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A Surface Enhanced Raman Spectroscopy study of aminothiophenol and
aminothiophenol-C₆₀ self-assembled monolayers: evolution of Raman modes with
experimental parameters

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ABSTRACT: P-aminothiophenol (PATP) is a well known molecule for the preparation of self-assembled monolayers on gold via its thiol functional group. After adsorption, it has been demonstrated that this molecule is anchored to gold through its thiol group, and standing nearly upright at the surface with the amino functional group on top. This molecule has been extensively studied by Surface Enhanced Raman Spectroscopy but its exact SERS spectrum remains unclear. Here we demonstrate that it can be strongly affected by at least two experimental parameters: laser power and layer density. Those features are discussed in terms of a dimerization of the PATP molecules. The free amino group affords the adsorption of other molecules, like C_{60} . In this case, a complex multilayer system is formed and the question of its precise characterisation remains a key point. In this article, we demonstrate that surface enhanced Raman spectroscopy combined with X-Ray photoelectron spectroscopy can bring very important information about the organisation of such a self-assembled multilayer on gold. In our study, the strong evolution of Raman modes after C_{60} adsorption suggests a change in the organisation of aminothiophenol molecules during C_{60} adsorption. These changes, also observed when the aminothiophenol layer is annealed in toluene, do not prevent the adsorption of C_{60} molecules.

Keywords: Self-assembled monolayers, surface enhanced Raman spectroscopy, aminothiophenol, fullerene C_{60} .

INTRODUCTION

The huge development of nanoelectronics leads to the elaboration of more and more complex architectures and to the subsequent difficult question of their physical and chemical characterisations. Among those architectures, self-assembled monolayers (SAM) on metallic nanostructures is an extremely promising approach, with potential applications in photovoltaics¹, molecular electronics², optics, and so on. Thiol functionalized molecules are known to form well-ordered, densely packed self-assembled monolayers on metals like gold or silver. The P-AminoThioPhenol (PATP) molecule appears as extremely promising due to its combination of electron-donating and electron-accepting groups, which is commonly designed as a “push-pull” molecule. In addition it is possible to elaborate “sandwich” PATP SAMs between gold substrate and another molecule using its respective two functional groups, i.e., thiol and amino groups. For instance, as for silver³ or gold⁴ colloids, fullerene molecules⁵ can be in principle grafted on top of the terminal amine moieties of PATP self-assembled on gold as a stacked SAM. Indeed, fullerene C₆₀ anchoring on amine groups has been observed and shown in a previous work⁶. This is particularly challenging for molecular electronics since electronic properties of such π -conjugated molecules enable their application in various molecular devices⁷. Among potential applications, such a bottom-up approach with a two step grafting is very attractive to build C₆₀-based capacitive memory cells with improved control of the density of active storage sites compared to the more developed top-down approach⁸. However, the experimental study of such complex nanostructures is extremely difficult. Indeed, during the formation of PATP SAMs some molecules may be linked to gold via amine groups although less favourable than linking via the thiol function⁹, and further grafting process of fullerenes may modify the organization of the pre-formed PATP SAM. Among all characterisation tools, Surface Enhanced Raman Spectroscopy (SERS) is extremely interesting as it can bring different information: metal-molecules interaction, charge transfer mechanism, electronic properties, molecule orientation, and so forth. This spectroscopy is based on the giant electromagnetism enhancement observed in the near field of metallic nanoparticle such as gold or silver¹⁰ through a plasmonic mechanism. It is generally admitted that an additional feature, usually called the chemical enhancement, also contributes to the overall SERS signal. This last contribution is subject of many debates in the SERS community and its exact role remains unclear. In the case of P-aminothiophenol (PATP) SAM on gold or silver nanostructures, SERS has been extensively used¹¹⁻¹⁴. Due to the strong affinity of the thiol group of this molecule for metals, SERS measurements can be extremely easily performed. All previous studies have shown that a charge transfer (CT) between this molecule and the metal occurs: in the SERS spectrum this CT gives rise to strong evolution in the Raman modes, but the exact origin of this mechanism remains unclear¹⁵. It must also be said that this molecule has been studied in complex metal-PATP-metal sandwich nanostructures¹² using SERS too. In the case of C₆₀ molecules, SERS studies are much less numerous¹⁶: this molecule has no functional group with a strong affinity for metals and it is thus almost impossible to prepare a clean bundle-free thin film of C₆₀ like a SAM directly on the metal. A possible technique is to use an intermediate molecule to connect C₆₀ to the metal¹⁷, as afore discussed with PATP. It must be noticed that most of C₆₀ SERS studies

have been performed using 1064 nm excitation, which is quite far from the plasmon absorption of gold or silver nanostructures. This might explain why the connecting molecule is usually not clearly observed in such analyses. Thus, both PATP and C₆₀ molecules have already been studied by SERS. Nevertheless, to our knowledge, few SERS studies have been performed using more complex SAM of stacked molecules. In this study we focused on the SERS characterisation of PATP and stacked PATP-C₆₀ SAM prepared on gold. We clearly observed that the SERS spectrum of PATP strongly depends on two experimental parameters: the density of the layer and laser power. We observed a progressive but irreversible evolution of the modes under high power laser illumination. A possible chemical change of the PATP molecule is discussed. For PATP-C₆₀ SAM the SERS signal of both PATP and C₆₀ was measured. A clear evolution of the SERS spectrum of PATP is observed, suggesting a change in the organisation of the PATP layer during C₆₀ grafting. The evolution of the Raman modes is compared with X-ray photoelectron spectroscopy (XPS) of the carbon and nitrogen core levels.

I. EXPERIMENTAL SECTION

Gold wire (99,99%) was obtained from Neyco. P-aminothiophenol (PATP, 97%), and fullerene C₆₀ (sublimated, 99.9%) were purchased from Sigma-Aldrich. Anhydrous toluene (synthesis grade) and absolute ethanol (99.99%) were obtained from Fisher Chemicals. Both molecules and solvents were used without further purification.

The nanostructured surfaces are prepared by thermally evaporating 15 nm of gold onto typical glass microscope slides in a vacuum chamber. Deposition is made at a rate of $\sim 1 \text{ \AA}\cdot\text{s}^{-1}$ under a vacuum pressure less than 5.10^{-6} mbar. Those nanostructured substrates are made of gold island films¹⁸ and have a low electrical conductivity. As a consequence for XPS measurements we used different gold substrates prepared with ~ 200 nm of gold Joule evaporated ($\sim 1 \text{ \AA}\cdot\text{s}^{-1}$) on a thin chromium layer deposited on a silicon substrate. Gold substrates are then soaked for 12 hours in 1mM alcoholic solution of PATP. Substrates are then rinsed with pure ethanol to remove physisorbed molecules. C₆₀ are grafted on the top of PATP monolayer under reflux conditions in toluene, at a temperature of 110°C and under nitrogen flow. Afterward, PATP+C₆₀ SAM are sonicated in pure toluene to remove physisorbed molecules. PATP molecules should have an almost standing up⁹ orientation with the thiol functional group linked to the gold surface, and C₆₀ are adsorbed via the amino group, forming a two layer self-assembled structure. Few SERS studies have been performed on such nanostructures. It must be noticed that for a part of our samples we have observed locally an aggregation of C₆₀ molecules (confirmed by Raman measurements) whose origin remains unclear. As a consequence we performed measurements out of those aggregation areas.

The XPS spectra were recorded in normal emission geometry with a 125 mm radius hemispherical analyzer (with a pass energy of 20 eV) and a Al K α X-ray source ($h\nu=1486.6\text{eV}$) in ultrahigh vacuum (UHV). The

XPS spectra are referenced to the spectroscopic Fermi level, and no corrections have been made for any surface charging. The area intensities were obtained following the spectral fitting and integration with Voigt-like functions.

SERS measurements were performed using an Horiba Jobin Yvon HR800 spectrometer using 632.8 nm excitation. The laser was focused on the sample via a 50 × objective. Typical laser power on the sample was 1.5 mW (D1 filter) or 150 μW (D2 filter). Those powers insure a rather high signal to noise ratio with a reasonable acquisition time.

II. RESULTS AND DISCUSSION

The SERS signal of PATP deposited on metallic nanostructures has already been extensively studied. It has now been clearly established that this spectrum is different from the bulk one, in opposition with what is commonly observed for dye molecules (methylene blue, rhodamine etc..). This feature has been interpreted as a sign of a charge transfer mechanism via an Herzberg Teller vibronic coupling^{13,19,20}. However the full interpretation of the observed modes has led to many discussions in the literature and the exact origin of the enhancement is still under debate. In table one is reported the vibrational assignment of the observed Raman of PATP according to different studies. It can be clearly seen that this attribution remains unclear. In addition, a recent theoretical study has concluded that the abnormal

Raman modes of ATP ^a	SERS modes of ATP (high concentration)	SERS modes of ATP (low concentration)	SERS modes of ATP after annealing in toluene	SERS modes of ATP and C ₆₀	Vibrational assignment ^{a,b}	Vibrational assignment ^c
1093 cm ⁻¹	1075 cm ⁻¹	1075 cm ⁻¹	1075 cm ⁻¹	1075 cm ⁻¹	a ₁ mode from ATP SC stretching + NH ₂ rocking	ATP SC stretching
1171 cm ⁻¹	1140 cm ⁻¹	1140 cm ⁻¹ (very weak or not observed)	Not observed	Not observed	b ₂ mode from ATP CH bending	ATP CH bending
1208 cm ⁻¹	1180 cm ⁻¹	1180 cm ⁻¹ (very weak or not observed)	Not observed	Not observed	CN stretching + CH rocking	ATP CH bending
Not observed	1390 cm ⁻¹	1390 cm ⁻¹ (very weak or not observed)	Not observed	Not observed	b ₂ mode from ATP CC stretching + NH ₂ rocking + CH rocking	ATP Phenyl ring deformation + NH ₂ rocking
Not observed	1435 cm ⁻¹	1435 cm ⁻¹ (very weak or not observed)	1435 cm ⁻¹	1435 cm ⁻¹	b ₂ mode from ATP CC stretching + NH ₂ rocking	ATP Phenyl ring deformation + CN stretching
Not observed	Not observed	Not observed	Not observed	1460 cm ⁻¹	Mode from C ₆₀	Mode from C ₆₀
1596 cm ⁻¹	1575 cm ⁻¹	1575 cm ⁻¹	1575 cm ⁻¹	1575 cm ⁻¹	a ₁ or b ₂ mode from ATP CC stretching + NH ₂ bending	ATP CC stretching in phenyl ring + NH ₂ bending

Table 1: list of most intense SERS and Raman modes of PATP SAM.

a is Ref [13], b Ref [19] and c Ref [20].

enhanced peaks observed in SERS did not originate from PATP molecule itself¹⁵. As a consequence the complete SERS analysis of PATP SAMs is still necessary. In figure 1 are shown SERS spectra for two different densities of PATP molecules deposited on nanostructured gold. The high density layer is obtained after 12h dipping in a 1mM PATP solution whereas the low density layer is obtained after a one minute dipping in a 10⁻⁵ mol/L solution. For both layers, two spectra were measured: the first one with a 1.5 mW excitation, the second one with a 150 μ W. According to classical Raman theory, the spectrum should not depend on the density of the PATP layer, except for its global intensity. Surprisingly it is not what is observed in figure 1. Some Raman modes (1140, 1185, 1390 and 1435 cm⁻¹, mainly *b*₂ modes according to table 1) are clearly not observed or have a much lower intensity when the layer density is low. In addition, the intensity of some modes also depends on laser power, in particular for the high density layer.

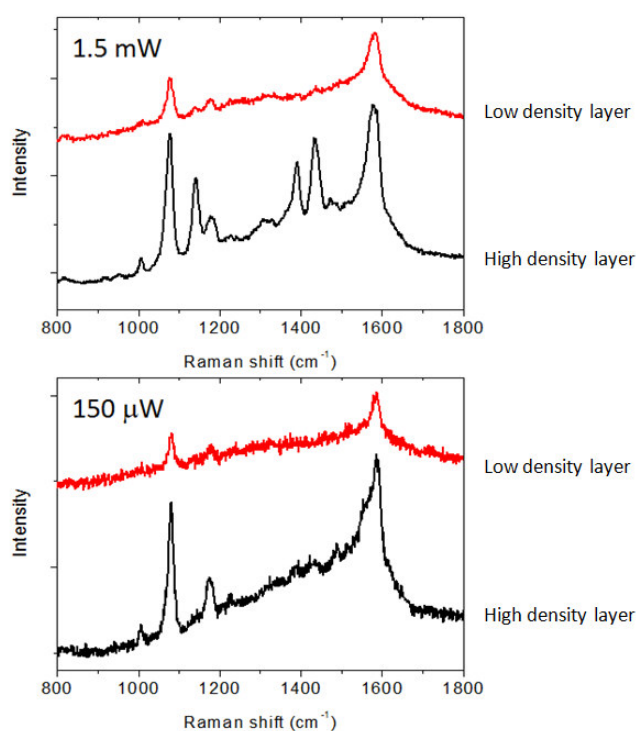


Figure 1: SERS spectra of low density layer (red) and high density layer (black) of PATP for two laser excitations: top 1.5 mW, bottom 150 μ W. Wavelength: 632.8 nm

It can immediately be remarked that Raman modes with pure *b*₂ symmetry are most affected by the density of the layer. According to spectra shown in figure 1 these modes are also most affected by laser power. As a first conclusion it appears clearly that the observed modes and their relative intensity in the SERS spectrum of PATP strongly depend on the experimental conditions. According to our measurements at least two factors can modify this spectrum: laser power and layer density. To gain better understanding on the influence of laser power we acquired short time spectra (1s) of the high density layer under a 1.5 mW laser illumination. Two spectra are shown in figure 2: one immediately after turning on the laser, and the other

after 30s of illumination. It appears clearly that an evolution of the SERS spectrum occurs during laser illumination. At the beginning the spectrum is similar to what is observed with low density layer or low power illumination. But after 30s the spectrum is different, and the intensity of b_2 modes (1140, 1185, 1390 and 1435 cm^{-1}) has considerably increased, giving rise to a spectrum similar to that in figure 1. As a consequence, it seems that SERS spectrum of PATP strongly depends on experimental conditions. It appears clearly that with low power laser illumination, b_2 modes are not enhanced, in accordance with recently published theoretical predictions¹⁵. The question is why the PATP SERS spectrum is progressively modified when laser power is increased. This evolution of PATP modes with laser illumination has already been reported by Kim et al.¹⁴ and interpreted as a consequence of threshold energy in the charge transfer enhancement. This feature should occur whatever the density of the PATP layer: as can be seen in figure 1, the low density layer has low b_2 modes even if laser power is high. As this threshold energy should reasonably not depend on the layer density this hypothesis cannot be considered for the observed evolution of Raman modes. Another possible explanation is a progressive degradation or modification of the PATP molecule induced by laser heating. Degradation is usually observed in Raman measurements when laser power density is too high. In this case, new Raman modes are present and the characteristic modes of the molecules disappear. As can be seen in figure 2 this is not what we observed: after a few seconds of laser illumination, the original modes of PATP are still present and their intensity has not decreased. This suggests that PATP is not totally damaged but it does not exclude a partial modification of the molecule as we clearly observed an evolution of the relative intensity between the different Raman modes of PATP. In previous studies the evolution of the b_2 modes was interpreted as a sign of the charge transfer, once again this mechanism should reasonably not depend on laser power and layer density. We can also assume that a change of the molecule orientation is induced by laser heating. This orientation change should modify the selection rules, lying to the observation of new modes. SERS has already been used to determine the orientation of C_{2v} molecules adsorbed on metals: phthalazine²¹, benzoic acid²², thiophenol²³...In the case of PATP, it has been demonstrated that this molecule is oriented with the C_2 axis almost normal to the surface. With this standing up orientation, electromagnetic enhancement theory cannot explain all the observed modes in the SERS spectrum of this molecule and chemical enhancement is introduced¹⁹. To explain the observed evolution of SERS modes of PATP we should suppose that under strong laser illumination this orientation changes, giving rise to new modes with different selection rules. This hypothesis appears as highly improbable. As a first conclusion, the most reasonable hypothesis for the evolution of the PATP modes with laser power and layer density is a chemical modification of the PATP molecules, excluding a total degradation of the molecule, as only modes reported in table 1 are observed. The dimerization of PATP molecules has already been reported on gold electrodes via an electrochemical mechanism²⁴. In addition, some recent studies²⁵ have suggested the production of dimercaptoazobenzene (DMAB) from PATP by catalytic coupling reaction on silver nanoparticles. This hypothesis has been advanced by Huang *et al.* to explain the evolution of PATP SERS modes²⁶, the supposed b_2 modes being in fact modes of DMAB. As can be seen in figures 1 and 2, we observed similar features. As a consequence, we suggest that the reported

evolution of SERS modes during high power laser illumination is induced by the transformation of PATP into a different molecule, certainly DMAB. This dimerization is not observed with low density layers as this reaction cannot occur if PATP molecules are too far one from the others.

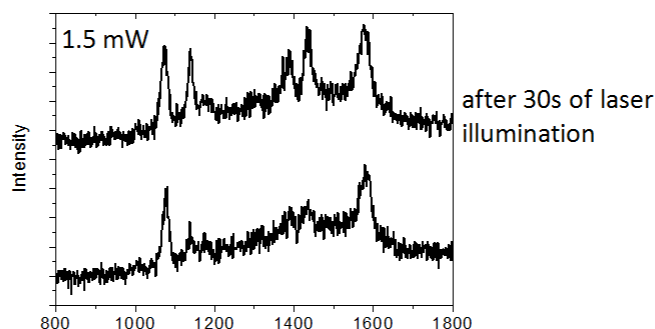


Figure 2: SERS spectra of a high density layer of PATP under a 1.5 mW laser illumination. The bottom spectrum is obtained immediately after the laser is turned on, the above spectrum is obtained 30s later. Wavelength: 632.8 nm Acquisition time: 1s

As explained in the introduction, PATP SAM can be interestingly used to graft fullerene molecules and obtain stacked SAM. The latter should then allow to obtain no more than one monolayer of C_{60} , with a surface density that could be varied depending on the targeted applications (e.g. isolated C_{60} bundles for memory cells). In figure 3 are shown the C1s and N1s core level XPS spectra for a PATP layer before and after C_{60} grafting. The effective grafting of the PATP layer on the gold surface is demonstrated by the well-resolved N1s peak corresponding to the amino group (binding energy of 399.2 eV, full width at half-maximum (FWHM) of 2.1eV). Remarkably, no binding energy peak usually attributed to N atoms involved in hydrogen bonds^{27,28} is observed, thus suggesting a low order in the organic layer or a good cleanliness of the film. The C1s spectrum is mainly composed of an intense peak at 284.3eV binding energy corresponding to C-C bonds and a minor peak around 286.3eV corresponding to carbon bonded to sulphur or nitrogen atoms.

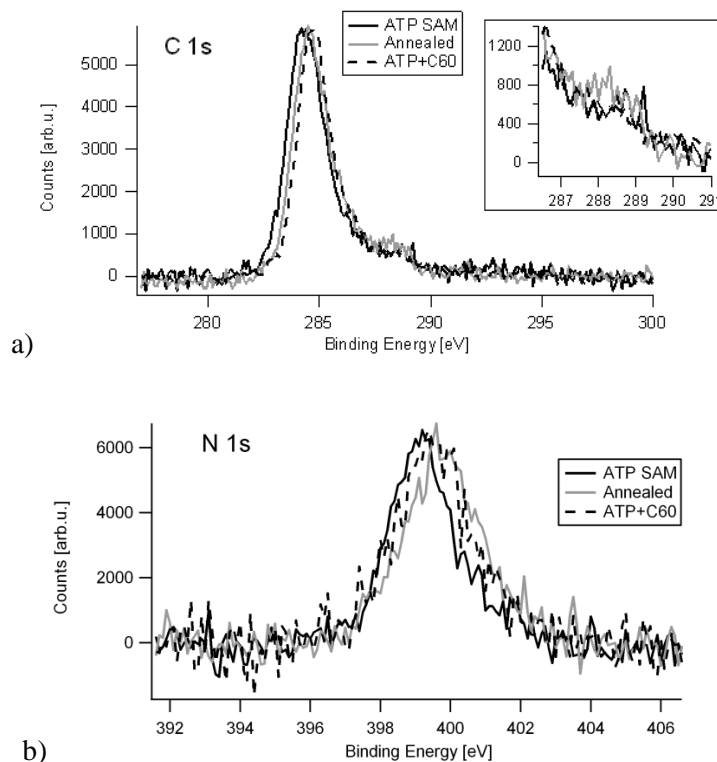


Figure 3: XPS spectra of the (a) C 1s and (b) N 1s core levels for different sample preparations (see text for details).

After C_{60} deposition XPS spectra are slightly modified. While the C1s spectrum remains nearly identical to that of PATP SAM, the nitrogen N1s peak is now centred at 399.5eV and slightly broader (FWHM 2.5eV). Such feature can be induced by the reaction of C_{60} molecules with the amino functional group or a change in the organisation of the PATP layer. No peak related to the formation of N bonded to C_{60} (binding energy above 400 eV²⁷) is observed, indicating a relatively low efficiency of the C_{60} grafting. In addition the quantitative analysis of the N/Au ratios indicates that the number of PATP molecules remained constant. A typical SERS spectrum after C_{60} adsorption is shown in figure 4 (black spectrum). Those spectra were obtained using 1.5 mW excitation (for lower laser power, the intensity is too low for a quantitative analysis). Comparing to spectra shown in figure 1, a global decrease of signal intensity is observed²⁹. Since the amount of PATP does not change after C_{60} adsorption, this decrease cannot be induced by a lower amount of PATP molecules. We cannot exclude that a part of gold nanoparticles are desorbed during C_{60} grafting decreasing the electromagnetic enhancement and the SERS signal. In final only 3 Raman modes of PATP are clearly observed: 1075, 1435 (weak intensity) and 1575 cm^{-1} . In addition a very broad band between 1400 and 1600 cm^{-1} is superposed to those Raman modes, probably associated to impurities. Moreover a new Raman mode, with a rather weak intensity, is observed at 1460 cm^{-1} . This mode is characteristic of C_{60} (as can be seen in

the green spectrum of figure 4) and clearly confirms that C_{60} molecules have been adsorbed on PATP molecules. The weak intensity of this mode is rather surprising since in previous studies from the literature the intensity of C_{60} modes excited at 1064 nm was extremely intense. But it must be noticed that this IR excitation is quite far from plasmonic absorption of gold or silver nanostructures. Further confirmation of the presence of C_{60} molecules was also brought by UV-Visible optical absorption (1-Transmission) measurements³⁰.

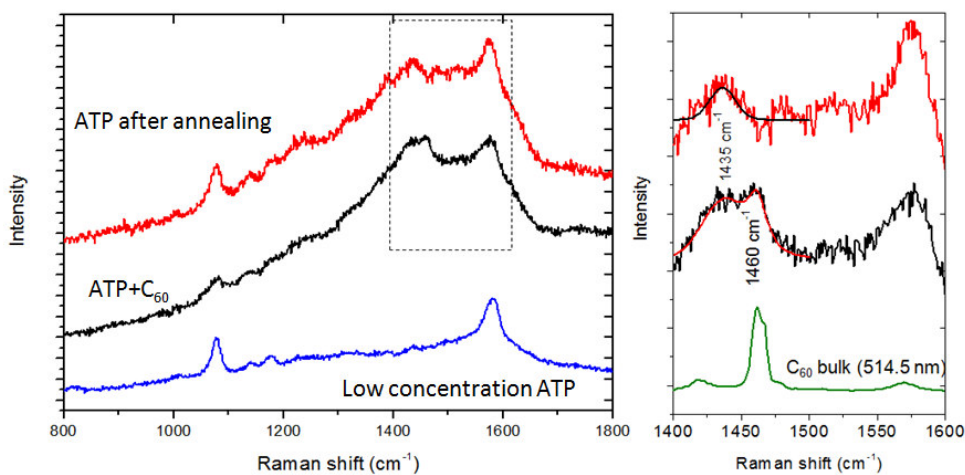


Figure 4: Left: SERS signal of PATP+ C_{60} SAM (black), PATP after annealing in toluene (red) and low concentration PATP SAM (10^{-5} mol/L, 1 minute dipping in blue). Right: an enlargement of black and red spectra. The characteristic peak of C_{60} at 1460 cm^{-1} can be clearly observed. The bulk spectrum of C_{60} obtained with 514.5 nm excitation is shown in green. For all other spectra, the wavelength excitation is 632.8 nm with a typical laser power of 1.5 mW.

The question is why the PATP SERS spectrum is modified after C_{60} adsorption. Two hypotheses can be put forward: those modifications are induced by C_{60} adsorption itself or by the heating process operated to promote the adsorption. This high temperature treatment in toluene can induce a desorption and a reorganisation of the PATP layer due to the labile nature of the thiol to gold bonding. However, XPS measurements suggest that no PATP molecule desorbed during C_{60} adsorption. To elucidate the effect of C_{60} grafting, we put an Au-PATP sample in a toluene solution heated at the temperature used in the grafting process of C_{60} ($\sim 110^\circ\text{C}$) but without dissolved C_{60} molecules. The XPS spectra of this sample can be seen in figure 3. The spectra of the nitrogen and carbon core levels are almost identical to the case of the C_{60} grafted sample. This important result suggests that the broadening of the nitrogen peak previously mentioned is certainly induced by a change in the organisation of the PATP layer due to the annealing process³¹. Concerning the C1s peak, one can note a small high binding energy shoulder at around 288 eV that is slightly decreased after C_{60} deposition. This peak may be induced by the C=O bond of an amide formed by reaction between amine and carbon dioxide³² and whose presence was detected in infrared spectra. The decrease of this signal after C_{60} deposition is consistent with the grafting of C_{60} that prevents from the

formation of amide groups. The area of the carbon peak for the C₆₀ grafted sample has increased by a factor of 1.7 as compared to the peak for the PATP SAM. This clearly indicates that C₆₀ molecules have actually been fixed on the PATP SAM. More surprisingly the intensity of this same carbon peak is also higher for the sample after heating in toluene (by a factor of 1.2 as compared to the peak for the PATP SAM). Certainly carbon impurities are deposited on the surface during heating in toluene. Comparing the intensity of the carbon peak between the C₆₀ grafted sample and the PATP sample heated in toluene, we can estimate the amount of C₆₀ molecules grafted if we consider that the same amount of impurities was deposited. The effective increase of the C/N ratio of 1.7-1.2=0.5 C/N corresponds to about 3 additional carbons for each PATP molecule, or one C₆₀ molecule deposited every 20 PATP molecules on the surface. This corresponds roughly to about half of the maximal C₆₀ density achievable on a ($\sqrt{3}\times\sqrt{3}$)R30° PATP SAM, even if this organisation has not been clearly confirmed for PATP molecules on gold³³. As a consequence the density of C₆₀ molecule is rather low, explaining why the C₆₀ Raman peak has a rather weak intensity. This result also explains why the peak corresponding to nitrogen bonded to C₆₀ was not observed in the XPS spectrum of figure 3: its absolute intensity is very low and the signal is furthermore screened by the presence of fullerene molecules. The SERS spectrum of a PATP SAM after heating in toluene is shown in figure 4 (red spectrum). Except for the C₆₀ peak at 1460 cm⁻¹, this spectrum is very similar to what we observed after C₆₀ adsorption. This indicates that the evolution of Raman modes of PATP is induced by the high temperature treatment in toluene. We also performed a similar high temperature treatment without any solvent in an incubator and the spectrum was almost identical to the standard SERS spectrum of PATP, despite a slight decrease of intensity³⁴. This indicates that it is not only temperature that plays a role in the evolution of PATP Raman modes, but also the presence of solvent. It must be noticed that both SERS spectra of C₆₀ grafted sample and of PATP SAM annealed in toluene are very similar to the SERS spectrum of the low density PATP SAM in figure 1. The only difference is the 1435 cm⁻¹ mode, whose intensity is very weak for the low density PATP SAM. XPS measurements indicates that carbon impurities are adsorbed during the heating in toluene (intensity increase of the carbon mode), which is confirmed by the presence of a broad band between 1400 and 1600 cm⁻¹ in the SERS spectrum³⁵.

Different hypotheses may explain the evolution of PATP modes in SERS spectra. The first one is a possible rearrangement of gold nanoparticles induced by high temperature treatment. It is well known that annealing of metallic nanoparticles can bring modifications in their organisation (size, distribution, etc.) and can have very important consequences for SERS measurements³⁶. With thiol molecules it is possible to create gold nanoparticle superlattices³⁷. In the special case of PATP, this molecule can induce the formation of “sandwich” structures, that is to say interlinked nanoparticles via a PATP molecule. This has been reported in several studies¹² and can have dramatic effects in the SERS spectrum of this molecule. Nevertheless, the formation of sandwich structures should prevent the adsorption of C₆₀ in so far as both functional groups of PATP molecules should be already linked to gold. And as explicitly mentioned above, XPS and SERS measurements confirm the presence of C₆₀ molecules. As a first conclusion we cannot totally exclude the possible formation of sandwich structures made of metallic nanoparticles interlinked by PATP molecules but

the presence of C_{60} molecules rather suggests that most of PATP and C_{60} molecules are bound otherwise C_{60} should have been washed off the surface during the rinsing step. To explain the observed modifications of SERS spectra of PATP after C_{60} adsorption or high temperature treatment, another hypothesis can be proposed: a strong reorganisation of the PATP SAM. It is known that the stability under high temperature treatment of thiol SAM on gold (nanostructured or not) is limited³⁸. Disorder, reconstruction and desorption can occur even at relatively moderate temperatures, typically around 100°C. In addition, this stability strongly depends on the chemical structure of the adsorbed molecules³⁹. The careful analysis of SERS spectra can bring information about the evolution of those SAM. Spectra reported in figure 4 are intermediate between the bulk and the SERS spectrum of PATP. They clearly show strong similarity with the low density modes, or the modes obtained with a low laser density. Once again only modes attributed to b_2 symmetry are affected. In previous SERS studies, the enhancement of those modes was attributed to a charge transfer mechanism based on a Herzberg-Teller contribution^{19,40}, that is to say a chemical enhancement. More recently, Uetsuki *et al.*⁴¹ have intercalated a thin spacer layer between the metal and the PATP molecules and have observed a strong decrease of the b_2 modes intensity at 1141, 1390 and 1437 cm^{-1} . With such sample, any possibility of chemical bonding between Au atoms and PATP molecules is completely eliminated and chemical enhancement is not expected, explaining the intensity decrease of those modes. In our experiments, we observed total disappearance of 1140 and 1390 cm^{-1} modes and a clear decrease of the 1437 cm^{-1} mode, whose intensity is now clearly lower than the 1575 cm^{-1} mode. This is very similar to what is reported by Uetsuki *et al.* This could suggest that chemical enhancement of PATP is also affected by the annealing in toluene and C_{60} adsorption. But as previously explained it is necessary to be very cautious with the chemical enhancement hypothesis due to the strong evolution of PATP modes with experimental parameters and the suggested dimerization of PATP molecules. In any case the strong similarity between the SERS spectrum of PATP annealed in toluene and the low density PATP layer suggests that this dimerisation do not occurs for those samples. In addition, the broadening and the shift of the N1s peak observed in XPS indicate that the layer is probably less organised. As a consequence we propose the following mechanism during C_{60} deposition: the annealing in toluene induces a desorption of PATP molecules, decreasing the layer density, and an introduction of carbon impurities. The layer organisation is also strongly decreased during this step. However no dimerization occurs and C_{60} molecules can be grafted on top of PATP molecules. It appears that this grafting is not very efficient as typically one C_{60} is grafted every 20 PATP molecules. This proposed mechanism is in accordance with all the reported experimental features: the evolution of the SERS modes (decrease of b_2 modes attributed to DMAB, appearance of broad modes between 1400 cm^{-1} and 1600 cm^{-1}) and XPS spectra (broadening and shift of the peaks).

CONCLUSIONS

We studied the SERS signal from PATP and PATP- C_{60} SAMs. We observed an evolution of the Raman signal of PATP depending on two parameters: the layer density and the laser power. It appears clearly that b_2 modes of PATP are not enhanced if one or both of these parameters is low, in accordance with theoretical

predictions. In opposition, if these parameters are both high, the intensity of b_2 modes is enhanced. The evolution of these modes is associated to a dimerisation of PATP molecules into DMAB. For PATP- C_{60} SAM we also observed strong evolution of the SERS spectra: b_2 modes have their intensity considerably decreased after C_{60} adsorption. Comparing SERS and XPS, we proposed a grafting reaction mechanism during C_{60} deposition in toluene. Impurities and disorganisation of the PATP layer are introduced, however those features do not prevent the adsorption of C_{60} molecules in a relatively low density, an issue that could be beneficial to build isolated discrete molecular storage sites for molecular memory cells. Our work demonstrates the capability of SERS for studying complex self-assembled layers on metallic nanostructures. Consequently, SERS could be envisaged as a challenging and powerful technique that could help better understanding electronic properties and processes involved within molecular devices.

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