon}_0)$	Litonsky (1977)	$1 - T/T_0$
Sokolowsky (1948)	$\log(1 + \dot{\epsilon}/\dot{\epsilon}_0)$	Vinh et al. (1980)	$\exp(W/T)$
Johnson and Cook (1983)	$1 + C \ln(\dot{\epsilon}/\dot{\epsilon}_0)$	Zhao and Gary (1996)	$1 - B(T - T_0)$
		Arrhenius	$\exp(-Q/RT)$

Table 2: List of multiplicative functions describing the effect of strain rate and temperature.

be cited. In particular, the Johnson and Cook (1983) model has been widely used to simulate relatively rapid processes such as machining and forging. This model is expressed by:

$$\sigma_{JC} = [A + B\epsilon^n] \left[1 + C \ln\left(\frac{\dot{\epsilon}}{\dot{\epsilon}_0}\right) \right] \left[1 - \left(\frac{T - T_0}{T_m - T_0}\right)^m \right] \quad (3)$$

where T_m is the melting temperature of the material, T_0 and $\dot{\epsilon}_0$ are the reference temperature and strain rate. Many authors have modified this formulation to better represent certain phenomena or specific applications. For instance, Holmquist and Johnson (1991) experimentally observed an increase in the stress with a large strain rate greater than that given by expression (3). Andrade et al. (1994) have empirically modified the temperature term to represent the effect of softening due to dynamic recrystallization when the temperature exceeds a limit value T_c .

3.2. Coupled Models

The decoupling between hardening, strain rate and temperature is a strong assumption in terms of constitutive modelling. This decoupling implies that the shape of the stress-strain curve does not change after a variation in strain rate or temperature. Several authors have shown the limits of these models. They found that the softening phenomenon is essential for the initiation and amplification of strain localization, which is the cause of the formation of adiabatic shear bands in the simulation of machining operations. Two types of modification to the decoupled empirical models have been proposed: the first consists of introducing a coupling between the three terms

(strain hardening, temperature sensitivity and strain rate sensitivity). However the final expression remains multiplicative (Khan and Liang (1999); Liang and Khan (1999); Calamaz et al. (2010); Bäker (2006)). The second category of coupled phenomenological models considers a quasi-static model (Table 1) and transforms its constants into internal variables that are functions of temperature and strain rate. Several authors have chosen the Voce (1948) law which tends to a saturation stress σ_s with increasing strain. For instance, Lurdos (2008) added a supplementary term $A\varepsilon^n$ to better model the stress observed in certain cases (Table 3-(B1)). The five parameters σ_0 , σ_s , r , A , and n (denoted β_i) vary with temperature and strain rate. They are expressed as a function of the Zener-Holloman parameter $Z = \dot{\varepsilon} \exp(Q/RT)$ (Zener and Holloman (1944)) connecting the temperature to the strain rate (Q is the apparent activation energy of the material and $R = 8.314 \times 10^{-3} \text{ kJ.mol}^{-1} \text{ K}^{-1}$ is the universal ideal gas constant).

Reference	Formulation
(A1) Khan and Liang (1999)	$\sigma = \left[A + B \left(1 - \frac{\ln \dot{\varepsilon}}{\ln \dot{\varepsilon}_0} \right) \varepsilon^n \right] [\dot{\varepsilon}]^C \left[1 - \left(\frac{T-T_0}{T_m-T_0} \right)^m \right]$
(A2) Bäker (2006)	$\sigma = \left[A \varepsilon^{n_0 f(T)} \right] \left[1 + C \ln \left(\frac{\dot{\varepsilon}}{\dot{\varepsilon}_0} \right) \right] f(T)$ with $f(T) = \exp \left[- (T/T_a)^m \right]$
(A3) Calamaz et al. (2010)	$\sigma = \sigma_{JC} \left[D + (1 - D) \tanh \left(\frac{1}{\varepsilon + \varepsilon_a} \right) \right]$ with $D = 1 - \left(\frac{p \varepsilon^r}{1 + p \varepsilon^r} \right) \tanh \left[\left(\frac{T-T_0}{T_{rec}-T_0} \right)^q \right]$
(B1) Lurdos (2008)	$\sigma = \sigma_s + (\sigma_0 - \sigma_s + A \varepsilon^n) \exp(-r \varepsilon)$ with $\beta_i = \beta_i^0 Z^{m_i}$, $\beta_i = \sigma_0, \sigma_s, r, A, n$
(B2) Mulyadi et al. (2006)	$\sigma = K \lambda Z^m (1 - \exp(-\beta \varepsilon))$ with $\lambda = \lambda_s + (1 - \lambda_s) \exp(-\alpha \varepsilon)$ and $\lambda_s = \lambda_0 Z^q$

Table 3: Different formulations of coupled constitutive equations.

Finally, these models assume the convergence of the flow stress to a saturation state. Their identification requires that this state be achieved experimentally. Montheillet (1996) showed that for materials with high stacking fault energy, very high amounts of deformation ($\varepsilon \approx 50$) are required to reach the saturation state. This saturation state is generally predicted by extrapolation of the experimental stress-strain curves during the identification process. This constitutes a weakness

of this type of phenomenological formulation. For these reasons, the model proposed in this work falls into the first category of empirical coupled laws. The aim is to improve the cost/accuracy ratio, that is, to propose a model that reproduces as faithfully as possible the experimental phenomena while keeping the number of model parameters to a minimum.

3.3. Proposed phenomenological coupled model

Drawing on the conclusions from the experimental study on the three steels investigated (Hor et al. (2013)), two different couplings are proposed in this model. The first involves coupling between the effects of work hardening and temperature. The second coupling is between the effects of strain rate and temperature. This proposition comes from the classification of the complete deformation domain into three sub-domains: cold, semi-hot and hot deformation. This classification highlights the fact that different behaviour was observed in each domain, especially concerning the viscosity.

$\sigma = \sigma_{\varepsilon}(\varepsilon, T) \cdot \sigma_T(T) \cdot \sigma_{\dot{\varepsilon}}(\dot{\varepsilon}, T)$	
(i)	$\sigma_{\varepsilon}(\varepsilon, T) = [A + B \varepsilon^n] \left[D + (1 - D) \tanh\left(\frac{1}{\varepsilon + \varepsilon_a}\right) \right]$ with $D = 1 - \left(\frac{p\varepsilon}{1 + p\varepsilon}\right) \tanh\left(\frac{T - T_0}{T_{rec} - T_0}\right)$
(ii)	$\sigma_T(T) = 1 - m_1 \left(\frac{T - T_0}{T_m - T_0}\right)^{m_2}$
(iii)	$\sigma_{\dot{\varepsilon}}(\dot{\varepsilon}, T) = 1 + C(T) \ln\left(\frac{\dot{\varepsilon}}{\dot{\varepsilon}_0}\right)$ with $C(T) = \frac{C_1 \exp\left(C_2 \frac{T}{T_m}\right)}{T_m}$

Table 4: Formulation of the proposed coupled constitutive law.

(i) *Modelling the competition between work hardening and softening.* The competition between strain hardening and softening is the origin of the variation in the slope of the stress-strain curve from one temperature to another. Empirical modelling of this competition requires a combination of two terms. This combination can be a simple addition of two functions: a work hardening function and a softening function (Hor (2011)) or it can have a multiplicative form as per the TANH model (Calamaz et al. (2010)). The first term is physically more justifiable but is very difficult to identify. In fact, additive laws are strongly coupled and do not allow the elimination of

terms in order to identify another term independently. The second term is more phenomenological, but it has the advantage of greater ease of identification. For these reasons, the second form was chosen here to model the hardening-softening competition. The expression of the function D (Table 4-(i)) has been simplified, by reducing the number of parameters, compared to the TANH model (Table 3-(A3)).

(ii) *Modelling the temperature sensitivity.* The sensitivity of the stress to the temperature is almost independent of the strain rate. It has been shown that a temperature variation has the same effect on the stress at different strain rates (Hor et al. (2013)). A strain rate variation does not affect the sensitivity of the stress to the temperature in the case of the three steels studied. From this finding and for reasons of simplicity, the temperature dependence can be introduced in a similar manner to that proposed by Johnson-Cook. A slight improvement was made in the proposed formulation (Table 4-(ii)): a second parameter was added to better approximate this temperature sensitivity (Figure 7).

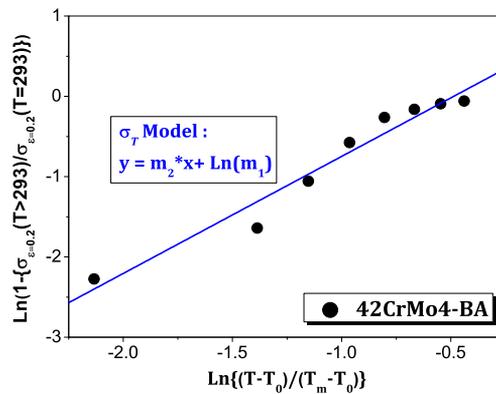


Figure 7: Identification of the parameters m_1 and m_2 from experimental database for the 42CrMo4-BA steel.

(iii) *Modelling the strain rate sensitivity.* The experimental study conducted by Hor et al. (2013) on three steels shows that this dependence varies with temperature. Indeed, for low temperatures, the stress is almost insensitive to the strain rate. As the temperature increases, the material becomes more viscous and the stress becomes more dependent on the strain rate (see Figure 3). The identification of the Johnson-Cook model based on experimental data leads to different values of

the parameter C (denoted C_i) according to the considered temperatures (Figure 8-(a)). To model this variable dependence, the Johnson-Cook formulation was modified by replacing the parameter C by a function $C(T)$ which depends on the temperature (Figure 8-(b)). The expression $C(T)$ (Table 4-(iii)) has been proposed based on experimental observations.

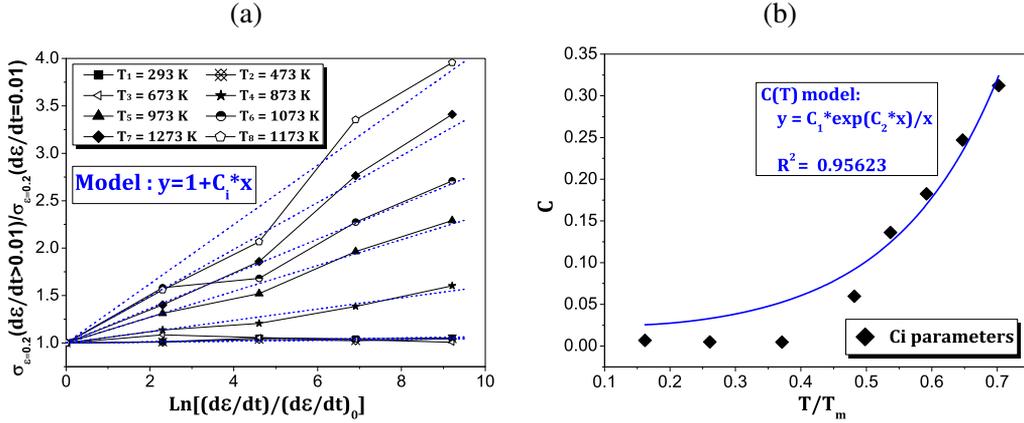


Figure 8: (a) Identification of the various parameters C_i related to different temperatures T_i and (b) correlation between the function $C(T)$ and the parameters C_i , for the 42CrMo4-BA steel.

4. Identification and comparison of the different models

Initially, compression tests were used to identify the various models presented in Table 5. The identified parameters are valid for a strain range between 0 and 0.5, temperatures between 293 and 1273 K, and for strain rates between 10^{-2} and 10^4 s^{-1} . The experimental stress-strain curves are sometimes extrapolated in order to estimate the saturation stress, necessary for the determination of some models. To facilitate the comparison, all models were identified from the same experimental database, even though some of them do not require such an extensive database. The sensitivity of the Johnson-Cook model parameters to the identification procedure and the loading modes will be discussed. To do so, shear tests and a minimum number of compression tests were used to identify the model.

In addition to the model parameters shown in Table 5, the models include certain coefficients that define the range of validity of the model (the reference temperature T_0 and reference strain

Model	Number of parameters	parameters to be identified
Johnson-Cook	5	A, B, n, C, m
Bäker	5	A, T_a, n_0, C, m
TANH	9	$A, B, n, C, m, p, q, r, \varepsilon_a$
Lurdos	10	$m_0, m_s, m_r, m_A, m_n, \sigma_0^0, \sigma_s^0, r^0, A^0, n^0$
proposed model	9	$A, B, n, p, \varepsilon_0, C_1, C_2, m_1, m_2$

Table 5: List of models investigated and the associated parameters to be identified.

rate $\dot{\varepsilon}_0$), define certain phenomena (i.e. the softening onset temperature T_{rec}) or intrinsic properties of the material (i.e. the melting temperature T_m and the apparent activation energy Q). These parameters are given for the three steels studied in Table 6.

Material	$\dot{\varepsilon}_0$ (s^{-1})	T_0 (K)	T_{rec} (K)	T_m (K)	Q ($kJ.mol^{-1}$)
42CRM04-FP	0.01	293	673	1813	300
42CRM04-BA	0.01	293	673	1813	300
100Cr6-FP	0.01	293	673	1760	300

Table 6: Values of the constants used in the different models for the three steels investigated.

4.1. Identification of the different models

An optimization program based on the Levenberg-Marquardt method (Marquardt (1963)) (derived from Newton's method) was used for the identification. This method required the introduction of a set of initial parameters which are of the same order of magnitude as the final parameters. To resolve this problem, a method of normalising the parameters (between 0 and 1) was used. For some cases a combination of the Simplex method and the Levenberg-Marquardt method was used to obtain the optimal set of parameters (i.e. the parameters minimizing the squared variation between the model and the experimental results). The uniqueness of the solution was verified by checking if the same optimum parameter set was obtained when the initial values were varied.

Sensitivity analyses were also performed for some models (e.g. the TANH model) to determine the influence of each parameter (Hor (2011)).

The least squares method was used for identification. The objective function was:

$$S(\beta_j) = \sum_{k=1}^N a_k s_k(\beta_j) = \sum_{k=1}^N \left[a_k \sum_{i=1}^{200} \left\{ (\sigma_{exp}^i)_k - L(\varepsilon_k^i, \dot{\varepsilon}_k, T_k, \beta_j) \right\}^2 \right] \quad (4)$$

The quantity $L(\varepsilon_k^i, \dot{\varepsilon}_k, T_k, \beta_j)$ was calculated from the analytical expression of the constitutive model for compression loading, or given by the FE model for shear loading. In this case, it was the efforts and displacements that were evaluated and not the stresses and strains.

N is the number of experimental tests (number of thermomechanical conditions) used for the identification, a_k is the weighting coefficients to allow favored one condition over another. (a_k is always equal to 1 in our case). The β_j are the model parameters to identify. A step of smoothing and interpolation was executed to obtain only 200 points for each curve. Self-heating effect was taken into account for some thermomechanical conditions. In fact:

- For compression tests, at cold deformation, for the strain rates less than 102 s^{-1} , we have seen that compression did not have much effect in the case of the materials studied (note 1). When the strain rate became greater than or equal to 102 s^{-1} , a correction was performed by using the FE simulations. The Johnson-Cook constitutive law was readjusted to find the same experimental curve as an adiabatic calculation. Then, a second isothermal calculation was made using the same material parameters. The newly corrected experimental curve was that obtained by the isothermal calculation (Figure 9). This procedure is rather heavy to implement. In addition, during tests on semi-hot and hot deformation domain, we did not observe a significant increase in temperature due to self-heating due to the feedback system. Disturbances (of a few degrees) were only observed. Thus the self-heating effects were neglected at hot deformation.
- For shear tests, the self-heating problem was considered in numerical simulation because the identification was performed based on a FE model (with an adiabatic calculation) due to the heterogeneity of the deformation.

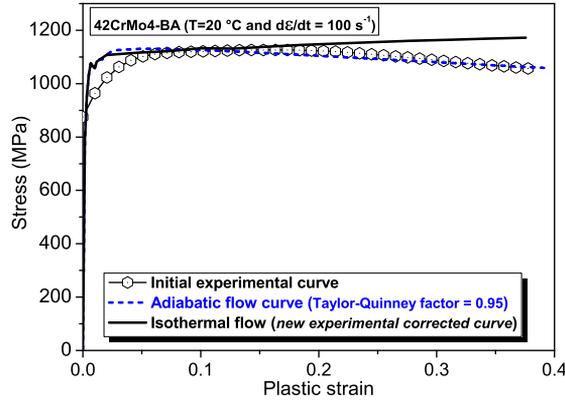


Figure 9: Comparison between the initial experimental curve, the curve obtained by FE simulation using an adiabatic flow and using an isothermal flow, for the 42CrMo4-BA steel.

The Johnson-Cook model. This model was identified using the complete experimental database. The Levenberg-Marquardt optimization algorithm was used. The five parameters were identified simultaneously.

Material	$A(MPa)$	$B(MPa)$	n	C	m
42CrMo4-FP	504	370	0.170	0.025	0.793
42CrMo4-BA	622	429	0.083	0.011	0.668
100Cr6-FP	594	359	0.059	0.003	0.718

Table 7: Parameters of the Johnson-Cook model identified using compression tests for the three steels investigated.

The Baker model. The number of parameters in this model was the same as the Johnson-Cook model. The identification of this model, for the three steels, was performed in the same way as the Johnson-Cook model. The identified parameters are given in Table 8.

The TANH Model. According to Calamaz et al. (2010), the advantage of this model is that the Johnson-Cook parameters, previously identified, can be used. Only five additional parameters, added to take softening into account, led to be identified. The value of T_{rec} (defined as the softening onset temperature) was determined directly from the experimental curves. Finally, using the same optimization algorithm, the model parameters are shown in Table 9.

Material	$A(MPa)$	n_0	C	$T_0(K)$	m
42CrMo4-FP	838	0.067	0.025	1039	4.005
42CrMo4-BA	1097	0.032	0.012	968	3.487
100Cr6-FP	918	0.020	0.005	963	3.186

Table 8: Parameters of the Baker model identified for the three steels investigated.

Material	$A(MPa)$	$B(MPa)$	n	C	m	p	q	r	ε_0
42CrMo4-FP	504	370	0.170	0.025	0.793	7.900	5	1	0.547
42CrMo4-BA	622	429	0.083	0.011	0.668	0.924	5	1	2.915
100Cr6-FP	594	359	0.059	0.003	0.718	5.103	5	1	0.674

Table 9: Parameters of the TANH model identified using compression tests for the three steels investigated.

The Lurdos model. The identification was carried out in two steps. Firstly, the stress-strain curves for different thermo-mechanical conditions were approximated by the equation:

$$\sigma = \sigma_s + (\sigma_0 - \sigma_s + A\varepsilon^n) \exp(-r\varepsilon) \quad (5)$$

An example of this approximation, as well as the evolution of the parameters σ_s and σ_0 , for the 42CrMo4-FP steel, is given in Figure 10.

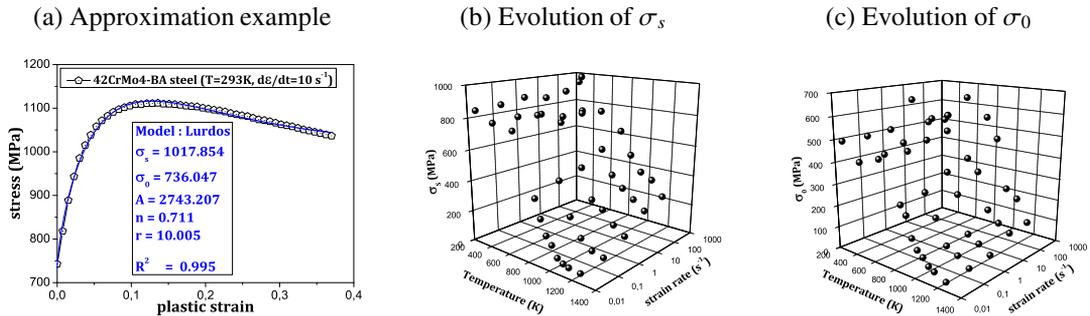


Figure 10: Example of the evolution of the parameters σ_s , σ_0 , r , A , and n as a function of T and $\dot{\varepsilon}$ - For the 42CrMo4-BA steel.

The second step was to use the values of σ_s , σ_0 , r , A , and n , determined for the different conditions in order to calculate the parameters of the equations $\beta_i = \beta_i^0 Z^{m_i}$. After determining the values of the Zener-Holloman parameters Z for the various tests, the variables σ_s , σ_0 , r , A and n were then plotted as a function of Z in a logarithmic diagram. An example is given in Figure 11 which shows the value of the parameters σ_s and σ_0 .

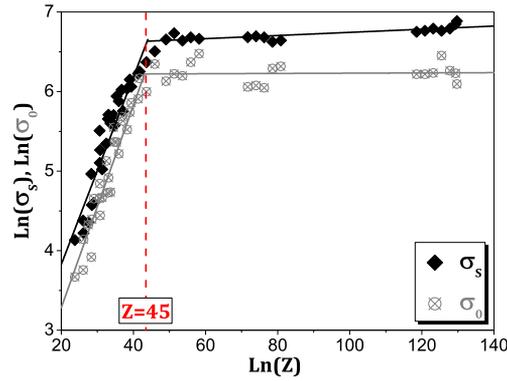


Figure 11: Experimental values and linear regression parameters (σ_s and σ_0) for the Lurdos model for the 42CrMo4-BA steel.

For the three steels studied, two different behaviours could be distinguished depending on the value of Z . Indeed, for the six variables (σ_s , σ_0 , r , A , and n), the linear regression changed slope at a value of $\ln(Z) \approx 45$ (Figure 11). This value marked the separation between the cold deformation, semi-hot deformation ($T/T_f \leq 0.5$) and hot deformation ($T/T_f \geq 0.5$) domains. The parameters of the Lurdos model are given for $\ln(Z) \leq 45$ and for $\ln(Z) > 45$. Table 10 summarises all the model parameters. The parameters σ_0^0 , σ_s^0 and A_0 are expressed in *MPa* and the other parameters are dimensionless.

The Proposed model. The simultaneous identification of nine parameters using the previously discussed optimization algorithm poses certain problems in terms of uniqueness of the solution, that is the solution depends on the initial parameter set. Hence, a simple identification procedure has been proposed. It consists of four steps: (1) identify the parameters m_1 and m_2 at the reference strain rate $\dot{\epsilon}_0$, (2) identify the function $C(T)$ and its parameters C_1 and C_2 , (3) determine the hardening parameters ($A = \sigma_0$, B and n) by assuming that during the reference test (i.e. at the

Material	σ_0^0	σ_s^0	r_0	A_0	n_0	m_0	m_s	m_r	m_A	m_n
For $\ln(Z) \leq 45$ or $T/T_f \geq 0.5$										
42CrMo4-FP	2.1	4.3	0.433	4.7	0.72	0.127	0.118	0.076	0.124	-0.007
42CrMo4-BA	1.3	6.3	4.461	30.1	0.573	0.142	0.105	0.007	0.054	-0.003
100Cr6-FP	2.3	5.5	3.399	8.5	0.537	0.121	0.105	0.015	0.094	-0.006
For $\ln(Z) > 45$ or $T/T_f \leq 0.5$										
42CrMo4-FP	499	696	2.986	462	0.329	0.001	0.002	0.014	0.001	0.001
42CrMo4-BA	534	781	17.5	215	0.242	0.002	0.002	-0.005	0.013	0.004
100Cr6-FP	498	656	2.344	219	0.133	0.002	0.003	0.012	0.014	0.009

Table 10: The parameters from the Lurdos and Montheillet model identified for the three steels investigated.

lowest strain rate and temperature) softening is negligible, (4) identify the softening parameters (p and ε_0). The resulting model parameters are given in Table 11.

Material	$A(MPa)$	$B(MPa)$	n	p	ε_0	C_1	C_2	m_1	m_2
42CrMo4-FP	505	449	0.186	7.900	0.547	$1.2 \cdot 10^{-3}$	7.42	1.61	1.18
42CrMo4-BA	622	584	0.175	0.924	2.915	$1.9 \cdot 10^{-4}$	10.1	1.73	1.26
100Cr6-FP	600	318	0.109	5.103	0.674	$8.7 \cdot 10^{-4}$	7.67	1.59	1.26

Table 11: Hardening and softening parameters identified for the three steels investigated.

At this point, it can be concluded that the increase in the number of model parameters increases the number of experimental tests necessary for the identification but not necessarily the difficulty of the identification. The latter is related to the degree of coupling between the various phenomena represented by the model and the mathematical formulation of the coupling. In the literature, the "cost" of a model is often related to the number of parameters which increases the number of tests needed for the identification. Other difficulties encountered during the identification, such as achieving the saturation state and the mathematical impossibility of decoupling certain physical phenomena (e.g. additive laws (Zerilli and Armstrong (1987))) should be considered in this

criterion.

4.2. Sensitivity of the model parameters to the identification method and the loading type

In the literature, it is common to find very different values for the parameters of a constitutive equation for the same material. In particular, several contradictions have been noted for the Johnson-Cook model (Tounsi et al. (2002); Huang and Liang (2003); Guo et al. (2006)). In order to study the influence of the numerical artifact causing this dispersion, the Johnson-Cook model was first identified using the compression tests with two different identification methods. The first method (method 1 in Table 12) is a sequential approach. It consisted of eliminating the terms σ_T and $\sigma_{\dot{\epsilon}}$ of equation 3 (by using tests at $T = T_0$ and $\dot{\epsilon} = \dot{\epsilon}_0$) to identify the hardening parameters (A , B and n). Then eliminate only the term $\sigma_{\dot{\epsilon}}$ by using tests at $\dot{\epsilon} = \dot{\epsilon}_0$) and identify the parameter m , and finally identify C . This method used only a part of the experimental database. The second method (method 2 in Table 12) used an optimization algorithm to determine the optimal parameters and minimize the difference between the models and experimental results.

Moreover, using an inverse approach based on a finite element model of the hat-shaped specimens coupled with an optimization algorithm (see 5-(b)), the five parameters of the Johnson-Cook constitutive equation that minimized the difference between calculated and experimental force for different thermomechanical conditions were determined (method 3 in Table 12). In this case, the loading type studied is shear, with the aim of analysing the effect of the loading type on the dispersion of the Johnson-Cook parameters.

The values of the Johnson-Cook parameters obtained by the three methods are different. The B and n parameters obtained from the shear tests are close to those obtained using the compression tests and the second identification method. This shows that the hardening is almost the same for both loadings. This result seems logical given that the two numerical methods of identification are almost identical. They were based on an optimization algorithm that minimized the difference between all the experimental curves and the results from an analytical model (for the compression tests) or via a finite element model (for the shear tests). These two methods used the complete experimental database in contrast to the first method where only few tests were used in the identification procedure.

Loading type / method	$A(MPa)$	$B(MPa)$	n	C	m
Compression (method 1)	622	584	0.175	0.009	0.468
Compression (method 2)	622	429	0.083	0.011	0.668
Shear (method 3)	668	478	0.09	0.012	0.89

Table 12: Values of the Johnson-Cook parameters identified using the compression tests results by the two direct identification methods and using the shear tests by the inverse method. For the 42CrMo4-BA steel.

However, the parameter m (controlling the temperature sensitivity) was higher for the shear loading condition. That is, the flow stress is more sensitive to the temperature when the material is loaded in shear. The choice of the loading mode for the identification of the constitutive law must therefore be consistent with the dominating loading mode of the forming process being studied. For example, the machining process can be characterized by highly localised deformation in a shear loading path. For this process the most appropriate identification method would be to use the shear test data. This is referred to as being physical (or experimental) simulation of the process (Hor et al. (2013)).

For both loading cases (compression and shear), the identified parameters were unique. In order to verify the uniqueness, the initial material parameters were varied usually and the program converged to the same parameters. However, the purpose of this paragraph was to show that when the size of the database, the type of loading or the identification method, were changed the set of parameters changes also, which explains the variability found in the literature for the same material. This was especially true when the model did not take into account all the observed phenomena (eg softening for the Johnson-Cook).

4.3. Comparison between the different constitutive models

Some predictions of the models identified in the previous paragraph, are presented below in the same stress-strain diagram for several thermomechanical conditions and the three domains of deformation (see Figure 12). In general, the difference between the model predictions and experimental results was very large for the Johnson-Cook, the Bäker and the TANH constitutive laws. The stress-strain curves from the proposed model and the Lurdos model were always closer

to the experimental curves for the three steels investigated. The proposed model performed better than the Lurdos model in the semi-hot domain. In the cold domain, the Lurdos model was more accurate, especially at high strain rate causing a decrease in strain hardening because it uses a formulation includes saturation whereas the proposed model is based on a power law. In the hot domain, the two models gave comparable results. The proposed model showed a tendency for the stress-strain curve to decrease in this domain although the experimental trend is the opposite.

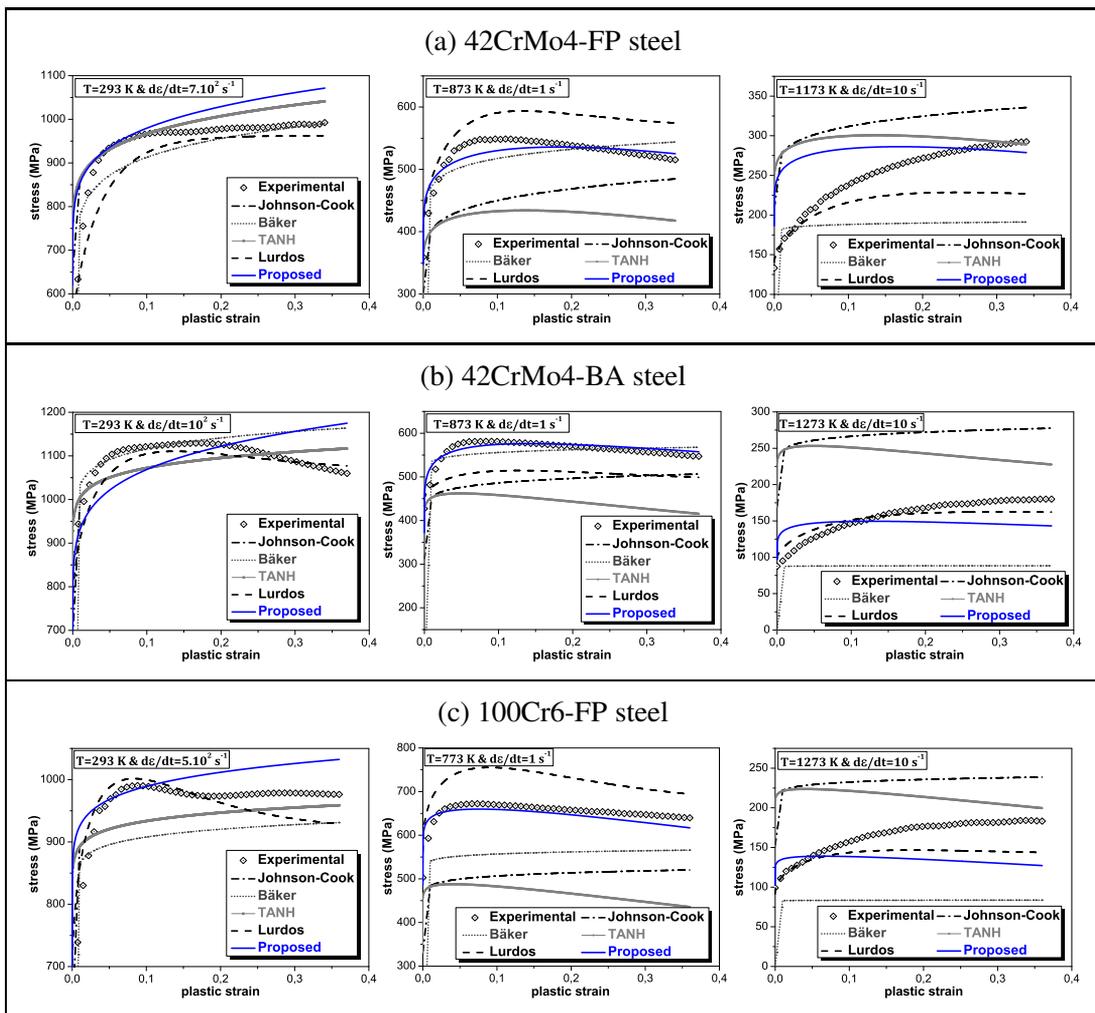


Figure 12: Comparison between the different constitutive models for different thermomechanical conditions for the three steels investigated: (a) 42CrMo4-FP, (b) 42CrMo4-BA and (c) 100Cr6-FP.

In order to quantify the difference between the predicted and experimental curves, a relative error was defined as the ratio of the area between the two curves to the area under the experimental

curve. For test i , the relative error E_i is given by:

$$E_i = \frac{\int_0^{\varepsilon_f} \left| (\sigma_{exper})_i - (\sigma_{model})_i \right| d\varepsilon}{\int_0^{\varepsilon_f} (\sigma_{exper})_i d\varepsilon} \quad (6)$$

The average relative error between the model predictions and the N experimental curves performed for a given material is therefore:

$$E_m = \frac{\sum_{i=1}^N E_i}{N} \quad (7)$$

Figure 13 represents the average relative error for the five constitutive models and the three steels investigated: the 42CrMo4-FP steel, the 42CrMo4-BA steel and the 100Cr6-FP steel. For these three materials, the error for the Johnson-Cook, Bäker and TANH models was between 10 % and 25 %. However, for the Lurdos model and the proposed model the error was less than 10 % for all three material. Despite the fact that the Lurdos model was identified in a piecewise manner (depending on the value of Z) which improved its capability, the proposed constitutive model was more accurate for the 42CrMo4-FP and 100Cr6-FP steels.

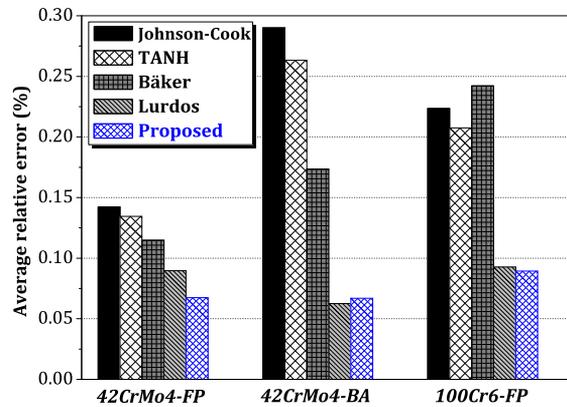


Figure 13: Average relative error between the experimental results and the predictions of the different constitutive laws identified for the three steels investigated. The proposed model is shown in blue.

Furthermore, for all of the models, the average relative errors were lower for the 42CrMo4-FP steel. They were greater for the 42CrMo4-BA and 100Cr6-FP steel. This is due to the number of thermomechanical conditions used for identification. For the 42CrMo4-FP steel, experimental

data for 45 different thermomechanical conditions were available. For the 100Cr6-FP steel only 33 different conditions were available and 25 conditions for the 42CrMo4-BA steel. Hence, it can be concluded that increasing the number of experimental tests used for the identification process improves the accuracy of the constitutive model. It was also shown that the choice of these conditions can improve the accuracy of the model in some deformation domains (i.e. cold, semi-hot or hot) (Hor (2011)). Indeed, if more experimental data is used in the identification which belongs to for example the semi-hot deformation domain, the identified model will be more accurate in that domain. This can be reproduced numerically by adding weight factors to several tests to make the model more accurate for these tests.

4.4. Summary

The stress-strain constitutive models investigated in this work belong to the family of phenomenological or empirical models. The Johnson-Cook model represents the group of uncoupled empirical models. These models typically have a low number of parameters. There is no coupling between the effects of strain hardening/softening, strain rate and temperature. Decoupling simplified the identification of these models and reduced the number of experimental tests needed for the identification. These models could not predict the experimentally observed behaviour for the three steels. Even though they do not represent reality, these models are widely used in the literature for the numerical simulation of forming processes.

Other investigations have shown the limitations of these models particularly for the simulation of machining operations (Calamaz et al. (2010)). Certain studies have shown that these models are unable to provide acceptable results, even qualitatively, for example to predict the chip shape during a machining operation. Following this work, several authors have introduced a coupling between the effects of strain hardening/softening, strain rate and temperature. This has been added in two different ways. First, by modifying the existing models. This group of models is represented in this work by the Bäker and TANH models which are an improvement compared to the Johnson-Cook law. Second, is to propose formulations based on simple strain hardening constitutive modes (such as the Voce law) in which the parameters are transformed into internal variables. These variables are then assumed to evolve with temperature and strain rate. This category of models is

represented here by the Lurdos model. Two types of coupling are used to take into account the influence of temperature and strain rate on the strain hardening.

Although, the number of parameters of Bäker model is not larger compared to Johnson-Cook model, the identification process is more complex. Bäker assumes that the appearance of softening is only a function of temperature. Its appearance diminishes the rate of strain hardening. This slope is always positive or zero. The results of this model are relatively inaccurate compared to the experimental observations. Small improvements can be seen compared to the Johnson-Cook law. The TANH model requires a higher number of experimental conditions for its identification. This model has four additional parameters compared to the Johnson-Cook law. It can reproduce reasonably well the competition between hardening and softening observed experimentally. However, the sensitivity of the flow stress to the rheological parameters is very different from the experimental results because it is identical to that proposed by Johnson-Cook. As for the Lurdos model, the identification procedure is simple but requires a vast experimental database. This is especially the case as the identification is done in a piecewise manner (depending on the value of Z), hence the need to have sufficient experimental data for each area. This model has ten parameters. It provides predictions that are close to the experimental results for the three steels.

The latter two models allow for strain softening via a competition mechanism between hardening and softening. The stress-strain curves can then be decreasing for certain thermomechanical conditions. In the Lurdos model, this softening is a function of temperature and strain rate, however in the TANH model only a temperature variation can cause the appearance of the softening phenomenon.

Finally, the proposed model gives more accurate predictions compared to those from the literature with a reasonable number of parameters (i.e. 9 parameters). It takes into account the phenomenon of softening accompanying plastic deformation at certain temperatures. The rate sensitivity depends on the temperature. Its identification is relatively easy but requires an extensive experimental database. It is also identifiable in a transitional regime and does not require the saturation stress to be achieved. It is therefore a good compromise between simplicity and accuracy, provided a sufficiently complete experimental database is available. The identification of the model for each domain, as per the Lurdos model, could further improve its accuracy even if this

would increase the number of parameters and require more experimental testing.

5. Validation of the different constitutive models for the simulation of orthogonal cutting

To highlight the ability of the different constitutive models to take into account phenomena such as softening, a numerical simulation of a machining operation was done and compared to experimental results. This results consists on the cutting forces, temperature and chip shape formed during the machining of the 42CrMo4-BA steel. Different cutting conditions were investigated experimentally in this planar cutting configuration and compared to finite element simulations using different stress-strain constitutive equations.

5.1. Experimental investigation of orthogonal cutting

Orthogonal cutting (Merchant (1945a), (1945b)) is a simplification of the machining problem, from both an experimental and a modelling point of view. Orthogonal cutting is a mechanical problem that occurs in a plane. If edge effects are neglected, this type of machining operation can be modelled in 2D. In this particular configuration, the three components of the cutting force usually encountered in three-dimensional cutting (F_a , F_r and F_t) are reduced to two components (F_r and F_t), and the cutting depth is equal to the feed rate. A complete experimental set-up has been put in place in order to characterize the cutting process from a mechanical, thermal and metallurgical point of view as a function of the cutting parameters (cutting speed and feed rate). For this study, tests were only performed on the 42CrMo4-BA steel. Four cutting conditions were chosen for investigation. The cutting force, the temperature fields and the chip shape were investigated.

Temperature field measurement by near-Infra-Red thermography

In order to measure the temperature in the cutting zone a near-infra-red camera was used (MSaoubi et al. (1998); Habak et al. (2007)). As shown in Figure 14, the thermographic images make it possible to determine the hottest point with respect to the cutting edge and to quantify the relative temperatures in the different shear zones, i.e. the primary shear zone observed on the cutting face (see the left most image in the column marked temperature field in Figure 14), and

the secondary and tertiary shear zones observed on the lateral face of the tool, (see the right image in the column marked temperature field in Figure 14). This thermal analysis showed precisely the evolution and position of the hot spot as a function of the cutting conditions. It was also observed that the temperature in the frictional zone with the tool increased when the feed rate or the cutting speed increased.

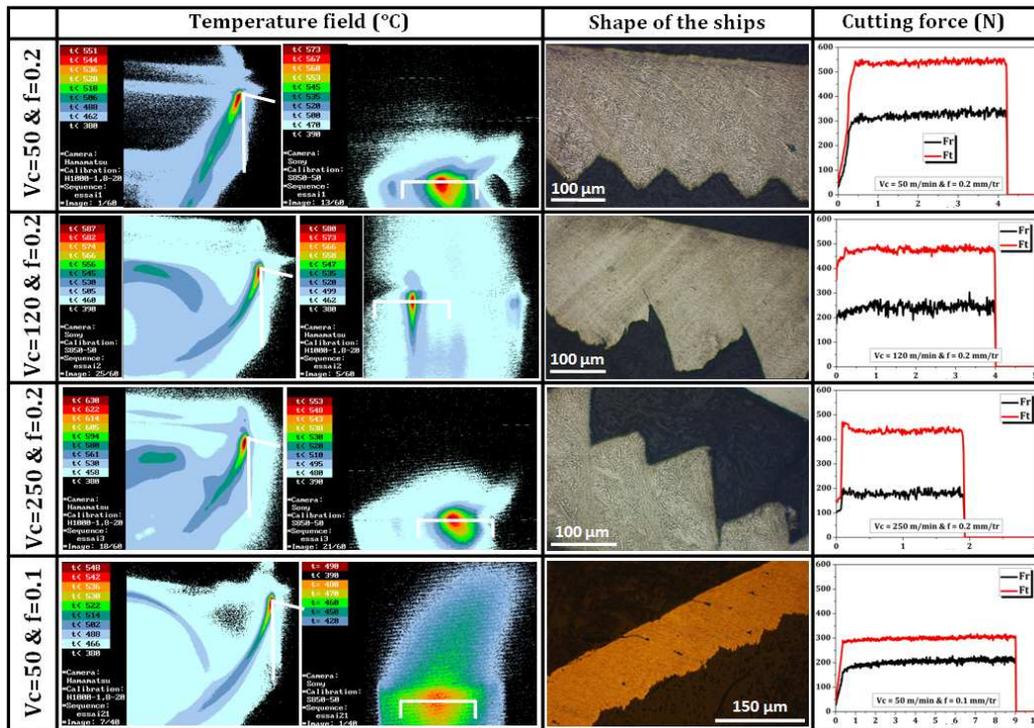


Figure 14: Evolution of the temperature field ($^{\circ}\text{C}$), the cutting force (N), and the chip shape as a function of the feed rate (mm/rev) and the cutting speed (m/min) for the machining of the 42CrMo4-BA steel.

Cutting forces

The measurement of the cutting forces was done using a Kistler table dynamometer. This type of equipment is widely used in machining because of its sensitivity, accuracy and above all its high stiffness which does not disturb the machining operation (Habak et al. (2007)). The different components (F_a , F_r and F_t) of the global cutting force were accessible. The evolution of these forces as a function of time for different cutting conditions is given in the third column of Figure

14. This force was very sensitive to the feed rate and the cutting speed. It increased when the feed rate increased and when the cutting speed decreased.

Analysis of the chip shape

Understanding the mechanisms involved in the machining process necessarily includes analysing the shape of the chips. After each test, the chips were collected and examined from a geometrical and metallurgical point of view using an optical microscope. The second column of Figure 14 shows the different chip morphologies obtained for different cutting conditions. It should be noted that the chip morphology depended strongly on the feed rate and slightly less on the cutting speed. For a feed rate of 0.1 mm/rev, the chips were observed to be continuous. However, when the feed rate was increased, the chips became increasingly discontinuous. These chips were formed by strain localization. This phenomenon, widely studied in machining, is called catastrophic shear or adiabatic shear band (Poulachon et al. (2002)). For a slow feed rate, the chip generating mechanism involved the formation of a crack followed by slip in the crack plane.

5.2. Numerical simulation of orthogonal cutting operations

In the literature, different simulation methods are used to model machining operations: finite element method, meshless methods, etc. For the finite element method, depending on the assumptions used, different approaches are possible. In this work, 2D plane strain models were used with strong thermo-mechanical coupling. This procedure takes into account frictional heating and plastic dissipation. The cutting model included a rigid tool moving at constant speed and a deformable metal part using a thermo-visco-plastic constitutive model. The separation of nodes was possible via the use of a failure law including a damage initiation criterion and a damage evolution law (Mabrouki et al. (2008)).

As part of this coupled problem, both mechanical and thermal contact properties must be defined. For the mechanical part, a frictional model with a Coulomb friction coefficient of 0.1 was used. For the thermal properties, a temperature increase due to friction was modelled with a conversion factor of 1.0 (i.e. 100 % of frictional energy converted to heat), and 50 % of this energy was transmitted to the surface of the tool and 50 % to the machined surface. No heat exchange

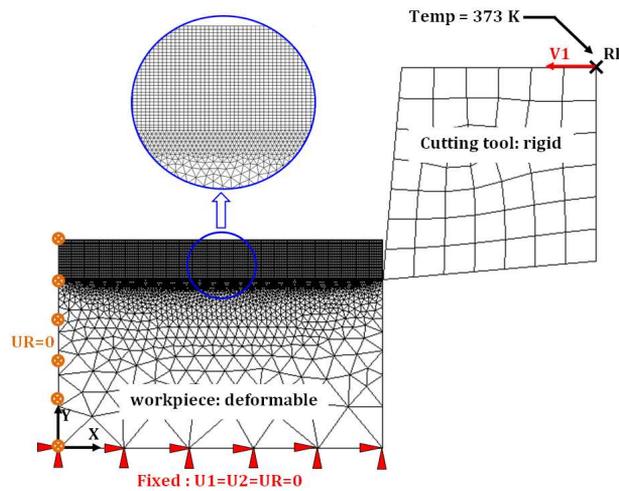


Figure 15: Finite element model of chip formation in orthogonal cutting.

via conduction between the tool and the workpiece was modelled. It was assumed that this heat transfer mode is probably low given the geometry of the tool.

Concerning the boundary conditions, the lower surface of the workpiece was completely fixed (Figure 15). The tool was considered to be rigid and isothermal, and moved with a constant speed in the negative X-direction (i.e. right to left horizontally). No movement in the Y-direction or rotation about the Z-axis was allowed. The initial temperature of the tool and workpiece was 293 K.

The tool and workpiece were meshed using thermo-mechanical elements having both mechanical and thermal degrees of freedom. For the workpiece, two types of elements were used. For the tool and the zone of chip formation, quadrilateral CPE4RT elements were used (i.e. 6336 elements in the workpiece and 47 for the tool). For the rest of the workpiece, triangular CPE3T elements were used (3811 elements). The mesh was refined in the zone of chip formation and development. The elements of the tool were stiffened by adding a type constraint "isothermal rigid body". In order to reduce the computation time, a coarse mesh was used for the tool (Figure 15). The commercial finite element code Abaqus Explicit 6.8 was used.

By observing the evolution of the temperature and the equivalent stress and strain, the process of chip formation, may be divided into three phases (Figure 16): the initiation phase (a), the

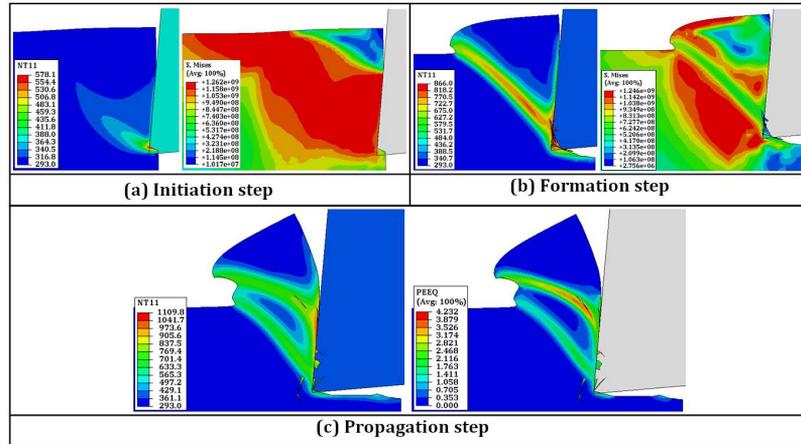


Figure 16: Description of the chip formation process in the numerical simulation.

formation phase (b) and the propagation phase (c).

5.3. Comparison between experimental and numerical results

The purpose of this section is to discuss the effect of the different stress-strain constitutive models that take into account softening, on the chip formation process, for different cutting conditions. To this end, simulations were conducted on one hand, using a model that does not take into account this phenomenon (i.e. the Johnson-Cook model (Johnson and Cook (1983))) and on the other hand, using models that accurately reproduced the phenomenon such as the TANH model (Calamaz et al. (2010)), the Lurdos model (Lurdos (2008)) and the proposed model. The same damage model was used for all these simulations: it was the Johnson-Cook failure criterion (Johnson and Cook (1985)).

Other than the constitutive laws and friction, the predicted chip shape depended on the damage initiation and evolution parameters. The Johnson-Cook damage initiation criterion was identified in Hor (2011). The damage evolution criterion was based on fracture mechanics and it depended on some numerical parameters such as mesh size (Mabrouki et al. (2008)). These parameters were adjusted by using the Johnson-Cook model to obtain chip shapes that are similar to the experimental observations for the same cutting conditions.

These parameters were then fixed for the other simulations, in which **only the constitutive**

model was changed. Figure 17 shows a comparison between the chip shape predicted using the different constitutive models. It is obvious that the chip shape which was closest to the experimentally observed shape was obtained using the Johnson-Cook model. This is because the principal parameters governing the chip formation process (i.e. the damage evolution parameters) were adjusted using this model. This comparison also shows that the most fragmented chip shapes were obtained using the TANH and the proposed constitutive relations. These two models are characterized by a more significant softening (i.e. the slope of the stress-strain curve becomes negative for certain conditions). The introduction of the softening phenomenon in the constitutive equation improved the chip fragmentation. These observations were also made by Calamaz et al. (2010). Hence, softening plays a role that is similar to damage.

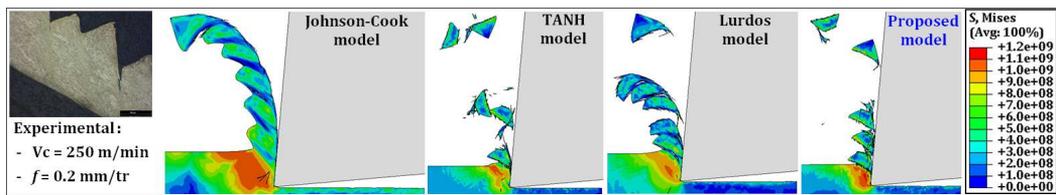


Figure 17: Comparison between the chip shapes predicted using the different constitutive equations and the experimental chip shape for the following cutting conditions $V_c = 250 \text{ m/min}$ and $f = 0.2 \text{ mm/rev}$.

Comparison with experimental observations was made by evaluating the cutting forces and temperatures in the primary and secondary shear zones (Figure 18). In terms of cutting forces, the four models studied in this section all overestimated the tangential force (F_t) and underestimated the radial force (F_r). This difference was due to inaccuracies in the frictional modelling. Nevertheless, a comparison between the tangential forces obtained in the various simulations was possible because the friction remained unchanged. This comparison shows, in particular, that the use of the Lurdos model or the proposed model improved the prediction of the cutting force. Indeed, the addition of softening in the TANH model reduced the cutting forces by 30 %. For the proposed model and the Lurdos model, this improvement was 35 and 37 % respectively.

Concerning the temperature (see Figure 18 on the right), it can be seen that the four constitutive models underestimated the temperature in the primary and secondary shear zones. This result

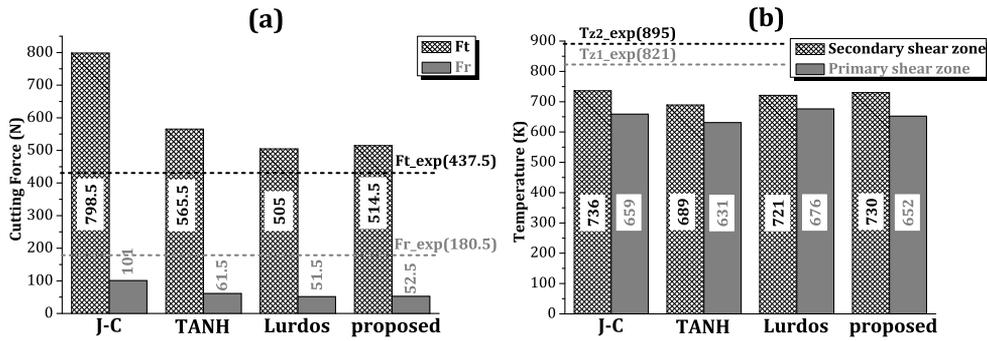


Figure 18: Comparison between the cutting forces (on the left) and the temperatures (on the right) obtained via numerical simulation using different constitutive models and corresponding experimental observations for the cutting conditions: $V_c = 250 \text{ m/min}$ and $f = 0.2 \text{ mm/rev}$.

explains the overestimation of the cutting forces previously discussed. In addition, the thermal predictions of the four models were very similar, the difference did not exceed 28°C in the primary shear zone and 47°C in the secondary shear zone. The fact that the predicted temperatures were lower than the experimentally observed values indicates that the identification of the stress-strain constitutive law based on compression tests could not properly account for certain effects such as the very localised self-heating observed in machining.

In order to improve the thermal predictions, the parameters of the Johnson-Cook equation were identified using shear test results obtained using "hat-shaped" specimens. The temperatures obtained in simulations using these parameters were similar or even higher than the experimental results. A comparison between the predicted temperatures obtained with two sets of parameters was undertaken. The first parameter set was obtained by direct identification (i.e. the second method) using the compression test results and the second parameter set was obtained by an inverse identification method using the shear tests data (Table 12). This comparison showed that, for numerical simulations of machining operations using the Johnson-Cook constitutive equation identified from the shear test data, the temperature increased from 659 to 764 K in the primary shear zone and 736 to 810 K in the secondary shear zone (Figure 19). These temperatures were close to the experimental observations (821 K in ZI and 895 K in ZII). This justifies the underestimation of the temperature by the different constitutive models identified using the compression

test data.

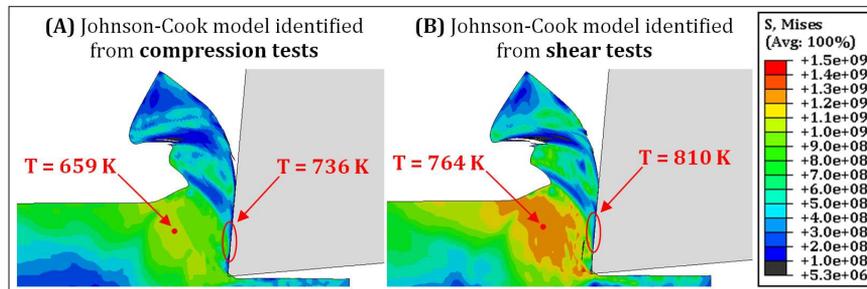


Figure 19: Comparison between the chip shapes, the equivalent stress field and the temperatures in ZI and ZII obtained numerically using the Johnson-Cook constitutive equation: (A) identified using compression test data (B) identified using shear test data, for the following cutting conditions: $V_c = 250 \text{ m/min}$ and $f = 0.2 \text{ mm/tr}$.

Finally, the use of empirical coupled constitutive models, such as the TANH, the Lurdos and the proposed model in the simulation of hat-shaped specimens and machining operations resulted in better chip fragmentation and improved the prediction of certain global measurements such as the cutting forces. The Lurdos model and the proposed model resulted in the best predictions of the tangential cutting forces and were 35 to 37 % lower than the predictions of the Johnson-Cook equation. However, the identification of these models using the compression test data underestimated some effects such as self-heating and localisation, resulting in lower temperatures than the experimentally determined ones. This was the case for all of the constitutive models investigated. The simulation performed using the Johnson-Cook constitutive model had a larger cutting force but not reached larger temperature. This is due to the chip segmentation process. Indeed, the models with softening allow a greater increase in temperature, but the cutting force remained low due this segmentation process.

6. Conclusion

A major problem with studies concerning rheology in machining and forging is the lack of experimental data, which limits those wishing to numerically model and analyse these operations to the use of very simple models (such as the Johnson Cook model) (Barge et al. (2005); Mabrouki et al. (2008)) whose identification requires only a few tests (Tounsi et al. (2002); Zemzemi et al.

(2009)). These constitutive models, once identified, have in general a very limited range of validity which does not cover the range of application of the machining process. In addition, very different parameters for the same materials can be found in the literature (Guo et al. (2006); Huang and Liang (2003)).

In this study, the limits of these constitutive models (especially the Johnson-Cook model) have been highlighted, through the use of a very rich experimental database. Also, the experimental and numerical reasons that can lead to large difference in their identification were discussed. More sophisticated empirical models were then investigated. These coupled models showed better accuracy but were sometimes very costly (in terms of the number of parameters), and their identification often required specific experimental or numerical manipulation. A coupled empirical model was developed and compared to these models. This comparison showed that it had a better cost-to-accuracy ratio. The comparison between these models and the experimental tests for three steels also showed that the increased number of experimental tests used for the identification improved the accuracy of their predictions. This was the case for all of the models investigated.

Finally, the application of these constitutive laws in the finite element modelling of machining operations showed that the inclusion of phenomena such as softening improved both qualitatively and quantitatively the accuracy of these simulations. It was also shown that using the parameters identified from experimental tests that simulated the process (shear tests in this case) increased the accuracy of these simulations and more closely simulated the phenomena experienced by the material during this process (i.e. localization and self-heating in this case).

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