

A new model for surface potential decay of corona-charged polymers

George Chen

To cite this version:

George Chen. A new model for surface potential decay of corona-charged polymers. Journal of Physics D: Applied Physics, 2010, 43 (5), pp.55405. $10.1088/0022-3727/43/5/055405$. hal-00569754

HAL Id: hal-00569754 <https://hal.science/hal-00569754v1>

Submitted on 25 Feb 2011

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers.

L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

A New Model for Surface Potential Decay of Corona-charged Polymers

George Chen

School of Electronics and Computer Science University Of Southampton, SO16 1BJ, United Kingdom

Email: gc@ecs.soton.ac.uk

Abstract. Surface potential measurement provides a useful tool to gauge electrical properties of materials. It has been observed that the potential of sample with an initial high surface potential decays faster than that with an initial lower surface potential, known as cross-over phenomenon. The phenomenon has been found a few decades ago and various theories and models have been proposed. The common feature of the existing models is based on single charge carrier injection from corona charged surface. With our recent space charge measurement results on corona charged sample, double injection from both electrodes has been verified. Based on this new fact, a new model based on bipolar charge injection is proposed and initial numerical simulation reveals that the surface potential cross-over phenomenon can occur under bipolar charge injection.

1. Introduction

Over the years, considerable interest has been shown in the surface potential decay of corona-charged polymeric materials. In its most basic form, a surface potential experiment involves charging the surface of an insulating material to a voltage V_0 . The charging is most conveniently achieved by a corona device which can deposit charges of positive or negative polarity onto the surface of the material. Following the charging process, the time dependence of the surface potential V(t) can be monitored via an electrostatic voltmeter. The measurement of the potential decay has been proven to be a simple and useful technique for characterizing insulating materials and the charging method. For example, this method enables a convenient determination of charge carrier mobility and trap parameters. One of the well-known effects in the observation of surface potential decay is the crossover phenomenon $[1]$, i.e. initially the surface potential of a sample charged to a high-potential decays more rapidly than one charged to a lower potential. Most of the theories addressed the time evolution of the surface potential in terms of surface conduction [2], charge injection [3–5], and polarization [6]. The recent literature on potential decay measurement is dominated by the hypothesis of the injection into the bulk of the charge deposited on the surface accompanied by a slow polarization processes within the bulk under the influence of the deposited charge. Clearly, a thorough understanding of the detailed physical process of kinetics of surface potential decay is required.

With the recent progress in space charge measurement techniques, it is possible to observe charge evolution within the bulk of the corona charged insulting material. Our papers [7, 8] demonstrated that bipolar charge injection has taken place in low density polyethylene sample during and after corona charging using the pulsed electroacoustic technique (PEA). In the light of this new experimental evidence, we propose a new model which incorporates charge injection from top and bottom surfaces and field dependent mobility.

2. Surface potential decay

A typical corona charging experimental setup is shown in Figure 1. The surface potential of the sample will be the same as the grid voltage providing that the potential difference between the needle and grid is sufficient high so the electric field around the needle is greater than the breakdown strength of air.

Figure 1 Schematic diagram of corona charging setup.

Once the corona charging stops, the surface potential starts to decay. The decay takes various forms depending on the corona charging conditions such as the grid voltage, charging time and sample thickness. Figure 2 shows a typical potential decay observed in low density polyethylene (LDPE) polymeric materials when the grid voltage is changed. The charging time was 2 minutes for all the samples. Negative surface potential was obtained, however, for easy graphing, the absolute surface potential is used in this paper.

Figure 2 Potential decay for 50µm LDPE sample for different corona voltages.

It has been observed that the potential of sample with initial high surface potential decays faster than that with an initial lower surface potential, known as cross-over phenomenon [1]. It can be seen that the cross-over point moves towards shorter time with increases in charging voltage. Although the

phenomenon has been found a few decades ago and many efforts have been spent to investigate the mechanisms, a satisfactory explanation, however, has not been found so far.

It has also been noticed that charging time also has a significant influence on the surface potential decay when the grid voltage is kept constant. Figure 3 illustrates a typical surface potential decay showing the influence of the charging time.

It can be seen the surface potential of the sample charging longer time has a fast rate of decay compared with that of sample with a shorter charging time.

Figure 3 Surface charge decay in 50µm LDPE sample corona charged at 4 kV

3. The existing models

The surface potential decay and the cross-over phenomenon have been observed in various polymers [9, 10] and several models and theories have been proposed. Batra *et al* [11 – 13] assumed the fielddependent mobilities and negligible penetration depth of the surface charge but neglecting the effect of partial instantaneous injection and trapping. Wintle, in his papers [14-16], developed theories that include field-dependent mobilities of various forms as well as trapping but none of them explained the crossover effect. Wintle also made an assumption the depth of penetration of the initial charge is fieldindependent. Batra proved that Wintle's theories also cannot count for the crossover [17]. Later, Sonnonstine and Perlman came out with two distinct theories in their work [10]. The first one is the modification of Batra's theory to include both instantaneous partial injection and field-dependent mobility. The second theory assumes time-dependent detrapping of charge carriers at the corona charged surface. Although the approximate forms of the decay curves were theoretically predicted in the paper, they were not satisfactory as the charging conditions were neglected.

Baum *et al* [18-19] were able to demonstrate that the crossover phenomenon depended upon the sign and the duration of the corona charging process. In their papers, it was shown that the crossover phenomenon did not occur for positive corona voltage, in apparent, contradicts the findings of Ieda *et al* [1, 20]. They suggested that the excited molecules and photons generated in the corona discharge process caused the charge which originally deposited in deep surface to be injected into the bulk where it becomes mobile thus increasing the decay rate of the surface potential especially for high initial potential. They were also able to prove that crossover is charging-time dependent by demonstrating that no crossover appeared for charging time less than ~25 *ms*. The proposal of Baum and co-workers was then supported by the work of Kao *et al* [21] who revealed a deep surface trap distribution centered at 95 ºC and a shallow surface bulk distribution centered at 55 ºC on negatively-coronacharged low-density polyethylene (LDPE) by using the thermally-stimulated discharge technique. Using the same technique, they were able to show that the trapped charge in the shallow surface can be released by exciting molecules from the corona discharge. In 1980, Toomer and Lewis [22] introduced the existence of both deep and shallow surface traps in the sample. They also showed that negative charges penetrate more readily into the bulk and the bulk traps exists for both sign of carriers.

In addition to various assumptions which were not evident, one of the common features in the models proposed so far is that all the models are based on single charge carrier injection. Our new experimental evidence has shown this is not always the case especially when the cross-over is concerned. Bipolar charge injection has been verified by the measurement of space charge in corona charged sample as shown in the section below. This new finding challenges the existing surface potential decay models which were developed based on a single charge carrier injection.

4. New evidence

In last two decades, significant progresses have been made in developing techniques to map space charge in solid dielectrics. The pulsed electroacoustic technique (PEA) is one of the methods widely used. Author is one of the fewer people use the technique to measures space charge in corona charged polymeric materials [7]. It has been confirmed without any doubt that bipolar charge injection taking place during charging. For a typical sample, the presence of bulk space charge in the PEA results is observed through charge profile or peak between the two electrode charge peaks. However, for a thin sample bulk charge profile or peak may overlap with the two electrode charge peaks, consequently, the presence of the bulk charge can be identified by the changes in the two electrode charge peaks i.e. broadening or narrowing depending on the polarity of the bulk charge.

Figure 4 shows the space charge measurement in a -2 kV corona charged LDPE film. To minimize the disturbance to the deposited charge, top and bottom surface of the samples were protected by an extra layer 50µm LDPE film. There are four distinctive charge peaks present across the sample from the left to right. The first and the fourth peaks are known as induced charge peaks on the PEA electrodes. The presence of the other two peaks is due to the existence of charge in the sample. The second peak corresponds to the bottom surface of the 50µm corona-charged film while the third peak to the top surface. The third negative peak is expected as the sample was exposed to negative corona. The first space charge measurement was performed 2 minutes after the sample has completed corona charging as the sample has to be transferred to the PEA setup. The bulk charge is not evident when a short charging time is adopted. However, the bulk charge is clearly seen when the charging time is extended to 10 minutes as shown in Figure 4 (b). It can also be seen that a small amount of negative charge presents across the sample. This may be served as the evidence that charge injection from the top has taken place.

Figure 4 also shows that the charge decay rate is different for different charging times. Charge decay is more rapid when a longer charging time is adopted. As the amount of charge in the sample is closely related to the surface potential measured, this observation is consistent with the results shown in Figure 3.

(a) $t_c = 2$ minutes

(b) $t_c = 10$ minutes

Figure 4 Space charge distribution in 50µm sample corona charged at -2 kV for (a) 2 minutes and (b) 10 minutes.

When the corona charging voltage increases, charge injection becomes more obvious as shown in Figure 5 where a -8 kV grid voltage was applied. Comparing with Fig. 4 where a lower corona charging voltage was used, the second and third peaks (corresponding to the bottom and top surfaces of the sample) in the PEA measurement become broader towards the bulk of the sample, indicating the presence of bulk charge. These bulk charges can only be injected into the sample from the corresponding surface. It is clear that both positive and negative charges are present in the bulk of the sample. When the charging time is extended to 10 minutes significant positive charge injection can be observed as shown in Figure 5 (b). It is worth noting that the scale for charge density reflecting rapid decay of charge within first 2 minutes. Charge decay rate increases significantly compared with the sample charged at 2 kV. This result is in agreement with the cross-over phenomenon observed in Figure 2. The sample arrangement for the PEA measurement differs from that in surface potential measurement. The two added layers and induced charge on the electrodes may affect the charge movement, therefore, the charge decay rate. Nevertheless, our early work on comparison between the surface potential obtained through either the PEA measurement or direct surface potential measurement indicating little difference, suggesting the added two layers have insignificant influence on the charge transport.

(a) $t_c = 2$ minutes

(b) t_c = 10 minutes

Figure 5 Space charge distribution in 50µm sample corona charged at -8 kV for (a) 2 minutes and (b) 10 minutes.

To further verify the occurrence of bipolar charge injection in the corona charged sample, two layers of 50µm films were corona charged for 2 minutes with a grid voltage of -8 kV. The interface of LDPE is known to be able to trap both positive and negative charges due to surface states. The injected positive and negative charges can be captured by the interface during transport process. To observe these trapped charges, space charge measurements were performed on the two layered corona charged sample first and then on both top and bottom layer separately. The results are shown in Figure 6, where two 50µm LDPE films were attached to both sides of each layer. From these distributions it is evident that negative charge is present at the top surface of the bottom layer (bottom layer curve or pink curve) and positive charge at the bottom surface of the top layer (top layer curve or green curve). The amount of negative charge is greater than that of positive charge, so overall it shows negative charge peak (2layer curve or blue curve). The revelation of both positive and negative charges at the interface between the top and bottom layer is a clear indication of bipolar charge injection. In this instance, positive charge tends to move upwards therefore is trapped at the bottom surface of the top layer. Similarly, negative charge is trapped at the top surface of the bottom layer.

Figure 6 Space charge distribution at different layer of corona charged sample (- 8kV and 2 min).

5. New model and preliminary simulation result

Recently, the bipolar charge injection model has been used extensively to simulate charge transport, trapping and recombination under dc voltage. The simulation results show a good match with experimental results. Considering the new evidence of bipolar charge injection during the corona charging, we reckon the model can be applied to explain the surface potential decay with some modifications. The new modified model takes bipolar injection into consideration. Additionally, a tunneling process has been proposed to account for charge injection from the top surface while the conventional Schottky injection is used for the bottom surface.

Based on the space charge measurement from the PEA, the initial charge distribution immediately after the corona charging in a sample can be illustrated in Figure 7.

Figure 7 Schematic diagram of surface charge and space charge distribution immediately after corona charging.

Here V₀ is the grid voltage. $\sigma_1(t_0)$, $\rho(x, t_0)$ and $\sigma_2(t_0)$ are surface charge on the top surface, space charge in the sample and induced charge on the metal electrode respectively. The initial values will depend on the grid voltage and charging time. They all change with time leading to a change in surface potential $V(t)$.

The principle on all space charge models lies in the description of the charge conduction and electrical transport mechanism across the material. Alison-Hill [23] model aims to effectively describe the bipolar transport and space charge phenomena in solid dielectrics under high dc stress. The bipolar transport is being described as a conduction process governed by an effective mobility. This feature distinguishes the model from the others. In effect, charge carriers are injected from the electrodes, electrons from the cathode and holes from the anode. Injection occurs based on the *Schottky* mechanism [24] whereby overcoming a potential barrier W at the interfaces.

$$
J_S = AT^2 \exp\left(-\frac{W - \sqrt{\frac{e^3 E}{4\pi \epsilon_r \epsilon_0}}}{kT}\right) \tag{1}
$$

where A is a constant related to the material, T the temperature, k the Boltzmann constant, e the electron charge, E the electric field at the interface, ε , the relative permittivity of the material and ε_0 the permittivity of vacuum.

It can be seen that the potential barrier is lowered down because of the application of the electric field. This will lead to an increase in the injected current.

After penetrating into the material, the carriers, under the influence of the applied field, will drift across the material characterized by an effective mobility. Throughout its motion, some carriers are trapped in the localized states i.e. deep trap centres and therefore, the total amount of charges moving across reduces. However, no extraction barrier is introduced in the model and on the other hand, they are prone to recombine with their opposite species (electrons with holes).

Due to the fact that oppositely charged species, electrons and holes are being considered in the numerical computation, charge trapping and mutual annihilation or recombination between these species were introduced into the model. The model now contains four *species* of charge particles namely mobile electrons (e_u) and holes (h_u) , and immobile electrons (e_v) and holes (h_v) (at trap sites) as shown in Figure 8.

Figure 8 Schematic representation of the conduction and trapping model. S_i are recombination coefficients, n_{eu} , n_{et} , n_{ht} , n_{hu} are mobile and trapped electron and hole densities. B_e and B_h are electron and hole trapping coefficients.

Charge transportation in solid dielectrics is essentially governed by a set of basic equations. They describes the behaviour of charge carriers in the system through a time and space dependent total flux $j(x,t)$ and by neglecting diffusion [23]:

Transport equation:
$$
j_c(x,t) = \mu n(x,t) E(x,t)
$$
 (2)

Continuity equation:
\n
$$
\frac{\partial n(x,t)}{\partial t} + \frac{\partial j(x,t)}{\partial x} = s
$$
\n(3)
\n*Poisson's equation:*
\n
$$
\frac{\partial E(x,t)}{\partial x} = \frac{\rho(x,t)}{\varepsilon}
$$
\n(4)

where μ is the mobility of carriers, n the density of mobile species, E the electric field, i the current density, x the spatial coordinate, t the time, s the source term, ε the dielectric permittivity and ρ the net charge density.

The above model has been widely used for simulation of space charge buildup and dynamics and achieved a satisfactory match when the parameters are appropriately selected [25]. It has also been used to model transient current and anomalous discharge current successfully [26].

For a corona charged sample, once charges are injected into the bulk, the rest description of charge trapping, detrapping, recombination and transport can be readily applied using the existing bipolar model.

It has been well accepted that the mobility of charge carriers is a function of the electric field. In fact several models have attempted to use field dependent mobility to account for the cross-over phenomenon without success. The field-dependent mobility can also be easily implemented in the proposed model. A power law proposed by Wintle [16] has been given as below:

$$
\mu = cE^{n-1} \qquad n \ge 1 \tag{5}
$$

where c is a constant and n is a fixed component. Both of them are material dependent.

The charge transport in the bulk of the sample is determined by the electric field. The electric field in the sample at any time consists of contributions from the three components, i.e. space charge $p(x, t)$, surface charge density at the top $\sigma_1(t)$ and the induced surface charge density at the bottom electrode $\sigma_2(t)$. Let us assume the field components are represented by $E_\rho(t)$, $E_{\sigma_1}(t)$ and $E_{\sigma_2}(t)$ respectively. The surface potential across the sample can be calculated by integrating the total electric field:

$$
V_0(t) = \int_0^d [E_\rho(t) + E_{\sigma_1}(t) + E_{\sigma_2}(t)]dx
$$
 (6)

In addition, the total charge in the system at any time must in balance, i.e.

$$
\sigma_1(t)S + \sigma_2(t)S + \int_0^d \rho(x, t)Sdx = 0 \tag{7}
$$

where S is the surface area where charges are present.

Clearly, $\sigma_1(t)$, $\sigma_2(t)$ and $\rho(x, t)$ are not independent quantities. Based on the modified model, it is possible to calculate $\rho(x, t)$ during the corona charging until a predefined charging time t=t₀. The quantities $V_0(t_0)$ and $\rho(x, t_0)$ are the initial condition for surface potential decay. This allows one to determine $\sigma_1(t_0)$ and $\sigma_2(t_0)$ using the above two equations.

Once these initial four quantities are determined, one can calculate new space charge distribution $p(x,$ $t_0+\Delta t$) based on the proposed model and the two surface density $\sigma_1(t_0+\Delta t)$ and $\sigma_2(t_0+\Delta t)$ using the Fowler – Nordheim (FN) tunnelling and Schottky injection respectively. In addition, as the system is an open circuit, the injected charge must satisfy the following condition

Maxwell's equation for the total current:
$$
J(t) = j_c(x, t) + \varepsilon \frac{\partial E(x, t)}{\partial t} = 0
$$
 (8)

The surface potential at time $t=t_0+\Delta t$ can finally be computed using equation (6). Repeating the above process the relationship between the surface potential and time can be calculated.

Our initial simulation work shows it is possible to achieve surface potential cross-over providing the initial surface potential and space charge distribution are set right as shown in Figure 9. In this simulation the parameters are similar to those used in [25]. The only difference here is that the mobility equation (5) has been utilised in the present study instead of a constant mobility. Additionally, the phenomenon presented in Figure 3 is readily explained by th he modified model as a long charging time will result in more charge injection and less surface charge densities for both top and bottom. The charge distribution favours the charge recombination and therefore a rapid surface potential decay. In the present simulation, Schottky injections have been used for both surfaces. Schottky injection typically describes the injection from metal to semiconductor or insulator, so the FN tunnelling may be more appropriate for corona charged surface. More simulation is under way where FN tunnelling will be investigated and it is important to point out that the present model allows one to extract much more material information.

Figure 9 Initial simulation showing cross-over phenomenon in surface potential decay (Power-Law mobility: μ =4.5×10⁻¹⁶Eⁿ⁻¹, n=1.15).

6. Conclusions

Based on the new experimental evidence from the space charge measurements on corona charged low density polyethylene film, a modified bipolar charge injection model has been proposed to account for surface charge decay. The new model can take care of field dependent carrier mobility and is readily explain the surface potential cross-over phenomenon and surface potential decay with different charging time.

More simulation based on the proposed model will be carried out and the results will be compared with the experiments.

References

[1] M. Ieda, G. Sawa and I. Shinohara, A decay process of surface electric charges across polyethylene film, J. Appl. Phys. 6 (1967), pp. 793–794.

[2] P. Jestin, R. Coelho and M. Goldman, Surface potential decay on epoxy resin, Inst. Phys. Conf. Ser. No. 85, sec 2 (1987), pp. 193–196.

[3] E.A. Baum, T.J. Lewis and R. Toomer, The lateral motion of charge on thin films of polyethylene terephthalate, J. Phys. D: Appl. Phys. 1 (1978), pp. 963–977.

[4] F. Mady, R. Renoud, J.P. Ganachaud, Modelling of potential decay experiment. Influence of the microscopic charge transport processes on the macroscopic observable phenomena. 3rd Conf. French Society of Electrostatics, Toulouse, 2002.

[5] H.V. Berlepsch, Interpretation of surface potential kinetics in HDPE by a trapping model, J. Phys. D: Appl. Phys. 18 (1985), pp. 1155–1170.

[6] P. Molinié, Surface potential decay on corona-charged epoxy samples due to polarization processes, J. Phys. D: Appl. Phys. 28 (1995), pp. 1601–1610.

[7] G. Chen, Z. Xu and L. W. Zhang, 2007, 'Measurement of the surface potential decay of coronacharged polymer films using the pulsed electroacoustic method', Meas. Sci. Technol. 18, p. 1453- 1458.

[8] Zhiqiang Xu, Linwen Zhang and George Chen. 2007, 'Decay of electric charge on corona charged polyethylene', J. Phys. D: Appl. Phys. 40 (2007). p. 7085- 7089.

[9] S. Salhi, A. Bellal, Z. Ziari, A. Kahlouche and Y. Segui, Mesure and analysis of potential decay in polypropylene films after negative corona charge deposition, J. Electrostat. 57 (2003), pp. 169–181.

[10] T.J. Sonnonstine and M.M. Perlman, Surface potential decay in insulators with field dependent mobility and injection efficiency, J. Appl. Phys. 41 (1975), pp. 3975–3981.

[11] I. P Batra, K. Keiji Kanazawa and H. Seki. 1970, 'Discharge characteristics on photoconducting insulators', J. Appl. Phys. 41(8), July 1970. p. 3416-3422.

[12] I. P Batra, K. Keiji Kanazawa, B. H. Schechtman and H. Seki. 1971, 'Charge carrier dynamics following pulsed photoinjection', J. Appl. Phys. 42(3), 1 March 1971. p. 1124-1130.

[13] H. Seki and I. P. Batra. 1971, 'Photocurrents due to pulse illumination in the presence of trapping. II', J. Appl. Phys. 42(6), May 1971. p. 2407-2420.

[14] H. J. Wintle. 1970, 'Decay of static electrification by conduction processes in polyethylene', J. Appl. Phys. 41(10), Sept 1970. p. 4004-4007.

[15] H. J. Wintle. 1971, 'Decay of Surface Electric Charge in Insulators', Japan. J. Appl. Phys. 10, 1971. p. 659-660.

[16] H. J. Wintle. 1972, 'Surface-Charge Decay in Insulators with Non-constant Mobility and with Deep Trapping', J. Appl. Phys. 43(7), July 1972. p. 2927- 2930.

[17] I. P. Batra and K. Keiji Kanazawa. 1972, 'On the `cross-over' effect in surface potential decay', Japan. J. Appl. Phys. 10, 1971. p. 267-268.

[18] E A Baum, T J Lewis and R Toomer. 1977, 'Decay of electrical charge on polyethylene films', J. Phys. D: Appl. Phys., Vol. 10, 1977a. p. 487-497.

[19] E A Baum, T J Lewis and R Toomer. 1977, 'Further observations on the decay of surface potential of corona charged polyethylene films', J. Phys. D: Appl. Phys., Vol. 10, 1977b. p. 2525- 2531.

[20] T Mizutani, A Oota and M Ieda, 1978, Surface Potential Decay of Corona-Charged High-Density Polyethylene, Jpn. J. Appl. Phys. 17 pp. 2157-2158

[21] K. J. Kao, S. S. Bamji and M. M. Perlman. 1979, 'Thermally stimulated discharge current study of surface charge release in polyethylene by corona generated excited molecules, and the crossover phenomenon', J. Appl. Phys. 50(12), Dec 1979. p. 8181-8185.

[22] R Toomer and T J Lewis. 1980, 'Charge trapping in corona-charged polyethylene films', J. Phys. D: Appl. Phys., 13 (1980). p. 1343-56.

[23] J. M. Alison and R. M. Hill, "A model for bipolar charge transport trapping and recombination in degassed crosslinked polyethylene", J. Phys. D: Appl. Phys. Vol. 27 pp. 1291-1299, 1994.

[24] K. C. Kao and W. Hwang, Electrical Transport in Solids, Pergamnon Press, Oxford, 1981.

[25] G. Chen and S. H. Loi, "Space Charge Modelling in Solid Dielectrics under High Electric Field Based on Double Charge Injection Model" MRS Fall Meeting, Boston USA, 2005.

[26] S.L. Roy, P. Segur, G. Teyssedre and C. Laurent, "Description of bipolar charge transport in polyethylene using a fluid model with a constant mobility: model prediction", *J. Phys. D: Appl. Phys.* **37**, pp. 298–305, 2004.